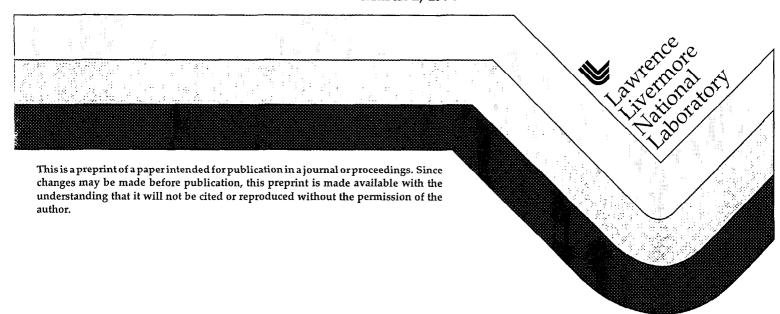
Picosecond Nonradiative Processes in Neodymium-doped Crystals and Glasses: Mechanism for the Energy Gap Law

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Picosecond Nonradiative Processes in Neodymium-doped Crystals and Glasses: Mechanism for the Energy Gap Law*

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Abstract

We present measurements of the ${}^4G_{7/2}$ emission lifetime for 26 Nd-doped materials. A model of nonradiative decay based on dipole-dipole energy transfer is developed and found to be supported by our data.

Introduction

It has been extensively verified in the past that the nonradiative decay rate between rare earth energy levels is predominantly determined by the energy gap and the particular host medium. The *energy gap law* is embodied in the simple expression for the nonradiative rate:

$$W_m = (1/\tau_0) \exp(-a \cdot p), \qquad (1)$$

where τ_0 and β are constants characteristic of the host medium, and p (= $\Delta E / h \nu_{max}$) is the number of phonons needed to bridge the gap, The expression $h \nu_{max}$ is related to the highest phonon frequency of the host medium and ΔE is the energy gap between the populated rare earth state and the next energetically lower level. The constant, a, is related to the details of the electron-phonon coupling. In nearly all cases reported in the literature, eqn. (1) is regarded as adequately describing the measured decay rates over several orders of magnitude [1-3].

In the present paper we report the nonradiative decay rates of the ${}^4G_{7/2}$ state of Nd³⁺ in 26 different crystals and glasses, with the goal of providing a data base relevant to the relaxation rate of the ${}^4I_{11/2}$ state which has a similar energy gap as the ${}^4G_{7/2}$ state [4]. The ${}^4I_{11/2}$ state of Nd³⁺ is a particularly important one from a practical point of view, since it can potentially "bottleneck" during lasing and give rise to transient absorption at the laser wavelength (constituting loss).

Much of the motivation for establishing this correlation is that emission lifetimes are much simpler to measure and can be widely applied to numerous Nd-doped crystals and glasses, while the pump-probe technique used to directly assess the ${}^4I_{11/2}$ decay time is complex to set up and execute.

The relevant energy levels are depicted in Fig. 1, where the $\tau_{5/2}$, $\tau_{7/2}$, and $\tau_{11/2}$ lifetimes are identified on

the diagram along with the 532 nm pump wavelength and the ~600 nm detection wavelength. One of the details that must be handled in the numerical analysis of the data is the effect of the overlapping ~600 nm emission arising from the ${}^4G_{5/2}$, ${}^2G_{7/2}$ states on the measured results (i.e. $\tau_{5/2}$), as depicted in Fig. 1. The numerical analysis and resulting fits to the data will be discussed.

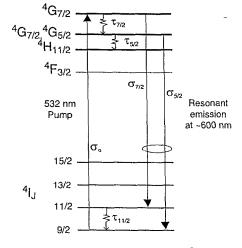


Fig.1 Energy levels and transitions for Nd³⁺

Experimental

The picosecond emission lifetimes [4] were measured using the time-correlated single photon counting system together with a Coherent mode-locked laser producing ~90 psec pulses at 76 MHz and then doubled to 532 nm. The sample emission was detected at ~600 nm with a monochromator followed by a multichannel plate photomultiplier (MCP-PMT). The (deconvoluted) temporal resolution of the data is about 50 psec. For the case of the fluoride samples, the lifetimes of ~10 nsec were too long to be measured with the time-correlated photon-counting apparatus. To handle these longer lifetimes, we employed a conventional set-up consisting of a Q-switched Nd:YAG laser (12 nsec pulsewidth), along with a monochromator, oscilloscope, and MCP-PMT

Results

The results of the emission lifetime experiments are contained in Table 1. Based on an analysis of the best fit to the data, the uncertainty in the value of the $\tau_{7/2}$ lifetimes is $\Delta \tau_{7/2}$ (<1 ns) = \pm 50 ps for data with lifetimes less than 1 ns and $\Delta \tau_{7/2}$ (>1 ns) = \pm 200 ps for data with lifetimes greater than 1 ns.

Many of the trends that can be gleaned from the data in Table 1 are expected. For the case of the fluorides, the $\tau_{7/2}$ values are the longest, being 4,000 - 41,000 psec. This is anticipated, since their phonon

frequencies are the lowest among the hosts listed, being <600 cm⁻¹. Whereas the phosphate and silicate glasses are in the range of 150 to 250 psec owing to the rather high vibrational frequencies of the SiO₄ and PO₄ anions. It is noteworthy that comparisons of previous $\tau_{7/2}$ measurements in the literature are satisfactory, including [5]: 1400 versus 1090 psec for YAlO3; 370 versus 200 psec for Y₃Al₅O₁₂; and 8400 psec versus 9100 psec for LiYF₄; 56,400 psec versus 41,000 psec

Table 1. Emission lifetimes of the ${}^4G_{7/2}$ excited state

$(\tau_{7/2})$		
Name	Formula	τ _{7/2} , psec
Phosphate	P ₂ O ₅ +Al ₂ O ₃ +	
glasses	modifiers	
LG-750		228
APG-1		215
APG-x		210
APG-2		150
LG-812	P ₂ O ₅ +fluorides+ modifiers	~1400
Silicate	SiO ₂ +Al ₂ O ₃ +	
glasses	modifiers	
LG-660	modificis	215
LG-650		210
Sol-gel	SiO_2	245
Vanadate	5102	243
YVO ₄	YVO_4	190
Tungstate	1 4 04	190
CaWO₄	CaWO ₄	510
Oxide	Cavr O ₄	510
YALO	YAlO ₃	1090
GSGG	$Gd_3Sc_2Ga_3O_{12}$	715
YAG	$Y_3Al_5O_{12}$	200
GGG	$Gd_3Ga_5O_{12}$	530
LLGG	$La_3Lu_2Ga_3O_{12}$	1200
Apatites	La ₃ Lu ₂ Ca ₃ C ₁₂	1200
C-FAP	Ca ₅ (PO ₄) ₃ F	70
S-FAP	$Sr_5(PO_4)_3F$	175
C-VAP	$Ca_5(VO_4)_3F$	200
S-VAP	$Sr_5(VO_4)_3F$	330
5- 1 AI	Sr ₅ (VO ₄) ₃ Cl	380
Fluorides	315(V O4)3C1	360
ZBLAN	ZrF ₄ -BaF ₂ -AlF ₃ -	18,000
202 IIV	LaF ₃ -NaF	10,000
YLF	LiYF ₄	9100
Na ₃ Sc ₂ Li ₃ F ₁₂	Na ₃ Sc ₂ Li ₃ F ₁₂	4000
KY ₃ F ₁₀	KY ₃ F ₁₀	9000
YF_3	YF ₃	
LaF ₃	LaF ₃	22,000
Lai'3	டகாვ	41,000

Analysis and Discussion

As mentioned earlier, measurements of the 4G_{7/2} population decay time have long been suspected to be similar in magnitude to that of the ⁴I_{11/2} level of Nd³⁺. Using the 4I11/2 lifetimes from a direct measurement in [7] we find that the ${}^4I_{11/2}$ and ${}^4G_{7/2}$ lifetimes are correlated with each other to within about a factor of two, across a variety of different host media. This concurrence is reasonable, since some differences in the energy gap and perhaps the electron-phonon coupling may be expected, in addition to the experimental and analytical uncertainties. Furthermore, due to the small difference in the size of the energy gap for a given host medium, this data offers some experimental validation that the specific characteristics of the electronic states (symmetry, crystal field interactions, spin, etc.) do not strongly influence the nonradiative decay

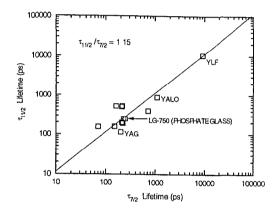


Fig. 2 Comparison of the ⁴I_{11/2} nonradiative relaxation time from [4] and the ⁴G_{7/2} emission lifetime.

Another theory of nonradiative decay derives the multi-phonon rate on basis of energy-transfer theory, as originally described by Forster [6] and Dexter [7]. With the assumption that dipole-dipole interactions dominate the process, and that the phonon absorption $\alpha_{\text{ph}}(\lambda)$ is a slowly varying function compared to the emission cross section spectrum $\sigma_{em}(\lambda)$, we derive that.

$$W_{nr} = \frac{2c_{3}^{\prime}}{(\pi n)^{2}} \frac{\int \sigma_{em} d\lambda}{V_{min}} \alpha_{ph}, \qquad (2)$$

where V_{min} is the minimum volume that is nonabsorbing in nature (centered on the rare earth ion), n is the refractive index, and c is the speed of light. This expression essentially emerges from the well-known Forster-Dexter spectral overlap integral between the emitting and absorbing species, and a volume integration over a uniform acceptor (phonon) concentration, (a derivation similar to other reports in the literature [8-11]). It is also noteworthy however, that the multi-phonon absorption spectrum is commonly described with an exponential expression expression [12,13]

$$\alpha_{\rm ph} = A_{\rm host} \exp \left(-\gamma_{\rm host} h \upsilon_{\rm ph} / h \upsilon_{\rm char} \right),$$
 (3)

where v_{ph} is the phonon frequency, v_{char} is a characteristic phonon frequency of the host material, and γ_{host} and A_{host} are other host-dependent constants. So if we combine eqns. (3) and (4), we can suggest an alternative route to deriving the form of the energy gap law:

$$W_{nr}(v_{gap}) = \frac{2c_{3}^{2}}{(\pi n)^{2}} \frac{\int \sigma_{em} d\lambda}{V_{min}} A_{bust} \exp \left[-\gamma_{host} \left(\frac{hv_{gap}}{hv_{max}} \right) \right], \tag{4}$$

where we have identified υ_{ph} as υ_{gap} and υ_{char} as υ_{max} (maximum phonon frequency), the . The γ_{host} parameter turns-out to be similar for many crystals, for instance being in the range of 4-5 for alkali and alkaline-earth halide crystals [13]. The main point to note regarding eqn (4) is that the exponentiated factor in square brackets only contains information concerning the host medium, while the rare earth properties are exclusively represented in the pre-exponential factor – implying that we may expect reasonably good adherence to the form of the energy-gap law, eqn. (1),

We can explore the validity of eqn. (4) by inputting reasonable estimates for the terms in the pre-exponential factor and deducing the magnitude of this constant. Using $V_{min} = 5 \times 10^{-24} \text{ cm}^3$, $A_{host} = 20,000 \text{ cm}^{-1}$ (average from ref. [13]), and $\sigma_{em} * \Delta \lambda = 1.8 \times 10^{-25} \text{ cm}^2$ (calculated from Judd-Ofelt theory using the average Ω_J parameters of YAG and YLF [14]), we obtain a value of $0.6 \times 10^{12} \text{ sec}^{-1}$, or $\tau_0 = 1.7$ psec, which is defined from eqn. (1) as:

$$1/\tau_{o} = \frac{2c/3}{(\pi n)^{2}} \frac{\int \sigma_{em} d\lambda}{V_{min}} A_{host}, \qquad (5)$$

We are now in a position to compare this calculation to the data in Table 1, where we use the energy gap ($\Delta E_{7/2}$) and highest phonon frequencies (hv_{max}) to calculate number of phonons $p=\Delta E_{7/2}/hv_{max}$. If we also group all of the phosphate glasses into a single datum, and all of the silicates into a second one, then there are potentially ten usable points. Finally, if the LaF₃ result is eliminated because of its strongly nonexponential character, the results of the exercise may be displayed as shown in Fig. 3, where we have plotted the nonradiative decay time, $\tau_{7/2}$, against the number of phonons, p. The data is then fitted to the reciprocal of the energy gap law:

$$\tau_{nr} = \tau_0 \exp(\mathbf{a} \cdot \mathbf{p}) \tag{6}$$

with the result of the numerical fit yielding $\tau_0 = 3.3$ psec and a = 3.1. We are very encouraged by this result because it may be compared with theoretical calculation of $\tau_0 = 1.7$ psec noted above in connection with eqn.

(5). Now, if we associate γ_{host} with a, and $h\nu_{char}$ with $h\nu_{max}$, the a=3.1 value from Fig. 3 appears to be within the range of what one may expect from the energy-transfer theory of nonradiative decay encompassed in eqn. (4), since the phonon spectra of many materials is characterized by $\gamma_{host}=4-5$.

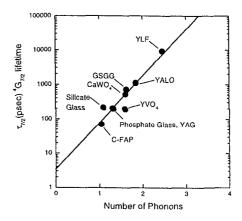


Fig. 3 Plot of the nonradiative decay time against the number of phonons needed to bridge the gap.

In referring back to the more general energy-transfer-based description of nonradiative decay relating to eqn. (2), we can apply another test of this model by plotting W_m of the rare-earth ion against the absorption coefficient α_{ph} of the host and fitting these data to the simple expression:

$$W_{nr} = k \alpha_{ph}, \qquad (7)$$

where k can be theoretically calculated and related to τ_0 by:

$$k = 1/(A_{host} * \tau_0) = \frac{2c_3}{(\pi n)^2} \frac{\int \sigma_{em} d\lambda}{V_{min}}$$
 (8)

based on eqns. (2) and (5). In Fig. 4 we have chosen to focus this analysis on data derived exclusively from the YAG, YALO and YLF hosts, because the phonon spectra of these materials have been reported in the literature [15, 16, 17]; in addition, the nonradiative decay rates and the associated energy gaps derived from refs. [18, 19, 20, 21, 22] entail a wide variety of different electronic states and rare earth ions (in contrast to the prior exclusive attention given to the ${}^4G_{7/2}$ state of Nd³⁺). While we notice that the scatter in the data of Fig. 4 is substantial, it does span many orders of magnitude and the fitted or experimental value of $k_{exp} = 1.0 \times 10^7$ cm/sec based on the data is found to be near

the independently theoretical value of $k = 3 \times 10^7$ cm/sec deduced from eqn. (8) for the ${}^4G_{7/2}$ state.

Employing the experimental value of k in the simple relationship of eqn. (8): $\tau_o = 1/(k_{exp}*A_{host})$, we find a third estimate for τ_o of 5 psec, a value that is in reasonable accord with the theoretically derived value of 1.7 psec from eqn. (5), and the other experimentally determined magnitude of 3.3 psec from Fig. 3 and eqn. (6).

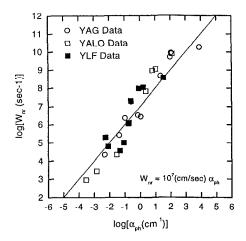


Fig 4. Plot of the nonradiative decay rate of various states of rare earth ions in YAG, YALO, and YLF, as a function of the phonon absorption coefficient of the host at the particular energy gap.

It is important to keep in mind that the results from the first two calculations are specific to the $^4G_{7/2}$ state, while the treatment in the last calculation entails numerous different rare earth ions and states (Fig. 4). The concurrence of these three independent approaches in estimating τ_0 is remarkable and an implicit endorsement of the usefulness of the energy-transferbased mechanism of nonradiative decay.

Acknowledgments

(1967) 252.

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H. W. Moos, J. Luminesc. 1/2 (1970) 106.
 W. D. Partlow and H. W. Moos, Phys. Rev. 157

[3] L. A. Riseberg, W. B. Gandrud, and H. W. Moos, Phys. Rev. 159 (1967) 262.

[4] C. Bibeau, Ph. D. Dissertation, "Evaluation of the ⁴I_{11/2} terminal level lifetime for several neodymium-doped laser crystals and glasses," April 25, 1995.

[5] T. T. Basiev, A. Yu. Dergachev, Yu. V Orlovskii, and A. M. Prokorov, J. Luminesc. 53 (1992) 19

[6] T. Forster, Z. Naturf. 49 (1949) 321.

[7] D. L. Dexter, J. Chem. Phys. 21 (1953) 836.

[8] N. T. Timofeev and E. B. Sveshnikova, Opt. Spectrosc. (USSR) 54 (1983) 595.

[9] V. L. Ermolaev and E. B. Sveshnikova, J. Luminesc. 20 (1979) 387.

[10] V. L. Ermolaev and E. B. Sveshnikova, Chem. Phys. Lett. 23 (1973) 349.

[11] E. B. Sveshnikova, N. T. Timofeev, and V. M. Zolotarev, Izvestiya Akademii Nauk SSSR. Seriya Fizicheskaya 44, (1980) 722.

[12] T. C. McGill, R. W. Hellwarth, M. Magir, and H. V. Winston, J. Phys. Chem. Solids 34 (1973) 2105.

[13] L. L. Boyer, J. A. Harrington, M. Haas, and H. B Rosenstock, Phys. Rev. B 11 (1975) 1665.

[14] R. P. Bauman and S. P. S. Porto, Phys. Rev 161 (1967) 842.

[15] V. P. Gaponsev, M. R. Sirtlanov, A. K. Gromov, A. A. Isineev, *Proceedings on the International Conference on Lasers '81* (STS Press, McLean, 1981), p. 763.

[16] G. A. Slack, D. W. Oliver, R. M. Chrenko, and S. Roberts, Phys. Rev. 177 (1969) 1308.

[17] E. B. Sveshnikova, A. A. Stroganov, and N. T.Timofeev, Opt. Spectrosc. (USSR) 64 (1988) 43.

[18] M. J Weber, Phys. Rev. B 8 (1973) 54.

[19] A. M. Tkachuk, A. V. Khilko, and M. V. Petrov, Opt. Spectrosc. (USSR) 58 (1985) 216.

[20] T. T. Basiev, Yu. V. Orlovskii, K. K. Pukhov, V. B. Sigachev, M. E. Doroshenko, and I. N. Vorob'ev, J. Luminesc. 68 (1996) 241.

[21] G. M. Zverev, G. Ya. Kolodnyi, and A. M. Onishchenko, Sov. Phys. - JETP 33 (1971) 497.

[22] T. T. Basiev, A. Y. Dergachev, Yu. V. Orlovskii, and A. M Prokhorov, N. Vorob'ev, OSA Proceedings on Advanced Solid State Lasers 10 (1991) 358.

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