EXAFS Study of Rutile and Anatase

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Z. Naturforsch. 36a, 1192-1195 (1981); received August 5, 1981

Two crystallographic forms of TiO₂ (rutile and anatase) were subject to an extended x-ray absorption fine structure study at the K threshold of titanium. Data analysis was carried out by Fourier methods and curve fitting techniques. Using the theoretical phases, Ti-O bond distances of 1.93(1) Å were derived both for rutile and anatase, as compared to the crystallographic values of 1.948 - 1.980(1) Å (rutile) and 1.934 - 1.980(1) Å (anatase). The ratio of the cation coordination numbers $N_r/N_a = 1.07$ is close to the theoretical value.

Introduction

As part of a program of characterization of high yield Ziegler-Natta catalysts, we have undertaken extended x-ray absorption fine structure (EXAFS) studies of various titanium compounds with an allchlorine or all-oxygen first coordination shell. In a preceeding paper [1] we have described our results for some titanium chlorides; here we present our results for a Ti-O first shell compound, namely TiO_2 in the polymorphic forms of rutile and anatase.

Experimental

Materials: Rutile and anatase were commercial SIBIT products (impurity level < 0.5%). After admixture of chemically inert and x-ray transparent BN, about 0.1 g of the powder was pressed in tabular form $(1 \times 2 \text{ cm})$. The TiO₂/diluent ratio was such that $\mu x \approx 2$ (including the absorption due to BN) in order to optimize contrast at the high-absorption side of the Ti K edge. The tablets were then inserted in a stainless steel frame together with a 4μ thick titanium sheet (Goodfellow Ltd., 99.5%) which was used as a reference material to the energy scale.

Data Collection: EXAFS spectra were recorded at the Synchrotron Radiation Facility (PULS) at the National Laboratories (INFN), Frascati (Italy), using the x-ray beam line. The experimental ap-

Reprint requests to Dr. G. Vlaic, Istituto G. Donegani, Via Fauser 4, 28100 Novara/Italien. paratus has been described previously [2] and data collection followed our standard procedures [3]. All EXAFS spectra were taken in air at room temperature, at $\approx 1.5 \text{ eV}$ intervals over a scan range of about 1300 eV, extending up to 800 eV above the K threshold of Ti, measuring for 1 sec. each.

Data processing was carried out on a UNIVAC 1100/20 computer using a locally written program set with general graphical and mathematical routines due to B. Pianzola.

Data Evaluation: For reliable comparison of the results obtained from the two TiO₂ samples, the EXAFS spectra were extracted from the experimental absorption data μx using a uniform procedure (thus excluding bias due to the data handling [4]). At variance of the procedures of Ref. [1] and in view of the rather extended pre-edge absorption [5], the smooth absorption background $\mu' x$ was subtracted using a Victoreen fit [6]; the atomic-like contribution $\mu_0 x$ was obtained by means of a Fourier filtering technique [1, 7]. The EXAFS signal was then calculated as

$$\chi(k) = [\mu(k) x - \mu'(k) x - \mu_0(k) x]/\mu_0(k) x.$$

In converting the absorption data from the photon energy E_x to the wave vector of the outgoing electron

$$k = [(E_x - E_0) 2 m/\hbar^2]^{1/2}$$

the reference energy E_0 was fixed, with an arbitrary but uniform choice, at the inflection point of the *K*-edge of titanium (4983.8(2), 4983.1(2) and

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4964.5 eV for rutile, anatase and metallic titanium, respectively, with the latter being used for calibration of the energy scale [1]). Figs. 1 and 2 report the EXAFS signal $\chi(k)$ vs k for the two samples.

According to the single scattering theory, the EXAFS signal [8-10] is given by

$$\chi(k) = \frac{1}{k} \Sigma_i A_i(k) \sin\left[(2 k R_i + \Phi_i(k))\right],$$

where the sum is over *i* coordination shells, R_i is the distance of the *i*th shell from the absorber and $\Phi_i(k)$ the phase shift of the photoelectron caused by the potential of the absorbing and backscattering atoms. The amplitude envelope A_i is given by

$$A_{i}(k) = (N_{i}/R_{i}^{2}) \exp(-2 \sigma_{i}^{2} k^{2}) F_{i}(k),$$

where N_i is the number of atoms at distance R_i and σ_i^2 the Debye-Waller like term. In this expression the amplitude $F_i(k)$ is given by

$$F_{i}(k) = f_{i}(\pi, k) \exp\left(-2 R_{i}/\lambda\right),$$

where $f_i(\pi, k)$ is the backscattering factor [11] for each of the N_i neighbouring atoms in the *i*th shell



Fig. 1. EXAFS signal $\chi(k)$ vs k for rutile.



Fig. 2. EXAFS signal $\chi(k)$ vs k for anatase.

and λ is the mean free path of the photoelectrons [12].

The EXAFS signals, weighted by k for compensating amplitude reduction [12], were Fourier transformed from k to R space according to

$$FT(R) = \left(\frac{1}{2\pi}\right)^{1/2} \\ \cdot \int_{k_{\min}}^{k_{\max}} W(k) k \chi(k) \exp\left(-2ikR\right) dk$$

with $k_{\min} = 3.9$ and $k_{\max} = 13.8$, and W(k) being a gaussian window function [1]. As may be seen from Figs. 3 and 4, the modulus of FT(R) of anatase shows a strong peak in correspondence to the first shell and several weak higher shell contributions. On the other hand, |FT(R)| of rutile shows three peaks which may be attributed to the first Ti-O coordination sphere at about 1.96 Å, Ti-Ti distances at 2.959 Å and Ti-O contributions at about 3.50 Å.

The first peak of each radial distribution function was then backtransformed (FT^{-1}) in the



Fig. 3. Modulus of the Fourier transform for EXAFS signal of rutile.



Fig. 4. Modulus of the Fourier transform for EXAFS signal of anatase.

range 1.20-1.90 Å for rutile and 1.26-1.90 Å for anatase, according to

$$\chi_1(k) = \sqrt{\frac{2}{\pi}} k^{-1} W(k)^{-1} \\ \cdot \int_{R_1}^{R_2} FT(R) \exp(2ikR) dR.$$

Results and Discussion

Distance information was derived by first dividing the backtransformed signal $\chi_1(k)k$ by $|FT^{-1}|$ to obtain the sinusoidal term

$$\varphi_1(k,R) = \sin\left[2\,k\,R_1 + \Phi_1(k)\right]$$

and by subsequent least-squares fitting of $\varphi_1(k, R)$. The phase shift function $\Phi_1(k)$ was parametrized in the form

$$egin{aligned} \varPhi(k) &= a_0 + b_0 - \pi + (a_1 + b_1) \, k \ &+ (a_2 + b_2) \, k^2 + (a_3 + b_3) / k^3 \end{aligned}$$

according to [13]. During the fitting the phase parameters were constrained to the theoretical values [14] (Table 1). This restricts the variables to R_i and ΔE_0 , where the latter is required for the definition of

$$k' = (k^2 + 0.262467 \Delta E_0)^{1/2}$$

with k being defined as above and ΔE_0 corresponding to an energy shift with respect to a previously reported value [10, 13]. This fitting procedure is rapid and avoids strong correlations between the parameters [4], which may compromise their physical significance. The maximum number of independent parameters, N_{free} , in a fitting procedure has been given by Eisenberger [15] as

$$N_{\rm free} \approx 0.6 \Delta R \Delta k$$
,

where ΔR is the Fourier filter length $(R_2 - R_1)$ and Δk the interval in k space in the fit. The fitted functions for rutile and anatase are reported in Figs. 5 and 6 $(k_{\min} = 4.5, k_{\max} = 12.5)$. Comparison of the resulting parameter values with the crystallographic Ti-O distances [16, 17] shows excellent agreement (Table 2). The ratio of the titanium coordination numbers in the two samples has been

$a_0 + b_0 - \pi$	2.73440
$a_1 + b_1$	- 1.36929 Å
$a_2 + b_2$	$0.03393 \ { m \AA^2}$
$a_3 + b_3$	39.05040 Å-3

Table 1. Fitted theoretical absorber and backscatter phase parameters for Ti-O.



Fig. 5. Curve fitting analysis for rutile: fit (---) of the experimental signal $\varphi_1(k, R_1)$ vs. k (---).



Fig. 6. Curve fitting analysis for anatase: fit (---) of the experimental signal $\varphi_1(k, R_1)$ vs. k (---).



Fig. 7. Logarithmic ratio of $|FT^{-1}|$ for rutile and anatase vs k^2 (—) with fit (---) for the definition of the coordination number of anatase ($N_{\text{rutile}} = 6.0$).

Table 2. Results of fitting $\sin[2kR_1 + \Phi_1(k)]$.

Sample	$\Delta E_0 ({ m eV})$	R (Å)	Cryst. value (Å)	Ref.
Rutile	- 16.0	1.93 (1)	$\begin{array}{c} 1.948(1) \ (4\times) \\ 1.980(1) \ (2\times) \end{array}$	[16]
Anatase	-13.4	1.93 (1)	$\begin{array}{c} 1.934(1) \ (4\times) \\ 1.980(1) \ (2\times) \end{array}$	[17]

calculated according to the procedure proposed by Sayers et al. [18] from the moduli of FT^{-1} :

$$\frac{A_{\rm r}(k)}{A_{\rm a}(k)} = \frac{N_{\rm r}}{N_{\rm a}} \frac{R_{\rm a}^2}{R_{\rm r}^2} \exp\left[-2\left(\sigma_{\rm r}^2 - \sigma_{\rm a}^2\right)k^2\right].$$

The value of $N_{\rm a} = 6.45$ is obtained for $N_{\rm r} = 6$ by plotting

 $\ln(A_{\mathbf{r}}(k)/A_{\mathbf{a}}(k)) \operatorname{vs} k^2$.

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The discrepancy with respect to the theoretical value is within the error limits of the method [5] and derives probably from the strong overlap between the first shell in rutile with the background peak and second shell, which leads to distorsions in the A_r term. The calculated value of

$$\Delta \sigma^2 (= \sigma_{\mathbf{r}}^2 - \sigma_{\mathbf{a}}^2) = -0.0022 \text{ Å}^2$$

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