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1. Introduction

1 Thermal barrier coatings (TBCs) are employed to protect the metallic components against corrosion and
2 oxidation at high temperature. Atmospheric plasma spraying (APS) is one of the most popular technologies
3 to obtain TBCs because of its flexibility and low cost [1]. The material of TBCs is currently yttria-stabilized
4 zirconia (YSZ) due to its low thermal conductivity, relatively low coefficient of thermal expansion, phase
5 stability and high corrosion resistance [2]. The increase in the operational temperature of gas turbines
6 resulted in the demand to find coatings enabling better thermal isolation [3]. Consequently, an intense
7 research activity has started to reduce the thermal conductivity of TBCs [4,5,6]. In general, high porosity
8 associated with cracks in the coating results in low thermal conductivity, but the shape of pores and cracks
9 and their size distribution may also influence considerably this parameter.

10 More recent research has focused on the use of nanostructured feedstocks [7] and they have demonstrated that
11 the nanostructured coatings have lower thermal conductivity than that of their conventional counterparts,
12 basically due to the presence of unmolten and porous nano-zones.

13 One way of obtaining nanostructured coatings consists of using a carrier liquid to inject a suspension of
14 nanoparticles inside the plasma instead of dry powder. This process, known as suspension plasma spraying
15 (SPS), has undergone extensive development in the last years due to the fact that the feedstock preparation
16 process is much easier than that of agglomeration of nanostructured solids by spray-drying. SPS coatings
17 with improved properties, in comparison with those of coatings obtained from conventional (microstructured)
18 APS feedstocks, have been reported [8]. The previous research has also shown that TBCs obtained by SPS
19 exhibit very low thermal conductivity and higher thermal shock resistance without substantially varying the
20 hardness or elastic modulus [9]. However there are only very few studies which investigate qualitatively the
21 relationship between microstructure and thermal conductivity [10].

22 From the above it can be deduced that nanostructured YSZ coatings obtained with the use of SPS process
23 have the potential to become part of the next generation of TBCs. Nevertheless, there is a need to
24 quantitatively assess the relationship between microstructure and thermal conductivity by modeling
25 important microstructural features. Thus the present research addresses a first attempt to present a simple,
26 quantified correlation between the cracks present in YSZ SPS coatings and thermal conductivity. The
27 coatings described in this work were obtained and characterized in our previous study [11].

2. Experimental

2.1. Materials and coating deposition

A commercial aqueous suspension of 3 mol% yttria-stabilized zirconia (YSZ) nanoparticles (MELox Nanosize 3Y, Mel Chemicals, United Kingdom) was deposited on AISI 304 stainless steel substrates by suspension plasma spraying (SPS). The coatings were obtained in a set of experiments following a factorial model which aimed to addressing the effect of the operational spray parameters on microstructure and mechanical properties of the sprayed coatings. The studied parameters are displayed in Table 1 and procedure details can be viewed at [11].

2.2. Coating characterization

The coating's cross-sections were characterized by field emission scanning electron microscopy (FEG-SEM, S-4800, Hitachi, SCSIE of the University of Valencia). In addition, the different types of cracks present in the coatings were identified and quantified by image analyzer (Visilog 5.3, Noesis S.A.). In order to obtain a representative, complete scan of the entire cross-sectional area, 10 FEG-SEM backscattered electron micrographs were analyzed (x1000 magnification) for each sample. Phase identification by X-ray diffraction as well as mechanical properties (hardness and elastic modulus) by nanoindentation are described elsewhere [11].

Thermal conductivity was calculated from the following equation:

$$\lambda(T) = \alpha(T) \cdot c_p(T) \cdot \rho \quad (1)$$

in which λ is thermal conductivity in W/(m·K), α is thermal diffusivity in m²/s, c_p is specific heat in J/(kg·K) and ρ is apparent density of the coating at room temperature in kg/m³. The specific heat was calculated by differential calorimetric technique following the equation:

$$c_p(T) = \frac{1}{m_g} \cdot \frac{\Delta H}{\Delta T} \quad (2)$$

in which ΔH is the enthalpy in J, ΔT is the temperature variation in K and m_g is the amount of free-standing coating in kg.

The coating porosity, which was also determined from the image analysis procedure set out above, was taken into account in order to determine the coating density. The coating density was calculated from the following expression:

$$\rho = \rho_0 \cdot (1 - P) \quad (3)$$

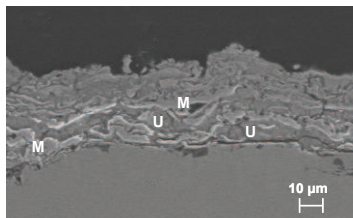
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2 in which ρ_0 is the non-porous material density of YSZ which was estimated on 5890 kg/m^3 [12] and P is the
3 porosity.

4 The thermal diffusivity measurements were performed by lamp flash method using LFA 447 NanoFlash[®]
5 Flash Lamp System (Netzsch, Germany). The setup enables measurements in the temperature range from
6 room temperature to 573 K in air. **Three measurements for each of the five samples were performed.**

7
8 The possible impact of the various types of cracks on thermal conductivity was studied. The correlations
9 were ratified by analysis of variance (ANOVA) method. Test F was performed in which critical F value
10 means the probability to exist a significant influence between the thermal conductivity and the considered
11 parameters. When critical F value is lower than 0.05 the correlation is valid. In contrast, when critical F
12 exceeds 0.05, the correlation is not valid and the number of parameters must be changed in order to reduce
13 critical F.
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22 3. Results and Discussion

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24 Cross-sectional FEG-SEM images of one of the as-sprayed coatings is shown in Fig. 1. The figure reveals the
25 two-zone microstructure which consists of nanostructured zones formed by unmolten nanoparticles (marked
26 U in Fig.1) surrounded by lamellar (splats), molten powder matrix (marked M in this figure). Similar
27 microstructures were observed in the rest of the coatings which correspond to those reported in the literature
28 for SPS-YSZ coatings [7].
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41 **Figure 1.**

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43 The arrows in Fig. 2 show that different types of cracks can be distinguished in the coatings as reported
44 elsewhere [4,13]. On the one hand, several microcracks are produced in the direction normal to the substrate.
45 These intra-splats cracks appearing through coating's thickness could have been generated during splat
46 quenching (*inner cracks*). Some of them grow from the bond coat-top coat interface to surface top coat
47 (*segmentation cracks*) and may affect the mechanical and thermal properties [14]. On the other hand, the
48 third type of cracks evolves parallel to the substrate at the molten grains boundaries (*inter-splats*) indicating
49 somehow poor lamellae cohesion (*horizontal cracks*). This crack morphology is depicted in Figure 2 together
50 with the original micrograph where the actual cracks come from. The percentage of each type of crack
51 assessed by image analysis is shown in Table 1.
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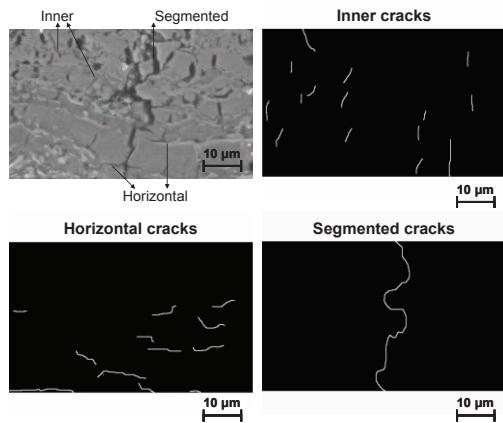


Figure 2.

Thermal diffusivity at different temperatures as well as specific heat and porosity were measured in order to calculate the thermal conductivity, which is also showed in Table 1.

As previously mentioned, the TBCs coatings obtained by SPS process exhibit very low thermal conductivities (typically 0.6-0.8 W/m·K) being lower than those of the coatings obtained by conventional atmospheric plasma spray (APS) process (usually between 0.8-1.1 W/m·K at room temperature [3]).

According to Table 1, the samples 2, 3 and 4 had the smallest thermal conductivity. The sample 3 displayed an extremely low thermal conductivity of $\lambda=0.43$ W/m·K at 573 K. The explication of such low thermal conductivity in YSZ SPS coatings can be the peculiar pore network architecture mainly made up of nanometric-sized pores [15]. In addition, this pores network is difficult to quantify using conventional techniques due to their low resolution. In the present research porosity values (determined from FEG-SEM images) ranged from 6 % ± 1 to 8 % ± 1 which correspond to typical SPS coatings. Despite the difficulty in quantifying this porosity, the significant differences in thermal conductivity among the coatings confirm the strong dependence of this characteristic with the coatings microstructural features.

Different correlations for the all five samples data according to the types of cracks set out in Fig. 2 were attempted. Correlation coefficient (R^2) and critical F of these correlations which are detailed in Table 2, change according to the kind of crack that are considered.

First, in case 1, each type of crack (*inner*, *segmentation* and *horizontal*) was analyzed separately. Although correlation coefficient is higher than 0.9, critical F value exceeds 0.05 in all the cases therefore the correlation is invalid. This is because only one degree of freedom is taken into account. For this reason, the number of parameters was reduced in order to increase the degrees of freedom.

For this purpose, in case 2, *inner* and *segmentation* cracks were linked in the same type of crack now called *vertical* cracks. In fact as reported elsewhere *segmentation* cracks develop from inner cracks when a low energy release rate is promoted during splat quenching [16]. Thus SPS coatings form easily segmentation

1 cracks due to the low energy release rate associated with their thinner splats [14]. Critical F was still higher
2 than 0.05 but lower than that of the first case. Moreover, relative error as worked out from correlation
3 coefficients was very high, in particular the contribution to this error by *horizontal* cracks. Again the
4 correlation was weak.
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7 Finally, in case 3, only *vertical* cracks (sum of *inner* and *segmentation* cracks) were considered. In spite of a
8 R^2 value little lower than that of the other cases, critical F value was lower than 0.05 with an acceptable error
9 what indicates that a correlation between *vertical* cracks and thermal conductivity does exist. A positive
10 correlation number means that a higher *vertical* cracks density gives rise to an increase of thermal
11 conductivity. Although it is widely accepted that defected microstructure made up of pores and cracks
12 contribute substantially to the reduced thermal conductivity of YSZ TBC coatings, *vertical* and in particular
13 *segmentation* cracks are extensively reported to produce the opposite effect. In fact, the thermal conductivity
14 of coatings with high *segmentation* cracks density can rise up to 50% [16]. This is because the contribution of
15 the flow of high temperature gas into the pores to heat transfer makes more significant [13]. Recent FEM
16 modeling of TBCs has also confirmed the dominant role of the longest transverse crack to the thermal
17 conductivity of the coating when compared with any individual defect [17].
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20 The findings obtained allow confirming some of the previous research carried out with YSZ TBC coatings
21 obtained by APS processes which have been extensively treated in the literature due to their industrial
22 applications. However further research is still necessary so as to better understand the correlation between
23 microstructure and thermal properties in SPS coatings since as widely recognized nanostructured YSZ
24 coatings show clear potential to become part of the next generation of TBCs.
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27 4. Conclusions

28 3YSZ coatings were obtained by SPS from an aqueous commercial suspension. Thermal diffusivity was
29 measured using commercial NanoFlash system in the temperature range from room temperature to 573 K.
30 Thermal conductivity was then calculated from thermal diffusivity measurements. The different types of
31 cracks were identified by SEM inside the coating microstructure namely; *inner* microcracks and
32 *segmentation* cracks associated with splat quenching process and *horizontal* interlamellae cracks.
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35 Very low thermal conductivities values were obtained which did not correlate with coatings porosity data.
36 ANOVA test was performed to try to relate these types of cracks identified in the coatings with thermal
37 conductivity values. A good correlation between *vertical* cracks which include *inner* microcracks and
38 *segmentation* cracks and thermal conductivity was found. The findings also confirmed the increase of
39 thermal conductivity associated with this type of cracks as reported in the literature.
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Figure captions

Figure 1. FEG-SEM cross section micrographs (sample 2 from reference [11]).

Figure 2. Different types of cracks identified by image analysis from the original micrograph of sample 5 (reference [11]).

Tables

Sample	Spray parameters		Type of cracks (%)			Thermal conductivity (W/(m·K))	
	Linear speed (mm/s)	Injection pressure (bar)	Segmentation	Inner	Horizontal	T=323 K	T=573 K
1	250	1.5	2.16	3.18	3.63	1.3±0.1	1.3±0.1
2		3.5	-	2.33	2.33	0.7±0.1	0.6±0.1
3	300	2	-	1.38	3.70	0.5±0.1	0.4±0.1
4	350	1.5	-	1.72	1.98	0.6±0.1	0.5±0.1
5		3.5	2.74	2.34	4.36	0.9±0.1	0.9±0.1

Table 1. Spray parameters, cracks percentage and thermal conductivity of the coatings (samples reference from [11]).

Case	Degree of freedom	Critical F	Variables (type of cracks)	Coefficients	Error (%)	R ²
1	1	0.37	Segmentation	0.048	401	0.91
			Inner	0.392	65	
			Horizontal	0.046	429	
2	2	0.14	Vertical	0.193	33	0.86
			Horizontal	-0.070	173	
3	3	0.03	Vertical	0.170	26	0.83

Table 2. Correlation coefficients and results of the ANOVA test.