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Cr as a promoter for the In₂O₃-catalyzed hydrogenation of CO₂ to methanol



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ABSTRACT

The utility of Cr as a promoter for In₂O₃ catalysts in the hydrogenation of CO₂ to methanol is investigated. Uniform precursors to binary CrOx-In2O3 and ternary NiO-CrOx-In2O3 catalysts are prepared by flame spray pyrolysis. For the CrO_x-In₂O₃ samples, the highest methanol rate is obtained at a Cr content of 2 mol%, exceeding the methanol rate of In₂O₃ by 55 %. With increasing Cr content, the CO₂ conversion rate does not increase, albeit the methanol selectivity decreases. Characterization of the samples supported by density functional theory calculations provides insight into the role of Cr. At low content, Cr is mainly doped into the lattice of In_2O_3 , which leads to more oxygen vacancy (O_v) sites. The In₂O₃ surface sites close to Cr-oxide clusters present on the surface are also activated towards Ov formation, offsetting the decrease in Ov due to coverage of the In2O3 surface by Cr-oxide with increasing Cr content. Cr₂O₃ dispersed on the surface of the In₂O₃ particles suppresses sintering of In₂O₃ under reducing conditions, which is especially evident at a Cr content above 2 mol% and higher reaction temperatures. Introducing Ni to the CrOx-In2O3 catalysts results in a higher methanol formation rate compared to CrO_x-In₂O₃. The methanol rate increases with the Ni content with the highest activity obtained at Ni and Cr contents of 22 mol% and 8 mol%, respectively. The optimum Ni(22)-Cr(8)-In₂O₃ catalyst displays twice the methanol rate of a Ni(22)-In₂O₃ reference. Ni and Cr play different promoting roles in achieving an increased and more stable rate of methanol formation compared to In2O3: Ni promotes the hydrogenation of formate and methoxy surface intermediates to methanol, while Cr results in more O_v sites and suppresses sintering of In₂O₃.

1. Introduction

 CO_2 hydrogenation to methanol is expected to become an important technology in the energy transition, because methanol is a versatile energy carrier and chemical building block, contributing in this way to closing the carbon cycle [1–4]. Among the many catalysts investigated, conventional Cu-based catalysts are less preferred for CO_2 hydrogenation to methanol due to their reverse water–gas shift activity and deactivation caused by active phase sintering [5,6]. In₂O₃ stands out because of its high methanol selectivity in CO_2 hydrogenation [7]. The active sites are comprised of oxygen vacancies in the In₂O₃ surface and adjacent In-O sites, which are involved in CO_2 adsorption and H₂ dissociation, respectively [8]. For practical applications, the activity of In₂O₃ is too low and it also deactivates during CO_2 hydrogenation [9].

The use of promoters for In_2O_3 has been well investigated. Pérez-Ramírez's group systematically examined the role of transition metal promoters, such as Pd, Pt, Ru, Rh, Ir, Ag, Au, Co, and Ni, introduced by flame spray pyrolysis (FSP) on methanol formation [10]. The presence of well-dispersed In_3M or In_2M_2 ensembles (M = Pd, Rh, Pt, Ru, Ir) on the In_2O_3 surface significantly enhanced the methanol yield by facilitating homolytic H_2 dissociation. Additives like Ag and Au were found to rapidly sinter on the In_2O_3 surface, which hinders H_2 homolytic splitting. The resulting particles also obstruct the active sites of In_2O_3 , explaining the strongly reduced catalytic activity in the presence of Ag and Au. It has also been observed that a too high rate of H_2 activation can have a detrimental effect on catalyst stability because of overreduction of In_2O_3 , resulting in metallic In, which is also prone to agglomeration. We also reported a similar promoting role of Ni clusters as a dopant in H_2 activation for In_2O_3 [11,12]. The same group also used density functional theory calculations to investigate the impact of single atoms (SAs) of Ni, Pd, Pt and Rh on In_2O_3 on CO_2 hydrogenation over In_2O_3 , mainly confirming that CO is the main product [13].

In recent years, Zr has also emerged as a promising structural promoter, improving the thermal stability of In_2O_3 . Yang et al. investigated the influence of the ZrO_2 phase on the methanol yield [14]. Among these phases, monoclinic ZrO_2 (m- ZrO_2) increased the methanol yield the

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most, which is thought to include an effect of electron transfer from monoclinic ZrO₂ to In₂O₃. The resulting electron-rich In₂O₃ phase exhibits enhanced H₂ dissociation and hydrogenation activity of surface intermediates toward methanol [14]. Zhang et al. showed that specific InOx species on m-ZrO2 play an important role in methanol formation [15], emphasizing that the dynamic behavior of In₂O₃ on m-ZrO₂ surface with temperature can affect the product distribution. It was observed that, at high temperatures (~400 °C), In was included in m-ZrO₂, forming a solid solution in which In cations are present in the surface region of the zirconia support. This transformation lowers the number of active sites and, thus, the catalytic activity. The resulting catalyst showed a relatively high activity in the reverse water-gas shift reaction. By correlating the catalytic performance to the extent of In₂O₃ reduction, it was found that a In_2O_3/m -ZrO₂ composite containing ~ 7 wt% In has the highest methanol rate at a reaction temperature 300 °C, which was explained by optimum coverage with an InO_x surface phase.

Besides ZrO₂, the oxides of Al and Cr are also frequently used to structurally stabilize catalysts. For instance, Cr is a typical promoter in Fe-oxide catalysts used for the high-temperature water-gas shift (WGS) reaction. The main aspects of high-temperature WGS catalysts were reviewed by Zhu and Wachs [16]. Ariëns et al. showed that doping Cr into Fe-oxides prepared by co-precipitation can decrease the particle size of the active phase (magnetite) and enhance the CO conversion [17]. It is important to note that hexavalent chromium (Cr^{6+}) is a strong carcinogen, posing a threat to both human health and the environment [18]. Exposure through inhalation and ingestion of contaminated water can lead to cancer and severe damage to human organs and skin. In contrast, trivalent chromium (Cr³⁺) exhibits very low toxicity and serves as a nutrient for the human body [18]. Cr promotion has also been reported in the context of CO₂ hydrogenation catalysis. Wang et al. reported that Cr increased the activity in a bifunctional InCrOx/SAPO-34 catalyst used in the hydrogenation of CO₂ to light olefins [19]. The incorporation of Cr resulted in more surface oxygen vacancies in InOx and stronger binding of formate intermediates, as supported by density functional theory calculations. The Cr-promoted InOx phase was prepared by a sol-gel method, which led to a heterogeneous dispersion of CrO_x species onto In₂O₃.

Herein, we prepared Cr-doped In₂O₃ particles in a single step using flame spray pyrolysis, which brings benefits in terms of control over the In_2O_3 particle size and homogeneous dispersion of the dopant [20,21]. Reference samples were prepared by impregnating Cr on a In₂O₃ sample prepared by FSP. The structure of the catalysts was characterized by Xray diffraction (XRD), transmission electron microscopy (TEM), Raman and X-ray adsorption spectroscopy (XAS) techniques. CO2-TPD (temperature-programmed desorption) and quasi-in situ X-ray photoelectron spectroscopy (XPS) were employed to understand the promoting role of Cr in increasing the number of oxygen vacancy (O_v) sites. Density functional theory (DFT) calculations were used to reveal the role of different Cr species in O_v formation in In₂O₃. The catalytic performance of these samples was evaluated in CO₂ hydrogenation. An optimized Crpromoted In₂O₃ catalyst was further promoted by Ni to render a highly active Ni-In₂O₃-Cr₂O₃ catalyst for CO₂ hydrogenation to methanol. In situ IR spectroscopy was used to investigate the reaction mechanism of methanol formation.

2. Experimental

2.1. Catalyst preparation

In₂O₃, CrO_x and In₂O₃-CrO_x catalysts with varying Cr molar contents (1 mol%, 2 mol%, 5 mol% and 10 mol% Cr) were synthesized by FSP using a Tethis NPS10 apparatus. The precursor solution was prepared by dissolving appropriate amounts of In(NO₃)₃·xH₂O (99.999%, Alfa Aesar) and Cr(NO₃)₃·9H₂O (99%, Sigma Aldrich) in an equivolumetric solvent mixture of ethanol (HPLC, Sigma Aldrich) and 2-ethylhexanoic acide (99%, Sigma Aldrich) at room temperature. The total metal

concentration (In + Cr) was 0.15 mol/L. This precursor solution was injected into the flame at a flow rate of 1.0 mL/min. The flame was sustained by a flow of 1.5 L/min methane and 3.0 L/min oxygen. The catalyst powders were collected on a glass-fiber filter (Hahnemühle FineArt GmbH, GF6, 257 mm in diameter) placed above the FSP nozzle. The resulting samples are denoted as Cr(x)-In₂O₃, where *x* stands for atomic percentage of Cr metal normalized to the total metal content. For comparsion, a reference sample was prepared by wet impregnation. In₂O₃ obtained by FSP was suspended in a solution of $Cr(NO_3)_3$ ·9H₂O in 50 mL H₂O, which was stirred overnight, followed by drying in a rotary evaporator, and calcination in air at 300 °C for 3 h. The sample is denoted by $Cr(2)/In_2O_3$.

Ni-In₂O₃-CrO_x mixed oxides were also prepared using the same FSP method, where Ni(NO₃)₂·6H₂O (99%, Sigma Aldrich) was added to the precursor solution. All other conditions for FSP synthesis were the same as described above. The resulting samples are denoted Ni(*x*)-Cr(*y*)-In₂O₃, where *x* and *y*, respectively, stand for atomic percentage of Ni and Cr metal when normalized to the total metal content.

2.2. Characterization

The textural properties of the as-prepared catalysts were analyzed by N₂ physisorption at -196 °C using a Micrometrics TriStar II 3020 instrument. Before the measurements, the samples were pretreated at 120 °C in a nitrogen flow overnight. The Brunauer-Emmett-Teller (BET) method was used to determine the specific surface area. Cr and Ni loading were analyzed using Inductively coupled plasma optical emission spectroscopy (ICP-OES) (Ametek SPECTROBLUE EOP). Prior to measurements, samples were dissolved in an equivolumetric mixture of nitric acid and water at room temperature. The particle size distribution of as-prepared and used catalysts was analyzed by Transmission electron microscopy (TEM) using a FEI Tecnai (type Sphera) instrument operating at an acceleration voltage of 200 kV. For sample preparation, appropriate amounts of samples were dispersed in ethanol under ultrasonic exposure and deposited on holey copper grids. XRD patterns were recorded using a Bruker D2 Phaser diffractometer with Cu Ka radiation (λ = 1.5406 Å) between 10 and 90° with a step size of 0.02° at a scan rate of 0.23 s/step.

Infrared (IR) spectra were obtained using a Nicolet FT-IR spectrometer equipped with a cryogenic MCT detector. Typically, samples were diluted with ZrO2 (prepared by flame spray pyrolysis with an injection rate of 1.0 mL/min, surface area of 204 m²/g, tetragonal and monoclinic mixed phases) in a 1:10 ratio by mass. An amount of ca. 15 mg of diluted sample was pressed into a pellet and placed in a homemade cell equipped with CaF2 windows for in situ measurements. The sample was pretreated in a He flow of 50 mL/min at 260 °C for 1 h. After cooling to 40 °C in He, a background spectrum was recorded. For in situ IR measurements at elevated pressure, the cell was pressurized to 10 bar in a 50 mL/min flow of 25 vol% CO_2 in H_2 . The cell was then heated to 260 °C, during which IR spectra were recorded. After reaching 260 °C and a dwell of 1 h, the gas mixture was switched from CO₂/H₂ mixture $(H_2/CO_2 = 3, 50 \text{ mL/min}, 10 \text{ bar})$ to H_2 (50 mL/min, 1 bar). During this replacement of the gas volume in the cell, IR spectra were recorded for 1 h.

Raman spectra were recorded using a WITec UHTS300 spectrometer equipped with a WITec WMT50 confocal Raman microscope. The spectra were collected using a 532 nm laser with the laser power of 5 mW. X-ray photoelectron spectroscopy (XPS) measurements were carried out using a K-Alpha XPS instrument (Thermo Scientific) equipped with a monochromatic X-ray source (Al K α = 1486.68 eV, 72 W, spot size of 400 µm) and a 180° double focusing hemispherical analyzer. Survey scans were measured at a constant pass energy of 200 eV and region scans at 50 eV. As-prepared samples were placed onto a doublesided carbon tape. Used samples (reaction in a CO₂ (20 vol%)/H₂(60 vol %)/N₂(20 vol%), total flow 50 mL/min, 260 °C, 30 bar, 14 h; cooling in an N₂ flow of 50 mL/min) were transferred to a glovebox and placed in a gas-tight holder for transfer into the XPS analysis chamber to avoid air exposure. Quasi-in situ XPS was conducted in a Kratos AXIS Ultra 600 spectrometer equipped with a monochromatic Al K α X-ray source. Region scans were recorded at a pass energy of 40 eV and survey scans at 160 eV. Typically, a powder sample was pressed into a pellet, which was placed in a Kratos high-temperature reaction cell attached to the XPS analysis chamber. Before XPS analysis, the sample was pretreated in an Ar flow (50 mL/min, 1 bar) for 1 h followed by cooling to 50 °C. The reaction was performed in a CO₂(20 vol%)/H₂(60 vol%)/Ar(20 vol%) gas flow of 50 mL/min at 1 bar at 260 °C or 300 °C for 2 h. The XPS spectra were analyzed using the CasaXPS software (version 3.2.23) and energy calibration was performed against the C 1 s peak of adventitious carbon at a binding energy of 284.8 eV.

XAS measurements at Cr and Ni K-edges were carried out at the CLÆSS beamline of the ALBA synchrotron (Barcelona, Spain). Energy selection was done with a Si(111) double-crystal monochromator. A mixture of the samples with boron nitride (BN) were pressed into thin pellets and sealed in Kapton tape in the home laboratory. Spectra were recorded at room temperature. The Cr K-edge XAS spectra were recorded in fluorescence mode using a 4-element silicon drift detector for the samples with a Cr loading of 2 mol%, while spectra of samples containing 9 mol% Cr were recorded in transmission mode. A Cr(NO₃)₃ sample was measured simultaneously in transmission mode for energy calibration. The Ni K-edge XAS spectra were measured in transmission mode with a Ni foil acting as a reference. Data analysis included energy calibration, normalization and fitting of the EXAFS part using the Demeter package (Athena/Artemis software). Scattering paths were calculated using the FEFF6 code based on crystal structures of Cr₂O₃, NiO, and Ni metal. A single Ni - In scattering path was included for EXAFS fitting. The amplitude reduction factor (S_0^2) was obtained by fitting the Cr(NO₃)₃ or Ni foil reference under the same conditions. The distance (ΔR), coordination number (CN), and Debye-Waller factor (σ^2) were fitted. The Fourier-transformed EXAFS spectra will be displayed without phase-correction.

Cr L-edge and O K-edge NEXAFS (Near-edge X-ray absorption fine structure) analysis was performed at the RIXS station of the SPECIES beamline at the MAX IV Laboratory (Lund, Sweden). A plane-grating monochromator illuminated with collimated light (cPGM) was used for energy selection. A retractable photodiode detector was placed in the main chamber to collect NEXAFS spectra in total fluorescence yield (TFY) mode. Before the measurements, the samples were diluted with boron nitride (BN) in a 1:5 ratio by mass, pressed into self-standing pellets and placed on an ultra-high-stable four-axis ultra-high-vacuum (UHV) manipulator using double-sided carbon tape (SPI supplies). Spectra were recorded at room temperature in UHV using an X-ray beam with vertical polarization.

Temperature-programmed reduction (H₂-TPR) measurements were performed in a Micromeritics AutoChem II setup. Typically, about 50 mg of sample was loaded into a quartz U-tube between two quartz wool layers. The sample was pretreated in a He flow (50 mL/min) at 120 $^\circ$ C for 1 h. TPR profiles were recorded by heating the sample from 40 to 800 °C at a rate of 10 °C/min in a 50 mL/min flow of 4 vol% H_2 in He. H_2 consumption was monitored by a thermal conductivity detector (TCD). Calibration of the TCD signal was done using a CuO reference sample. The CO₂ adsorption capacity of the as-prepared samples was determined by temperature-programmed desorption (CO₂-TPD) using a plug flow setup coupled with a mass spectrometer (Balzers TPG251). Before the CO₂-TPD analysis, the samples were pretreated in a reaction mixture flow (CO₂/H₂/N₂ = 10:30:10 mL/min, 30 bar) at 260 $^{\circ}$ C for 12 h, followed by a switch to a mixture of $\rm H_2$ and $\rm N_2$ (H_2/N_2 = 30:20 mL/min) at 260 °C. Subsequently, the samples were sealed with two three-way valves and transferred to a glovebox. They were then loaded into a quartz reactor without air exposure for CO₂-TPD experiments. CO₂ adsorption was carried out in pure CO2 (30 mL/min) at 50 °C for 2 h. Then, the reactor was purged with He for 1 h to remove the weakly adsorbed CO2. The TPD step involved heating in a He flow of 30 mL/min

from 50 to 600 °C at a 10 °C/min rate. The amount of desorbed CO₂ was quantified by integrating the area of the corresponding desorption profile. The amount of desorbed CO₂ was calibrated by decomposition of MgCO₃. In CO₂ + H₂-TPD experiments, samples were pretreated in He (50 mL/min) at 260 °C for 1 h. The feed gas was then switched to a mixture of CO₂ and H₂ (CO₂/H₂ = 12.5:37.5 mL/min, 1 bar) at 260 °C for 2 h. After cooling to 40 °C, the feedstock was switched back to He (30 mL/min). The TPD step followed the same procedure as described above.

2.3. Catalytic activity measurements

High-pressure CO₂ hydrogenation tests were carried out in a downflow stainless-steel reactor (ID = 4 mm). Typically, the catalyst was pressed, crushed and sieved into a 125–250 µm fraction, loaded into the reactor (100 mg) and pretreated in a He flow of 50 mL/min at 260 °C using a rate of 5 °C/min at 1 bar, followed by a dwell of 1 h. The catalyst was then exposed to the reaction mixture flow (CO₂/H₂/N₂ = 10:30:10 mL/min), followed by an increase of the pressure to 30 bar using a backpressure regulator. The reaction was initially tested at 260 °C and subsequently at 300 °C. To prevent condensation of reaction products, the stainless-steel tubes downstream of the reactor were heated to 130 °C. The product composition was analyzed by an online gas chromatograph (Interscience CompactGC) equipped with Rtx-1 (FID), Rt-QBond and Molsieve 5A (TCD), and Rt-QBond (TCD) columns. Response factors f(i)for each compound *i*, respective to the internal standard of nitrogen, in the GC analysis were determined by Eq. 1:

$$f(i) = \frac{A(i) \times F(N_2)_{in}}{A(N_2) \times F(i)_{in}}$$
(1)

where A(i) is the integrated area determined for the peak of compound *i*, and $F(i)_{in}$ is the corresponding known volumetric flow rate at the reactor inlet. The unknown effluent volumetric flow rate of product was determined using Eq. 2:

$$F(i)_{\text{out}} = \frac{A(i)}{A(N_2) \times f(i)} \times F(N_2)_{\text{out}}$$
⁽²⁾

The CO₂ conversion (*X*), product selectivity (*S*) and methanol formation rate (*STY*) were calculated using the following equations:

$$X(\text{CO}_2) = \frac{F(CO)_{\text{out}} + F(\text{CH}_3\text{OH})_{\text{out}} + F(\text{CH}_4)_{\text{out}}}{F(\text{CO}_2)_{\text{out}} + F(\text{CO})_{\text{out}} + F(\text{CH}_3\text{OH})_{\text{out}} + F(\text{CH}_4)_{\text{out}}}$$
(3)

$$S(\text{product}) = \frac{F(\text{product})_{\text{out}}}{F(CO)_{\text{out}} + F(CH_3OH)_{\text{out}} + F(CH_4)_{\text{out}}}$$
(4)

$$STY(CH_{3}OH) = \frac{F(CH_{3}OH)_{out} \times Mw(CH_{3}OH)}{V_{m} \times m_{cat}}$$
(5)

where *F* stands for the volumetric flow rate determined based on the N₂ internal standard using calibrated response factors, M_w (CH₃OH) is the molecular weight of methanol and V_m is the molar volume of ideal gas at standard conditions.

2.4. Density functional theory calculations

Density functional theory (DFT) calculations were conducted using the projector augmented wave (PAW) method [22] and the Perdew–Burke–Ernzerhof (PBE) [23] exchange–correlation functional as implemented in the Vienna Ab Initio Simulation Package (VASP) software [24,25]. The valence 5s and 5p states of In were treated explicitly as valence states within the scalar-relativistic PAW approach. To find solutions to the Kohn-Sham equations, we employed a plane-wave basis set with a cut-off energy of 400 eV. The Brillouin zone was sampled using a $3 \times 3 \times 1$ Monkhorst-Pack grid. All calculations were spinpolarized. Gaussian smearing ($\sigma=0.1~\text{eV})$ was employed.

The In₂O₃(111) surface termination was chosen as a support for Cr species, as it is more stable than the (110) and (100) surfaces, as shown in a previous computational study [26]. The stoichiometric $In_2O_3(111)$ surface consists of a three-dimensional slab with periodic boundary conditions. We introduced a 15.0 Å vacuum region in the direction perpendicular to the surface (*c*-direction) to avoid spurious interactions between neighboring cells. The upper two layers were allowed to move, whereas the bottom two were frozen. The supercell had dimensions of 14.57 Å \times 14.57 Å \times 26.01 Å and contained 96 O atoms and 64 In atoms, distributed in four atomic layers. We also verified that the electron density approached zero at the edges of the periodic super cell in the *c*direction. A structural model for a single atom (SA) of Cr (denoted by Cr- In_2O_3) was evaluated from a set of structures in which surface In atoms were replaced one at a time by Cr atoms. The most stable model was chosen by determining the substitution energies for each model, as defined earlier [12,13]. Several Cr-oxide clusters, each comprising 6 Cr atoms, supported on the $In_2O_3(111)$ surface were generated by hand, followed by geometry optimization. The electronic energies of gaseous H₂, H₂O, and CO₂ were calculated using an cubic unit cell (8 \times 8 \times 8 Å³) using a $1 \times 1 \times 1$ Monkhorst-Pack grid (Γ -point only). Gaussian smearing was employed. The electronic energies of the surface models were corrected for zero-point energies (ZPE) of adsorbates. The ZPE correction was also employed to the energies computed for the gasphase molecules.

The influence of oxygen vacancies on the reaction energetics in the CrO_x - In_2O_3 models was investigated by removing oxygen atoms from the $In_2O_3(111)$ lattice and the CrO_x clusters. We used H₂O as reference to calculate the energy required to remove a surface oxygen to form a vacancy according to Eq. (6):

$$\Delta E_{\rm O_v} = E_{\rm defective \ slab} - E_{\rm stoichiometric \ slab} + E_{\rm H_2O} - E_{\rm H_2} \tag{6}$$

where $E_{\text{defective slab}}$ is the electronic energy of the catalyst containing an oxygen vacancy, $E_{\text{stoichiometric slab}}$ is the reference energy of the stoichiometric slab, $E_{\text{H}_2\text{O}}$ and E_{H_2} are the electronic energies of gas-phase H₂O and H₂, respectively.

We determined stable and transition states for the different models relevant to the formation of O_v via removal of H₂O. Stable states were calculated using the conjugate-gradient algorithm, while the climbingimage nudged elastic band (CI-NEB) method was employed to find candidate transition states [27]. A frequency analysis was performed to confirm the occurrence of stable and transition states. It was verified that stable states have no imaginary frequencies and transition states have a single imaginary frequency in the direction of the reaction coordinate [28]. The mass-weighted Hessian matrix in this frequency analysis was constructed using a finite difference approach with a step size of 0.015 Å for displacement of individual atoms along each Cartesian coordinate. The corresponding normal mode vibrations were also used to calculate the ZPE correction and the vibrational partition functions.

3. Results and discussion

3.1. Characterization as-prepared In₂O₃-CrO_x

The XRD patterns of the as-prepared Cr-In₂O₃ and In₂O₃ samples feature the typical diffraction lines of cubic In₂O₃ (Fig. 1a). No XRD diffraction lines related to Cr-containing phases were observed, indicating a high Cr dispersion. There were no systematic trends in the very small shifts in the positions of the diffraction lines. Therefore, from these data we cannot conclude whether the isomorphous substitution of In by Cr in In₂O₃ took place. Impregnation of Cr on a In₂O₃ sample prepared by FSP (denoted as Cr(2)/In₂O₃) resulted in sharper diffraction lines of the In₂O₃ phase, when compared to the Cr(2)-In₂O₃ at the same Cr content. This means that some agglomeration of In₂O₃ occurred during the preparation steps of the Cr(2)/In₂O₃ sample. The In₂O₃ particle size was estimated by using the Scherrer equation (Table 1). This analysis revealed a slight decrease of the particle size in the Cr-doped samples as compared to In₂O₃. While In₂O₃ has on average 5.6 nm particles, the average particle size of Cr(10)-In₂O₃ is 4.6 nm. The morphology and

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P	hysicoc	hemical	properties	of	the as-prepared	l and	l used	ln_2	$O_{3}-0$	CrC) _x s	ampl	les
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Sample	d _{XRD} (nm) ^a	Specific surface area $(m^2 \cdot g^{-1})$	d _{XRD} (nm) ^a	Specific surface area $(m^2 \cdot g^{-1})$					
	As-prepar	red ^b	Used ^c						
In ₂ O ₃	5.6	150	12.5	67					
Cr(1)- In ₂ O ₃	5.0	162	9.9	54					
Cr(2)- In ₂ O ₃	5.0	140	11.0	60					
Cr(5)- In ₂ O ₃	5.2	135	9.3	80					
Cr(10)- In ₂ O ₃	4.6	119	6.3	95					
Cr(2)/ In ₂ O ₃	5.8	85	7.7	53					

^a Particle size based on XRD line broadening and Scherrer equation.

 $^{\rm b}\,$ As-prepared samples prepared by FSP.

 $^{\rm c}\,$ After CO_2 hydrogenation (300 $^{\circ}\text{C},$ 30 bar, CO_2/H_2/N_2 = 10/30/10 mL/min, 14 h).



Table

Fig. 1. (a) XRD patterns and (b) Raman spectra of the as-prepared In₂O₃-CrO_x samples.

particle size of In_2O_3 -CrO_x samples were investigated by TEM (Fig. S1). All the FSP-prepared Cr-In₂O₃ samples exhibit a similar morphology consisting of globular nanoparticles with a comparable particle size distribution centered at 4–5 nm. According to TEM, the average particle size of the impregnated Cr(2)/In₂O₃ sample was 5.2 nm, which is larger than the average size of the FSP-prepared Cr(2)-In₂O₃ sample. The specific surface area of the FSP-prepared samples ranges from 119 to 162 m²/g (Table 1). Samples with a higher Cr loading have a lower surface area, which can be related to the small interparticle voids being filled by Cr species. The surface area of the Cr(2)/In₂O₃ sample is only half of the surface area measured for Cr(2)-In₂O₃. This decrease can be mainly attributed to the presence of larger In₂O₃ particles.

Raman spectroscopy was used to study the In_2O_3 - CrO_x samples (Fig. 1b). The peaks at 132, 307 and 497 cm⁻¹ can be assigned to the In-O vibrations of InO_6 structural motifs.[14] The peak at 365 cm⁻¹ corresponds to an In-O-In stretching vibration [14,29,30]. The peak observed at 132 cm⁻¹ in the In_2O_3 sample broadens and shifts to lower wavenumbers with increasing Cr content. These changes point to a distortion in the local coordination environment of In^{3+} , caused by the incorporation of Cr^{3+} ions into the lattice of In_2O_3 [19]. In the samples with higher Cr content (≥ 5 mol%), a broad peak emerges at ~845 cm⁻¹, which corresponds to surface tetrahedral Cr-O species CrO_x [19,31,32], and indicates the presence of small CrO_x clusters on the In_2O_3 surface. Moreover, the peaks were sharper in the impregnated sample compared to Cr(2)- In_2O_3 , which implies a higher crystallinity of In_2O_3 . This is in line with the other characterization results of this sample.

The reducibility of the as-prepared In₂O₃-CrO_x samples was investigated by H₂-TPR (Fig. 2). The reduction feature at low temperature range can be assigned to the reduction of the surface of In₂O₃, which leads to the formation of O_v [7]. The high-temperature feature can be assigned to reduction of bulk In₂O₃ [31]. Reduction of Cr₂O₃ only occurs at temperatures above 1000 °C [33], while Cr⁶⁺ species reduce at lower temperature. Although the as-prepared samples contain small amounts of Cr⁶⁺ as will be shown by XPS below, the H₂-TPR profiles lack clear reduction peaks due to Cr⁶⁺ reduction.

Surface reduction of In₂O₃ occurred at a lower temperature in the samples with low Cr content (≤ 2 mol%), suggesting facile formation of O_v. The surface reduction feature shifts to a higher temperature again, as the Cr content exceeds 5 mol%. This is tentatively explained by the formation of small CrO_x clusters on the In₂O₃ surface at higher Cr content, as supported by the Raman results. Values of the H₂ consumption related to surface reduction normalized to the catalyst weight are summarized in Table S1. The effect of Cr on the surface reducibility follows from the H₂ consumption related to the low-temperature



reduction peak. The H₂ consumption increases from 0.40 mmol/g_{cat} for In2O3 to 0.80 mmol/gcat for Cr(2)-In2O3 and 0.98 mmol/gcat for Cr (10)- In_2O_3 . The value for the impregnated Cr(2)/ In_2O_3 sample of 0.31 mmol/g_{cat} is lower than that of Cr(2)-In₂O₃ prepared by FSP. Notably, the H₂ consumption for In₂O₃ with a particle size of \sim 5 nm is almost 10 times higher than the value of $0.042 \text{ mmol/g}_{cat}$ for In₂O₃ with a particle size of ~ 9 nm [11]. This points to a higher reducibility of smaller In₂O₃ crystallites. Such a strong influence of crystallite size has also been reported for CeO₂ crystallites with a size of 4–15 nm [34]. From the above, it can be stated that the inclusion of Cr in In₂O₃ using FSP enhances the reducibility of the In2O3 surface. This can be due to the destabilization of the In₂O₃ lattice by Cr dopants, but an effect of the crystallite size should also be considered. At higher Cr content, there can also be a stabilizing effect of CrO_x species on In₂O₃, explaining the smaller size of the In₂O₃ particles. As observed earlier for In₂O₃-Al₂O₃ samples prepared in a similar manner [35], the interface between In_2O_3 and a surface oxide phase can result in a broader temperature range of In₂O₃ reduction.

The composition and oxidation state of the surface of the as-prepared In₂O₃-CrO_x samples were investigated by XPS. Fig. 3 presents Cr 2p and O 1s XPS spectra of the In₂O₃-CrO_x samples. The Cr 2p_{3/2} peak at 576.6 eV corresponds to Cr^{3+} , while the Cr $2p_{3/2}$ peak at 579.1 eV is due to Cr^{6+} [36]. The Cr^{6+} contribution in the as-prepared samples increases from 28% for Cr(1)-In₂O₃ to 46% for Cr(10)-In₂O₃, which is likely due to the increasing contribution of surface CrO_x species (Table S2). The In 3d XPS spectra are dominated by the In 3d_{5/2} peak at 444.3 eV due to In₂O₃ (Fig. S2). Compared to In₂O₃, all Cr-containing samples display a minor shift of this feature to lower binding energies. Wang et al. attributed a similar shift for In₂O₃-CrO_x catalysts to the strong interactions between In and Cr in their oxide form [19]. The O 1s XPS spectra can provide information about the nature of surface oxygen species. The O 1s region of these samples can be deconvoluted into three contributions (Fig. 3b). The main peak at ~529.7 eV is assigned to lattice oxygen atoms, while the peak at \sim 531.9 eV is due to oxygen atoms of hydroxyl groups [7]. The assignment of the peak at \sim 531.1 eV is less clear, as it can be due to O atoms near O_v sites or O atoms of surface carbonates [37]. From the C 1s spectra, which can be deconvoluted into four components at 284.8, 287.5, 288.5 and 289.3 eV due to C-C, C=O, O-C=O and carbonate groups (Fig. S2) [38], it follows that the surface contains carbonates. Following a procedure from the literature [39], we estimated the amount of O_v by removing the carbonate contribution from the O 1 s spectra using the C 1 s feature at ~289.5 eV due to carbonate and assuming an O/C ratio of 3. The resulting fraction of O_v at the surface increases from 22% for In₂O₃ to a maximum value of 27% for Cr(2)-In₂O₃, followed by a nearly unchanged fraction for the samples with a higher Cr content (Table S3). The Cr(2)/In₂O₃ sample also contains slightly more O_v (24%) than In₂O₃, yet less than Cr(2)-In₂O₃. Notably, the fraction of O_v in In_2O_3 in this work is higher than the fraction of ~ 10 % typically reported in the literature, which may be attributed to differences in In₂O₃ particle size.

The Cr valence and electronic structure was investigated by NEXAFS for the as-prepared In₂O₃-CrO_x samples (Fig. S3a). The Cr L-edge positions in Cr(2)-In₂O₃ and Cr(10)-In₂O₃ samples are shifted to higher energies as compared to Cr₂O₃, indicating the presence Cr species with a higher oxidation state than +3 [40], which is in agreement with presence of Cr⁶⁺ according to XPS. Fig. S3b shows the normalized O K-edge spectra of the In₂O₃-CrO_x samples, which reflect the electronic transitions from the O 1s orbital to unoccupied states with O 2p character hybridized with In 3d [40-42]. The observation that the shape of the O K-edge feature of Cr(2)-In2O3 and Cr(10)-In2O3 is different from the one of Cr₂O₃ points to a different structure of Cr species, which is likely due to the high dispersion of Cr in contact with In₂O₃. In addition, the broad peak observed in the In₂O₃-CrO_x samples is split up due to the presence of t_{2g} and e_g orbitals. As shown before, the intensity ratio of the t_{2g} and e_g components can be correlated to the number of O_v [43]. The I_{eg}/I_{t2g} peak ratio for Cr(2)-In₂O₃ and Cr(10)-In₂O₃ are comparable, indicating a similar O_v density, which agrees with the XPS findings.



Fig. 3. (a) Cr 2p and (b) O 1s XPS spectra of the as-prepared In₂O₃-CrO_x samples.

Taken together, the characterization data above show that Cr species is present in the In_2O_3 -CrO_x samples as doped cations in the In_2O_3 lattice and as highly dispersed CrO_x on the In_2O_3 surface. The addition of Cr to In_2O_3 leads to a higher O_v concentration.

at a pressure of 30 bar and a H_2/CO_2 molar ratio of 3 (Fig. 4). The CO_2 conversion was kept below 10%. Fig. 4a shows that the CO_2 conversion at 260 °C increases with the Cr content up to a value of ca. 1.5% for Cr (2)-In₂O₃, indicating a promoting effect of Cr. With increasing Cr content, the CO₂ conversion remains nearly the same. The methanol selectivity for In₂O₃, Cr(1)-In₂O₃ and Cr(2)- In₂O₃ is nearly constant at ~60%, while a further increase of the Cr content leads to a slightly lower methanol selectivity. At a reaction temperature of 300 °C, the CO₂

3.2. Catalytic performance

The catalytic performance of the In_2O_3 -CrO_x samples was evaluated



Fig. 4. CO_2 conversion and product distribution at (a) 260 °C, (b) 300 °C, and (c) methanol space–time yield (STY) as a function of the Cr content on In_2O_3 (methanol STY of $Cr(2)/In_2O_3$ is shown by triangles); (d) Weight and surface-area-normalized methanol space–time yield (STY) for In_2O_3 , $Cr(2)-In_2O_3$, and $Cr(2)/In_2O_3$ catalysts at 300 °C. Reaction conditions: P = 30 bar, $H_2:CO_2 = 3$, and WHSV = 30000 mL h⁻¹ g⁻¹_{cat}.

conversion is highest at 6.5% for Cr(10)-In₂O₃, which is substantially higher than the value of 3.7% for In₂O₃. The trend in the methanol selectivity at 300 °C is similar to the trend observed at 260 °C (Fig. 4b). At both reaction temperatures, the CO₂ conversion of the Cr(2)/In₂O₃ was significantly lower than that of Cr(2)-In₂O₃, albeit that the methanol selectivity was comparable. The highest methanol yield of 0.24 g_{MeOH} g_{cat}⁻¹ h⁻¹ was obtained with the Cr(2)-In₂O₃ catalyst at 300 °C (Fig. 4c). Fig. 4d reports the methanol formation rates at 300 °C normalized by surface area for In₂O₃, Cr(2)-In₂O₃ and Cr(2)/In₂O₃. These findings show that Cr promotion is more effective when Cr is introduced during FSP synthesis.

Fig. 5 shows the stability of In_2O_3 and Cr(2)- In_2O_3 during CO_2 hydrogenation at 260 °C. While the In_2O_3 catalyst lost about one third of its initial activity after 60 h, the decrease for Cr(2)- In_2O_3 catalyst was only 12%. The methanol formation rate of Cr(2)- In_2O_3 and In_2O_3 decreased ~10% and ~25%, respectively, during the reaction. These results point to an improved stability due to Cr. Fig. S4 shows the performance of all catalysts at 300 °C for 10 h. It can be seen that, under these more demanding conditions, the activity of In_2O_3 decreases much stronger compared to the Cr-containing catalysts. The methanol rate is much more stable for the catalysts with a Cr content of 2% and higher. Thus, Cr acts as a promoter for methanol formation and also suppresses deactivation.

3.3. Characterization used In₂O₃-CrO_x

Fig. S5 shows the XRD patterns for the used In₂O₃-CrO_x catalysts. The absence of diffraction peaks due to Cr-oxides indicates that extensive sintering of Cr did not occur during CO₂ hydrogenation. The narrower In₂O₃ diffraction lines of the used samples compared to the as-prepared samples indicates sintering of the In₂O₃ particles. Whereas no significant shift was observed between Cr(10)-In₂O₃ and In₂O₃ in the as-prepared state, the sharper lines of the used samples allow resolving small shifts and the doping Cr^{3+} ions in the In₂O₃ lattice. The small shifts to higher angles can be explained by the slightly smaller ion radius of Cr^{3+} (0.62) Å) compared to the radius of In^{3+} (0.80 Å) [44]. While the particle size of used In₂O₃ of 12.5 nm is nearly twice the size of the as-prepared In₂O₃ particles (Table 1), we observe that the extent of sintering of the Crcontaining samples is less and decreases with increasing Cr content. However, the particle size of used Cr(2)-In₂O₃ is slightly larger than the particle sizes of Cr(1)-In₂O₃ and Cr(5)-In₂O₃, thus deviating from the general trend. This could be due to an upper limit in the amount doped Cr^{3+} that can be doped in the lattice of In_2O_3 . Doped Cr^{3+} distorts the lattice of In2O3 and facilitates the reduction of the surface In2O3, resulting in particle agglomeration. Cr(10)-In₂O₃ presents the smallest increase in particle size, which is consistent with the highest stability during CO2 hydrogenation at 300 °C. Also, the Cr(2)/In2O3 exhibits a better resistance against sintering compared to Cr(2)-In₂O₃, suggesting that the stabilization is due to surface Cr-oxides. This is also consistent

with the effect of Cr content on the stability, as characterization shows accumulation of surface CrO_x species on the In_2O_3 surface with increasing Cr content. The differences in the surface areas in Table 1 are consistent with the variations in In_2O_3 particle size. TEM analysis confirms that In_2O_3 particles are more stable, when they contain more Cr (Fig. S6). The Raman features of the used samples are qualitatively similar to those of the as-prepared samples (Fig. S7). The sharpening of the peak at 845 cm⁻¹ for used Cr(5)-In_2O_3 and Cr(10)-In_2O_3 indicates that some agglomeration of CrO_x on the surface of In_2O_3 occurred during CO₂ hydrogenation.

The Cr 2p and O 1s XPS spectra of the used In₂O₃-CrO_x catalysts are shown in Fig. 6. The Cr 2p XPS spectra are given in Fig. 6a. All the used $In_2O_3\mbox{-}CrO_x$ catalysts prepared by FSP contain a \mbox{Cr}^{6+} contribution of \sim 16%, which is lower than in the corresponding as-prepared samples (Table S2). This shows that some of the surface Cr^{6+} species were reduced Cr^{3+} during the reaction. The Cr^{6+} contribution of used Cr(2)/In₂O₃ is 22%, similar to the value in the as-prepared state. Fig. 6b displays the O 1s spectra of the used In₂O₃-CrO_x catalysts. Among the used In₂O₂-CrO_x catalysts, the O_y fraction in the Cr(10)-In₂O₃ catalyst reaches a maximum value of 14%. This finding is consistent with the highest H₂ consumption observed for Cr(10)-In₂O₃ by TPR, indicating that the addition of more Cr promotes oxygen vacancy formation during the reaction. The In 3d_{5/2} XPS spectra of the used In₂O₃-CrO_x catalysts are shown in Fig. S8. For all the used catalysts, the In $3d_{5/2}$ feature is consistently observed at 444.3 eV, signifying that In predominantly exists in the +3 oxidation state during the CO₂ hydrogenation reaction. Table S4 shows a slightly higher atomic Cr/In surface ratio for the used In₂O₃-CrO_x samples prepared by FSP when compared to the corresponding as-prepared ones. While in principle such differences can point to a higher Cr dispersion, it is more likely that some doped Cr atoms from the bulk migrated to the surface region during CO₂ hydrogenation. Instead, the Cr/In ratio for the Cr(2)/In2O3 sample did not change during the reaction, indicating the high structural stability of Cr species on the In₂O₃ surface.

3.4. Promoting role of Cr

Quasi-in situ XPS measurements were conducted to investigate changes in the electronic state of metal species and the O_{ν} fraction among In_2O_3 , Cr(2)-In_2O_3 and Cr(10)-In_2O_3 catalysts during CO_2 hydrogenation (CO_2/H_2/Ar = 10/30/10 mL/min, 1 bar). XPS spectra were recorded in the as-prepared state and after reaction at 260 and 300 °C for 2 h. The In 3d spectra did not change for these catalysts (Fig. S9), indicating that In remains in the +3 oxidation state. Fig. S10 shows that the Cr^{6+} contribution in the Cr-containing catalysts slightly decreased upon reaction at 260 °C. After reaction at 300 °C, the Cr^{6+} contribution decreased to 17% for Cr(2)-In_2O_3, which is higher than the value of 11% for Cr(10)-In_2O_3. This points to the higher stability of Cr^{6+} species for Cr (2)-In_2O_3.



Fig. 5. Time on stream (TOS) behavior of (a) CO_2 conversion, (b) methanol selectivity and (c) methanol STY over In_2O_3 and Cr(2)- In_2O_3 catalysts. Reaction conditions: T = 260 °C, P = 30 bar, H_2 : $CO_2 = 3$, and WHSV = 30000 mL h⁻¹ g⁻¹_{cat}.



Fig. 6. (a) Cr 2p and (b) O 1s XP spectra of the used In_2O_3 -CrO_x catalysts. Reaction conditions: T = 260 °C, P = 30 bar, H₂:CO₂ = 3, and WHSV = 30000 mL h⁻¹ g⁻¹_{cl}.

We also studied the surface O_v fraction in these samples using O 1s XPS spectra (Fig. 7). As done above, we took into account the contributions of carbonate species in the O 1s spectra, utilizing the C 1s spectra (Fig. S11). For In₂O₃, it is observed that the O_v fraction was lower after reaction at 300 °C than after reaction at 260 °C. This is consistent with the fast deactivation of In₂O₃ at 300 °C, as supported by the catalytic results. Compared to used In₂O₃, the O_v fractions in the used Cr(2)-In₂O₃ and Cr(10)-In₂O₃ samples are higher and they are not affected by reaction at the higher temperature of 300 °C. Overall, these data suggest that Cr stabilizes the In₂O₃ particles against sintering. The smaller In₂O₃ particles cause the higher reducibility of surface In₂O₃, thereby enhancing the O_v concentration on In₂O₃.

CO₂ adsorption on the In₂O₃-CrO_x catalysts was investigated using CO₂-TPD. Prior to CO₂-TPD, the samples were evaluated in the CO₂ hydrogenation reaction (260 °C, 30 bar, CO₂/H₂/N₂ = 10/30/10 mL/min) for 14 h, followed by transfer without air exposure to the TPD setup

and CO₂ adsorption at 50 °C. Fig. 8 displays the CO₂-TPD profiles. The peak observed around 130 °C corresponds to weakly adsorbed CO₂. In₂O₃ has few hydroxyl (OH) groups on its surface. CO₂ can weakly bind to these OH groups on the In₂O₃ surface [45]. While the peak ranging from 200 to 450 °C is attributed to CO₂ adsorbed on O_v sites [46]. With increasing Cr content above 2 mol%, this peak broadens and shifts to higher temperatures. Such changes are also seen for the Cr(2)/In₂O₃ sample. A CO₂-TPD measurement for the Cr₂O₃ reference samples showed a CO₂ desorption peak at a much higher temperature of 385 °C. XPS confirms that most Cr species are in the +3 state in the used In₂O₃-CrO_x samples. Accordingly, we speculate that the shift in the CO₂ desorption peak is caused in part by the presence of Cr₂O₃ domains on the In₂O₃ surface.

The amount of adsorbed CO₂ quantified by integrating the desorption peak in the 200 to 450 $^\circ C$ range increases from low values of 16.1 and 36.5 $\mu mol/g_{cat}$ for In₂O₃ and Cr(1)-In₂O₃ to the highest value of



Fig. 7. O 1s XP spectra for (a) In₂O₃, (b) Cr(2)-In₂O₃ and (b) Cr(10)-In₂O₃ catalysts after treatments in various conditions.



Fig. 8. (a) CO_2 -TPD profiles of In_2O_3 -CrO_x catalysts with different Cr content and Cr_2O_3 . (b) The relation between methanol space–time yield (STY) over the In_2O_3 -CrO_x catalysts during CO_2 hydrogenation at 260 °C (Fig. 4c) and the amount of CO_2 desorbed, which was obtained by quantifying the TPD desorption peak in the range of 200 to 450 °C (Table S1).

284.9 µmol/g_{cat} for Cr(10)-In₂O₃ (Table S1). The value of 7.0 µmol/g_{cat} for $Cr(2)/In_2O_3$ is much lower than that of 187.9 μ mol/g_{cat} for Cr(2)-In₂O₃. Although a portion of CO₂ absorbed on Cr₂O₃ domains, the significantly increased CO2 adsorption with increasing Cr content aligns with the trend in O_v fraction, suggesting enhanced formation of O_v at high Cr content. Fig. 8b shows the correlation between the amount of adsorbed CO_2 and the methanol rates. At low Cr content ($\leq 2 \mod \%$), the methanol rate correlates well with the CO₂ adsorbed, suggesting that the introduction of small amounts of Cr to In₂O₃ can generate more O_v and enhance CO₂ adsorption. However, as the Cr content exceeds 2 mol%, the CO₂ adsorption increases and the methanol rate decreases. This can be explained by the increasing contribution of CO2 adsorbed on Croxides at the surface, which is likely inactive in CO₂ hydrogenation to methanol. CrO_v prepared by FSP did not exhibit any CO₂ hydrogenation activity (260 °C, 30 bar, $CO_2/H_2/N_2 = 10/30/10$ mL/min). Next to CO_2 desorption, we also observed the formation of CO at the same temperature and the CO amount was seen to increase with the Cr content (Fig. 8 and Table S1). Although bulk CrO_x is not catalytically active, we still observed CO formation in In₂O₃-CrO_x samples below 300 °C during the CO2-TPD experiments, and a higher Cr content led to more CO. This is likely related to the presence of much better dispersed CrO_x domains on the In₂O₃ surface, contributing to CO formation during both CO₂-TPD and CO_2 hydrogenation. According to the XPS, Cr^{3+} is the dominant oxidation state of Cr in the used In2O3-CrOx catalysts. Therefore, we speculate that the decrease in methanol rate for samples with a high Cr content is due to the higher coverage of the In₂O₃ surface by Cr-oxide species.

Characterization revealed that Cr species are present as doped Cr atoms in the In₂O₃ lattice and as dispersed CrO_x on the In₂O₃ surface. XPS confirmed that the majority of surface Cr⁶⁺ species is reduced to Cr³⁺ during CO₂ hydrogenation. We employed DFT calculations to understand the role of doped and surface Cr species in O_v formation and CO2 adsorption. The three different surface models considered are shown Fig. 9a, namely (i) the stoichiometric $In_2O_3(111)$ surface, (ii) a Cr atom doped in the first In layer in the In₂O₃(111) surface, and (iii) a Cr_6O_9 cluster placed on the $In_2O_3(111)$ surface. We first calculated the oxygen vacancy formation energies (ΔE_{Ov}) using Eq. (6) from section 2.4. The results are given in Fig. S12 and Table S5. The In₂O₃(111) surface contains 12 different surface O atoms, for which the oxygen vacancy formation energies (Fig. S12a and Table S5) range between -62and +8 kJ/mol, as reported before [12,13]. Doping a Cr atom in the In₂O₃ surface makes oxygen removal slightly more difficult, with ΔE_{Ov} ranging between -38 and +66 kJ/mol (Fig. S12b and Table S5). The O atoms with the lowest ΔE_{OV} bind to the Cr atom (oxygens 1–3 in Fig. S12b, $\Delta E_{OV} = -38$ kJ/mol). The Cr-O bond distance of 1.74 Å is much shorter than the original In-O bond distance at the site of doping (2.19 Å). The lattice contraction due to Cr doping brings the O atoms around the Cr dopant closer to each other than in the stoichiometric In₂O₃ case, enhancing electron-electron repulsion. This weakens the Cr-O bonds, explaining the destabilization of the O atoms around Cr. This line of reasoning is similar to the one advanced for oxygen vacancy formation for Ni-doped In₂O₃(111) [11,12]. Overall, the ΔE_{OV} for the Cr-doped case is slightly less favorable than the most favorable ΔE_{OV} for the stoichiometric In₂O₃ surface ($\Delta E_{Ov,Cr-doped} = -38$ kJ/mol and $\Delta E_{Ov,}$



Fig. 9. (a) DFT surface models and (b) potential energy diagrams for oxygen vacancy formation via H_2O formation. The oxygens that have been removed to create a vacancy are highlighted with dashed circles (O atoms in red, In atoms in brown, and Cr atoms in blue). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

 $_{In2O3} = -62$ kJ/mol). Nevertheless, doping Cr ions into the In_2O_3 can increase the number of oxygen vacancies, as it creates more surface sites with a favorable ΔE_{Ov} .

For the Cr₆O₉-In₂O₃ model, we considered oxygen vacancies belonging to the In₂O₃ surface and the Cr₆O₉ cluster (Fig. S12c). On In₂O₃, the ΔE_{Ov} values range between -10 and +117 kJ/mol, while higher values are found for the Cr₆O₉ cluster itself, with ΔE_{Ov} values ranging between 104 and 124 kJ/mol (Table S5). The unfavorable oxygen vacancy formation energies for the O atoms on the Cr₆O₉ cluster compared to In₂O₃ is in line with the TPR results, pointing to higher reduction temperatures for the CrO_x overlayer. Moreover, it is in keeping with the higher stability of Cr₂O₃ (Δ H_f, _{298 K}) = -1131.2 kJ/mol) compared to In₂O₃ (Δ H_f, _{298 K} = -922.9 kJ/mol) [47,48].

CO2 hydrogenation conditions favor a surface configuration, in which one oxygen vacancy per unit cell is exposed, as also employed as a model in previous computational reports [24]. Next, we studied the pathways of oxygen vacancy formation on the models in Fig. 9a. The potential energy diagrams for the removal of the oxygen atom with the lowest ΔE_{OV} for each surface (O_v in Fig. 9a) are shown in Fig. 9b (data in Table S6, structures of initial, transition and finale states in Tables S7-S9). For the stoichiometric In₂O₃ surface and the Cr-doped model, homolytic dissociation of H₂ results in two OH groups (Tables S7-S8). On the In₂O₃ surface, H₂ dissociation has an activation energy of 98 kJ/mol and the reaction energy is $\Delta E_r = -197$ kJ/mol. These values agree well with an earlier DFT study of H₂ dissociation on In₂O₃ [49]. Doping of a Cr atom in the In₂O₃ surface renders H₂ activation less favorable with activation and reaction energies of 137 and -140 kJ/mol, respectively. For the Cr₆O₉ cluster, H₂ activation (Fig. 9b and Table S9) has a relatively low activation energy of 20 kJ/mol ($\Delta E_r = -60$ kJ/mol). The transition state involves Cr-H and O-H moiety, while the final state is characterized by one Cr-OH and an OH moiety on In₂O₃. The next step involves OH hydrogenation to H2O via proton migration. On In₂O₃(111), this step has an activation energy of 97 kJ/mol and is endothermic by 75 kJ/mol. On Cr-doped In₂O₃, proton migration has a slightly higher activation energy of 119 kJ/mol ($\Delta E_r = 49$ kJ/mol). Proton migration from the Cr cluster to In₂O₃ results in facile formation of H₂O on the In₂O₃. This step has an activation energy of 51 kJ/mol and is slightly exothermic ($\Delta E_r = -9 \text{ kJ/mol}$). Desorption of H₂O from In₂O₃, Cr-doped In2O3 and Cr6O9 on In2O3 cost 10, 10 and 63 kJ/mol, respectively. The overall barriers (in brackets reaction energies) with respect to gas-phase H₂ are 98 (-62 kJ/mol), 137 (-38 kJ/mol), 20 (-10 kJ/mol), for In₂O₃, Cr₁-dopded and Cr₆O₉. Thus, Cr doping leads to a slightly higher barrier for O removal as H₂O. This will not suppress the role of such O_v in methanol formation, as the overall barrier for O_v formation is still low compared to the overall barriers for methanol formation on In₂O₃ [21,22]. These theoretical investigations show that Cr can result in more O_v sites by replacing In³⁺ in the In₂O₃ (Cr doping) and through modifying the electronic structure of In-O-In moieties close to Cr-oxide clusters formed on the surface.

We next explored different adsorption configurations of CO₂ on the CrO_x-In₂O₃ models after O removal. The optimized structures are shown in Fig. S13. For the Cr_1 -doped model, CO_2 absorption led to a surface carbonate species with two stable adsorption configurations being identified. The most stable configuration ($\Delta E_{ads} = -74$ kJ/mol, Fig. S13a) features CO₂ adsorbed with the carbon atom close to the Cr single atom. This adsorption configuration is similar to the one observed for the In₂O₃(110) surface, which involved slightly weaker adsorption energy ($\Delta E_{ads} = -41$ kJ/mol) [8]. The small difference in adsorption energies between In₂O₃ and Cr₁-In₂O₃ may be due to the fact that we included van der Waals corrections in our DFT calculations unlike Ye et al [8]. In the less stable position ($\Delta E_{ads} = -36 \text{ kJ/mol}$, Fig. S13b), CO₂ is further away from the Cr atom. On the Cr₆O₉ model, CO₂ adsorbs preferentially at the interface between the Cr₆O₉ cluster and In₂O₃ with one O atom filling the oxygen vacancy in the In₂O₃ surface ($\Delta E_{ads} =$ -73 kJ/mol, Fig. S13c). Adsorption on the Cr-oxide cluster with one oxygen vacancy (removing O13 from the model shown in Fig. S12) was

also explored, but lead to a weaker binding compared to adsorption on In₂O₃ ($\Delta E_{ads} = -10$ kJ/mol, Fig. S13d). In an earlier study of a Ni atom doped inside In₂O₃, the hydrogenation of carbonate to formate (HCOO*) involved a high activation barrier [12], excluding its involvement in methanol synthesis. The presence of Cr-H species upon H₂ dissociation on the Cr₆O₉ cluster can lower the barrier for the protonation of the carbon atom in adsorbed CO₂, resulting in the facile formation of HCOO*. Accordingly, we conducted CO₂ + H₂-TPD experiments on In₂O₃, Cr(2)-In₂O₃, and Cr(5)-In₂O₃ catalysts to compare their ability to activate H₂ (Fig. S14). After pretreatment in the reaction mixture (CO₂/H₂ = 12.5/37.5 mL/min, 1 bar) for 2 h, a TPD was carried out from 40 to 600 °C. The H₂ desorption peak was seen to increase with increasing Cr loading, supporting the conclusion of H₂ activation being facilitated by Cr-oxides on the In₂O₃ surface.

3.5. Ni-Cr-In₂O₃

Although we found that the presence of Cr as Cr-oxide species can facilitate O_v formation through facile H₂ activation, typically transition metal promoters like Pt, Pd and Ni are preferred to promote H2 dissociation on In₂O₃ catalysts for CO₂ hydrogenation. For this purpose, Ni-In₂O₃-CrO_x samples were synthesized by FSP at various Ni and Cr contents (Table S10). The catalytic performance among In₂O₃, Cr(10)-In2O3, Ni(12)-In2O3 and Ni(12)-Cr(9)-In2O3 catalysts is shown in Fig. 10. Notably, Cr(10)-In₂O₃ and Ni(12)-Cr(9)-In₂O₃ catalysts were prepared at the same Cr/In atomic ratio of 1/9. Fig. 10a shows that Ni (12)-Cr(9)-In₂O₃ exhibits the highest CO₂ conversion of 5.1% at 260 °C, significantly surpassing the values of 1.0% for pure In₂O₃, 1.5% for Cr (10)-In₂O₃ and 4.3% for Ni(12)-In₂O₃. However, the methanol selectivity in Ni(12)-Cr(9)-In₂O₃ (~40%) is similar to that Ni(12)-In₂O₃, lower than those of In₂O₃ and Cr(10)-In₂O₃. Fig. 10b shows that the methanol rate of Ni(12)-In2O3 decreases substantially during the first 30 min on stream. In contrast, the Ni(12)-Cr(9)-In₂O₃ catalysts hardly deactivates, indicating that the stabilizing effect of Cr observed above is retained in the presence of Ni. The highest methanol rate is observed in Ni(12)-Cr(9)-In₂O₃. These findings suggest that Ni and Cr can synergistically improve CO₂ hydrogenation synthesis with In₂O₃ catalysts.

Fig. S15 shows the catalytic performance of the Ni-In₂O₃-CrO_x samples with different Ni and Cr contents in CO2 hydrogenation. The simultaneous presence of Cr and Ni increases the CO2 conversion. For instance, at a Ni content of 12 mol%, the addition of 2 mol% Cr increases the CO₂ conversion at 260 $^{\circ}$ C from 4.3% to 5.3%. At a Ni content of 22 mol%, the CO₂ conversion of Ni(22)-Cr(8)-In₂O₃ is nearly two times higher than that of Ni(22)-In₂O₃. The methanol selectivity does not substantially change for these catalysts. The highest CO₂ conversion at 300 °C is 15% for Ni(22)-Cr(8)-In₂O₃, which can be compared to a CO₂ conversion of 10% for Ni(22)-In₂O₃. The CO₂ conversion in Ni(22)-Cr (8)-In₂O₃ is lower than the CO₂ equilibrium conversion of \sim 30% under the given conditions, implying that increasing the active sites by adjusting the Ni or Cr content can further enhance the catalyst performance. The methanol selectivity in all the catalysts is comparable and below 20% at 300 °C. In Fig. S15c, the Ni(12)-Cr(9)-In₂O₃ catalyst has a 25% higher methanol rate than Ni(12)-In2O3 after stabilization at 260 °C. The Ni(22)-Cr(8)-In₂O₃ catalyst exhibits the highest methanol rate of 0.24 $g_{MeOH}\,g_{cat}^{-1}\,h^{-1},$ which is approximately twice that of Ni(22)-In₂O₃. At a temperature of 300 °C, Ni(12)-In₂O₃, Ni(12)-Cr(x)-In₂O₃ and Ni(22)-Cr(8)-In₂O₃ catalysts exhibit similar methanol rates with Ni(22)-In₂O₃ having a significantly lower methanol rate. These data indicate the optimal Ni and Cr contents for promoting methanol synthesis in Ni-In2O3-CrOx samples. We also compared the Ni-In-Cr catalysts to other Cu-based and Ni-promoted catalysts. As shown in Table S11, Ni-In₂O₃ prepared by FSP shows a slightly lower methanol rate than a Ni-In₂O₃ sample described in the literature [45], which could be due to the lower pressure and H₂/CO₂ ratio. The methanol rate of Ni-In₂O₃-CrO_x is similar to that of Cu-based catalysts, such as Cu-ZnZr and Ce-CuZn samples [50,51].



Fig. 10. (a) CO₂ conversion and product selectivity at 260 °C and (b) methanol space–time yield (STY) along with the time on stream over In₂O₃, Cr(10)-In₂O₃, Ni (12)-In₂O₃ and Ni(12)-Cr(9)-In₂O₃ catalysts. Reaction conditions: P = 30 bar, H_2 :CO₂ = 3:1, and WHSV = 30000 mL h⁻¹ g⁻¹_{cat}.

XRD analysis was employed to determine the phases present in the as-prepared Ni-In₂O₃-CrO_x samples (Fig. 11a). All diffraction lines of Ni (12)-In₂O₃ can be attributed to cubic In₂O₃, with peak positions shifting slightly to higher angles compared to pure In₂O₃. This shift suggests the incorporation of Ni into the In₂O₃ lattice in line with the earlier work of Zhu et al [11]. However, as the Ni content is increased to 22 mol%, the diffraction lines become broader compared to those in Ni(12)-In₂O₃, pointing to smaller In₂O₃ particles. No diffraction peaks indicative of Croxides were observed for the Cr-containing samples. The average particle size for the as-prepared Ni(12)-In₂O₃ is ~3.3 nm (Table S12). The estimation of the average particle size based on the Scherrer equation for the as-prepared Ni-In₂O₃-CrO_x and Ni(22)-In₂O₃ samples cannot be realized due to the broad diffraction lines, suggesting a smaller particle size than 3.3 nm.

The XRD patterns of the used Ni-In₂O₃-CrO_x catalysts are shown in Fig. S16. The used catalysts only display diffraction peaks of In₂O₃, showing that Ni and Cr remain highly dispersed. Compared to the asprepared catalysts, the diffraction lines due to In₂O₃ are sharper, pointing to sintering of the In₂O₃ particles during the reaction. The average size of the In₂O₃ particles in the used Ni(12)-In₂O₃ is 19.3 nm (Table S11). The In₂O₃ catalysts are similar at ~17 nm. At a Ni content of 22 mol%, the In₂O₃ particle size in the used Ni(22)-In₂O₃ catalyst is 51 nm, whereas the particle size in Ni(22)-Cr(8)-In₂O₃ is smaller, i.e. 38 nm. These findings show that the stabilizing effect of Cr on In₂O₃ is retained in the presence of Ni.

H₂-TPR was used to study the reduction behavior in the Ni-In₂O₃-CrO_x samples (Fig. 11b). Ni-In₂O₃ and Ni-In₂O₃-CrO_x samples exhibit a similar peak at \sim 240 °C, corresponding to reduction of the In₂O₃ surface

[7]. The H_2 consumption corresponding to this surface reduction amounts to 0.33 mmol/g_{cat} for Ni(12)-In₂O₃, while it is higher at 0.49 mmol/g_{cat} for Ni(12)-Cr(2)-In₂O₃ and 0.41 mmol/g_{cat} for Ni(12)-Cr(9)-In₂O₃ (Table S13). Ni(22)-In₂O₃ exhibits a lower H₂ consumption compared to Ni(12)-In₂O₃. This may be attributed to NiO clusters covering part of the In₂O₃ surface, hindering the reduction of In₂O₃ [52]. The addition of 8 mol% Cr to Ni(22)-In₂O₃ results in a significantly enhanced H₂ consumption of 0.65 mmol/g_{cat}. These results indicate that the introduction of Cr to Ni-In2O3 facilitates the reduction of surface $In_2O_3\text{,}$ similar to Cr promotion of $In_2O_3\text{.}$ The reduction feature at ~280 °C for Ni(12)-In₂O₃ is due to the NiO reduction to Ni metal [11]. This feature is located at higher temperature for the Ni(12)-Cr(9)-In₂O₃ and Ni(22)-Cr(8)-In₂O₃ samples, suggesting that Ni-oxide might preferentially interact with Cr-oxide at the surface delaying Ni reduction. The H_2/Ni ratio associated with this feature is 1.70 for Ni(12)-In₂O₃, pointing to the further reduction of In₂O₃ to In due to H₂ spillover from metallic Ni. The H₂/Ni ratio increases to 1.85 and 2.25 for Ni(12)-Cr(2)- In_2O_3 and Ni(12)-Cr(9)-In₂O₃, respectively, which may be due to reduction of Cr⁶⁺ species present in the as-prepared samples.

The normalized XANES spectra at the Cr K-edge are shown in Fig. 12a. The pre-edge feature located at 5993.7 eV due to the 1s to 3d transition for the CrO_x sample points to the presence of Cr^{6+} [53], which is further supported by the higher edge energy compared to the Cr_2O_3 reference. It is also in line with the high Cr^{6+} fraction of 83% in CrO_x according to XPS (Fig. S17). The as-prepared Ni(12)-Cr(2)-In₂O₃ and Ni (12)-Cr(9)-In₂O₃ samples also contain a pre-edge feature due to Cr^{6+} , while the broad feature ranging from 6010 to 6030 eV points to the presence of Cr(III) hydrate [54]. The XANES of the used Ni(12)-Cr(2)-In₂O₃ and Ni(12)-Cr(9)-In₂O₃ samples are similar to the XANES



Fig. 11. (a) XRD patterns and (b) TPR profiles of the as-prepared Ni-In₂O₃-CrO_x samples.



Fig. 12. Normalized XANES spectra at the (a) Cr K-edge and (b) Ni K-edge of the as-prepared and used Ni(12)-In₂O₃-CrO_x samples.

spectrum of Cr_2O_3 , while the features corresponding to Cr(III) hydrate and Cr^{6+} disappeared. EXAFS spectra were obtained at Cr K-edge for asprepared and used Ni(12)-Cr(2)-In₂O₃ and Ni(12)-Cr(9)-In₂O₃ samples (Fig. S18) and the fit results are collected in Table S14. The first and second shell at uncorrected distances of 1.6 and 2.5 Å correspond to the Cr-O and Cr-O-Cr contributions, respectively, in Cr₂O₃ reference. In the as-prepared Ni(12)-Cr(2)-In₂O₃ and Ni(12)-Cr(9)-In₂O₃ samples, only Cr-O contribution is observed, suggesting that Cr atoms are well dispersed in In₂O₃. After the reaction, the Cr-O coordination numbers decrease, while a Cr-O-Cr shell appears, indicating the clustering of Cr species on the In₂O₃ surface during the reaction. Notably, the Cr samples after the reaction show a Cr-O-Cr shell with a coordination number (CN) of ~1, which is much lower than the CNs of the Cr-O-Cr shell (~4) in bulk Cr₂O₃, pointing to highly dispersed Cr clusters on the In₂O₃ surface.

The Ni K-edge XANES of the Ni-containing samples are shown in Fig. 12b. The edge energy at 8344 eV is similar to that of the NiO reference. The edge energy of the used Ni(12)-In₂O₃ sample is \sim 8340 eV, indicating the reduction of NiO to metallic Ni. Nevertheless, some NiO is likely present as the edge energy of the used Ni(12)-In₂O₃ is slightly higher than the edge energy of the Ni foil. The used Ni(12)-Cr (2)-In₂O₃ and Ni(12)-Cr(9)-In₂O₃ exhibit higher edge energies than to Ni (12)-In₂O₃, indicating that the Cr-containing samples contain a larger amount of NiO. The EXAFS results show a decrease of the Ni-O and Ni-O-Ni shells after the reaction (Fig. S19, Table S15). A shell at 2.5 Å in the Ni (12)-In₂O₃ catalyst can be related to metallic Ni-In bonds [55]. In the used Ni(12)-Cr(2)-In2O3 and Ni(12)-Cr(9)-In2O3 catalysts, a Ni-Ni (metallic) shell at ~2.2 Å was observed, while the Ni-In shell was absent. This observation suggests that the introduction of Cr to Ni-In₂O₃ weakens the interaction between Ni and In in the reduced Ni species. In addition, EXAFS reveals CNs of ~ 2 for the metallic Ni-Ni shell in Ni- In_2O_3 -CrO_x and ~2 for the metallic Ni-In shell in Ni-In₂O₃. These CNs are much lower than those expected for nanoparticles (CN > 10), suggesting a high dispersion of the Ni phase in the former samples.

We next investigated reaction intermediates of CO₂ hydrogenation by in situ IR spectroscopy for the Cr(10)-In₂O₃, Ni(12)-In₂O₃ and Ni(12)-Cr(9)-In₂O₃ samples (Fig. 13a-13c). Exposing the samples to a model reaction mixture (H₂/CO₂ = 3, 10 bar) at 40 °C resulted in the formation of monodentate carbonate (m-CO₃², 1422 and 1335 cm⁻¹), bicarbonate (HCO₃, 1613 and 1227 cm⁻¹) and bidentate carbonates (b-CO₃², 1633 and 1528 cm⁻¹) [56–58]. As the temperature increases, the bands due to bicarbonates disappear and the formation of polycarbonates (1484 cm⁻¹), formates (1367, 1380, 1571–1592, 2872, and 2973 cm⁻¹) and methoxy (1045, 1142, 2827, and 2931 cm⁻¹) is apparent [57]. The hydrogenation behavior of these intermediates was evaluated by switching the feed from CO₂ + H₂ (10 bar, 50 mL/min) to pure H₂ (1 bar, 50 mL/min) at 260 °C (Fig. 13a-13c). This led to stronger CH₃O* bands and weaker HCOO* bands, suggesting that the reaction follows the formate pathway, where HCOO* is considered as the key intermediate for methoxy and methanol formation, which is in line with suggestions from DFT calculations [12]. Overall, the evolution of the IR bands was qualitatively comparable for the three investigated samples.

The formate IR bands of the three samples were compared after 1 h CO₂ hydrogenation at 260 °C (Fig. 13d). The v(CH) band of HCOO* in all samples is located at the same position of 2872 cm⁻¹. In Cr(10)-In₂O₃, two distinct features at 1571 and 1592 cm⁻¹ in the C-O region can attributed to the O-C-O ($v_{as}(OCO)$) vibration of HCOO*. The $v_{as}(OCO)$ band at 1571 cm⁻¹ in HCOO* has been reported to represent formate on In₂O₃ [59]. The ν_{as} (OCO) band at 1592 cm⁻¹ in Cr(10)-In₂O₃ is similar to the band observed in In2O3-Al2O3 samples in our previous study of Alpromoted In₂O₃ [35]. By analogy, we assign this to formate adsorbed on an oxygen vacancy close to highly dispersed Cr. In Ni(12)-In₂O₃, we also found two $v_{as}(OCO)$ bands of HCOO* at 1591 and 1571 cm⁻¹. The feature in the formate region at 1591 cm^{-1} can be assigned to formate on Ni sites, as this band was also observed for Ni/SiO₂ during CO₂ hydrogenation [60]. Based on the EXAFS analysis of used Ni-In₂O₃, it can be said that Ni strongly interacts with In₂O₃ during CO₂ hydrogenation. This strengthens the assignment of the 1591 cm⁻¹ band to HCOO* on Ni sites close to In₂O₃. Such a formate intermediate may also involve coordination to oxygen vacancies in In₂O₃ at the Ni-In₂O₃ interface. Additionally, the formate band in Ni(12)-Cr(9)-In₂O₃ occurs at a lower wavenumber of 1573 cm⁻¹, which is comparable to the location of HCOO* on In₂O₃ and different from Ni(12)-In₂O₃. This observation implies a larger contribution of adsorbed formate on In2O3 in Ni(12)-Cr (9)-In₂O₃ than in Ni(12)-In₂O₃.

The amount of HCOO* and CH₃O* species under CO₂ hydrogenation conditions (260 °C, 10 bar, 60 min) at a semi-steady state in the IR experiment was estimated by integrating the bands at 2872 and 2931 cm^{-1} , respectively (Fig. 13d and 13e). In our prior IR study on In_2O_3 carried out under identical conditions, the peak areas for formate and methoxy were ~5 and ~3, respectively.[35] Ni(12)-Cr(9)-In₂O₃ displays significantly higher formate and methoxy areas of 10.4 and 7.2, respectively, surpassing values of 8.5 and 5.9 for Cr(10)-In₂O₃, 9.4 and 6.7 for Ni(12)-In₂O₃ and those for In₂O₃. These differences shows that the Ni(12)-Cr(9)-In₂O₃ surface contains more HCOO* and CH₃O* intermediates during the ongoing reaction than Ni(12)-In₂O₃, Cr(10)-In₂O₃ and In₂O₃. The presence of Ni as a hydrogenation promoter can explain the larger amount of reaction intermediates compared to In₂O₃. Based on XRD, XAS and TPR, it can be inferred that Cr decreases the size of In₂O₃ particles and facilitates the formation of O_v on the In₂O₃ surface, thereby enhancing CO₂ adsorption yielding more formate and methoxy intermediates. Overall, these data point to a synergistic effect between Ni and Cr in promoting the formation of formate and methoxy intermediates in methanol formation.



Fig. 13. In situ FTIR spectra as a function of temperature and hydrogenation time over (a) Cr(10)- In_2O_3 , (b) Ni(12)- In_2O_3 and (c) Ni(12)-Cr(9)- In_2O_3 . (d) FTIR spectra at reaction temperature at 260 °C over Cr(10)- In_2O_3 , Ni(12)- In_2O_3 and Ni(12)-Cr(9)- In_2O_3 samples. (e) Comparison of peak area of HCOO* and CH_3O^* collected from the spectra at 260 °C under reaction mixture for 1 h over Cr(10)- In_2O_3 , Ni(12)- In_2O_3 and Ni(12)-Cr(9)- In_2O_3 samples. Reaction conditions: CO_2 : $H_2 = 1:3$, gas flow rate = 50 mL/min, P = 10 bar. Peak areas of IR spectra are normalized by the weight of catalyst pellet.

4. Conclusion

We synthesized binary In₂O₃-CrO_x and ternary Ni-In₂O₃-CrO_x catalysts via a one-step flame spray pyrolysis method to investigate the promoting role of Cr. For binary In₂O₃-CrO_x samples, the optimal Cr content was 2 mol%, which resulted in a more than 50 % increase in the methanol formation rate at 300 °C compared to In₂O₃. While the CO₂ conversion rates were similar at higher Cr content, the methanol selectivity decreased. Characterization of these samples and complimentary DFT calculations demonstrated that (i) doping of Cr into the In₂O₃ can lead to more O_v sites, explaining the increased performance at low Cr content, and (ii) Cr-oxide clusters destabilize nearby O atoms in the In2O3 surface sites, thereby also increasing the Ov site density and offsetting the decrease in Ov due to coverage of the In2O3 surface by Croxide. Cr2O3 formed above 2 mol% Cr suppressed the sintering of In₂O₃ during CO₂ hydrogenation, which was especially evident at the higher reaction temperature of 300 °C. The methanol rates of the ternary catalysts containing Ni were further enhanced as compared to Cr-In₂O₃ reference catalysts. The methanol formation rate correlated strongly with the Ni content, an optimum being reached at a Ni content of 22 mol % and a Cr content of 8 mol%. The resulting Ni(22)-Cr(8)-In₂O₃ catalyst displayed twice the methanol rate of a Ni(22)-In₂O₃ reference. Also with Ni promoter, Cr suppressed sintering of In₂O₃ and, thereby, catalyst deactivation. Overall, the results indicate that Ni and Cr play separate promoting roles in achieving an increased and more stable rate of methanol formation compared to In₂O₃: Ni promotes the hydrogenation of formate and methoxy surface intermediates to methanol, while Cr results in more O_v sites and suppresses sintering of In₂O₃.

CRediT authorship contribution statement

Liang Liu: Writing – original draft, Visualization, Validation, Methodology, Investigation. Francesco Cannizzaro: Investigation. Anouar Kaychouhi: Investigation. Nikolay Kosinov: Writing – review & editing, Supervision, Investigation. Emiel J.M. Hensen: Writing – review & editing, Funding acquisition, Funding acquisition, Conceptualization.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

Data will be made available on request.

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Appendix A. Supplementary data

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L. Liu et al.

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