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3. Title: Transport Phenomena and Interfacial Kinetics in Planar Microfluidic Membraneless Fuel Cells

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4. Final Report

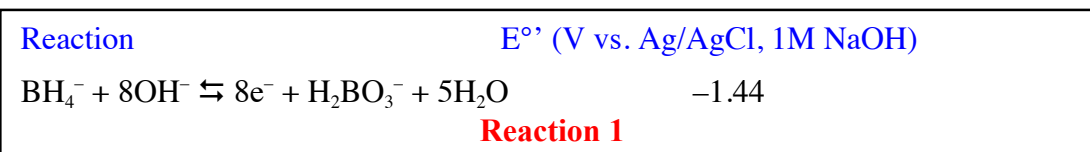
Abstract:

Our work is focused on membraneless laminar flow fuel cells, an unconventional fuel cell technology, intended to create a system that not only avoids most typical fuel cell drawbacks, but also achieves the highest power density yet recorded for a non-H₂ fuel cell. We have employed rigorous electrochemistry to characterize the high-energy-density fuel BH₄⁻, providing important mechanistic insight for anode catalyst choice and avoiding deleterious side reactions. Numerous fuel cell oxidants, used in place of O₂, are compared in a detailed, uniform manner, and a powerful new oxidant, cerium ammonium nitrate (CAN), is described. The high-voltage BH₄⁻/CAN fuel/oxidant combination is employed in a membraneless, room temperature, laminar-flow fuel cell, with herringbone micromixers which provide chaotic-convective flow which, in turn, enhances both the power output and efficiency of the device. We have also been involved in the design of a scaled-up version of the membraneless laminar flow fuel cell intended to provide a 10W output.

5. Research Findings

A. Electroanalytical investigation of borohydride (BH₄⁻): Oxidation mechanism, best catalysts, and performance optimization.

Our search for a highly soluble fuel for microfluidic fuel cell applications quickly focused on BH₄⁻, most commonly used as its sodium salt, NaBH₄. This compound has considerable solubility in aqueous solutions (>14 M), a high electron recovery per molecule (8e⁻, Reaction 1), a very low potential for oxidation in base (-1.44 V vs. Ag/AgCl), and extremely fast kinetics at room temperature:



Thus, it has approximately the same low onset potential for oxidation as H₂, allowing for high voltage systems. However, because of its very high solubility, about 14,000 times that of H₂, it can have a theoretical maximum current density that is multiple orders of

magnitude greater. NaBH₄ also has significant specific and volumetric energy density (9.4 kWh/kg, 10 kWh/L for combustion with O₂), exceeding the values for the much more intensively studied fuel methanol (MeOH; 6.1 kWh/kg, 4.8 kWh/L). While MeOH (6e⁻, -0.4 V vs. Ag/AgCl in base) would appear similar to NaBH₄ based on these thermodynamic, theoretical measures, a comparison of activity yields a very different conclusion. Figure 1 shows RDE voltammograms for the room-temperature oxidation of NaBH₄ and MeOH at their best known catalysts, Pt and PtRu, respectively. NaBH₄ exhibits a current density 200 times greater than MeOH, producing 10 times the current at 20 times lower concentration. Thus, it is clear that thermodynamics alone cannot predict the activity of a fuel, and that NaBH₄ represents a significantly more power-dense fuel than either H₂ or MeOH.

Though BH₄⁻ has many attractive features as a fuel, its utilization is hindered by its deleterious hydrolysis reaction, which decomposes the fuel to H₂ and boric acid (Reaction 2).^{1,2} Hydrolysis can proceed partially or completely, converting any number of BH₄⁻'s eight reducing equivalents to H₂ or partial fractions thereof. Hydrolysis occurs readily in bulk solution, in both acidic and neutral aqueous environments, as well as at the surfaces of various catalysts. Hydrolytic decomposition at electrode surfaces is especially detrimental, as H₂ bubbles block electrodes for further reaction, decreasing the effective surface area of fuel cell catalysts, and disrupt laminar flow across the electrode. Once H₂ is formed, the solubility of the reducing equivalents dramatically decreases, eliminating one of BH₄⁻'s principal advantages as a fuel. Thus hydrolysis must be avoided in BH₄⁻ fuel cell technologies.

Prior to our studies, it was generally accepted in the literature that Au was an ideal catalyst for BH₄⁻ oxidation, providing all 8e⁻ while precluding hydrolysis.^{3,4} In contrast, Pt was reported and believed to hydrolyze BH₄⁻ extensively, and often provide only 4 to 6e⁻.⁵ Moreover, all of the low-potential current at Pt was attributed to H₂ produced from electrolysis, and only the current at high potentials was thought to originate from the direct oxidation of BH₄⁻. However, nearly all of these previous

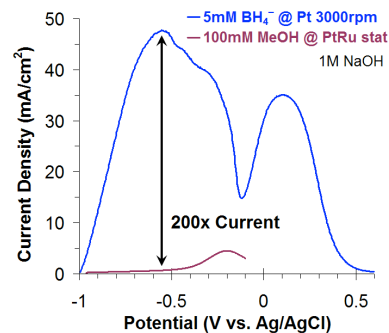


Figure 1: Comparison of 5 mM BH₄⁻ vs. 100 mM MeOH, both in 1 M NaOH

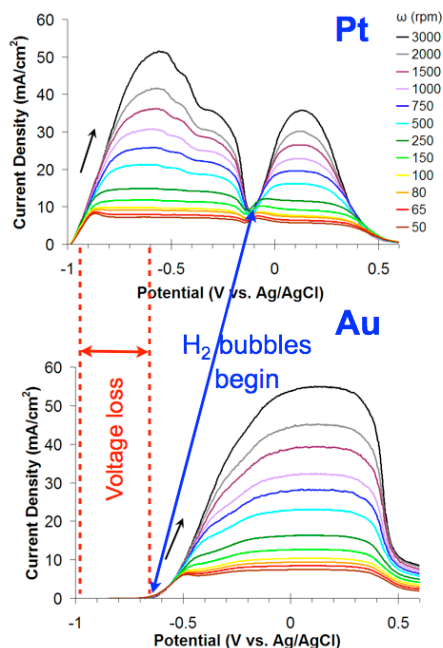
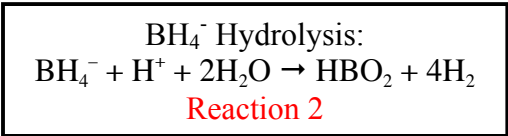


Figure 2: Comparison of BH₄⁻ oxidation at Pt vs. Au. RDE voltammograms of 5mM NaBH₄ in 1M NaOH are shown for each electrode.

studies employed techniques that were not well-suited to establishing mechanistic parameters for fuel oxidation.

To gain an in-depth understanding of the mechanistic details of the oxidation of BH_4^- at Pt and Au electrode catalysts, we employed rotating-disk electrode (RDE) voltammetry, which readily allows distinction between surface-adsorbed and diffusional processes, as well as the determination of kinetic reaction rates and mass transport parameters for a given fuel.⁶ Our RDE study represents the first, and we believe the most rigorous, to date, comparison between Pt and Au catalysts for BH_4^- , and provides a fully different (relative to conventional views) perspective on BH_4^- 's oxidation mechanism.⁷ Two distinct oxidative reactions were observed at Pt: The first occurred at low potentials (between -1 to -0.1 V vs. Ag/AgCl, Figure 2), and corresponded to a $7e^-$ oxidation of BH_4^- , while the second reaction occurred at higher potentials (-0.1 to +0.3 V), and involved only $5.5e^-$. Contrary to literature reports, Pt's low potential current was directly derived from BH_4^- , rather than H_2 , as H_2 's solubility is insufficient to produce the high currents involved. That is, our 5 mM fuel gave $7e^-$ in its oxidation, generating a current corresponding to 35 mM e^- equivalents, while H_2 , with <1 mM solubility and only $2e^-$ available, could produce, at most, a current equivalent to ~ 2 mM e^- equivalents (assuming comparable diffusion coefficients). Additionally, the current from BH_4^- at low potentials scaled perfectly linearly with $[\text{BH}_4^-]$, as the limiting current at 3000 rpm increased from 50 mA/cm² to **0.5 A/cm²**, (specific surface area), when increasing $[\text{BH}_4^-]$ from 5 to 50 mM (Figure 3). Such high current densities and quantitative scaling suggest that direct BH_4^- oxidation at low potentials is even more likely, and the sheer current density available makes the low-potential BH_4^- oxidation at Pt a highly attractive fuel oxidation process to harness.

Au also exhibited two oxidation processes, with the one at low potentials (-0.6 to -0.3 V) yielding about $4.5e^-$, and the one at higher potentials (-0.2 to +0.4 V) providing the expected $8e^-$ (Figure 2). Not only did Pt provide nearly equivalent e^- recovery at lower potentials (which would result in a higher fuel cell voltage) than Au, but the onset potential was about 400 mV more negative. Thus, the use of Pt instead of Au as an electrode will increase the open-circuit voltage of a BH_4^- fuel cell by 400 mV, and its operating voltage (tuned to a potential of maximal e^- recovery; highest power) by 600 mV, while providing close to full recovery of BH_4^- 's $8e^-$.

We found that BH_4^- 's two reaction mechanisms at Pt seem to be intricately tied the state of the Pt surface. A comparison of the anodic sweep of a stationary cyclic voltammogram (CV) of BH_4^- at Pt compared to a CV of polycrystalline Pt in electrolyte alone (1M NaOH) is shown

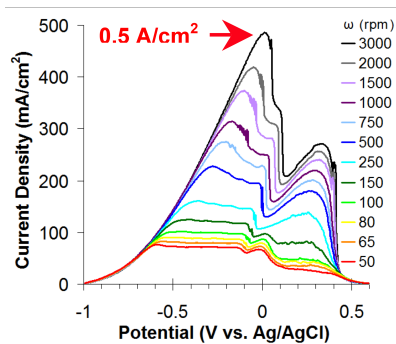


Figure 3: RDE voltammograms for 50 mM NaBH_4 in 1M NaOH at Pt, showing current scaling to 0.5 A/cm².

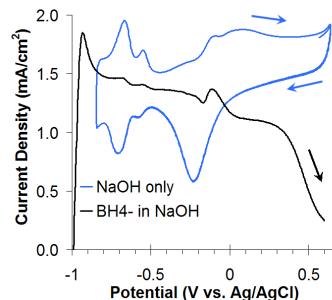
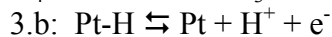
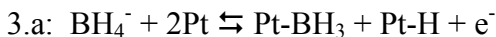


Figure 4: CV's of 5 mM BH_4^- in 1 M NaOH at Pt (black line) vs. CV of Pt in 1 M NaOH without any fuel present (blue line).

in Figure 4. The two small peaks at -0.7 and -0.55 V in BH_4^- 's CV overlap perfectly with the positions for surface hydrides (adsorbed hydrogen) at Pt, suggesting that BH_4^- 's low potential reaction may depend on the formation of surface hydrides as intermediates in its oxidation (Reactions 3.a and 3.b).⁸



Reactions 3.a and 3.b

The peak at -0.1 V in BH_4^- 's CV corresponds very well to OH^- adsorption at Pt. This peak in BH_4^- 's CV is also immediately preceded by a dip in current, which marks the end of the low-potential reaction in the RDE voltammograms (Figure 2). Thus it seems likely that at Pt, at higher potentials, binding of OH^- to the surface decreases the activity of the low-potential BH_4^- oxidation reaction, and that the high-potential reaction proceeds at an OH^- and Pt oxide covered surface.

These mechanistic findings have guided our selection of alternative catalysts for BH_4^- oxidation, as we have targeted metals that support surface hydrogen species, and have paid close attention to effect of oxides on BH_4^- 's oxidation reaction. (see Proposed Research below). They also point to the importance of having a fundamental understanding of the mechanistic details of the electrode reactions in the selection of fuels (and oxidants) for fuel cell applications.

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Much to our surprise, we found that while Au had been widely reported not to be active in BH_4^- 's hydrolysis reaction, we noticed copious H_2 bubble formation at Au's surface as soon as BH_4^- oxidation began (Figure 2). In contrast, BH_4^- oxidation at Pt occurred hydrolysis-free during the low potential reaction, and did not begin until the onset of the higher-potential reaction at -0.1 V. Pt's low-potential reaction is therefore of critical significance to BH_4^- fuel cell operation, as it provides high e^- recovery, high fuel cell voltage (low potential of oxidation), and can proceed free of BH_4^- 's hydrolysis reaction.

Although Pt's low-potential BH_4^- oxidation offers a tremendous gain for BH_4^- fuel cells, it also undergoes an undesirable, though reversible, poisoning reaction. The current was found to decrease by 50-90% within 10-20 min, depending on potential. Similarly, Au also exhibited poisoning behavior with a 40% loss in current over 15 min.

However, we were able to develop a cleaning protocol, based on our RDE study of BH_4^- at Pt and Au.⁷ We found that by briefly stepping the potential to +0.60V we could recover virtually all of the electrocatalytic activity. In essence, the $7e^-$ BH_4^- oxidation at Pt was well-maintained for over 20 min, while stepping positive for only 0.5 s every 300s (Figure 5). That is, the potential of the electrode remained at -0.7 V for 600 times the length of time it was held at +0.6 V. The procedure was also successfully applied to an Au electrode held in Au's low potential region (-0.4 V), though the electrode spent only 200 times as long at low potential as it did at +0.6 V. The preliminary demonstration of these cleaning techniques in an actual BH_4^- fuel cell is discussed in Section E, Performance of $\text{BH}_4^- / \text{CAN}$ and $\text{BH}_4^- / \text{Cr}_2\text{O}_7^{2-}$ fuel cells, below, and suggests that *in-situ* cleaning is instrumental to operation of BH_4^- fuel

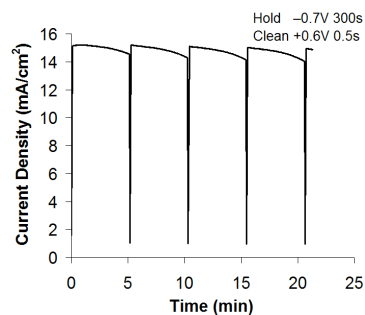


Figure 5: Long-term activity under pulse-cleaning of 5mM BH_4^- in 1M NaOH at Pt, 250 rpm.

cells at high voltage.

Though BH_4^- exhibits appreciable performance at Pt, the cost of a Pt-based fuel cell system will always be of concern. Fortunately, BH_4^- is known to exhibit activity at a number of less precious metals, such as Pd, Ag, Ni, and even misch metals.⁹ This feature gives BH_4^- fuel cells an important cost flexibility that is simply unachievable for H_2 or MeOH systems. Preliminary results of BH_4^- oxidation at alternative metal catalysts is discussed below under Proposed Studies.

B. Oxidation of BH_4^- in nonaqueous solvents as a means to address BH_4^- hydrolysis:

While we were able to overcome BH_4^- hydrolysis at electrode catalyst surfaces via the employment of Pt electrodes at low potential, hydrolysis in bulk solution still occurred at high concentrations of BH_4^- (>0.15 M), even in 3M NaOH. This strictly limited the concentrations we were able to test in our fuel cell. Given that NaBH_4 has an aqueous solubility of 14.5 M,¹⁰ the current and power densities observed (see Section E below) represent about 1% of that theoretically obtainable from BH_4^- .

In an attempt to boost the concentration of BH_4^- in our fuel cell and avoid hydrolysis altogether, we studied the oxidation mechanism of BH_4^- in nonaqueous solvents.¹¹ First, we assessed a first-order solubility of NaBH_4 in a number of solvents of varying polarity (Table 1). We found that many solvents would solubilize NaBH_4 above our target of 0.15 M, but most were incompatible with NaBH_4 . In accord with literature reports, MeOH and EtOH directly reacted with BH_4^- , and performed methanolysis and ethanolysis, respectively, of BH_4^- to H_2 .^{12,13} These alcohols therefore supplant BH_4^- hydrolysis with equally deleterious reactions. Though NaBH_4 is "soluble" in multiple glymes,¹⁴ we found that it made a cloudy suspension, and we concluded BH_4^- would have greatly diminished activity if not fully dissolved. Most of the other, predominantly hydrophobic solvents (i-PrOH, THF) did not appreciably solubilize NaBH_4 . That left us with just DMSO and DMF as prospective solvents for NaBH_4 , with solubilities three times higher than we could readily achieve under aqueous conditions (0.42 and 0.43 M, respectively, vs. 0.15 M in water).

After finding an appropriate solvent window of stability, NaBH_4 was found to exhibit a well-behaved oxidation at Pt in DMSO, without any interference from coupled chemical reactions (Figure 6). The reaction was similar at Au, but with slightly higher overpotential for reaction. In contrast

Table 1: Solubilities of NaBH_4 in organic solvents of varying polarity.

Solvent	BH_4^- Solubility (M)	Comments
DMSO	0.42	
DMF	0.43	
Acetonitrile	-	not soluble
Diglyme	0.46	suspension
Ether	-	not soluble
Methanol	0.33	bubbles
Ethanol	0.49	bubbles
Isopropanol	-	not soluble
1-propanol	-	not soluble
THF	-	not soluble

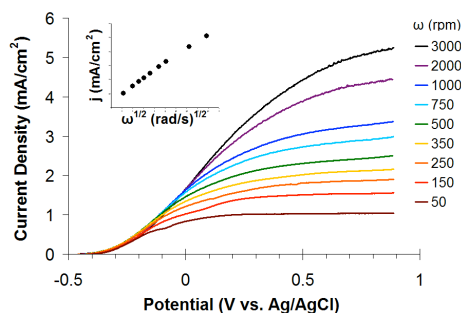


Figure 6: RDE voltammograms of 5mM NaBH_4 in DMSO at Pt, with 1M NaNO_3 supporting electrolyte.

to DMSO, it was very difficult to obtain reproducible voltammograms in DMF, and slow poisoning over time seemed evident.

Overall, the currents obtained in nonaqueous solvents were disappointingly low, corresponding to an electron recovery of only about $1e^-$ from BH_4^- , and overpotentials were 0.3 to 0.7 V higher (less favorable) than those observed in base. Voltammograms of 5mM BH_4^- at Pt and Au in 1M NaOH are contrasted with those obtained in DMF and DMSO in Figure 7. The observed currents were ten times lower in some cases, reflecting poor diffusivity of BH_4^- in the solvents in addition to BH_4^- 's lower electron recovery. Literature suggested that methoxy ligands may help stabilize boron in higher oxidation states, and possibly improve electron recovery.¹⁵ However, inclusion of 5 and 30 mM MeOH with 5 mM BH_4^- did not significantly affect current density.

In addition to these challenges with low current recovery, BH_4^- decomposition to H_2 remained a challenge. While BH_4^- decomposition (hydrolysis) in bulk solution was mitigated via implementation of nonaqueous solvents, BH_4^- still reacted to form H_2 at Pt and Au surfaces during oxidation, with visible H_2 bubble formation. At the moment, nonaqueous solvents do not seem to be a promising avenue for BH_4^- fuel cell research, but further studies in solvents not examined here may help address issues with hydrolysis and decomposition in BH_4^- fuel cells. We also propose physical methods – pressure and confinement – to extend the range of stable concentrations of BH_4^- in aqueous electrolyte.

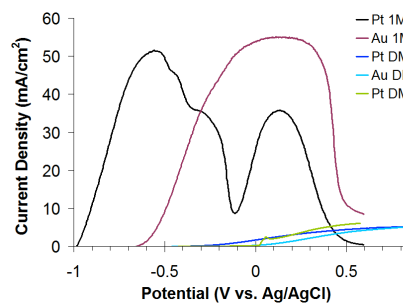


Figure 7: Activity comparison of 5mM BH_4^- in 1M NaOH_(aq), DMSO, and DMF at Pt and Au. RDE voltammograms at 3000 rpm are shown.

C. Comparison of previously employed, alternative oxidants for high-power and laminar flow fuel cells

BH_4^- is capable of significant current density in aqueous solutions, producing **0.5 A/cm²** from just 50 mM fuel when convected (Figure 3, 3000 rpm). In order to balance this reaction in the fuel cell, using a cathode of similar size, highly soluble, kinetically fast oxidants must be employed. O_2 , with a maximum solubility of just 1.3 mM, produces a current of around 6 mA/cm² at 3000 rpm, falling about two orders of magnitude short of BH_4^- 's current. In order to find a more suitable oxidant for BH_4^- , we examined a number of alternative oxidants that have been employed in fuel cells previously.⁷ While various performance parameters for many of these oxidants have already been established, the literature was lacking in a uniform comparison of oxidants tested under identical conditions. Our goal was to provide not only mechanistic information for the various oxidants, but to make a valuable reference highlighting the strengths and weaknesses of each oxidant, allowing other fuel cell researchers to quickly

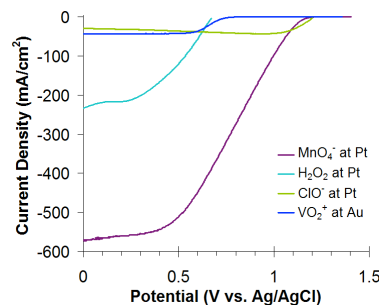
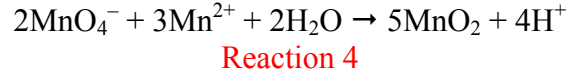


Figure 8: Metacomparison of previously employed oxidants. RDE voltammograms for each oxidant at its best catalyst are shown for 100 mM oxidant in 0.5M H_2SO_4 , rotated at 3000 rpm.

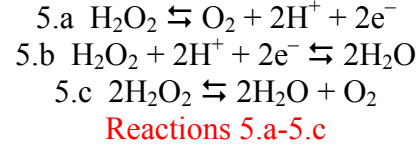
compare among fuel cell chemistries.

The oxidants permanganate (MnO_4^-), peroxide (H_2O_2), dioxovanadium (VO_2^+), and hypochlorite (bleach, ClO^-) were studied and compared to O_2 . The RDE voltammograms for the reduction of the oxidants at their best catalysts and uniform concentration (100 mM), electrolyte (0.5 M H_2SO_4), and rotation rate (3000 rpm) are shown in Figure 8. Oxidant reduction reactions are shown in Table 2, and the oxidants' mass-transport parameters are summarized in the upper half of Table 3. Each oxidant had a unique blend of transport properties, kinetic character, and mechanistic complexity.



MnO_4^- had the highest potential and current density of any of the oxidants studied, and was the only oxidant that, at 100 mM, could potentially match BH_4^- 's current density at 50 mM. If paired with BH_4^- , MnO_4^- 's onset potential of +1.2 V would result in an open-circuit voltage of **2.2 V** with a current density **> 0.5 A/cm²**, specific surface area, with convected fuel of just 0.1 M (Figure 8).

Unfortunately, MnO_4^- suffers from a precipitation reaction, which quickly occludes the electrode surface for further reaction at high current densities. It is possible that MnO_4^- could



be directly reduced to MnO_2 , but this seems unlikely, as this would involve a 3e^- transfer (Table 2), while the RDE data was consistent with a 5e^- reduction (Table 3). The precipitation is most likely due to MnO_4^- 's 5e^- reduction product, Mn^{2+} (Table 2), reacting with excess MnO_4^- to form the intermediate compound MnO_2 , a brown precipitate (Reaction 4).^{16,17}

The concept that the precipitation results from side reactions of MnO_4^- 's reduction product, and is not an inherent feature of MnO_4^- reduction, implies that various measures could be taken to avoid precipitation.

For example, sequestration of Mn^{2+} via chelators or other materials would prevent the reaction between Mn^{2+} and MnO_4^- . Our characterization of MnO_4^- 's reduction mechanism to confirm the 5e^- pathway has therefore given direction to new approaches for addressing MnO_4^- reduction, and utilizing this remarkably potent oxidizer.

Table 2: Reduction reactions for previously employed and novel oxidants.

Oxidant	Possible Reactions:	E ⁰ (V vs. Ag/AgCl)
O_2	$\text{O}_2 + 2\text{H}^+ + 2\text{e}^- \rightleftharpoons \text{H}_2\text{O}_2$	0.498
	$\text{O}_2 + 4\text{H}^+ + 4\text{e}^- \rightleftharpoons \text{H}_2\text{O}$	1.032
MnO_4^-	$\text{MnO}_4^- + \text{e}^- \rightleftharpoons \text{MnO}_4^{2-}$	0.361
	$\text{MnO}_4^- + 4\text{H}^+ + 3\text{e}^- \rightleftharpoons \text{MnO}_2 + 2\text{H}_2\text{O}$	1.482
	$\text{MnO}_4^- + 8\text{H}^+ + 5\text{e}^- \rightleftharpoons \text{Mn}^{2+} + 4\text{H}_2\text{O}$	1.310
H_2O_2	$\text{H}_2\text{O}_2 + 2\text{H}^+ + 2\text{e}^- \rightleftharpoons 2\text{H}_2\text{O}$	1.579
VO_2^+	$\text{VO}_2^+ + 2\text{H}^+ + \text{e}^- \rightleftharpoons \text{VO}^{2+} + \text{H}_2\text{O}$	0.794
	$\text{VO}^{2+} + 2\text{H}^+ + \text{e}^- \rightleftharpoons \text{V}^{3+} + \text{H}_2\text{O}$	0.140
ClO^-	$\text{HClO} + \text{H}^+ + 2\text{e}^- \rightleftharpoons \text{Cl}^- + \text{H}_2\text{O}$	1.285
	$\text{HClO} + \text{H}^+ + \text{e}^- \rightleftharpoons \frac{1}{2}\text{Cl}_2 + \text{H}_2\text{O}$	1.414
$\text{Ce}(\text{NH}_4)_2(\text{NO}_3)_6$	$\text{Ce}^{4+} + \text{e}^- \rightleftharpoons \text{Ce}^{3+}$	1.523
$\text{S}_2\text{O}_8^{2-}$	$\text{S}_2\text{O}_8^{2-} + 2\text{H}^+ + 2\text{e}^- \rightleftharpoons 2\text{HSO}_4^-$	1.926
IO_3^-	$\text{IO}_3^- + 6\text{H}^+ + 6\text{e}^- \rightleftharpoons \text{I}^- + 3\text{H}_2\text{O}$	0.888
IO_3^-	$\text{IO}_3^- + 6\text{H}^+ + 5\text{e}^- \rightleftharpoons \text{I}_2 + 3\text{H}_2\text{O}$	0.998
HCrO_4^-	$\text{HCrO}_4^- + 7\text{H}^+ + 3\text{e}^- \rightleftharpoons \text{Cr}^{3+} + 4\text{H}_2\text{O}$	1.153
$\text{Cr}_2\text{O}_7^{2-}$	$\text{Cr}_2\text{O}_7^{2-} + 14\text{H}^+ + 6\text{e}^- \rightleftharpoons 2\text{Cr}^{3+} + 7\text{H}_2\text{O}$	1.035

H₂O₂, like O₂, is only reduced at Pt after Pt's oxide has undergone significant reduction/removal at about +0.65 V vs. Ag/AgCl. At potentials above this value, H₂O₂ is oxidized to O₂. As a result, H₂O₂ provides about 0.5 V less voltage than other oxidants studied. During reduction, H₂O₂ undergoes a catalytic decomposition to O₂ (Reactions 5.a-5.c).¹⁸ This decomposition vigorously produces O₂ gas bubbles at the surface, causing the same disruptions in fluid flow and available electrode area discussed previously for H₂ production during BH₄⁻ oxidation. The resulting decrease in current and pressurization in a fuel cell makes H₂O₂ very difficult to implement, and our initial fuel cell studies using H₂O₂ were plagued with low current densities and leaks. Aside from this side reaction, H₂O₂ has a high diffusion coefficient and two e⁻ per molecule, resulting in a substantial current density of **0.22 A/cm²**. Because H₂O₂ is fully miscible in water, and is a liquid in its pure state, it can be brought to very high concentration, which would otherwise allow for a significant scale-up of its current density in a practical device. In moving forward, as with BH₄⁻, we propose strategies to exploit H₂O₂ at high concentrations.

VO₂⁺ is a specialty oxidant seen essentially exclusively in vanadium "redox batteries," or reversible vanadium fuel cells.¹⁹ The primary advantage of such systems is the reversibility of the redox reactions and the simplicity of having vanadium in both positive (VO₂⁺ and VO²⁺) and negative (V⁺³ and V⁺²) half-cell reactions. Aside from its ability to be used in reversible charge-storage systems, VO₂⁺ is lacking in many important features for fuel cell applications. It participates in just a 1e⁻ transfer reaction and has low solubility (< 3 M, Table 3), and so has a strictly limited maximum possible current density. We also found VO₂⁺ to have a remarkably low diffusion coefficient²⁰ that is four times lower than any of the other oxidants studied, and is two to six times lower than most known soluble species in general.¹⁰ This combination of mass transport parameters gave VO₂⁺ the lowest current density of the alternative oxidants in our comparison. Factoring in VO₂⁺'s relatively lower onset potential for reduction (about 0.5 V lower than MnO₄⁻, Figure 8), it is unlikely that VO₂⁺ will play a role in high power density fuel cells.

Table 3: Mass transport parameters relevant to current density for previously employed and novel oxidants.

	Oxidant	# of e ⁻	Diffusion Coefficient (x10 ⁻⁵ cm ² /s)	Solubility (M)	Current Density* (mA/cm ²)	Complication
Previously Employed Oxidants Increasing Current Density ↑	NaMnO ₄	5	1.2	7.3	560	Precipitation
	H ₂ O ₂	2	1.5	42.4	220	Bubbles
	VO ₂ ⁺	1	0.25	< 3 [†]	43	Reversible Poisoning at Au
	NaClO	1?	1.1	10.7	> 50	Irreversible Poisoning at Pt, Au, GC
	O ₂	4	1.3	< 0.001	6.2	
Novel Fuel Cell Oxidants Increasing Current Density ↑	Na ₂ Cr ₂ O ₇	6	0.8	7.1	540	Oscillations, Poisoning
	NaIO ₃	5-6?	1.1	0.5	340	
	Na ₂ CrO ₄	3	0.92	5.4	290	
	Na ₂ S ₂ O ₈	2	0.6	2.3	150	
	CAN	1	0.42	2.6	57	

The mass-transport limited current density for 100 mM of the given oxidant at 3000 rpm is shown.
VO₂⁺ is typically generated from the precursor VO²⁺, so the solubility for VOSO₄ is shown.

That said, we did find that the power density of vanadium redox battery systems can probably be increased significantly by switching cathode material from glassy carbon (GC) to Au. Most vanadium systems use carbon-based cathodes to decrease cost. However, the onset potential and $E^{1/2}$ for reduction are shifted positive by 0.6 V at Au (Figure 9), indicating a 0.6 V gain in both open-circuit and operating voltages for vanadium fuel cells that use Au cathodes. Previously, it was determined that VO_2^+ would slowly poison Au cathodes,²⁰ but we discovered that occasionally pulsing the electrode to high potentials ($> +0.6$ V vs. Ag/AgCl, 0.5 M H_2SO_4) would remove surface-bound poisons and fully restore current density in a manner analogous to our findings with BH_4^- .

ClO^- has only recently appeared as a fuel cell oxidant,^{21,22} and we were understandably skeptical of its effectiveness. It is expected for ClO^- to be reduced to either $\text{Cl}_{2(g)}$ or Cl^- (Table 2), the latter of which is a well-known poison for noble metals. Poisoning was indeed evident in our study. In fact, ClO^- reduction poisoned Pt so quickly that we were unable to complete a single RDE voltammogram without significant loss of current during the scan (Figure 8). ClO^- reduction poisoned Au as well, and even glassy carbon. It is unclear to us why the poisoning was not noticed by other researchers. It is possible that, in some cases, the electrodes used may have been pre-poisoned by the use of Cl^- electrolytes.²¹ Other fuel cells used high-surface area electrodes,²² and complete poisoning may not have occurred on the time scale of the experiment.

Aside from its undesirable reaction products, ClO^- reduction served as a potent process. The reduction itself occurred at exceptionally high potentials at Pt (+1.2 V, Figure 10), practically at the limit of an aqueous system, allowing for a very high voltage fuel cell. Though only $1e^-$ is involved in the reduction, NaClO has very high solubility (10.7 M, Table 3), which would allow for significant scale-up of the process.

Unlike MnO_4^- , unfortunately, ClO^- 's poison formation is inherent to the reduction of the oxidant itself, and will probably be difficult to avoid. We do not think that ClO^- will be a candidate oxidant for future fuel cells.

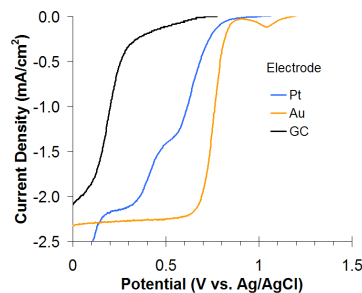


Figure 9: RDE voltammograms of 5mM VO_2^+ in 0.5M H_2SO_4 at Pt, Au, and glassy carbon, 3000 rpm, cathodic sweeps at 20 mV/s.

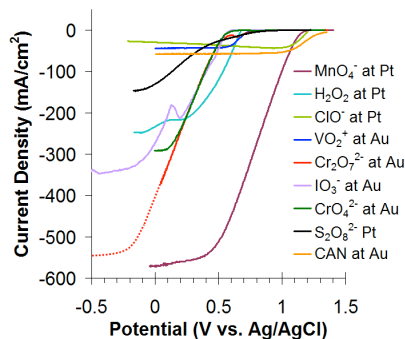


Figure 10: Metacomparison of conventional and new oxidants. RDE voltammograms for 100 mM oxidant at its best catalyst in 0.5M H_2SO_4 , rotated at 3000rpm, are shown.

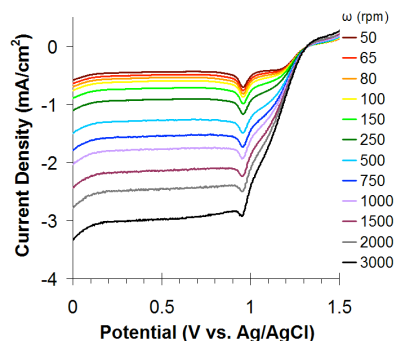


Figure 11: RDE of 5mM CAN in 1M HNO_3 at Au.

D. Investigation of novel, superior oxidants for high-power and laminar flow fuel cells

Because none of the alternative oxidants studied were ideal, we instead turned to novel oxidants not previously employed in fuel cell devices. From an array of known oxidant reduction reactions, we chose those that we thought would be the most promising. Cerium ammonium nitrate, $\text{Ce}(\text{NH}_4)_2(\text{NO}_3)_6$ (hereafter CAN), and persulfate, $\text{S}_2\text{O}_8^{2-}$, both have exceptionally high potentials for reduction, and could possibly contribute to high voltage fuel cells (Table 2). CAN was also reported to have a facile kinetic rate of reduction,²³ so it is expected to deliver higher current with a minimal voltage penalty. Based on our studies of alternative oxidants, the number of e^- transferred in the reaction is the most critical factor in determining the limiting current density of a given oxidant. Thus iodate, chromate, and dichromate were chosen because they undergo reductions involving 3 or more electrons (Table 2), and so should provide significant current densities. The RDE voltammograms for the various novel oxidants are shown alongside those for the previously employed oxidants in Figure 10.

CAN was one of the best behaved oxidants we worked with, and as a result, was the most extensively studied for our fuel cell work (see Section E, Performance of $\text{BH}_4^- / \text{CAN}$ and $\text{BH}_4^- / \text{Cr}_2\text{O}_7^{2-}$ fuel cells). At Au, CAN's reduction began immediately negative that of destructive Au etching and dissolution (about +1.3 V), providing a high potential (high voltage) reaction at the limit of the electrode's stability in aqueous solution (Figure 11). Kinetics were rapid, as described in the literature.²³ Perhaps most importantly, the reduction of CAN did not show any interference whatsoever from poisoning or other disadvantageous side reactions, allowing for immediate implementation of CAN into laminar flow fuel cells.

Though CAN's expected voltage is high, its gravimetric energy density and current density are low. For the weight of a lanthanide series metal, only $1e^-$ is delivered (Table 2). In addition to this low electron delivery, CAN also has a diffusion coefficient two and a half times lower than the other oxidants studied ($\sim 0.42 \times 10^{-5} \text{ cm}^2/\text{s}$ ^{23,24}, Table 3), on par with VO_2^+ s. Also like VO_2^+ , CAN has a low solubility, precluding effective scale-up of the oxidant. While CAN proved easy to work with in our fuel cell, oxidants with higher current densities were sought to further increase power density.

With an E^0 of +1.926 V, $\text{S}_2\text{O}_8^{2-}$ had the highest standard potential of any oxidant studied, and combined with its $2e^-$ reaction (Table 2), it appeared very promising for delivering both voltage and

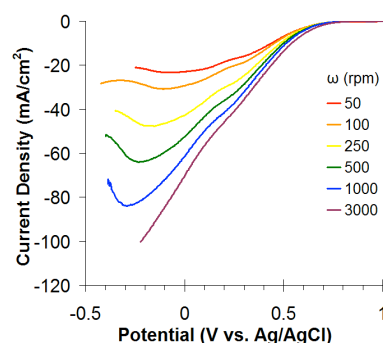


Figure 12: RDE voltammograms of 100mM $\text{S}_2\text{O}_8^{2-}$ in 0.1M HClO_4 at Au.

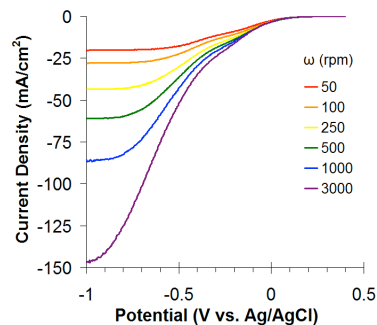


Figure 13: RDE voltammograms of 100 mM $\text{S}_2\text{O}_8^{2-}$ in 1M NaOH at Pt.

current. However, like many of the compounds we studied (such as BH_4^- at Au and H_2O_2 at Pt), the thermodynamics were not realized in an electrochemical setting. At Au in 0.1 M HClO_4 , $\text{S}_2\text{O}_8^{2-}$ does not begin its reduction until +0.7 V (Figure 12), consistent with literature results,²⁵ and its reaction kinetics were rather slow at every pH, electrolyte, and catalyst studied. Utilizing $\text{S}_2\text{O}_8^{2-}$ reduction is difficult, as it is thought to undergo a dissociative adsorption to form adsorbed SO_4^{2-} at Pt surfaces, and the adsorbed SO_4^{2-} blocks Pt for further reaction until SO_4^{2-} is removed at very low potentials.^{26,27} Our data supported this concept, showing mostly a self-poisoning peak at Pt. Some researchers have speculated that OH^- enhances $\text{S}_2\text{O}_8^{2-}$'s reduction reaction,^{25,27} and our results confirmed that $\text{S}_2\text{O}_8^{2-}$ reduction will proceed to a mass-transport limited current at Pt in 1 M NaOH (Figure 13). Of course, switching from acidic to basic medium shifts pH dependent reactions negative by ~ 0.8 V (0.059 V per pH unit via the Nernst equation²⁸), so the potential of the reduction becomes significantly less attractive using this approach.

Despite these drawbacks, $\text{S}_2\text{O}_8^{2-}$, like CAN, did not appear to undergo any deleterious side reactions or poisoning at Au in acid, and can therefore find immediate utility in fuel cells. Though its onset potential for reduction was much lower than expected, it was still reasonably high, on par with H_2O_2 and O_2 , and could produce significantly higher current than CAN (Figure 10). Many fuel cell researchers currently using H_2O_2 , who are frustrated with O_2 bubble formation, can obtain trouble-free operation with only a moderate decrease in activity by switching to $\text{S}_2\text{O}_8^{2-}$, especially if low overpotentials are used.

IO_3^- was a very attractive oxidant, with expected reductions involving 5 and 6e⁻ (Table 2). Indeed, IO_3^- produced the third-highest limiting current of any of the oxidants studied (0.35 A/cm², Figure 10). Complications quickly arose, though, when assessing current delivered over time. Oscillatory behavior manifested, in which the reaction seemed to switch between the 5 and 6e⁻ processes. The process repeated in a cyclical pattern for several minutes, while the overall current slowly diminished. A similar process has been described under galvanostatic conditions,²⁹ and may be akin to the Briggs-Rauscher "iodine clock" experiment.

The highest-impact findings of our exhaustive search through novel oxidant chemistries were, by far, the massive current densities afforded by the two chromium species, chromate, CrO_4^{2-} , and dichromate, $\text{Cr}_2\text{O}_7^{2-}$ (Figure 10). The activities (onset potential, kinetic rate of reaction, etc.) were identical, with the exception that $\text{Cr}_2\text{O}_7^{2-}$ produced about double the current density of CrO_4^{2-} , in accord with $\text{Cr}_2\text{O}_7^{2-}$'s doubled number of accepted electrons (6 vs. 3e⁻). $\text{Cr}_2\text{O}_7^{2-}$ is unique as an oxidant, possessing two transition metal atoms in the same molecule, doubling the current density at a given concentration while preserving fast kinetics. Indeed, we found $\text{Cr}_2\text{O}_7^{2-}$'s kinetics to be immeasurably fast by RDE methodology, with RDE voltammograms appearing as, essentially, step functions instead of smooth waves (Figure 14).

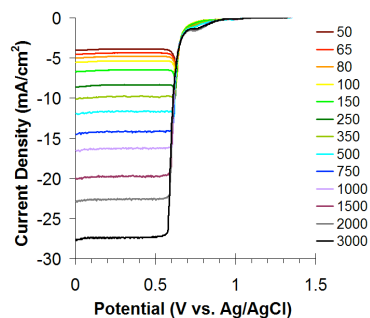


Figure 14: RDE voltammograms of 5mM $\text{Cr}_2\text{O}_7^{2-}$ in 0.5M H_2SO_4 at Au.

Like CAN, reduction of $\text{Cr}_2\text{O}_7^{2-}$ does not appear to have any poisoning or adverse side reactions, and we were able to observe sustained current densities $> 0.4 \text{ A/cm}^2$, specific surface area, for 20 min. Because $\text{Cr}_2\text{O}_7^{2-}$ accepts 6 times the number of e^- as CAN, and has a diffusion coefficient about twice as large, its current density is nearly ten times greater (Figure 14 vs. Figure 11). Though current is significantly higher, $\text{Cr}_2\text{O}_7^{2-}$ manifests an onset potential about 0.5 V lower (less favorable) than CAN. When coupling to a low-potential fuel like BH_4^- (-1 V), however, this may represent a voltage drop of only about 20-30% for an order of magnitude gain in cathodic current.

In our search of novel oxidants, $\text{Cr}_2\text{O}_7^{2-}$ was the oxidant most comparable to MnO_4^- , providing a similar current density of $\sim 0.5 \text{ A/cm}^2$ at 0.1 M concentration (Figure 10) to match BH_4^- 's current density at just 0.05 M (Figure 3). Though $\text{Cr}_2\text{O}_7^{2-}$'s reduction does not begin until 0.5 V more negative (less favorable) than MnO_4^- 's, the reaction is free of poisoning and side reactions, allowing it to be implemented today without further investigation or development. Though we have not studied it as thoroughly in our fuel cell as CAN, as we discovered $\text{Cr}_2\text{O}_7^{2-}$ later in our oxidants search, our preliminary results are in agreement with our analytical RDE investigation, and we are continuing to study fuel cell operation at higher concentrations and power densities (see Section E, Performance of $\text{BH}_4^- / \text{CAN}$ and $\text{BH}_4^- / \text{Cr}_2\text{O}_7^{2-}$ fuel cells).

E. Performance of $\text{BH}_4^- / \text{CAN}$ and $\text{BH}_4^- / \text{Cr}_2\text{O}_7^{2-}$ fuel cells

Our $\text{BH}_4^- / \text{CAN}$ fuel cell combines two reactions occurring at the limit of solvent stability to manifest very high voltages in fuel cells. The convection in our laminar flow fuel cell allows for the use of dual (asymmetric) electrolytes, which can result in a significant increase in the open circuit potential. This arises from the fact that at high pH the potential for fuel oxidation is shifted negative, while in acid the potential for oxidant reduction is shifted positive. With our fuel in 1 M NaOH and our oxidant in 1 M HNO_3 , the 14 unit pH difference further increases the voltage by $\sim 0.84 \text{ V}$, as predicted by the Nernst equation ($0.059 \text{ V/pH unit}^{28}$). Thus we achieved open-circuit potentials of $> 2 \text{ V}$ (Figure 15), about 0.8 V greater than the traditional 1.2 V limit obtainable in aqueous solutions. This open-circuit potential value is in very close agreement with the difference in analytically obtained onset potentials for fuel and oxidant (-1 V for BH_4^- at Pt in 1 M NaOH, +1.2 V for CAN in 1 M HNO_3 , Figures 2 and 11).

Under conditions of laminar flow, using just 0.15 M fuel and 0.5 M oxidant, we achieved a current density of 0.15 A/cm^2 (Figure 15) and a power density of 0.125

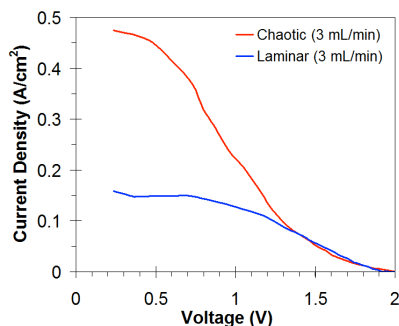


Figure 15: Load Curves for fuel cell using 0.15M NaBH_4 in 3M NaOH at Pt, 0.5M CAN in 1M HNO_3 at Pt.

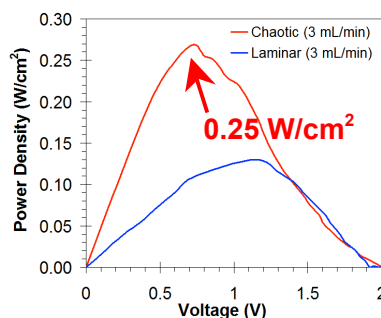


Figure 16: Power Curves for fuel cell using 0.15M NaBH_4 in 3M NaOH at Pt, 0.5M CAN in 1M HNO_3 at Pt.

W/cm² (Figure 16). Using the chaotic mixers described in Section F, mass transport to the electrodes tripled, in agreement with our theoretical predictions,³⁰ and therefore tripled our transport-limited current density to nearly **0.5 A/cm²**. However, because more voltage had to be sacrificed to drive reaction rates faster to support this current density, via Butler-Volmer kinetics,²⁸ the power doubled, instead of tripled, to **0.25 W/cm²**.³¹ This, however, represents a remarkable achievement in terms of power density.

Performance comparison with other high power fuel cells in the literature is difficult, as other groups use high-surface area electrodes but normalize to geometric area. By normalizing outputs to catalyst loading, our system achieved the highest power density yet recorded for a non-H₂ fuel cell. Whereas MeOH and BH₄⁻ fuel cells use 1.5-3 M fuel at 60°C and provide up to 30 and 220 mW/mg catalyst, respectively,^{9,32} our system used 0.15 M fuel at room temperature and produced **1,230 mW/mg Pt** (Figure 17). This value represents a **50% increase** in Pt utilization from a typical H₂/O₂ fuel cell (800 mW/mg Pt³³).

The preliminary results from our BH₄⁻/Cr₂O₇²⁻ fuel cell indicate that this power density can be pushed many times higher. Under conditions of laminar flow, and fuel limitation, 15 mM BH₄⁻ in 1 M NaOH at Pt and 20 mM Cr₂O₇²⁻ in 0.5 M H₂SO₄ at Au generated a current density of 60 mA/cm² and 50 mW/cm² (Figures 18 and 19, respectively). If these performance values scale proportionately when increasing the concentrations of the fuel and oxidant ten times to 0.15 and 0.2 M, similar to the BH₄⁻ / CAN system, it would produce 0.5 W/cm² under laminar flow. In our proposed work, we aim to stabilize BH₄⁻ out toward its solubility limit (14 M) and reduce the internal resistance of our planar FC. In this limit, we can look toward current and power densities upward of 10 A/cm² and 10 W/cm².

Furthermore, in our BH₄⁻/Cr₂O₇²⁻ fuel cell, we performed a cursory investigation of pulse cleaning to avoid BH₄⁻ poisoning at Pt, allowing for sustained 7e⁻ current at low anode potential (high fuel cell voltage), as described in Section A, “*Electroanalytical investigation of BH₄⁻*”. Repeatedly pulsing the potential of the anode to the region of Pt oxide formation successfully enabled sustained currents of 50-60 mA/cm² for about 60 min, as predicted by our analytical results (Figure 5). We will perform longer-term studies of BH₄⁻ fuel cells under pulse cleaning to further demonstrate durability and reliability of the fuel oxidation reaction.

It is notable that pulse cleaning of our BH₄⁻/Cr₂O₇²⁻ fuel cell does not consume any outside energy. In Figure 18, the fuel cell load curve closely mirrors the RDE voltammogram of BH₄⁻ oxidation seen in Figure 2, indicating that the fuel cell is

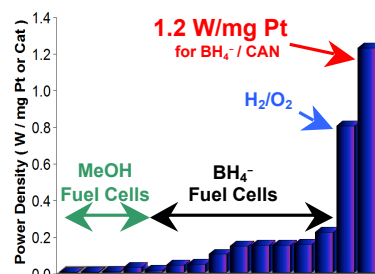


Figure 17: Specific power density normalized to weight of catalyst for BH₄⁻ / CAN fuel cell vs. other reported and common fuel cells.

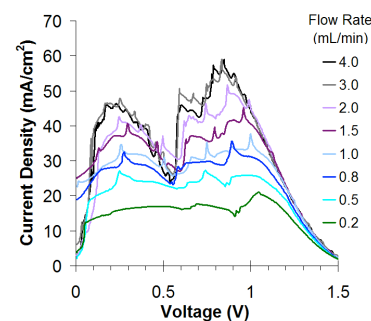


Figure 18: Load curves for fuel cell using 0.015M NaBH₄ in 3M NaOH at Pt / 0.02M Na₂Cr₂O₇ in 0.5M H₂SO₄ at Au. Laminar flow only.

completely fuel limited at all potentials. In this configuration, the high potential of the oxidant can pull the potential of the anode well into an electrode-cleansing region of Pt oxide, which occurs near the short-circuit current (0 V) in Figure 18. Thus only temporarily decreased current, rather than outside energy, is all that is necessary for the fuel cell to self-cleanse.

F. Enhancement of rates with three-dimensional laminar flows

Our analysis predicts^{34,35} that the introduction of three-dimensional (3-D) flows within laminar systems can provide significantly higher rates of interfacial transfer. To test our predictions quantitatively, we have worked in microfluidic potential cells (Figure 20(a)) with structured electrodes (Figure 20(b)). Figure 20(c) presents the rates of mass transfer as the average Sherwood number, Sh_{ave} ($Sh = k_M H/D$, where k_M is the mass transfer coefficient, H is the characteristic dimension of the channel, and D is the diffusivity of the reactive species). Measurements are shown for smooth electrodes (triangles), grooved electrodes that generate non-chaotic flow (circles), and grooved electrodes that generate chaotic flow (crosses and squares)³⁶. For the stirred, non-chaotic flow, the grooved boundary acted as the counter electrode (circles); for the chaotic flow, the grooved boundary acted as both the counter (crosses) and the working electrode (squares). These measurements demonstrate the following important points: 1) the induction of either chaotic or non-chaotic flow adjacent to a reactive boundary can significantly increase transport-limited current relative to an unstirred, laminar flow. When the stirring is induced from the counter, this increase is ~ 2 -fold (circles and crosses vs. triangles); when the stirring is induced from the working electrode, the increase is ~ 8 -fold (squares vs. triangles). 2) In stirred flows, the entrance length is short and Sh goes to a Pe -dependent, asymptotic value, Sh_{aym} that can be much greater than one. This behavior can be seen by the absence of the constant region at low Pe for the stirred flows. 3) The measured rates match quantitatively with the predictions of our modified Grätz theory^{34,35} with no adjustable parameters: the prefactors of the best fits lines in Figure 20(c) match the predictions within 20%; the exponents ($-1/3$) are those predicted.

This solid theoretical basis and the extremely high rates achievable with structured electrodes will serve us well as we develop methods to exploit our fast, high concentration fuels and oxidants. We note that for $Sh_{ave} = 40$ in a channel of depth, $H = 50 \mu\text{m}$ and a typical diffusivity ($D = 10^{-9} \text{ m}^2/\text{s}$), we can achieve a mass transfer coefficient, $k_M = 8 \times 10^{-2} \text{ cm/s}$, a value that is comparable the kinetic rate constants measured for our fast fuels and oxidants ($k > 0.02 \text{ cm/s}$)^{6,7}. To take full advantage of these rates and the high energy densities of our fuels and oxidants in this format, we propose to develop methods to stabilize high concentration solutions of BH_4^- (and H_2O_2) and incorporate rigid flow separators with high ionic conductance.

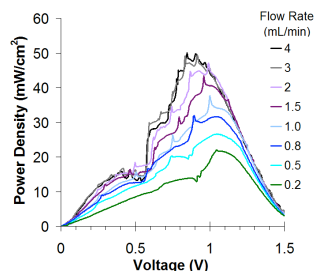


Figure 19: Power curves for fuel cell using 0.015M NaBH_4 in 3M NaOH at Pt / 0.02M $\text{Na}_2\text{Cr}_2\text{O}_7$ in 0.5M H_2SO_4 at Au. Laminar flow only.

G. Reactive bubbly flow in Fuel Cells

In meeting the challenge of combining high power density with high fuel efficiency, the fuel/oxidant couple $\text{NaBH}_4/\text{H}_2\text{O}_2$ is a very good candidate with a theoretical high open circuit voltage of 3V (Reaction 1, Table 2) and high current density (Figures 3 and 8). However, as mentioned earlier, H_2O_2 reacts via a catalytic decomposition (Reaction 5.c) to form bubbles at the electrode surface, resulting in reduced area available for mass transfer and fuel/oxidant crossover. These phenomena adversely affect the power output of the fuel cell system. In order to overcome these challenges, we have conducted experiments to study the different multiphase flows present in the fuel cell. This study is aimed at managing the multiphase flow to achieve spontaneous stirring of the reagent streams to meet the challenges of high power density and high fuel utilization efficiency. We report here our progress toward the characterization and understanding of the fluid dynamics in these reactive multiphase flows. Below, we propose approaches to exploit this knowledge for the development of multiphase FCs that can exploit the exceptional kinetics and energy density of NaBH_4 and H_2O_2 .

Experimental Characterization of multiphase flows to study the different regimes in a fuel cell system. In understanding the role of gas-liquid flow for interfacial mass transfer in the fuel cell system, we focused our efforts on characterizing the flow patterns during the catalytic decomposition of H_2O_2 on Pt (reaction 5.c). Developing a map of different flow patterns has been performed for boiling³⁷ but not for heterogeneous reactive flows. Characterization was done in a microchannel as in Figure 21. The pattern of electrodes in each channel is designed to work like a fuel cell with working and counter electrodes on

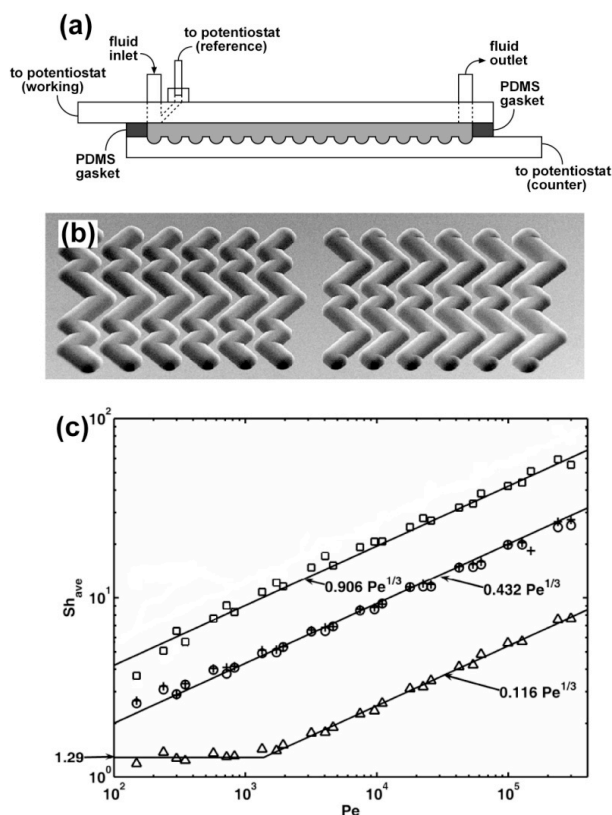


Figure 20. Electrochemical measurements of rates of mass transfer in microfluidic potential cells. **(a)** Schematic diagram of microfluidic potential cell. Top and bottom walls of channel act as working and counter electrodes, respectively. Grooves of herringbone mixing structure are indicated on bottom wall. **(b)** Scanning electron micrograph of counter electrode with grooves in staggered herringbone motif. **(c)** Globally averaged Sherwood numbers evaluated from measured current for uniaxial flow over smooth counter electrode (triangles), 3-D, chaotic flow over counter with staggered herringbone motif (crosses), 3-D, non-chaotic flow over counter with symmetrical herringbone motif (circles) and over working with herringbone motif (squares). Experiments were run with the reversible redox species, ferricyanide/ferrocyanide run at the transport-limited current.

the same surface 400 microns apart and was used to clean the surfaces electrochemically within the assembly³⁸. The multiphase flow is visualized using a high speed camera.

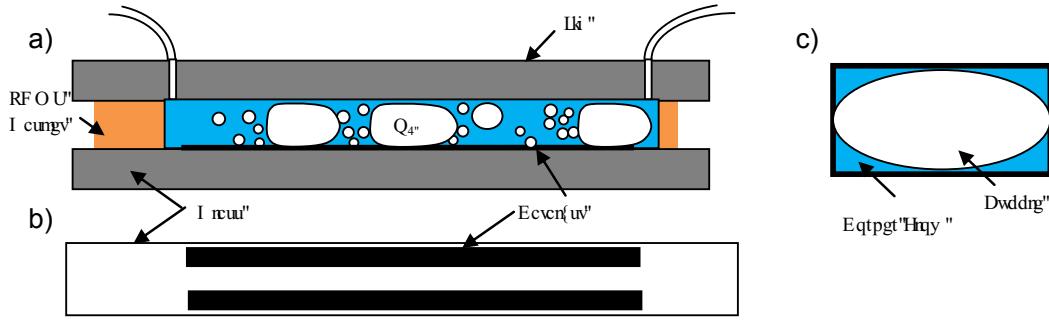


Figure 21: (a) Schematic of the experimental setup - Assembly of the device with the jig, PDMS gasket and the glass patterned with Pt. (b) Patterning of the catalyst in the Ferringo fuel cell geometry. (c) Cross section of the channel with a bubble, showing the corner flow past the bubble.

Experiments were performed by varying the average velocity U and the concentration C_0 of the reactant, H_2O_2 . We identify the capillary number, $Ca = \mu U / \sigma$, where μ [kg/m s] is the viscosity, U [m/s] is the average flow speed, and σ [J/m²] is the surface tension; Ca captures the competition between the viscous and capillary forces defining the dynamics in a bubbly flow and allows for generalization of conclusions. We represent the concentration as a molar flow rate to serve as a measure of the ratio of the gas to liquid volume flow rate. Figure 22(a) shows the flow map we obtained in the system for the parameter space explored. The churn/slug flow (Figure 22(b)) covers most of the parameter space accessed. At higher molar flow rates and higher capillary numbers we find the transition to a regime with dispersed bubbles throughout the channel (churn – Figure 22(b)). At low molar flow rates, transition to a regime with no bubbles occurs. The churn/slug regime approximates the desired Taylor regime that would provide efficient mass transfer and allow for steady operation. However, the length of gas to liquid slugs varies with time as well as with the concentration and capillary number (Figure 23(a)). These variations must be controlled in order to achieve a useful Taylor regime.

Understanding the dynamics of bubble growth, departure, and propagation. To build a basis for engineering these flows, we have further analyzed the dynamics of the growth and departure of bubbles. Figure 23 presents the scaling of the bubble length (a) and the time delay (b) between departures of bubbles from the leading edge of the electrodes. Interestingly, both quantities present robust power-law scaling over a broad range of conditions. The lengths scale as $Ca^{-1/2}$ and are independent of the concentration. This observation indicates a purely hydrodynamic effect (independent of chemical rate) in which the resistance to flow around the bubble grows with its length until the threshold

for break-off from the surface is achieved. This resistance is controlled by the form of

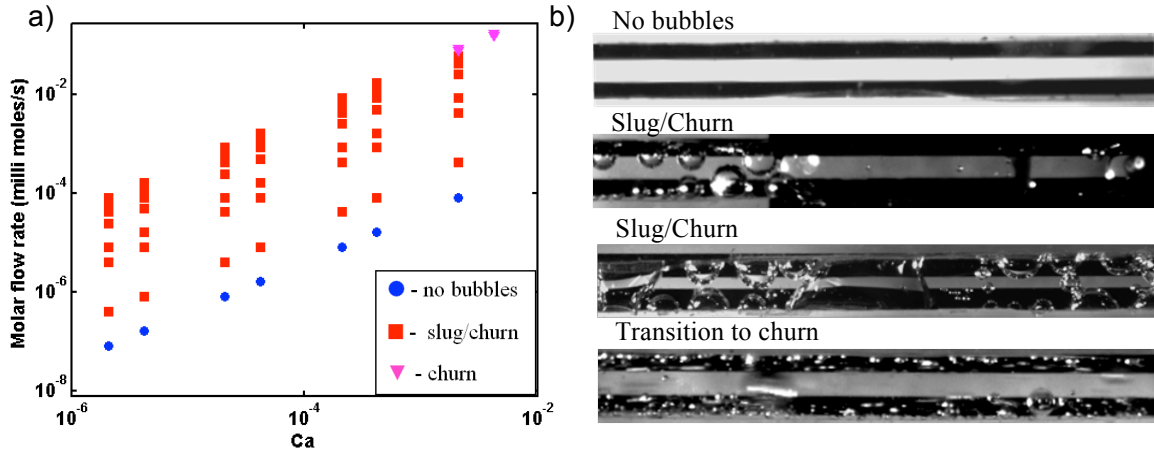


Figure 22: (a) The flow map showing the multiphase patterns for the parameter space explored. (b) The different flow patterns over different values of Ca number are shown on the right.

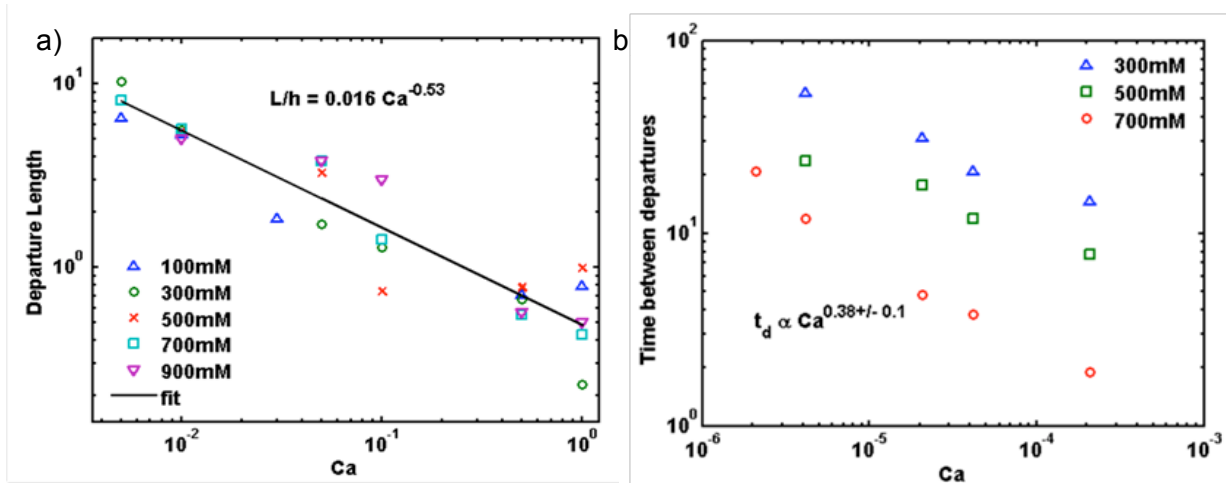


Figure 23: (a) Plot of the average departure length of gas bubble with Capillary number. (b) Plot of the average time between departures of two bubbles with Capillary number.

the liquid channels that surround the bubble (Figure 21(c))^{39 40}; this observation points to an opportunity to manipulate this behavior by changing the shape of the channel's cross-section. The time delay scales as $Ca^{-1/3}$ and exhibits dependence on the concentration. This observation indicates that the growth of the bubble is controlled by the mass transfer-limited rate of decomposition of H_2O_2 to O_2 . This process cannot be manipulated independently of the operation of the electrochemical process. Below, we propose to introduce regulation of the absolute pressure imposed on the flow in order to manipulate the relative volumes of the gas (compressible) and liquid (incompressible) phases. The introduction of this additional variable, when coupled with the robust dynamics that we have elucidated, will allow us to navigate the distinct flow regimes more freely and thus access the desired Taylor regime (Figure 26).

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