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Transitions between Nanoporous crystalline α and β forms of poly (2,6-dimethyl-1,4-phenylene) oxide (PPO)

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1. Introduction

Poly(2,6-dimethyl-1,4-phenylene) oxide (PPO) is a commercial engineering plastic, being amorphous after melt processing $[1-11]$ $[1-11]$. It is well known that this amorphous polymer can be crystallized by suitable solvent treatments $[12-21]$ $[12-21]$, generally leading to co-crystalline (CC) forms with low-molecular-mass guest molecules [16–[21\].](#page-5-0) CC forms of PPO were divided in two Classes: *i*) with few specific guest molecules (decalin, tetralin and α-pinene) leading to highly ordered and large crystallites (even single crystals) and exhibiting small diameter 4/1 polymer helices with *inter-helical guest location* [16–[18\];](#page-5-0) *ii*) with most organic guest molecules leading to poorly ordered and small crystallites and exhibiting large diameter helices with *intra-helical guest location* [19–[21\]](#page-6-0).

Guest removal from CC PPO phases of Class *i* leads to polymer amorphization while guest removal from CC PPO phases of Class *ii* leads to nanoporous-crystalline (NC) phases, i.e. crystalline phases with density lower than for the corresponding amorphous phase [\[19](#page-6-0)–21]. These NC phases exhibit high uptakes in their crystalline channels of organic guest molecules, even when present in traces in air or water [21–[30\]](#page-6-0).

Detailed studies, mainly based on wide-angle X-ray diffraction (WAXD) and Fourier Transform Infrared (FTIR) characterizations, have shown the presence of two sub-classes of Class *ii* CC forms, named *α* and $β$ [\[21,31,32\].](#page-6-0) These *α* and *β* CC forms, after guest removal, are transformed in two NC forms also named *α* and *β*, respectively [\[20,21\]](#page-6-0). NC *α* and β -forms of PPO exhibit large diameter helices $[32]$ with slightly different chain conformations, with layer-line periodicity of 0.53 nm and of 0.55 nm, respectively [\[20,21\].](#page-6-0) Guest molecules that can induce crystallization in *α* or *β* CC forms of PPO (by sorption in amorphous films and by solution casting) were defined as *α* or *β*-guests, respectively. A list of *α* and *β*-guests of PPO is reported in [Table 1](#page-1-0) of Ref[.31](#page-6-0).

Thermal transitions between these two NC forms of PPO are not observed, because both NC forms remain stable up to their melting [\[33,34\].](#page-6-0) Sorption/desorption procedures of liquid α guests are instead able to induce transition [\[35\]](#page-6-0) from the lower melting NC β-form ($T_{m,\beta}$ ≈245 ◦C) toward the higher melting NC α-form (245 ◦C *< Tm,α <* 265 ◦C) [\[34\]](#page-6-0).

In this paper, an extended study on sorption of liquid α and β guests in PPO unoriented and oriented films exhibiting NC β and α phases, respectively, is reported. This study not only allows establishing suitable conditions for guest induced $\alpha \to \beta$ and $\beta \to \alpha$ transitions but also gives

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Table 1

A list of guest molecules that can induce formation of CC and NC β-forms by sorption in amorphous PPO films (β-guests). Sorption of liquid β-guests, being good PPO solvents, induce $\alpha \rightarrow \beta$ transition as well as loss of crystal phase orientations.

| β Guest molecules | α to β transition α form film immersion in the liquid at room temperature | Loss of crystal phase orientation | PPO solubility |
|---------------------------------------|---|--------------------------------------|-----------------------------|
| Dichloromethane | Film dissolution | | Yes at rt ^b |
| Dichloromethane/ acetone $(1/1)$ | Yes | Yes | |
| CHCl ₃ | Film dissolution | | Yes at rt ^b |
| CHCl ₃ /acetone (1/1) | Yes | Yes | |
| Benzene | Film dissolution | | Yes at rt ^b |
| Benzene/acetone (1/1) | Yes | Yes | |
| Perchloroethylene | Film dissolution | | Yes at rt ^b |
| Perchloroethylene/ acetone $(1/1)$ | Yes | Yes | |
| Dichloroethane | Film dissolution | | Yes at rt ^b |
| Dichloroethane/acetone (1/1) | Yes | Yes | |
| 2-Methyl | Yes | Yes | Yes at |
| tetrahydrofuran | | | $60-65$ °C |
| Ethyl benzene | Yes | Yes | Yes at |
| | | | 40-45 $°C$ |
| 1,4-Dioxane | Yes | Yes | Yes at |
| | | | 60-65 $°C$ |
| 1,2-dichloro propane | Yes | Yes | Yes at |
| | | | $60-65$ °C |
| Methyl phenyl ether | Yes | Yes | Yes at |
| | | | 50-55 $°C$ |
| Methyl ethyl ketone | No ^a | No ^a | No for T \leq |
| | | | $T_{\rm h}^{\rm c}$ |
| Acetone | No ^a | No ^a | No for T \leq |
| | | | T _b ^c |
| Hexanal | No | No | No for T \leq |
| | | | $T_{\rm b}^{\rm c}$ |
| Ethyl acetate | No | No | No for T \leq |
| | | | $T_{\rm b}^{\rm c}$ |
| Diethyl ether | No | No | No for T \leq |
| | | | T_b^c |

^a No at room temperature and yes at 60 °C.
^b rt is room temperature.
c T_b is the boiling temperature.

relevant information on their mechanism.

2. Experimental section

PPO (Ultra High P6130 grade; $M_w = 350 \text{ kg mol}^{-1}$) was kindly supplied by Sabic, the Netherlands. All used solvents (of Tables 1 and 2) were purchased from Sigma Aldrich and used without further

Table 2

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processing.

All NC PPO films were prepared by guest extraction by sorption/ desorption of acetonitrile from CC films. Unoriented CC α form films were prepared by casting at room temperature of 0.5–1 wt% PPO solutions in trichloroethylene. CC α form films with $c_$ orientation [36] were prepared by guest induced crystallization of amorphous PPO films, by immersion in liquid limonene for 12 h at room temperature. CC α form films with axial orientation were prepared by limonene induced crystallization (immersion in liquid, for 3 h at room temperature) of axially stretched amorphous PPO films. Amorphous films with a thickness of nearly 150 μm were stretched at 220 °C up to draw ratio of nearly 3, at a stretching rate of 10 mm/min. Amorphous PPO films, were obtained by casting at 60 °C of 0.5–1.0 wt% polymer solutions in CHCl₃. CC β form films with *c*// orientation [\[36\]](#page-6-0) were prepared by casting at room temperature of 0.5–1.0 wt% polymer solutions in benzene.

Sorption of β guests in α -form films as well as sorption of α guests in β-form films were conducted by immersion in pure liquids, at room temperature or at 60 ◦C, for 10–240 min. For some guests being very strong solvents for PPO, also test with 50/50 vol/vol guest/acetone solutions were used, to avoid film dissolution.

To study kind and degree of planar orientation in the films, twodimensional wide angle X-ray diffraction (2D-WAXD) patterns were obtained by a D8 QUEST Bruker diffractometer (CuKα radiation), by sending the X-ray beam parallel to the film surface (EDGE patterns). For films with orientation of the chain axes of the crystalline phases preferentially parallel and perpendicular to the film plane (named $c_{//}$ and c_{\perp} orientations, respectively), the degree of orientation $\chi_{(001)}$, was formalized on a quantitative numerical basis using the usual Hermans orientation function:

$$
\chi_{(001)} = \left(3\overline{\cos^2 y} - 1\right)\bigg/2\tag{1}
$$

where $cos^2\gamma$ was experimentally evaluated by intensity of the azimuthal distribution of 001 reflection. In these assumptions, when $\chi_{(001)}$ is equal to 0, there is a random crystallite orientation while when $\chi_{(001)}$ is equal to -0.5 or $+1$, **c** axes of all crystallites are perfectly perpendicular or parallel to the film surface, respectively.

Fourier Transform Infrared Spectroscopy (FTIR) was obtained at 2.0 cm^{-1} resolution with a Vertex 70 Bruker spectrophotometer. It is equipped with a deuterated triglycine sulfate (DTGS) detector and Ge/ KBr beam splitter. For polarized infrared spectra, it is fitted with SPE-CAC 12000 wire grid polarizer. 32 scans were averaged to reduce the noise level.

For the kinetics study of [Fig. 7,](#page-4-0) c_\perp NC α PPO films with a thickness of nearly 50 µm were immersed in liquid THF at room temperature for 0–5 min. These films were removed from liquid THF at different times and then rapidly dipped in ACN. This step was needed to quench further crystallization.

Differential scanning calorimetry (DSC, TA Q2000 equipment) was

^a Other α guests, like toluene, o-xylene, chlorobenzene, 1,2-dichloro benzene and 1,1,2-trichloroethylene lead to complete film dissolution at rt. ^b No at rt and Yes at 60 °C.

used to evaluate melting enthalpies of the NC PPO films. The heating/ cooling scans were conducted under the control program, at a rate of 10 \degree C/min. Degree of crystallinity (X_c) of NC PPO samples was determined based on the following formula:

$$
X_c = \frac{\Delta H_m}{\Delta H^{\circ}_m} \tag{2}
$$

where $\Delta H_{\rm m}$ is the melting enthalpy and $\Delta H^{\circ}{}_{\rm m}$ is the melting enthalpy of a fully crystalline α -form PPO sample (42 \pm 3 J g⁻¹) [\[29\]](#page-6-0).

3. Results and discussion

3.1. α → *β transitions as induced by β*-guests

A WAXD pattern of an unoriented NC α-form PPO film is shown in Fig. 1a and exhibits the typical 100 and 010 peaks located at $2\theta_{\text{CuK}\alpha}$ = 4.5◦ and 7.2◦[\[20\]](#page-6-0), respectively. The WAXD pattern of this film after sorption at room temperature of a liquid β-guest (THF) followed by guest removal by sorption/desorption of acetonitrile, is shown in Fig. 1b. This pattern shows hk0 peaks located at $2\theta_{\text{CuK}\alpha} = 5.2^\circ$ and 7.7°, typical of the NC β form [\[20\].](#page-6-0) The occurrence of a THF induced $\alpha \rightarrow \beta$ transition is confirmed by the corresponding FTIR spectra reported in Fig. 1a' and 1b'. In fact, the untreated film shows the typical peaks of the α form (e. g., at 773 cm^{-1} and 414 cm^{-1}) while the THF treated film shows the typical peaks of the β form (e.g., at 777 cm^{-1} and 418 cm^{-1}).

The THF induced $\alpha \rightarrow \beta$ transition also occurs for oriented α form films, both exhibiting axial orientation (2D pattern of Fig. 2a) as well as planar orientations (as shown for *c*⊥ orientation in the 2D pattern of [Fig. 3a](#page-3-0)). This $\alpha \to \beta$ transition is clearly shown by the angles of the observed diffraction peaks, more clearly apparent in the equatorial profiles of Fig. 2a',b' and [3a](#page-3-0)',b'. Beside $\alpha \rightarrow \beta$ transitions, the 2D patterns of Figs. 2 and 3 also show complete loss of crystal phase orienta-tion. In fact, the diffraction arcs of Fig. 2a and [3](#page-3-0)a, after $\alpha \rightarrow \beta$ transitions, are all replaced by uniform diffraction rings (Fig. 2b and [3](#page-3-0)b).

The occurrence of a THF induced $\alpha \rightarrow \beta$ transition in oriented films is confirmed by the corresponding FTIR spectra reported in [Fig. 4.](#page-3-0) In fact, the untreated oriented films show the typical peaks of the α form (e.g., at 773 cm^{-1} and 414 cm^{-1}) while the corresponding THF treated films shows the typical peaks of the β form (e.g., at 777 $\rm cm^{-1}$ and 418 $\rm cm^{-1}$).

Polarized FTIR spectra of an axially stretched NC α PPO film and of the derived NC β PPO film, are shown (for the of 790–730 cm⁻¹ range) in [Fig. 5](#page-4-0)a and 5b, respectively. It is apparent that the crystalline α peak at

Fig. 2. 2D WAXD patterns of a NC α-form film with axial orientation before (a) and after sorption of liquid THF (b) at room temperature, showing complete loss of orientation. The corresponding equatorial profiles shown in (a') and (b') indicate the transition from the NC α-form to the NC β form.

773 cm⁻¹ is highly dichroic [\(Fig. 5](#page-4-0)a) whereas the crystalline β peak at 777 cm⁻¹, observed after α→β transition as induced by liquid THF sorption, does not exhibit linear dichroism ([Fig. 5b](#page-4-0)). This confirms that the $\alpha \rightarrow \beta$ transition occurs with complete loss of crystalline phase orientation.

Behaviors strictly like that one observed for THF sorption ($\alpha \rightarrow \beta$) transition associated with complete loss of crystal phase orientation, Figs. 1-5) are observed for other β-guests, being able to dissolve PPO, which are listed in the upper part of [Table 1.](#page-1-0) β-guests being poor PPO solvents, which are listed in the lower part of [Table 1,](#page-1-0) on the contrary lead to CC α forms (rather than CC β forms), with maintenance of crystal phase orientation. For the last three guests of [Table 1](#page-1-0), the formation of CC α forms and maintenance of their possible crystal phase orientations, even occurs for their sorption at 60 ◦C.

It is worth adding that, the first five β-guests of [Table 1](#page-1-0) are very strong solvents and fully dissolve α-form PPO films. To study the

Fig. 1. WAXD patterns (a,b) and FTIR spectra (a',b') of an unoriented NC α-form PPO film, before (a,a') and after (b,b') temporary sorption of THF (β-guest). Both WAXD patterns and FTIR spectra indicate the transition to the NC β-form.

Fig. 3. 2D WAXD EDGE patterns of a NC α-form film with *c*⊥ orientation before (a) and after sorption of liquid THF (b) at room temperature, showing complete loss of orientation. The corresponding meridional profiles shown in (a') and (b') indicate the transition from the NC α-form to the NC β form.

possible ability of these β-guests to induce $\alpha \rightarrow \beta$ transition, tests were conducted by using 50/50 by volume solutions with acetone, i.e., with a β-guest which is a poor PPO solvent (lower part of [Table 1\)](#page-1-0). As reported in the upper part of [Table 1,](#page-1-0) the five considered $β$ -guest/acetone 50/50 by volume solutions induce $\alpha \rightarrow \beta$ transition, again associated with complete loss of crystal phase orientations. Strictly similar phenomena are observed for β-guest/acetone solutions, with 75/25 vol ratio.

3.2. β → *α transitions as induced by α-guests*

 $\beta \rightarrow \alpha$ transitions were described in a previous paper for two α guests: dibenzyl ether and limonene (by sorption at 60 $^{\circ}$ C) [\[35\].](#page-6-0) In this paper we extend our analysis to other α guests (listed in [Table 2\)](#page-1-0), by also conducting liquid sorption tests on β form films exhibiting crystal phase orientations. Our analysis was mainly conducted on NC β PPO films exhibiting the $c_{//}$ orientation, i.e., with the crystalline chain axes

preferentially parallel to the film plane, which are generally obtained by solution casting with $β$ guests [\[36\]](#page-6-0).

WAXD EDGE patterns of a NC β-form film with c_{ℓ} orientation before and after sorption of a liquid α -guest (CCl₄) are shown in [Fig. 6a](#page-4-0) and 6b, respectively. The hk0 diffraction rings being polarized on the equator and the 001 peak being polarized on the meridian of the pattern of [Fig. 6](#page-4-0)a, clearly indicate the presence of *c*// orientation. In particular, the intensity distribution for the 001 peak allows evaluating a degree of orientation $\chi_{(001)} \approx 0.3$. The presence of only uniform diffraction rings in [Fig. 6](#page-4-0)b clearly indicates that also $β \rightarrow α$ transition occurs with complete loss of crystal phase orientation. The corresponding equatorial profiles, shown in [Fig. 6](#page-4-0)a' and 6b', clearly show the occurrence of a β \rightarrow α transition.

Behaviors like that one observed for CCl₄ sorption ($\beta \rightarrow \alpha$ transition associated with complete loss of crystal phase orientation, [Fig. 6\)](#page-4-0) are observed for other β-guests, being able to dissolve PPO, which are listed in the upper part of [Table 2](#page-1-0). Sorption of liquid α-guests being poor PPO solvents, which are listed in the lower part of [Table 2,](#page-1-0) on the contrary lead to CC β forms (rather than CC α forms), and to NC β form after guest removal, with maintenance of crystal phase orientation.

It is worth adding that also the α guests being poor PPO solvents (lower part of [Table 2\)](#page-1-0) can induce $\beta \rightarrow \alpha$ transition associated with complete loss of crystal phase orientation, if guest sorption is conducted at 60 ◦C.

3.3. Sorption of guests leading to CC phases with maintenance of helical conformation and of crystalline phase orientations

As described in the two previous sections, sorption of some guests being poor solvents of PPO (lower rows in [Tables 1 and 2](#page-1-0)) can lead to CC phases, without inducing $\alpha \to \beta$ or $\beta \to \alpha$ transitions and with maintenance of crystalline phase orientations.

Also sorption of liquid α and β guests by NC α and β form films leads to CC α and β forms, respectively, with maintenance of the possibly present crystalline orientation. In fact, axial orientation as well as both kinds of planar orientations (c_{\perp} and $c_{//}$) [\[36\]](#page-6-0) remain essentially unaltered [\[35\].](#page-6-0) Analogous polymorphic behavior was observed for transitions between PPO CC forms of Class *i* (i.e., of CC forms with decalin, tetralin and α-pinene). In fact, both the observed uniplanar orientations are fully retained after guest exchange procedures [\[37\].](#page-6-0)

It is also interesting the polymorphic behavior observed for two

Fig. 4. FTIR spectra of NC α-form films with c⊥ orientation (a, a') or with axial orientation (b,b'), before (a,b; black spectra) and after sorption (a',b'; red spectra) of liquid THF.

Fig. 5. Polarized FTIR spectra (in the range of 790 cm^{-1} to 730 cm^{-1}) of an axially oriented NC α form film before (a) and after sorption of liquid THF guest at room temperature (b). Green line spectra refer to unpolarized light, while red and blue lines spectra were collected with polarized light being parallel and perpendicular to the film stretching direction, respectively. The high dichroism of the crystalline α peak at 773 cm⁻¹ and the absence of dichroism for the crystalline β peak at 777 cm⁻¹, are clearly apparent.

lead to CC α phase by solution casting at 60 °C. Immersion of NC α and β form films in both guests at rt leads to CC α and β form films, respectively, with maintenance of helical conformation as well as of crystalline phase orientation. The maintenance of the crystalline helical conformation as well as of their packing (α or β) can be explained by the similar thermodynamic stability of CC α and β form crystallites, for these two specific guests.

3.4. Mechanism of α → *β and* $β$ → *α transitions*

Data reported in the previous sections show that transitions between the two PPO phases (both $\alpha \to \beta$ and $\beta \to \alpha$) are always associated with complete loss of crystalline phase orientations, which are possibly present in the films. These observations rule out the possibility that $\alpha \rightarrow \beta$ and $\beta \rightarrow \alpha$ could be solid–solid phase transitions.

The reported data also show that $\alpha \rightarrow \beta$ and $\beta \rightarrow \alpha$ transitions occur by sorption of β and α guests, respectively, only when they are good PPO solvents. This indicates transition mechanisms that involve dissolution of a NC phase followed by formation of a host/guest CC phase with change of chain conformation. In this respect, it is worth noting that both α and β forms (NC as well as CC) exhibit large helices, being suitable for intrahelical guest inclusion, but with different helix diameters (larger for α form than for β form) [\[32\]](#page-6-0).

To get additional information on the mechanism of these transitions, kinetics were studied by FTIR. Just an example, variations of the FTIR spectra in the range 785–745 $\rm cm^{-1}$, as a consequence of THF sorption at room temperature in a α form PPO film having a thickness of nearly 50 $μ$ m, are shown in Fig. 7. A change of α form to β form is clearly shown by the gradual shift of the 773 cm^{-1} peak up to 777 cm^{-1} (already complete after 1 min of guest sorption), which is mainly due to the change of chain conformation, leading to a reduction of the crystalline polymer helix diameter. Correspondingly the intensity of the crystalline peak at 756–754 cm^{-1} , and hence the degree of crystallinity, remains essentially

Fig. 6. 2D WAXD EDGE patterns of a NC β-form film with c// orientation before (a) and after sorption of a liquid α -guest (CCl₄) (b), showing the loss of orientation ($\chi_{(0\,0\,1)}$ decreases from 0.3 to zero). The corresponding equatorial profiles shown in (a') and (b') indicate the transition from the NC β-form to the NC α form.

specific guests (methylbenzoate and 1,3,5-trichlorobenzene), which are good PPO solvents but cannot be defined neither α nor β guests. In fact, both guests lead to CC β phase by amorphous film immersion at rt while

Fig. 7. FTIR spectra of a NC α -form film (50 μ m thick) after different times of sorption of liquid THF at room temperature: 0 min (black line), 0.1 min (red line), 0.4 min (blue line), 1 min (green line), and 5 min (pink line). A complete transformation from α form to β form (as pointed out by the peak shift from 773 cm^{-1} up to 777 cm^{-1}) occurs, without any significant change of degree of crystallinity (constant intensity of the 756–754 cm^{-1} peak).

unaltered, for the entire transition kinetics. The invariance of degree of crystallinity is confirmed by the unaltered intensity of other crystalline peaks (like, e.g., those observed at 495 cm^{-1} and 414 cm^{-1}).

The maintenance of the degree of crystallinity is further confirmed by evaluating melting enthalpies (ΔH_m) of NC PPO films before and after phase transition, by DSC scans like those shown in Fig. S2. Δ*H*m of NC PPO films after different times of sorption of liquid THF are all similar, within a narrow range $(20.0-21.5 \text{ J/g})$. Correspondingly, the degree of crystallinity of NC PPO films is in the range 47.5–51.0 %.

Gradual crystal phase transitions, with poor variation of degree of crystallinity, can be rationalized by recrystallization kinetics faster than kinetics of advancement of the guest front in thick PPO films.

Hence, the observed phase transition kinetics, with maintenance of degree of crystallinity for the entire recrystallization time and with complete loss of orientation of the crystalline phase, can be rationalized by local dissolution of NC α or β phases (controlled by kinetics of guest diffusion in the polymer film) and faster local recrystallizations leading to CC β or α phases, which after guest removal lead to NC β or α phases, respectively.

It is worth adding that the complete loss of orientation, as observed for both $\alpha \to \beta$ and $\beta \to \alpha$ transitions, fully confirms complete separation between the two NC phases (as suggested by more recent structural data) [\[20\]](#page-6-0) while rules out the previous hypothesis [\[19\]](#page-6-0) of occurrence of many (if not a continuum of) modifications between two limit ones.

A comparison with the polymorphic behavior of syndiotactic polystyrene (sPS), i.e. the only other polymer being able to form NC phases, is also informative. For sPS, differently from PPO, crystalline phase orientations are maintained for all crystalline phase transitions involving CC and NC phases. In fact, axial, uniplanar [\[38,39\]](#page-6-0) and uniplanar-axial $[40]$ orientations of the NC and CC δ form $[41]$, can be maintained not only for $NC \rightarrow CC$ and $CC \rightarrow NC$ transitions but also for transitions toward CC and NC ε phases $[42, 43]$ as well as toward the dense γ phase [\[38\].](#page-6-0) Crystalline phase orientations are also maintained after guest-exchange procedures [\[39\]](#page-6-0), which transform a CC phase with a given guest into a CC phase with a different guest [\[44](#page-6-0)–46]. The maintenance of crystalline phase orientation, for crystal phase transitions of sPS, can be rationalized by the presence for all these crystalline forms (NC and CC δ, NC and CC ε, and γ) of the same helical confor-mation (s(2/1)2) [\[41,42,47\]](#page-6-0). The opposite behavior observed for $\alpha \rightarrow \beta$ and $\beta \rightarrow \alpha$ transitions of PPO, i.e. complete loss of crystalline phase orientation, can be rationalized by the presence for α and β phases of two different helical conformations.

4. Conclusions

An extended study on sorption of liquid guests, in PPO films exhibiting NC α and β phases, is reported. Sorption of some α and β guests in NC β and α-form PPO samples induces formation of CC α and β forms (and of the corresponding NC α and β forms, after guest removal), respectively. Sorption of other guests in NC PPO samples instead leads to their inclusion in the available NC form, thus inducing formation of the corresponding CC form.

When sorption of liquid guests is by oriented samples (e.g., films with axial or planar orientations), maintenance of crystalline form occurs with maintenance of crystalline phase orientation while $\alpha \rightarrow \beta$ and $\beta \rightarrow \alpha$ transitions occur with complete loss of crystalline phase orientation. These observations rule out the possibility that $\alpha \rightarrow \beta$ and $\beta \rightarrow \alpha$ could be solid–solid phase transitions.

Collected data for many guest molecules show that transitions between the two phases (both $\alpha \rightarrow \beta$ and $\beta \rightarrow \alpha$), and the associated loss of crystalline phase orientation, occur only for guests being good PPO solvents. Moreover, these transitions occur with maintenance of degree of crystallinity, for the entire transition kinetics. This indicates the occurrence of only local dissolution of a NC α or β phases and local recrystallizations leading to CC β or α phases (which after guest removal lead to NC $β$ or $α$ phases), respectively.

Moreover, the complete loss of crystal phase orientation, as observed for both $\alpha \to \beta$ and for $\beta \to \alpha$ transitions, fully confirms complete separation between the two NC phases while rules out the previous hypothesis of occurrence of a continuum of modifications between two limit ones.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

Data will be made available on request.

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Appendix A. Supplementary material

Supplementary data to this article can be found online at [https://doi.](https://doi.org/10.1016/j.eurpolymj.2023.112532) [org/10.1016/j.eurpolymj.2023.112532](https://doi.org/10.1016/j.eurpolymj.2023.112532).

References

- [1] [G.D. Staffin, C.C. Price, I.X. Polyethers, Poly-\(2, 6-dimethyl-1, 4-phenylene oxide\),](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0005) [J. Am. Chem. Soc. 82 \(1960\) 3632](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0005)–3634.
- [2] F.E. Karasz, J.M. O'Reilly, Thermal properties of poly(2,6-dimethyl phenylene ether), J. Polym. Sci. B 3 (1965) 561–563, [https://doi.org/10.1002/](https://doi.org/10.1002/pol.1965.110030708) [pol.1965.110030708](https://doi.org/10.1002/pol.1965.110030708).
- [3] [K. Toi, G. Morel, D.R. Paul, Gas sorption and transport in poly \(phenylene oxide\)](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0015) [and comparisons with other glassy polymers, J. Appl. Polym. Sci. 27 \(1982\)](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0015) 2997–[3005.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0015)
- [4] [G. Guerra, C. De Rosa, V.M. Vitagliano, V. Petraccone, P. Corradini, Effects of](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0020) [blending on the polymorphic behavior of melt-crystallized syndiotactic](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0020) [polystyrene, J. Polym. Sci. B Polym. Phys. 29 \(1991\) 265](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0020)–271.
- [5] [K.C. Khulbe, T. Matsuura, G. Lamarche, H.J. Kim, The morphology characterisation](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0025) [and performance of dense PPO membranes for gas separation, J. Memb. Sci. 135](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0025) [\(1997\) 211](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0025)–223.
- [6] [O.M. Ilinitch, V.B. Fenelonov, A.A. Lapkin, L.G. Okkel, V.V. Terskikh, K.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0030) [I. Zamaraev, Intrinsic microporosity and gas transport in polyphenylene oxide](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0030) [polymers, Micropor. Mesoporous Mater. 31 \(1999\) 97](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0030)–110.
- [7] [J. Pak, M. Pyda, B. Wunderlich, Rigid amorphous fractions and glass transitions in](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0035) [poly \(oxy-2, 6-dimethyl-1, 4-phenylene\), Macromolecules 36 \(2003\) 495](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0035)–499.
- [8] M. Khayet, J.P.G. Villaluenga, J.L. Valentin, M.A. López-Manchado, J.I. Mengual, [B. Seoane, Filled poly \(2, 6-dimethyl-1, 4-phenylene oxide\) dense membranes by](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0040) [silica and silane modified silica nanoparticles: characterization and application in](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0040) [pervaporation, Polymer 46 \(2005\) 9881](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0040)–9891.
- [9] [T. Xu, D. Wu, L. Wu, Poly \(2, 6-dimethyl-1, 4-phenylene oxide\)\(PPO\)](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0045)—a versatile [starting polymer for proton conductive membranes \(PCMs\), Prog. Polym. Sci. 33](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0045) [\(2008\) 894](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0045)–915.
- [10] D. Zhou, H. Huth, Y. Gao, G. Xue, C. Schick, Calorimetric glass transition of poly (2, [6-dimethyl-1, 5-phenylene oxide\) thin films, Macromolecules 41 \(2008\)](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0050) [7662](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0050)–7666.
- [11] [R. Rea, S. Ligi, M. Christian, V. Morandi, M. Giacinti Baschetti, M.G. De Angelis,](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0055) [Permeability and selectivity of PPO/graphene composites as mixed matrix](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0055) membranes for $CO₂$ capture and gas separation, Polymers 10 (2018) 129.
- [12] [J. Hurek, E. Turska, X-ray studies of crystallization of poly \(2, 6-dimethyl-1, 4](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0060) henylene oxide) on swelling in certain systems, Acta Polym. 35 (1984) 201-207.
- [13] [O.M. Ilinitch, G.L. Semin, M.V. Chertova, K.I. Zamaraev, Novel polymeric](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0065) [membranes for separation of hydrocarbons, J. Memb. Sci. 66 \(1992\) 1](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0065)–8.
- [14] [A. Alentiev, E. Drioli, M. Gokzhaev, G. Golemme, O. Ilinich, A. Lapkin, V. Volkov,](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0070) [Y. Yampolskii, Gas permeation properties of phenylene oxide polymers, J. Membr.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0070) [Sci. 138 \(1998\) 99](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0070)–107.
- [15] [K.C. Khulbe, T. Matsuura, G. Lamarche, A.-M. Lamarche, X-ray diffraction analysis](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0075) [of dense PPO membranes, J. Membr. Sci. 170 \(2000\) 81](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0075)–89.
- [16] [J.M. Barrales-Rienda, J.M.G. Fatou, Single crystals of poly \(2, 6-dimethyl-1, 4](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0080) [phenylene\) oxide, Kolloid-Zeitschrift Und Zeitschrift Für Polym. 244 \(1971\)](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0080) 317–[323.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0080)
- [17] [S. Horikiri, Single crystals of poly \(2, 6-dimethylphenylene oxide\), J. Polym. Sci. 10](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0085) [\(Part A-2\) \(1972\) 1167](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0085)–1170.

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European Polymer Journal 200 (2023) 112532

- [18] [O. Tarallo, V. Petraccone, C. Daniel, G. Fasano, P. Rizzo, G. Guerra, A chiral co](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0090)[crystalline form of poly \(2, 6-dimethyl-1, 4-phenylene\) oxide \(PPO\), J. Mater.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0090) [Chem. 22 \(2012\) 11672](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0090)–11680.
- [19] [C. Daniel, S. Longo, G. Fasano, J.G. Vitillo, G. Guerra, Nanoporous crystalline](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0095) [phases of poly \(2, 6-dimethyl-1, 4-phenylene\) oxide, Chem. Mater. 23 \(2011\)](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0095) [3195](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0095)–3200.
- [20] [B. Nagendra, A. Cozzolino, C. Daniel, P. Rizzo, G. Guerra, F. Auriemma, C. De Rosa,](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0100) M.C. D'[Alterio, O. Tarallo, A. Nuzzo, Two nanoporous crystalline forms of poly \(2,](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0100) [6-dimethyl-1, 4-phenylene\) oxide and related Co-crystalline forms,](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0100) [Macromolecules 52 \(2019\) 9646](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0100)–9656.
- [21] [F. Auriemma, C. Daniel, M. Golla, B. Nagendra, P. Rizzo, O. Tarallo, G. Guerra,](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0105) [Polymorphism of poly \(2, 6-dimethyl-1, 4-phenylene\) oxide \(PPO\): co-crystalline](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0105) [and nanoporous-crystalline phases, Polymer 125290 \(2022\)](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0105).
- [22] [M. Galizia, C. Daniel, G. Fasano, G. Guerra, G. Mensitieri, Gas sorption and](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0110) [diffusion in amorphous and semicrystalline nanoporous poly \(2, 6-dimethyl-1, 4](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0110) [phenylene\) oxide, Macromolecules 45 \(2012\) 3604](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0110)–3615.
- [23] [C. Daniel, M. Pellegrino, V. Venditto, S. Aurucci, G. Guerra, Nanoporous-crystalline](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0115) [poly \(2, 6-dimethyl-1, 4-phenylene\) oxide \(PPO\) aerogels, Polymer 105 \(2016\)](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0115) $96 - 103$ $96 - 103$.
- [24] [C. Daniel, P. Rizzo, B. Nagendra, A. Cozzolino, G. Guerra, High diffusivity dense](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0120) [films of a nanoporous-crystalline polymer, Polymer 229 \(2021\), 124005](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0120).
- [25] [A. Cozzolino, B. Nagendra, P. Rizzo, C. Daniel, G. Guerra, Fast uptake of organic](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0125) [pollutants from dilute aqueous solutions by nanoporous-crystalline PPO films with](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0125) [c-perpendicular orientation, Eur. Polym. J. 161 \(2021\), 110864.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0125)
- [26] [A.Y. Alentiev, I.S. Levin, M.I. Buzin, N.A. Belov, R.Y. Nikiforov, S.V. Chirkov, I.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0130) [V. Blagodatskikh, A.S. Kechekyan, P.A. Kechekyan, V.G. Bekeshev, Gas transport](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0130) [parameters, density and free volume of nanocrystalline poly-2, 6-dimethylpheny](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0130)[lene oxide, Polymer 226 \(2021\), 123804](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0130).
- [27] [A.Y. Alentiev, I.S. Levin, N.A. Belov, R.Y. Nikiforov, S.V. Chirkov, D.A. Bezgin, V.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0135) [E. Ryzhikh, J.V. Kostina, V.P. Shantarovich, L.Y. Grunin, Features of the gas](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0135)[permeable crystalline phase of poly-2, 6-dimethylphenylene oxide, Polymers 14](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0135) (2021) 120.
- [28] [A.Y. Alentiev, S.V. Chirkov, R.Y. Nikiforov, I.A. Levin, A.S. Kechekyan, P.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0140) [A. Kechekyan, N.A. Belov, Poly \(2, 6-Dimethyl-1, 4-Phenylene Oxide\) as a Polymer-](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0140)[Polymer Nanocomposite: Mechanical and Gas Transport Characteristics, Membr.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0140) [Membr. Technol. 4 \(2022\) 1](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0140)–10.
- [29] [B. Nagendra, A. Cozzolino, C. Daniel, P. Rizzo, G. Guerra, High surface area](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0145) [nanoporous-crystalline polymer films, Macromolecules 55 \(2022\) 2983](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0145)–2990.
- [30] [B. Nagendra, S. Salman, H. Peter Bloch, C. Daniel, P. Rizzo, R. Fittipaldi, G. Guerra,](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0150) [Poly \(phenylene oxide\) Films with Hydrophilic Sulfonated Amorphous Phase and](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0150) [Physically Cross-Linking Hydrophobic Crystalline Phase, ACS Appl. Polym. Mater.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0150) [\(2023\)](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0150).
- [31] [M. Golla, A. Cozzolino, B. Nagendra, E. Vignola, C. Daniel, P. Rizzo, G. Guerra,](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0155) F. Auriemma, M.C. D'[Alterio, Molecular features behind formation of](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0155) α or β Co[crystalline and nanoporous-crystalline phases of PPO, Front. Chem. 9 \(2022\),](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0155) [809850.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0155)
- [32] [M. Golla, B. Nagendra, C. Daniel, P. Rizzo, F. Auriemma, O. Tarallo, G. Guerra,](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0160) [Intrahelical empty channels of nanoporous-crystalline](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0160) α and β forms of PPO, Eur. [Polym. J. 179 \(2022\), 111568](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0160).
- [33] [C. Daniel, D. Zhovner, G. Guerra, Thermal stability of nanoporous crystalline and](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0165) [amorphous phases of poly \(2, 6-dimethyl-1, 4-phenylene\) oxide, Macromolecules](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0165) [46 \(2013\) 449](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0165)–454.
- [34] [B. Nagendra, M. Golla, C. Daniel, P. Rizzo, G. Guerra, Melting of nanoporous](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0170)[crystalline and co-crystalline solution cast films of poly \(2, 6-dimethyl-1, 4](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0170) [phenylene\) oxide, Polymer 228 \(2021\), 123935](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0170).
- [35] [B. Nagendra, A. Cozzolino, M. Golla, C. Daniel, P. Rizzo, G. Guerra, Guest induced](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0175) transition from β to α [nanoporous crystalline forms of PPO, Polymer 187 \(2020\),](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0175) [122083.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0175)
- [36] [P. Rizzo, C. Gallo, V. Vitale, O. Tarallo, G. Guerra, Nanoporous-crystalline films of](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0180) [PPO with parallel and perpendicular polymer chain orientations, Polymer 167](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0180) [\(2019\) 193](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0180)–201.
- [37] [P. Rizzo, G. Ianniello, S. Longo, G. Guerra, Uniplanar orientations and guest](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0185) [exchange in PPO cocrystalline films, Macromolecules 46 \(2013\) 3995](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0185)–4001.
- [38] [A.R. Albunia, P. Rizzo, G. Guerra, Polymeric films with three different orientations](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0190) [of crystalline-phase empty channels, Chem. Mater. 21 \(2009\) 3370](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0190)–3375.
- [39] [P. Rizzo, A. Spatola, A. De Girolamo Del, G.G. Mauro, Polymeric films with three](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0195) [different uniplanar crystalline phase orientations, Macromolecules 38 \(2005\)](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0195) [10089](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0195)–10094.
- [40] [A.R. Albunia, P. Rizzo, G. Guerra, Two Different Uniplanar-Axial Orientations of](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0200) [Syndiotactic Polystyrene Films, Macromolecules 44 \(2011\) 5671](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0200)–5681.
- [41] [C. De Rosa, G. Guerra, V. Petraccone, B. Pirozzi, Crystal structure of the emptied](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0205) clathrate form (δ[e form\) of syndiotactic polystyrene, Macromolecules 30 \(1997\)](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0205) [4147](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0205)–4152.
- [42] [V. Petraccone, O. Ruiz de Ballesteros, O. Tarallo, P. Rizzo, G. Guerra, Nanoporous](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0210) [polymer crystals with cavities and channels, Chem. Mater. 20 \(2008\) 3663](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0210)–3668.
- [43] [T. Sano, H. Ebihara, S. Sano, T. Okabe, H. Itagaki, The ways of connecting](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0215) [crystalline phases having tubular cavities like stringing beads: New conductive](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0215) [polymer composites prepared by the polymerization of aniline in highly oriented](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0215) ε [crystalline phase of syndiotactic polystyrene, Eur. Polym. J. 138 \(2020\), 109975](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0215).
- [44] A. Yoshioka, K. Tashiro, Polymer− [solvent interactions in crystalline](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0220) δ form of [syndiotactic polystyrene viewed from the solvent-exchange process in the](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0220) δ form [and the solvent evaporation phenomenon in the thermally induced](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0220) $\delta-\gamma$ phase [transition, Macromolecules 36 \(2003\) 3593](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0220)–3600.
- [45] [Y. Uda, F. Kaneko, T. Kawaguchi, Selective guest uptake from solvent mixtures in](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0225) [the clathrate phase of syndiotactic polystyrene, Macromol. Rapid Commun. 25](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0225) [\(2004\) 1900](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0225)–1904.
- [46] [Y. Uda, F. Kaneko, T. Kawaguchi, Guest Exchange with n-Alkanes and Host](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0230)− Guest [Interactions in the Clathrate Phase of Syndiotactic Polystyrene, Macromolecules 38](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0230) [\(2005\) 3320](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0230)–3326.
- [47] [F.J. Torres, B. Civalleri, A. Meyer, P. Musto, A.R. Albunia, P. Rizzo, G. Guerra,](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0235) [Normal Vibrational Analysis of the Syndiotactic Polystyrene s\(2/1\)2 Helix, J. Phys.](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0235) [Chem. B 113 \(2009\) 5059](http://refhub.elsevier.com/S0014-3057(23)00715-2/h0235)–5071.