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Ramp-Compression Experiments on Tantalum at the NIF and Omega Lasers

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July 7, 2011

JOWOG37

Aldermasten, United Kingdom

July 11, 2011 through July 15, 2011

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HEDP

high energy density physics



Ramp-Compression Experiments on Tantalum at the NIF and Omega Lasers

JOWOG 37, AWE

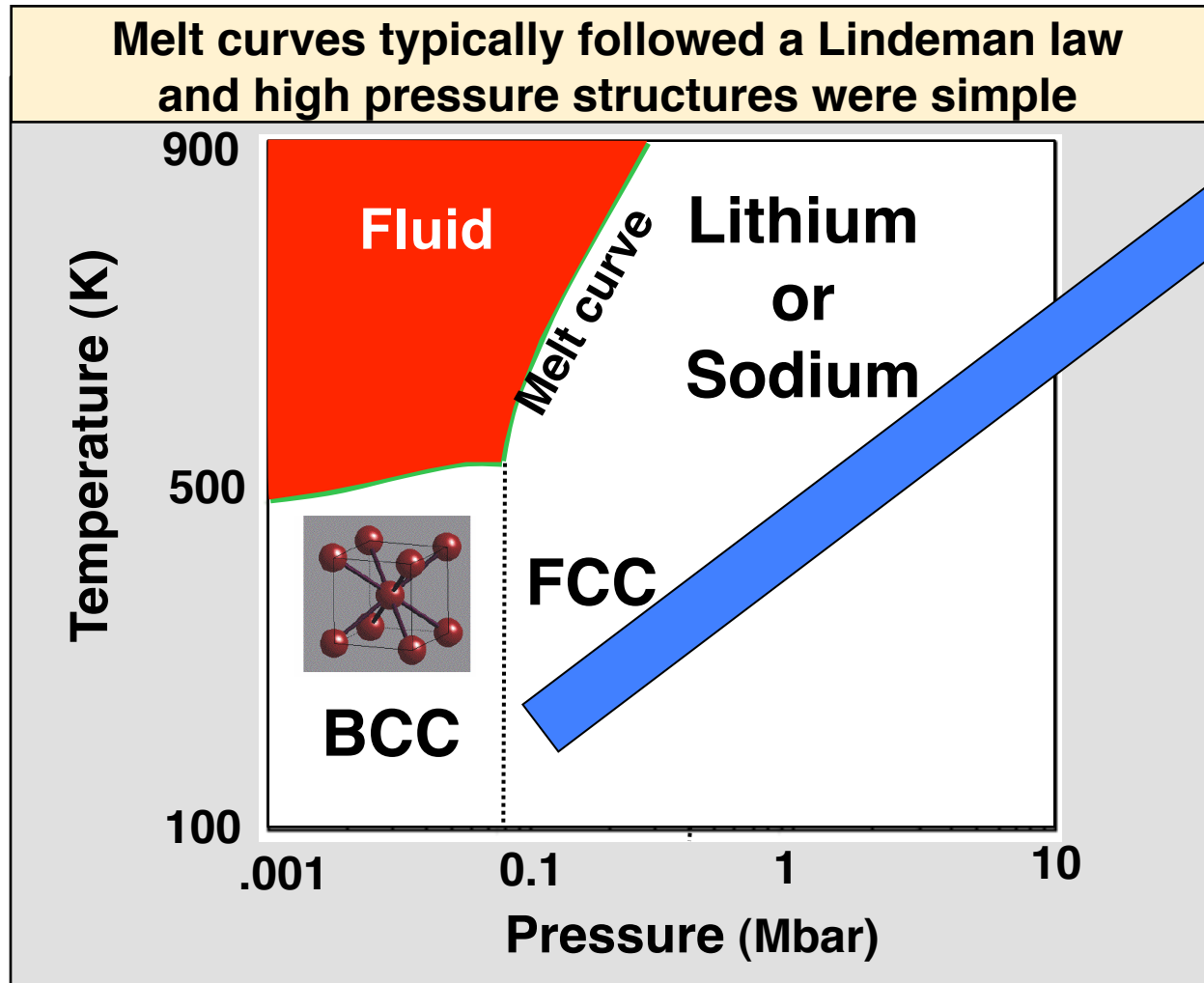
July 11, 2011

Jon Eggert,

**Ray Smith, Dave Braun, Reed Patterson,
Peter Celliers, Gilbert Collins, Ryan Rygg, Jim Hawreliak, Ted Perry**

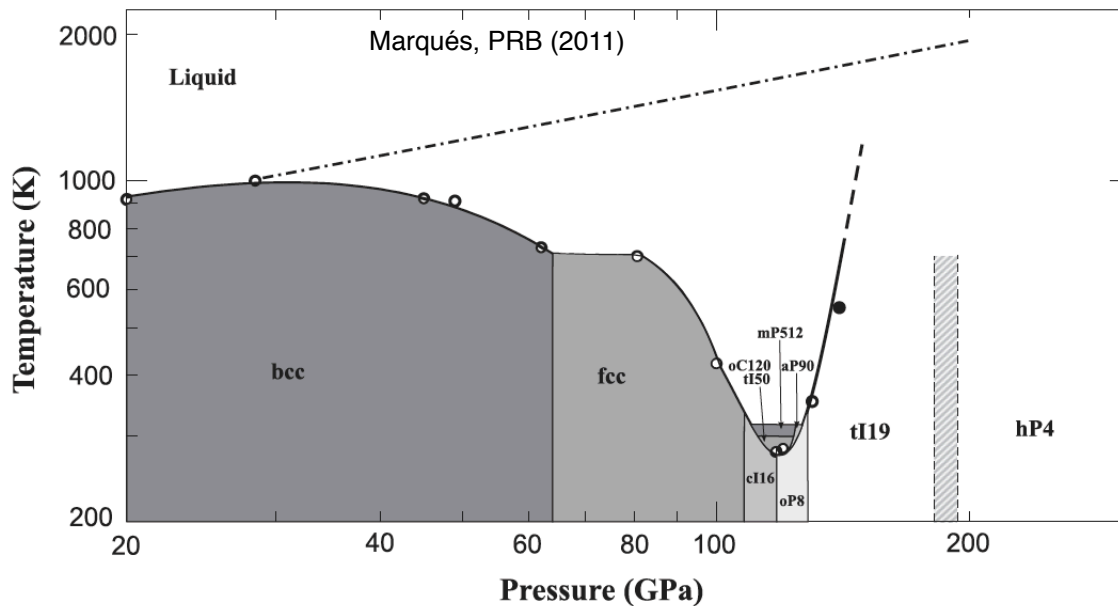
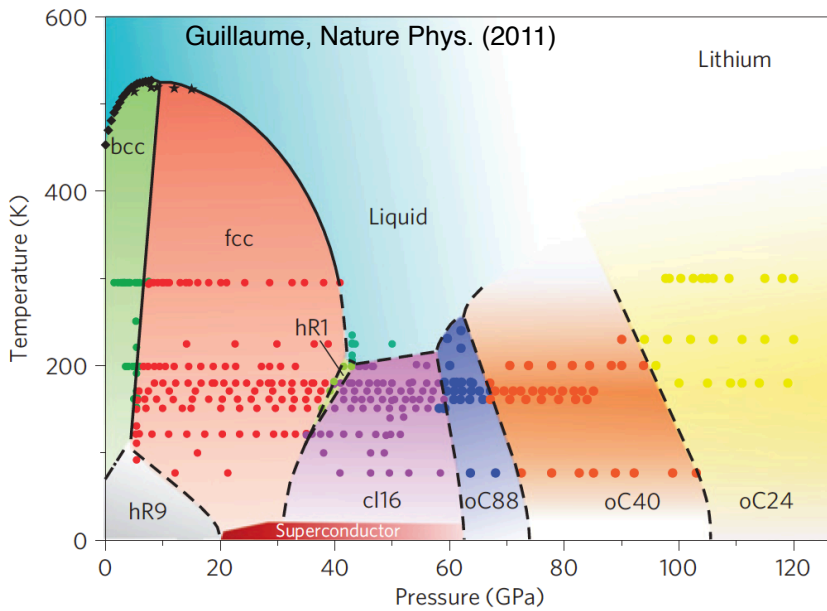
Lawrence Livermore National Laboratory • High Energy Density Physics

Just a few years ago, ultra-high pressure phase diagrams for materials were very simple



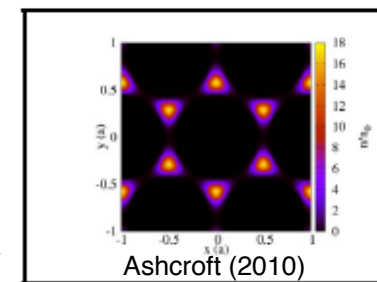
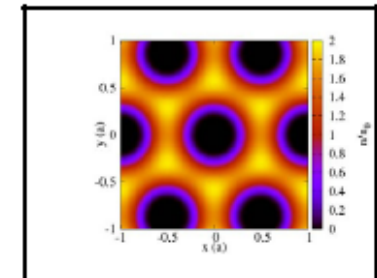
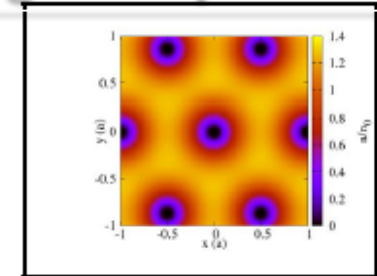
Physics Gets Simple!

A few recent observations and calculations suggest suggest very complex high-pressure behavior



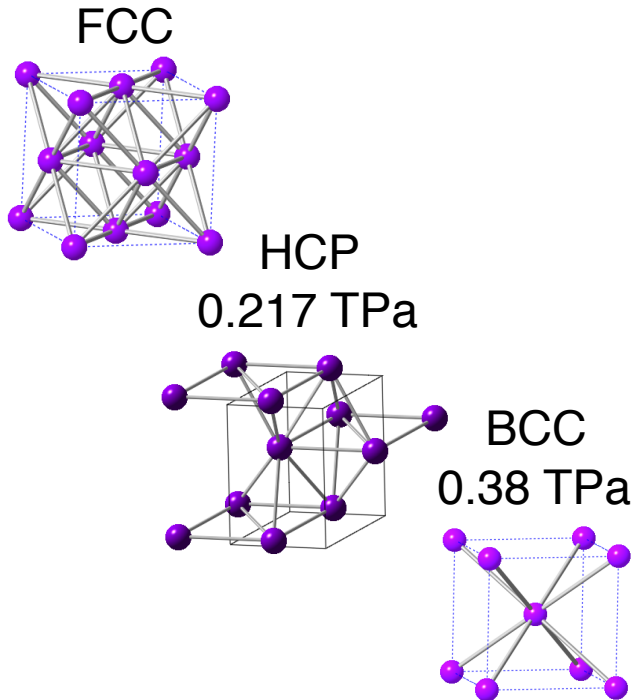
Simulations predict electronic localization through compression

Increasing Compression



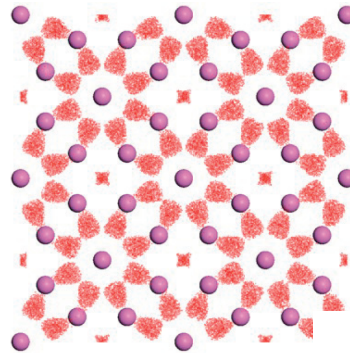
Driven by orthogonalization and Pauli Exclusion to interstitial regions

High pressures phases of aluminum are also predicted to be complex

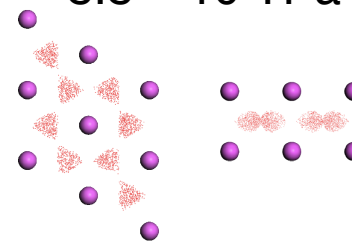


Pickard and Needs, Nature Materials (2010).

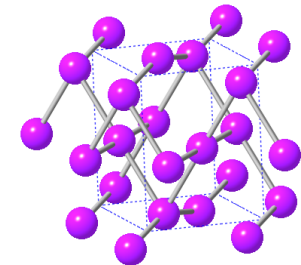
Host-Guest structure
of Ba-IVa
(Incommensurate
Electride)
3.2-8.8 TPa



Simple Hexagonal
Electride
8.8 – 10 TPa



CMMA
Electride
> 10 TPa



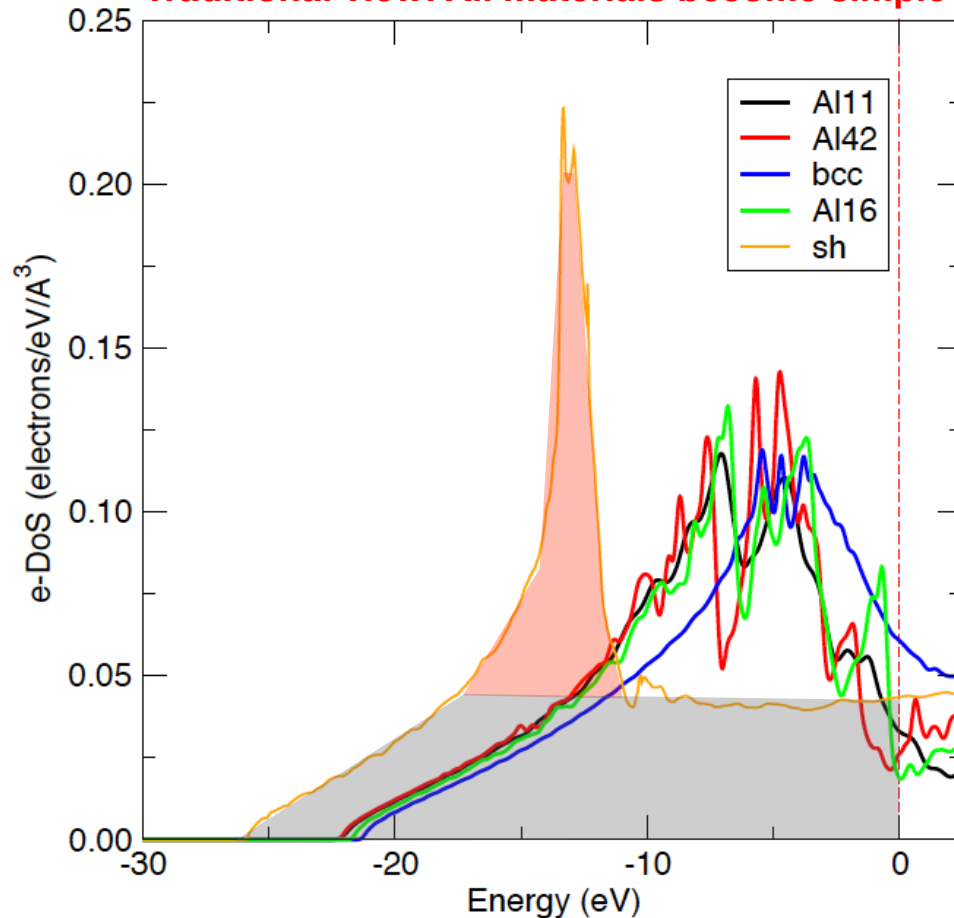
Electrides predicted to form above about 3.2 TPa

“all structures near 30 TPa are far from close packed”

The energies involved in these transitions are enormous!

Pickard and Needs, Nature Materials (2010).

Traditional view: All materials become simple at high pressure appears to be incorrect!



Up to 10 TPa aluminum is still metallic, but the electronic structure changes are enormous!

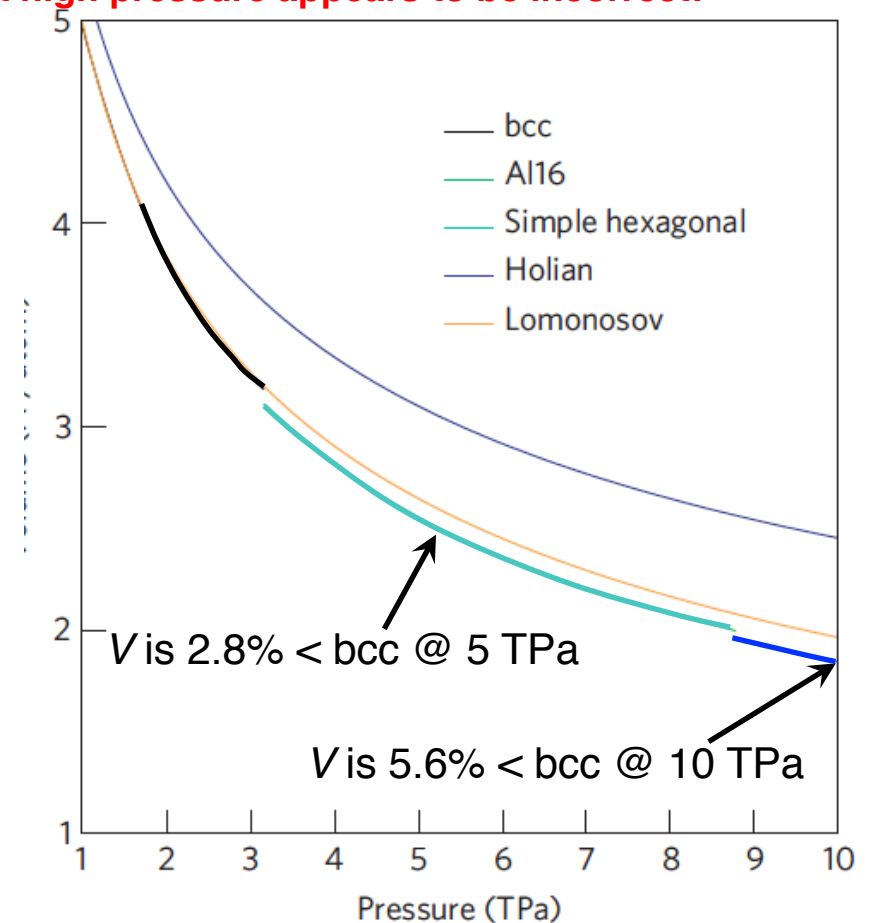
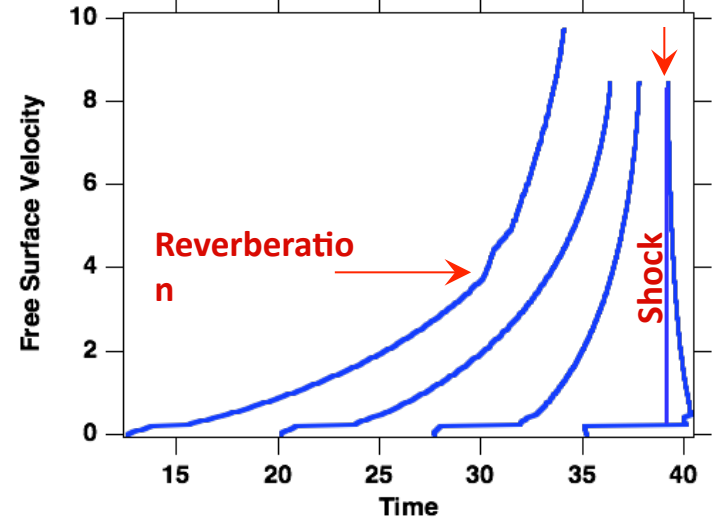
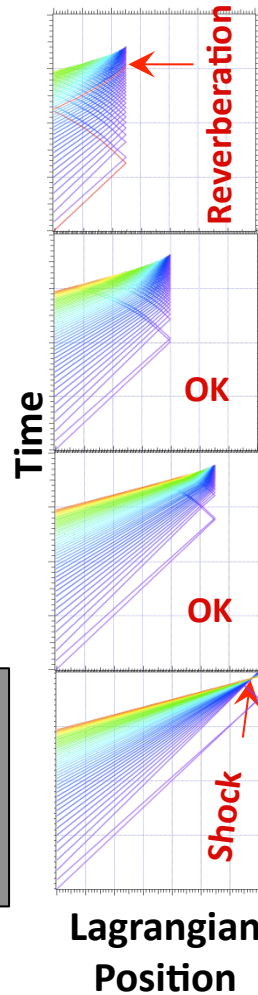
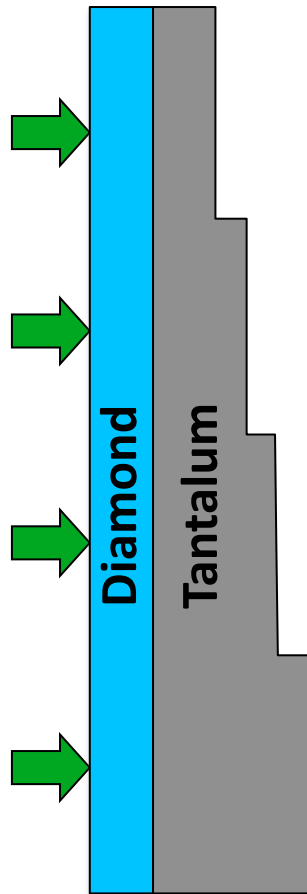
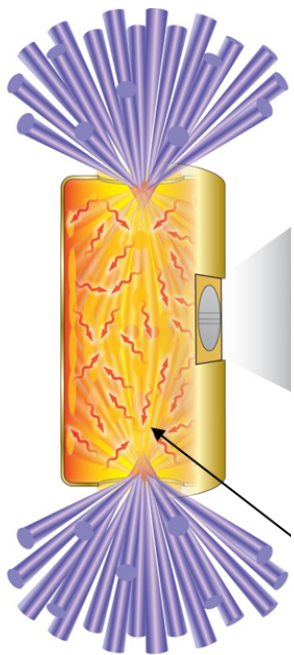
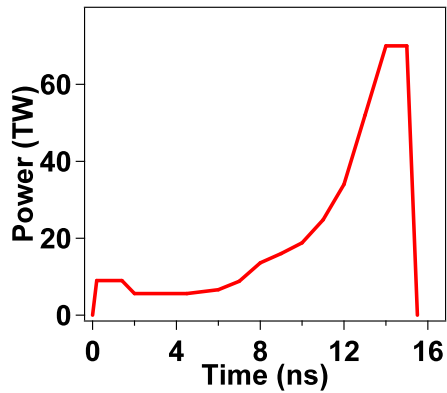


Figure 3 | Equation of state of Al. The volume-pressure relationship calculated in this work and from the EoSs of Lomonosov^{4,5} and Holian¹⁸.

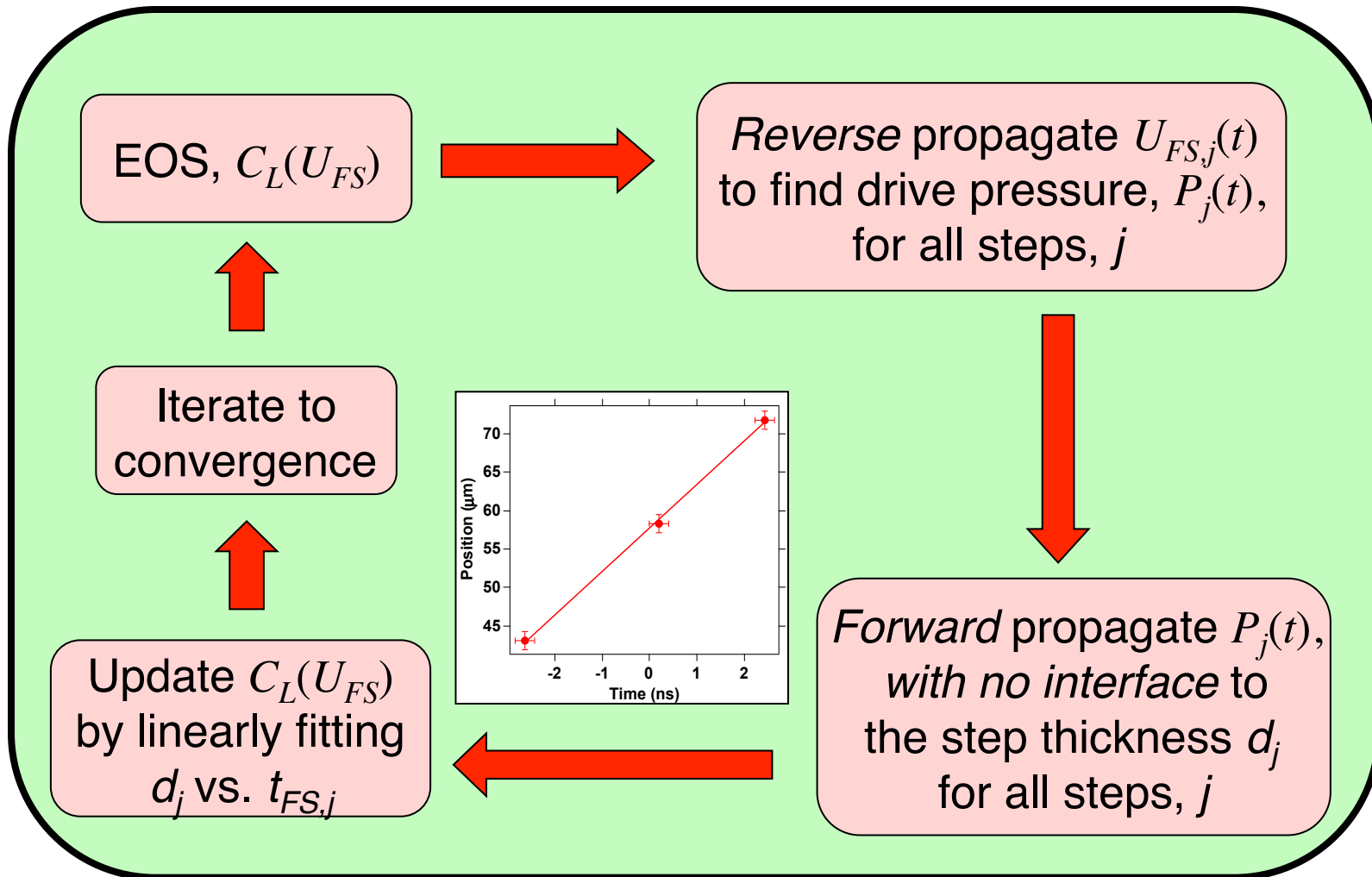
Laser-driven ramp-compression experiments

Under ramp loading, EOS (stress-density) can be determined from free-surface velocity measurements



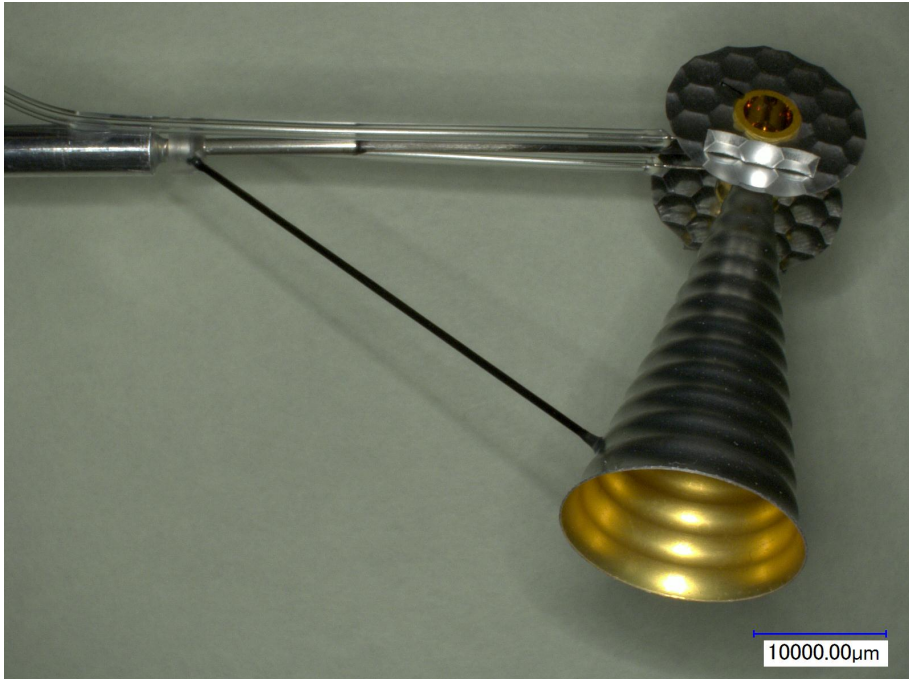
Design constraints:
No reverberation
and no shock

Iterative Analysis: Correction for free-surface wave interactions. Rothman, et al. J. Phys. D (2005)



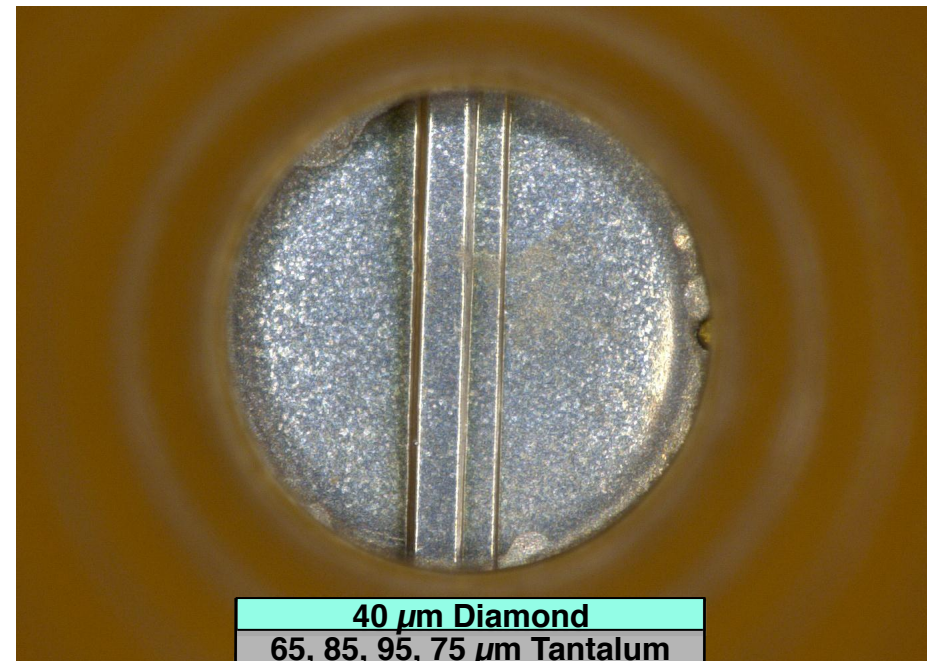
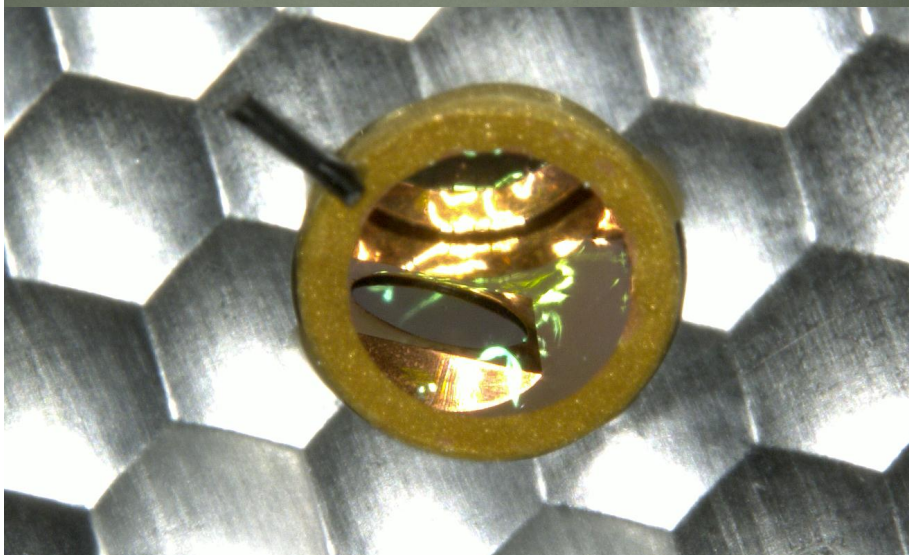
Method requires both reverse and forward propagation steps

NIF: Materials EOS Drive targets



Gas-filled, room-temperature, stepped target mounted on side of Hohlräum with VISAR cone.

- 0.2 atm. Neopentane gas fill
- $L = 11$ mm
- $\phi = 6$ mm
- $LEH = 4.5$ mm.

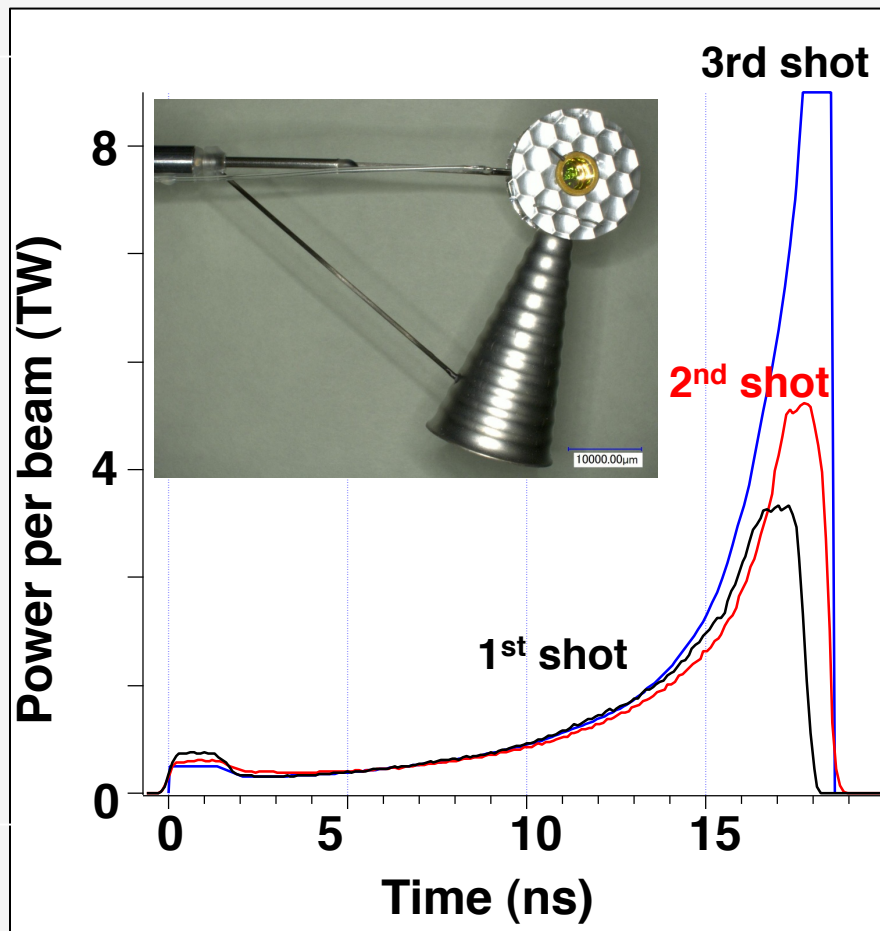


40 μm Diamond
65, 85, 95, 75 μm Tantalum

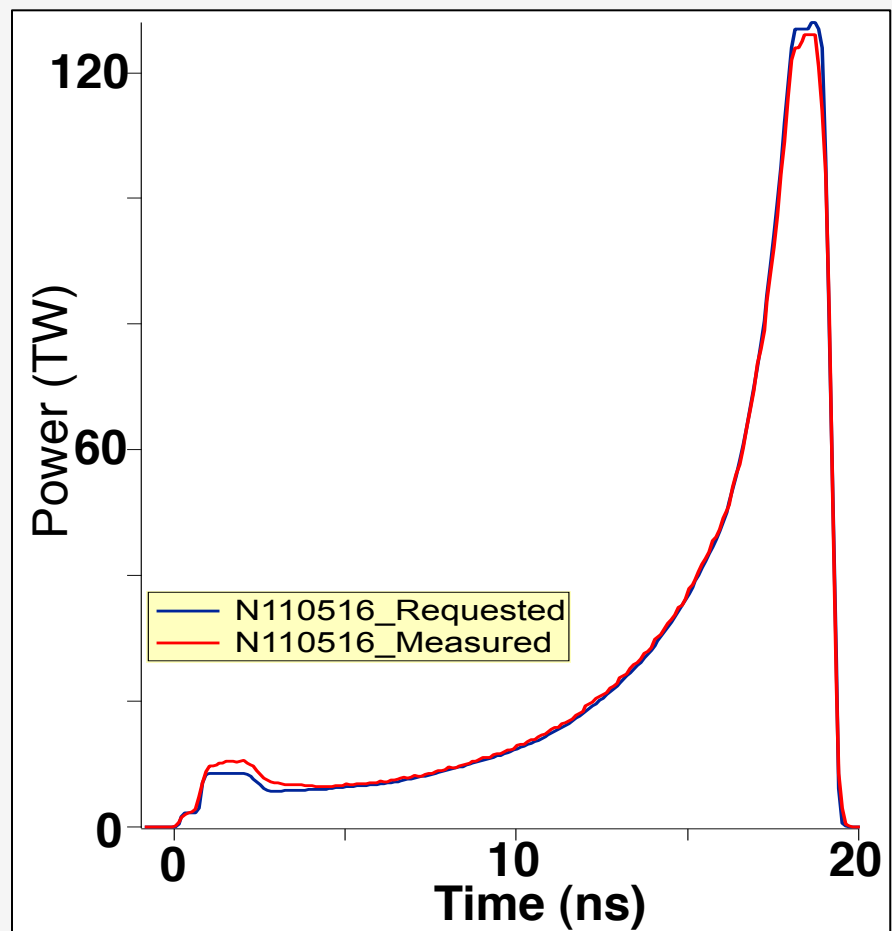
Nano-crystalline diamond EOS on NIF

In addition to the higher energy, pulse shaping was key to achieving higher pressure

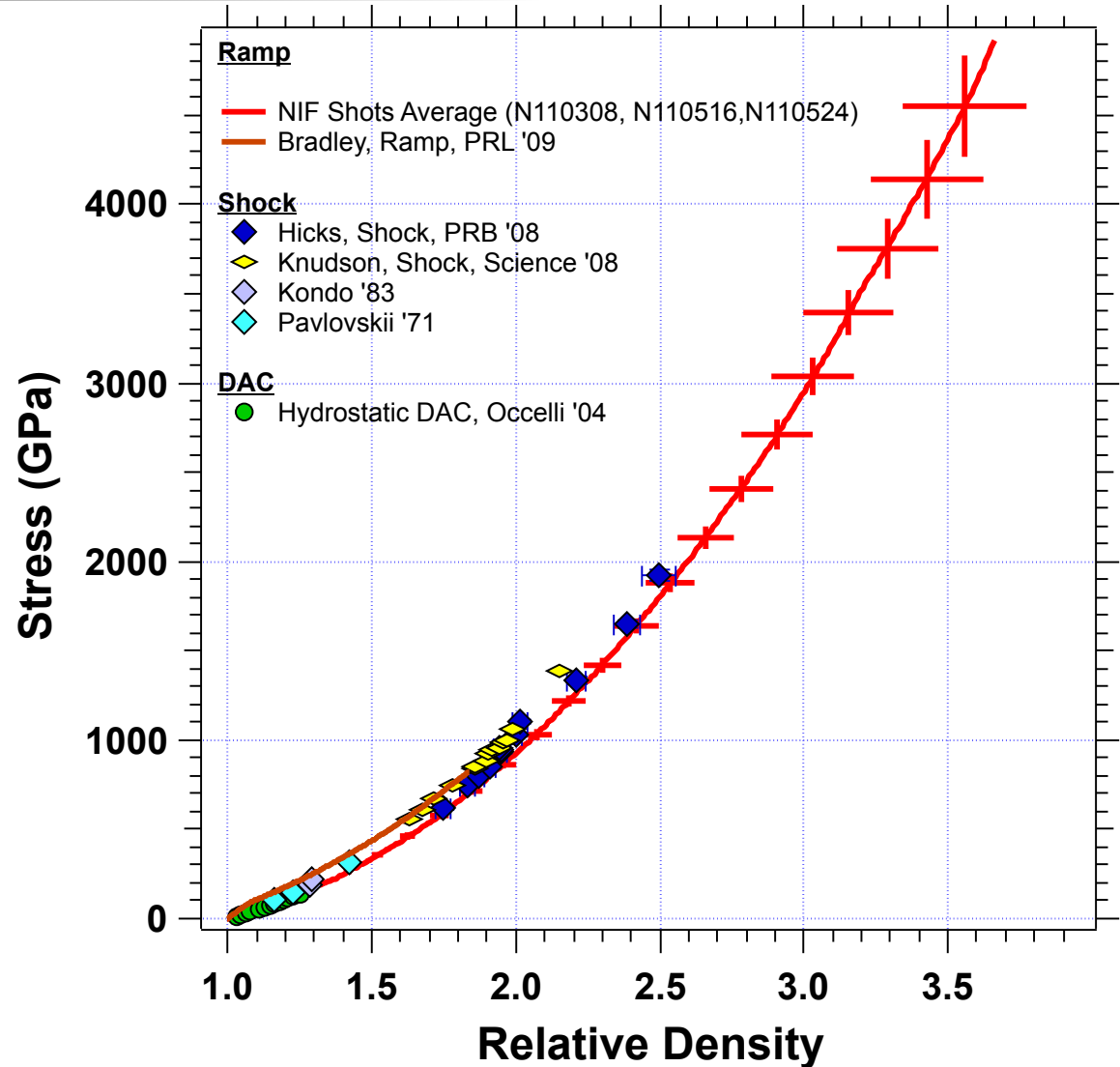
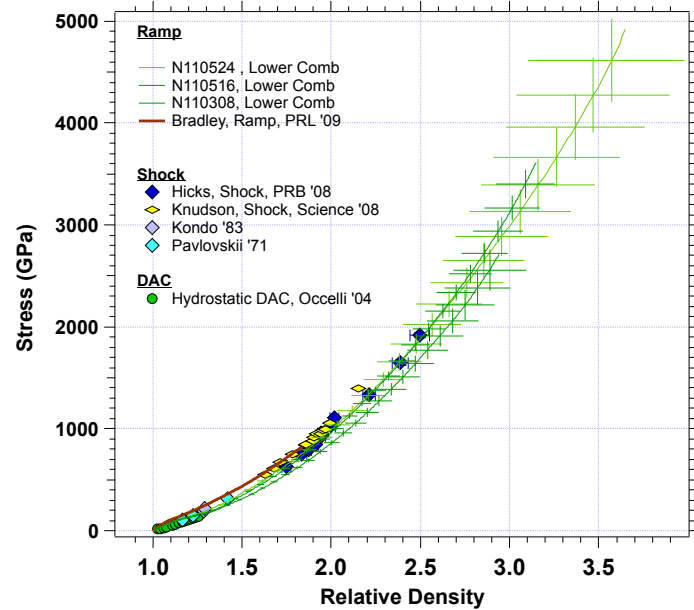
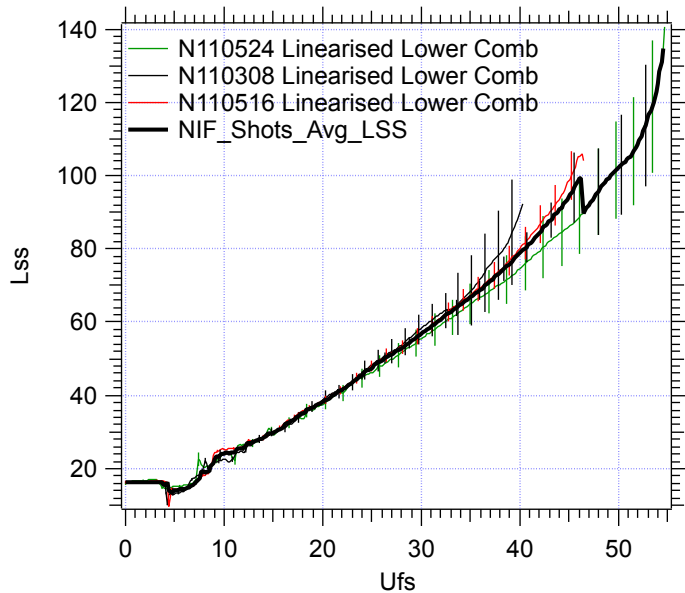
Pulse Shape evolution for diamond ramp experiments



Requested vs delivered pulseshapes for ramp experiments



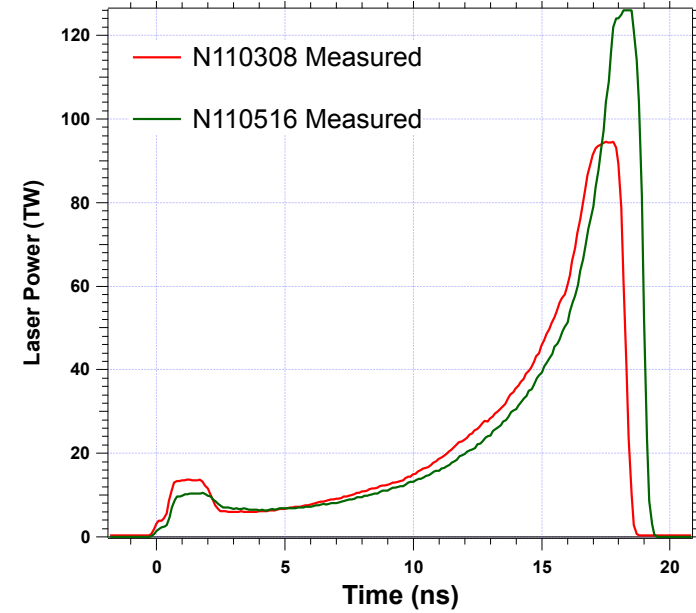
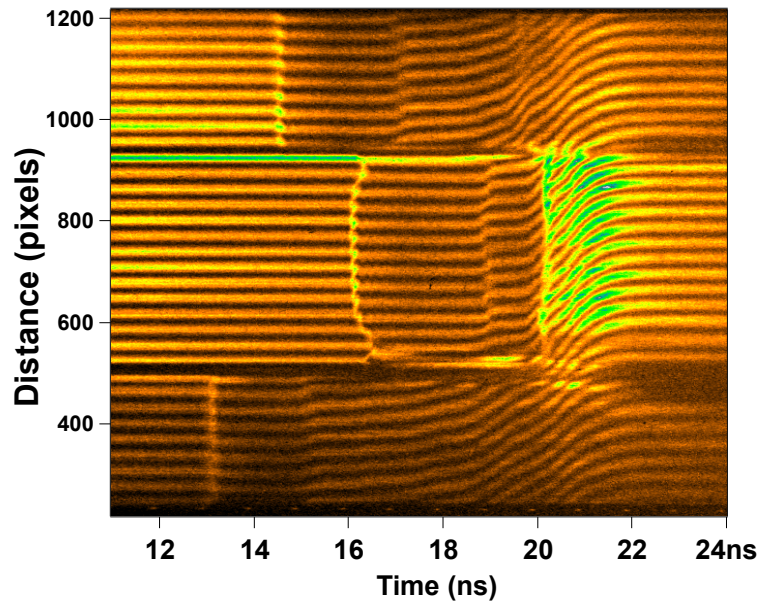
Ramp-compression EOS of nano-crystalline diamond to 50 Mbar.



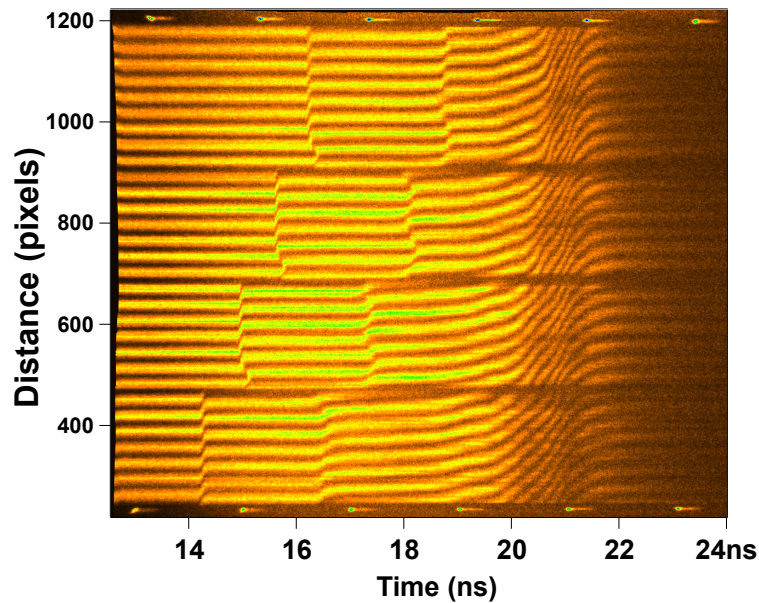
Average of three shots

Pulse shaping required very small adjustments to laser-power drive: First two data shots.

27 Mbar
N110308

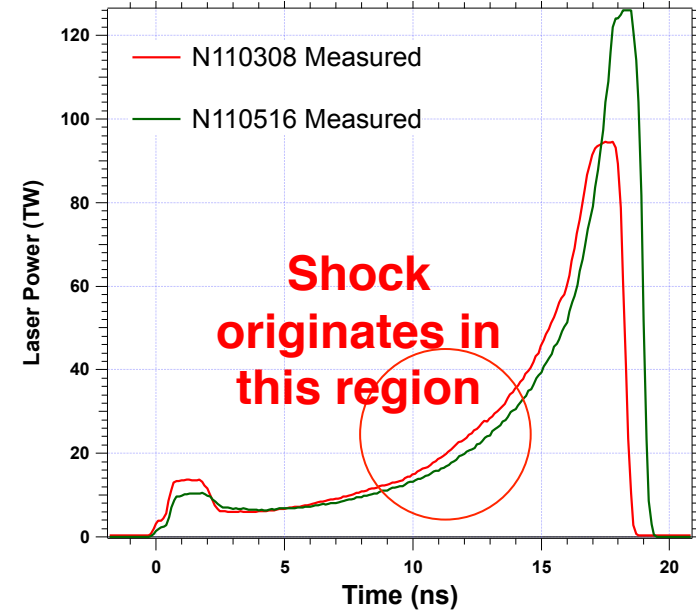
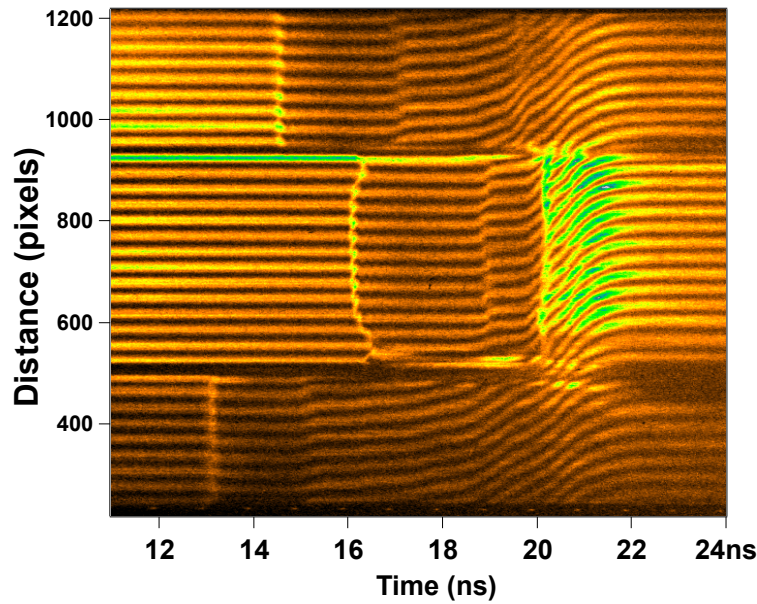


37 Mbar
N110516

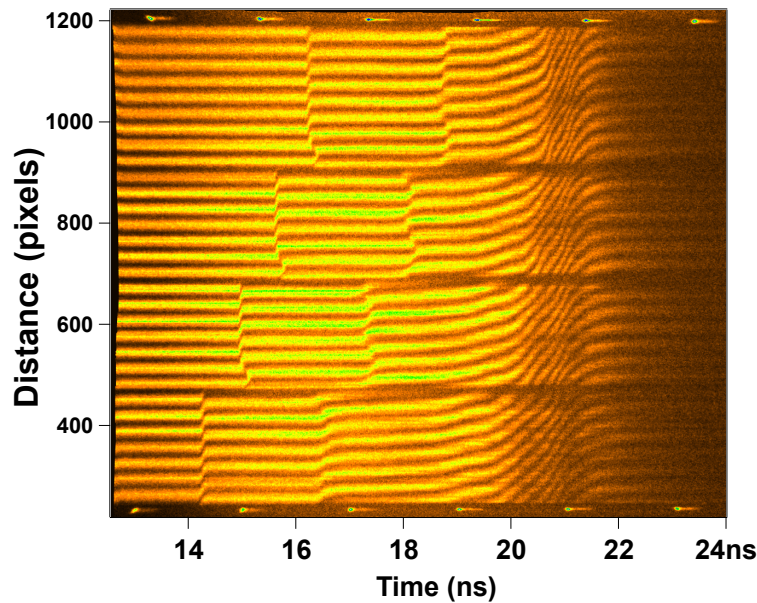


We were able to identify and correct regions responsible for growing shocks.

27 Mbar

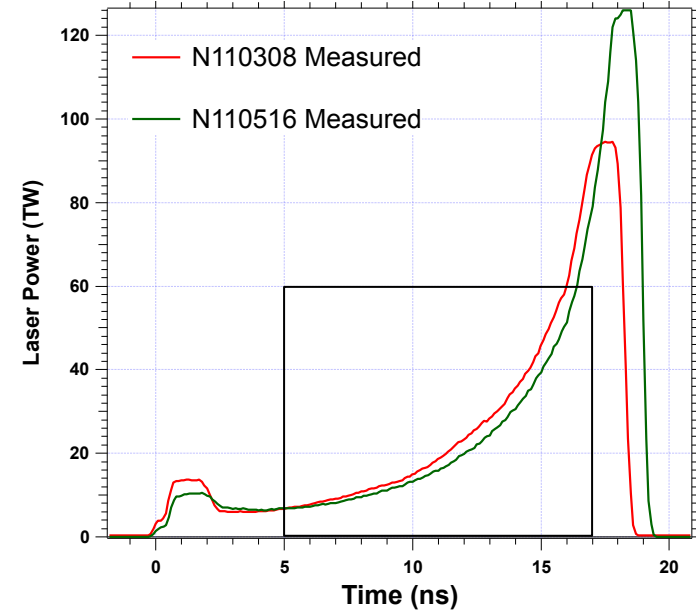
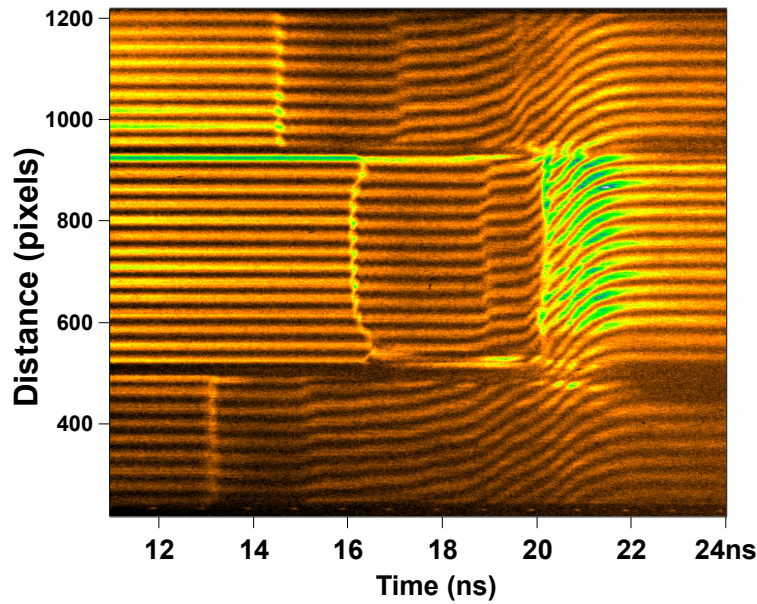


37 Mbar

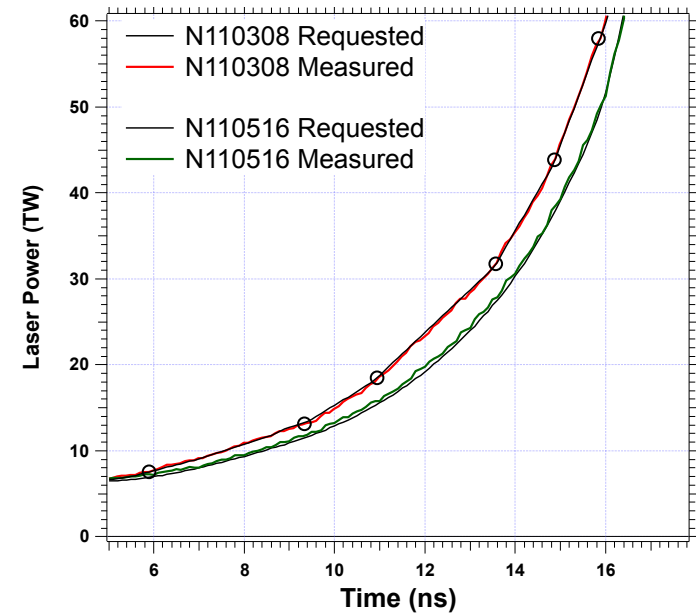
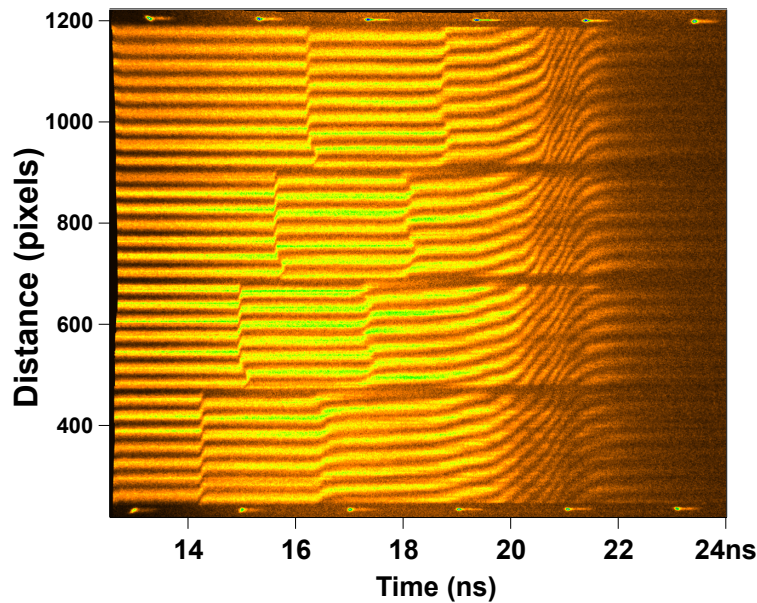


Pulse-shape correction worked extremely well.

27 Mbar

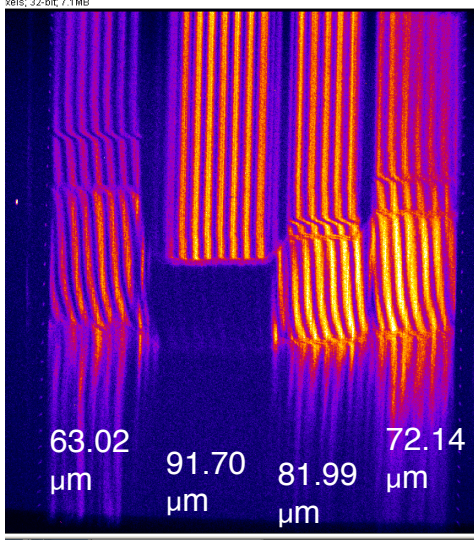


37 Mbar

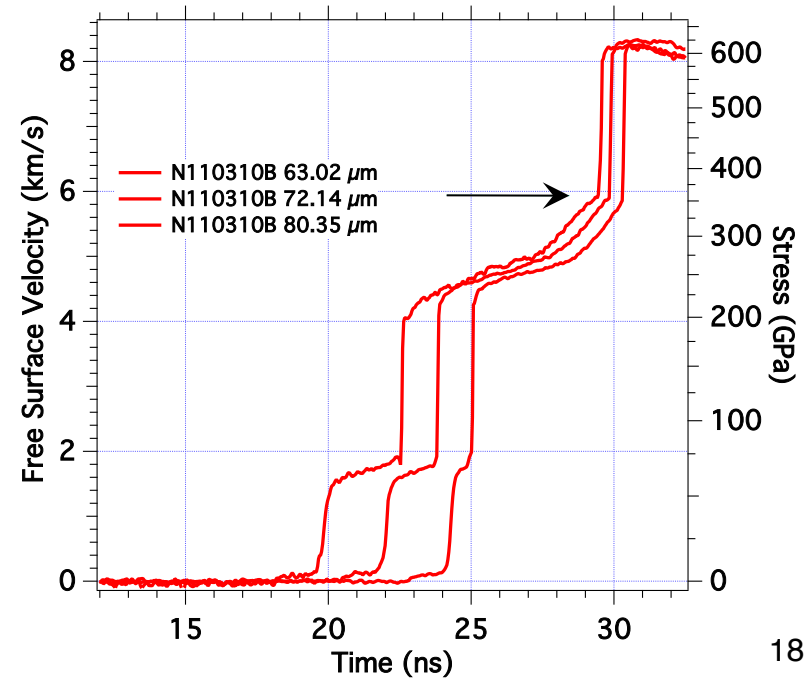
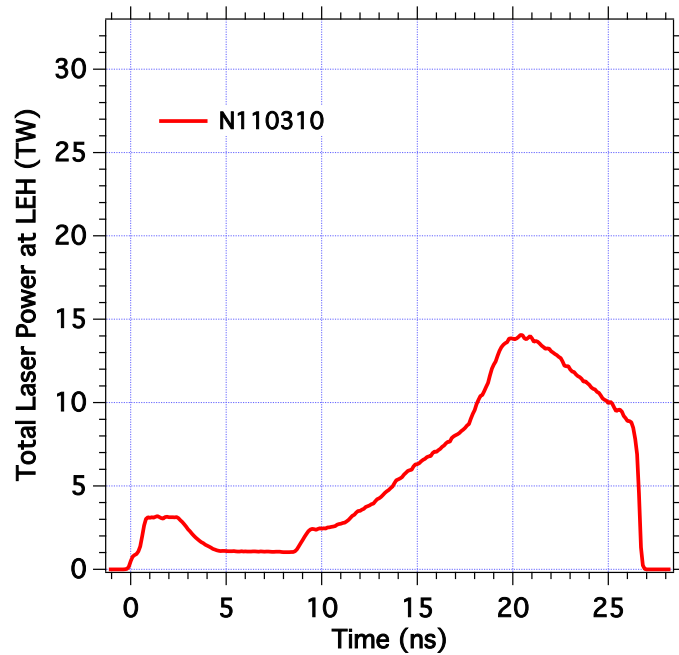


Vapor-deposited tantalum EOS on NIF

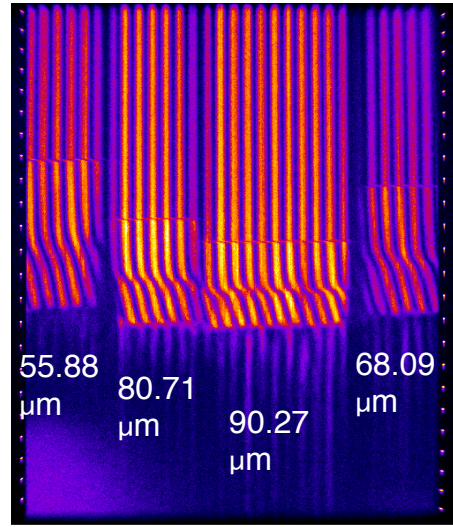
On the NIF, Tantalum shocked up independent of thickness



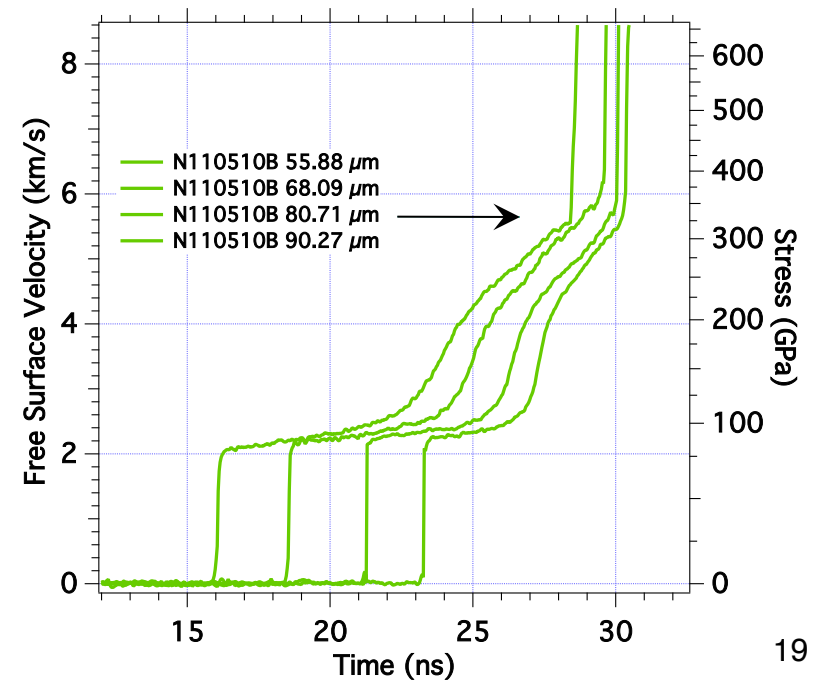
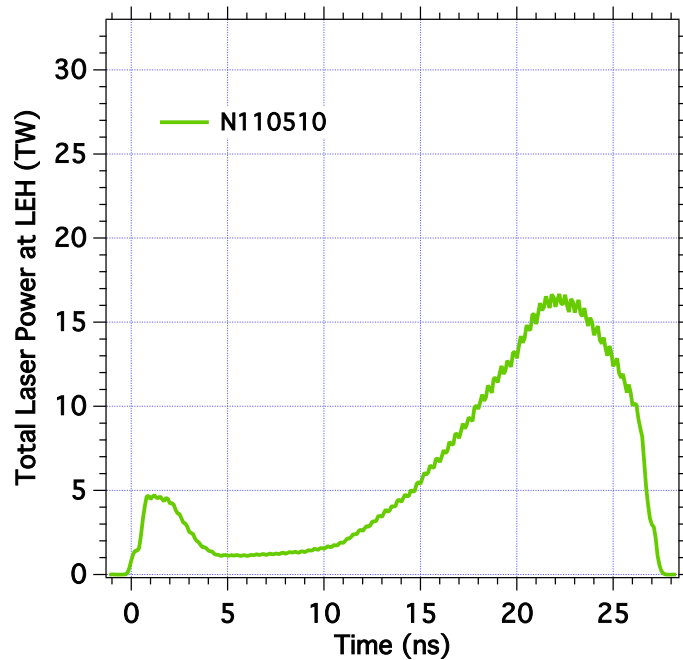
< Eliminate multiple shocks >



After eliminating 2nd shock, tantalum still shocked up at the same velocity on each step

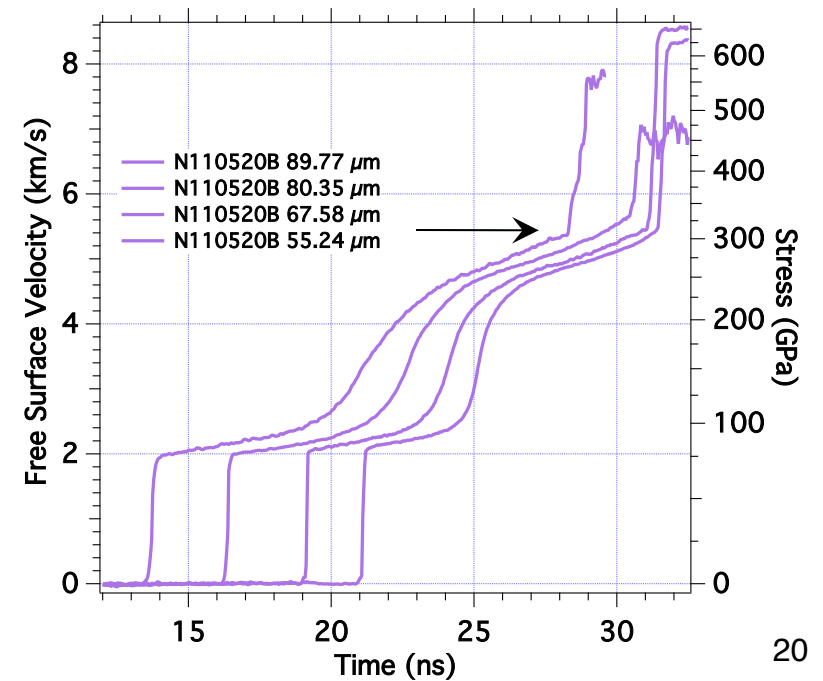
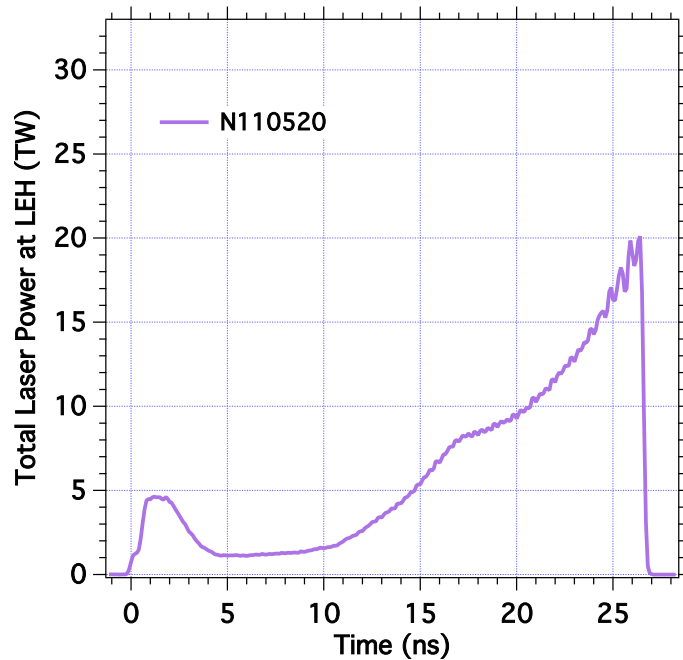
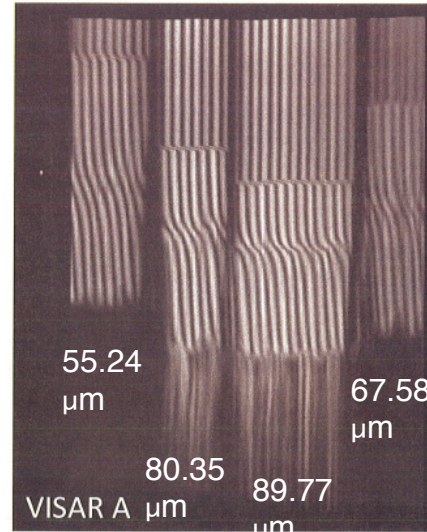


< Reduce strain rate at shock >



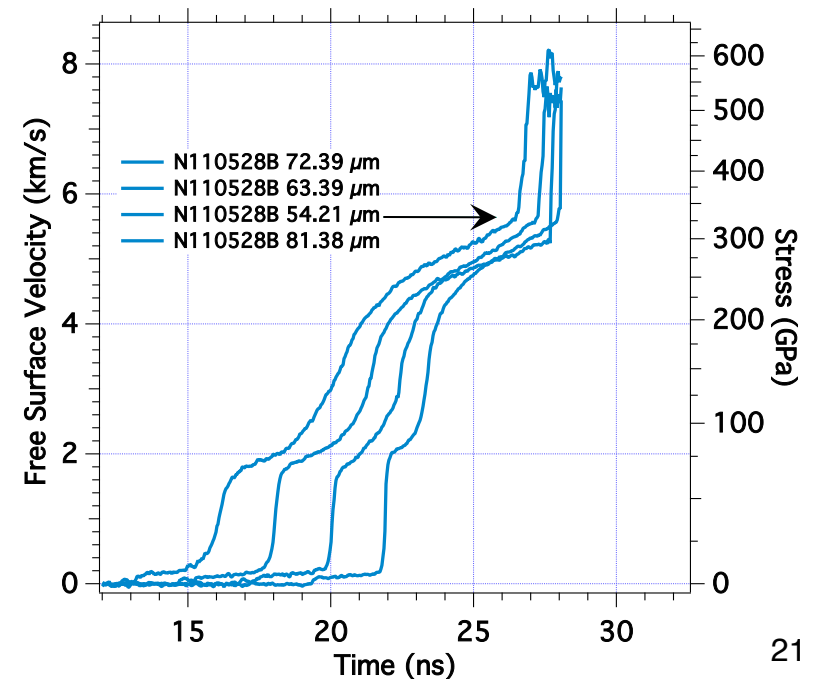
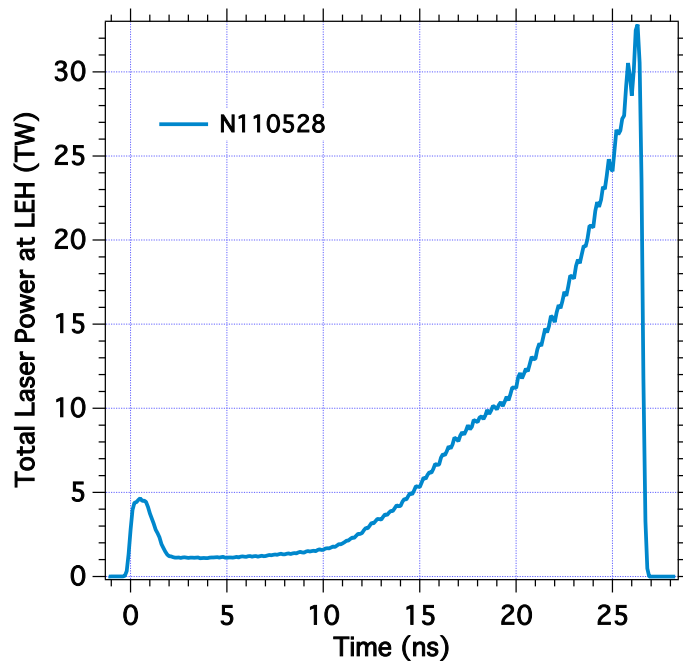
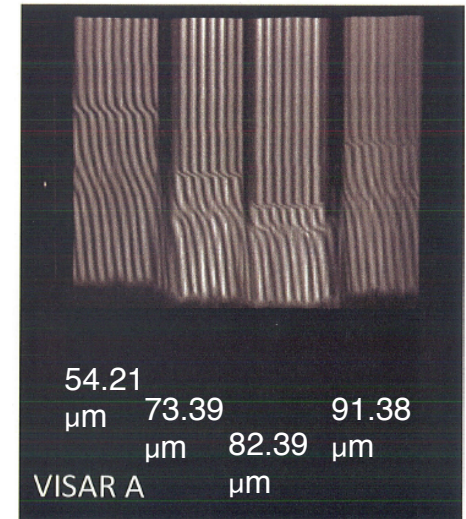
On the NIF, Tantalum shocks up at the same velocity on each step

< Increase final pressure >

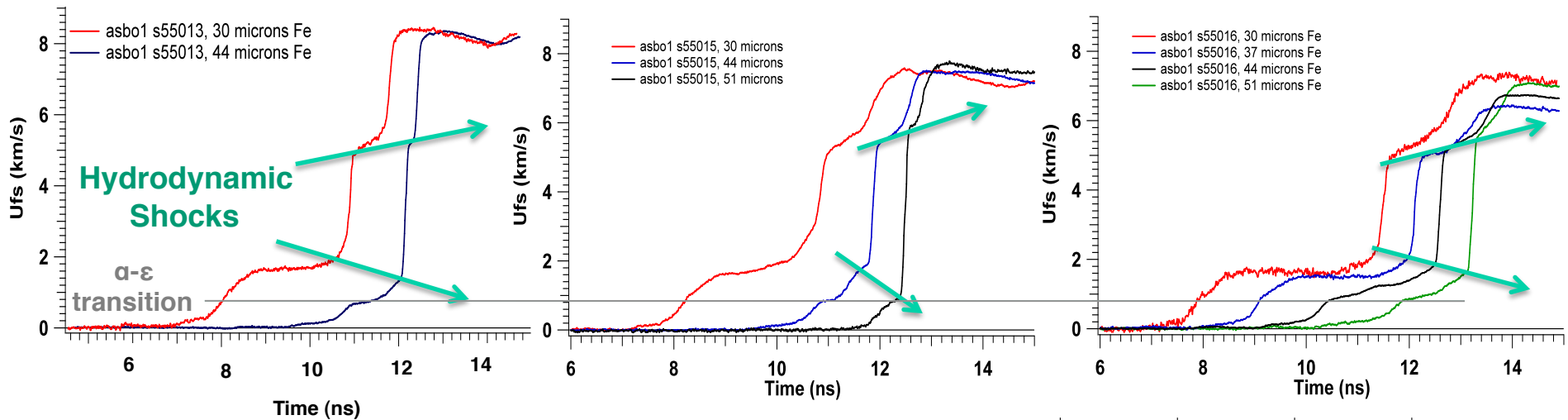


On the NIF, Tantalum shocks up at the same velocity on each step

Shock initiation at ~ 3.4 Mbar repeatedly with minor dependence on sample thickness or drive profile.



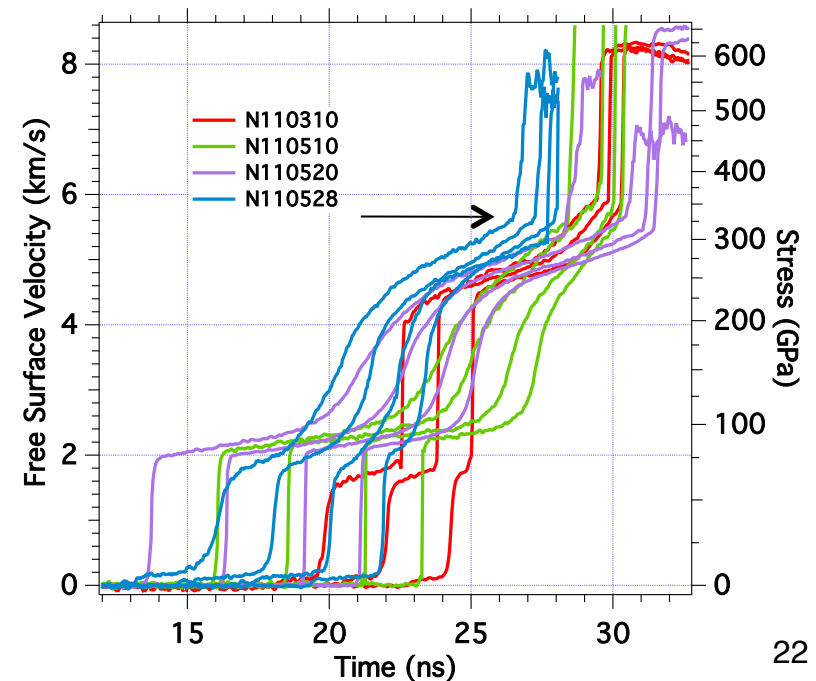
Hydrodynamic shocks in iron experiments at Omega



- Hydrodynamic shocks grow with thickness at both upper and lower limit of shock.

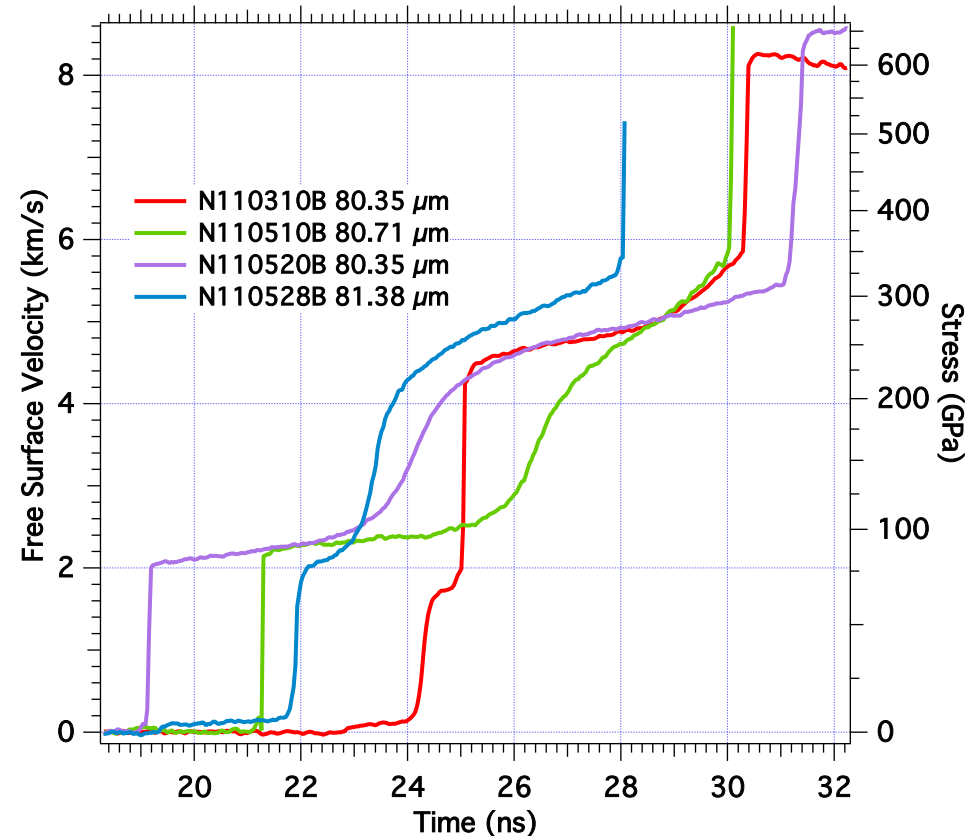
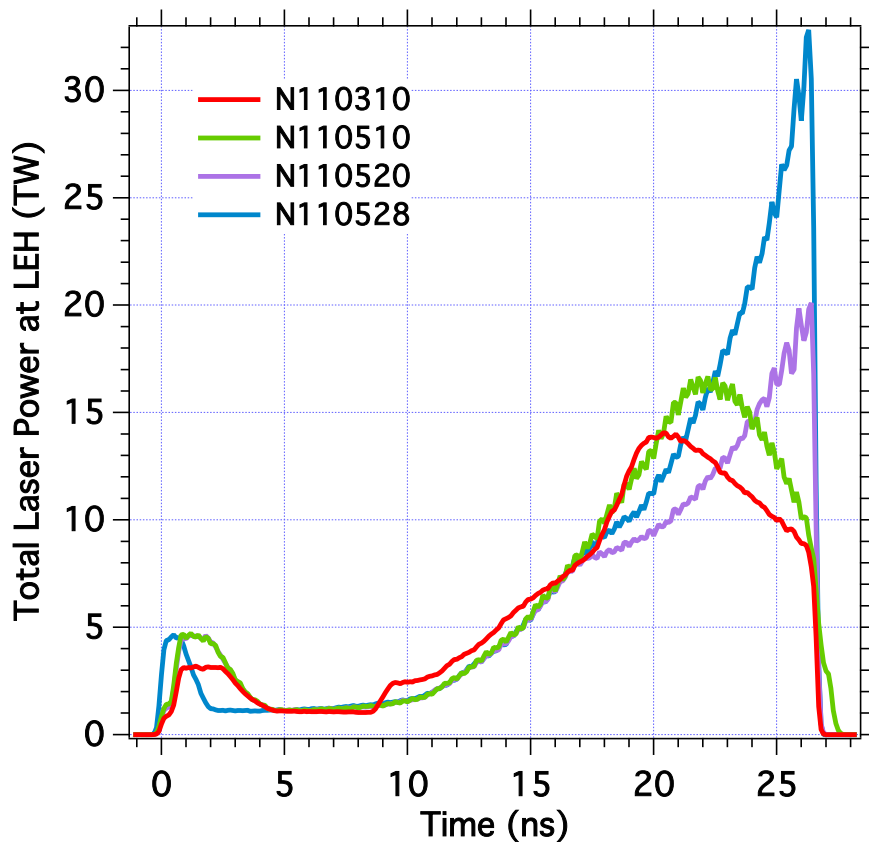
- Phase transition plateaus and shock initiation velocity are fixed with thickness

- On NIF, the Ta shocks do not appear to be hydrodynamic steepening in origin



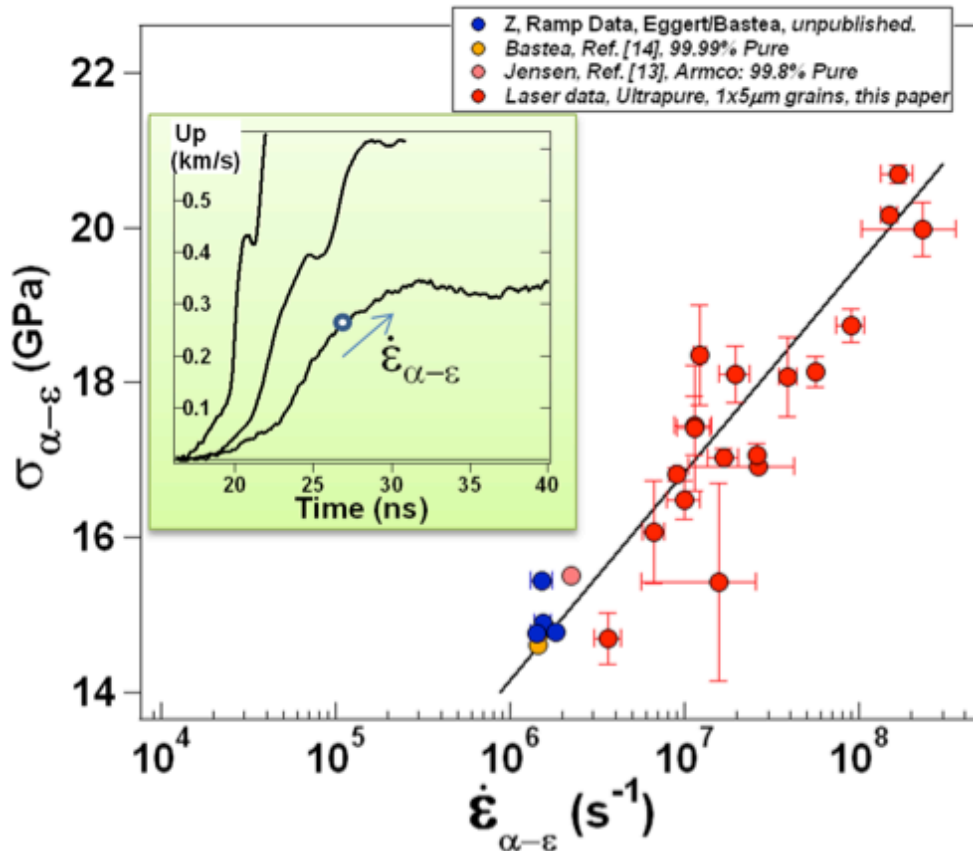
For tantalum, shock initiation is very independent of pressure-drive profile

Variation of pressure drive (stress rate) is much larger than was needed for diamond. Shock does not appear due to hydrodynamic steepening.

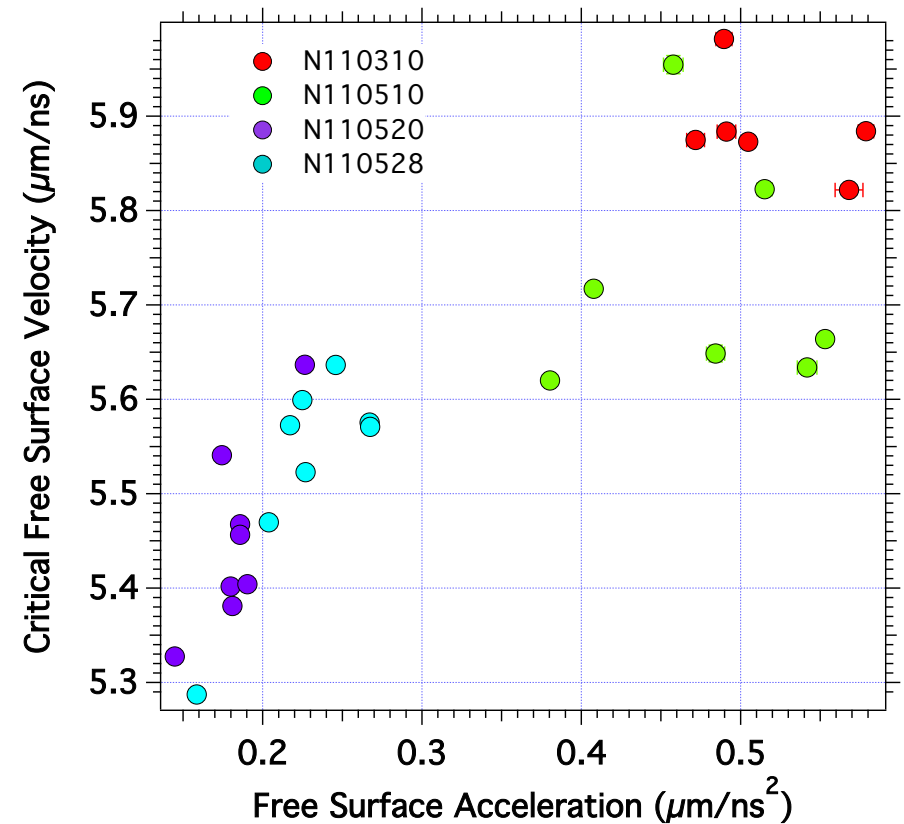


Critical stress for phase transitions increase with free-surface acceleration (strain rate)

Data for iron on Omega and Z



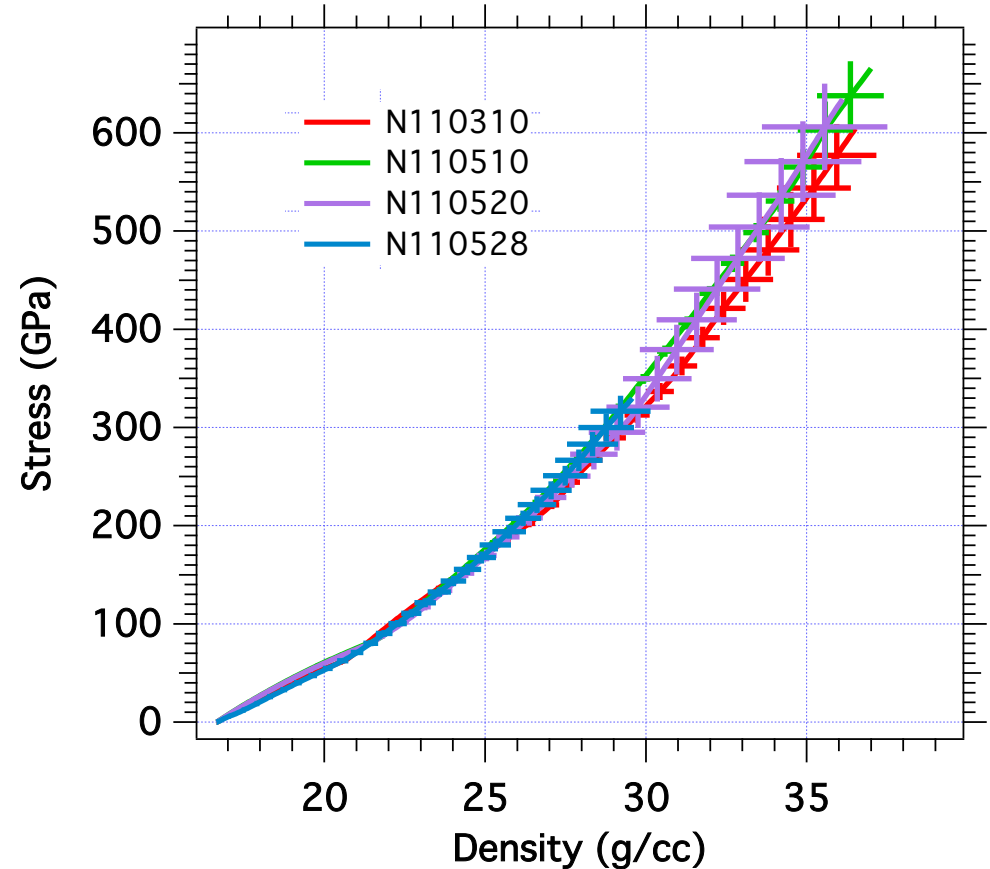
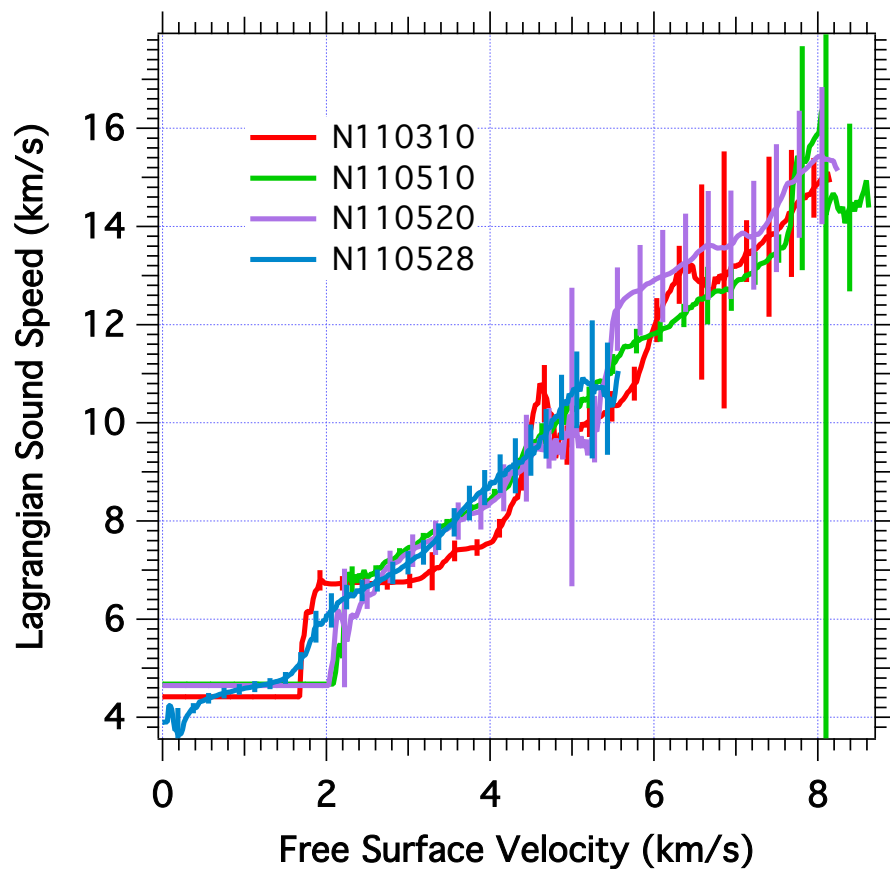
Similar results for Ta on NIF



Results for tantalum on NIF appear most consistent with a phase transition

Since the shock was steady, we can extract an EOS to 6.7 Mbar

Independent analyses for each shot



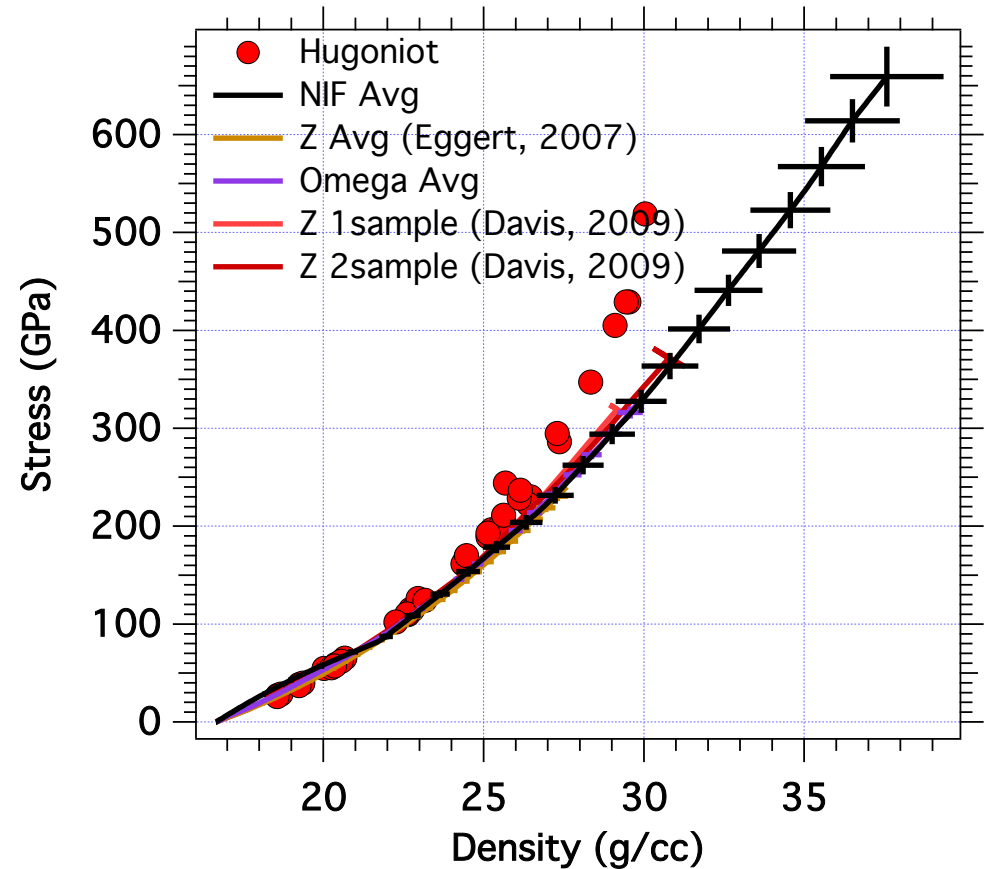
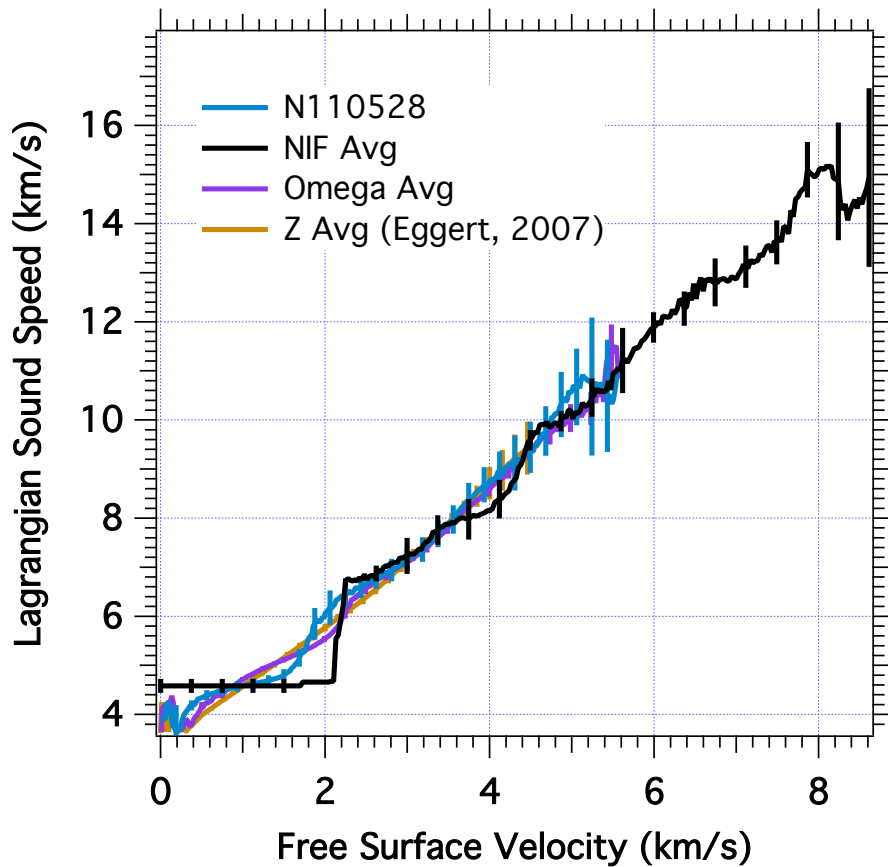
Lagrangian R-H Equations:

$$P_2 - P_1 = \rho_0 U_{SL} (U_{p2} - U_{p1})$$

$$\rho_0 U_s = \rho_2 \left(\frac{\rho_0}{\rho_1} U_s - (U_{p2} - U_{p1}) \right)$$

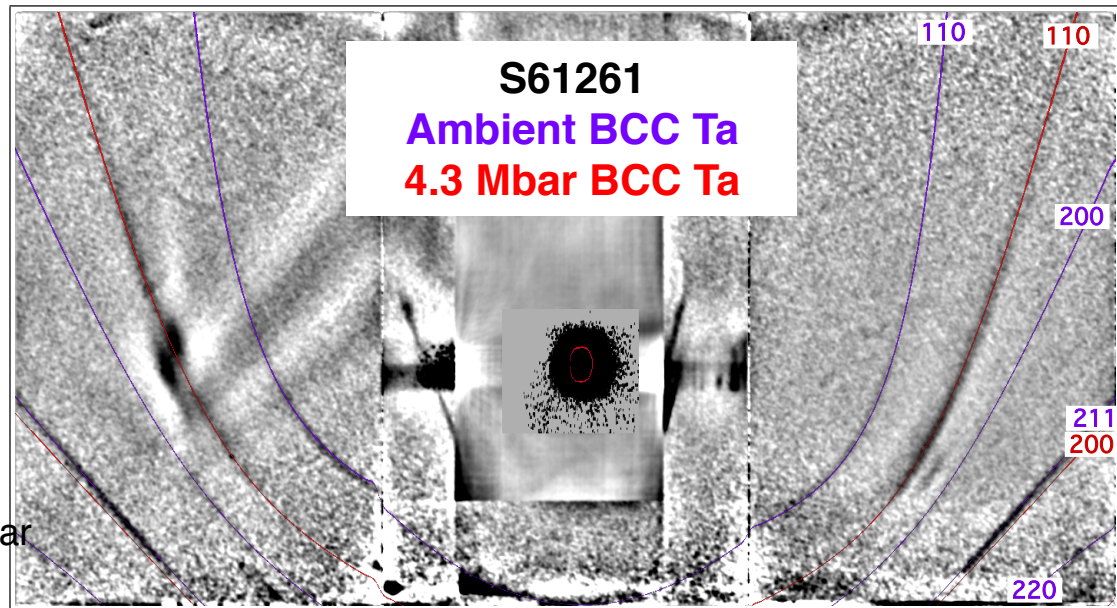
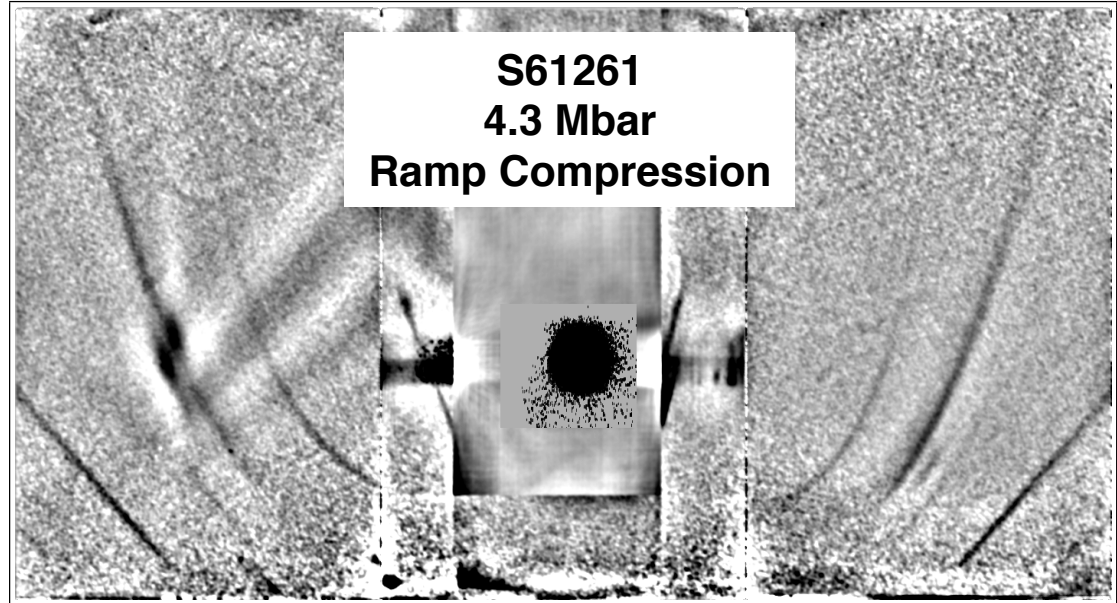
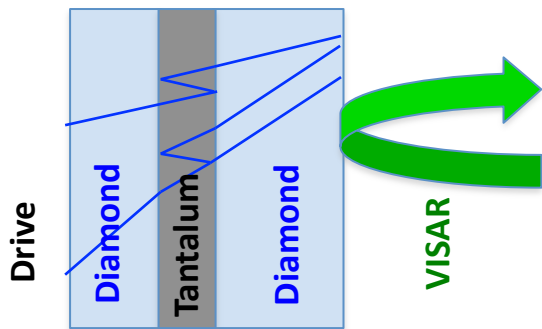
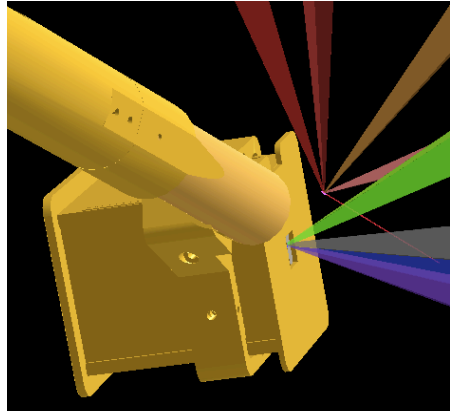
We average the sound speed for multiple shots

Excellent agreement with previous data from Z and Omega



Powder x-ray diffraction of rolled tantalum on Omega

We have also performed high-pressure x-ray diffraction on tantalum at the Omega laser



RampDiff-11B (Ta, ramp5)

Shot **61261**, OMEGA 2011-0223

Target: C[17]**Ta**[3]C[40], **BL: Fe**

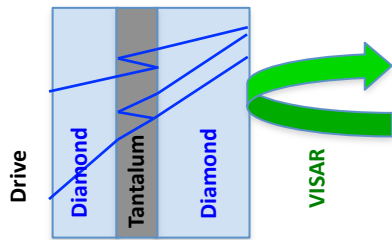
Ramp drive: 246 J ($t_{BL} = 4.6$ ns)

$P_{\text{expected}}: 5.0$ Mbar, $P_{\text{max_exp}} = 4.34 \pm 0.09$ Mbar

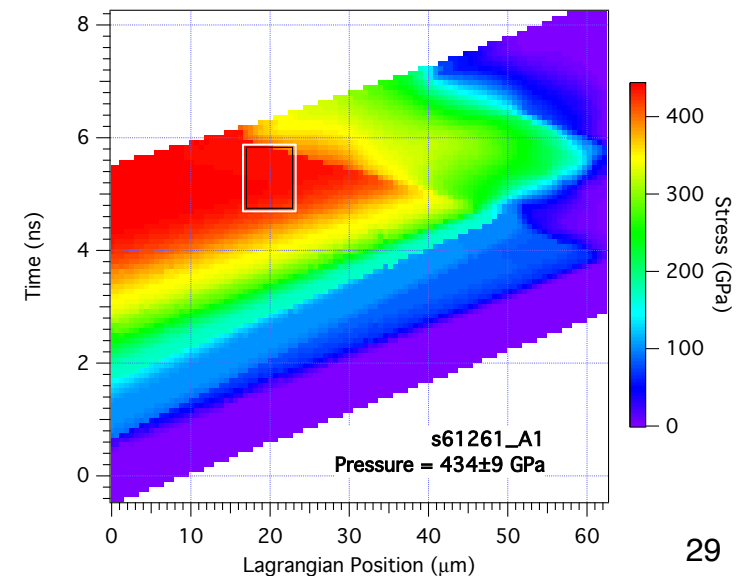
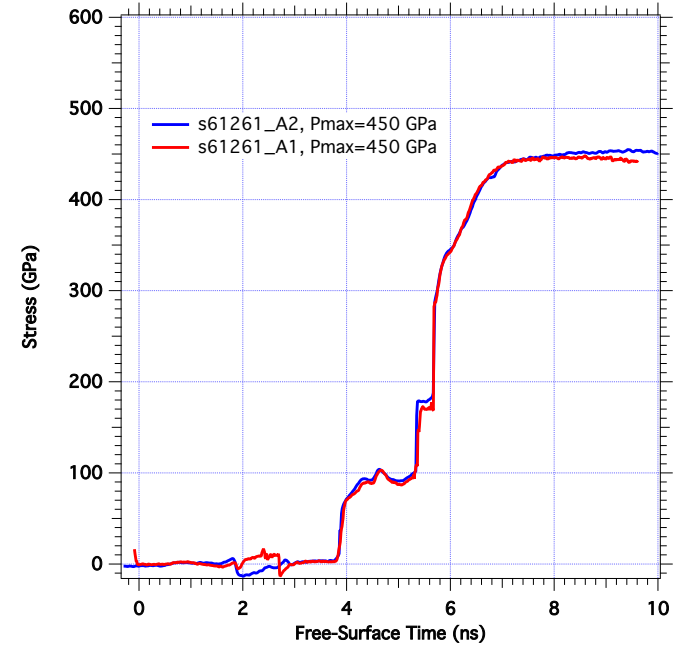
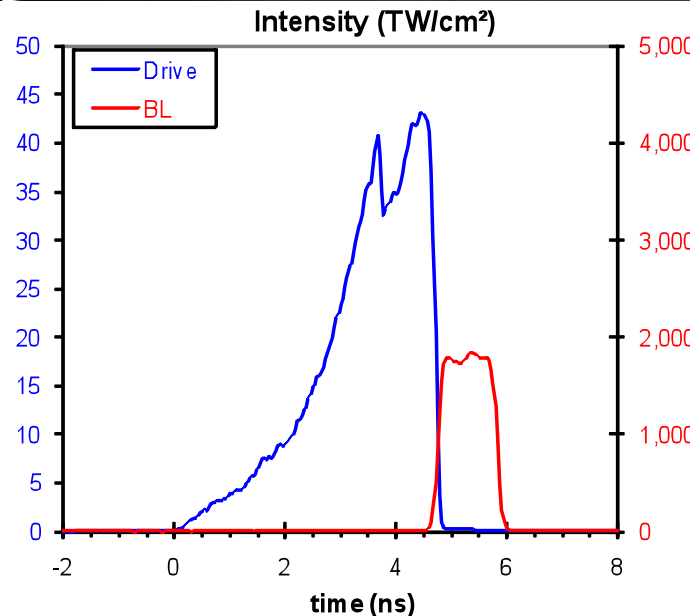
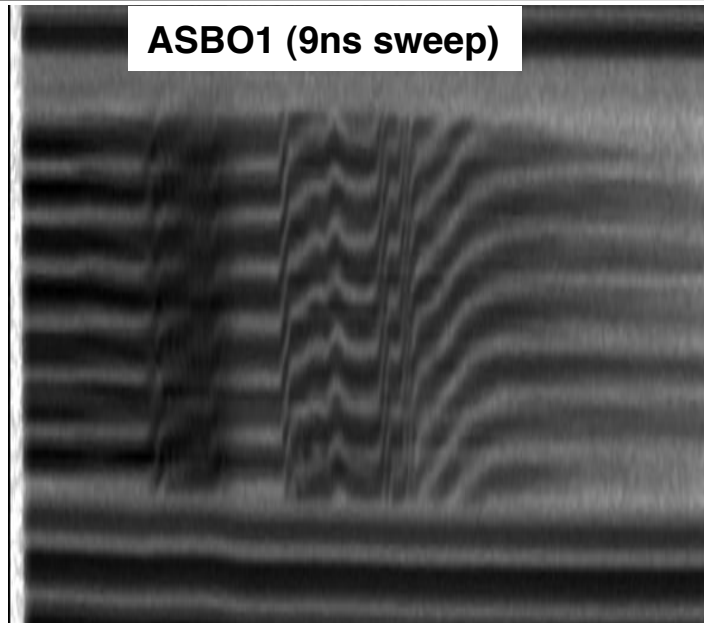
We determine stress by backward integration of diamond free-surface velocity

Shot **61261**,
OMEGA 2011-0223

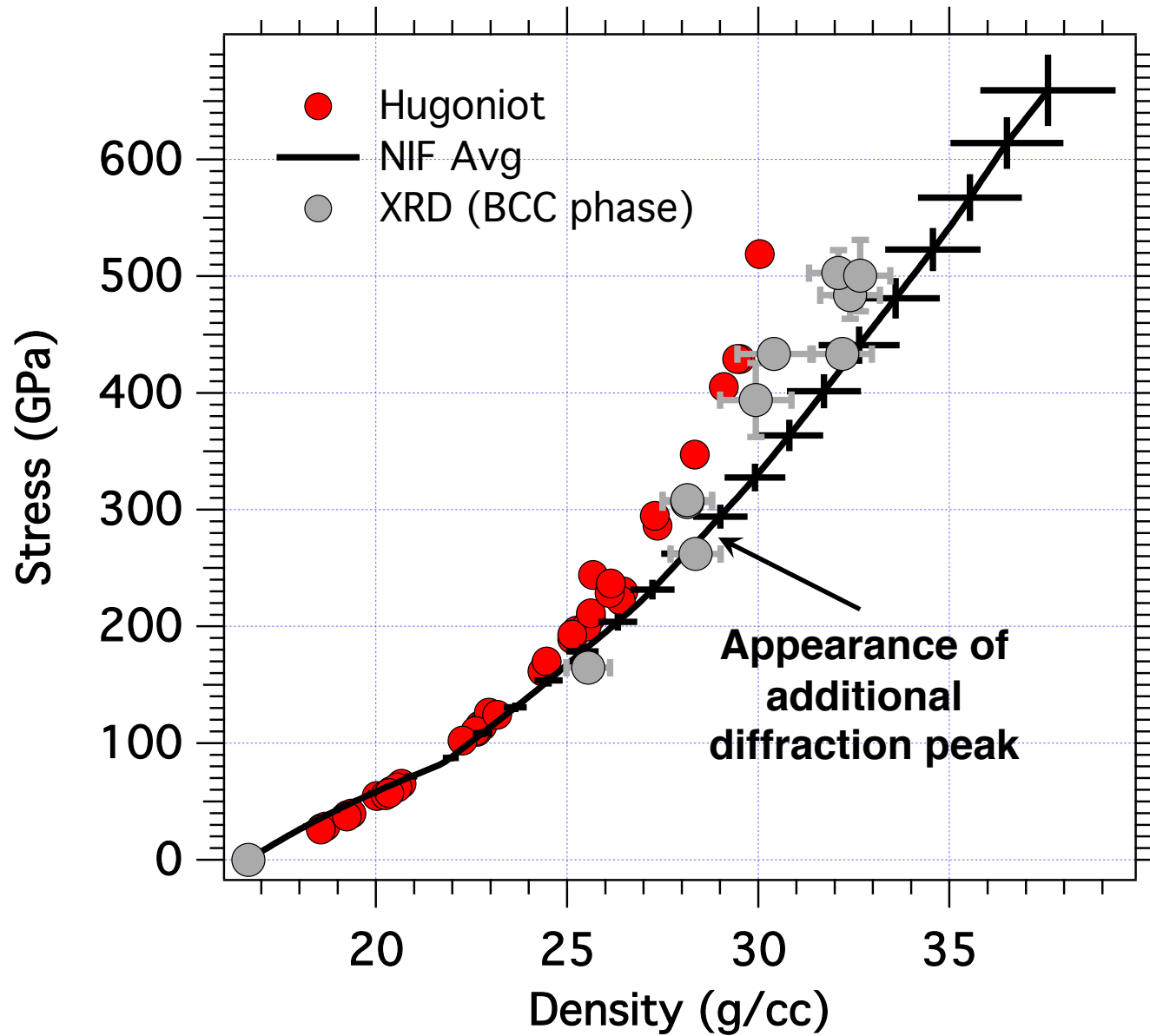
Target: C[17]**Ta**[3]C
[40], **BL: Fe**



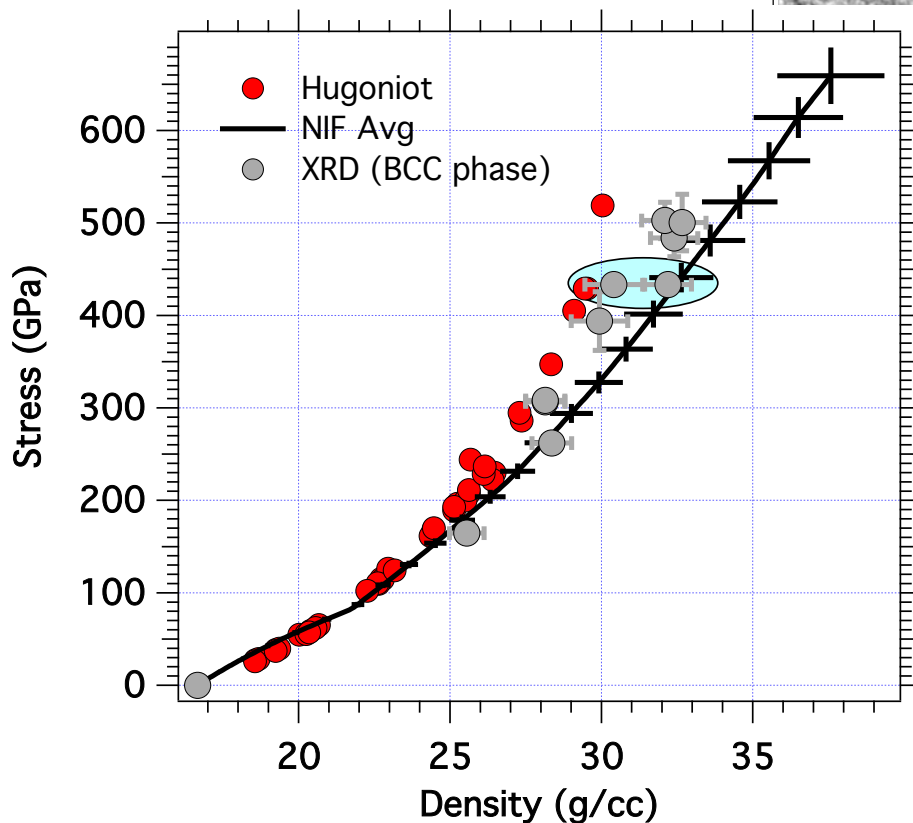
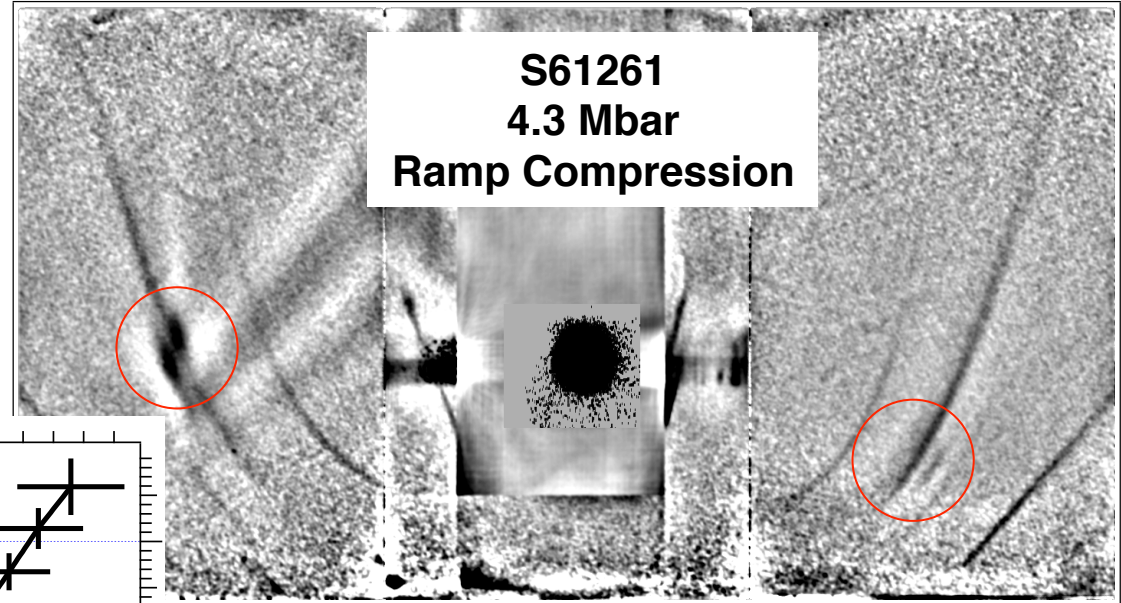
Ramp drive: 246 J
($t_{BL} = 4.6$ ns)
 $P = 4.34 \pm 0.09$ Mbar



We have measured the (assumed) BCC density for 8 shots.



Diffraction for S61261 can be interpreted as two phase coexistence



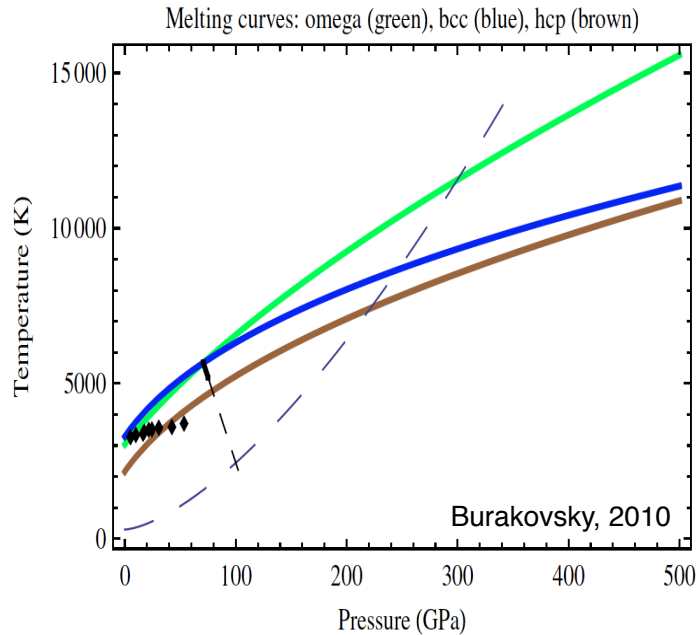
A consistent understanding of both the NIF EOS and the Omega diffraction data can be had by positing a Ta phase transition near 3.4 Mbar.

e.g. Burakovski (2010) predict ω -phase.

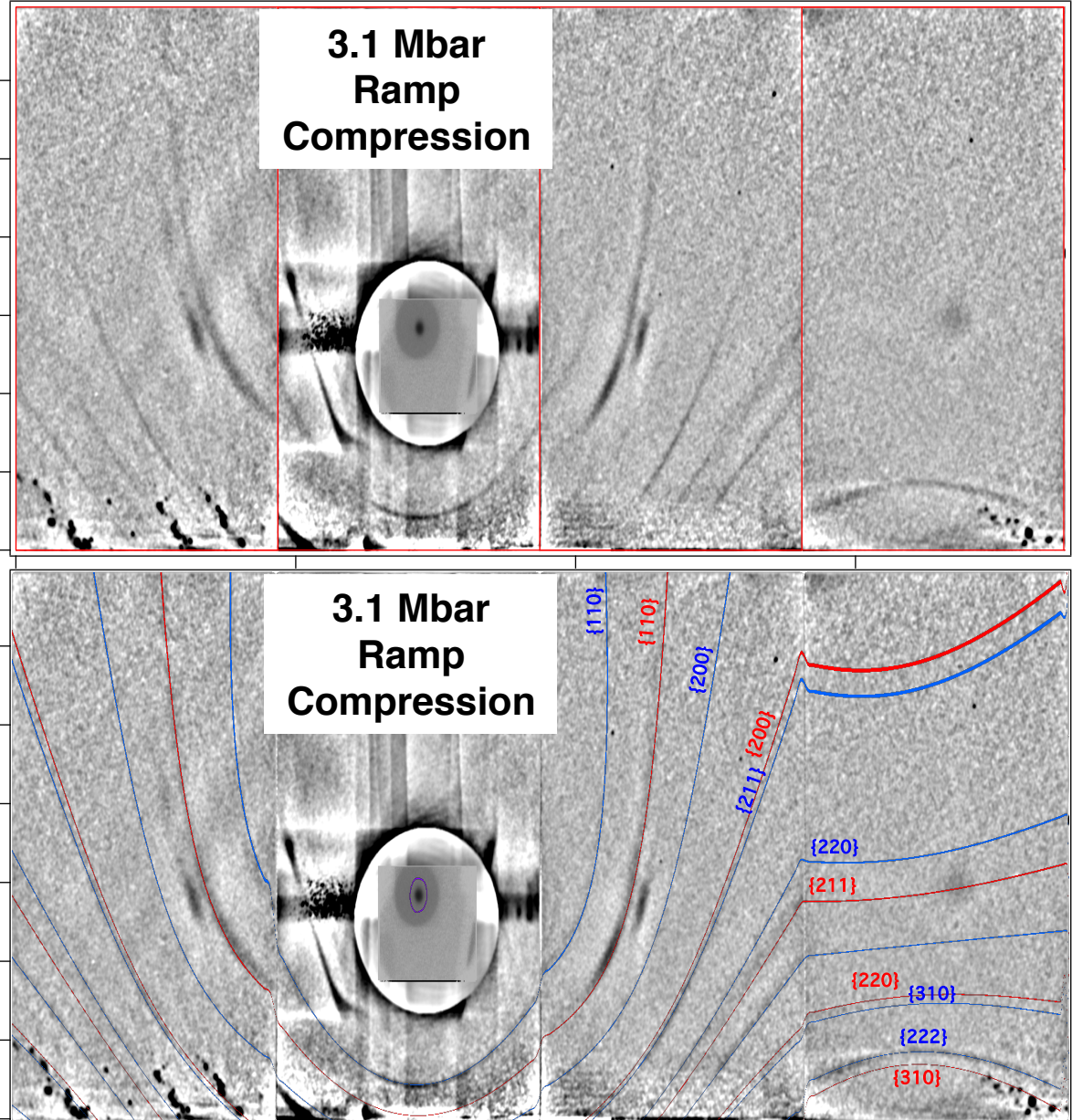
High-Pressure X-Ray Diffraction of Tantalum: Possible phase transition.

RampDiff-11B (Ta, ramp4)
Shot **61267**, OMEGA 2011-0223

Target: C[22]**Ta[3]**C[40], **BL: Cu**
Ramp drive: 171 J ($t_{BL} = 4.7$ ns)
P = 3.1 Mbar

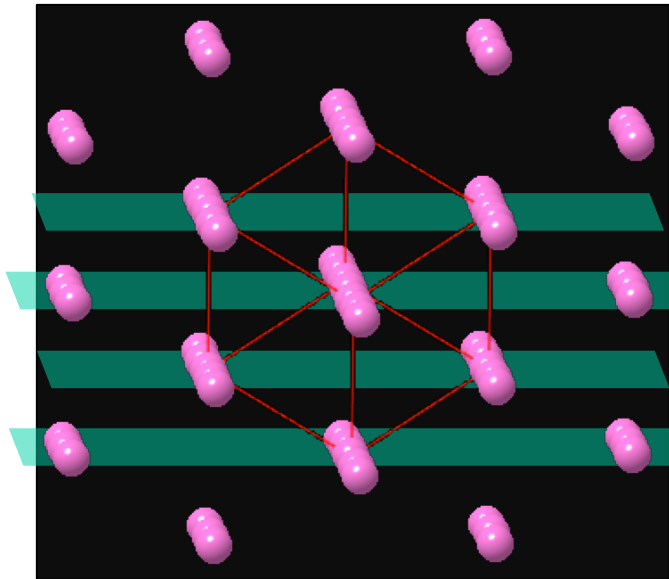


$\{211\}_{BCC}$ line is exactly 90° between plane normal and compression axis.

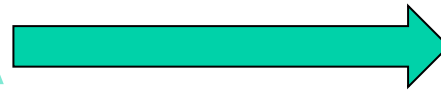


Burakovsky (2010) predicted Omega phase with $c/a \sim 0.572$

BCC

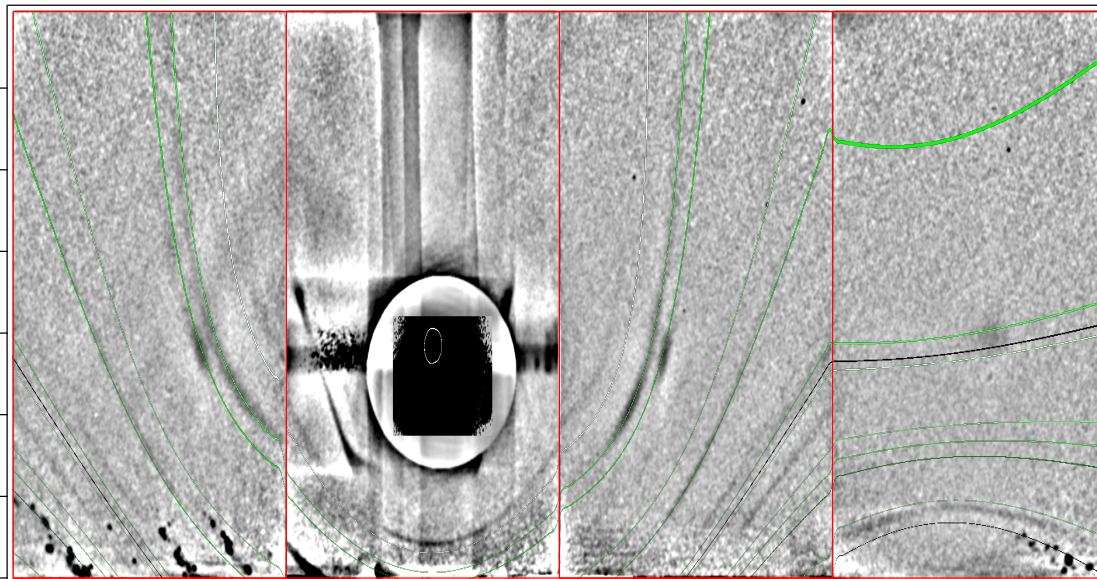
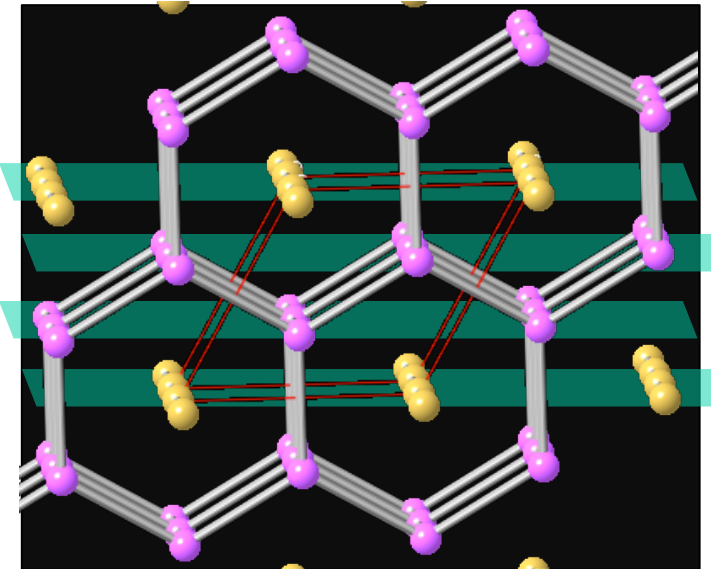


$c/a=0.612$ and a shuffle every third $(112)_{\text{bcc}}$ plane



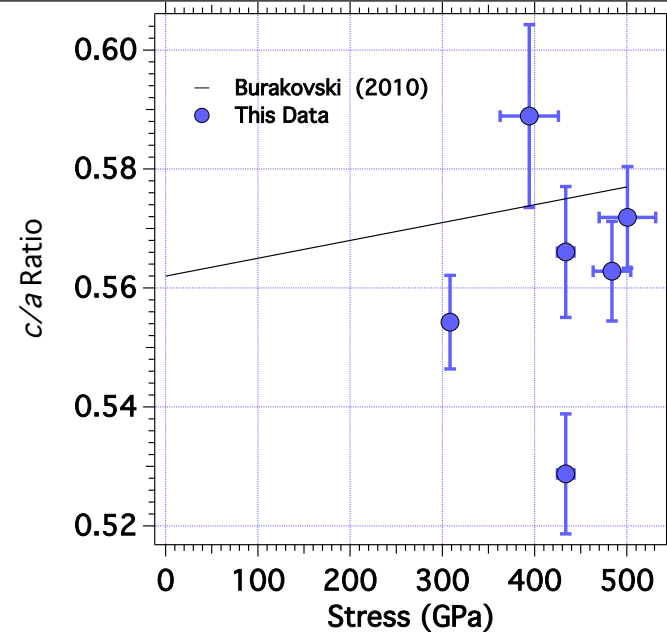
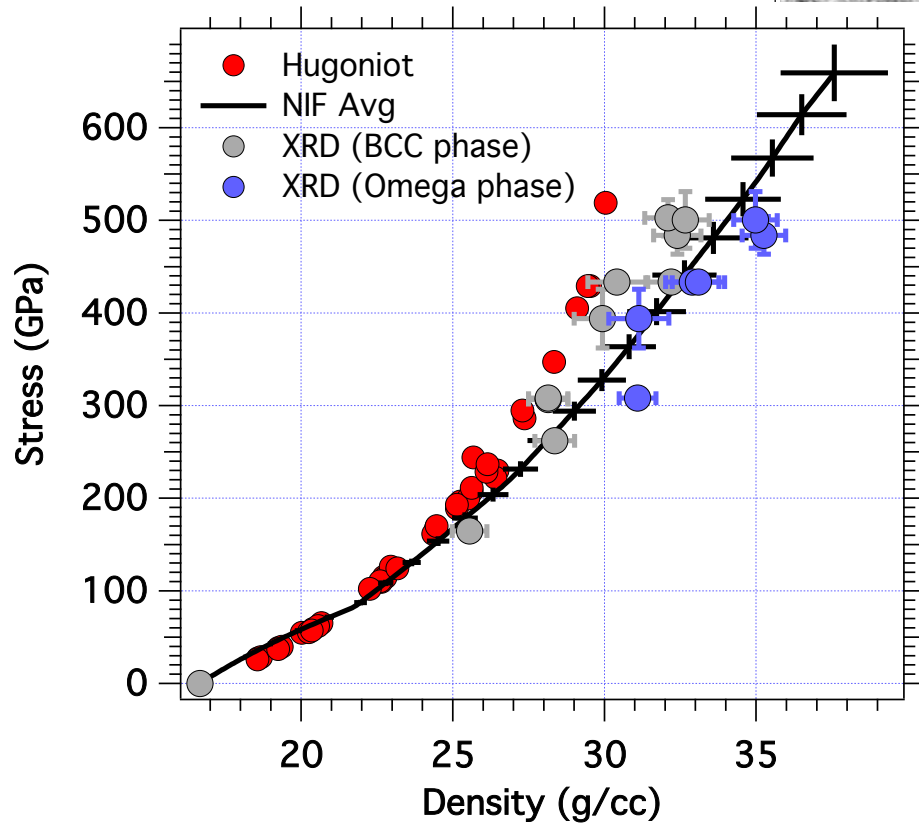
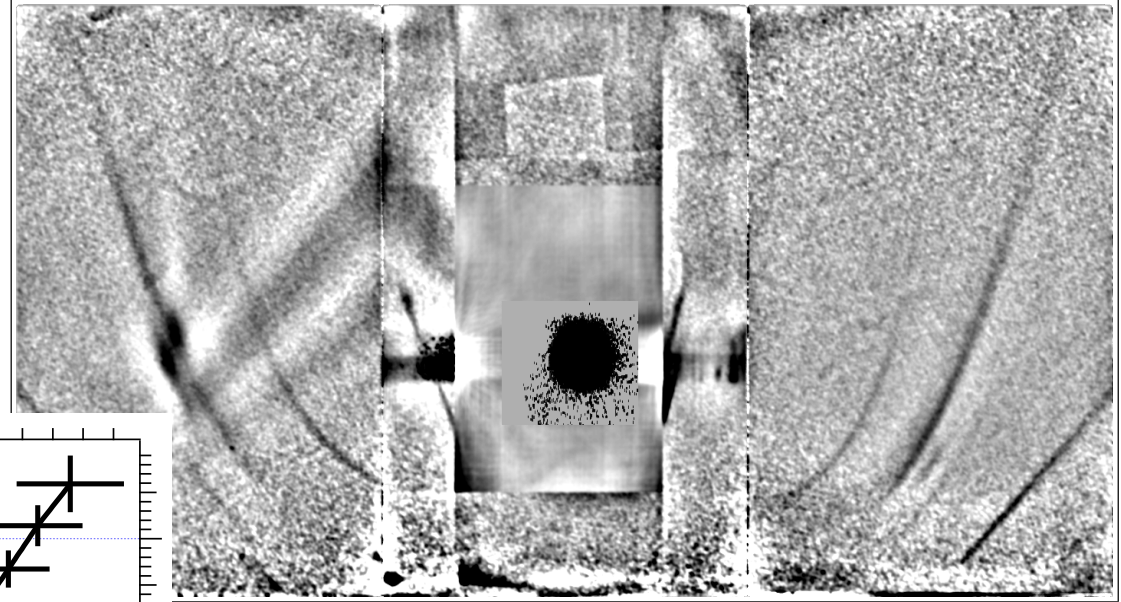
Yields the Omega phase with a $(112)_{\text{bcc}} \rightarrow (300)_{\Omega}$ Correspondence

Omega



Omega best fit with $V=29 \text{ cc}$, and $c/a=0.56$

We can also determine density assuming the omega phase



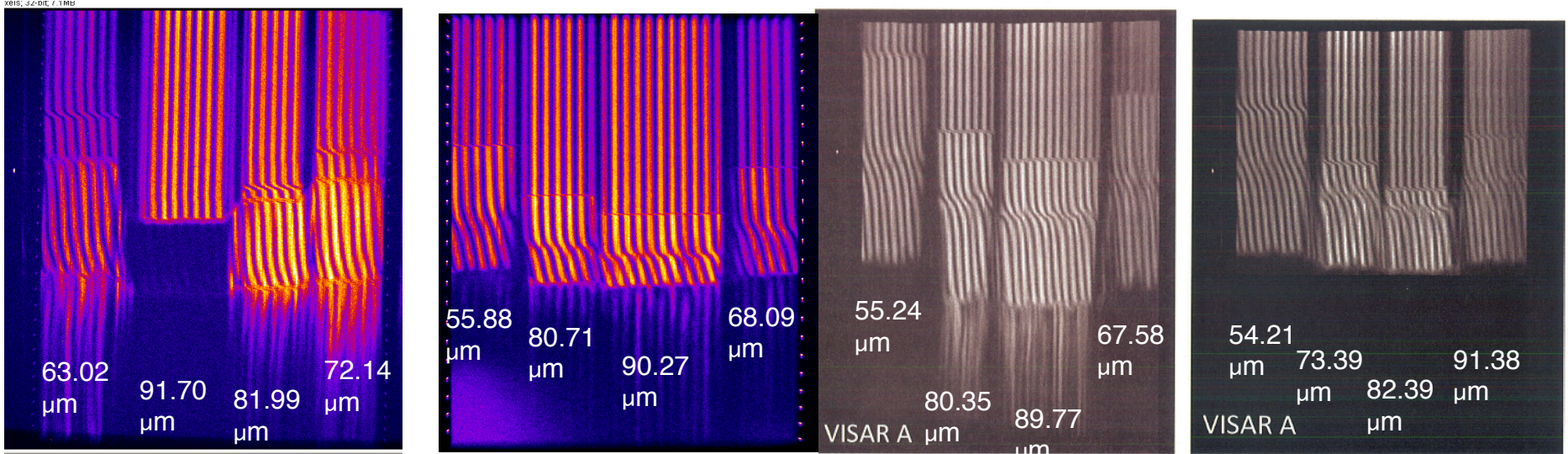


About 30% of NIF's capacity was needed to reach 50 Mbar on ramped diamond, and about 8% of capacity to reach 9 Mbar in Ta.

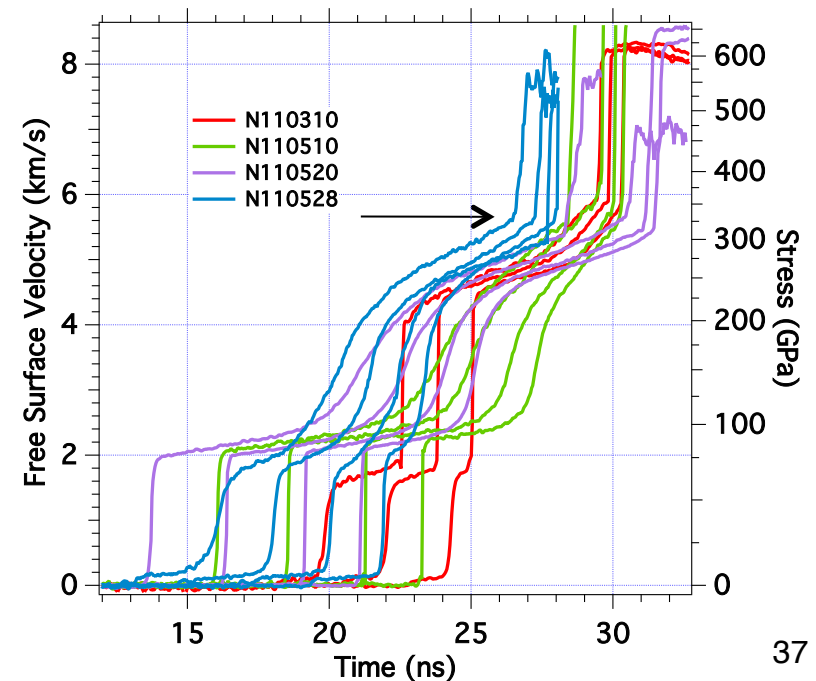
We need to continue to develop experimental techniques for this regime.

Supporting Slides

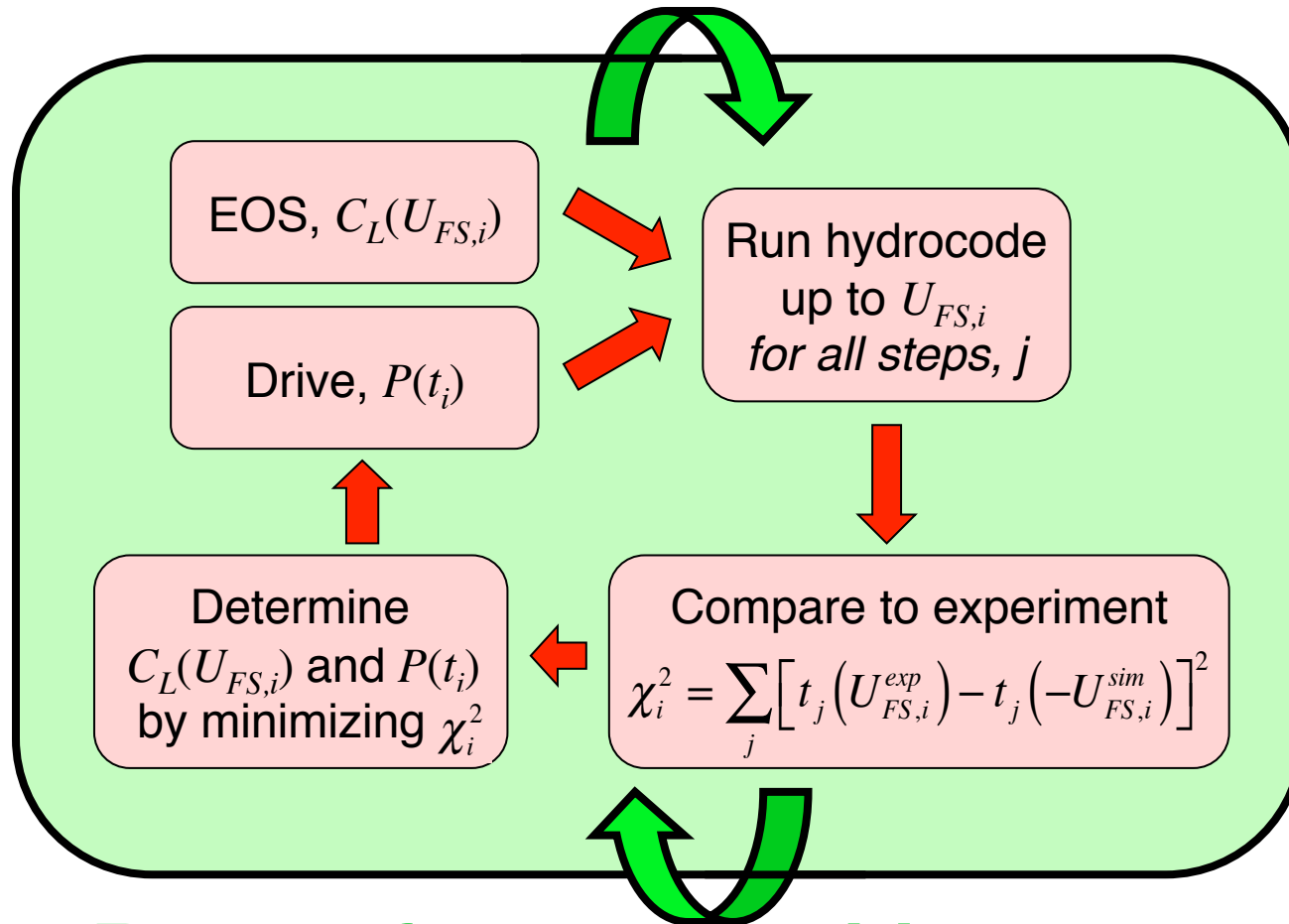
On the NIF Tantalum shocks up at the same velocity on each shot



Shock initiation at ~ 3.4 Mbar repeatedly with minor dependence on sample thickness or drive profile.



We will use a Forward-Only Analysis Method to deal with time-dependent transitions and growing shocks.



Repeat for all velocities, $U_{FS,i}$

- This method still requires a model for time-dependent phase transitions.
- Exact methods being developed by Evan Reed and by Bryan Reed potentially offer a very attractive alternative.