1	Behaviour of metakaolin-based geopolymers incorporating s
2	sludge ash (SSA)
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11	ABSTRACT
12	In recent years, geopolymers have become a widely researched binding material. There are te
13	and environmental advantages to using this type of binder instead of Portland cement. In this
14	binary systems of geopolymers were produced by using mixtures of metakaolin (MK), a well-
15	aluminosilicate raw material, and a residue from sewage sludge incineration: sewage sludge a
16	This ash was used to partially replace the metakaolin in proportions of 0–20%. The mixtures
17	activated with alkaline solutions and they were cured by using two different conditions: at roc
18	temperature (25 °C) and in a thermal bath (65 °C). The samples were assessed by X-ray diffra
19	scanning electron microscopy (pastes) and compressive strength (mortars). The results from t
20	showed zeolite formation (faujasite) in geopolymers cured in the thermal bath, which caused
21	the compressive strength of the alkali-activated mortars. Replacement of MK with SSA cause
22	reduction in the compressive strength of mortars cured at 65 °C. However, at room temperature
23	mechanical strength was observed for the MK and MK-SSA systems. These results demonstr
24	SSA is a suitable mineral precursor for partial replacement of MK in geopolymer production.
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26	KEY WORDS: alkali-activated binder, microstructure, residue, X-ray techniques
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viour of metakaolin-based geopolymers incorporating sewage sludge ash (SSA) uque¹, L. Reíg², J.C.B. Moraes¹, J.L. Akasaki¹, M.V. Borrachero³, L. Soriano³, J. Payá³, J.A. Malmonge¹ and M.M. Tashima¹* ESP — Univ Estadual Paulista, Campus de Ilha Solteira, São Paulo, Brazil. ²EMC, Universitat Jaume I, Castelló de la Plana, Spain. CH — Instituto de Ciencia y Tecnología del Hormigón, Universitat Politècnica de Valencia, Valencia, Spain. Corresponding author: maumitta@hotmail.com; tel. +55 18 37431217 CTears, geopolymers have become a widely researched binding material. There are technological nmental advantages to using this type of binder instead of Portland cement. In this study, ems of geopolymers were produced by using mixtures of metakaolin (MK), a well-known icate raw material, and a residue from sewage sludge incineration: sewage sludge ash (SSA). as used to partially replace the metakaolin in proportions of 0-20%. The mixtures were vith alkaline solutions and they were cured by using two different conditions: at room e (25 °C) and in a thermal bath (65 °C). The samples were assessed by X-ray diffraction, lectron microscopy (pastes) and compressive strength (mortars). The results from these studies olite formation (faujasite) in geopolymers cured in the thermal bath, which caused a decrease in essive strength of the alkali-activated mortars. Replacement of MK with SSA caused a lower n the compressive strength of mortars cured at 65 °C. However, at room temperature, similar

strength was observed for the MK and MK-SSA systems. These results demonstrated that

1. INTRODUCTION Geopolymers are a new class of material obtained by a chemical reaction of an aluminosilicate material and a highly concentrated alkaline solution [1]. This binding material can be used as a construction material due to their high strength and durability, replacing Portland cement (OPC) in concrete [2]. Metakaolin (MK) is usually used as the aluminosilicate source in geopolymers [3-5]. Previously studies on metakaolin-based geopolymers have shown high compressive strength after a few hours of curing at temperatures ranging from 40 to 95 °C [3]. However, research has shown that some geopolymers, especially metakaolin-based ones cured at high temperatures, tend to form crystalline structures: zeolites [6-9]. These crystalline phases significantly reduce the compressive strength of geopolymers, a critical behaviour for building materials [2,9]. In this sense, the combination of different raw materials containing silicon and/or aluminium oxides on their composition are being carried out (binary systems) in order to reduce the zeolite formation [10,11]. Sewage sludge ash (SSA), a waste generated in large amounts (1.7million tons per year) has been studied extensively in blended Portland cements [12-14]. The first study related to the use of SSA in geopolymers were reported by Yamaguchi et al. where authors used fly ash/SSA yielding the maximum flexural strength (about 5.5 MPa) for mixture containing 75%SSA [15]. In this paper is presented the influence of SSA on the mechanical strength and on the crystallization process (zeolite formation) of metakaolin-based geopolymers. Specimens were cured at both hightemperature and 25 °C and they were assessed through compressive strength, X-ray diffraction and scanning electron microscopy. Geopolymers are a new class of material first introduced by Davidovits in 1978 [1] that can be used as a construction material due to their high strength and durability, replacing Portland cement in concrete [2]. Geopolymers have environmental advantages over Portland cement because less energy in consumed in their production with lower CO₂ emissions [3,4]. A geopolymeric binder is obtained when an aluminosilicate material is combined with a highly

concentrated alkaline solution [5]. Metakaolin is usually used as the aluminosilicate source in

geopolymers [6-8]. In the field of aluminosilicate materials research, interesting studies of geopolymers

obtained from binary systems, such as metakaolin/fly ash, are being carried out [9]. The present study presents a binary system of metakaolin and sewage sludge ash (SSA) a residue that has been studied recently in blended Portland cement [10,11]. Large volumes of sewage sludge must be managed, and its transformation to ash is an interesting solution due to the volume reduction obtained. SSA is generated in large amounts worldwide (1.7 million tons per year) [10], and due the implementation of wastewater treatment plants in many cities, the amount of SSA produced is increasing significantly. Therefore, since there is a considerable quantity of this residue, building materials may be a suitable use for the ash. Previously studies on metakaolin based geopolymers have shown high compressive strength after a few hours of curing at temperatures ranging from 40 to 95 °C [6]. However, research has shown that some geopolymers, especially metakaolin based ones cured at high temperatures, tend to form crystalline structures: zeolites [12-15]. These crystalline phases significantly reduce the compressive strength of geopolymers, a critical behaviour for building materials [2,15]. This study aimed to assess the use of SSA as partial replacement (0-20%) for metakaolin in geopolymer synthesis in order to diminish the problems related to strength decrease of high temperature cured mixtures. Specimens cured at 25°C were also studied in order to assess the influence of SSA, X ray diffraction and scanning electron microscopy studies on pastes, and compressive strength measurements of mortars were carried out.

2. MATERIALS AND METHODS

2.1 Materials and Equipment

Metakaolin was supplied by Metacaulim do Brasil®. Sewage sludge ash was obtained from an auto-combustion process of sewage sludge from São José do Rio Preto city (São Paulo-Brazil). The chemical composition of MK and SSA are shown in Table 1. The mean particle diameter, d_{50} and d_{90} of MK were 23.90, 18.16 and 53.96 μ m, respectively; and for SSA they were 20.28, 11.77 and 52.45 μ m, respectively. For mortar preparation, siliceous sand (Castilho city, São Paulo-Brazil) with a fineness modulus of 2.05 and specific gravity of 2.67 ton/m³ was used. Sodium hydroxide (98% purity) and sodium silicate (18% Na₂O, 63% SiO₂) were used for the preparation of alkaline solutions (both supplied by Dinâmica Química).

X-ray diffraction (XRD) patterns for raw materials and geopolymeric pastes were obtained using a Shimadzu XRD-6000 system. The 2θ range was 5– 60° using Cu-K α radiation and a Ni filter, at a voltage of 30 kV, a current intensity of 40 mA, an angle step of 0.02° , and a step time of 1.20 s/step. Scanning electron microscopy (SEM) images of fractured surface pastes were obtained using a ZEISS model EVO LS15. The compressive strength of mortars was measured in an EMIC Universal machine with a 200-ton load limit.

Table 1 – Chemical composition of MK and SSA in percentage by mass

Oxide (%)	SiO ₂	Al ₂ O ₃	Fe ₂ O ₃	CaO	MgO	Na ₂ O	K ₂ O	SO ₃	TiO ₂	Others	LOI
MK	58.39	35.47	2.71	0.01	0.30	-	1.44	-	1.51	0.07	0.10
SSA	38.28	20.72	11.27	5.51	1.91	0.70	0.73	4.18	3.73	9.25	3.72

2.2 Geopolymer preparation

Three different proportions of MK replacement by SSA were assessed in this study: 0% (control), 10% and 20% (by mass). The H_2O/Na_2O and SiO_2/Na_2O molar ratios were maintained constant at 9.26 and 2.00, respectively. For mortars, the sand/binder ratio was 2.5 (the binder amount being the sum of the masses of MK and SSA). Two different curing temperatures were applied at a relative humidity greater than 95%: 25 °C (room temperature) and 65°C (using a thermal bath). The compressive strength of the mortars was determined after one, three and seven days of curing. XRD studies were performed on pastes after the same curing times. SEM studies were performed only after three days of curing. The samples used in this paper are named as MK*c-xx*, where c is the curing temperature (c, R: room temperature, B: thermal bath) and xx is the percentage of SSA incorporated (xx = 0, 10 or 20).

3. RESULTS AND DISCUSSION

The compressive strengths of the different mortars are shown in Figure 1. For mortars cured at 65 °C (Fig. 1a), the compressive strength of all mixtures decreased with increasing curing time. Similarly, the strength of the mortars after one day of curing at 65 °C decreased with increasing replacement of MK by SSA. After three days of curing at 65 °C the compressive strength of the mortars decreased by 34% for

MKB-0, 36% for MKB-10 and 37% for MKB-20. The compressive strength of mixtures incorporating 10 or 20% of SSA decreased by a similar percentage as the control (MKB-0, 36%). After seven days of curing, the compressive strength values and their respective percentage loss of compressive strength compared to the values after three days of curing were 20.3 MPa (16.8%), 16.6 MPa (12.6%) and 15.0 MPa (9.1%) for MKB-0, MKB-10 and MKB-20, respectively. These results show that the relative decrease in compressive strength with curing time is lower in mortars containing up to 20% SSA than in the control sample. These results suggest that the use of SSA in the production of metakaolin-based geopolymers stabilizes the compressive strength in mortars prepared with long curing times. Mortars cured at room temperature (Fig. 1b) behaved differently to samples cured at 65 °C. At room temperature, mortars did not show a decrease in compressive strength with curing time. After one day of curing at room temperature, the strength decreased when SSA content was increased, similar to the situation for mortars cured at 65°C. After three and seven days of curing, the compressive strength of MKR-0 increased slightly, whereas samples incorporating SSA presented an important strength gain. After seven days of curing at room temperature, the MKR-10 sample achieved a similar strength to MKR-0 (27.9 and 28.8 MPa, respectively). This result suggests that the use of SSA in metakaolin-based geopolymers may be an interesting possibility. XRD studies were carried out on MKB-0, MKB-20, MKR-0 and MKR-20 pastes in order to examine the formation of crystalline phases, both at 25°C and 65°C (Figure 2). The raw materials MK and SSA show a baseline deviation in the range 16–32° and 18–32°, respectively, which is characteristic of the presence of an amorphous phase. Quartz (SiO₂, PDFcard#331161), kaolinite (Al₂Si₂O₅(OH)₄, PDFcard#140164) and muscovite (KAl₃Si₃O₁₀(OH)₂, PDFcard#210993) were found in MK, and quartz, anhydrite (CaSO₄, PDFcard#371496), anorthite (CaAl₂Si₂O₈, PDFcard#411486) and hematite (Fe₂O₃, PDFcard#130534) were found in SSA.

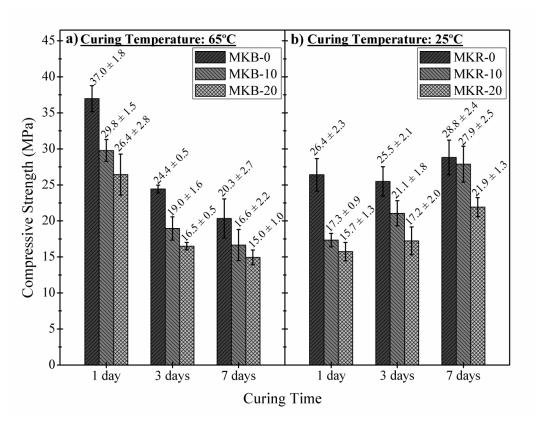


Figure 1 – Compressive strength of mortars: a) cured at 65°C; and b) cured at 25°C

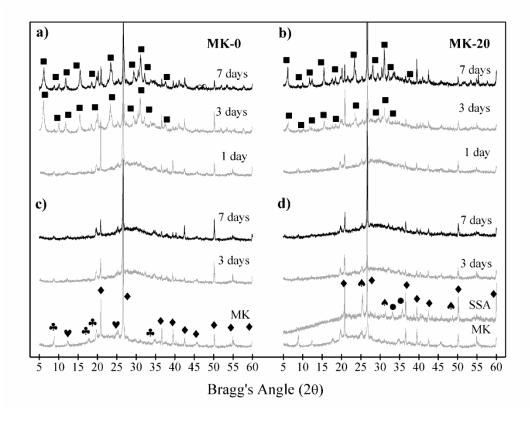


Figure 2 – XRD patterns for MK/SSA pastes: a) MKB-0; b) MKB-20; c) MKR-0; d) MKR-20 (Key: ◆:

Quartz; ♥: Kaolinite; ♣: Muscovite; ♠: Anhydrite; •: Hematite; ■: Faujasite)

For the geopolymeric pastes, all samples presented a baseline deviation line between 16 and 40°, which can be attributed to the amorphous phase of the geopolymeric gels. This shift of the baseline to higher 20 values compared to the MK and SSA amorphous phases due the geopolymerisation reaction has also been observed in others studies [16]. For pastes cured at 65°C, faujasite (Na₂Al₂Si₄O₁₂.8H₂O, PDFcard#391380) formation was observed after three days of curing (Fig. 2a and 2b). However, the presence of SSA influences the zeolite formation, since a lower zeolite peak intensity is observed after three days of curing at 65°C compared to MKB-0. No signals attributed to zeolites were distinguished by XRD analyses on pastes cured at room temperature, either in MKR-0 (Fig. 2c) or MKR-20 (Fig. 2d), whatever the curing time (three and seven days). Both geopolymeric gel and zeolite formation are directly related to the reactivity of the raw materials [2] and to the curing temperature [2,17]. For high alkaline environment, high curing-temperatures favours the crystallization of aluminosilicate gels forming zeolite-type structures and, according to Bosnar et al., the crystallization process is sharply reduced with the increase on the SiO₂/H₂O [17]. In this paper, MK presented higher reactivity than SSA, so it was expected that geopolymers with higher amounts of MK would present more intense zeolite formation and, consequently, greater reduction in compressive strength. It is due to the microporous-crystalline structure based on 3D-cage system of zeolites that reduces the compressive strength of mortars when compared to the amorphous structure based on 3Dnetwork of aluminate and silicate tetrahedral of geopolymers [2,18].

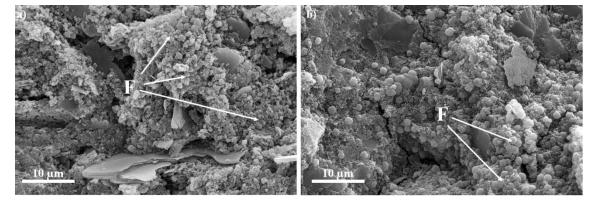


Figure 3 – SEM micrographs of geopolymer fractured surfaces: a) MKB-0; b) MKB-20 (Key: F-

163 faujasite)

	165	Faujasite was also observed in SEM on fractured samples of MKB-0 and MKB-20 after three days of
1 2	166	curing (Fig.3). Rounded crystalline particles of 2–4 µm size were formed. Since the raw material mainly
3 4	167	contains metakaolin, faujasite was formed in both pastes. However, mortars containing higher amounts of
5 6	168	SSA presented a smaller decrease in compressive strength.
7 8	169	
9 10	170	
11 12	171	
13 14	172	4. CONCLUSION
15 16	173	
17 18	174	Metakaolin-based geopolymers with partial replacement of MK with SSA were studied. XRD analysis
19 20	175	showed that geopolymers cured at 65 °C produced faujasite after three days of curing. This zeolite
21 22 23	176	formation caused a decrease in compressive strength with the curing age at 65 °C. The addition of SSA
24 25	177	(up to 20%) to the mixture resulted in a smaller loss of compressive strength in mortars cured at 65°C
26 27	178	when compared to the control without SSA. In addition, in samples cured at 25°C, those containing 10%
28 29	179	SSA presented similar compressive strength as the control mortar after seven days of curing. Thus partial
30 31	180	replacement of metakaolin with SSA showed advantages in both curing conditions.
32	181	
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42	186	(processo 309015/2015-4).
44 45	187	
46 47	107	
48 49	188	REFERENCES
50 51	189	[51] P. Duxson, A. Fernández-Jiménez, J.L. Provis, G.C. Lukey, A. Palomo, J.S.J. van Deventer,
52 53	190	Geopolymer technology: the current state of the art. J Mater Sci 42 (2007) 2917–2933.
54 55	191	[1] J. Davidovits, Synthesis of New High Temperature Geo Polymers For Reinforced
56 57	192	Plastics/Composites. SPE PACTEC 79 Society of Plastic Engineers, Brookfield Center, 1979.
5 <i>7</i> 58 59	193	[2] J.L. Provis, J.S.J. van Deventer, Geopolymers: Structure, Processing, Properties and Industrial
60 61	194	Applications. first ed., Woodhead Publishing Limited: Oxford, 2009.

	224	[16] M.M. Tashima, J.L. Akasaki, J.L.P. Melges, L. Soriano, J. Monzó, J. Payá, M.V. Borrachero, Alkali
1 2	225	activated materials based on fluid catalytic cracking catalyst residue (FCC): Influence of SiO ₂ /Na ₂ O and
3 4	226	H ₂ O/FCC ratio on mechanical strength and microstructure, Fuel 108 (2013) 833–839.
5 6	227	[17] S. Bosnar, J. Bronic, D. Brlek, B. Subotic, Chemically controlled particulate properties of zeolites:
7 8	228	Towards the face-less particles of zeolite A. 2. Influence of aluminosilicate batch concentration and
9	229	alkalinity of the reaction mixture (hydrogel) on the size and shape of zeolite A crystals, Micropor
11 12	230	Mesopor Mater 142 (2011) 389-397.
13 14	231	[18] J. Li, A. Corma, J. Yu, Synthesis of new zeolite structures, Chem Soc Rev 44 (2015) 7112-7127. [3]
15 16	232	B. McLellan, R. Williams, J. Lay, A. Van Riessen, G. Corder, Costs and carbon emissions for
17 18	233	geopolymer pastes in comparison to ordinary Portland cement. J Clean Prod 19 (2011) 1080–1090.
19 20 21	234	[4] J. Davidovits, Geopolymer Chemistry and Applications. third ed., Institut Géopolymère: France,
22 23	235	2001.
23 24 25	236	[5] P. Duxson, A. Fernández-Jiménez, J.L. Provis, G.C. Lukey, A. Palomo, J.S.J. van Deventer,
26 27	237	Geopolymer technology: the current state of the art. J Mater Sci 42 (2007) 2917-2933.
28 29	238	[6] M.S. Muñiz Villarreal, A. Manzano Ramírez, S. Sampieri Bulbarela, J.R. Gasca Tirado, J.L. Reyes-
30 31	239	Araiza, J.C. Rubio Ávalos, J.J. Pérez Bueno, L.M. Apatiga, A. Zaldivar Cadena, V. Amigó Borrás, The
32 33	240	effect of temperature on the geopolymerization process of a metakaolin based geopolymer, Mater Lett 65
34 35	241	(2011) 995-998.
36 37	242	[7] C. Kuenzel, T.P. Neville, S. Donatello, L. Vandeperre, A.R. Boccaccini, C.R. Cheeseman, Influence
38 39	243	of metakaolin characteristics on the mechanical properties of geopolymers, Appl Clay Sci 83-84 (2013)
40 41	244	308-314.
42 43	245	[8] M.R. Wang, F.C. Jia, P.G. He, Y. Zhou, Influence of calcination temperature of kaolin on the
44 45	246	structure and properties of final geopolymer. Mater Lett 64 (2010) 2551–2554.
46 47	247	[9] Z. Zhang, H. Wang, Y. Zhu, A. Reid, J.L. Provis, F. Bullen, Using fly ash to partially substitute
48 49	248	metakaolin in geopolymer synthesis, App Clay Sci 88-89 (2014) 194-201.
50 51	249	[10] S. Donatello, C.R. Cheeseman, Recycling and recovery routes for incinerated sewage sludge ash
52 53	250	(ISSA): A review, Waste Manage 33 (2013) 2928 2940.
54 55	251	[11] B.J. Zhan, C.S. Poon, Study on feasibility of reutilizing textile effluent sludge for producing concrete
56 57	252	blocks, J Clean Prod 101 (2015) 174–179.
58 59	I	

[12] J. Zhang, Y. He, Y. Wang, J. Mao, X. Cui, Synthesis of a self supporting faujasite zeolite membrane using geopolymer gel for separation of alcohol/water mixture, Mater Lett 116 (2014) 167–170.

[13] N. Granizo, A. Palomo, A. Fernandez Jiménez, Effect of temperature and alkaline concentration on metakaolin leaching kinetics, Ceram Int 40 (2014) 8975–8985.

[14] H. Takeda, S. Hashimoto, H. Yokoyama, S. Honda, Y. Iwamoto, Characterization of zeolite in zeolite geopolymer hybrid bulk materials derived from kaolinitic clays, Materials 6 (2013) 1767–1778.

[15] T. Bakharev, Geopolymeric materials prepared using Class F fly ash and elevated temperature euring, Cem Concr Res 35 (2005) 1224–1232.

[16] M.M. Tashima, J.L. Akasaki, J.L.P. Melges, L. Soriano, J. Monzó, J. Payá, M.V. Borrachero, Alkali activated materials based on fluid catalytic cracking catalyst residue (FCC): Influence of SiO₂/Na₂O and H₂O/FCC ratio on mechanical strength and microstructure, Fuel 108 (2013) 833–839.