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Highly Photocatalytic Titanium Oxide/Carbon Nitride Heterojunctions Obtained via Dual Asymmetric Centrifugation

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ABSTRACT

In this work dual asymmetric centrifugation technique was applied to obtain new TiO₂/carbon nitride heterojunctions (TiO₂/g-C₃N₄) with high photocatalytic performance. Materials were obtained from TiO₂ and g-C₃N₄, with an optimal mass ratio of 1.0:0.1 for achieving the highest photocatalytic activity in dyes degradation. The TiO₂/g-C₃N₄ (1.0/0.1) photocatalyst was characterized via X-ray diffraction, confirming the crystalline phases and chemical composition of the material. UV-VIS spectroscopy data showed an increase in the visible light absorption, indicating an effective interaction between the heterojunction components. XPS data showed the formation of Ti³⁺ ions and oxygen vacancies, boosting the photocatalytic process. The TiO₂/g-C₃N₄ (1.0/0.1) showed the highest photocatalytic performance degrading 99.9% of rhodamine B and 99.6% of methylene blue in 60 min under LED light irradiation. Radicals trapping experiments indicated O₂^{-•} radicals as the major photoactive species, through a Z-scheme type mechanism. The product of RhB photodegradation did not show toxicity to *Artemia franciscana* larvae.

KEYWORDS: Carbon nitride, TiO₂, speed-mixer, dual asymmetric centrifugation, dye photodegradation

1. Introduction

The growth of water disinfection and decontamination technologies is dependent on the development of photocatalytic materials capable of degrading organic pollutants with high efficiency. Among these high-performance materials, titanium dioxide (TiO_2) has stood out as a low-cost, non-toxic, and chemically stable semiconductor with appreciable photocatalytic activity [1]. However, TiO_2 has serious limitations such as large bandgap, poor visible-light absorption, high electro-hole recombination rate and low photogenerated charge separation/migration [2]. Combine TiO_2 with other visible light active semiconductors with narrow band gap to form heterojunctions has emerged as an excellent strategy to boost the photocatalytic performance [1–3]. On the other hand, carbon nitride (g- C_3N_4) is a promising metal-free low-coat photocatalyst, obtained through thermal poly-condensation of many organic molecules such as urea, thiourea and melamine [4,5]. Compared to TiO_2 , g- C_3N_4 has an appreciable visible light absorption near 400–460 nm, also showing high thermal stability and high chemical stability in acidic or alkaline medium [4,5]. However, shortcomings low electrical conductivity, low visible light absorption above 460 nm, and especially high charge carriers' recombination rate are still obstacles for practical applications. Several synthetic strategies such as doping have been adopted to improve the g- C_3N_4 photocatalytic properties [6–10]. Thus, the development of g- C_3N_4 -based heterojunctions are a viable alternative to improve separation/transfer efficiency of photogenerated electron–hole pairs [4,5].

In this context, TiO_2 /carbon nitride (TiO_2 /g- C_3N_4) heterojunctions show prominent position in the literature, being applied for photodegradation of dyes, drug-waste, and photoreduction of heavy ions as well [1,2]. Many synthetic strategies have been used to obtain TiO_2 /g- C_3N_4 heterojunctions such as simultaneous calcination of the precursors [11,12], loading g- C_3N_4 on the pre-obtained TiO_2 [13,14] or growing TiO_2 crystals over g- C_3N_4 surface [15,16]. Solvent-free methods were also largely applied to form TiO_2 /g- C_3N_4 heterojunctions, such as ball-milling [17,18] or manual grinding in a mortar [19,20]. These dry methods provided an environmental-friendly, low cost, fast and practical synthetic routes to obtain highly photoactive heterojunctions. For example, Li and co-workers reported the synthesis of rutile- TiO_2 /g- C_3N_4 photocatalyst manually grinding the precursors together for only 10 minutes. This heterojunction was able to totally photodegrade rhodamine B dye (RhB) under simulated sunlight irradiation in 150 minutes [20]. Wang *et. al* also reported the synthesis of TiO_2 /g- C_3N_4 heterojunction obtained by ball-milling/microwave method capable to photodegrade 78.27% of RhB and 82.62% of methylene blue (MB) under visible light irradiation in 4 hours [21].

Dual asymmetric centrifugation (DAC), operating in speed-mixer, is a nonconventional and particular type of centrifugation where the vial is turned around the main rotation axis at a fixed distance and also spins around its own centre (a vertical axis) [22]. As a result, the main rotation pushes the sample against the centrifugal force, while the spin movement leads the sample opposite direction [22,23]. This coupled movements reduce the sample's translational degree of freedom along the central axis, generating a high centrifugal acceleration and very strong shear force, dependent on the frequency of the motion [22,23]. This intense acceleration (700x gravitational force at 3500 rpm) combined with the presence of ceramic beads inside the sample vial, transform this technique a powerful tool to mixer high-viscosity fluids, make homogeneous dispersions, grind powders to a reduced size and synthesized composites [23].

Although it was first used to triturate silver amalgam alloys for dental applications [24], DAC is widely applied for the preparation of lipide-based biomaterials for drug delivery [25], nanocellulose [23], and high-viscosity polymer-based composites [26–29]. Womble and co-workers prepared for the first time TiO₂-based polymeric composite using DAC synthesis, exploring the influence of the nanoparticles on the dielectric properties of polypropylene [27,30]. Alrahlah *et al.* also utilized DAC to produce ZrO₂/TiO₂/PMMA (PMMA = polymethyl methacrylate) composites as dental material for interim restoration applications [31]. Recently, we demonstrated the use of DAC to obtain high-performance S-doped carbon nitride and TiO₂-based photocatalysts, capable of degrading rhodamine B dye in only 60 minutes [32]. Here, we reported the utilization of DAC to synthesize new TiO₂/g-C₃N₄ heterojunctions as photocatalysts. Heterojunctions were synthesized from a commercial P25 TiO₂ and g-C₃N₄ obtained from urea, and characterized by X-ray powder diffraction (XRD), infrared spectroscopy (FT-IR), X-ray photoelectron spectroscopy (XPS), solid-state UV-VIS spectroscopy, scanning electron microscopy (SEM), high-resolution transmission electron microscopy (HRTEM) and energy dispersive X-ray spectra (EDS). Photodegradation of rhodamine B (RhB) and methylene blue (MB) were conducted at room temperature in aqueous solution under visible LED light irradiation. Photocatalytic mechanism was also investigated by radical trapping technique using scavengers. The toxicity of the RhB photodegradation products were investigated using *Artemia* mobility assay.

2. Experimental

Urea (99%), Rhodamine B (RhB, 95%), Ethylenediaminetetraacetic acid (EDTA, 99.4%), p-Benzoquinone (BQ, 98%) and Isopropyl alcohol (IPA, 99.8) were acquired from Sigma-Aldrich. Titanium (IV) oxide (P25) and barium sulphate were purchased from Acros

Organics and Nacalai Tesque INC, respectively. All chemicals were used without previous purification.

2.1. Synthesis of the g-C₃N₄ nanosheets

Bulk g-C₃N₄ was prepared via thermal polymerization of urea at 550 °C for 2 hours and heating rate of 3 °C/min. g-C₃N₄ nanosheets were obtained via thermal exfoliation at 500 °C (10 °C/min) for 2 hours.

2.2. TiO₂/g-C₃N₄ heterojunctions

Titanium oxide nanopowder (Evonik P25, 100 mg) and g-C₃N₄ nanosheet (10 mg) were mixed in a plastic vessel (10 mL) using a SpeedMixer™ (DAC 150.1 FVZ) in 2000 rpm for 30 s. After this period, seven zirconia balls (3 mm diameter) were added to the vessel and the mixture was ground for 2.5 minutes in 2000 rpm (Figure 1). This heterojunction was named TiO₂/g-C₃N₄ (1.0/0.1), indicating the mass ratio between the components. Heterojunctions with other mass proportion were also prepared using the same methodology, and labelled as (1.0/0.3), (1.0/0.5), (1.0/1.0), (0.5/1.0), (0.3/1.0) and (0.1/1.0).

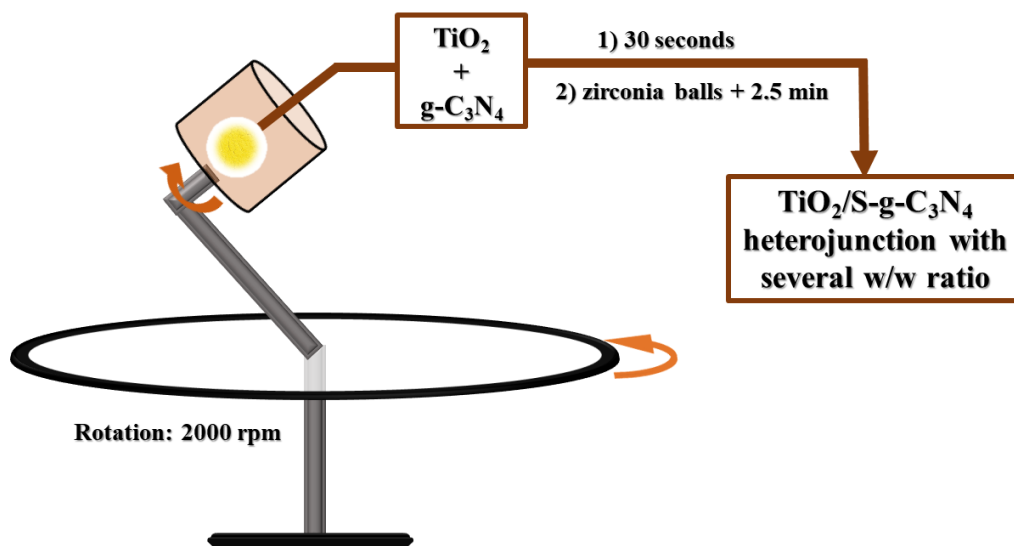


Figure 1: DAC synthesis scheme.

2.3. Characterization

The vibrational spectrum (FT-IR) between 55-4000 cm⁻¹ was collected in a PerkinElmer FT-IR spectrometer with an attenuated total reflection attachment (ATR). Powder X-ray diffraction patterns (5-80°) were obtained in a Bruker D8 Advance diffractometer with a PSD

LynxEye detector and Cu radiation ($K_{\alpha}(\text{Cu}) = 1.5406 \text{ \AA}$), step width 0.02° and acquisition time of 2 s/step. Solid state UV-VIS measurements were performed in a Shimadzu UV-2600 spectrophotometer in absorbance mode using BaSO_4 as reference. The surface chemical composition was analyzed by X-ray photoelectron spectroscopy (XPS). The sample was pressed with high purity indium and the spectra was obtained in a Scienta Omicron Argus Analyzer with monochromatic Al K_{α} X-ray source at 270 W. CasaXPS software was used for spectra deconvolution and data processing. Photoluminescence spectra were recorded in a spectrofluorometer Agilent, Cary Eclipse model, with $e_{\text{excitation}} = 350 \text{ nm}$, and excitation/emission slits of 5 nm. The heterojunction morphology was investigated *via* scanning electron microscopy (SEM) and high-resolution transmission electron microscopy (HRTEM) analysis. For SEM, samples were coated with high purity silver and micrographs were acquired in a JSM-IT300 (JEOL, Japan), voltage 15 kV, working distance of 10 mm, using both secondary and backscattered electron detectors. Energy dispersive X-ray spectra (EDS) were collected from a X-max 80 mm^2 (Oxford Instruments, UK). TEM and STEM (HAADF mode) images were collected in a JEM-2100F microscope from JEOL at 200 kV, equipped with an Orius SC1000 camera from Gatan-US. The samples were drop-casted on copper grids which were coated with a thin film ($\sim 5 \text{ nm}$) of carbon (high purity graphite), using a Q150 ES from Quorum, USA.

2.4. Photocatalytic assays

Photodegradation investigations were conducted with rhodamine B (RhB) and methylene blue (MB) dyes in aqueous solution (10 ppm and 50 mL). The respective photocatalyst (100 mg) was suspended into the dye solution and kept in the dark under magnetic stirring to achieve adsorption/desorption equilibrium. Then, the solution was irradiated using a blue LED lamp light source (Deltech FD50BL, 50 W, 475-495 nm) positioned vertically 5 cm from the solution. The photodegradation reaction was monitored via UV-VIS spectrophotometry (Shimadzu UV-2600) and the photocatalytic activities were evaluated using the maximum absorbance peak of each dye. Concentrations along the reaction were calculated by Beer-Lambert law, and photodegradation percentages were measured by the following equation:

$$\text{Photodegradation (\%)} = \frac{C_0 - C_t}{C_0} \quad (1)$$

where C_0 is the concentration before the adsorption/desorption equilibrium and C_t are the concentrations after each 30 minutes of irradiation. Radicals trapping technique was employed

to investigate the photocatalytic mechanism using the same experimental procedure, except by the addition of isopropyl alcohol (IPA, 6.4×10^{-1} mol/L), EDTA (6×10^{-4} mol/L) and p-benzoquinone (BQ, 2×10^{-3} mol/L), for scavengers of hydroxyl radicals, photogenerated holes and superoxide radicals, respectively [32].

2.5 Artemia culture and toxicity test

Artemia survival/mobility assays has been used to investigate the toxic effect of photodegraded RhB[33]. For this purpose, cysts of *Artemia franciscana* were obtained from Artemia International LLC (Texas, USA; hatching rate 90%). For nauplii hatching, cysts (10 mg) were placed in Petri dishes containing artificial seawater (ASW) for 24 h (25 ± 1 °C; 3000 - 4000 lumens). Hatched nauplii at instar I stage were transferred to Petri dishes containing fresh ASW and cultured for additional 24 h (25 ± 1 °C; dark) until the instar II stage [34].

Nauplii at instar II stage (10 larvae/well; 2 mL ASW/well; 24-well plates) were exposed to different concentrations of untreated and photodegraded RhB (4, 8, 16, 24 and 32 ppm). Larvae mobility was used as toxicity endpoint and was analysed 24 h and 48 h post-exposure under light microscope [34]. Larvae were classified as immobile if they did not exhibit any swimming behavior for longer than 10 s of observation [35]. The control group was cultured in Milli-Q[®] water (80%) plus ASW (20%) since pure Milli-Q[®] water (100%) was toxic for *A. franciscana*. Serial dilutions of untreated RhB and photodegraded RhB were prepared using Milli-Q[®] water supplemented with 20% of ASW.

Mobility percentages were calculated by the following equation:

$$Mobility (\%) = \frac{S - I}{S} \times 100 \quad (2)$$

where S is the number of swimming larvae, and I is the number of immobile larvae. The experiments were carried out in triplicate (technical replicate) and repeated four times in an independent manner (biological replicate).

3. Results and Discussion

Pure g-C₃N₄ was obtained through thermal polymerization of urea. XRD pattern (Figure S1) shows the main signal at 27.57° related to the (002) diffraction plane due to interlamellar aromatic π - π stacking interactions, typical of graphite-like materials. This corresponds to an interlayer distance of 0.32 nm, in agreement with the literature [36]. The signal regarding to the (100) diffraction plane was also observed at 12.95° and d-spacing of 0.68 nm, corresponding to the intralayer distance [36]. DAC was employed to obtain TiO₂/g-C₃N₄ heterojunctions with

different mass ratio, however, since they are similar, only the characterization of sample with mass ratio (1.0/0.1) was reported here due to its superior photocatalytic activity. The X-ray diffraction pattern of TiO₂/g-C₃N₄ (1.0/0.1) (Figure 2 (a) blue line) shows the typical 2θ values for anatase (ICSD 082082) and rutile (ICSD 085492) crystalline phases. The most intense signal for each phase is at 25.6° and 27.720°, related to the (101) and (110) diffraction planes of anatase and rutile, respectively.

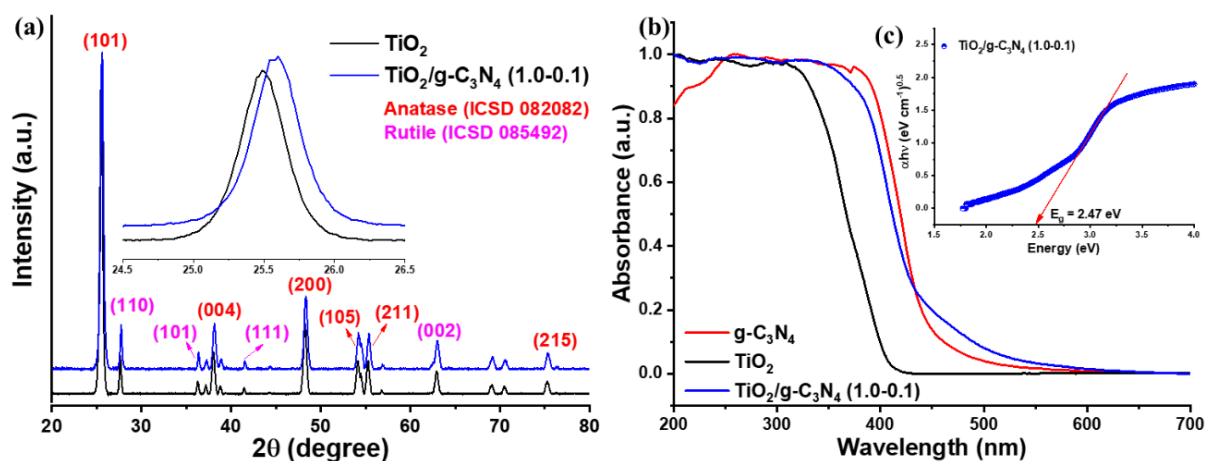


Figure 2: (a) Experimental powder pattern of pure TiO₂ (black line) and TiO₂/g-C₃N₄ (1.0/0.1), highlighting the shifts in the XRD signal (insert). (b) Solid-state UV-VIS absorption spectra of the precursors, TiO₂/g-C₃N₄ (1.0/0.1) (blue line) and (c) Tauc plot for the heterojunction.

Crystallite sizes were estimated by Scherrer' equation:

$$D = \frac{K\lambda}{\beta \cos\theta} \quad (3)$$

where D is the crystallite size, K is the shape factor (≈ 0.90), λ is the K_α wavelength of the Cu radiation source (1.5406 Å) and β is the full width at half maximum (FWHM) of each individual peak and θ is the Bragg's angle. Considering the first eight signals, D values of 22.58 nm and 23.83 nm were obtained, respectively for TiO₂/g-C₃N₄ (1.0/0.1) and the pure TiO₂. This decrease in the crystallite size can be justified by the grinding process during the DAC synthesis. As expected, no signals related to the g-C₃N₄ were observed due to its low content, in agreement with similar systems in the literature [21,32]. On the other hand, the main vibrational bands of g-C₃N₄ can be observed in Figure S2, due to its strong absorption in the infrared region compared to TiO₂. The band centered at 3336 cm⁻¹ corresponds to the asymmetric N-H and O-H stretching of g-C₃N₄ terminal amine groups and hydroxyl groups on

the metal oxide surface. Signals at 1604 cm⁻¹ and 1029 cm⁻¹ are due to the C-N/C=N stretching of the heptazine rings (C₆N₇), while the signal regarding the triazine ring (C₃N₃) are located at 1420 cm⁻¹. Signals related to the asymmetric C-H stretching (2907 cm⁻¹), asymmetric Ti-O stretching (600 cm⁻¹) and N-H bending (883 cm⁻¹) are also observed. All signals observed agrees with similar heterojunctions in the literature [37–41].

UV-VIS solid-state absorption spectra of TiO₂ (black line), g-C₃N₄ (red line) and TiO₂/g-C₃N₄ (1.0/0.1) (blue line) are shown in Figure 2 (b). An improvement in the visible light absorption of the heterojunction was observed compared to the pure TiO₂, red shifting the maximum absorption peak. TiO₂/g-C₃N₄ (1.0/0.1) also shows a higher absorption at λ > 450 nm compared to the pure g-C₃N₄, extending the absorption far in the visible region and favoring the photocatalytic activity. Li *et al.* also observed this phenomenon for rutile-TiO₂/g-C₃N₄ heterojunction obtained by manual grinding method in a mortar [20]. Other heterojunctions with higher visible absorption compared to the pure g-C₃N₄ were also obtained by hydrothermal [11], simultaneous calcination [14] and impregnation [42,43] methods. The materials' bandgaps (E_g) were calculated by Tauc plot using the following equation:

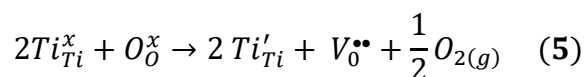
$$(\alpha h\nu)^n = A(h\nu - E_g) \quad (4)$$

where A is a constant, and α, h, and ν are the absorption coefficient, Plank's constant and the photon frequency, respectively. Considering TiO₂ and g-C₃N₄ as semiconductors with direct and indirect bandgap (n = 0.5 or n = 2) [44,45], E_g values were 3.25 eV and 2.63 eV, respectively (Figure S3 and S4). For the TiO₂/g-C₃N₄ (1.0/0.1) heterojunction (Figure 2 (c)), the estimated E_g was 2.47 eV, because of the improvement in the absorption in visible region. This reduction in the bandgap compared to the pure g-C₃N₄ was also noticed by Li and co-workers in the heterojunctions synthesized through impregnation in methanol solution [42] and by Ma *et al.* in heterojunctions obtained by growing TiO₂ crystals on the g-C₃N₄ surface [46].

Surface chemical composition of the TiO₂/g-C₃N₄ (1.0/0.1) heterojunction and the precursors were investigated via XPS. Carbon 1s XPS spectrum (Figure S5) for pure g-C₃N₄ shows signals at 285.9, 286.9 and 291.3 eV, due to the presence of the C=C, C-N-C and N-C=N organic groups of heptazine and triazine motifs, respectively [32,47]. N1s spectrum (Figure S5) also confirms the signals related to the main nitrogenated groups -C=N-C- (396.4 eV), -NC₃ (398.2 eV) and -NH₂ (402.3 eV). O 1s and Ti 2p high-resolution XPS spectra for pure TiO₂ are shown in Figure S6. Deconvolution for oxygen 1s spectrum indicates the presence of lattice oxygen (530.44 eV) and oxygen vacancies (531.76 eV). Quantitatively (Table S1), oxygen vacancies represent 9.93% on the TiO₂ surface, with the remaining percentage (90.07%) corresponding to lattice oxygen atoms. Thus, the presence of oxygen vacancies leads to the

reduction of $Ti^{4+} \rightarrow Ti^{3+}$ on the oxide surface, as confirmed by the presence of signals at 456.97 eV (Ti 2p_{3/2}) and 461.28 eV (Ti 2p_{1/2}) of the Ti^{3+} ions (Figure S6), in agreement with other reports in the literature [48,49]. The majority presence of Ti^{4+} ions in pure TiO_2 was confirmed by the peaks near 464.88 eV (Ti 2p_{1/2}) and 459.16 (Ti 2p_{3/2}), with concentration of 97.80% (Table S1), far larger than Ti^{3+} (2.20%), agreeing with the low content of oxygen vacancies.

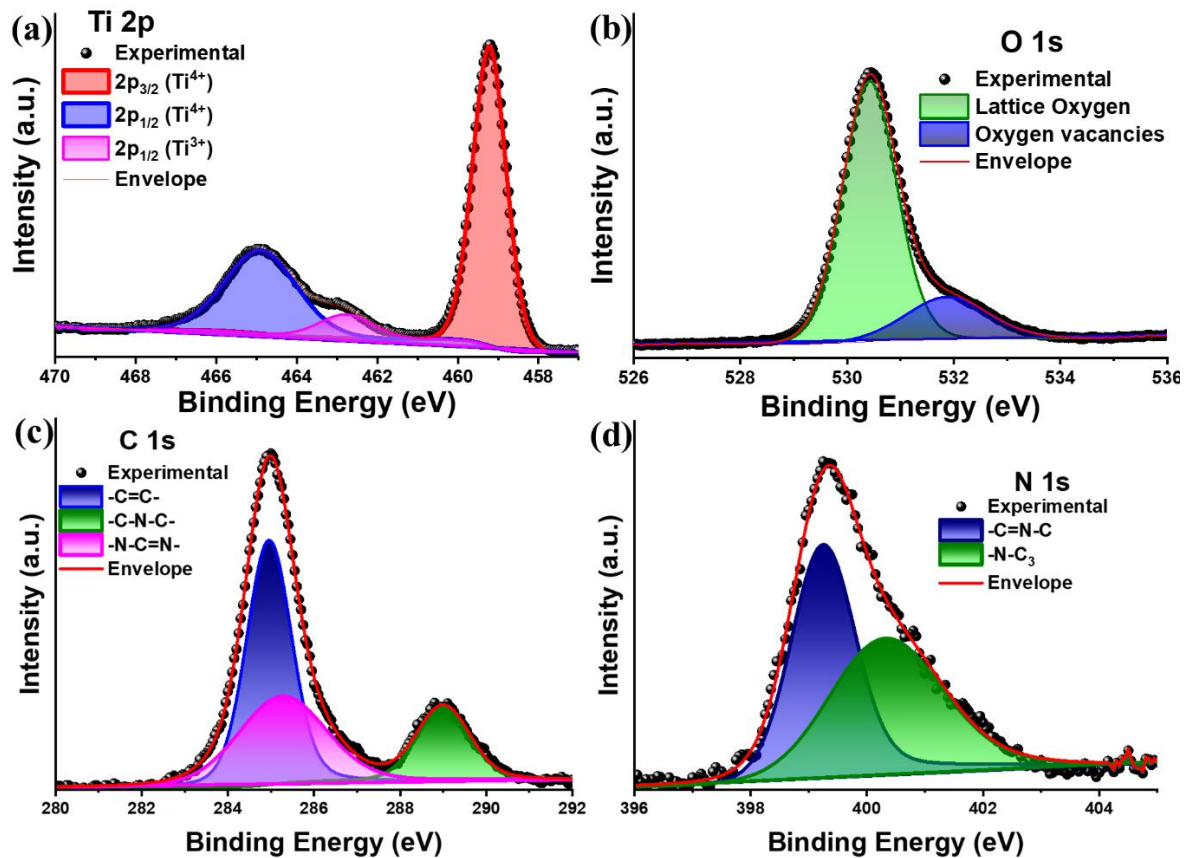
For $TiO_2/g-C_3N_4$ (1.0/0.1), the survey spectrum (Figure S7) reveals the presence of Ti, O, C and N, as expected. The largest percentage of the material surface is composed of oxygen (45.67%) and titanium (32.43%) atoms, in accordance with the higher TiO_2 content in the sample compared to $g-C_3N_4$ (Table 1). The Ti 2p high-resolution spectrum (Figure 3(a)) was deconvoluted into three peaks at 459.21eV (Ti^{4+} 2p_{3/2}), 464.89 eV (Ti^{4+} 2p_{1/2}) and 462.82 (Ti^{3+} 2p_{1/2}). The intense energy used in the DAC synthesis process led to more intense reduction of Ti^{4+} ions, increasing the percentage of Ti^{3+} from 2.20% to 9.70% on the surface (Table S2). For O 1s high-resolution XPS spectrum (Figure 3(b)) an intense signal at 530.42 was noticed, related to the lattice oxygen atoms into the TiO_2 structure. Additionally, the increase in the percentage of oxygen vacancies in the composite (19.40%, Table S3 and Figure 3(b)) was also observed. Thus, high energy in the grinding process can lead to a recombination of oxygen atoms in the surface to form O_2 , reducing metal ions and generating oxygen vacancies. This process is summarized in the following equation using Kröger-Vink notation.



Deconvolution of carbon 2p spectrum (Figure 3(c)) indicates the presence of three peaks at 284.9 eV, 285.3 eV, 289.0 eV, corresponding respectively to C=C, C-N-C and N-C=N organic groups of heptazine and triazine motifs into the $g-C_3N_4$. For N 1s spectrum (Figure 3(d)), two signals at 399.2 eV and 400.3 eV were observed, related to C=N-C and N-C₃ organic groups in the $g-C_3N_4$. The binding energy shifts observed in the Ti 2p, C 1s and N 1s spectra demonstrate the formation of the heterojunction, as well as an effective interaction and charge migration between the components, as observed in similar to other $TiO_2/g-C_3N_4$ composites in the literature [11–17,42,43,50].

Table 1: XPS survey deconvolution parameters for TiO₂/g-C₃N₄ (1.0/0.1) heterojunction.

Element	Position (eV)	FWHM	Area	Atomic percentage (%)
Ti 2p	451.50	2.19	2877.9	32.43
O 1s	530.50	1.51	1520.3	45.67
C 1s	285.00	1.63	220.4	19.40
N 1s	399.50	1.76	51.1	2.50

**Figure 3:** High-resolution XPS spectra of (a) Ti 2p, (b) O 2p, (c) C 2p and (d) N 2p for TiO₂/g-C₃N₄ (1.0/0.1) heterojunction.

The efficiency in separating the photogenerated charges was investigated using solid-state photoluminescence spectroscopy at room temperature, and the results are shown in Figure S8. Under excitation at 350 nm, the free g-C₃N₄ presents a broad blue emission band centered at 460 nm (Figure S8, red line). According to Zhang and co-workers, this typical blue emission is related to the electronic transitions between lone pair (LP) states in valence band and the π^* antibonding states in conduction band [51]. TiO₂/g-C₃N₄ (1.0/0.1) heterojunction also exhibits a large emission band, blue-shifted by 16 nm (Figure S8, blue line). The heterojunction

formation also led to a significant decrease in photoluminescence intensity, indicating a decrease in the recombination processes of photogenerated charges [18,21,47], boosting the photocatalytic performance.

Morphological investigations of the TiO₂/g-C₃N₄ (1.0/0.1) heterojunction reveal the presence of high aggregated particles (Figure 4(a)), like other samples prepared through mechanical methods *e.g.*, ball milling [18,21,38,47]. TEM images (Figure 4(b) and S9) indicate a distribution of TiO₂ nanoparticles on the g-C₃N₄ nanosheet surface, also confirmed by the presence of the (101) diffraction plane of TiO₂ with d-spacing 0.356 nm (Figure 4(c)), similar to the one found in the literature [21,47] and near the value obtained in the XRD data (0.350 nm). The TiO₂ particle size distribution (Figure 4(d)) shows a mean size of 22.67 nm, consistent with the crystallite size obtained by experimental diffraction patterns. EDS mappings (Figure 4(e-h)) confirm the Ti, O, C and N elemental composition and the presence of TiO₂ nanoparticles on the surface of the carbonaceous material. However, the high content of carbon and nitrogen around the TiO₂ nanoparticles (Figure 4(g and h)) can also be noticed. More detailed elemental mappings of TiO₂ nanoparticles (Figure S10) confirm the presence of a g-C₃N₄ covering (\approx 6.5 nm thickness) around the TiO₂ nanoparticles, similar to other works in literature [17].

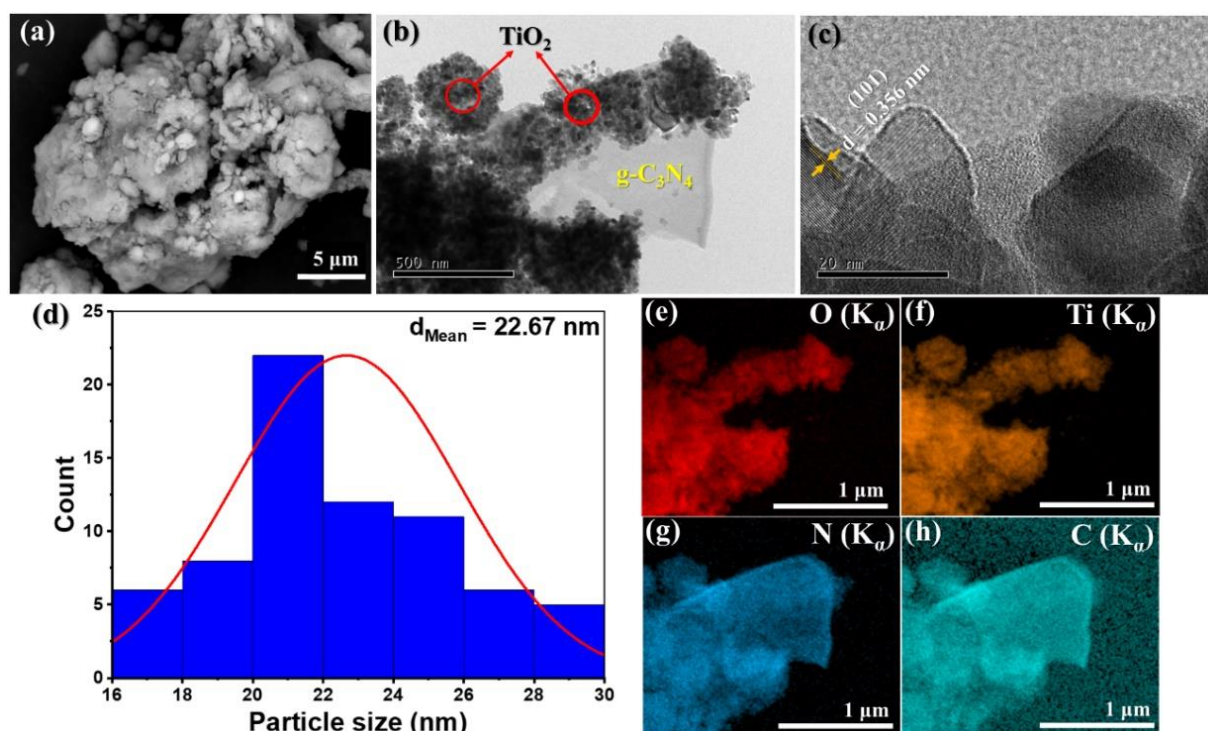


Figure 4: (a) SEM and (b,c) HRTEM images, (d) particle size distribution and (e-h) EDS mapping of O, Ti, N and C of TiO₂/g-C₃N₄ (1.0/0.1) heterojunction.

Photocatalytic performance of the $\text{TiO}_2/\text{g-C}_3\text{N}_4$ heterojunctions for the degradation of RhB and MB dyes were investigated compared to the pure precursors. For RhB, the system was left until the adsorption/desorption equilibrium for 30 minutes (Figure 5(a)), resulting in adsorption uptakes between 38.9-13.8% (Table S4) with the high value for the $\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0/0.1), may indicate higher surface area and higher RhB affinity of this heterojunction compared to the others. For the pure TiO_2 and $\text{g-C}_3\text{N}_4$, the adsorption percentages were 5.17% and 20.1%, respectively, suggesting that DAC synthesis improves the materials' surface area for $\text{TiO}_2/\text{g-C}_3\text{N}_4$ mass ratio ≥ 1 . After light was on, RhB started to photodegrade, indicated by the decreasing in the absorption bands and concentrations (Figure 5(a) and S8). The dye absorption bands profile during the photocatalytic process (Figure S11) indicates a degradation pathway via *N*-deethylation process as observed before for similar photocatalysts [32,52,53]. The heterojunctions degradation profile and percentages are shown Figure 5(b,c) and Table S4, showing the $\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0/0.1) with the highest photocatalytic performance, degrading 99.9% of RhB in 60 minutes of irradiation. All heterojunctions show superior photocatalytic activity compared to the pure TiO_2 (59.1%, Figure S12) and the $\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0/0.1) sample also has higher performance than the free $\text{g-C}_3\text{N}_4$ (97.8%, Figure S13). The moderate photocatalytic activity of TiO_2 can be explained dye-sensitized energy transfer mechanism, due to the presence of RhB on the metal oxide surface [54].

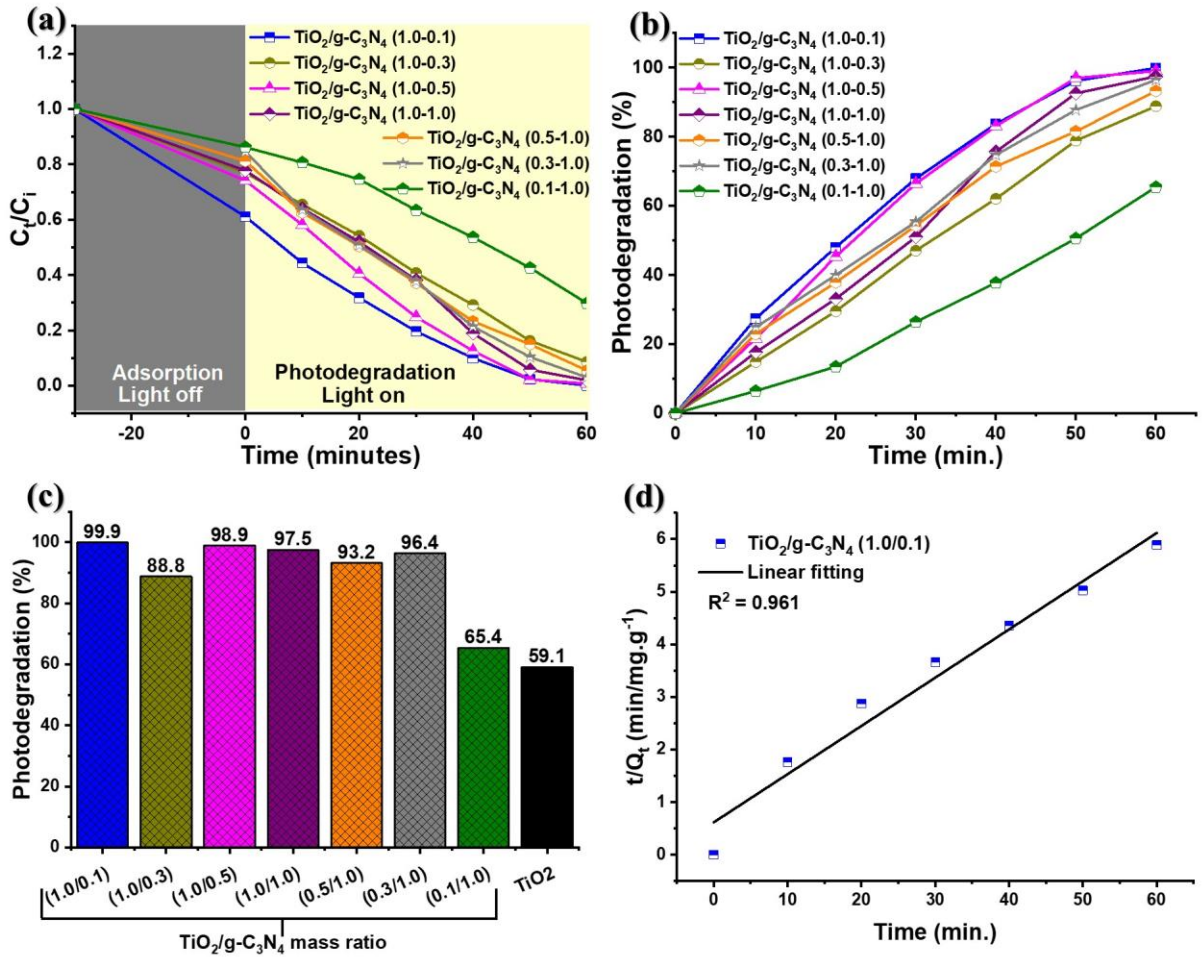


Figure 5: (a) C_t/C_i ratio with decreasing in the RhB concentration during adsorption and photodegradation processes, (b) photodegradation profile and (c) degradation percentages of the heterojunctions, and (d) Pseudo-second order plot of the $\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0/0.1).

From the $\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0/0.1) photodegradation data, the reaction kinetics were investigated, and result is shown in Figure 5(d). The experimental data agree with the pseudo-second order model, when the reaction rate is dependent on the amount of the solute adsorbed on the photocatalyst surface and the amount adsorbed at equilibrium, as described by the equation (5):

$$\frac{t}{Q_t} = \frac{1}{k_2 Q_e^2} + \frac{t}{Q_e} \quad (6)$$

where t , Q_t , Q_e and k_2 are the time, amount of dye adsorbed at each time, amount of dye adsorbed at the equilibrium and the pseudo-second order reaction rate, respectively [55,56]. Pseudo-second order kinetic models were also used to describe photocatalytic and adsorption properties of other carbon nitride-based materials in literature [57–59]. Pure TiO_2 follows a

pseudo-first order kinetic model as shown in the Figure S12, commonly found for several photocatalysts in the literature including $\text{TiO}_2/\text{g-C}_3\text{N}_4$ systems [17,21,37]. However, the photodegradation of RhB in the presence of pure $\text{g-C}_3\text{N}_4$ also fits well with the pseudo-second order model, as shown in Figure S13. Clearly, the $\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0/0.1) heterojunction presents superior performance compared with the pure precursors. Table 2 shows results of the photocatalytic performance of $\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0/0.1) sample in the RhB photodegradation, compared to the other heterojunctions with similar compositions and obtained through different synthesis methods. The photocatalyst obtained here shows similar or superior activity, as observed, indicating the DAC technique as a promising method to obtain high photoactive heterojunctions.

Table 2: Photocatalytic activity of $\text{TiO}_2/\text{g-C}_3\text{N}_4$ -based heterojunctions in the degradation of RhB.

Photocatalyst	Synthesis method	Degradation (%)	Time (min)	Light source	Ref.
$\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0-0.1)	DAC	99.9	60	50 W blue LED	This work
N- $\text{TiO}_2/\text{g-C}_3\text{N}_4$	grinding	98.8	35	800 W Xe lamp with a UV cut-off filter	[37]
rutile- $\text{TiO}_2/\text{g-C}_3\text{N}_4$	grinding	99	150	500 W Xe lamp	[20]
$\text{g-C}_3\text{N}_4/\text{TiO}_2$	ball milling	78.27	240	500 W Xe lamp with UV cut-off filter	[21]
$\text{g-C}_3\text{N}_4/\text{TiO}_2$	ball milling	99.41	120	60 W, 460 nm lamp	[60]
$\text{Ti}^{3+}\text{-TiO}_2/\text{O-g-C}_3\text{N}_4$	hydrothermal	99	100	30 W LED cold visible light	[61]
$\text{g-C}_3\text{N}_4/\text{TiO}_2$	hydrothermal	80	300	500 W Xe lamp UV-cut off filter (365 nm)	[62]
m $\text{TiO}_2\text{-ex-C}_3\text{N}_4$	hydrothermal	100	540	150 W halogen lamp with UV-cut off filter (400 nm)	[63]
$\text{g-C}_3\text{N}_4/\text{TiO}_2$	hydrothermal	100	10	300 W Xe lamp	[64]
$\text{g-C}_3\text{N}_4/\text{reduced TiO}_2$	hydrothermal	88.6	90	350 W Xe lamp UV-cut off filter (420 nm)	[65]
TiO_2 nanorods/ $\text{g-C}_3\text{N}_4$ nanosheets	hydrothermal	99.9	12	300 W Xe lamp	[66]
$\text{TiO}_2/\text{g-C}_3\text{N}_4$ nanosheets	hydrothermal	99.9	60	500 W Xe lamp	[67]
$\text{TiO}_2/\text{Graphitic Carbon Nitride Nanosheets}$	impregnation in solution	100	10	500 W Xe lamp	[68]
core-shell $\text{TiO}_2@\text{g-C}_3\text{N}_4$	impregnation in solution	93.3	100	300 W Xe lamp with a UV-cut off filter (420 nm)	[69]

g-C ₃ N ₄ @TiO ₂ hollow microspheres	impregnation in solution	90.8	90	350 W Xe lamp	[70]
g-C ₃ N ₄ /TiO ₂ with exposed (001)	impregnation in solution	55	300	300 W halogen lamp	[71]
g-C ₃ N ₄ -TiO ₂	impregnation in solution	99	20	500W Xe lamp	[72]
TiO ₂ hollow sphere/g-C ₃ N ₄	Impregnation in solution	98.1	120	250 W Xe lamp with UV-cut off filter (420 nm)	[73]
2D/2D g-C ₃ N ₄ /TiO ₂ with exposed (001)	Impregnation in solution	99	120	500 W Xe lamp	[74]

Photocatalytic performance of the TiO₂/g-C₃N₄ (1.0/0.1) heterojunction for degradation of methylene blue (MB) was also investigated (Figure 6 and S14), compared to the free precursors (Figure S15-17). The systems achieved the adsorption/desorption equilibrium after 60 minutes with MB uptake of 1.5%, 22.3% and 21.5% for TiO₂, g-C₃N₄ and TiO₂/g-C₃N₄ (1.0/0.1), respectively. After irradiation, the MB concentration decreased in the presence of all photocatalysts (Figures 6(a), S15 and S16), and the blueshifts observed in the main absorption band at 664 nm indicate that photodegradation follows a mechanism via loss of dimethylamino groups [75]. As shown in Figures 6(b) and (c), the TiO₂/g-C₃N₄ (1.0/0.1) heterojunction shows the highest photodegradation percentage (99.6%) in 150 minutes, compared to pure TiO₂ (60.1%) and pure g-C₃N₄ (84.4%).

In opposition to the RhB photodegradation, both precursors present reaction kinetics consistent with a pseudo-first order model, as observed in Figures S17 and S18. Nonetheless, the experimental data of the TiO₂/g-C₃N₄ (1.0/0.1) heterojunction also agrees with a pseudo-second order kinetic reaction, as observed for RhB photodegradation. (Figure 6(d)). Table 3 shows the MB photodegradation and reaction time values for several TiO₂/g-C₃N₄ based photocatalysts, compared to those ones obtained in this work. The TiO₂/g-C₃N₄ (1.0/0.1) heterojunction presents photocatalytic performance equal to or superior to the other photocatalysts with a lower power visible light source, when compared to light sources conventionally used in the literature (300 W Xe lamp). In general, higher photodegradation reaction times for MB were observed in the literature, compared to those found for the degradation of RhB. However, the reaction time obtained for the TiO₂/g-C₃N₄ (1.0/0.1) heterojunction in the total degradation of MB (120 minutes) is far inferior to other catalysts of similar composition.

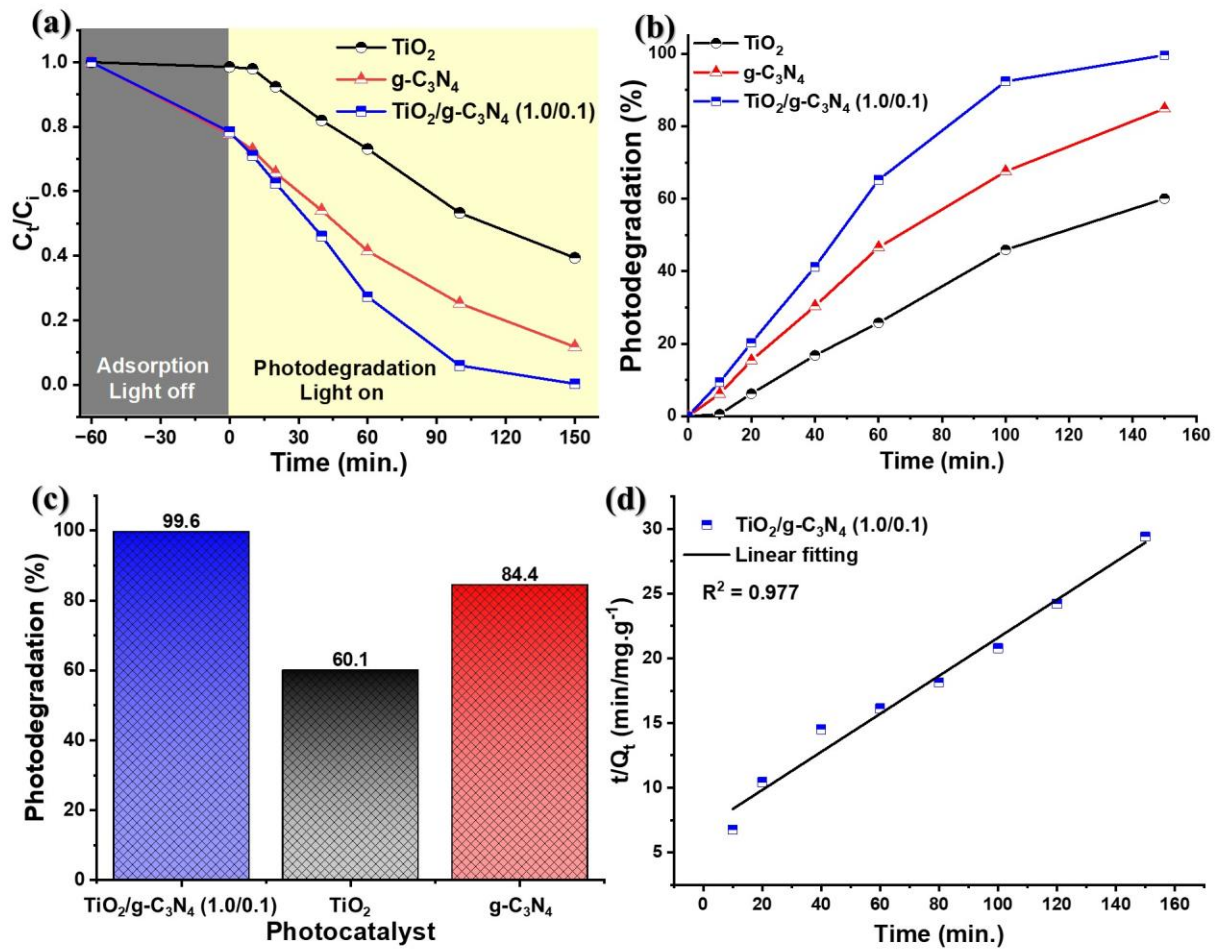


Figure 6: (a) C_t/C_i ratio with decreasing in the MB concentration during adsorption and photodegradation processes, (b) photodegradation profile and (c) degradation percentages of $\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0/0.1) compared to the precursors, and (d) Pseudo-second order plot of the $\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0/0.1).

Table 3: Photocatalytic activity of $\text{TiO}_2/\text{g-C}_3\text{N}_4$ -based heterojunctions in the degradation of MB.

Photocatalyst	Degradation (%)	Time (min)	Light source	Ref.
$\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0-0.1)	99.6	150	50 W blue LED	This work
$\text{g-C}_3\text{N}_4/\text{TiO}_2$	82.62	120	300 W Xenon lamp with a cut-off filter ($\lambda > 420$ nm)	[21]
$\text{g-C}_3\text{N}_4/\text{TiO}_2$ films	68	180	50 W Halogen lamp with a cut-off filter ($\lambda > 400$ nm)	[76]
$\text{g-C}_3\text{N}_4/\text{TiO}_2$	91	360	500W Xenon lamp with a cut-off filter ($\lambda > 400$ nm)	[77]
$\text{g-C}_3\text{N}_4/\text{TiO}_2$	98	100	100 W Halogen lamp with a cut-off filter ($\lambda > 400$ nm)	[78]

Na-g-C ₃ N ₄ /TiO ₂	90	75	300 W Xenon lamp with a cut-off filter ($\lambda > 400$ nm)	[79]
N/Ti ³⁺ -codoped triphasic g-C ₃ N ₄ /TiO ₂	81.64	100	Xenon lamp with a cut-off filter ($\lambda > 420$ nm)	[80]
Mesoporous TiO ₂ /g-C ₃ N ₄ nanocomposite	96	90	400 W visible lamp	[81]
g-C ₃ N ₄ /TiO ₂	94	150	300 W Xenon lamp with a cut-off filter ($\lambda > 420$ nm)	[82]
mpg-C ₃ N ₄ /anatase TiO ₂	64	360	500 W Xenon lamp	[83]
TiO ₂ /g-C ₃ N ₄	100	330	15 W of UV-B lamps	[84]
g-C ₃ N ₄ /TiO ₂ nanotube	81	180	Xenon lamp with a cut-off filter ($\lambda > 420$ nm)	[85]
g-C ₃ N ₄ /TiO ₂ nanocomposite	100	120	300 W halogen lamp with a UV cut-off filter	[86]
Ti ³⁺ Self-Doped TiO ₂ /g-C ₃ N ₄	99	100	30 W LED visible light	[87]
TiO ₂ /g-C ₃ N ₄	73	80	250 W Xenon lamp with a cut-off filter ($\lambda > 420$ nm)	[88]
g-C ₃ N ₄ /N-TiO ₂	65	80	500 W Xenon lamp with a cut-off filter ($\lambda > 420$ nm)	[89]
g-C ₃ N ₄ /C-TiO ₂	99	150	400 W Xenon lamp with a cut-off filter ($\lambda > 420$ nm)	[90]
g-C ₃ N ₄ /TiO ₂	95	360	500W Xenon lamp with a 420 nm cut-off filter	[91]
g-C ₃ N ₄ /TiO ₂	91.75	120	60 W lamp (460 nm)	[60]

Radical trapping technique was applied to investigate the RhB photocatalytic mechanism of the TiO₂/g-C₃N₄ (1.0/0.1), using BQ, IPA and EDTA as scavengers for O₂^{-•}, HO[•] and h⁺, respectively. Results in Figure S19 show a photodegradation of 34%, 68.6% and 81.2%, respectively when scavengers of superoxide radicals, hydroxyl radicals and holes were applied. Thus, experimental data indicate that the three photogenerated species (O₂^{-•}, HO[•] and h⁺) have influence on the RhB degradation process, however the major photocatalytic mechanism is through the formation of superoxide radicals, followed by the HO[•] radicals. Since the formation of photogenerated species is related to the energetic position of the valence (E_{VB}) and conduction bands (E_{CB}), Valence Band XPS (VB-XPS) measurements were conducted in order to obtain the energy diagram, seeking a better understanding of the photocatalytic mechanism.

The experimental potentials for the valence band of pure TiO₂ and g-C₃N₄ obtained (Figure S20-21) were converted to standard hydrogen potential (E_{VB-NHE}) through the equation (6) [92]:

$$E_{VB-NHE} = \varphi + E_{VB-XPS} - 4.44 \quad (7)$$

where φ is XPS analyzer work-function (4.55 eV) and E_{VB-XPS} is the potential obtained from the VB-XPS spectrum, being 3.127 eV and 0.337 eV for TiO₂ and g-C₃N₄, respectively. The energy positions of the valence band (E_{VB}) were +3.24 eV and +0.45 eV for TiO₂ and g-C₃N₄, respectively. Considering the experimental E_g values obtained from the UV-VIS data the conduction band energy positions (E_{CB}) for TiO₂ and g-C₃N₄ were +0.40 eV and -2.18 eV, respectively.

Since O₂[•] and HO[•] radicals play the most important role in the photocatalytic degradation of RhB (Figure S19), it is possible to investigate the formation of these photogenerated species, bearing in mind that the superoxide radicals are obtained by reduction of atmospheric O₂ ($E^\circ(\text{O}_2/\text{O}_2^\bullet) = -0.33$ eV) by the electrons at conduction band (CB) and hydroxyl radicals are formed by water ($E^\circ(\text{H}_2\text{O}/\text{HO}^\bullet) = +1.99$ eV) or hydroxyl ions oxidation ($E^\circ(\text{OH}^-/\text{HO}^\bullet) = +2.72$ eV) through the holes at the valence band (VB) of the materials. Notoriously, the O₂[•] radicals are the main photoactive species during the RhB degradation, however, these radicals only can be formed at the g-C₃N₄ conduction band ($E_{CB} = -2.18$ eV) due to its higher negative potential. In addition, comparing the E_{VB} positions of TiO₂ (+3.24 eV) and g-C₃N₄ (+0.45 eV), high h⁺ concentration must be positioned at VB of TiO₂ to generate HO[•] radicals from water molecules and/or hydroxyl ions [52]. Holes preferentially positioned in the TiO₂ valence band can also contribute to the direct oxidation of RhB, however, the trapping experiments results (Figure S19) indicate this as a less favored pathway when compared to the dye oxidation via superoxide and hydroxyl radicals.

Therefore, considering the results observed and the E_{VB} and E_{CB} energy positions of the materials, the electrons must be preferentially positioned at the CB of the g-C₃N₄, and holes must be preferentially located in the VB of TiO₂, and these observations are compatible with a Z-scheme type mechanism (Figure 7). In this case, there is a selective recombination between holes located in the VB of g-C₃N₄ and electrons located in the CB of TiO₂. This results in the remaining photogenerated charges located in the higher positive VB and in the more negative CB, favoring a high photocatalytic performance. These results are similar to the photodegradation mechanism found for other TiO₂/g-C₃N₄ based materials [20,21,32,64,74].

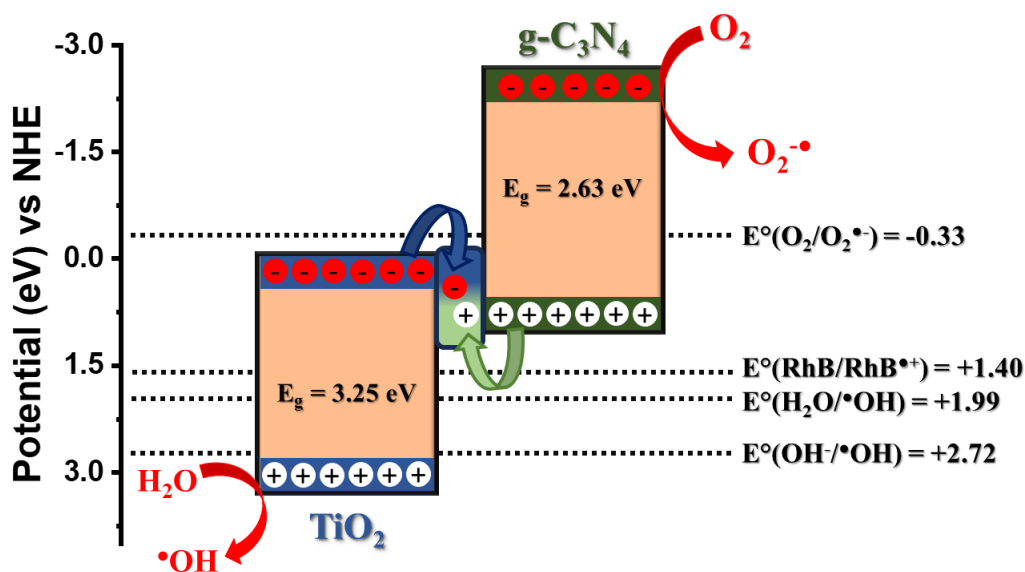


Figure 7: Z-scheme mechanism in the photodegradation of RhB in the presence of TiO₂/g-C₃N₄ (1.0/0.1) photocatalyst.

Artemia toxicity test has been widely used to evaluate the harmful effect of several compounds - from pharmaceutical drugs to environmental contaminants - to the biological systems [93,94]. Our experimental protocol intended to evaluate two different stages of *Artemia franciscana* development: instar II (24 h of exposure) and instar III (48 h of exposure), being the last the most sensitive to toxic compounds [34,95]. Our results showed that photodegraded RhB did not exhibit deleterious effect on *A. franciscana* larvae even after 48 h of exposure for all concentration tested (Figure 8(b); compared to the control group; Kruskal-Wallis test). No difference was found in larvae mobility after exposure to untreated RhB or photodegraded RhB. The mobility rates at the highest concentration tested were $88.3 \pm 3.66\%$ and $80 \pm 4.77\%$ (mean \pm SEM; untreated RhB and photodegraded RhB, respectively; 32 ppm; 48 h of exposure; Kruskal-Wallis test). So, our data strongly suggests that no toxic compounds were produced as result of RhB photodegradation by TiO₂/g-C₃N₄, pointing out the method as a safe.

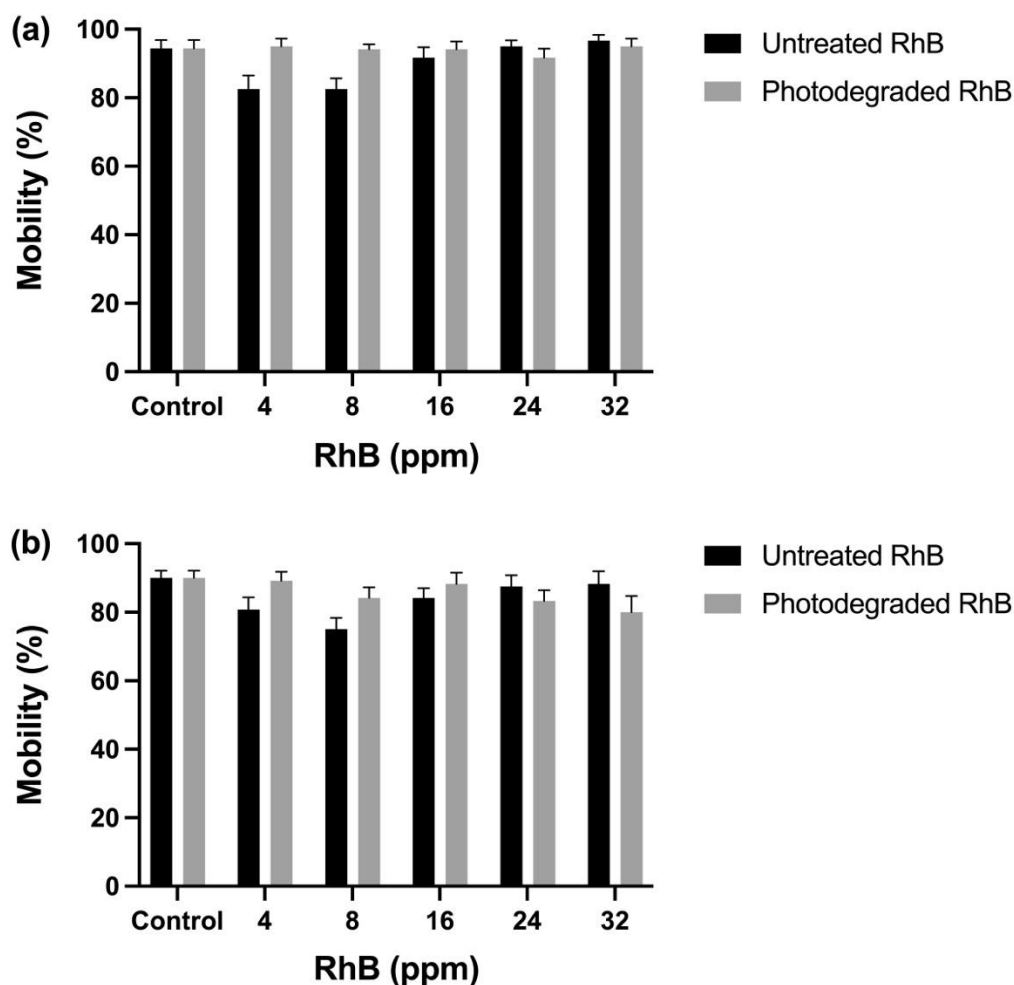


Figure 8: Mobility of *Artemia franciscana* larvae after 24 h (a) and 48 h (b) of exposure to untreated RhB and photodegraded RhB. Data are expressed as mean \pm SEM.

4. Conclusion

Here, we demonstrate the application of the DAC technique for the synthesis of new $\text{TiO}_2/\text{g-C}_3\text{N}_4$ heterojunctions, confirmed by TEM and XRD analysis. The synthesis methodology has proved to be effective for producing heterojunctions with greater absorption in the visible light region, when compared to the precursors. XPS data shows a $\text{Ti}^{4+} \rightarrow \text{Ti}^{3+}$ reduction during the synthesis, leading to the formation of oxygen vacancies, improving the photocatalytic process. Photodegradation experiments indicate a mass ratio of 1:0.1 as optimal for the highest photocatalytic performance in RhB and MB degradation. Radical trapping experiments show the $\text{TiO}_2/\text{g-C}_3\text{N}_4$ (1.0/0.1) heterojunction functions via a Z-scheme mechanism. All RhB photodegradation products shown no ecotoxicity against *A. franciscana* larvae.

Supporting Information

The Supporting Information is available free of charge and shows additional data such as infrared spectra, UV-VIS spectra, and XRD details.

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Conflict of Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

CRedit authorship contribution statement

Fausthon Fred da Silva: Conceptualization, Methodology, Validation, Resources, Visualization, Investigation, Writing original draft, Formal analysis, Project administration.

André L. Menezes de Oliveira: Methodology, Validation, Visualization, Investigation, Writing original draft and Formal analysis.

Thayná Carla Barbosa de Araújo: Methodology, Validation, Visualization, Investigation, Writing original draft and Formal analysis.

Luis Fernando Marques-Santos: Methodology, Validation, Resources, Visualization, Investigation and Writing original draft and Formal analysis.

Bo Su: Conceptualization, Methodology, Validation, Resources, Visualization, Investigation, Funding acquisition, Writing original draft, Formal analysis, and Project administration.

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