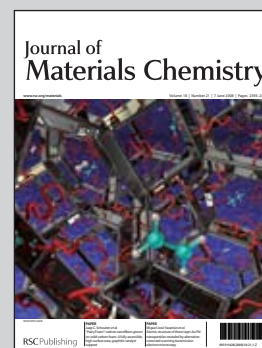


Showcasing work from Yacamán Group at The University of Texas at Austin, High Temperature Materials Laboratory at Oak Ridge National Laboratory, Advanced Materials Research Center, México (CIMAV) and Center for Innovation, Research & Development in Engineering & Technology (CIIDIT) at UANL, México.

Title: Atomic Structure of Three-Layer Au/Pd Nanoparticles Revealed by Aberration-Corrected Scanning Transmission Electron Microscopy.

Aberration-corrected imaging is employed under the scanning transmission electron microscopy (STEM) mode to precisely determine the chemical ordering of Au/Pd nanoparticles. These colloids exhibit a core-shell structure with a three-layer arrangement that surprisingly presents alloying events in the observed layers.

As featured in:



See Domingo Ferrer, Douglas A. Blom, Lawrence F. Allard, Sergio Mejía, Eduardo Pérez-Tijerina and Miguel José-Yacamán, *J. Mater. Chem.*, 2008, **18**, 2442

Atomic structure of three-layer Au/Pd nanoparticles revealed by aberration-corrected scanning transmission electron microscopy

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The study of nanomaterials can be greatly improved with the use of aberration-corrected transmission electron microscopy (TEM), which provides image resolutions at the level of 1 Å and lower. Sub-Ångström image resolution can yield a new level of understanding of the behavior of matter at the nanoscale. For example, bimetallic nanoparticles are extremely important in catalysis applications; the addition of a second metal in many cases produces much-improved catalysts. In this paper, we study the structure and morphology of Au/Pd bimetallic particles using primarily the high-angle annular dark-field (HAADF) imaging mode in an aberration-corrected STEM/TEM. It is well established that, when recorded under appropriate illumination and collection geometries, incoherent HAADF-STEM images are compositionally sensitive and provide direct information on atomic positions. We matched the experimental intensities of atomic columns with theoretical models of three-layer Au/Pd nanoparticles, in different orientations. Our findings indicate that the surface layer of the nanoparticle contains kinks, terraces and steps at the nanoscale. The effect of adding a second metal induces the formation of such defects, which might very likely promote the well-known improved catalytic activity of this system.

Introduction

Since the development of practical aberration correctors for electron microscopy,¹ a new possibility for understanding matter at the nanolevel has been opened. In particular, aberration-corrected scanning transmission electron microscopy (STEM) offers the possibility of studying atomic structures at a resolution of <1 Å,^{2–4} using primarily the high-angle annular dark-field (HAADF) imaging mode of an instrument with an aberration corrector fitted on the incident probe side. The incoherent nature of HAADF images, recorded with carefully controlled illumination and collection geometries, makes them free of dynamical interference effects (particularly for the case of crystalline particles of small dimensions) and therefore more straightforward to interpret than conventional high-resolution transmission electron microscopy (HRTEM) images.^{5–8}

The structure of nanoparticles is of great interest in many applications of nanotechnology such as catalysis, plasmonics, sensing and others. Using two metals, we can achieve a broader range of properties; for instance, a core-shell structure will

scatter light in a different way than will an alloy particle.⁹ We can in principle tune the properties by changing the way the metals combine. The case of Au/Pd is particularly interesting. In the bulk, these two metals are fully soluble at all compositions. However, the Au/Pd nanoparticle structure strongly depends on the preparation method. For instance, core-shell structures have been reported by several authors^{10–12} when chemical synthesis methods are used. On the other hand, Au/Pd particles grown by sputtering show an alloy structure with an icosahedral structure.¹³ It is also possible to produce a three-layer structure, as shown by Marzan *et al.*^{14,15} In this work we describe a novel three-layer structure in Au/Pd nanoparticles, and demonstrate that by combining aberration-corrected imaging with conventional techniques such as theoretical simulation and single particle diffraction, we can obtain significant new information about nanoscale colloids.

Experimental

Bimetallic nanoparticle synthesis

The colloidal dispersions of Au/Pd bimetallic clusters, stabilized by a polymer, were fabricated as previously described¹⁶ by successive alcoholic reduction of their constituent metal ions. Solutions of 0.1 M tetrachloroauric acid (HAuCl₄) and 0.05 M palladium chloride (PdCl₂) were prepared by dissolving the corresponding crystalline material in deionized water. The protective agent poly(*N*-vinyl-2-pyrrolidone) (PVP, 0.4 g, 3.6 mmol of monomeric units) and PdCl₂ solution (2 mL, 0.1 mmol) were then mixed in ethylene glycol. This solution was stirred and heated to reflux at 140 °C in air. The reduction of the Pd precursor was indicated by a sudden change in the color of the solution from yellow to

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brown. Following reduction of the Pd, the aqueous solution of HAuCl_4 (1 mL, 0.1 mmol) was added and subsequently reduced. This mixture was finally refluxed again at 140°C for 3 h in air. The total amount of both metals was always kept as 0.2 mmol in 50 mL of the mixed solution. Nanoparticles with different compositions of gold and palladium (namely AuPd, $\text{Au}_{0.75}\text{Pd}_{0.25}$, and $\text{Au}_{0.25}\text{Pd}_{0.75}$), and monometallic nanoparticles of Au and Pd, were fabricated by the above method.

Transmission electron microscopy characterization

Nanodiffraction patterns were acquired using a JEOL 2010F (200 kV) transmission electron microscope equipped with a field-emission gun and an ultra-high-resolution pole piece with STEM attachment. Aberration-corrected microscopy was performed with a JEOL 2200FS STEM with a hexapole corrector (CEOS GmbH) for the electron probe. The geometric aberrations of the probe-forming system were controlled to allow a beam convergence of 26.5 mrad half-angle to be selected. The HAADF images were acquired from 100 to 170 mrad scattering semi-angle, easily satisfying the requirement for the detector to eliminate contributions from unscattered or low-angle scattered beams. The theoretical models of the nanoparticles' structure were obtained using the Materials Studio software (Accelrys Inc.). In all cases the structure was relaxed to obtain the minimum energy configuration. The resultant atomic coordinates were fed into image simulation programs. We employed the SimulaTem¹⁷ software, modified for STEM, and the software WIN-HAADF (HREM Research Inc.).¹⁸

Results and discussion

The fabricated particles presented a three-layer structure, as has been previously reported.¹⁶ They exhibit a Pd-rich core, with the first shell highly enriched in Au, and a Pd-rich outer shell. A typical HAADF image of the three-layer Au/Pd particles is shown in Fig. 1a. The inset indicates a sketch of element-rich locations in the layers. By tuning the ratio of Au and Pd, the thickness of the layers systematically changed. A higher content of gold increased the diameter of the central Au-rich layer at the expense of the Pd-rich core and external thickness, and *vice versa*. The three-layer structure observed in these nanoparticles was found to be dependent on the diameter.¹⁶ It was observed that 25% of the nanoparticles had a diameter of $5.5 \pm 1.3\text{ nm}$, and only displayed two clear layers. The remaining 75% of the nanoparticles showed a size of $12.7 \pm 0.8\text{ nm}$, with three layers being evident.

The intensity profile through a typical Au/Pd nanoparticle is displayed in Fig. 1b, and depends on the atomic number and column thickness. The crystallographic structure of these particles was directly determined by means of electron microdiffraction techniques from isolated nanostructures. A beam of 2–5 nm was precisely positioned on a particle and the diffraction pattern was obtained. Fig. 1c shows the microdiffraction patterns from two particles, consistent with $\langle 100 \rangle$ and $\langle 110 \rangle$ orientations of an FCC structure. The lattice parameter was measured by using the standard non-linear form of the Bragg Law,¹⁹ and a value of 3.97 \AA was obtained.

An ultra-high-resolution HAADF image of Au/Pd nanoparticles is shown in Fig. 2a. The atomic columns are clearly

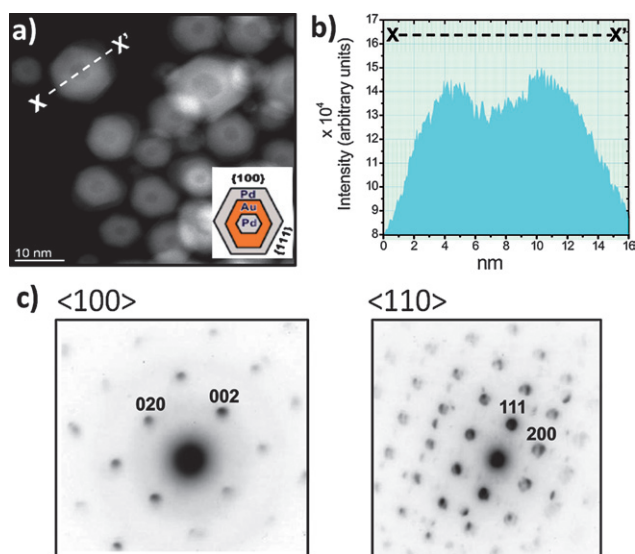


Fig. 1 (a) Image of Au/Pd nanoparticles with cuboctahedral shape. The contrast is due to a core-shell structure consisting of three layers as sketched in the inset. Projected facets of the cuboctahedral particles can be observed. (b) The experimental intensity profile of a typical HAADF-STEM image shows a lower magnitude on the central portion on the particle (indicated as $x-x'$) due to the fact that $Z_{\text{Pd}} < Z_{\text{Au}}$. (c) Nanodiffraction patterns of an individual Au/Pd nanoparticle showing a single-crystalline structure. Two orientations, $\langle 100 \rangle$ and $\langle 110 \rangle$, can be observed. Note that extra spots due to twins are not observed.

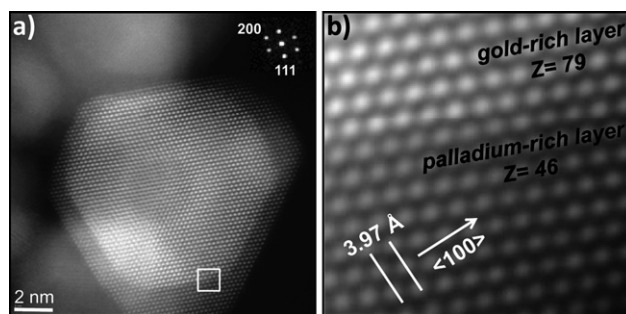


Fig. 2 Aberration-corrected STEM images. (a) Atomic-resolution HAADF-STEM image of a cuboctahedral Au/Pd nanoparticle. The contrast of the three distinct regions can be clearly seen. Bright dots represent atomic columns. The inset corresponds to the fast Fourier transform of the nanoparticle. (b) Enlarged image of a small part of the exterior layer of nanoparticle (indicated as a white square in (a)), exhibiting $\langle 110 \rangle$ crystal orientation.

resolved. A detailed image of the outer shell is shown in Fig. 2b. The inset corresponds to the digital diffractogram of this image, which indicates that the nanoparticle is in the $\langle 110 \rangle$ orientation. By measuring the distance between the individual atomic column peaks in the plot of intensity profile *versus* distance, an interplanar spacing was obtained, and the lattice parameter determined. The values obtained by the two methods were consistent, with a mean value of $3.97 \pm 0.08\text{ \AA}$ being obtained. This value is consistent with the one expected from Vegard's law,²⁰ and indicates that despite the observed three-layer structure, which exhibits different composition domains, the lattice parameter is not significantly altered. Moreover, no

extra diffraction spots, which would have suggested the presence of twinning or defects at the core–inner-shell and inner-shell–outer-shell interfaces, were found. This strongly suggests that coherent interfaces are produced in these layered nanoparticles.

Another interesting feature of the particles studied is their single-crystal structure. We can simulate the HAADF contrast observed in the nanoparticles of Figs. 1 and 2 by assuming that the whole shape of the particle corresponds to a cuboctahedron. It is well established that projected atomic columns in HAADF-STEM images appear in bright contrast. Since the images are incoherent, no interference effects are produced between the Bloch waves propagating through the crystal. The contrast strongly depends on the atomic number. To a very good approximation, the detailed image is a convolution

between a probe function and the projected atomic structure. We can approximate the image intensity as

$$I = I_p \otimes I(V_p) \quad (1)$$

where I_p is the probe effective intensity distribution and $I(V_p)$ is the contribution of the projected potential. Now if we assume that two kinds of atoms are present, in any column of atoms, we can write:

$$I(V_p) = F(NAu_i + NPd_j) \times I(Z) \quad (2)$$

where $F(NAu_i + NPd_j)$ corresponds to the atomic distribution of atoms in the particle, and $I(Z)$ is the intensity that originates from thermal diffuse scattering (TDS). In order to determine

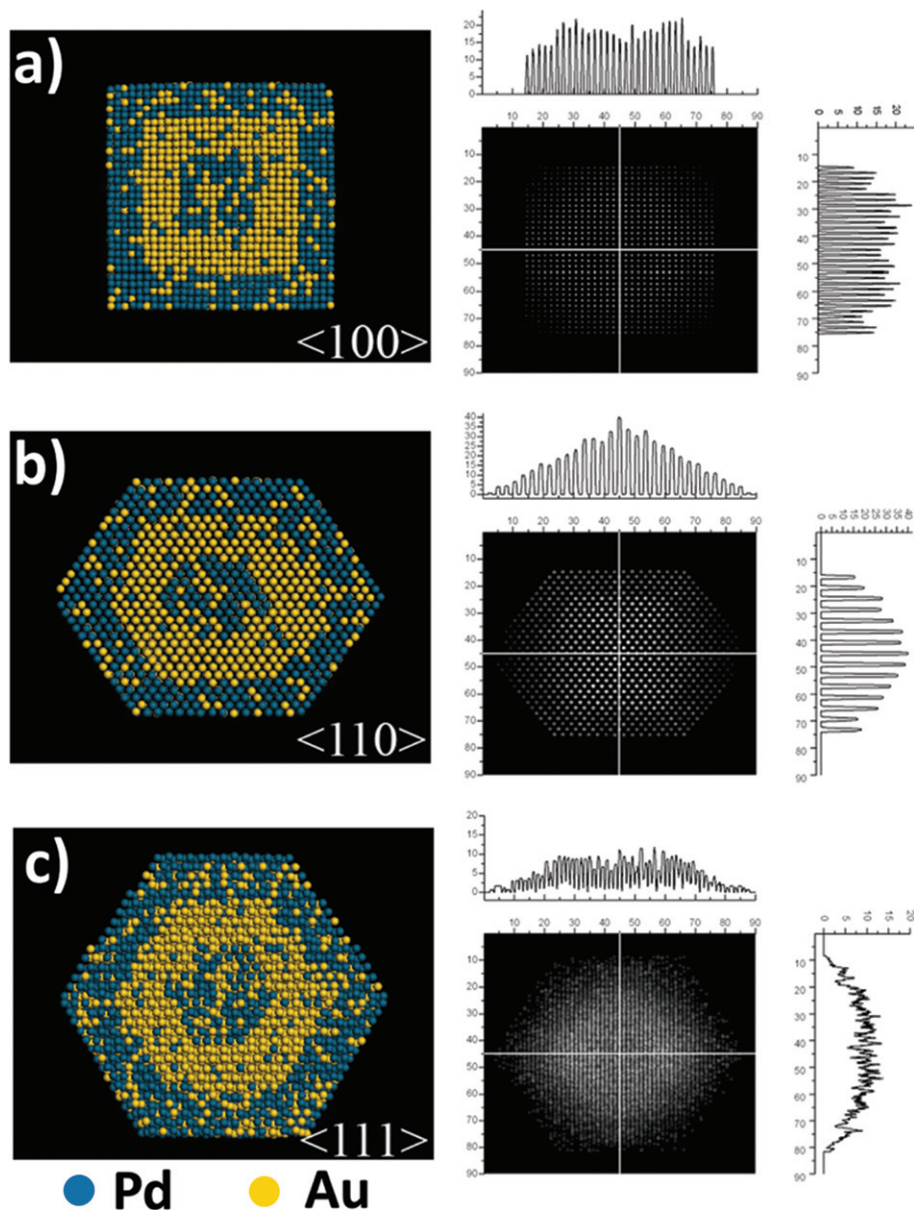


Fig. 3 Simulated HAADF-STEM of a cuboctahedral particle consisting of three layers: a Pd-rich inner-core with a thickness of five layers, an Au-rich intermediate-shell with a thickness of five layers, and a Pd-rich exterior-shell with a thickness of five layers. The cross-sectional view of Au/Pd models, their corresponding STEM images, and intensity profiles are shown for the (a) <100>, (b) <110>, and (c) <111> orientations, respectively.

$F(\text{NAu}_i + \text{NPd}_j)$, we assume a cuboctahedral-shaped particle formed of three layers: a central core containing 60% palladium, an intermediate shell containing 90% gold and an exterior shell containing 75% palladium. The $I(Z)$ was calculated assuming that the main contribution corresponded to TDS.²¹ The results for three different zone axes are shown in Fig. 3a–c. The displayed orientations, namely $\langle 100 \rangle$, $\langle 110 \rangle$ and $\langle 111 \rangle$, exhibit a good agreement between theoretical and experimental images, in all cases.

A measurement of the theoretical inter-atomic column spacing was also consistent with the experimental measurements. However, a systematic study of the aberration-corrected HAADF-STEM images shows deviations from the ideal shape. For instance, in several cases the central core is more rounded than in the simulated case. When we compared an experimental HAADF intensity profile in a picture without atomic resolution

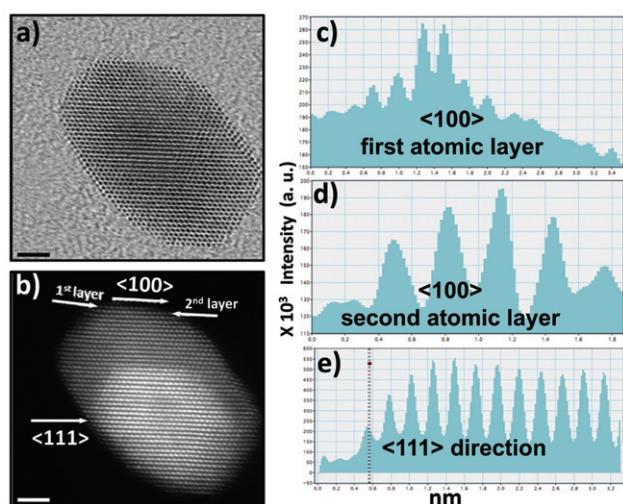


Fig. 4 Image of Au/Pd nanoparticle showing deviation from ideal cuboctahedral morphology and its corresponding intensity profile along different crystal orientations. By employing aberration-corrected HRTEM under (a) bright field and (b) HAADF-STEM modes, the atomic positions and chemical species can be clearly analyzed. The scale bar corresponds to 2 nm in both cases. The intensity profiles for (c) the first and (d) the second atomic layers of the bimetallic nanoparticle along the $\langle 100 \rangle$ surface and (e) the $\langle 111 \rangle$ face show a Gaussian distribution.

(Fig. 1a,b) we found a profile that agrees quite well with the theoretical profile for a $\langle 110 \rangle$ -oriented cuboctahedral shape. The profile is consistent with a three-layer particle with a Pd-rich central core. No twins or stacking faults are seen on the interface between layers, indicating that the interfaces between layers are fully coherent along the axis. However, more detailed information can be obtained from aberration-corrected images. Another cuboctahedral particle is shown in Fig. 4a,b in STEM bright-field (BF) mode and HAADF mode, respectively. The fast Fourier transform (FFT) of both images can be readily indexed as corresponding to the $\langle 110 \rangle$ zone axis. Since the HAADF-STEM intensity is sensitive to the atomic number and to the mass thickness, we can obtain profiles of different rows of atomic columns. The results are shown in Fig. 4c–e for the first and second atomic rows of the $\langle 100 \rangle$ surface and the $\langle 111 \rangle$ atomic row. The measurements were taken along the directions indicated by the arrows shown in Fig. 4b. As can be seen, the profiles show an increase in intensity and then a decrease, displaying a Gaussian distribution.

A close examination of the regular cuboctahedral structure shown in Fig. 5 indicates that the intensity in the $\langle 100 \rangle$ direction should always be increasing (Fig. 5a), and that the profile in the $\langle 111 \rangle$ direction should display constant intensity. The only possibility to reconcile the experimental data with the Au/Pd three-layer model (Fig. 5b) is that two phenomena are occurring. First, the particle is not smooth, and contains edges, steps and kinks (Fig. 5c), and secondly there is atomic mixing of Au and Pd in the columns on or near the surface. From consideration of the theoretical models included in this report, we conclude that the addition of a second metal introduces defects to the surface of the particles. The lack of extra diffraction spots in the electron diffraction patterns of the particles (Fig. 1c) indicates that these are single crystals. The absence of defects in the electron diffraction patterns is an intriguing fact that requires additional experimental analysis.

Conclusions

In summary, this paper reports the direct observation of individual atomic columns of nanoparticles with three-layer morphology. This enabled us to investigate their structure–composition relations. A very important principle for bimetallic

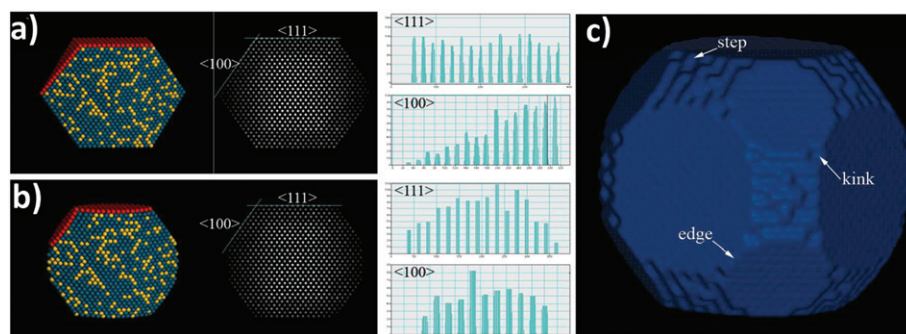


Fig. 5 Models corresponding to (a) a regular cuboctahedron and (b) Au/Pd modified cuboctahedron. The thickness of the atomic columns increases for a $\langle 100 \rangle$ face, whereas is constant for a $\langle 111 \rangle$ face for the regular cuboctahedron. The model of the reconstructed shape of Au/Pd nanoparticles is consistent with the observed intensity profiles. (c) Solid polyhedron representing an Au/Pd nanoparticle, showing steps, edges and kinks on the surface. Locally, these features produce high-index surfaces.

particles was elucidated, namely that for nanoparticles displaying alloying and a core-shell structure, differences in atomic radii and atom mobilities causes the formation of high Miller index facets. However, these facets are unstable and decompose into steps formed composed of {100} and {111} facets. It is very likely that the stepped surface of analyzed nanoparticles plays an important role in the catalytic activity of bimetallic systems.

Acknowledgements

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