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1	Numerical Evaluation of Pin-Bearing Strength for the Design of Bolted
2	<b>Connections of Pultruded FRP material</b>
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Abstract: Presented in this paper are finite element (Abaqus) predictions for the strength of a pultruded fibre reinforced polymer material subjected to pin-bearing loading with hole clearance. One of the distinct modes of failure in steel bolted connections is bearing. It is caused by the compression action from the shaft pressing into the laminate, and when there is no lateral restraint the mechanism observed at maximum load shows 'brooming' for delamination failure.

21 Each lamina in the glass fibre polyester matrix material is modelled as a homogeneous, anisotropic 22 continuum and a relative very thin resin layer is assumed to contain any delamination cracking between stacked layers. A cohesive zone model is implemented to predict the size and location of the 23 initial delamination, as well as the load-carrying capacity in a pin-bearing specimen. Finite element 24 simulations (as virtual tests) are performed at the mesoscale level to validate the modelling 25 methodology against experimental strength test results with delamination failure, and to show how pin-26 bearing strength varies with parameter changes. For an example of the knowledge to be gained for the 27 design of bolted connections, the parameteric study where the mat reinforcement is either continuous 28 strand or triaxial (+45°/90°/-45°/chopped strand) shows the latter does not provide an increase in pin-29 bearing strength. 30

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Author keywords: Damage mechanics; Bearing failure; Finite element modelling; Pultruded
 material.

## 35 Introduction

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37 Shapes made by the pultrusion composite processing method consist of thin-walled laminated panels of glass Fibre Reinforced Polymer (FRP) matrix connected to form open or closed cross-38 sections. Because I, wide flange, channel, box and leg-angle shapes mimic steel sections it is natural 39 40 that frame construction follows what is seen in conventional steelwork (Creative Pultrusions 2016; Strongwell 2016; Turvey 2000). Being lightweight and resistant to corrosion, and having expected 41 low life cycle costing, pultruded structures are increasingly used where these attributes meet the 42 43 requirements of the construction project, such as for pedestrian bridges (Anonymous 2016). Introduced in Mottram and Zafari (2011) is the rationale for steel bolting to be a main method 44 of connection (Creative Pultrusions 2016; Strongwell 2016), and the requirement that, for strength 45 design, a reliable test method is needed to determine bearing strengths. The *bearing strength* is an 46 important material parameter for static strength design (Bank 2006; Mottram and Turvey 2003), and 47 48 can be characterized as the resistance of the material to a fastener loaded hole. Strength depends on a number of parameters including: FRP thickness (t); FRP mechanical properties; FRP material 49 orientation to the bearing force; bolt material; bolt diameter (d); clearance hole size. In bolted lap-50 51 joints the end distance  $(e_1)$ , which is the distance from the centre of the hole to the free edge perpendicular to the loading direction has to exceed a limiting value for the bearing failure mode to 52 53 govern.

An advantage for bolted connection design failing in the bearing mode is that it might provide
the connection/joint with a degree of damage tolerance and structural integrity. Bearing failure can
be enhanced by: (i) correctly sizing the connection, usually end distance to fastener diameter ratio *e*<sub>1</sub>/*d* ≥ 3 providing the width to fastener ratio is 4 or higher, see Bank (2006), and Girão Coelho and
Mottram (2015a); (ii) the choice of the fibre reinforcements in the lamination stacking sequence.
Mottram and Zafari (2011) justify why the measurement of bearing strength for design
calculations must be for the lower bound pin-bearing value. The pin-bearing condition assumes the

steel bolt shaft has no thread in bearing and that there is no lateral restraint, which will not initially 61 exist in practice because the washer, nut, bolt combination should be tightened. The paper explains 62 the reasons for having a strength test method for pin-bearing strength with a plate-shaped coupon 63 64 that is no larger in size than, say  $80 \times 80 \times t$  mm. Using an in-house test approach, detailed in Mottram and Zafari (2011) or Matharu (2014), numerous test results have been reported from programmes of 65 characterization work (see also Zafari and Mottram 2012) to understand the variations in pin-66 bearing strength and characteristic values for pultruded materials. Two different materials were 67 studied, and in the test matrix parameters were: four bolt diameters, e.g. 9.53 mm (3/8 in.), 12.7 mm 68 (1/2 in.), 15.8 (5/8 in.) and 25.4 mm (1 in.) (Mottram and Zafari 2011) or M10, M12, M16 and M20 69 70 (Matharu 2014); different material orientations (e.g. 0°, 90° and 45°); non-aged and hot-wet aged materials. One aim of the characterization work was to obtain data on how pin-bearing strengths 71 might change in the field due to the long-term effect of exposure to a site's environment. 72 Both materials were from the American pultruder Creative Pultrusions (2016) having glass 73 fibre reinforcement in the form of alternative layers of unidirectional rovings and a mat, and a fire-74 75 retardant matrix of an isophthalic polyester polymer. For the Mottram Zafari (2011) and Zafari and 76 Mottram (2012) tests the pultruded material was from the standard 1525 series having mat reinforcement of a Continuous Strand Mat (CSM) having a random arrangement of continuous 77 fibres. Matharu's (2014) work was with material from the Pultex® SuperStructural 1525 series, and 78 the main difference is that the mat reinforcement is triaxial, having unidirectional fibres at 90°, and 79  $\pm 45^{\circ}$  and a 'thick' backing of a chopped strand fibres. Another difference in the test matrix with 80 Matharu is that testing was with and without bolt thread in bearing; only the latter connection 81 82 condition can be linked to the Finite Element (FE) work presented in this paper. The FE modelling methodology applied by the authors uses options solely available from the 83 general purpose software Abaqus with its implicit solver (2016). Girão Coelho et al. (2016) 84 85 introduce, and show how the approach is modelling the various failure modes, and, in particular, for

the parametric studies given later the critical mode of progressive delamination using cohesive zone

models. A similar approach using Abaqus/Explicit has been reported by Du et al. (2016) with their 87 progressive damage analysis implemented via a user subroutine VUMAT. Because no customized 88 user-subroutines are required with the author's modelling methodology the computational results 89 90 using Abaqus implicit can be universally reproduced. Simulation outputs for pin-bearing strengths 91 will be validated using non-aged test results from Matharu (2014), and in a strength comparison 92 using data from Mottram and Zafari (2011) for changing the mat reinforcement. The main 93 contribution of this paper is the predictions from a series of parametric studies to investigate how 94 pin-bearing strength might vary with changes in the fibre reinforcement.

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# 97 Fundamental Behaviour and Analysis Approach

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Bearing failure is a mode showing local crushing and delamination of the laminated material in 99 100 direct contact with the steel bolt shaft (Bank 2006; Mottram and Turvey 2003). The bearing strength of monolithic materials (e.g. steel) is generally evaluated without lateral restraint or through 101 thickness constraint from a bolt tightening. Application of this test condition with FRP composites 102 allows delamination fracturing to occur. In this paper the word *delamination* has the meaning for the 103 formation and growth of a flat flaw in an initially flawless interface (between two previously 104 105 'bonded' laminae) that grows into a sizeable delamination crack. With PFRP materials this failure mechanism leads to the lowest bearing strength. 106 The presence of composite material viscoelasticity (Mottram 2005), and the influence of 107 structural actions on the bolted connection ensure that it will be unreliable to assume there can 108

always be lateral restraint at the end of the design working life, which can for a structure be, 50 or

110 100 years. It is for this reason (Mottram and Zafari 2011) that the *pin-bearing strength* has to be the

strength used in design calculations for bolted connection strength design.

The FE study presented next to model the bearing behaviour includes a *continuum damage* 112 model with a cohesive zone approach. Continuum damage models address the intralaminar failure 113 mechanisms from a global standpoint, whereas if the analyst used individual damage mechanisms 114 115 they would be homogenized and constructed around a failure criterion (Knops 2008). This approach is the least complex and uses the composite lay-up modeller tool within the Abaqus (2016) pre-116 117 processor to define the individual laminae through a laminate thickness. Each layer is cohesively bonded together to form the *lamination with interfaces* assumed to have a thickness of  $10^{-3}t_{lav}$ , 118 where  $t_{lay}$  is the thickness of the thinnest layer either side of the interface. Values for  $t_{lay}$  are 119 introduced in the section for the Description of the Model. This modelling dimension plays the role 120 of a length scale, and it has been shown by Girão Coelho (2016) that the thickness of the cohesive 121 interface does not affect the computational performance provided its thickness is small enough 122 123 compared to  $t_{\text{lay}}$ .

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## 126 Modelling of Progressive Delamination Using Cohesive Zone Models

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Delamination failure is the separation of reinforcing layers from each other, as a consequence 128 of shear stresses acting in planes parallel to the layers' interfaces and/or tensile stresses acting in the 129 through-thickness direction. This phenomenon is a typical *crack growth* problem and is treated in 130 the framework of fracture mechanics (Girão Coelho 2016). Today, the most popular computational 131 method for the prediction of delamination failure is based on *cohesive zone models* that provide a 132 natural bridge between strength-based models and energy-based models for fracture. This allows 133 134 delamination to be described by a single framework that covers a range of applications for which, on their own, neither a strength nor an energy criterion might not be sufficient. 135

Cohesive zone models consider fracture as a gradual phenomenon in which separation takesplace across an extended crack tip, or cohesive zone, and is resisted by cohesive tractions (Ortiz and

Pandolfi 1999). Thus cohesive zone elements do not represent any physical material, but describe the cohesive forces which occur when layers in the lamination are being pulled apart. In FE modelling, cohesive zone elements are placed between the continuum elements used to model the individual laminae. A key analysis feature of these elements is that they include the effect of first delamination failure, and the subsequent crack propagation by means of critical strain energy release rates.

In our modelling methodology, see Girão Coelho et al. (2016), the simulation of interlaminar 144 damage is based on the cohesive zone approach using the Abaqus three-dimensional cohesive 145 element COH3D8. The study is performed in the quasi-static regime. The traction-separation law 146 formulation assumes a non-zero elastic stiffness of the cohesive zone, which is physically motivated 147 by the reduced stiffness of the matrix-rich very thin interface layer  $(t_{lay})$  as compared to a perfect 148 bond assumed to exist between the fibres and the surrounding polymer matrix. From a numerical 149 point of view, this elastic stiffness can be understood as a penalty-type enforcement of displacement 150 continuity in the elastic range. 151

A quadratic stress criterion is used for the *damage initiation criterion*. To specify the conditions for separation in the cohesive zone model the following expression is chosen (Brewer and Lagace 1988; Camanho *et al.* 2003):

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$$\left(\frac{\langle \sigma_{\rm n} \rangle}{f_{\rm I}}\right)^2 + \left(\frac{\sigma_{\rm s}}{f_{\rm II}}\right)^2 + \left(\frac{\sigma_{\rm t}}{f_{\rm III}}\right)^2 = 1$$
(1)

where  $\sigma_n$  is the stress in pure opening mode,  $\sigma_s$  is the stress in the first shear direction,  $\sigma_t$  is the stress in the second shear direction,  $f_1$  (opening),  $f_{II}$  (sliding)) and  $f_{III}$  (tearing) are the peak strength values in the same directions, and:

159 
$$\langle \sigma_{n} \rangle = \sigma_{n} \text{ for } \sigma_{n} > 0 \text{ and } \langle \sigma_{n} \rangle = 0 \text{ for } \sigma_{n} \le 0.$$
 (2)

160 The latter modelling constraint in Eq. (2) is because compressive normal stresses cannot open a161 delamination crack.

162 Progression of damage at the interfaces is modelled using a linear softening law and a critical 163 mixed mode energy behavior based on the Benzeggagh-Kenane criterion (1996), which is described 164 by the following expression:

165 
$$G_{\rm c} = G_{\rm I,c} + \left(G_{\rm II,c} - G_{\rm I,c}\right) \left[G_{\rm II} / \left(G_{\rm I} + G_{\rm II}\right)\right]^{\eta}.$$
 (3)

In Eq. (3)  $G_{\rm m,c}$  (with m = I, II, III) is for the total critical strain energy release rate associated 166 with delamination mode m, and  $\eta$  is for the semi-empirical criterion exponent applied to 167 delamination initiation and growth. Based on the argument given by Girão Coelho (2016) exponent 168  $\eta$  is assumed to be 1.5. Illustrated in Figs. 1(a) to 1(c) are the three distinct opening modes that can 169 occur singularly or interact together to cause the initiation and formation of a delamination failure. 170 171 The use of cohesive zone models requires that a very fine mesh specification is used to ensure that sufficient interface elements exist within the cohesive zone length where the crack tip is 172 moving. If the mesh design happens to be too coarse, the cohesive stress at the discontinuity may 173 not even reach the interfacial strength and, as a result, the required failure mode is missed. Falk et 174 al. (2001) suggest a minimum of two to five elements in this cohesive zone length in order to 175 perform a reliable simulation. Turon et al. (2000) indicate that for typical graphite-epoxy or glass-176 epoxy FRP materials, the length of the cohesive zone should be smaller than one or two millimetres. 177 As a consequence, the mesh size required in order to have more than two elements in the cohesive 178 179 zone should be smaller than half a millimetre. For full-sized structural models of pultruded shapes and structures this has the obvious consequence of a computationally expensive solution. Current 180 numerical simulations for the 80×80×9.5 mm pin-bearing specimen comply with the cohesive zone 181 182 element mesh size requirements specified by the first author (Girão Coelho 2016) to allow stable numerical simulations of interface delamination. 183

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# 186 Finite Element Validation of Strength Behaviour

A general FE model for the pin-bearing problem is developed and validated in this section using 188 experimental strength results from the PhD work by Matharu (2014). Note that the objective of the 189 FE work is to obtained predictions for the pin-bearing strength, which is established by the 190 191 maximum test load (or stress, which is this load divided by the projected bearing area given by pin diameter times specimen thickness). There are engineering/scientific reasons why the FE 192 193 simulations might not be numerically reliable in predicting the onset of damage and all of the modes of failure that occur prior to peak load. There is a likelihood that piror to loading there are already 194 matrix cracks (running parallel to the unidirectional fibre) because of residual thermal strain owing 195 to the cooling down from a temperature of 150°C in the pultrusion composite processing. What is 196 197 known from the extensive programme of static strength tests by Matharu (2014), and by Mottram and Zafari (2011), is that when the bearing load is aligned in the direction of pultrusion the load-198 stroke curve (testing machine displacement) is virtually linear elastic until to ultimate failure, which 199 always occurs with a noticeable load reduction and audible acoustic emissions. This signal of 200 ultimate failure is too pronounced to be for initial/new matrix cracking, and when the post-failure 201 202 load is released inspection of the bearing surface shows there to be interfacial delamination fractures. This is the authors' evidence-based justification for how the FE invetigation was carried 203 out and reported in this paper. 204

Matharu (2014) tests are for pultruded material taken from the web of a Pultrex® 205 SuperStructural wide flange shape with nominal thickness of 9.53 mm (3/8 in.) (Creative 206 Pultrusions 2016); the measured thickness is closer to 9.6 mm. The test matrix has plain steel pins 207 of four metric diameters from 10 to 20 mm diameter with a clearance hole size that introduced the 208 209 maximum allowable fabrication tolerance using the guidance in Anonymous (2012) from the Pultrusion Industry Council, USA. The technical reason for the maximum is that the bigger the hole 210 clearance is, the lower is the pin-bearing strength (Yuan 1996), and for a safe design the lowest 211 212 characteristic strength is required (Mottram and Zafari 2011). Without the fabrication tolerance the recommended clearance size is 1.6 mm (or 1/16 in.) (Anonymous 2012; Creative Pultrusions 2106; 213

214	Strongwell 2012). The nominal clearance $(d_0 - d)$ is 2.2 mm (given by 1.6 mm clearance +0.4 mm
215	tolerance) for M10 and 2.4 mm (1.6 mm clearance +0.8 mm tolerance) for M12, M16 and M20 bolt
216	sizes. After drilling a pilot hole of diameter less than required the final hole dimeter to $\pm 0.02$ mm
217	was prepared using a Cinccinati Arrow 450 Milling machine. Batches of ten nominally identical
218	specimens were tested to determine the pin-bearing strength that is statistically analysed and
219	critically evaluated by Matharu (2014).
220	The geometry was taken comparable to the experimental set-up, with the pin diameter as FE
221	modelling parameter. Fig. 2 defines the specimen geometry and shows the layered structure of the
222	SuperStructural web material.
223	Specific characteristics and attributes that were incorporated into the modelling are:
224	1. Based on the relatively high stiffness of the steel pin with respect to the longitudinal web
225	material, c.f. 210 kN/mm <sup>2</sup> (BS EN 1993-1-1:2005) to 21.3 kN/mm <sup>2</sup> (Creative Pultrusions
226	2016)), the bearing load is exerted by a rigid (circular smooth) pin, this is shown in Fig. 2.
227	2. Interaction between pin and hole is not only modelled with a normal bearing load, but also
228	with a through-thickness friction having a friction coefficient taken equal to 0.25, based on
229	the testing by Mottram (2005) to account for the FRP/steel contact.
230	3. Residual thermal stresses resulting from the pultrusion thermal-induced and cure-induced
231	shrinkage processes are not considered. Chen et al. (2001) and Zhang et al. (2004) developed
232	micro-mechanical models that have been successfully employed in computational analysis
233	and showed that initial residual stresses within pultruded structural shapes dissipate over time
234	owing to viscoelastic creep relaxation (Bank 2006).
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237	Material Properties

This section describes the physical and mechanical properties of the original material used in the extensive testing programme by Matharu (2014). The pultruded material is from the Pultex<sup>®</sup>

SuperStructural 1525 series of 'off-the-shelve' shapes. Identifier 1525 means the thermoset 240 polyester matrix is Class 1 Flame Retardant. The E-glass FRP fibre architecture consists of mat 241 layers interspersed with nominally constant thickness layers of UniDirectional (UD) rovings having 242 243 a fibre volume fraction of 60%. The mat reinforcement is type E-TTXM 4008, a Triaxial Stitched Fabric Mat (TSFM) from Vectorply Corporation (http://vectorply.com/wp-244 content/uploads/2015/06/E-TTXM-4008.pdf). The pultruded shape has an outer (relatively thin at 245 246 0.03 mm (see Table 1)) surface veil (non-structural) for UV protection and a barrier to moisture diffusion. The NEXUS<sup>®</sup> veil layer is resin-rich, non-structural and consists of randomly orientated 247 short 100% melt polyester fibres, which has a mass per unit area  $< 100 \text{ g/m}^2$ . 248 As seen in Fig. 2 there are five alternating layers of UD and TSFM, with two TSFM layers at 249 mid-thickness having no UD layer between them. TSFM is of a stitched continuous fabric having 250 four layers with the lay-up sequence of  $+45^{\circ}/90^{\circ}/-45^{\circ}/random$  chopped strand. Assuming the 251 volume fraction of fibres in the continuous layers is 60% the chopped strand backing layer in TSFM 252 has a volume fraction of 16%. The thicknesses of the layers are 0.35 mm for the 90°, 0.51mm for 253 the  $\pm 45^{\circ}$  and 0.54 mm for the chopped strand. It is known from Creative Pultrusions Inc. that the 254 UD layers comprise of 56 yield rovings. 255

The thermoset resin is the Reichhold 31031 unsaturated isophthalic polyester resin. To produce a matrix for the pultrusion composite processing there are additives and fillers. The largest proportions in the formulation, which consist of 13 constituent parts, is for the polyester resin, at approximately 80% of the bulk matrix.

A series of resin burn-off tests were conducted by Matharu (2014) to estimate the volumetric proportions of UD and TSFM fibres, and matrix, as well as to establish the nominal thicknesses of each layer. The resin burn-off procedure was adapted from the method described in Appendix B of the PhD thesis by Lane (2002), and with reference made to the testing procedure given in Ye *et al.* (1995) and ASTM D2584-11 (2011). Table 1 summarizes the results from the resin burn-off tests.

Because the reinforcing effect of the Surface Veil (SV) layers is minimal it can be seen that the total

amount of glass fibre with the two main reinforcing laminae of UD and TSFM are equal at 34% volume fraction. The volume percentages of UD, TSFM and matrix are in the range 30 to 34. The final row in Table 1 reports the nominal thicknesses of the layers (the  $t_{lay}s$ ) on the assumption that they are constants for a particular reinforcement type in the original material. Note that both the veil and TSFM thicknesses are predetermined by their construction. Only the UD layers have a freedom to have a processing variation for layer thickness.

The mechanical properties presented in Table 2 are defined with respect to the local co-ordinate 272 system, with: 1 for the pultrusion direction; 2 for the in-plane direction normal to the direction of 273 pultrusion; 3 for the through-thickness direction. In the literature the direction of pultrusion is often 274 275 referred to as the longitudinal (or length-wise) direction and the perpendicular (in-plane) direction as the transverse (cross-wise) direction. Notation  $f_1$  is therefore for the in-plane longitudinal 276 strength,  $f_2$  for the in-plane transverse strength,  $f_{1,S}$  for the in-plane shear strength, and  $f_{2,S}$  for the 277 transverse shear strength. Values in the table for Tensile (T) and Compressive (C) actions are for in-278 plane strengths. 279

280 The determination of the elastic constants for the three different laminae are reported in Table 2 was carried out using micromechanical modelling with volume fractions of the constituents 281 established by using the resin burn-off method, as described in Lane (2002). In making the 282 micromechanical modelling calculations, it is assumed that the densities of matrix and glass fibres 283 are 1.1 g/cm<sup>3</sup> (Reichhold, 2006) and 2.56 g/cm<sup>3</sup> (Hancox and Mayer, 1994), respectively. The 284 modulus of elasticity for the matrix constituent is assumed to be 3.2 kN/mm<sup>2</sup> (Reichhold, 2006) and 285 for the fibres it is taken as 72 kN/mm<sup>2</sup> (Hancox and Mayer, 1994). The constituent Poisson's ratios 286 are assumed to be 0.36 and 0.22, respectively. Using the rule of mixtures approach with the two 287 lamina stiffnesses in Table 2 predictions for the elastic constants of the 9.6 mm thick panel are  $E_1$  = 288 25.0 kN/mm<sup>2</sup>,  $E_2 = 16.7$  kN/mm<sup>2</sup>,  $G_{12} = 4.5$  kN/mm<sup>2</sup> and  $v_{12} = 0.36$ . These elastic constants are seen 289 to be different to the tabulated properties of  $E_1 = 21.3 \text{ kN/mm}^2$  and  $E_2 = 9.6 \text{ kN/mm}^2$  in the Design 290 Manual from Creative Pultrusions (2016). The Design Manual stiffnesses are for a range of 291

pultruded shape sizes and the larger the size is, the lower can be the volume proportion (and maybe 292 fibre volume fraction) of the unidirectional reinforcement. As a consequence of this fact the 293 tabulated moduli of elasticities are for minimum measured from all the shapes in the series range. 294 295 Because the published stiffnesses are for the whole 1525 series they are less reliable, and this is the engineering reason why the micromechanical modelling laminae stiffnesses in Table 2 are input 296 297 data for the FE analyses. The web material consists of alternative reinforcing layer of UD and the 298 TSFM having a balanced lay-up, comprising TSFM1 layer at 1.38 mm thickness, UD1 at 1.93 mm, 299 TSFM2 at 1.42 mm, TSFM3 at 1.39 mm, UD2 at 2.01 mm and TSFM4 at 1.47 mm. Note that midthickness layers TSFM2 and TSFM3 do not have UD reinforcement between them. Accounting for 300 301 experimental errors in the resin burn-off testing the thickness of the four TSFM layer can be taken to be constant at 1.4 mm. The UD layers have a nominal constant thickness of 2.0 mm. 302

The strengths listed in the middle column in Table 2 are for the laminate. PFRP materials 303 having the individual fibre reinforcement types (and the same matrix) are not pultruded and so their 304 strength (and stiffness) properties cannot be measured. Matharu (2104) conducted a series of 305 306 tension and compression coupon tests (in batches of 10) to determine the panel's in-plane strengths. The longitudinal (compression) strength ( $f_{1,C}$ ) for the batch ranges from 280 to 328 N/mm<sup>2</sup>, whereas 307  $f_{2,C}$  is found to range from 99 to 153 N/mm<sup>2</sup>. The mean longitudinal strength in tension ( $f_{1,T}$ ) at 294 308 N/mm<sup>2</sup> is 4% lower than the mean compression, whereas the transverse tensile strength ( $f_{2,T}$ ) is, as 309 expected, much lower, at 63% of the compression mean of 133 N/mm<sup>2</sup>. The mean in-plane shear 310 strength  $f_{1,s}$  is 91 N/mm<sup>2</sup> was taken from D'Alessandro (2009). In the FE work the transverse shear 311 strength  $(f_{2,s})$  is taken to be  $f_{1,s}$ . 312

Each (continuum) damage evolution law includes a corresponding fracture energy (or fracture toughness), *G*<sub>c</sub>, that governs crack growth for the modes illustrated in Fig. 1. Fracture energy is defined as the work needed to create a unit area of a fully developed crack. Guidelines for the evaluation of these fracture properties can be found in Pinho *et al.* (2006) and Maimi *et al.* (2007). The four values adopted in this study are listed in the last column in Table 2. They were not experimentally determined, but estimated based on research expertise and data available from the

319 literature (Kelly and Zweben 2000; Girão Coelho *et al.* 2015).

The assumed modelling inputs for interfacial mechanical properties are summarized in Table 3. 320 321 To establish the peak strengths  $f_1$  and  $f_{II} = f_{III}$  for Eq. (1), in the absence of experimental measurements, the in-plane strengths  $f_{2,T}$  and  $f_{1,S} = f_{2,S}$  were factored down by adopting a 322 323 weakening factor  $f_w$ , according to guidance from the failure criterion work by Puck and Schürmann (2002), see Knops (2008), and from the computational modelling work by Wimmer et al. (2009). 324 The critical normal interface traction (for opening mode) of the cohesive zone elements is therefore 325 assumed to be equal to the transverse tensile strength,  $f_{2,T}$ , times the weakening factor. The critical 326 327 shear interface tractions (for sliding and tearing modes) are both assumed to be equal to the shear strength,  $f_{2,S}$ , times the same weakening factor. 328

It was found in a sensitivity analysis from a series of FE simulations that the overall response 329 for  $f_w = 0.25$  gives the one closest to the pin-bearing experimental results from Matharu (2014). 330 With  $f_I = 21 \text{ N/mm}^2$  and  $f_{II} = f_{III} = 23 \text{ N/mm}^2$  the analysis indicates a relatively weak interface 331 between the laminae, as defined by the FE modelling data given in Fig. 2 and Table 2. A similar  $f_w$ 332 has already been proposed by Girão Coelho et al. (2016) for this same pultruded material when 333 analysing the different structural engineering problem of delamination failure in a pultruded leg-334 angle shape subjected to tying force, as would be found in a web clip connection between beam and 335 column members. 336

The piecewise linear traction-separation law of the cohesive elements is described according to the lamina strength presented in the last column in Table 2. Using the guidance from Camanho *et al.* (2003) the elastic stiffnesses (per unit area) of the interface ( $K_{\rm I}$ ,  $K_{\rm II}$  and  $K_{\rm III}$  associated with  $f_{\rm I}$ ,  $f_{\rm II}$ and  $f_{\rm III}$ ) are assumed to be equal to 10<sup>6</sup> N/mm<sup>3</sup>. For modes I, II and III illustrated in Fig. 1 the critical energy release rates of  $G_{\rm I,c}$  and  $G_{\rm II,c} = G_{\rm III,c}$ , were taken as 0.2 (200) and 0.5 (500) N/mm ( $J/m^2$ ), respectively (Girão Coelho 2016).

## 345 Description of the Model

Defined in Fig. 2 are the dimensions for the pin-bearing specimen that is to be modelled and 346 347 analysed. Each plate of 9.6 mm thickness (t) is 80 mm long  $(e_1)$  and 80 mm wide (w). The pin diameter (d) is set equal to 11.8 mm for M12 and to 19.8 mm for M20 steel bolting, while the hole 348 349 diameter  $(d_0)$  was taken 2.4 mm larger, based on practical dimensions. This allowed for the maximum clearance hole tolerance size (Anonymous 2012) and the shaft diameter to be identical to 350 the bolt diameter specification. The pin was placed centrally in the hole, which itself is mid-351 positioned in the top surface. The UD layers are oriented with the fibres aligned to the direction of 352 the bearing load. 353

The FE model is constructed from stacked continuum shell elements with the individual UD 354 and TSFM layers being modelled separately. Although the element SC8R possesses the geometry 355 of a brick, its kinematic and constitutive behaviour is similar to those of conventional shell elements 356 (Abaqus 2016). The continuum shell elements are able to reproduce reliable results in simulations 357 358 of thin-walled laminated structures by means of only one element in a lamina thickness, owing to a higher-order displacement field (Parisch 1995; Remmers et al. 2003). Cohesive interface elements 359 are used to connect together an TSFM and an UD layer to allow for (multi-)delamination failures to 360 occur. As shown in the diagram in Fig. 2 four cohesive interfaces I1 to I4 are modelled. Justification 361 for no cohesive interface between the two touching TSFM layers at mid-thickness is that Matharu 362 (2014) did not observe, after tests, delamination failure over this interface. 363

The in-plane mesh geometry for the interfaces had to be the same as that for the laminae. A typical mesh with its very refined mesh specification is illustrated in Fig. 3, where adjacent to the hole a finer mesh is used to correctly capture delamination and the important contact situation. Note that using mirror symmetry about the two TSFM layers at mid-thickness the mesh is for a thickness of 4.8 mm (or t/2). The thickness of interface layers I1 and I2 are  $10^{-3}t_{lay}$ , based on the 1.4 mm TSFM layer thickness. Element size was increased towards the plate edges to reduce calculation

time. The FE mesh specification in our study of the original material has a total of 27834 (M12) 370 continuum shell SC8R elements (or 32520 elements for the M20 specimen), and 18556 COH3D8 371 cohesive elements (M12 and two interfaces) (or 21680 elements for M20 specimen and two 372 373 interfaces). Interfaces I1 and I2 (or I3 and I4) are included in the FE model. As mentioned earlier the (steel, smooth shafted) pin is modelled as a three-dimensional 374 analytical rigid body revolved shell. 375 Loading of the specimen is simulated by displacing the bottom edge against the fixed pin as 376 indicated by the 'load' arrows in Fig. 2. Vertical deformation is resisted by the contact surface for 377 pin and hole perimeter at the top of the specimen. Contact behavior is modelled with the 'hard' 378 379 surface-to-surface contact formulation in the normal direction and the friction is modelled using an isotropic friction model with a coefficient of 0.25. Note that the coefficient of friction was not 380 measured (Mottram 2005) for the steel-FRP contact in the through-thickness direction. This 381 represent one of several modelling data assumptions made to complete a successful FE simulation 382 and they have been reported in this paper. 383 384 The Abaqus implicit analysis is run with the laminae properties listed in Tables 2 assigned to the appropriate UD or TSFM layers and with the interfacial properties in Table 3. For each lamina 385

the Hashin failure criterion (Hashin 1983) is used to predict damage onset (Girão Coelho 2016), and
because interfacial delamination failure always occurs first this criterion is of secondary

importance. To account for damage progression after crack initiation there is a fracture energy

calculated for each of four failure modes using the fracture energies in Table 2 (Girão Coelho *et al.* 

2015). To assist with numerical convergence of the Abaqus solver the authors used previous

simulation ouputs (Girão Coelho *et al.* 2015) to set a global stabilization factor to  $2 \times 10^{-4}$ .

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## 394 Modelling Strategy Validation

Results from static analyses are compared with maximum pin-bearing loads from experiments. In the series of tests a specimen's stiffness was not measured because it had no meaning to the aim of the strength investigations in Matharu's PhD work (2014), which were to establish characteristic values for pin-bearing strengths. In this paper the web material studied by Matharu is referred to as the original (non-aged) material.

400 One finding from the FE simulations is that delamination failure always governs the magnitude of the load for pin-bearing strength and this matches experimental observations. As a direct 401 consequence the relevant Hashin parameters were found to be always below 1.0. This shows that it 402 has been acceptable for the authors to have made the approximate assumption that the laminate 403 404 strengths in Table 2 could be taken to be the individual laminae strengths. The pin-bearing strength, here expressed in terms of applied load, is reached when there is extensive delamination growth at 405 the two interfaces (I1 and I2), and there is a sudden load drop-off during the displacement 406 controlled loading. Fig. 4 is for plots constructed from Abaqus outputs of load (kN) versus the axial 407 displacement (mm) in the bearing force direction. The grey shaded circle symbols are for numerical 408 409 results with the larger black filled circle at the load when initiation of delamination fracturing is numerically predicted. In the legend there is (I1) to identify that the critical delamination is at 410 Interface 1, its location is defined in Fig. 2. The batch test results from Matharu (2104) are reported 411 412 by three representative loads. The batch mean is given by the solid horizontal line, and the maximum and minimum specimen loads are given by the upper and lower dashed lines. The load 413 range between the two dashed lines gives the measured variation from testing a batch of 10 414 nominally identical specimens. The coefficient of variation for the pin-bearing strength (Matharu 415 416 2014) for the M12 batch is 10.4 and for the M20 batch is 8.8%.

Fig. 4(a) is for the M12 pin, and using the same axis scales the equivalent computational results for the M20 pin are reported in Fig. 4(b). The predicted load-displacement response is seen to start off linear, and progress with non-linearity above 15 kN for M12 and 22 kN for M20. The computational response is seen to differ from the virtually linear load-stroke relationship recorded

by the testing machine. Above the loading (defined by filled circle symbol) for delamination 421 initiation there is a reduction in structural stiffness and the load-displacement response softens with 422 progressive delamination growth, before ultimate failure. Stiffness degradation after delamination 423 424 initiation and progressive growth is not an outcome in testing since delamination onset is believed to occur at the same instance the peak load is reached. Importantly, should the FE modelling 425 methodology presented herein be used to aid the design of bolted connections the predicted load for 426 delamination initiation is useful because as Figs. 4(a) and 4(b) show it is below the lower bound test 427 result. 428

Returning to the information in Fig. 4, the peak (Abaqus) loads of 23 kN (Fig. 4(a) for M12) and of 35 kN (Fig. 4(b) for M20) compare favourably with the mean experimental loads (Matharu 2014) of 22 kN and 34 kN, respectively. The predictions are in very good agreement with computed strengths only 4.5% and 6% higher. The positive correlation shown by the results in Figs. 4(a) and 4(b) for two independent batches provides good validation for applying the FE modelling approach to determine pin-bearing strengths.

In Fig. 5 the deformed shape is shown at peak load (magnification factor is 2.5). Fig. 5(a) is for the M12 pin and the maximum load of 23 kN, whereas Fig. 5(b) is for the M12 pin at maximum load of 36 kN. In both parts the deformations clearly shows the typical 'brooming' for compression induced delamination failure at interfaces I1 (nearest) and I2 (farthest).

Inspection of the localised through-thickness deformation in Fig. 5 helps to explain why 439 delamination failure is the mechanism that governs the pin-bearing strength. Under the increasing 440 pressure from the bearing (rigid) pin the existence of Poisson's ratio effect is for the laminate to 441 442 need to expand freely in the through-thickness direction. This physically cannot happen because of 443 the friction restraint over the steel pin/FRP contact area and the volume of the surrounding FRP material resisting the bearing load. As seen in Figs. 5(a) and 5(b) the deformation is in the form of a 444 bulge with a short wavelength ( $\cong 0.44d_0$ ). A complex, through-thickness stress field with tension (in 445 the *z*-direction) is generated local to the hole perimeter, and within the influence of the bulge zone. 446

It can be speculated that when the maximum through-thickness tensile stress reaches  $f_i$ , or the limit given by Eq. (1) when  $f_i$  interacts with  $f_{II}$  and  $f_{III}$  there will be cohesive failure for initiation of delamination failure. Propagation of a delamination crack to the final size shown, for example, in Fig. 6 is numerically controlled by fracture mechanics (Girão Coelho 2016) and the energies listed in Table 3.

The contour plots presented on interface surfaces in Fig. 6 show that, at maximum load, 452 delamination fracturing at the two interfaces has progressed from the hole into the body of the 453 specimen. Fig. 6 parts (a) and (b) are for the M12 and M20 bolt diameters, respectively. Damage 454 progression can be tracked, from 0 (none) to 1 (complete), using the Abaqus output parameter 455 Stiffness DEGradation (SDEG), which indicates the state of damage in the cohesive elements and 456 thereby provides insight into damage initiation and propagation. The red coloured area (SDEG is 457 0.7 to 1.0) is for complete interface separation and the thin (green coloured) zone around the 458 damage zone's perimeter is for partial damage (SDEG is 0.3 to 0.7). Where the material on the 459 interface surface is coloured dark blue there is no damage (for SDEG < 0.3), with full continuity 460 across the interface. Interface I1 is on the left-side and interface I2 on the right-side. One 461 observation from the contouring presented in Fig. 6 is that the delamination damage has progressed 462 from an initial state with a stable crack front; this is evidence for bearing failure offering the bolted 463 connection a level of damage tolerance. 464

465 Overall, we find that the predicted delamination cracks are fairly repeatable and overall are 466 representing the visual observations of bearing surface cracking made by Matharu (2014). The 467 numerical results presented in Figs. 4 to 6 demonstrate the ability of the FE modelling methodology 468 to appropriately predict the pin-bearing mode of failure where interaction between intralaminar and 469 interlaminar modes is strongly coupled.

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# 472 Changing the Bearing Strength for Bolted Connections

It is well-known that by increasing the size of the clearance hole the bearing strength decreases 473 (Yuan et al. 1996). Figs. 7 and 8 are equivalent to Figs. 4 and 6. In Fig. 7 there are two further load-474 displacement results for clearance holes of 0 mm (tight fitting) and 1.6 mm (without fabrication 475 tolerance (Anonymous 2012)). Although there is a gain in delamination initiation and maximum 476 477 loads on reducing clearance from 2.6 to 1.6 mm, there is a significant increase in their values, at 48% (M12) and 39% (M20), for the no clearance situation. Even the change in load on having the 478 479 clearance equal to the nominal design guideline of 1.6 mm (Creative Pultrusions 2016; Strongwell 480 2016) justifies why testing for the characteristic strength (Mottram and Zafari 2011; Matharu 2014) has to be with the maximum practical clearance. Illustrated in Fig. 8 are the delamination zones for 481 no clearance and the maximum loads of 25 kN for M12 (Fig. 8(a)) and of 37 kN for M20 (Fig. 482 8(b)). By inspection it can be seen that the shape of the delamination cracks at interfaces I1 and I2 483 are broadly similar to those in Figs. 6(a) and 6(b) when the clearance hole is 2.6 mm. Because the 484 tight fitting situation generates pressure over the whole semi-circular notch there is interfacial 485 failure in the laminate beyond the boundary of the hole. This finding shows that the test method 486 applied by Matharu (2014) will not give the actual pin-bearing strength if it is practical on- or off-487 488 site to assemble a bolted lap-joint with no hole clearance. By applying our FE modelling approach it is feasible to establish the clearance size at which the delamination growth at peak load first goes 489 beyond the confines of the bolt hole perimeter. 490

Keeping all other parameters constant to those in Matharu's testing, the next change to be evaluated by a sensitivity analysis is the relative thicknesses of the UD and TSFM layers (for a constant specimen thickness (*t*) of 9.6 mm). For both bolt diameters Table 4 reports the numerical results of: elastic stiffness ( $S_{el}$ ); pin-bearing load for initiation of delamination ( $R_{br,ID}$ ); maximum pin-bearing load ( $R_{br,max}$ ); the two load ratios of  $R_{br,ID}/R_{br,max}$  and  $R_{br,max}/R_{br,max,baseline}$ , where  $R_{br,max,baseline}$  is the maximum pin-bearing load for the original material (i.e. 23 and 36 kN for M12 and M20, respectively). The original material's predictions are presented in the first row of 498numbers, with three parametric variations given in the next three rows. The percentage of TSTM in499t is 58% in the original material and ranges from 42 to 67% in the three sensitivity analysis500materials. As expected,  $S_{el}$  increases as the UD percentage increases. Combining the M12 and M20501pin results it is found that neither  $R_{br,ID}$  nor  $R_{br,max}$  show any definite trends.  $R_{br,max}$  is seen to remain502virtually constant at 23-24 kN (M12) or 35-36 kN (M20). These observations are highlighted by the503virtually constant load ratios for the two bolt sizes.

504 The next study investigates the effect of changing the thickness of the mid-thickness TSFM. One reason for this parametric study is that the maximum load is governed by delamination failure 505 at interface I1, not I2 that is for the outer interfaces of the mid-thickness mat layer. The presentation 506 507 of information and computed predictions in Table 5 is the same as in Table 4. The parameter change is given in the second column where the thickness of the mid-thickness TSFM reduces in four 508 increments to  $0t_{\text{orig}}$  from its original specification of 2.8 mm (1.0 $t_{\text{orig}}$ ). The loss in the thickness of 509 the middle TSFM is replaced by UD reinforcement, shared equally by the two equal-thickness UD 510 layers. One observation is that the material with  $0t_{\text{orig}}$  and the M12 pin has the highest  $R_{\text{br,ID}}$  at 21 511 512 kN. With the M20 bolt size the highest  $R_{br,ID}$  of 33 kN is for  $0.15t_{orig}$ . There is, again, not a significant change in R<sub>br,max</sub> with the parametric variation, which for M12 is either 23 or 25 kN, and 513 for M20 is 36, 37 or 38 kN. 514

515 Note that the delamination contours for the specimens covered in Tables 4 and 5 were found to 516 be similar to those shown in Fig. 6. They are not reported in this paper because of lack of space.

Returning to the original study parameters (Matharu 2014), the next variable to be examined is reversing, in the laminate, the stacking sequence of layer TSFM1 or layer TSFM2 or of both. In the original material interfaces I1 and I2 had UD fibres on one side and +45° fibres from TSFM on the other side. By reversing the TSFM layer the interface has UD and chopped strand fibres in contact. Information in Table 6 follows Tables 4 and 5 with the stacking sequence changes given in the second column. The results are difficult to interpret with little change in predictions with the M12 pin and a significant increase to 1.16 in ratio  $R_{br,max}/R_{br,max,baseline}$  when the larger M20 pin is employed.

In the 1990s the wide flange shapes (Mottram and Zafari 2011) from Creative Pultrusions Inc. 525 (2016) did not have a triaxial mat reinforcement. Prior to introducing the Pultrex® SuperStructurals 526 product range the mat reinforcement was an CSM, having a local glass fibre volume fraction of 527 ≅23%. For the next study the material has either TSFM or CSM for the mat reinforcement. For 528 these simulations the model did not have the mid-thickness mat reinforcement. Predictions are 529 presented in Figs. 9 and 10 in the usual way. In the plots in Fig. 9 the load-displacement response 530 for the original laminate (Matharu 2014) is given by the shaded filled circle symbols. Superimposed 531 on this curve are the predicted results when the TSFM is fully replaced with CSM. The higher pair 532 of curves in Figs. 9(a) and 9(b) is for the situation where the mid-thickness mat is replaced by UD 533 reinforcement, thereby increasing  $S_{el}$ . A difference observed is that the maximum load is slightly 534 535 higher with the 'older' mat reinforcement and it occurs at a higher axial displacement. The positions of the open unfilled circle for the load at delamination initiation show that another difference on 536 having CSM reinforcement is that failure is delayed by about 2 kN (M12) or 4 kN (M20). A key 537 finding is that the maximum loads are the same at 25 kN (M12) and only slightly different at 38 and 538 40 kN (M20) after replacing the TSFM with CSM. An important conclusion from this study is that 539 the presence of  $\pm 45^{\circ}$  fibres in the TSFM layers has not increased the pin-bearing strength. 540 What is observed from looking at the contour plots in Figs. 10(a) and 10(b) for delamination 541 growth at maximum load is that the shape of the fractured zone is much more centrally located and 542

goes further into the specimen when the mat reinforcement is CSM. The area associated with
partially damaged material is seen to be larger too. These figures show that the size and shape of a
delamination crack will depend on the orientations of the fibres in the two layers adjacent to the
failed interface.

547 The final two FE simulations were performed to find out if pin-bearing strength increases when 548 the chopped strand backing in the TSFM layer is replaced by proportional increases in the 90° and 549  $\pm 45^{\circ}$  fibre reinforcements. The rationale for this parametric study is that the chopped strand layer has a relatively low fibre volume fraction at 16% and occupies 39% of the TSFM's 1.4 mm 550 thickness. The predicted results are reported in Figs. 11 and 12. As observed from the two curves in 551 Fig.11 there is little difference in the response up to when delamination failure initiates. Perhaps, 552 unexpected it is found that the original TSFM with the chopped strand backing has the higher 553 maximum load for both M12 (i.e. 23 and 18 kN) and M20 pins (i.e. 35 and 29 kN). Comparing the 554 equivalent contour plots in Fig. 12 with those presented in Fig. 6 it is seen that shape and size is 555 dependent on the fibre reinforcement local to the failing interface. 556

Results for this, and other parametric studies reported in this paper have shown that pin-bearing strength can be highest when the mat reinforcement has random continuous or chopped fibres at the relatively low volume fraction of < 25%.

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## 562 **Concluding Remarks**

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A FE modelling methodology using Abaqus software, without customized user-subroutines, has 564 been formulated that can be employed in parametric and/or sensitivity analyses to determine the 565 566 pin-bearing strength of laminated composite materials. Numerical outputs are able to predict the initiation and growth of delamination failures that governs what a pin-bearing strength is. Using 567 batch test results for a 9.6 mm thick web from a pultruded structural shape the simulation results for 568 the two pin diameters of 12 and 20 mm are shown to give good strength predictions. To assist 569 designers of bolted connections an interesting finding is that initiation of delamination cracking is 570 571 not at the peak load, and that this predicted load is close to, yet below the lowest pin-bearing strength measured from a batch of 10 specimens. There was, however, no observable evidence that 572 there was stable delamination failure(s) during testing before the pin-bearing strength load was 573 reached. Although this finding shows that the computational solution differs from the actual pin-574

bearing response an important finding is that the FE modelling methodology can reliably establishthe ultimate failure load for pin-bearing failure.

To demonstrate the potential of applying the modelling approach to investigate how the pin-577 578 bearing strength might vary with parameter changes a number of parametric studies are presented. While maintaining constant the other parameters in the test series, numerical predictions are 579 580 reported for changing the: clearance hole size; relative thickness of the unidirectional and triaxial mat layers; thickness of the mid-thickness triaxial mat layer; stacking sequence of the triaxial mat 581 layer; triaxial mat to a continuous strand mat; triaxial mat's construction so that the 39% thickness 582 of chopped strand backing is replaced proportionally with reinforcement having the orientations of 583  $\pm 45^{\circ}$  and  $90^{\circ}$ . 584

The main findings from the numerical results can be summarized as follows. New 585 understanding is obtained from contour plots for the Abaqus output Stiffness DEGradation showing 586 the size and shape of delamination failure(s) over interfaces. As expected, pin-bearing strength is 587 found to increase with reduced hole clearance and is significantly higher (at 48% and 39% for the 588 two pin diameters) when there is no clearance. Under the tight fitting condition it is observed that 589 the test method (Matharu 2014) becomes invalid because, at peak load, delamination failure is not 590 contained within the semi-circular hole. A less expected finding is that the strength is not sensitive 591 592 to the relative thicknesses of the unidirectional and mat layers, and there is no benefit on having 593 replaced the continuous strand mat with a triaxial mat (which is more expensive). Furthermore, tabulated results from parametric studies do not offer obvious trends, and the authors cannot give 594 any scientific explanation to why computed results are seen to be highly dependent on the pin 595 596 diameter being either 12 or 20 mm. The final study shows that strength is reduced on replacing the chopped strand backing in the triaxial mat layers; the opposite might be expected since the 597 598 additional  $\pm 45^{\circ}$  fibres ought to be effective in resisting a higher bearing load.

599 The satisfactory performance of the computational predictions in determining the response at 600 peak load encourages the authors to recommend that the Abaqus modelling methodology may be

- used to: (i) design the laminate (produced by the pultrusion composite processing method) for a 601
- specified pin-bearing strength; (ii) predict the pin-bearing strengths for bolted connections having 602 parameters to be scoped in a structural design standard.

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- **Fig. 1.** Modes of delamination failure: a) Mode I (m = I): opening; b) Mode II (m = II): sliding; c)
- 707 Mode III (m = III): tearing.
- **Fig. 2.** Geometry: schematic of plate and pin with applied load and composite lay-up showing
- 709 locations of four interfaces I1 to I4.
- **Fig. 3.** Typical plate mesh for the evaluation of the (12 mm) pin-bearing behaviour of a pultruded
- specimen, with the symmetry plane facing backwards.
- 712 Fig. 4. Load-displacement plots from implementing the FE model, and comparison with
- experimental data from Matharu (2014): a) M12; b) M20.
- **Fig. 5.** Specimen deformation at maximum predicted load (plots on deformed structure,
- magnification factor 2.5): a) M12; b) M20.
- **Fig. 6.** Delamination at interfaces I1 (left) and I2 (right) at maximum predicted load (contour
- plotting on deformed mesh): a) M12 at 23kN; b) M20 at 36 kN.
- **Fig. 7.** Effect of changing the size of the hole clearance: a) M12; b) M20.
- **Fig. 8.** Delamination at interfaces I1 (left) and I2 (right) at maximum load (plots on deformed
- structure) when specimen has no clearance hole: a) M12 has maximum load of 25 kN; b) M20 has
- 721 maximum load of 37 kN.
- **Fig. 9.** Plots to maximum load having mat reinforcement TSFM replaced with CSM (filled markers
- TSFM; unfilled markers CSM): a) M12; b) M20.
- **Fig. 10.** Delamination growth at interface I1 at maximum load for mat reinforcement of TSFM
- (left-side) or of CSM (right-side): a) M12 has maximum load of 25 kN | 25 kN); b) M20 has
- 726 maximum load of  $38 \text{ kN} \mid 40 \text{ kN}$ )

- **Fig. 11.** Comparison of original material with a new material when the chopped strand layer in the
- 728 TSFM is replaced by the continuous fibre reinforcements (filled markers TSFM; unfilled markers
- 729 TSFM without chopped strand layer): a) M12; b) M20.
- **Fig. 12.** Delamination at interfaces I1 (left) and I2 (right) at maximum load when chopped strand
- ration layer in the TSFM is not present (plots on deformed structure): a) M12 has maximum load of 18
- 732 kN; b) M20 has maximum load of 28 kN

















# **Table 1.** Material constituent properties for the Pultrex® Superstructural material of 9.6 mm

## 787 thickness.

		Constituent pa	art	
	SV	TSFM	UD	Matrix
Number of layers in thickness	2	4	2	-
Fibre architecture	Random fibre veil	+45°/90°/-45°/CSM	156 (56 yield)	-
% volume fraction	3	34	33	30
% total of fibre reinforcement	5	48	47	-
Nominal layer thickness, $t_{lay}$ (mm)	0.03	1.4	2.0	_

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792	Table 2. Lamina	properties for the	laminate defined i	in Fig. 2.
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Elastic lamin	na proper	ties (kN/mm <sup>2</sup> )		Laminate s	trength	Fracture energy			
	Const	ituent part		properties (	(N/mm)	J/mm)			
	UD	TSFM							
		Mat layer (+45°/90°/-45°)	CSM						
$E_1$	44.7	13.1	7.7	<i>f</i> <sub>1,T</sub>	294	$G_{1,\mathrm{T,c}}$	100		
$E_2 = E_3$	14.0	25.5	7.7	$f_{1,\mathrm{C}}$	306	$G_{1,\mathrm{C},\mathrm{c}}$	100		
$G_{12} = G_{13}$	4.1	6.1	2.7	$f_{2,\mathrm{T}}$	84	$G_{2,\mathrm{T,c}}$	1.2		
<i>V</i> <sub>12</sub>	0.28	0.41	0.41	<i>f</i> 2,C	133	$G_{2,\mathrm{C},\mathrm{c}}$	5		
				$f_{1,S} = f_{2,S}$	91				

Interfacial properties	
$K_{\rm I} = K_{\rm II} = K_{\rm III} (\rm N/mm^3)$	10 <sup>6</sup>
$f_{\rm I}$ (N/mm <sup>2</sup> )	21 <sup>1</sup>
$f_{\rm II} = f_{\rm III} ({\rm N/mm^2})$	23 <sup>1</sup>
G <sub>I,c</sub> (N/mm)	0.2
$G_{\rm II,c} = G_{\rm III,c} (\rm N/mm)$	0.5

**Table 3.** Interfacial properties for the pultruded FRP laminate defined in Fig. 2.

Note. These values correspond to a weakening factor of 0.25.

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# **Table 4.** Modelling results: relative thickness of layers.

		M12					M20				
9.6 mm	ન	S <sub>el</sub> (kN/mm)	R <sub>br,ID</sub> (kN)	R <sub>br,max</sub> (kN)	$R_{ m br,ID}/R_{ m br,max}$	$R_{ m br,max}/R_{ m br,max,baseline}$	Sel (kN/mm)	R <sub>br,ID</sub> (kN)	R <sub>br,max</sub> (kN)	R <sub>br,ID</sub> /R <sub>br,max</sub>	$R_{ m br,max}/R_{ m br,max,baseline}$
Original material											
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$	58) 42)	59	17	23	0.75	1.00	82	27	36	0.75	1.00
Parametric variation											
$\begin{array}{c cccc} t_{\text{TSFM}} \ (\text{mm}) \ (\% \ \text{of} \ t) & 1.0 \ (\% \ \text{of} \ t) & 2.8 \ (\% \ \text{of} \ t) & 0 \ (\ \$	42) 58)	69	19	23	0.82	1.02	95	31	35	0.89	1.00
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$	50) 50)	65	15	23	0.63	1.01	74	25	35	0.73	1.00
$\begin{array}{c c} t_{\text{TSFM}} \ (\text{mm}) \ (\% \ \text{of} \ t) & 1.6 \ (mm) \ t_{\text{UD}} \ (\text{mm}) \ (\% \ \text{of} \ t) & 1.6 \ (mm) \ t_{\text{UD}} \ (\text{mm}) \ (\% \ \text{of} \ t) & 1.6 \ (mm) \ t_{\text{UD}} \ (\text{mm}) \ (\% \ \text{of} \ t) & 1.6 \ (mm) \ t_{\text{UD}} \ (mm) \ (\% \ \text{of} \ t) & 1.6 \ (mm) \ t_{\text{UD}} \ (mm) \ (\% \ \text{of} \ t) & 1.6 \ (mm) \ t_{\text{UD}} \ (mm) \ (\% \ \text{of} \ t) & 1.6 \ (mm) \ t_{\text{UD}} \ (mm) \ (mm) \ (\% \ \text{of} \ t) & 1.6 \ (mm) \ t_{\text{UD}} \ (mm) \ (mm) \ (mm) \ (mm) \ t_{\text{UD}} \ (mm) \$	67) 33)	58	20	24	0.87	1.05	88	28	35	0.81	1.01

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# **Table 5.** Modelling results: thickness of internal TSFM layer.

i	[SFM1 2 ↓ I.4 mm	M12					M20				
9.6 mm		S <sub>el</sub> (kN/mm)	R <sub>br,ID</sub> (kN)	R <sub>br,max</sub> (kN)	$R_{ m br,ID}/R_{ m br,max}$	$R_{ m br,max}/R_{ m br,max,baseline}$	S <sub>el</sub> (kN/mm)	R <sub>br,ID</sub> (kN)	R <sub>br,max</sub> (kN)	$R_{\rm br,ID}/R_{\rm br,max}$	$R_{ m br,max}/R_{ m br,max,baseline}$
Original material	Original material										
$t_{\text{TSFM,inner}} (\text{mm})$ $t_{\text{UD}} (\text{mm})$	(1.0 <i>t</i> <sub>orig</sub> ) 1.4 2.0	59	17	23	0.75	1.00	82	27	36	0.75	1.00
Parametric variation	ļ										
<i>t</i> <sub>TSFM,inner</sub> (mm) <i>t</i> <sub>UD</sub> (mm)	(0.5 <i>t</i> <sub>orig</sub> ) 0.7 2.7	72	19	23	0.83	0.99	94	30	37	0.82	1.02
<i>t</i> <sub>TSFM,inner</sub> (mm) <i>t</i> <sub>UD</sub> (mm)	(0.25 <i>t</i> <sub>orig</sub> ) 0.35 3.05	77	20	25	0.80	1.11	102	31	37	0.83	1.03
$t_{\text{TSFM,inner}}$ (mm) $t_{\text{UD}}$ (mm)	(0.15 <i>t</i> <sub>orig</sub> ) 0.21 3.19	80	17	25	0.68	1.10	99	33	36	0.92	1.01
$t_{\text{TSFM,inner}}$ (mm) $t_{\text{UD}}$ (mm)	(0t <sub>orig</sub> ) 0 3.4	75	21	25	0.83	1.08	105	32	38	0.84	1.06

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# **Table 6.** Modelling results: stacking sequence TSFM layer.

	45 <sup>∞</sup> \$0.25 mm		M12					M20				
1.4 mm	0.35 mm 0.25 mm 0.55 mm	S <sub>el</sub> (kN/mm)	R <sub>br,ID</sub> (kN)	R <sub>br,max</sub> (kN)	R <sub>br,ID</sub> /R <sub>br,max</sub>	$R_{\rm br,max}/R_{\rm br,max,baseline}$	S <sub>el</sub> (kN/mm)	R <sub>br,ID</sub> (kN)	R <sub>br,max</sub> (kN)	R <sub>br,ID</sub> /R <sub>br,max</sub>	R <sub>br,max</sub> /R <sub>br,max,baseline</sub>	
Original mate	Original material											
<i>t</i> <sub>TSFM,1</sub> <i>t</i> <sub>TSFM,2</sub>	+45°/90°/-45°/CSM +45°/90°/-45°/CSM	59	17	23	0.75	1.00	82	27	36	0.75	1.00	
Variation												
t <sub>TSFM,1</sub> t <sub>TSFM,2</sub>	+45°/90°/-45°/CSM CSM /-45°/90°/45°	59	19	24	0.77	1.05	82	27	35	0.79	0.96	
<i>t</i> TSFM,1 <i>t</i> TSFM,2	CSM /-45°/90°/45° +45°/90°/-45°/CSM	59	20	23	0.87	1.01	82	33	37	0.88	1.08	
t <sub>TSFM,1</sub> t <sub>TSFM,2</sub>	CSM /-45°/90°/45° CSM /-45°/90°/45°	59	20	23	0.88	1.01	82	33	40	0.82	1.16	

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