# Competing order parameters for increased $T_c$ in "polytype" multilayer Cu-O systems

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Using a simple phenomenological model with coupled order parameters for polytype multilayer copper oxide systems, it is demonstrated that polytypism can increase  $T_c$ . Explicit expressions and results are given for  $T_c(N)$  where N is the polytype number  $(N=2,3,\ldots,\infty)$ . Classes of structures investigated are monolayer polytypes such as  $[-(CuO_2)_N-]$ ; bilayer TIO polytypes such as  $[-(TIOTIO(CuO_2)_N-]$ ; and monolayer TIO polytypes; and 1:2:3 polytypes such as  $[-(CuO_2)_n-]$ . Two types of nearest-layer bilinear coupling were studied: weak link (Josephson) and spin-spin. Polytypism is predicted to increase  $T_c$  in all classes; except in one case. For the same N, monolayer TIO polytypes have lower  $T_c$  than the bilayer TIO analog. Using reasonable values of parameters we predict a maximum  $T_c$  of 140 K in the monolayer and bilayer TIO series.

## I. INTRODUCTION

In this work we present results of a "Landau" version of a Ginzburg-Landau theory which has been developed<sup>1</sup> for the copper oxide high- $T_c$  superconductors. We shall give predictions for the dependence of superconducting transition temperature upon "polytypism," i.e., multilayer repeated stacking of planes or layers in different structures.

The physical idea is that there are distinct "active" elements in the structure. To each active element (type of plane, or chain) denoted a in the structure is associated a Ginzburg-Landau order-parameter field  $\phi_a(\mathbf{r})$ . The fields  $\phi_a(\mathbf{r})$  can be different types of superconducting entities, i.e., "planes" and "chains" in 1:2:3 structures, or competing superconducting and normal planes in the TlBa structures. Each of these  $\phi_a(\mathbf{r})$  has some distinct "bare" transition temperature  $T_a^0$  in that structure. The  $\phi_a(\mathbf{r})$  fields are Josephson (weak link) coupled, and hence compete to be primary order parameters: This produces a shift from the bare temperatures to the observed transition temperature  $T^*$ , as well as electrodynamic effects. (In this paper the notation  $T^*$  means the predicted transition temperature.)

The full Ginzburg-Landau theory is a generalization to competing order parameters of the work of Lawrence and Doniach, Klemm, Beasely and Luther, Katz, Bulaevski, and others.<sup>2</sup> The free energy in the general Ginzburg-Landau theory<sup>1</sup> is real, gauge invariant, and possesses the relevant space-group symmetry elements of the structure; it is written  $F = \int_{(\tau)} f(\mathbf{r}) d\tau$  where  $f(\mathbf{r})$  is the free-energy density per cell. We may take  $f(\mathbf{r})$  as

$$f(\mathbf{r}) = \sum_{a} \left[ \alpha_{a} | \phi_{a}(\mathbf{r}) |^{2} + \beta_{a} | \phi_{a}(\mathbf{r}) |^{4} + \frac{h^{2}}{2m_{a}} \left| \left( -i \nabla_{a} - \frac{2e}{hc} \mathbf{A}_{a} \right) \phi_{a}(\mathbf{r}) \right|^{2} \right] + \sum_{a,b} \eta_{ab} \left| \phi_{a}(\mathbf{r}) \exp\left( -\frac{2ei}{hc} \int \mathbf{A} \cdot d\mathbf{l} \right) - \phi_{b}(\mathbf{r}) \right|^{2},$$
(1)

here  $\eta_{ab}$  is the relevant coupling coefficient;  $(\mathbf{V}_d, \mathbf{A}_d)$  are the *d*-dimensional gradient, and appropriate vector potential,  $\alpha_a(T) = (\alpha'_a/T_a^0)(T - T_a^0)$ , and the weak-link Josephson coupling and bare terms are in gauge-invariant form. This Ginzburg-Landau theory is based on the known crystal structure of several series of high- $T_c$  oxides, and has as its objective to provide a framework for the prediction of electrodynamic effects, such as  $H_{c2}$ , vortex structure, etc.

In the simplified Landau version discussed here, the competing order-parameter fields  $\phi_a(\mathbf{r})$  are taken spatially homogeneous (thus neglecting terms  $\nabla \phi_a$  in the free energy), and also no vector potential  $\mathbf{A}$  is present. We are

concentrating on the temperature region just near  $T_c^*$ . If  $T > T_c^*$ , all order parameters are zero ("normal" state), while for  $T < T_c^*$  some  $\phi_a \neq 0$  for minimum free energy. Sufficiently close to  $T^*$  terms such as  $\beta |\phi_a|^4$  in the free energy are small, and so we neglect them. The original motivation of this work concerned the competition between Cu-O chains and CuO<sub>2</sub> planes (as two active entities) in the 90° or 1:2:3 materials such as  $YBa_2Cu_3O_{7-\delta}$ . Development of the two series of Bi, and Tl-based mixed copper oxide high- $T_c$  superconducting systems has provided even more interesting classes of materials to which the theory applies with suitable changes, leading to testable predictions. In the homogeneous limit of (1), the free-

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energy density is a bilinear form in the spatially independent order parameters:

$$f = \sum_{a} \alpha_{a}(T) |\phi_{a}|^{2} + \sum_{a,b} \eta_{ab} |\phi_{a} - \phi_{b}|^{2}.$$
 (2)

In this limiting form of the "Lawrence-Doniach" (LD) Josephson interaction<sup>2</sup> the quadratic terms for each active entity are shifted,

$$\alpha_a(T) | \phi_a |^2 \longrightarrow \sum_b [\alpha_a(T) + \eta_{ab}] | \phi_a |^2,$$

leading to a first-order "mass" renormalization, or lowering of the bare  $T_a^0$ . This is a proximity effect which is a result of taking the interaction as in Eq. (2). However, omitting the diagonal terms  $\eta_{ab}(|\phi_a|^2 + |\phi_b|^2)$  we can take the free-energy density in another gauge-invariant form. This reduces to terms analogous to the bilinear spin-spin interaction (SS) in magnetic systems

$$f = \sum_{a} \alpha_{a}(T) |\phi_{a}|^{2} + \sum_{a,b} (-) \eta_{ab} \phi_{a} \phi_{b}^{*} + \text{c.c.}$$
(3)

A priori there seems no compelling reason<sup>3</sup> to choose either the LD "proximity effect" form (2) or the spin-spin form (3) for the Landau f.

Strictly, the bilinear free energy (2) or (3) cannot be minimized (for minimization the omitted nonlinear terms  $\beta_a | \phi_a |^4$  must be included). We can extremize f by solving the linear equations  $\partial f / \partial \phi_a = 0$ . The secular equation resulting is of the form  $|| (\alpha_a(T) + p\eta_{ab}) \delta_{a,b} - \eta_{ab} || = 0$ in case (2) [in this case, for finite number of elements (planes) the terms for boundary planes and interior planes differ: p=2 for interior and p=1 for boundary] and  $|| \alpha_a(T) - \eta_{ab} || = 0$  in case (3). The transition temperature  $T^*$  is the largest root. The corresponding form of f (rotated to principal axes) is

$$f = \sum_{\lambda} A_{\lambda}(T) |\zeta_{\lambda}|^2,$$

and  $A_{\lambda}(T^*) = 0$  at the greatest root, while  $\zeta_{\lambda}$  is the eigenvector, i.e., that linear combination of the bare order parameters  $\phi_a$  which first has nonzero amplitude at  $T^* = T$ .

We now apply this simple idea to several classes of high- $T_c$  copper oxide systems: the TlBa class Tl<sub>2</sub>Ba<sub>2</sub>-Ca<sub>n</sub>Cu<sub>n+1</sub>O<sub>6+2n</sub>, a related monolayer TlO class,<sup>4</sup> and the 1:2:3 class typified by YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7-s</sub>.<sup>5</sup> With some modification the model can apply to the 2:1:4 class typified by La<sub>2-x</sub>Ba<sub>x</sub>CuO<sub>4</sub>;<sup>6</sup> and the BiSr class typified by Bi<sub>2</sub>Sr<sub>2</sub>Ca<sub>n</sub>Cu<sub>n+1</sub>O<sub>6+2n</sub>.<sup>7</sup> In the last named case, other structural features omitted here, such as incommensurate modulation, may be important.

In all the classes we shall take one of the  $\phi_a$  to be the superconducting order parameter of CuO<sub>2</sub> plane, the other  $\phi_b$  will depend on the systems. Here, the objective is to predict the dependence of transition temperature  $T^*$  upon order of stacking N or polytypism. To implement this idea, we introduce polytype multilayer models as shown on Figs. 1(a)-1(c). In Fig. 1(a) polytypes of a single active layer are shown; we take this as the CuO<sub>2</sub> layer. In Fig. 1(b) the members of the bilayer TlBa class are shown (shaded layers are TlO, open layers are CuO<sub>2</sub>). In Fig. 1(c) we show member of the 1:2:3 class with CuO<sub>2</sub> as open layers and the CuO chain-containing layer interleaved. Modifications needed for the monolayer TlO systems are evident.



FIG. 1. Model for polytypes of different classes. Polytypes N = 1,2,3 are illustrated. (a) Monolayer polytypes: The basic layer is taken as a CuO<sub>2</sub> superconducting layer. (b) Polytypes of the bilayer TlO type in the TlBa class. TlO layers are cross-hatched, CuO<sub>2</sub> layers as above. For monolayer TlO type, delete one TlO layer and close up. (c) Polytypes of 1:2:3 class. The layers with filament are the layers of CuO chains; the open squares are CuO<sub>2</sub> layers.

# II. POLYTYPES OF CuO2 MONOLAYERS

The CuO<sub>2</sub> plane is a common entity in all classes if we neglect orthorhombicity, and also the "rippling" by which O and Cu ions are slightly noncoplanar, and take a tetragonal model for the CuO<sub>2</sub> array. First make a very simplified ansatz: only the CuO<sub>2</sub> planes are active in determining  $T^*$ . Consider polytypes of the basic layer: -(CuO<sub>2</sub>)<sub>N</sub>- [see Fig. 1(a)]. Further, take only coupling between nearest-neighbor CuO<sub>2</sub> layers  $\eta_{ab} = \eta_2 \delta_{n,n \pm 1}$  and use the SS form Eq. (3). The transition temperature  $T^*(N)$  for an N-layer polytype is the maximum eigenvalue of the N-dimensional Jacobi matrix

$$\begin{vmatrix} \alpha(T) & -\eta_2 & 0 & 0 & \cdots \\ -\eta_2 & \alpha(T) & -\eta_2 & 0 & \cdots \\ 0 & -\eta_2 & \alpha(T) & -\eta_2 & \cdots \\ \vdots & \vdots & \vdots & \vdots & \ddots \end{vmatrix} = 0.$$
(4)

In (4),  $a(T) = (a'_p/T^0_p)(T - T^0_p)$ . Both  $T^*(N)$  and the eigenvectors can be analytically obtained for N = 2, ..., 5. For  $N \to \infty$  we make a "Bloch" ansatz for order parameter  $\phi_n$  at the *n*th layer  $\phi_n = e^{i\theta}\phi_{n-1}$  and it is easy to see that  $T^*(\infty)$  occurs for  $\theta = 0$ . For this monolayer polytype sequence the results are of form

$$T^{*}(N) = T_{p}^{0}[1 + r_{N}(\eta_{2}/\alpha_{p}')],$$

TABLE I. Transition temperature for monolayer polytypes  $-(\text{CuO}_2)_N$ . The table lists the maximum eigenvalue of the determinant  $\Delta_N(T)$ , which is the predicted transition temperature  $T^*(N)$ . The coefficients  $r_N$  were found analytically. The experimental data used for illustration are for the bilayer TlO class of materials, as given in Refs. 6 and 8. In this monolayer model, we take  $T^*(1) = 80 \text{ K} = T_p^0$  for one layer CuO<sub>2</sub>;  $T^*(2) = 110 \text{ K}$  for two layers. Then the coupling constant  $\eta_2/a'_p = 0.38$ . The theory column is our predicted values.

$T^{*}(N) = [1 + r_{N}(\eta_{2}\alpha'_{p})]T_{p}^{0}$								
Number N	r <sub>N</sub>	$T^*$ -(N) for TlBa class Experiment Theory						
		20 K						
1	U	80 K						
2	1	110 K						
3	$\sqrt{2}$	125 K	123 K					
4	$(1+\sqrt{5})/2$		129 K					
5	$\sqrt{3}$		133 K					
:	:		:					
~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~	2		141 K					

where  $T_p^0$  is the bare-CuO<sub>2</sub>-layer transition temperature for N=1; the maximum is  $T^*(\infty)$ . In Table I the values are given for  $r_N$ .

Note that  $T^*(N)$  is an increasing function of N, and an important result is that for five layers (N=5) one has reached 95% of the maximum  $(N \rightarrow \infty)$  possible transition temperature, for reasonable values of parameters. A second important result is that the eigenvector for  $T^*(N)$  is always a *symmetric* combination of the bare layer order parameters. There is no coupling between symmetric and antisymmetric combinations. In the more general Ginzburg-Landau theory, Josephson phase factors provide such coupling.

Finally, note that if this model is applied to the TlBa and BiSr classes we would take  $T_p^0$  (the bare CuO<sub>2</sub> plane temperature) to be 80 K, which is the value for a single CuO<sub>2</sub> layer, i.e., in 2:0:2:1 there is one CuO<sub>2</sub> layer per cell. Then, using the measured temperature of 110 K for two-layer material (2:1:2:2) we obtain  $\eta_2/\alpha'_p = 0.34$ . It follows that  $T^*(\infty)$  which we predict as the maximum possible temperature for infinite monolayer sequence is  $T^*(\infty) = 138$  K for [2, (N-1), 2, N] with  $N \rightarrow \infty$  (see Table I).

## III. POLYTYPES IN TI-Ba AND Bi-Sr CuO<sub>2</sub> CLASS

At the time of writing several members of the Tl-Ba and Bi-Sr classes have been prepared, their  $T_c$  and crystal structures determined.<sup>6,7</sup> These are so-called types 2:0:2:1, 2:1:2:2, 2:2:2:3, and 1:2:2:3 with  $T_c = 80$ , 110, 125, and 110 K, respectively. In this notation, 2(N-1)2N identifies a polytype with

$$[-TlO-TlO-BaO-CuO_2-(Ca-CuO_2)_N-BaO-]$$
.

In the structural model we now adopt, the layers, e.g., TIO, BaO,  $CuO_2$ , Ca are stacked above one another, we

ignore the lateral shifts. In this polytype there are (N+1) layers of CuO<sub>2</sub>, with a layer of Ca between each, and  $N=0,1,2,\ldots$  per cell.

In recent work of relevance to our model, a material with monolayer TlO, composition 1:2:2:3 and  $T_c = 110$  K was prepared.<sup>8</sup> These authors determined the structure and report it involves identical layering as the 2:2:2:3 material, modulo replacing the bilayer TIO-TIO by the TIO monolayer, and closing up in the z direction. In the framework of polytype models, this permits the conclusion that TIO-TIO interaction is important in this class and suggests a natural generalization of the previous model. We make the following assumptions: (1) Each  $CuO_2$ plane has identical  $\alpha_p(T)$ . (2) The TIO planes are nonsuperconducting and  $\alpha_T = \text{const.}$  (3) Nonzero coupling between CuO<sub>2</sub>-CuO<sub>2</sub>, CuO<sub>2</sub>-TlO, and TlO-TlO. (The effects of the interleaved BaO and Ca layers are assumed incorporated in the appropriate coupling constants.) (4) We consider a periodic  $(\infty)$  repeat of the basic polytype cell (defined below). (5) We use the spin-spin form of interaction Eq. (3).

The scheme used for 2:2:2:3, with the layer-layer couplings defined, is

-[TIO- $\bar{\eta}_0$ -TIO- $(\eta_0)$ -CuO<sub>2</sub>- $(\eta_2)$ -CuO<sub>2</sub>- $(\eta_2)$ -CuO<sub>2</sub>- $(\eta_0)$ -].

The parameters of the model are bare  $T_p^0$ , normalized coupling constants  $\eta_0/\alpha'$ ,  $\bar{\eta}_0/\alpha'$ ,  $\eta_2/\alpha'$  if we take  $\alpha'_T = \alpha'_p = \alpha'$ . An infinite periodic array is assumed in the z direction. Since we only take nearest-neighbor couplings the resulting determinant is tridiagonal (not of Jacobi type). We solved analytically for the maximum root  $T^*(N)$  for the following members of the bilayer TIO-TIO class: N=1 (or TTC), N=2 (or TTCC), N=3 (or TTCCC), N = 4 [or  $TT(C)_4$ ], and  $N = \infty$  [or  $TT(C)_{\infty}$ ], using an obvious notation. We also obtained analytical solutions for the members of the monolayer TIO class  $[T(C)_N]$ , where N=3 (or TCCC) is the new 110-K material. Results are presented in Table II. Note that using the observed transition temperatures for materials 2:0:2:1, 2:1:2:2, and 1:2:2:3, we can obtain the parameters needed. Then we calculate  $T^*(3) = 123$  K for 2:2:2:3 (instead of measured 125 K), as well as  $T^*(\infty) \cong 140$  K.

Several points are worth noting about the results. (1) For this model, the bare  $T_p^0$  of CuO<sub>2</sub> will be determined from the measured  $T_c^{exp}$  for TTC,  $T_c = 80$  K. But now the relation between bare  $T_p^0$  and  $T^*(1) = T_c^{exp}$  (see Table II where various reduced ratios of coupling coefficients are defined) gives  $T_p^0 < T^*(1)$ . We obtain  $T_p^0 = T^*(1)/(1+2\zeta_2)$  where  $T^*(1)$  is the observed transition temperature, which is 80 K for (TTC). Then  $T_p^0 < 80$  K, and in our parameter set  $T_p^0 \approx 40$  K. (This is the value of  $T_c^*$  for La<sub>2-x</sub>Ba<sub>x</sub>CuO<sub>4</sub> with a single sheet of CuO<sub>2</sub>!) (2) The symmetric combination of order parameters  $\phi_a$  is the eigenvector for maximum root  $T^*(N)$ . (3) The maximum  $T^*(\infty)$  for the  $[TT(C)_{\infty}]$  polytype is predicted at  $\approx 140$  K. As a check, results for this (TlBa) or (BrSr) class go over to the monolayer polytype case when  $\eta_0 = \bar{\eta}_0 = 0$  and  $\alpha_T = 1$ . Essentially the same maximum  $T^*(\infty) \approx 140$  K is predicted for the series  $[T(C)_{\infty}]$  of polytypes.

Predictions for other members of the series are also

Number N	$-[TT(C)_n] - class T^*(N) = [1 + f^{TT}(N)]T_p^0$			$-[T(C)_N]$ - class $T^*(N) = [1 + f^T(N)]T_n^0$		
	$f^{TT}(N)^{a}$	Experiment	Theory <sup>b</sup>	$f^T(N)^{\mathrm{c}}$	Experiment	Theory <sup>b</sup>
1	2ζ2	85 K	88 K	4ζ3	NA	56 K
1	а	110 K	112 K	с	85	96 K
3	а	125 K	121 K	с	110 K	112 K
4 :	а		124 K	C	122 K	120 K
œ	$2\zeta_1$		136 K	$2\zeta_1$		136 K

TABLE II. Transition temperature for polytypes of the TlBa class. The table gives predicted transition temperature  $T^*(N)$  for polytypes of bilayer TlO and monolayer TlO members of the TlBa class. Theory column is our predicted values. Experimental values taken from Refs. 6 and 8.

<sup>a</sup>Functions  $f^{TT}(N)$  are defined as  $f^{TT}(2) = (\zeta_1 + \zeta_2)$ ;  $f^{TT}(3) - 0.5(\zeta_2 + [\zeta_2^2 + 8\zeta_1^2]^{1/2})$ ;  $f^{TT}(4) - 0.5(\zeta_1 + \zeta_2 + [(\zeta_1 - \zeta_2)^2 + 4\zeta_1^2]^{1/2})$ ; here  $\zeta_1 = (\eta_2/\alpha'_c)$ ;  $\zeta_2 = [\eta_0^2/\alpha'_c(\alpha_T - \bar{\eta}_0)]$ . Basic coupling  $\eta_2$ ,  $\eta_0$ ,  $\bar{\eta}_0$ , and  $\alpha'_c$  and  $\alpha'_T$  are defined in the text.

<sup>b</sup>The SS coupling was used in the calculation here. Parameter values are chosen to fit to the known experimental data  $T_p^0 = 40$  K,  $\zeta_1 = 1.2$ ,  $\zeta_2 = 0.6$ ,  $\zeta_3 = 0.1$ . Predicted values of  $T^*$  are shown in the theory columns.

<sup>c</sup>Functions  $f^{T}(N)$  are defined as  $f^{T}(2) = (\zeta_{1} + 2\zeta_{3}); f^{T}(3) = \zeta_{3} + [\zeta_{3}^{2} + 2\zeta_{1}^{2}]^{1/2}; f^{T}(4) = (\zeta_{1}/2 + \zeta_{3}) + [(\zeta_{1}/2 - \zeta_{3})^{2} + \zeta_{1}^{2}]^{1/2}; here \zeta_{3} = (\eta_{\delta}^{2}/\alpha_{c}'\alpha_{T}).$ 

given in Table II. In so far as the (BiSr) series is isostructural, the same predicted sequence of  $T^*(N)$  arises. However, incommensurate modulation in (BiSr) may play an important role, not included in the model.

Further observations: The asymptotic (maximum)  $T^*(\infty)$  for these series are essentially the same as for the monolayer polytypes ( $\approx 140$  K). The rate of approach to  $T^*(\infty)$  differs, with  $[TT(C)_N]$  reaching  $T^*(\infty)$  more rapidly in N than  $[T(C)_N]$ . These results use the spin-spin form of coupling as in Eq. (3) (see Table I).

#### **IV. POLYTYPES IN THE 1:2:3 CLASS**

The prototype material of this class is  $YBa_2Cu_3O_{7-\delta}$ . The crystal structure has been carefully determined<sup>9</sup> and the ordered vacancy model is well established. For our model, we also take account of the fact that there is conclusive experimental evidence<sup>10</sup> that replacing Y by rareearth ions carrying localized magnetic moments does not affect  $T^*$ . Thus, we assume that each slice of crystal consisting of the planes between Y ions can be treated in isolation from neighboring slices, the assumed composition being stacked planes in the z direction:

$$\dots$$
 -[-Y-CuO<sub>2</sub>-BaO-CuO-BaO-CuO<sub>2</sub>-]- $\dots$ 

(again neglecting orthorhombicity and the puckering of the CuO<sub>2</sub> planes). We identify two types of active entities as the CuO<sub>2</sub> planes (denoted p as before) and the plane containing parallel filaments of CuO chains (denoted c). Now, the basic assumption is that we have competing chain and plane superconducting order parameters, which produce  $T^* = 95$  K for the three-layer prototype 1:2:3. For the model, we take couplings between plane and chain (nearest neighbors) and plane and plane (second neighbors), and define the coupling constants via

$$\begin{bmatrix} -CuO_2 - (\eta_1) - CuO - (\eta_1) - CuO_2 - \end{bmatrix}.$$

This is the basic three-layer sandwich [see Fig. 1(c)]. We now assume (1) interaction terms between layers either LD proximity type [Eq. (2)] or SS type [Eq. (3)]; (2) each CuO<sub>2</sub> plane has identical  $\alpha_p(T)$  as before, with  $\alpha_p(T) = (\alpha'_p/T_p^0)(T - T_p^0)$ ; (3) the CuO chain layer has  $\alpha_c(T) = (\alpha'_c/T_c^0)(T - T_c^0)$ . We examined cases  $T_c^0 > T_p^0$ and  $T_p^0 > T_c^0$ . The coefficients  $\eta_1$  (and  $\eta_2$ ) are assumed to include the effect of exchange coupling via the BaO layer [more specifically, via the O(4) ion].

The general structure of polytype N of the 1:2:3 class is

$$[-CuO_2 - (CuO - CuO_2)_n -]$$
 with  $N = (2n+1)$ ,

where n=1 is the three-layer 1:2:3 case, n=2 gives a five-layer polytype, etc. [see Fig. 1(c)]. We use the ansatz that coupling  $\eta_1$  is present between nearest CuO<sub>2</sub>-CuO layers and  $\eta_2$  between second neighbor CuO<sub>2</sub>-CuO<sub>2</sub> layers. Then, the secular determinant is no longer triadiagonal, its structure is illustrated for n=2 for SS coupling:

$$\Delta(5) = \begin{vmatrix} x & y & z & 0 & 0 \\ y & x' & y & 0 & 0 \\ z & y & x & y & z \\ 0 & 0 & y & x' & y \\ 0 & 0 & z & y & x \end{vmatrix} = 0.$$

.

This form applies in case the interaction is taken as the spin-spin type (3), and then  $\alpha_p(T) = x$ ,  $y = -\eta_1$ ,  $z = -\eta_2$ ,  $\alpha_c(T) = x'$ . The maximum eigenvalue  $T^*(N)$  can be found analytically for N = 3,5 (N = 2n + 1). A general iterative algorithm relating determinants  $\Delta(N)$  of increasing order has been obtained,<sup>11</sup> but for  $N \ge 7$  nu-

merical solution is required for the maximum eigenvalue,  $T^*(N)$  even after factorizing  $\Delta(N) = P[(N-1)/2]Q \times [(N+1)/2]$  where P(N) and Q(M) are polynomials of N(M)th degree in T.

In case the LD coupling is used as in Eq. (2) the (1,1) and (N,N) elements in  $\Delta(N)$  are replaced by  $x^0$  and the identification is now  $x^0 = [\alpha_p(T) + \eta_1 + \eta_2]$ ,  $x = [\alpha_p(T) + 2(\eta_1 + \eta_2)]$ ,  $x' = [\alpha_p(T) + 2\eta_1]$ , y and z are as above.

Detailed results for  $T^*(N)$  as function of the parameters depend on whether LD or SS coupling is used, but certain general features are similar. For orientation we begin with the case of the three-layer prototype, and assume  $T_c^0 > T_p^0$ , and consider  $\Delta(3)$  for the LD coupling: This form is the limit of the Ginzburg-Landau theory for two superconducting types of order parameter, e.g., chain and plane as suggested by NQR experiments.<sup>12</sup> The transition temperature [maximum root  $T^*(3)$ ] is determined from the equation

$$\Delta(3) = [x'(x^0 + z) - 2y^2](x^0 - z) = 0,$$

and an examination shows that the first factor is relevant:

$$x'(T)[x^{0}(T)+z]-2y^{2}=0$$
.

This equation gives  $T^*(3)$  and corresponds to an eigenvector that is a linear combination of  $\phi_c$  for the chain, plus a symmetric superposition of the  $\phi_p^+ = \phi_p^-$  [+ (-) refers to CuO<sub>2</sub> plane above (below) the CuO chains]. From the definition of  $x^0$  we note that this equation is independent of z (i.e., independent of the CuO<sub>2</sub>-CuO<sub>2</sub> second-neighbors couplings  $\eta_2$ ). Hence,

$$\{\alpha_{c}[T^{*}(3)] + 2\eta_{1}\}\{\alpha_{p}[T^{*}(3)] + 2\eta_{1}\} - 8\eta_{1}^{2} = 0$$

and  $T^*(3)$  is directly obtained.

For the polytype with five layers,  $\Delta(5)$  factorizes into a quadratic and a cubic (in T) and  $T^*$  is determined from the maximum root of

$$xx^{0}x' - 2x^{0}y^{2} - xy^{2} + 4y^{0}z - 2x'z^{2} = 0$$

For normalized coupling parameters  $\eta_1$  and  $\eta_2 < 1$ , it can be shown directly that

$$T^*(5) > T^*(3)$$
.

Since, in this class we expect  $\eta_1 > \eta_2$ , if we neglect  $\eta_2$  (z =0) our equation becomes

$$xx' - 3y^2 = 0$$

which evidently gives a larger root  $T^*(5)$  than  $T^*(3)$ . Again the eigenvector corresponding is a symmetric combination of the chain, plus a symmetric combination of plane bare order parameters. For the next polytypes in this class with seven layers,  $\Delta(7)$  factorizes and  $T^*(7)$  is the maximum root of a fourth-degree equation, which is (taking z=0)

$$(xx')^2 - 4(xx')y^2 + 2y^4 = 0$$

or

$$xx' - (2 + \sqrt{2})v^2 = 0$$
.

We then find

$$T^*(7) > T^*(5) > T^*(3)$$

With some effort the case z = 0 can be studied explicitly for a few larger values of N, and

$$T^*(N) > T^*(N-1)$$
.

For  $N \to \infty$ ,  $T^*(\infty)$  with z = 0 is determined by

$$xx' - 4y^2 = 0$$

Thus in this case  $(T_c^0 > T_p^0)$  with LD coupling we obtain

$$T^*(\infty) > \cdots > T^*(N) > T^*(N-1) \cdots > T^*(3)$$
.

Keeping the LD coupling, and now letting  $T_p^0 > T_c^0$ , we obtain a descending sequence of maximum eigenvalues

$$T^{*}(3) > \cdots > T^{*}(N-1) > T^{*}(N) > \cdots > T^{*}(\infty)$$

for this case.

For SS coupling we find  $T^*(N)$  is an increasing function of N for both  $T_p^0 > T_c^0$  and  $T_p^0 < T_c^0$ .

Further details and illustrative calculations for the 1:2:3 class are given in the Appendix.

# **V. SUMMARY AND CONCLUSIONS**

In this work we introduced a simple model for the copper oxide high- $T_c$  superconductors based on Ginzburg-Landau theory in the homogeneous or Landau approximation. The model utilized the assumption that in each of the classes of materials considered there are active entities to which an ordered parameter (field)  $\phi_a$  should be assigned. A common active entity in the systems considered is the  $CuO_2$  plane. We assign a superconducting field  $\phi_p$  to each CuO<sub>2</sub> plane present. In the layerlike TlBa (or BiSr) systems we assume there is a second normal order parameter  $\phi_T$  for each TlO plane. The set of order parameters is coupled via nearest-neighbor plane-plane interaction. For the 1:2:3 class we take the second type of parameter to represent assumed superconducting order on the CuO chains and assign to the layer of coupled 1D CuO chains an order parameter  $\phi_c$  (it stands for a layer of coupled parallel chains, i.e.,  $\phi_c = \sum_m \phi_m$  so  $\phi_c$  is a Bloch sum of individual chain parameters  $\phi_m$ ). The planes and chains are coupled bilinearly: plane-plane and planechain couplings are included.

In the simplest version of the model we consider polytype or multilayer stacks of identical CuO<sub>2</sub> planes with nearest-neighbor coupling. For more elaborate models to apply to TlBa, and 1:2:3 types of systems, several types of planes are introduced [see Figs. 1(a)-1(c)]. In all these cases, the intermediate layers (e.g., BaO, Ca, etc.) are assumed to play a role by affecting the interlayer coupling constants  $\eta_{ab}$ . The transition temperature  $T^*$  is obtained by extremizing the free energy, and is the maximum eigenvalue of a certain determinant for that structure.

Our major results and predictions in the framework of this coupled order parameter model are the following.

(1) Polytypes increase  $T^*$  for N increasing (when SS coupling applies) in (a) monolayers  $-[CuO_2]_N$ ; (b) the TlBa classes  $-[TlO-TlO-[CuO_2]_N]$ - and  $-[TlO-TlO-[CuO_2]_N]$ - and -[TlO-TlO-[CuO\_2]\_N]- and -[TLO-[CuO\_2]\_N]- and -[TLO-TLO-[CuO\_2]\_N]- and -[TLO-[CuO\_2]\_N]- and -[TLO-TLO-[CuO\_2]\_N]- and -[TLO-[CuO\_2]\_N]- and -[TLO-[CuO\_2]\_N

 $[CuO_2]_N]$ -; (c) the 1:2:3 structure - $[CuO_2-(CuO-CuO_2)_n]$ - if  $T_c^0 > T_p^0$ .

(2) Polytypes increase  $T^*$  for N increasing (with LD coupling) in the 1:2:3 structure if  $T_c^0 > T_p^0$ .

(3) Polytypes decrease  $T^*$  for N increasing (when LD coupling applies) in the 1:2:3 structure if  $T_c^0 < T_p^0$ .

(4) In the TlBa classes for reasonable values of parameters, the model accounts for experimental  $T_c$ ; and 95% of the maximum possible  $T^*$  is reached for polytype N = 5.

(5) The monolayer TIO compounds  $[T(C)_N]$  have lower  $T^*$  than corresponding bilayer TIO  $[TT(C)_N]$ .

(6) The maximum possible  $T^*$  for the monolayer TIO and bilayer TIO classes is  $T^*(\infty) \approx 140$  K.

Note added. After our work was completed, we became aware of other work with some similarity to our model.  $^{13-16}$ 

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## APPENDIX: ILLUSTRATIVE CALCULATIONS ON 1:2:3 CLASS POLYTYPES

Returning to the general secular determinant  $\Delta(N)$  for 1:2:3 class, we now neglect the second-neighbor interaction (take z = 0). For prototype 1:2:3 in our model, N = 3(with the effect of the layers of BaO incorporated into the coupling coefficients  $\eta_1$  and  $\eta_2$ ) and the active entities are planes of CuO<sub>2</sub> (P) and layers of CuO chains (C). The structure is P-C-P. Polytypes of 1:2:3 are as above: P-(C-P-)<sub>n</sub>, so N = (2n+1). The secular determinant  $\Delta(N)$ is

$$\Delta(N) = \begin{vmatrix} x & y & 0 & 0 & 0 & \cdots \\ y & x' & y & 0 & 0 & \cdots \\ 0 & y & x & y & 0 & \cdots \\ 0 & 0 & y & x' & y & \cdots \\ \vdots & \vdots & \vdots & \vdots & \vdots & \ddots \end{vmatrix} = 0.$$

First consider SS coupling [Eq. (3)] and let  $y = -\eta_1$ ,  $x = (\alpha'/T_p^0)(T - T_p^0)$ ,  $x' = (\alpha'/T_c^0)(T - T_c^0)$  and take  $\alpha'_p = \alpha'_c = \alpha'$ . Let  $T_p^0 = 3T_c^0$ . The relevant coupling parameter is  $\eta_1/\alpha'$ . Figure 2 shows  $T^* = T_c$  [maximum root of  $\Delta(N)$ ] for N = 3,5,7 and  $N = \infty$ , as a function of  $\eta_1/\alpha'$ .



FIG. 2. Superconducting transition temperature  $T^*(N)$  as function of reduced coupling parameter  $\eta_1/a'$  for 1:2:3 polytypes with N layers [see Fig. 1(c)]. The system is composed of two types of layers:  $L_1$  and  $L_2$ , which are stacked alternately  $[L_1-L_2-L_1-...]$  total N layers, or if  $L_1$ =plane,  $L_2$ =chain, then [P-C-P-...]. For this figure,  $T_p^0=3T_c^0$  and  $T_c=T^*(N)$  are given in units of  $T_p^0$  or  $T_c^0=3T_p^0$ , and  $T_c$  is given in units of  $T_c^0$ . Spinspin coupling between nearest layers is assumed. Note  $T_c=T^*(N)$  is an increasing function of N (for any  $\eta_1/a'$ ) reaching 95% of its asymptote  $T^*(\infty)$  at N=5. Labels on the curves refer to N.

As previously,  $N = \infty$  is the periodic repeat case. Results clearly show that  $T_c = T^*(N)$  is an increasing function of N, reaching (for  $\eta_1 < 1$ ) 95% of the maximum possible value  $T^*(\infty)$  by N=5. All results for  $T_c$  are normalized in terms of  $T_p^0$ : i.e., at y=0  $T^*(N)=T_p^0$  independent of N. Still retaining the SS interaction, let us now take



FIG. 3. System as described in Fig. 1(c). Here LD nearestlayer coupling is used [see text Eq. (2)] and  $T_c^0 = 3T_p^0$ . Note  $T_c = T^*(N)$  is an increasing function of N at any  $\eta_1$ .  $T^* = T_c$  is given in units of  $T_c^0$ .



FIG. 4. System as described in Fig. 1(c). Here LD nearestlayer coupling is used [see text Eq. (2)] and  $T_p^0 = 3T_c^0$ . Note  $T_c = T^*(N)$  is a decreasing function of N at any  $\eta_1$ .  $T^*(N) = T_c$  is given in units of  $T_p^0$ .

 $T_c^0 = 3T_p^0$ . Then Fig. 2 still applies except now  $T_c = T^*(N)$  is in units of  $T_c^0$ ; i.e., at y = 0  $T^*(N) = T_c^0$  in-

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dependent of N. Clearly  $T^*(N)$  for SS coupling is an increasing function of N.

Now turn to the Lawrence-Doniach coupling (2). Keep nearest-neighbor coupling  $y = -\eta_1$ . In the secular determinant  $\Delta(N)$  the two end diagonal entries are now  $\Delta(N)_{11} = \alpha_p(T) + \eta_1 = \Delta(N)_{NN}$ , while the remaining diagonal entries are  $x = \alpha_p(T) + 2\eta_1$ ;  $x' = \alpha_c(T) + 2\eta_1$ . Now let  $T_c^0 = 3T_p^0$ . Figure 3 shows  $T^*(N)$  for different polytypes as function of reduced coupling strength. To be noted is that  $T^*(N)$  is an increasing function of N.

For LD coupling but  $T_p^0 > T_c^0$ , results are shown in Fig. 4. In this case  $T^*(N)$  is a decreasing function of N at any  $\eta_1/\alpha'$  to the limit  $T^*(\infty)$ .

These graphical computations indicate that except for LD coupling in the 1:2:3 structure, with  $T_p^0 > T_c^0$ , polytypism should always increase  $T^*(N)$  with N. The maximum value  $T^*(\infty)$  can be obtained from the figures for given  $\eta_1/\alpha'$  as a fraction of the bare temperature  $T_p^0$  or  $T_c^0$ .

For example, using LD coupling with  $T_c^0 = 200$  K (bare temperature for a layer of 1D chains) and  $T_p^0 = 60$  K (bare temperature for a CuO<sub>2</sub> plane), then from Fig. 3 with  $\eta_1 \approx \frac{1}{4}$ , we would find  $T^*(\infty) \approx 134$  K as the maximum in 1:2:3 class. These assignments of  $T_c^0$ ,  $T_p^0$  follow the original report of Warren *et al.*<sup>12</sup> but have since been retracted [see Pennington *et al.*<sup>12</sup> and Walstedt *et al.*], and assignments are now reversed with  $T_p^0 > T_c^0$ .

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