Large dynamic scissoring mode displacements coupled to band gap opening in the cubic phase of the methylammonium lead halide perovskites

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Abstract. Hybrid perovskites are a rapidly growing research area, having reached photovoltaic power conversion efficiencies of over 25 %. There is a increasing consensus that the structures of these materials, and hence their electronic structures, can not be understood purely from the time and space averaged crystal structures observable by conventional methods. We apply a symmetry-motivated analysis method to analyse X-ray pair distribution function data of the cubic phases of the hybrid perovskites MAPb X_3 (X = I, Br, Cl). We demonstrate that, even in the cubic phase, the local structure of the inorganic components of MAPb X_3 (X = I, Br, Cl), are dominated by scissoring type deformations of the PbX_6 octahedra. We find these modes to have a larger amplitude than equivalent distortions in the A-site deficient perovskite ScF_3 and demonstrate that they show a significant departure from the harmonic approximation. Calculations performed on an inorganic perovskite analogue, FrPbBr₃, show that the large amplitudes of the scissoring modes are coupled to a dynamic opening of the electronic band gap. Finally, we use density functional theory calculations to show that the organic MA cations reorientate to accommodate the large amplitude scissoring modes.

1. Introduction

Molecular perovskites, also known as hybrid perovskites, are a fast growing research area in photovoltaics, due to their low cost to make and rapid increase in efficiency (from 3.9 % in 2009[1] to > 26 % today[2, 3, 4, 5, 6]). These materials have the familiar topology, connectivity and chemical formula of traditional perovskites (ABX_3) , but where they differ is that the A site cation is organic. The most frequently studied of this class of materials are the methylammonium (MA) lead halides, which have the general formula $CH_3NH_3PbX_3$ (X = I, Br, Cl), commonly abbreviated to MAPbX₃. In addition to their high conversion efficiency, this class of hybrid perovskites have other desirable photovoltaic properties, such as long charge carrier lifetimes[7], mobility[8] and diffusion lengths [9], a high absorption coefficient [10], and a direct band gap[1]. These properties couple together to create a device that has a high density of charge carriers with a strong barrier against recombination, all whilst needing much less material than traditional solar cell materials, and without the need for a high energy input manufacturing process[11].

Whilst perovskite oxides are a well studied class of materials due to the wide range of desirable properties exhibited by them, less is understood about the structure-property relationship in halide perovskites, particularly the hybrid perovskite family. Having a methylammonium ion rather than a metal ion at the A site results in the A site possessing an electric dipole moment rather than a point charge, so the dynamics of these ions are the focus of a lot of research in these hybrid perovskites. In the higher temperature tetragonal and cubic phases, the alignment of the ions appears to be disordered [12, 13, 14, 15], however they could form small domains below the length scale required for coherent diffraction where the molecules are aligned. The dynamics of their rotations, and any local order, could have a large contribution to the properties of the material. For example, the interaction between phonons and the rotational degrees of freedom of the MA cations has been shown to have an impact on thermal conductivity[16]. The changes in dynamics are thought to be closely linked to the structural changes of the material with temperature, and it is still unknown how the dynamics affect the properties of this material as a photovoltaic. Another question that has still not been fully solved is whether the configuration of the MA cations lead to this class of materials being ferroelectric[12, 17, 18, 19, 20, 21, 22].

Use of X-ray single crystal and powder diffraction has led to a good understanding of the different structural phases of these materials. Similarly to a large number of perovskites, all of the single-halide MAPbX₃ materials have cubic symmetry at high temperatures and undergo symmetry-lowering phase transitions to tetragonal and orthorhombic structures at lower temperatures [23]. Most experimental studies agree that there are 3 structural phases for $MAPbI_3$ and MAPbCl₃, however there is a 4^{th} phase for MAPbBr₃ which is preferred for a small temperature range (ca. 150-155 K), commonly thought to be an incommensurate phase[24]. In the cubic phase, the MA cation is thought be fully disordered, with recent advances made using techniques such as NMR and quasi-elastic neutron scattering showing that the MA cation is close to having the orientational freedom of a free MA cation [25, 26]. As the inorganic framework undergoes structural phase transitions, lowering the average symmetry from cubic $Pm\bar{3}m$, the orientational freedom of the MA cation is restricted, becoming fully ordered in the orthorhombic phases [27, 14]. This shows that the organic molecular and inorganic framework dynamics in MAPb X_3 are inherently linked [28, 29]. In addition to experimental studies, computational methods have seen a lot of use in this, and other, areas of research in hybrid perovskites [30, 31, 32, 33, 34].Both classical molecular dynamics and Density Functional Theory (DFT) simulations have demonstrated a link between the different phases of MAPbI₃ and the preferred orientations of the MA cations[35]. Work from Quarti et al has demonstrated that the configuration of the MA cations has a significant effect on the properties of the material, such as its electronic band structure [36, 37]. This underlines why it is important to fully understand the structureproperty relationship in these materials. Despite the knowledge that the organic molecular and inorganic framework dynamics are linked *via* hydrogen bonding interactions, it is currently unclear how this interaction affects the dynamics as a whole.

The bands forming the top of the valence bands and the bottom of the conduction band in the electronic structure of the methylammonium lead halides will be dominated by Pb and X (X = I, Br, Cl) electrons[38, 29]. It has been demonstrated that phonons involving distortions of bonds between these elements interact with electrons and holes in the band edges, leading to effects such as ultrafast intraband charge carrier relaxation[39, 40]. Therefore, regardless of the role of the MA cation in stabilising particular distortions, it is necessary to establish good models for the dynamic distortions in the PbX_3 framework. In this work, we aim to probe the dynamics of the inorganic framework of the cubic phases of the three single-halide $MAPbX_3$ materials. We have recently demonstrated how by using a symmetry motivated approach to analysing pair distribution function (PDF) data we can gain extra information on disorder and dynamics within a system. Our



Figure 1. Pair distribution functions of MAPbBr₃ are shown in three different phases (orthorhomic, tetragonal and cubic, shown top to bottom). Each PDF is shown with a offset of 4 between them. Similar plots for X = I and Cl are given in the Supplemental information.

study on BaTiO₃ has shown that this method is very sensitive to primary order parameters and is a powerful tool to analyse order-disorder phase transitions[41, 42]. Both this study and our more recent work on the negative thermal expansion materials ScF₃, CaZrF₆[43] and ReO₃[44] has demonstrated that this method is also sensitive to soft phonon modes and has also revealed substantial deviations from the crystallographic average structure in these materials. Here, we use X-ray total scattering data, which is much more sensitive to the inorganic framework than the molecular cations, to probe the characters of the low lying excitations of the cubic phases of the methylammonium lead halides.

2. Experimental Details and Data Analysis

MAPbI₃ was prepared using the inverse temperature crystallisation method[45]. Briefly, equal molar amounts of MAI and PbI₂ were dissolved in a solvent (γ -butyrolactone) at room temperature. Then the obtained MAPbI₃ solution was heated to 110 °C for the crystal growth. Powder samples of MAPbBr₃ were prepared by the reaction of stoichiometric amounts of lead acetate and methylamine hydrobromide in



Figure 2. For each compound, the best individual fitting statistic is plotted for each irrep at each temperature. The Rfactor is shown relative to the R-factor for the refinement with no symmetry adapted displacement modes active in the refinement, such that a more negative value corresponds to an increased quality of fit. The temperature shown is relative to the cubic phase transition (T_1) as reported in the literature, such that the measurement temperature $T = T_1 + \Delta T$. T₁ for each compound is indicated on the plot. The irreps are labelled as follows: colour denotes the k-point of the irrep, with blue referring to the M-point, green to X, pink to R and yellow to Γ ; marker shape denotes the irrep number, with a circle referring to 1, an upward-pointed triangle to 2, a star to 3, a square to 4 and a downward-pointed triangle to 5; linestyle denotes the parity of the irrep, with a solid line referring to a "+" irrep, and a dashed line referring to a "-" irrep.

hydrobromic acid. The excess acid was then evaporated to leave an orange colored product which was washed with diethyl ether. Powder samples of MAPbCl₃ were prepared out of a solution of methylamine hydrochloride and lead acetate dissolved in hydrochloric acid. An excess of an approximately 8-10 molar ratio of methylamine hydrochloride was required to obtain these phase pure samples. The resulting powder was washed with diethyl ether.

For MAPbBr₃, synchrotron radiation X-ray total scattering experiments were conducted at the synchrotron facility PETRA III (beamline P02.1[46]) at DESY, Hamburg. A wavelength $\lambda = 0.2070 \text{ Å}$ was used to collect data. Data were collected at temperatures of 125, 140, 147, 152 K and at intervals of 25 K from 175 to 450 K. For MAPbI₃ and MAPbCl₃, Synchrotron radiation X-ray total scattering experiments were conducted at the synchrotron facility Diamond Light Source (beamline I15-1). A wavelength of $\lambda = 0.161669$ Å was used to collect data. Data were collected at 20 K intervals over the temperature ranges $100 - 460 \text{ K} (\text{MAPbCl}_3)$ and 100 - 560 K (MAPbI₃). The obtained 2D images were masked and radially integrated using the DAWN[47] software. G(r) and D(r) functions were computed using GudrunX[48], using Q_{max} values of 21,30 and 28 Å^{-1} for MAPbBr₃, MAPbCl₃ and MAPbI₃ respectively. GudrunX was also used to perform background subtraction, sample absorption and fluorescence corrections.

Analysis of the pair distribution functions was carried out using the symmetry-adapted PDF analysis (SAPA) method described in ref. [49] and implemented in our work on the local structures of $BaTiO_3$ [41], ScF_3 [43] and ReO_3 [44]. We consider only PDF data in which MAPbX₃ (X = I, Br, Cl) have cubic average symmetry in the following analysis. For each sample, a 2 \times 2 \times 2 P1 supercell of the $Pm\bar{3}m$ aristotype PbX₃ with Pb at (0.5, 0.5, 0.5) and X at (0.5, 0.5, 0) was generated and parameterised in terms of symmetry adapted displacements using the ISODISTORT software [50]. The generated mode listings were output in .cif format and then converted to the .inp format of the TOPAS Academic software v6 using the Jedit macros[51] and Python code[49]. In total, there were 96 modes which transformed according to one of 19 irreducible representations The physical characters associated with (irreps). the irreducible representations, with respect cubic perovskite structure, are detailed in [52]. The supercells were generated without the organic Asite cation included, since the contribution of pairs involving the organic components of the structure will have a negligible contribution to the overall PDF due to their comparatively weak scattering power for X-rays. This lack of sensitivity of X-ray total scattering to the organic elements of hybrid perovskites can be seen by

comparing recent publications by Malavasi et al [53, 54, 55]. For each irreducible representation (irrep) at each temperature, refinements of the corresponding modes were started from random starting mode amplitudes. This was repeated 500 times. For all samples, the refinements were carried out with a fitting range of 1.7 to 20 Å. This starting value for the fitting range was chosen to exclude any contribution from the intraatomic peaks of the MA cations. Refinements were also tested using a fitting range with a maximum of 10 Å and found to be broadly similar. When reported, mode amplitudes use the supercell normalised amplitude as defined by ISODISTORT. This amplitude for a given irrep is the square root of the sum of squares of the normalised mode amplitudes for all distortions transforming as that irrep.

The DFT calculations were performed using the Vienna Ab Initio Simulation Package (VASP)[56, 57, 58, 59, version 5.4.4. We employed the optB86bvdW exchange correlation potential[60] which includes VdW corrections previously found to suit hybrid perovskites[29]. Projector augmented-wave (PAW) pseudopotentials[61, 57] were utilised, as supplied within the VASP package. A plane wave basis set with a 1100 eV energy cutoff and a $4 \times 4 \times 4$ Monkhorst-Pack k-point mesh with respect to the parent cubic primitive cell (scaled accordingly for other supercells) were found suitable. The energy landscape of the various modes in the hybrid system were studied by fixing the halide framework while allowing for Pb and MA to relax until the forces were less than 5 meV/Å. Results were compared with FrPbBr₃, which we used as a hypothetical inorganic analogue to the hybrid perovskite, since Fr best matches the ionic radii of MA[29].

3. Results and Discussion

A key aspect of the local structure of the MAPb X_3 (X = I, Br, Cl) family of hybrid perovskites is that the first four peaks of the inorganic component of the PDF do not change much beyond that expected for simply changing the temperature, i.e. a change in peak width corresponding to a change in thermal energy, and a change in peak position corresponding to thermal expansion. For MAPbBr₃ (Fig. 1) and MAPbCl₃, the peaks stay the same from the low temperature orthorhombic phase into the high temperature cubic phase [54, 53, 55]. For MAPbI₃, there is a slight change upon the transition between the tetragonal and orthorhombic phases, but the peaks from the tetragonal phase persist in the cubic phase [62] (see Supplemental Information [?]). This has been taken to imply that the cubic phase consists of local symmetrybroken domains and there has been recent work to



Figure 3. (a-c) A breakdown of the atomic basis that spans the X_5^+ irrep. Shown in (d) is the structure resulting from a refinement of the *Pnma* order parameter direction of the X_5^+ irrep. (e) Mode energies with varying distortion mode amplitude for the *Pnma* order parameter direction of FrPbBr₃. Harmonic (2^{nd} order) and anharmonic (2^{nd} and 4^{th} order) fits to the potential well are shown. (f) Mean displacement values for the general X_5^+ order parameter direction in the cubic phase of MAPbX₃, obtained from SAPA refinements against experimental data.

support this hypothesis[62]. This would suggest that the distortions most responsible for the local structure should be the rigid-unit modes (RUMs) that drive these phase transitions.

To gain a more robust understanding of the local structure of MAPb X_3 (X = I, Br, Cl) in the cubic phase, we perform symmetry-adapted PDF analysis (SAPA)[43, 41] to elucidate the character of the dominant lattice dynamics associated with the inorganic cage. We note that we are insensitive to the orientation and displacement modes of the organic component in the present X-ray PDF study, and so no attempt is made to model these against The symmetry-adapted the experimental data. displacements which show the most improvement in the fitting statistic (R_w) for the models refined against the PDFs, for all compositions and all temperatures, are those which transform according to irreps that permit a scissoring motion of the X anions, *i.e.*, the Br-Pb-Br bond angles are distorted away from 90° but the Pb–Br bond lengths remain undistorted. An important additional note is that when the symmetry analysis is extended to include the MA cations, these scissoring modes also allow for multipolar ordering of the cations [63]. This result of the SAPA analysis does not imply that the RUMs are high energy modes,

it simply means that the majority of the motion of the halide anions arise from these scissoring modes at the temperatures studied. There is a likely entropic origin to this observation; the scissoring modes occupy a much larger volume in reciprocal space, as RUMs are restricted to a single line. This is supported by competitive two-phase refinements of the PDFs, in which we allow the X_5^+ scissoring-type displacements to refine in one phase and the displacements for one of the RUMs $(R_5^- \text{ or } M_2^+)$ in the other. These refinements show a preference for scissoring modes compared to the RUMs for all 3 samples, as evident from the refined scale factors of the two phases which show an approximate scissoring:rotation ratio of 2.3:1 (see the Supplemental Information [?] for more details). For context, this ratio is approximately 4:1 in ScF₃, which is isostructural to the inorganic framework of MAPb X_3 . The lower ratio compared to ScF₃ reflects a lower flexibility due to the presence of an A-site cation, which can interact with the inorganic framework via hydrogen bonding[28]. However, it is clear from our results that the majority of the halide anion motion in the cubic phase still arises from scissoring-type deformations of the octahedra.

The above results are in line with a recent reverse Monte Carlo (RMC) analysis of neutron PDFs of $MAPbI_3[64]$ between 10 and 400 K. This study demonstrates that a bending of the Pb-I-Pb bond angle dominates the local distortions of the PbI₆ octahedra. Our results show that the four best fitting modes all have scissoring character, of which it is the X_5^+ (Fig 3) (c&d)) that performs best across all three compositions and temperatures. This could be due to the fact that there are more degrees of freedom transforming as X_5^+ (18) than for other irreps (12 for M_5^- and 6 for $X_5^$ and M_5^+), but the improvement could arise from the anti-polar Pb displacements that enter into the irrep X_5^+ , although this is unlikely since they only have a small contribution to the overall displacements. The three distortions that span this irrep are shown in Fig. 3. We find the amplitudes of these scissoring modes to be quite large; refinements of X_5^+ and X_5^- in the tetragonal phase of MAPbBr₃ resulted in supercellnormalised mode amplitudes of ≈ 1.35 Å. This is close in magnitude to the equivalent amplitude of the R_5^- distortion (≈ 1.65 Å) which is frozen into the structure in the tetragonal phase. The domination of these scissoring modes in the local structure of $MAPbX_3$ reflects the high occupation these modes Experimental and computational work by have. Giustino *et al* has demonstrated the significance these scissoring types of modes have in determining the optoelectronic properties of hybrid perovskites, in particular limiting carrier mobility and increasing polaron effective masses[39, 65].

Given how large the local deviations are from the average cubic structure, it is reasonable to assume they will have a substantial effect on the band structure. We used DFT calculations to investigate the impact that the scissoring modes could have on the electronic band structure of the hybrid perovskites in the cubic phases. We chose to analyse MAPbBr₃, since it is cubic at room temperature where experimental band gap values have been reported, and to focus on the two X point modes that do the best job at describing the deviations away from local cubic symmetry, as evident in the PDF data. For a completely unrestrained order parameter direction transforming as X_5^+ , there are a rather large number of degrees of freedom (18 in total), so, to make our results more robust, and to facilitate a direct comparison to X_5^- , we take results from refinements using higher symmetry OPDs with no more than 5 parameters. We use structures from refinements against our data with X_5^+ OPDs with *Pnma* and *Cmcm* symmetry ((0, a; b, 0; 0, c) and (0, a; b, b; a, 0)respectively) and the X_5^- OPD with C2/c symmetry ((a,b;c,-c;-b,-a)) as input to our band structure calculations (See Supplemental information). For the two X_5^+ OPDs, only Br anion displacements were refined when generating the CIFs for the band structure calculations, although by symmetry, Pb

displacements also enter into the irrep. For X_5^- , Pb displacements are forbidden by symmetry. We also sampled points of different overall distortion amplitude along the X_5^+ OPD with *Pnma* symmetry and calculated the energy. These energy calculations were performed for the FrPbBr₃ structures used to calculate the band structure.

In the undistorted cubic phase, the calculated band gap was 1.717 eV, which is slightly higher than other calculated band gaps for cubic MAPbBr₃ (1.64 eV[66]) at the same level of theory, and is direct. Previous work has shown that substitution of Fr for MA opens up the band gap slightly in orthorhombic MAPbI₃[29]. For each tested distortion, the band gap opens up significantly to values of 2.025, 2.138 and 2.162 eV for the C2/c, Cmcm and Pnma distortions with an amplitude of $0.8 \times$ the maximum amplitude refined from PDF data for the X_5^+ distortions and $1.1 \times$ the maximum amplitude for X_5^- , respectively, and remains direct. These relative amplitudes were chosen so all 3 distortions were at similar mode amplitudes. These values are closer to the experimentally determined band gaps for MAPbBr₃ of ≈ 2.3 eV at room temperature[67], although this is likely due to a cancellation of errors. The distortions result in a reduced orbital overlap between Pb and Br p-orbitals, leading to a lower band curvature and therefore an increased effective mass in the distorted band structures (Fig 4 and Supplemental Information). The mobility of polarons is inversely proportional to the electron band effective mass [68], and this increased effective mass in the distorted structures may explain the discrepancy between experimental and calculated values[69].

Spin-orbit coupling (SOC) interactions, which play a large role in systems involving Pb, have not been accounted for. Consequently, the exact shape of the electron bands and size of the band gap won't be accurate, since inclusion of SOC has been shown to lead to unconventional dispersion relations [70]. The effects from SOC on band gap size in halide perovskites tend to be cancelled out by full treatment of electron Coulomb interactions beyond DFT[71, 72] and the principle effect is a renormalisation of the band gap[73]. Therefore, the trends we detect due to the different distortion modes will remain the same. In the two X_5^+ distortions, the degeneracy of the bands at the conduction band minimum (CBM) at the Γpoint are broken, leading to fewer available states at the CBM. Contrastingly, the X_5^+ Pnma distortion appears to have the largest DOS at the valence band maximum due to the reduced bandwidth. Fluctuations in the band gap of hybrid perovskites due to their highly dynamic structure has been previously predicted in both hybrid[37] and all-inorganic halide perovskites?], and is expected to assist the initial stages of charge separation. In addition, an increase of the band gap coinciding with a transverse displacement of I ions in MAPbI₃ due to an external strain field has been reported[38].

Our refinements against the PDF data show that all three models for the X_5^+ displacements have a large amplitude, with supercell-normalised mode amplitudes of 1.84, 1.82 and 1.36 Å for OPDs with Pnma, Cmcm and C2/c symmetries, respectively. These mode amplitudes correspond to maximum Br displacements of 0.486, 0.410 and 0.350 Å. Note that the refined distortions correspond to a time-averaged view of the structure, so these maximum Br displacements are a factor of $\sqrt{2} \times$ greater, in the harmonic approximation, than those found in the refined structures. As a consequence of their large amplitudes, the distortions would be expected to be anharmonic in nature, which is supported by the potential energy well we calculate for the X_5^+ (0, a; b, 0; 0, c) OPD in FrPbBr₃ (Fig 3 (f)), which has a significant quartic component when fit with a 4th order polynomial fit ($\Delta E = 127x^4 + 79.7x^2$ meV/f.u., where x is the distortion mode amplitude relative to its maximum value at 400 K). The significant degree of anharmonicity in this character of distortion will contribute to the ultra-low thermal conductivity observed in both hybrid and inorganic halide perovskites [74, 75, 76, 77]. Despite the presence of an A-site in these materials, the amplitude of these scissoring modes are greater than those in ScF_3 , suggesting the MA cations move to accommodate the large-amplitude modes. The implication of this, then, is that the band gap opening we detect as a response to the scissoring modes is likely influenced by the dynamics of the MA cations, although our refinements are only sensitive to the inorganic framework.

To investigate the above hypothesis, we consider the X_5^+ OPD with *Pnma* symmetry. This breaks the equivalency of the cubic < 100 > directions and leads to two distinct A-site symmetries (see Supplemental Information). Therefore, if the inorganic and organic dynamics are coupled together, we would expect to see the MA cations located at different points of the unit cell to respond differently to the distortion mode, to reflect the different local environments they would experience. To test this, we relaxed the 8 MA cations in the $2 \times 2 \times 2$ supercell from an initial anti-polar configuration with the C-N bonds aligned with the cubic $\begin{bmatrix} 1 & 0 & 0 \end{bmatrix}$ direction, in a structure with a $0.8 \times X_5^+$ (0, a; b, 0; 0, c) distortion (relative to the maximum amplitude at 400 K) frozen in. The MA cations showed significant reorientation, with the 4 "edge" ((0.5, 0, 0) and equivalents) and 4 "corner" ((0, 0, 0)) cations rotating to include significant components along c. There is a split amongst the

"face" cations, with two (at (0.5, 0.5, 0) and (0, 0.5, 0)(0.5)) rotating to include smaller components along the b- and c-axes. The remaining "face" cation and the cation located at the centre of the supercell both rotate to include a significant component along c and a smaller component along b. In all, there are 5 distinct C-N bond alignments, which coincides with the 5 distinct Br sites resulting from the X_5^+ OPD with Pnma symmetry. In addition, all cations show a slight displacement from the high-symmetryunique positions. Full details can be found in the Supplemental Information. This demonstrates that the MA cations can rotate to accommodate the distortions of the inorganic framework, indicating that the dynamics of the two components of the structure are linked. This is further supported by a 4^{th} order polynomial fit to the potential well, which, compared to the FrPbBr₃ analogue, shows a much reduced quartic component (($\Delta E = 60.0x^4 + 106x^2$) meV/f.u., where x is the distortion mode amplitude relative to its maximum value at 400 K). However, it is important to note that our calculations are effectively performed at 0 K, where the ground state is the fully ordered orthorhombic phase. It is quite possible that the configurational entropy associated with the MA orderings may effectively act to decouple these dynamics at higher temperatures in the cubic phase. Indeed, there is evidence to suggest the organic and inorganic dynamics are decoupled in $MAPbCl_3[15]$. Additionally, a similar computational result in CsPbBr₃ showing coupling between large amplitude distortions of the Br ions and head-to-head Cs motion [78] suggest this feature is not exclusive to hybrid inorganic systems. We have also shown that acoustic phonon lifetimes for the all-inorganic CsPbBr₃ are very similar to those in $MAPbCl_3[79]$, further supporting the idea that at high temperatures the MA rotational modes may have little effect on the lattice phonon modes.

There has been recent literature supporting the idea that cubic halide perovskites, rather than being treated as a single repeating unit, should be thought of as a network of polymorphs showing different symmetry-lowering deformations of the average structure, such as varying degrees of octahedral tilting or differing amplitudes of B-site displacement[80, 81]. The stereochemical behaviour of the Pb cation, in conjunction with the coupling between organic cation and inorganic framework dynamics, is likely to have a large impact on the possible polymorphs the material exhibits within this hypothesis of the nature of the structure of halide perovskites. However, our experimental work finds that even on a length scale of 20 Å, the local dynamic deviations from the cubic symmetry are dominated by distortions of very specific character, namely



Figure 4. The calculated electronic band structure of FrPbBr₃ for the undistorted structure (a), the *Pnma* and *Cmcm* order parameter directions of the X_5^+ irrep (b and c, respectively) and the C2/c order parameter direction of the X_5^- irrep (d). These figures were created using sumo[85]. The wave vectors used are described in detail in Supporting Information.

of the scissoring mode type. This likely arises due to the low energy and thus high occupation of these modes. There is a significant body of work suggesting that the cubic phases of hybrid perovskites are formed of dynamic domains of the lower symmetry structures exhibited at lower temperatures[53, 62, 54, 55, 82]. Some further work may be required to reconcile these two pictures, but the answer may lie in the significant vibrational entropy that is no doubt associated with the scissoring modes at non-zero temperatures. What is clear is that the dynamics of these systems depart significantly from the usual quasi-harmonic normalmode model typically used for crystalline solids[83, 84].

In summary, we have shown that large scissoring modes of the halide ions describe the dominant deviations from the average structure in the cubic phases of the hybrid perovskites. These modes have a similar amplitude to those of the static rigid unit modes below the phase transition temperature (\mathbf{R}_5^- (a,0,0) for MAPbI₃ and MAPbBr3, $\mathbf{R}_5^- \bigoplus \mathbf{M}_2^+$ (a,0,0|0;b;0) for MAPbCl₃‡). These distortions have the effect of opening up the band gap of the electronic structure of the cubic phases. In addition, we have shown that the organic cations move to accommodate the large amplitude distortions of the inorganic framework, suggesting that the dynamics of the two components are inherently linked, and that the inorganic lattice is significantly distorted from the average at a local level. Since the experimental probe used for this study is not sensitive to the organic cations, we cannot say anything concrete about the dynamics of the MA cations. However, group theoretical treatment of the scissoring mode distortions^[63] identified in the present work show that they would be expected to couple to the dipole (X_5^-, M_5^-) and quadrupole (X_5^+, M_5^+) moments of the MA cations, which is in contrast to the octahedral tilting modes (R_5^-, M_2^+) , providing a possible explanation as to why the former dominate the dynamics observed in the present study. These dynamic structures should be accounted for in simulations performed on the hybrid perovskites, since they have a significant effect on the calculated properties.

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Data Availability Statement

The data that support the findings of this study openly available.

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 $[\]ddagger$ Strictly speaking, the exact nature of the tetragonal phase of MAPbCl₃ is still disputed, so we have here provided the OPD for the observed tilting pattern in the orthorhombic phase

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