# New fluorescent polymeric nanocomposites synthesized by antimony dodecyl-mercaptide thermolysis in polymer

F. Capezzuto<sup>1</sup>, G. Carotenuto<sup>1\*</sup>, F. Antolini<sup>2</sup>, E. Burresi<sup>2</sup>, M. Palomba<sup>1</sup>, P. Perlo<sup>3</sup>

<sup>1</sup>Institute of Composite and Biomedical Materials, P.le Tecchio, 80 – 80125 Napoli, Italy <sup>2</sup>ENEA Research Center of Faenza, Dept. of Advanced Physical Technologies and New Materials Via Ravegnana 186 – 48018 Faenza (Ra), Italy

<sup>3</sup>FIAT Research Center, Strada Torino 50 – 10043 Orbassano (TO), Italy

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Abstract. In this work, the formation of semiconductive  $Sb_2S_3$  nanoparticles inside amorphous polystyrene has been achieved by thermal degradation of the corresponding antimony dodecyl-mercaptide,  $Sb(SC_{12}H_{25})_3$ . The thermolysis of the dodecyl-mercaptide precursor was studied as both pure phase and mercaptide solution in polystyrene. The thermal decomposition of the antimony mercaptide precursor at 350°C, under vacuum, showed the formation of a mixture of antimony trisulfide (stibnite,  $Sb_2S_3$ ) and zero-valent antimony (Sb) phase. X-ray Powder Diffraction (XRD) and Rietveld analysis carried out on the obtained nanostructured powder confirmed the presence of Sb and  $Sb_2S_3$  phases in 10.4 wt% and 89.6 wt% amount, respectively. The same pyrolysis reaction was carried out in the polymer and the resulting nanocomposite material was characterized by X-ray diffraction (XRD), transmission electron microscopy (TEM), UV-VIS spectroscopy, and fluorescence spectroscopy. The nanocomposite structural characterization indicated the presence of well-dispersed nanoclusters of antimony and stibnite (15–30 nm in size) inside the amorphous polymeric phase. Optical measurements on the obtained nanocomposite films showed a strong emission at 432 nm upon excitation at 371 nm, probably related to the presence of  $Sb_2S_3$  nanoclusters.

Keywords: nanocomposites, stibnite, antimony, mercaptide, fluorescence

## 1. Introduction

Polymer nanocomposites made of nanoparticles (spheres, rods, plates, etc.) embedded into a polymeric matrix extend the potentialities of polymers while maintaining the manufacturing and processing flexibility inherent to plastics [1–6]. In fact, the potential impact of polymeric nanocomposite materials lies in the synergistic combination of nanoparticles and polymers properties that can lead not only to the sum of the organic and inorganic phase characteristics, but also to new functionalities with respect to the single materials [6–9]. The advantages of these materials arise mainly from the filler size reduction to nanoscale, considering that the

As already said, it is well known that the embedding of nanoscale fillers into a polymeric matrix offers the possibility to achieve materials with properties combination not realizable by the single components. In particular, the combination of fluorescent semiconductor nanoparticles with optical polymers yields to a new class of materials characterised by both fluorescence and transparency characteristics. The luminescent properties of semicon-

concomitant quantum confinement effects induce new physical phenomena [10, 11]. In fact, nanoparticles have unique optical and electronic properties not observed in the corresponding bulk materials [12, 13].

<sup>\*</sup>Corresponding author, e-mail: giancaro@unina.it © BME-PT

ductor nanoparticles stem from the separation of the conductive and valence band due to the quantum-size effect. Nowadays, the luminescent properties of semiconductor nanoparticles are thoroughly studied for the development of new labelling systems for biomedical purposes [14] and for lightemitting devices applications [15–17]. Especially in the last case, the combination of nanoparticles and polymers is of special interest because the mechanical and electro-optical properties of polymers joined with the high stability of nanoparticles as light-emitters allow the set-up of industrially competitive systems for light-source production.

Antimony trisulfide, Sb<sub>2</sub>S<sub>3</sub> is an interesting material, with unique optical properties and currently under study as potential photovoltaic material [18, 19]. Sb<sub>2</sub>S<sub>3</sub> is a semiconductor which has been also studied for its high photosensitivity and thermoelectric power [20]. It has been used in television cameras with photoconducting targets, microwave devices, thermoelectric, electronic and optical devices, as well as in infrared (IR) spectroscopy [21]. The optical properties of the Sb-Sb<sub>2</sub>S<sub>3</sub> system were studied by fluorimetry since this material shows photoluminescence characteristics [22]. In addition, sphere-like nanomaterials have excellent optical, physical and chemical properties that could be widely applied in biology, as fluorescence probes [23-25].

In this work, the synthesis of luminescent stibniteantimony, Sb<sub>2</sub>S<sub>3</sub>-Sb, nanoparticles embedded into a polymer matrix, by thermolysis of antimony dodecyl-mercaptide, Sb(SC<sub>12</sub>H<sub>25</sub>)<sub>3</sub>, dissolved in polymer is reported. It has been shown that the thermal decomposition of mercaptides is a powerful synthesis method for a number of noble metals, semimetals and metal sulphide [26]. Sb<sub>2</sub>S<sub>3</sub> nanomaterials can be synthesised by several methods namely hydrothermal method, refluxing polyol process, or in solution at room temperature, however the proposed synthesis allows the material preparation at quite mild temperature conditions (150–250°C), absolutely compatible with the thermal stability of common polymers. Here, the structural and morphological characteristics of the Sb(SC12H25)3 thermolysis product and the luminescent behaviour of the Sb<sub>2</sub>S<sub>3</sub>-Sb/polystyrene system are given.

## 2. Experimental

Antimony dodecyl-mercaptide, Sb(SC<sub>12</sub>H<sub>25</sub>)<sub>3</sub>, was prepared by adding drop-by-drop an alcoholic solution of dodecanethiol, C<sub>12</sub>H<sub>25</sub>SH (Aldrich), to an antimony chloride solution, SbCl<sub>3</sub> (Aldrich, 99.9%) in ethanol (Fluka, 99.8%) at room-temperature, under stirring (see Equation (1)) [27]. Stoichiometric amounts of reactants were used. The mercaptide precipitation did not take place after reactants mixing, but a pH correction, by addition of ammonium hydroxide, NH<sub>3</sub>·H<sub>2</sub>O (Aldrich) was required (see Equation (2)). The mercaptide promptly precipitated in form of a white crystalline powder, which was separated by vacuum-filtration and then washed several times with ethanol. The antimony dodecyl-mercaptide was further purified from the NH<sub>4</sub>Cl by-product by dissolution in chloroform followed by filtration and solvent evaporation (Equations (1) and (2)):

 $SbCl_3 + 3C_{12}H_{25}SH \Longrightarrow Sb(SC_{12}H_{25})_3 + 3HCl$  (1)

$$HCl + NH_4OH \rightarrow NH_4Cl + H_2O$$
(2)

The thermal decomposition of the obtained antimony dodecyl-mercaptide at 350°C, under vacuum (2 mbar), by using a hotplate, is expected to give an inorganic product made of a mixture of stibnite, Sb<sub>2</sub>S<sub>3</sub>, and antimony, Sb, powders. A preliminary study of the mercaptide thermal degradation has been carried out by Differential Scanning Calorimetry (DSC-TA Instruments 2920) from room-temperature to 150°C, at 10°C/min, under fluxing nitrogen, using sealed aluminium capsules. From the DSC and X-ray Powder Diffraction analysis of the inorganic powder product of the antimony dodecyl-mercaptide pyrolysis, it results that the processes developed during the thermal treatment took place in the following steps: (i) mercaptide melting at ca. 50°C, (ii) mercaptide decomposition to nanometric antimony (Sb) and antimony sulfide  $(Sb_2S_3)$  at ca. 150°C, and (ii) evaporation of the organic by-products, which refluxed inside the flask at ca. 300°C. After cooling down at room temperature, the inorganic phase was washed by chloroform and separated by centrifugation at 8000 rpm for 10 minutes.

In order to obtain stibnite-antimony/polystyrene nanocomposite films,  $Sb(SC_{12}H_{25})_3$  was dissolved in chloroform and mixed with a chloroform solution of polystyrene (Aldrich, molecular weight

 $M_w = 230\ 000\ \text{gmol}^{-1}$ ). Sb(SC<sub>12</sub>H<sub>25</sub>)<sub>3</sub>/polystyrene blend consisting of 5% by weight of mercaptide component was prepared. The obtained system was cast onto a glass-substrate and allowed to dry at room temperature. The thermal annealing of the obtained Sb(SC12H25)3/polystyrene blend was performed at ca. 250°C, using a hot-plate, and it led to stibnite-antimony/polystyrene nanocomposite film. Both powder and nanocomposite films were investigated by X-ray Powder Diffraction (XRD). The Sb<sub>2</sub>S<sub>3</sub>-Sb powder (hand ground in an agate mortar) was loaded in an aluminium flat holder and the X-ray data were collected using a Philips conventional Bragg-Brentano vertical diffractometer with Cu-K<sub> $\alpha$ </sub> radiation. The investigated 2 $\Theta$  range was  $5-80^{\circ}$  in steps of  $0.02^{\circ}$  and 8 s per step. The phase identification was carried out by means of X' pert High Score software (data base ICDD PDF2 Sets 1-82). The observed data were refined with GSAS (Larson and Von Dreele). The nanocomposite film was investigated in the  $2\Theta$  range from 5 to  $80^{\circ}$  with a step size of 0.02° and 1 s per step by means of a Rigaku DMAX-IIIC, using Cu-K<sub> $\alpha$ </sub> radiation,  $\lambda =$ 1.5418 Å.

The morphology of the Sb<sub>2</sub>S<sub>3</sub>-Sb powder mixture was examined by Scanning Electron Microscopy (SEM, Cambrige-S360) and the Sb(SC<sub>12</sub>H<sub>25</sub>)<sub>3</sub>/ polystyrene nanocomposite film was investigated by Transmission Electron Microscopy (Philips EM208S microscope equipped with a MegaView Camera for digital imaging, using an accelerating voltage of 100 kV) and photoluminescence spectroscopy (Perkin-Elmer-LS55 spectrofluorometer using a pulsed xenon lamp as an excitation source). TEM specimens were prepared by dissolution of the nanocomposite material in chloroform and the obtained solution was cast on graphitized film supported on copper mesh grids.

### 3. Results and discussion

The DSC thermogram of pure  $Sb(SC_{12}H_{25})_3$  shows an endothermic peak at 48°C, corresponding to the antimony dodecyl-mercaptide melting point. The nature of the inorganic products resulting from  $Sb(SC_{12}H_{25})_3$  pyrolysis was investigated by X-ray Powder Diffraction (see Figure 1). In order to obtain an accurate and quantitative phase analysis, a Rietveld refinement was performed. Two main phases were identified: stibnite (Sb<sub>2</sub>S<sub>3</sub>, reference





pattern N° 01-073-0393 crystal system Orthorhombic, space group Pnma, crystal lattice parameters *a* = 11.3107 Å, *b* = 3.8363 Å, *c* = 11.2285 Å, *V* = 487.22 Å<sup>3</sup>) and antimony (Sb, reference pattern N° 01-085-1324, crystal system Rhombohedral, space group R-3m, crystal lattice parameters a =4.3007 Å, b = 4.3007 Å, c = 11.2220 Å, V =179.75 Å<sup>3</sup>). A preliminary quantization of the two component amounts in the investigated powder mixture, carried out using the RIR ratio, displays 95 wt% stibnite and 5 wt% antimony. However, a more sophisticate phase quantization was carried out by using the Rietveld method with the GSAS program. The final agreement factors of the Rietveld refinement for the investigated stibnite/ antimony mixture are  $R_{wp} = 16.7\%$ ,  $R_p = 12.9\%$ and  $\chi^2 = 1.6$ , with a phase estimation of 10.4 wt% Sb and 89.6 wt% Sb<sub>2</sub>S<sub>3</sub>. The refinement lattice parameters of these phases are: stibnite a =11.3103 Å, b = 3.8367 Å, c = 11.2289 Å, V =487.27 Å<sup>3</sup> and antimony a = 4.3073 Å, b =4.3073 Å, c = 11.2752 Å, V = 181.16 Å<sup>3</sup>. The result of the Rietveld method was obtained refining the background, both cells, profiles, intensities and orientation. As it can be seen, the calculated lattice crystal parameters of stibnite are in good agreement with the observed one, while the calculated lattice parameters of antimony are slightly higher with respect to the expected ones. Both methods show





**Figure 2.** Representative SEM micrograph of the powder product resulting from the Sb(SC<sub>12</sub>H<sub>25</sub>)<sub>3</sub> thermolysis at 350°C, under vacuum: a) Sb<sub>2</sub>S<sub>3</sub> prismatic crystals and b) Sb polyhedral crystals

that stibnite is the main phase, while the antimony is present only in small amount.

The SEM-micrograph of the inorganic phase  $Sb_2S_3/Sb$  (see Figure 2) shows two different solid phases: one, made of prismatic shaped crystals, corresponding to stibnite (Figure 2a) and the other one, made of aggregated polyhedral crystals, corresponding to zero-valent antimony (Figure 2b). The crystals size was evaluated from the SEM micrographies by image analysis using Sigma Scan Pro 5.0 and gave the following information: the prismatic stibnite crystals have a length ranging from 30 to 150 µm and ca. 10–20 µm in diameter, while the polyhedral antimony crystals have an average size of 2–3 µm.

After this preliminary study of pure antimony dodecyl-mercaptide thermal degradation, the mercap-



**Figure 3.** XRD pattern of the Sb(SC<sub>12</sub>H<sub>25</sub>)<sub>3</sub>/polystyrene blend thermally treated at 250°C

tide decomposition in polymer has also been investigated. The XRD pattern of stibnite-antimony/ polystyrene nanocomposite film (see Figure 3) shows a broad reflection at ca. 20°, corresponding to the diffuse alone of amorphous polymer matrix and different peaks belonging to diffraction patterns of the inorganic nanostructured phase. The phase analysis of dodecyl-mercaptide decomposed in polystyrene allows the identification of two main phases: stibnite (Sb<sub>2</sub>S<sub>3</sub>, reference pattern N° 01-073-0393) and antimony (Sb, reference pattern N° 01-085-1324), just like in the case of pure dodecyl-mercaptide thermal decomposition.

A representative TEM micrograph of the stibniteantimony/polystyrene system is shown in Figure 4. TEM samples were prepared by dissolving 100 mg of nanocomposite in 20 ml of chloroform. A drop of the resulting solution was placed onto TEM copper grid support. As visible, the average size of the



**Figure 4.** Typical TEM micrograph of the Sb(SC<sub>12</sub>H<sub>25</sub>)<sub>3</sub>/ polystyrene thermally treated at 250°C

inorganic nanocrystals (stibnite and antimony clusters) was of ca. 15–30 nm.

The photoluminescence spectra of the nanocomposite antimony-stibnite/polystyrene films are shown in Figure 5. It can be seen that the excitation and emission peaks are shifted to higher wavelength values with respect to the corresponding peaks of the used pure polystyrene which exhibits a maximum of emission band at 389 nm for a maximum excitation at 296 nm. In fact, the emission band peak of the stibnite-antimony/polystyrene system appeared at 432 nm when excited at 371 nm which is the maximum of the absorption spectrum (see Figure 5). These spectra show a shape similar, to that of organic luminescent dyes, because the excitation and emission spectra result quite symmetric (mirror-like shaped peaks) [14]. This feature is rarely observed with nanoparticles, since nanocrystals usually exhibit a broad absorption with a narrow symmetric photoluminescence spectra (fullwidth at half maximum of 25-40 nm) [14]. The presence of the tail in the emission spectrum of the nanocomposite can be explained by taking into account the size distribution and the not uniform shape of the nanoparticles inside the polymer producing different emission spectra. As reported by other authors [28-30], the presence of surface traps, that modulates the emission of nanocrystals, should be also taken into consideration.

In order to exclude that the photoluminescence signal was related to the polystyrene presence, the emission spectrum of the nanocomposite was recorded at 296 nm, i.e. the excitation wavelength typical of polystyrene. As a result this experiment gives an emission signal with a maximum at



Figure 5. Photoluminescence spectra of an antimony-stibnite/polystyrene nanocomposite film

424 nm (data not shown). This photoluminescence band shows a lower intensity and a shift of the maximum with respect to the band of the nanocomposite obtained with  $\lambda_{exc} = 371$  nm, excluding a possible role of the polystyrene matrix in the nanocomposite material photoluminescence. In addition, the approximate value of the nanocomposite energy band-gap ( $E_g = hc/\lambda_{em}$ ) has been evaluated from the maximum emission peak wavelength in the PL spectra. The measured value of this energy bandgap is ca. 2.87 eV ( $\lambda_{em} = 432$  nm) which is almost 1 eV higher than the Sb<sub>2</sub>S<sub>3</sub> bulk-value (1.6 to 1.9 eV [31–32]). This result is in accordance with a well-known quantum-size effect observed with nanosized semiconductor materials [10, 11].

### 4. Conclusions

The antimony dodecyl-mercaptide thermolysis has been studied as a pure phase and as solution in polymer. In both cases the thermal decomposition of the mercaptide gave a mixture of antimony trisulfide (Sb<sub>2</sub>S<sub>3</sub>) and zero-valent antimony powder. The stibnite/antimony mixture obtained from the thermolysis of pure Sb(SC12H25)3 is made of 89.6 wt% Sb<sub>2</sub>S<sub>3</sub> prismatic crystals (30–150 µm length and 10-20 µm in diameter) and 10.4 wt% of Sb aggregated polyhedral crystals (average size 2– 3 µm). The antimony-stibnite/polystyrene nanocomposite, prepared by the thermolysis of  $Sb(SC_{12}H_{25})_3$ dissolved in polystyrene, exhibits peculiar luminescent characteristics due to the stibnite nanocrystals with average dimensions ranging from 15 to 30 nm. The possibility to produce in situ polymer nanocomposites with new functionalities is of great importance for application in widespread fields and the here proposed synthesis method based on antimony dodecyl-mercaptide thermolysis in polystyrene is compatible with all current polymer manufacturing technologies.

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