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Title: Multiscale Simulations of Alloy Phase Stability

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## MULTISCALE SIMULATIONS OF ALLOY PHASE STABILITY

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### Abstract

First principles, atomic scale and continuum level models are combined to predict thermodynamic properties of alloys and stability of phases. Many-body interactions, as well as vacancies, defects, and non-stoichiometry are included in the modeling process and the structural stability of hypothetical phases is evaluated. The resulted thermodynamic functions and phase diagrams are integrated in a casting simulation computer program.

The process of relating microscopic modeling results to the macroscopic heat transfer and phase equilibrium calculations is detailed to emphasize the self-consistency of the approach and to identify the potential sources of errors. The sequence: data acquisition, modeling, prediction, experimental validation, is illustrated for several recent results in actinide based alloys.

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## MULTISCALE SIMULATIONS OF ALLOY PHASE STABILITY

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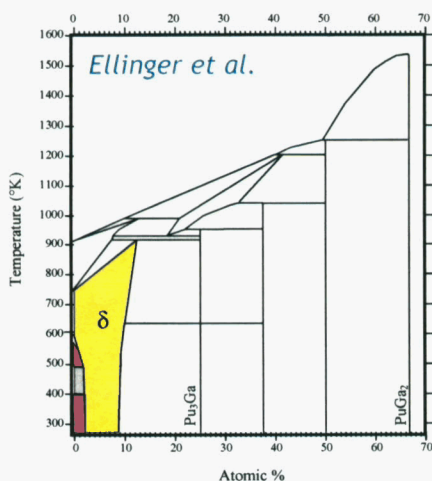
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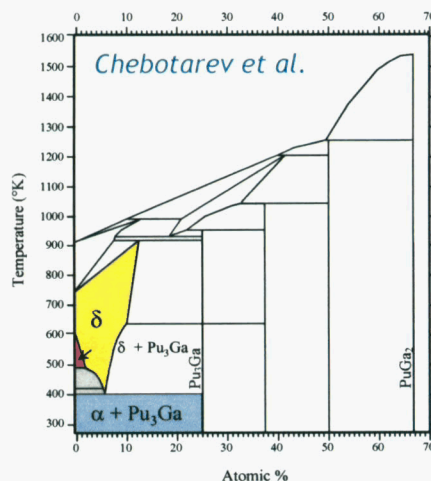
United States Department Of Energy



### Comparison of Portions of the Pu-Ga Phase Diagrams



\* F. H. Ellinger, C. C. Land, and V. O. Strebing, The Plutonium-Gallium System, *J. Nucl. Met.*, 12 (1964) p. 226.



\* N. T. Chebotarev, E. S. Smotriskaya, M. A. Adrianov, and O. E. Kostyuk, In Plutonium 1975 and Other Actinides, Proceedings of the Fifth International Conference on Plutonium and Other Actinides, Baden Baden, Germany, September 10-13, 1975, Edited by North-Holland Publishing Company, New York, 1976, p. 37-45.

\*\* L. T. Timofeeva, Low temperature equilibrium, aging under self-irradiation in binary alloys of Pu with elements of III-B group (to be published)



## A Good Plan

- Get thermodynamic data for perfect and imperfect structures from:
  - First principle calculations (Generalized Gradient Approximation)
  - Atomic-scale models (Modified Embedded Atom Method)
  - Molecular dynamics simulations (Cascade Models for defects)
  - Experimental measurements (Calorimetry, Dilatometry)
- Check the thermodynamic data sets for thermodynamic self-consistency.
- Evaluate uncertainty (Bayesian statistics, Genetic Algorithms, Markov Chain Monte Carlo).
- Calculate the equilibrium phase diagram (CALPHAD method).
- Incorporate the phase diagram into a casting simulation software.



## First Principles Calculations

- VASP
- Generalized Gradient Approximation
- Ultra-soft pseudopotentials or Projected Augmented Wave method
- Spin-polarized
- Monkhorst-Pack special k points
- Structural relaxations



## Results for PuGa<sub>3</sub>

- L1<sub>2</sub> structure
- a<sub>0</sub> = 4.2880 Angstrom
- Bulk Modulus = 71.78 Gpa
- B' = dB/dp = 4.8
- Magnetic moment = 5.4 Bohr magneton
- Ferromagnetic state is more stable



## Atomic Scale Modeling

### Embedded Atom Method (EAM)

$$E = \sum_i \left( F(\bar{\rho}_i) + \frac{1}{2} \sum_{j \neq i} \phi(R_{ij}) \right)$$

#### Embedding Function

$$F(\bar{\rho}) = AE_c \frac{\bar{\rho}}{\rho_0} \ln \frac{\bar{\rho}}{\rho_0}$$

#### Pair Potential

$$\phi(R) = \frac{2}{Z} \left\{ E^u(R) - F(\bar{\rho}^0(R)) \right\}$$

#### Background Electron Density

$$\bar{\rho} = \rho^{(0)} \sqrt{1 + \Gamma}$$

$$\Gamma = \sum_{l=1}^3 t^{(l)} \left( \rho^{(l)} / \rho^{(0)} \right)^2$$

angle dependent  
electron density (MEAM)

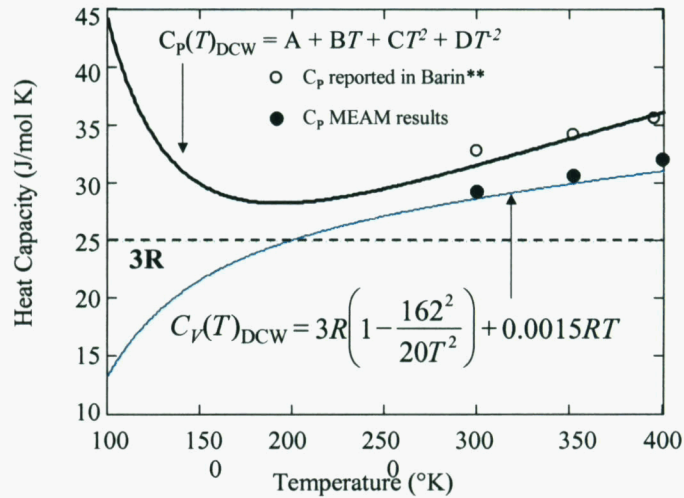
#### Universal EOS

$$E^u(R) = -E_c \left( 1 + a^* + \delta \alpha^* \frac{r_e}{R} \right) e^{-a^*}$$

$$a^* = \alpha \left( \frac{R}{r_e} - 1 \right) \quad \alpha^2 = \frac{9\Omega B}{E_c}$$



### C<sub>p</sub> of alpha-Pu (MEAM calculations)



\*D. C. Wallace, Electronic and phonon properties of six crystalline phases of Pu metal, Phys. Rev. B, 58, 1998, pp.14344-15439

\*\*I. Barin, Thermochemical data of pure substances, VCH, New York, 1993.



### Thermodynamic Modeling

$$C_p(T) = A + BT + CT^2 + DT^{-2}$$

$$H(T) = H_{298.15} + \int_{298.15}^T C_{p1}(\tau) d\tau + \Delta H_{T_1} + \int_{T_1}^T C_{p2}(\tau) d\tau$$

$$H(T) = H_0 + AT + B\frac{T^2}{2} + C\frac{T^3}{3} - DT^{-1}$$

$$S(T) = S_{298.15} + \int_{298.15}^T \frac{C_{p1}(\tau)}{\tau} d\tau + \frac{\Delta H_{T_1}}{T_1} + \int_{T_1}^T \frac{C_{p2}(\tau)}{\tau} d\tau$$

$$S(T) = S_0 + A \ln T + BT + C\frac{T^2}{2} - D\frac{T^{-2}}{2}$$

$$G(T) = H(T) - T \cdot S(T)$$

$$G(T) = G_1 + G_2T - AT \ln T - B\frac{T^2}{2} - C\frac{T^3}{6} - D\frac{T^{-1}}{2}$$

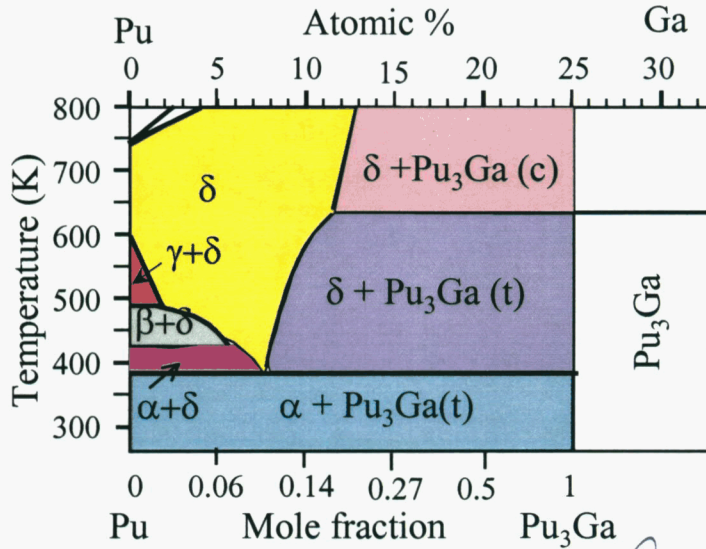
$$C_p(T) = -T \frac{\partial^2 G}{\partial T^2}$$

$$H(T) = \frac{\partial(G/T)}{\partial(1/T)}$$

$$S(T) = -\frac{\partial G}{\partial T}$$



Equivalent representations of the eutectoid point in the Pu-Ga and Pu-Pu<sub>3</sub>Ga systems



### A model consistent with the Pu-Ga eutectoid point

We propose a subregular model for the Gibbs free energy of the stabilized delta phase in the Pu-Pu<sub>3</sub>Ga pseudo-binary system (all values in kJ/mol). The model is intended to emphasize the possibility of the existence of an eutectoid point in the Pu-Ga phase diagram at:

$T^E = 370 \text{ K}$  and  $x^E = 7.9 \%$  in Pu-Ga

$x^E = 0.104$  in Pu-Pu<sub>3</sub>Ga

$$G_{\delta}(T, x) = (1-x)G_{\text{Pu}}(T) + xG_{\text{Pu}_3\text{Ga}}(T) + RT[(1-x)\ln(1-x) + x\ln x] + (1-x)x[(1-x)\Omega_{\text{Pu}} + x\Omega_{\text{Pu}_3\text{Ga}}]$$

$$G_{\text{Pu}_3\text{Ga}_{\text{cubic}}}(T) = -40 - (T - 298.15)0.185 \quad T^{\text{transf}} = 638 \text{ K}$$

$$G_{\text{Pu}_3\text{Ga}_{\text{tetragonal}}}(T) = -30 - (T - 298.15)0.215$$

When referenced to delta Pu:  $\Delta G_{\text{Pu}_3\text{Ga}}(T) = -72.4 - 0.0138T$

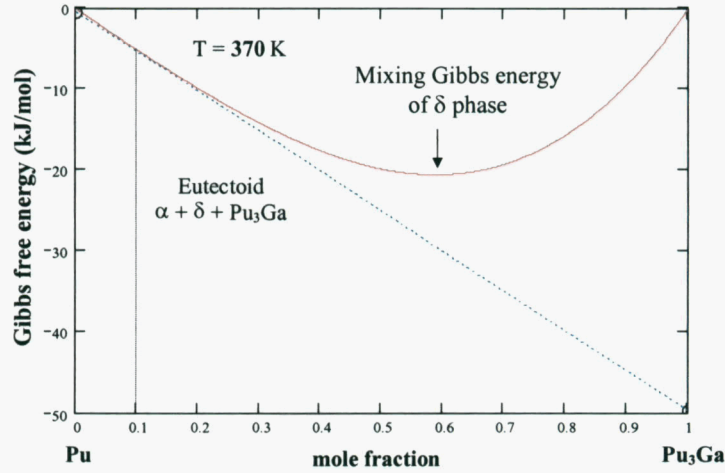
Interaction parameters:  $\Omega_{\text{Pu}} = -42$   $\Omega_{\text{Pu}_3\text{Ga}} = -102$





### Free Energies of Selected Phases at the Eutectoid Point

$$\Delta G_{Pu_3Ga}(T) = -72.4 - 0.0138T \quad \Omega_{Pu} = -42 \quad \Omega_{Pu_3Ga} = -102 \quad (\text{kJ/mole})$$



### Radiation Damage Modeling

#### Technique

- Empirical modified embedded atom potential for all phases of Pu (Baskes)
- Molecular dynamics simulations at constant volume and temperature
- Cells periodic in x,y,z
- Pure metal (no alloying)

#### Conditions (fcc Pu)

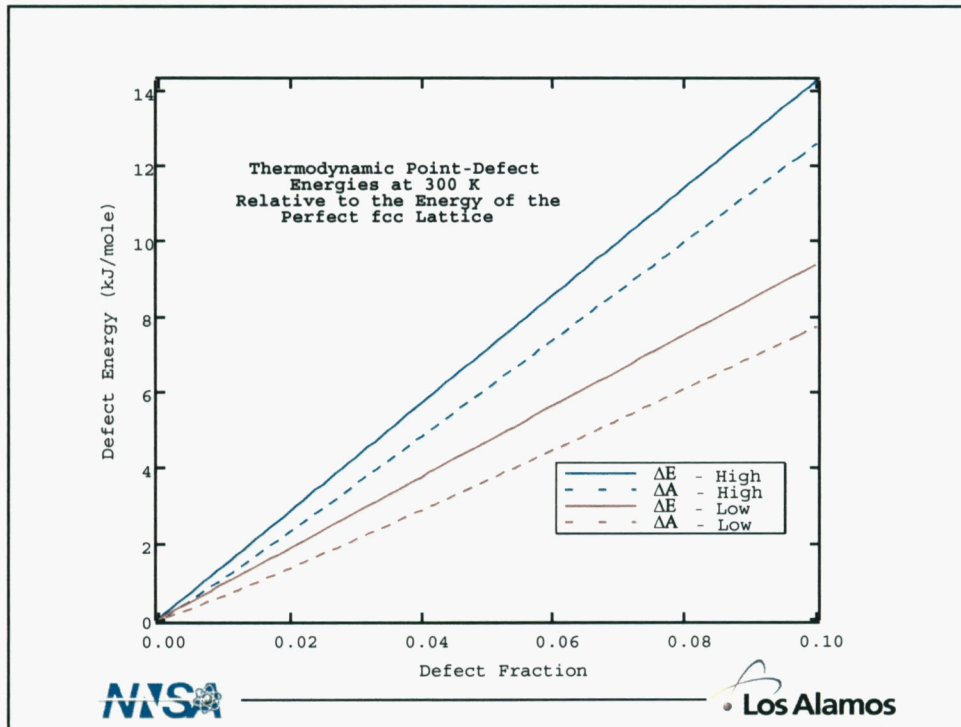
- 512 Pu atoms in relaxed, distorted,  $\delta$ -lattice (Sickafus)
- 300 K, constant volume to mimic alloy stabilization
- 20 eV Primary Knock-on Atom (PKA) energy; (121) direction

#### Results (fcc Pu)

- Estimate of displacement threshold energy: 8-10 eV
- Modified dumbbell defect structures suggested
- Nascent defect loops







## Truchas

**Complex code for simulation of casting  
and related phenomena. (ASCI project)**

Simulation capabilities:

- Mass transfer (flow)
- Heat transfer (conduction, convection, and radiation)
- Phase change (pure materials, alloys)
- Chemical reactions (curing)
- Thermo-mechanical (stress-strain)
- Electro-magnetic fields