

# THE UNIVERSITY of EDINBURGH

### Edinburgh Research Explorer

### **Dynamic Optical Spectroscopy and Pyrometry under Optical and** X-ray Laser Heating at European XFEL

#### Citation for published version:

EuXFEL Community 2023, 'Dynamic Optical Spectroscopy and Pyrometry under Optical and X-ray Laser Heating at European XFEL', *Journal of applied physics*, vol. 134, 055901, pp. 1-38. https://doi.org/10.1063/5.0142196

### **Digital Object Identifier (DOI):**

10.1063/5.0142196

Link: Link to publication record in Edinburgh Research Explorer

**Document Version:** Peer reviewed version

**Published In:** Journal of applied physics

#### **General rights**

Copyright for the publications made accessible via the Edinburgh Research Explorer is retained by the author(s) and / or other copyright owners and it is a condition of accessing these publications that users recognise and abide by the legal requirements associated with these rights.

Take down policy The University of Edinburgh has made every reasonable effort to ensure that Edinburgh Research Explorer content complies with UK legislation. If you believe that the public display of this file breaches copyright please contact openaccess@ed.ac.uk providing details, and we will remove access to the work immediately and investigate your claim.



### Dynamic Optical Spectroscopy and Pyrometry under Optical and X-ray Laser Heating at European XFEL

O.B. Ball,<sup>1</sup> C. Prescher,<sup>2,3</sup> K. Appel,<sup>4</sup> C. Baehtz,<sup>5</sup> M.A. Baron,<sup>6,7</sup> R. Briggs,<sup>8</sup> V. Cerantola,<sup>4</sup> J. Chantel,<sup>7</sup> S. Chariton,<sup>9</sup> A.L. Coleman,<sup>8</sup> H. Cynn,<sup>8</sup> H. Damker,<sup>2</sup> D. Dattelbaum,<sup>10</sup> L.E. Dresselhaus-Marais,<sup>8,11,12</sup> J.H. Eggert,<sup>8</sup> L. Ehm,<sup>13</sup> W.J. Evans,<sup>8</sup> G. Fiquet,<sup>6</sup> M. Frost,<sup>12</sup> K. Glazyrin,<sup>2</sup> A.F. Goncharov,<sup>14</sup> R.J. Husband,<sup>2</sup> H. Hwang,<sup>15,16</sup> N. Jaisle,<sup>17</sup> Zs. Jenei,<sup>8</sup> J-Y. Kim,<sup>18</sup> Y. Lee,<sup>15</sup> H.P. Liermann,<sup>2</sup> J. Mainberger,<sup>2</sup> M. Makita,<sup>4</sup> H. Marquardt,<sup>19</sup> E.E. McBride,<sup>12,20</sup> J.D. McHardy,<sup>1</sup> M.I. McMahon,<sup>1</sup> S. Merkel,<sup>7</sup> G. Morard,<sup>6,17</sup> E.F. O'Bannon III,<sup>8</sup> C. Otzen,<sup>2,21</sup> E.J. Pace,<sup>1</sup> A. Pelka,<sup>5</sup> C.M. Pépin,<sup>22</sup> J.S. Pigott,<sup>10,23</sup> C. Plückthun,<sup>4,24</sup> V.B. Prakapenka,<sup>9</sup> R. Redmer,<sup>24</sup> S. Speziale,<sup>25</sup> G. Spiekermann,<sup>26</sup> C. Strohm,<sup>2</sup> B.T. Sturtevant,<sup>10</sup> P. Talkovski,<sup>2</sup> L. Wollenweber,<sup>4</sup> U. Zastrau,<sup>4</sup> R.S. McWilliams,<sup>1, a)</sup> and Z. Konôpková<sup>4, b)</sup>

(EuXFEL Community Proposal #2292)

<sup>1</sup>) The School of Physics and Astronomy, Centre for Science at Extreme Conditions, and SUPA, University of Edinburgh, Peter Guthrie Tait Road, Edinburgh, EH9 3FD.

UK

<sup>2)</sup> Deutsches Elektronen-Synchrotron DESY, Notkestr. 85, 22607 Hamburg, Germany

<sup>3)</sup>Institute of Earth and Environmental Sciences (Geomaterials and Crystalline Materials),

Albert-Ludwigs-University Freiburg, Germany

<sup>4)</sup> European XFEL GmbH, Holzkoppel 4, 22869 Schenefeld, Germany

<sup>5)</sup> Institute of Radiation Physics, Helmholtz-Zentrum Dresden-Rossendorf, Bautzner Landstraße 400, 01328 Dresden, Germany

<sup>6)</sup>Inst. Minéralogie, de Physique des Matériaux et de Cosmochimie (IMPMC), Sorbonne Université,

UMR CNRS 7590, Museum National d'Histoire Naturelle, 4 Place Jussieu, Paris, France

<sup>7</sup>) Univ. Lille, CNRS, INRAE, Centrale Lille, UMR 8207 - UMET - Unité Matériaux et Transformations, F-59000 Lille, France

<sup>8)</sup>Lawrence Livermore National Laboratory, 7000 East Avenue, Livermore, CA 94550, USA

<sup>9)</sup>Center for Advanced Radiation Sources, University of Chicago, Chicago, IL 60637, USA.

<sup>10)</sup>Los Alamos National Laboratory, Los Alamos, New Mexico 87545, USA

<sup>11)</sup>Stanford University, Department of Materials Science & Engineering, Stanford, CA, USA

<sup>12)</sup>SLAC National Accelerator Laboratory, 2575 Sand Hill Road, Menlo Park, California 94025, USA

<sup>13)</sup> Mineral Physics Institute, Stony Brook University, Stony Brook, NY 11794, USA

 $^{14)}\mathit{Carnegie}\ Science$  , Earth and Planets Laboratory, 5241 Broad Branch Road, NW, Washington, DC 20015

<sup>15)</sup>Earth System Sciences, Yonsei University, 50 Yonsei-ro, Seodaemun-gu, Seoul 03722, Republic of Korea

<sup>16)</sup>School of Earth Sciences and Environmental Engineering, Gwangju Institute of Science and Technology,

123 Cheomdan-Gwagiro, Gwangju 61005, Republic of Korea

<sup>17)</sup> Université Grenoble Alpes, Université Savoie Mont Blanc, CNRS, IRD, IFSTTAR, ISTerre, 38000 Grenoble, France

<sup>18)</sup> Department of Physics, Research Institute for High Pressure, Hanyang University, 222 Wangsimni-ro, Seoul 04763, Republic of Korea

<sup>19)</sup> Department of Earth Sciences, University of Oxford, South Parks Road, OX1 3AN Oxford, United Kingdom

<sup>20)</sup> School of Mathematics and Physics, Queen's University Belfast, University Road, Belfast, BT7 1NN, United Kingdom

<sup>21)</sup>Institute of Geosciences, Friedrich Schiller University Jena, Carl-Zeiss-Promenade 10, 07745 Jena, Germany

<sup>22)</sup> CEA, DAM, DIF, F-91297 Arpajon, France and Université Paris-Saclay, CEA,

Laboratoire Matière en Conditions Extrêmes, 91680 Bruyères-le-Châtel, France

 <sup>23)</sup>Case Western Reserve University, 10900 Euclid Ave, Cleveland, OH 44106, United States

<sup>24)</sup> Universität Rostock, Institut für Physik, D-18051 Rostock, Germany

<sup>25)</sup> GFZ German Research Centre for Geosciences, Telegrafenberg, 14473 Potsdam, Germany

<sup>26)</sup> ETH Zürich, Rämistrasse 101, 8092 Zürich, Switzerland

(Dated: 8 March 2023)

Experiments accessing extreme conditions at X-ray free electron lasers (XFELs) involve rapidly evolving conditions of temperature. Here we report time-resolved, direct measurements of temperature using spectral streaked optical pyrometry (SOP) of X-ray and optical laser-heated states at the High Energy Density (HED) instrument of the European XFEL. This collection of typical experiments, coupled with numerical models, outlines the reliability, precision, and meaning of time dependent temperature measurements using optical emission at XFEL sources. Dynamic temperatures above 1500 K are measured continuously from spectrallyand temporally-resolved thermal emission at 450 - 850 nm, with time resolution down to 10 - 100 ns for 1  $-200 \ \mu s$  streak camera windows, including in single sweep mode. Targets include zero-pressure foils freestanding in air and *in vacuo*, and high-pressure samples compressed in diamond anvil cell multi-layer targets. Radiation sources used in the presented examples are 20-femtosecond hard X-ray laser pulses at 17.8 keV, in single pulses or 2.26 MHz pulse trains of up to 30 pulses, and 250-nanosecond infrared laser single pulses. A range of further possibilities for optical measurements of visible light in X-ray laser experiments using streak optical spectroscopy are also explored, including for study of X-ray induced optical fluorescence, which often appears as background in thermal radiation measurements. We establish several scenarios where combined emissions from multiple sources are observed and discuss their interpretation. Challenges posed by using X-ray lasers as non-invasive probes of sample state are addressed.

#### I. INTRODUCTION

The study of warm dense matter (WDM) and high pressure-temperature states of condensed matter is vital to understanding complex physical systems that exist in nature such as planetary interiors  $^{1,2}$  and in technologies such as internal confinement fusion<sup>3</sup>, however it is poorly described by theoretical models<sup>4</sup>. Laboratory experiments are therefore central to characterising WDM behaviour. The creation of high pressure (millions of atmospheres) and temperature (thousands of degrees of kelvin or higher) conditions are required to study these fundamental systems  $5^{-14}$ . Traditionally, these are created through intense irradiation  $15^{-19}$ , dynamic shock  $compression^{20-23}$ , or a combination of static compression with these techniques, often using a diamond anvil pressure cell  $(DAC)^{5-14,24}$ . However, these techniques are often plagued by short experimental lifetimes due to loss of high-density conditions. Damage to samples can also occur over long exposures through sample contamination from chemical reactions with surrounding material (e.g. diamond anvils) or cumulative thermal or mechanical damage. This can then lead to discrepancies between various experimental results<sup>7,12,25-27</sup>. At the boundary between traditionally fast and slow measurements a variety of experiments are possible which address common challenges of WDM experiments, however novel measurement techniques and diagnostics are often needed for such  $regimes^{9,11,28}$ .

X-ray Free Electron Lasers (XFELs) present a new frontier for intermediate timescale extreme conditions studies. Excited states following femtosecond pulsed irradiation can persist for beyond microsecond timescales on XFELs, for example due to volumetric heating of massive objects<sup>29</sup>. Meanwhile, pulse repetition rates in the MHz range, such as is currently available at the European X-ray Free Electron Laser (EuXFEL)<sup>30,31</sup>, offer opportunities for serial probing and excitation on similar, microsecond timescales.

Controlling samples subjected to WDM extremes over such experimental lifetimes as well as defining the pressure-density conditions reached, is made possible using DACs to contain and pressurize samples of interest<sup>29</sup>. This configuration is ideal to study WDM as it is possible to use pulsed X-ray heating to heat targets to tens of thousands of Kelvin<sup>27,29,32-35</sup>. In this senario hard X-ray energies greater than 10 keV have long absorption lengths ( $\sim$  millimeters) for light elements (i.e diamond), allowing penetration of millimeter thick diamonds, while absorption lengths in the micrometer range for high atomic number elements enable localized energy deposition within micrometer thick samples. XFELs are ideally suited to this type of experiment because they are able to produce short (femtosecond pulse length), highpower (~  $10^{12}$  photons per pulse), hard X-ray (>10 keV) pulses. These pulses also provide a fast and high power probe to observe the sample conditions 27,35-37, and are often available in a pair or train. This therefore naturally leads to the ability to perform serial pump-probe  $experiments^{15,38}$ .

A number of diagnostics can be employed to infer sample conditions, many using the pulsed X-ray source itself to obtain a snapshot of conditions in the sample bulk, including temperature. If the pump is also an X-ray pulse, the pump and probe are automatically spatially aligned, which is ideal for sampling the heated volume. X-ray diffraction (XRD) is a well-developed technique<sup>16</sup> to determine sample conditions through measurements of sample density and phase transformations, but is complicated by the difficulty in discriminating between the effects of pressure and temperature changes. Likewise, Xray Thomson scattering, often used in conjunction with X-ray heating, can estimate electron temperature but requires knowledge of how free electrons scatter incoming X-rays, with temperature uncertainties often on the order of a few  $eV^{39-43}$ , and therefore often requires high

<sup>&</sup>lt;sup>a)</sup>Electronic mail: rs.mcwilliams@ed.ac.uk

<sup>&</sup>lt;sup>b)</sup>Electronic mail: zuzana.konopkova@xfel.eu

resolution information on the probe X-ray spectrum as well as a clear understanding of the X-ray scattering to determine temperature. Such X-ray snapshots of temperature can be further limited when observing fast and/or dynamic continuous changes in sample conditions, e.g. due to availability of a single probe timing in each event, or insufficient time sampling options<sup>27</sup>.

An ideal temperature diagnostic will continuously determine sample temperature through a period of interest with high time resolution and small uncertainty independent of the detailed sample conditions. Pyrometry provides such a method, inferring sample temperature from optical emission released from heated samples. As opposed to active X-ray probing, pyrometry is passive so inherently provides a continuous record of the sample changes. Pyrometric measurements have long been used to estimate the temperature reached in static compression (i.e. laser-heated diamond anvil cell,  $LHDAC)^{5-14,44}$ , shock compression<sup>20-23</sup>, and fast radiative (i.e. isochoric) heating studies  $^{45,46}$ , though while  $proposed^{33}$  its application to XFEL heating has not been demonstrated. Pyrometry exploits the emission released during sample heating to determine temperatures reached<sup>10</sup> usually by assumption that radiation follows the behaviour of a grey-body radiator, as defined by Planck's Law<sup>20,47</sup>.

Streaked optical pyrometry (hereafter, SOP) provides one such method to measure *in-situ* sample temperatures with potentially high time resolution in the picosecond to microsecond range. Streak camera images record emission over a pre-determined time window, denoted the streak window. Changes in emission brightness as a function of time are recorded on the image as changes in intensity along the time axis<sup>48,49</sup>. The spatial, spectral and temporal shape of optical emission from any heated sample is difficult to obtain simultaneously<sup>20</sup>, thus the orthogonal axis of the streak image will resolve the emission either spatially or spectrally (i.e. producing a position vs. time linear intensity map across the target or a wavelength vs. time spectrogram from a particular area on the target). Each approach has advantages that depend on the sample properties, temperature range, target design, spatial scale of the event, and other factors. For example, the presence of optical windows in targets, such as those used when observing statically compressed samples or a shocked interface, adds complications such as the requirement that any optical emission pass through layers of other material, potentially affecting the final collected emission 50,51. In many cases the use of spectrally (as opposed to spatially) resolved detection is advantageous as it is less sensitive to emissivity, target transmission, other complex optical properties of targets, temperature inhomogeneity, or time dependence of these factors, within certain limits 52,53. The measured spectral shape also allows an independent verification that emission adheres to the predictions of the Planck model and thus provides a final test of whether the measured temperature can be valid, by resolving obvious distortions in

thermal emission or nonthermal contributions to signal. For these reasons, spectrally resolved temperature determination is generally considered more reliable in complex sample environments, such as those using tight focused radiative heating similar to the X-ray and optical heating employed in this study (e.g. in traditional laser heated DAC experiments).

The detection limit of SOP is predicated on the temperature, sweep window, number of events (shots) integrated, size of heated and observed areas, event duration, sample optical characteristics, spectral range detected, and spectrometer dispersion, and varies significantly depending on these factors for different applications. Lower detection limits for SOP fall in the range of 400 – 4000 K in typical single shot HED applications  $^{9,45,48,54,55}$ , and is observed to be 1500 - 3000 K for the system described here. Fundamentally, this is limited by the rapid reduction in emission with decreasing temperature (i.e. the Stefan–Boltzmann law). While there is no upper limit on detectable temperature, spectral form becomes insensitive to temperature where the Planck distribution peaks well below the observed wavelength band (i.e. Wien's law), which occurs, e.g., in the electron volt range for visible diagnostics; here absolute emission intensity remains a reliable – and possibly superior – indicator of temperature, in certain circumstances<sup>48</sup>.

Here we report the design and performance of a streaked optical spectroscopy diagnostic for pyrometry (SOP) implemented at the High Energy Density (HED) instrument of the EuXFEL, Germany. The system measures optical signals in the visible spectral range, in spectral-temporal space and can be deployed in a range of sample environments creating extreme conditions scenarios. It was specifically designed and tested to work in conjunction with DAC static compression techniques and optimized to detect thermal radiation from hot samples undergoing rapid temperature changes. Equipment for the system was provided to the HED instrument by the HIBEF user consortium, and the design and implementation of the SOP system was performed in collaboration between EuXFEL, Deutsches Elektronen-Synchrotron (DESY), and University of Edinburgh. Basic design parameters are discussed in Sec. II. the experimental procedure in Sec. III, samples in Sec. IV, and results in Sec. V. The system was developed at DESY, with first experiments using laser heating only (Section VA). It was moved to EuXFEL for the first online measurements with X-rays only, in association with the first User Community Assisted Commissioning Experiment (1st UCAC) Proposal #2292 in October  $2019^{36,37,56}$  (Sections VB, VC, VD, VE and VF). Laser heating and X-ray heating together were subsequently performed in further commissioning work (Proposal #2731) (Sections VG and VH). Numerical modelling of experimental conditions achieved for comparison to the experimental results are included in Sec. VI, while a discussion of the findings and conclusions for future work using the diagnostic are given in Sec. ??.

#### **II. EXPERIMENT SETUP**

#### A. Sources

European XFEL offers three self-amplified spontaneous emission (SASE1, SASE2, SASE3) sections that produce X-ray photons at wavelengths varying from 0.1 nm to 1.6 nm. The HED experimental hutch is located at the end of the SASE2, which produced photons in the hard X-ray regime between 0.05 - 0.25 nm<sup>57</sup>. X-ray pulses are grouped into trains of up to 2700 pulses, with a train repetition rate of 10 Hz and intra-train pulse repetition rate of fractions of 4.5 MHz, giving pulse separation of 221.5 ns (or 443 ns, 886 ns etc.) Energy intensity monitors are placed upstream and downstream of the experimental chamber in order to measure XFEL beam intensity and any fluctuations in intensity that result from beamline  $optics^{37}$ . Energy on target is determined to be  $\sim 30\%$  of the measured upstream beam intensity on the absolutely-calibrated X-ray intensity gas monitor (XGM) in the SASE2 beamline (SASE XGM)<sup>37,58,59</sup>, if there is no further attenuation of the beam. Downstream energy intensity monitors are placed in the beamstop after the experimental chamber (Fig. 1). The XFEL heating (XH) due to individual pulses and pulse trains are examined here. The XFEL beam is focused onto the target using several sets of on-axis chromatic compound refractive lenses, with spot sizes in the range of  $5-20 \ \mu m$  on target used in the described experiments (Table I).

Single or double-sided pulsed or continuous optical heating (LH) of targets is made possible using an NIR laser (Model SP-100P-A\_EP\_Z, 1070 nm, 1.2 eV) which was integrated with the SOP optical system (Figs. 1b) and 2c). The NIR laser has a repetition rate of up to 1 MHz, standard pulse lengths ranging from 20 - 420 ns and a collimated beam diameter of 10 mm. Focusing with the geoHeat system creates a spot size on the sample of 11 x 14  $\mu$ m (FWHM). Some laser waveforms possible are shown in Fig. 3. The maximum pulse energy is 1 mJ (depending on the waveform and repetition rate), and peak power of 10 kW. Incident laser power is attenuated from 100% to 0% by a combination of the polarizing beam splitting cube and the rotating quartz waveplates in the laser beam path (Fig. 1b). The upstream and downstream laser powers are controlled separately.

#### B. Optical Configuration

The SOP experiments at the HED instrument are designed to take place in Interaction Chamber 2 (IC2). Fig. 2a and 2b shows the optical configuration used during these experiments. Further details about the experimental set-up of IC2 can be found in Refs. 37, 60, 61, 62 and 63.

Optical access to the sample is possible from both upstream and downstream sides, relative to the XFEL beam (see Sec. II A). The near sample optics consist of turning mirrors at 45 degrees relative to the XFEL beam and objectives (AdlOptica geoHEAT-60-NIR) with 60 mm focal length focusing through the mirrors onto the sample plane, both upstream and downstream of the sample inside the vacuum chamber (Fig. 1). The mirror-objective system can be retracted when not in use in order to clear the X-ray beampath. The optical beams are guided through the chamber window to the custom-built optical table (Fig. 1b) where the rest of the optical components are placed.

The upstream turning mirror was initially a dielectric coated glass mirror (Semrock MGP01-650-1300) with a 300  $\mu$ m hole drilled in the centre to allow the X-ray beam to pass through. The downstream turning mirror was a custom made glassy carbon plate coated with Ag and was retracted during single-sided LH and XH experiments, as it was not necessary and it would unnecessarily affect the XRD images. Both the upstream and downstream mirrors were later replaced by 1 mm thick Ag-coated silica glass mirrors (from AHF) with holes drilled for X-ray direct beam transmission. This allowed for double-sided LH experiments with XH, as the downstream mirror was no longer required to be retracted whilst using the XFEL. The hole diameter is 1 mm downstream and 300  $\mu$ m upstream.

Acquisition of optical emission is performed only from the upstream side in the present experiments, whereas sample observation (and illumination) and optical laser heating can be performed from both sides. The downstream and upstream optical paths are split to allow for imaging with an optical microscope, heating with the NIR laser, and, on the upstream side, spectroscopic pyrometry by passing the light through to the spectrometer and streak camera system.

Online microscopy is achieved using long focal length lenses (f=750 mm upstream, f=1000 mm downstream) to image onto Basler CCD cameras, to provide sample viewing and alignment at high image magnification. Illumination from both sides of the sample is achieved by inserting pellicle beamsplitters in front of the viewport chamber window on the optical paths to deliver a red LED light source to the sample. Alignments are made using direct imaging of the NIR laser at low laser power and localized effects of X-ray irradiation, including target fluorescence or damage imprinting.

The optical path of the pyrometric signal includes an optional spatial filter consisting of a 50  $\mu$ m diameter pinhole and two 60 mm focal length lenses (~1:1 magnification) designed to exclude non-sample emissions and other background light (such as X-ray fluorescence from diamonds above and below the focal plane in DAC targets). The light is focused onto the spectrograph entrance slit by a f=150 mm lens for an effective magnification of ~ 2.5. The numerical aperture is ~ 0.16 at the imaging objective and spatial filter segments, and less at the spectrometer. Additional details on the optical configuration of the SOP can be found in Refs. 9, 44 and 64.



TABLE I: XFEL parameters used during the experiments discussed here.

FIG. 1: Schematic of optical diagnostic at the HED instrument. a) Detailed drawing of the immediate area surrounding the sample environment inside IC2. Upstream and downstream optical beam path is shown in green. b) Schematic of optics table layout showing the upstream and downstream optical paths from sample inside IC2 to either a microscope or the spectrometer and streak camera (black). NIR Laser heating pathway is shown in red.

#### C. Detector Configuration

In order to acquire time- and spectrally-resolved information on the emitted light, an optical spectrometer IsoPlane 160 (Princeton Instruments) is coupled to the front slit of the streak camera. The vertical slit of the spectrograph restricts input emission in the spectral direction. The spectrometer has a 203 mm focal length and f/3.88 aperture ratio.

The streak camera is a Hamamatsu system with S-20 photocathode (model C13410-01A) with spectral response in the region of 200 – 850 nm and a horizontal slit which restricts the input emission in the time axis direction, affecting the time resolution and point spread function (PSF). The MCP intensifier gain of the streak camera can be varied from 0 – 60. The sweep windows for the streak camera range between 0.5 ns to 1 ms. The streak tube is coupled to an ORCA-Flash 4.0 V2 Digital CMOS camera with 1344 (H) x 1016 (V) pixels in the working area and 6.5  $\mu$ m pixel size; sometimes this is binned 2 x 2, i.e 672 (H) x 508 (V) pixels, to im-

prove data handling. The readout speed of the camera is >100 frames/s. There are two different collection modes which the streak camera is operated with: enclosing trigger and sequential mode. In sequential mode, exposure to the CMOS only occurs during a single sweep regardless of how many events are incident, meaning that the visible signal from the interaction of each NIR or XFEL pulse is recorded on the final spectrogram, separated in the time domain by their intrinsic pulse separation (see Sec. II A). In most cases, the experimental data is taken using this mode. For data collected in the enclosing trigger mode, the spectrogram shows emission that has been integrated over many sweeps at a repetition rate of up to 1 kHz, with varying total integration time. Thereby, increasing integration time increases the number of exposures that are accumulated per spectrogram image, each exposure timed with a (presumably identical) repeating event (i.e. heating cycle). This mode is compatible with accumulation over EuXFEL pulse trains repeating at 10 Hz. Fig. 4 shows schematically how the two different collection schemes differ.

The wavelength axis of spectrograms is calibrated to

![](_page_6_Figure_1.jpeg)

FIG. 2: Photographs showing the optical set-up inside IC2 and on the adjoining optical bench. a) Sample stage inside IC2, including 6-chamber sample revolver, viewed from the downstream. b) Sample stage inside IC2, viewed from the upstream. c) Optics table layout showing the upstream and downstream optical paths to microscopy cameras and the spectrometer and streak camera. Upstream (US) and downstream (DS) optical paths are shown in blue and laser heating pathway is shown in red. Key experimental components are highlighted in pink. The AGIPD (adaptive gain integrating pixel detector) is used for time resolved X-ray diffraction (XRD) measurements<sup>37</sup>.

![](_page_6_Figure_3.jpeg)

FIG. 3: Example of measured pulse waveforms for the NIR pump laser (in this study we make use of WF\_0). Y-axis indicates oscilloscope voltage therefore is in arbitrary units.

![](_page_6_Figure_5.jpeg)

FIG. 4: Schematic of the relevant timings of the streak camera and pump pulses. Pump pulses can be either from a laser heating system or XFEL pulses. a) Series of pulses integrated in enclosing trigger mode, one pulse per sweep. b) Series of pulses captured in the same streak window in sequential mode, three pulses in a single sweep. Streak sweep denotes the voltage applied to the electrode in the streak camera; pump pulse denotes the timings of the radiation pulses incident on the sample; CMOS exposure denotes the streak camera

detector integration time.

the emission from a Ne calibration lamp resulting in a 451 - 955 nm observable wavelength range in this experiment. The optical response of the SOP system,  $F^{ideal}(\lambda)$ is a combination of quantum efficiency of the S20 photocathode, the reflectivity of the Semrock turning mirror, and the accumulated behavior of other optics (Fig. 5, see Sec. IIG). The transmission function is experimentally measured by imaging a W calibration lamp with a known greybody Planck distribution and comparing the intensity, as a function of wavelength and time, to the theoretical Planck distribution for the temperature at which the lamp is operating. Fig. 5a shows that between 575 - 775 nm the experimentally measured transmission  $(F_1(\lambda))$  closely agrees with the spectra predicted from the two main optical elements discussed above. Most spectral dips and oscillations in this transmission function were later eliminated by replacement of the dielectric turning mirrors with Ag-coated mirrors<sup>65</sup>, also enabling a somewhat broader spectral band for detection (Fig. 5b).

The combination of horizontal and vertical slits in the streak spectrometer, located at the imaging planes of the sample, also function as vertical and horizontal spatial filters, respectively, and ensure good time and spectral resolution for spatially extended emission sources. Here we report both slitted and slitless operation, the latter case being required during early EuXFEL experiments where X-ray beam pointing drift and instability precluded reliable alignment to the slits, with the small emitting area on the samples imaged to the detector and functioning as a point source in the time and space directions, with

![](_page_7_Figure_0.jpeg)

FIG. 5: SOP spectral response when using the initial dielectric (a) and final Ag (b) upstream turning mirrors. Measured spectral transmission functions, based on calibration lamp data, are  $F_1(\lambda)$ , blue, and  $F_2(\lambda)$ , red, respectively. Expected response is  $F^{ideal}(\lambda)$ , green, with the published reflectivity of the turning mirror, grey, and the quantum efficiency of the S20 photocathode<sup>66</sup>, dashed black. The wavelength region used to determine sample temperature is shown as two vertical dashed lines. The measured transmission function is scaled to the expected transmission function.

sufficient (but reduced) temporal and spectral resolution.

An example of the streak camera acquisition of optical light emitted from the W calibration lamp, designed to be a thermal greybody source at a set temperature (in this case 2900 K, cross calibrated to a NIST lamp), is illustrated in Fig. 6. The horizontal axis represents time (here a 5  $\mu$ s sweep window), and the vertical axis contains spectral information covering 451 – 955 nm. Poor optical sensitivity in the intensity measurement at the edges of the spectrogram are observed (Fig. 5), so we restrict the wavelength range when fitting temperatures to 575 – 775 nm. All reference lamp calibrations were obtained with slits and spatial filtering in place, in order to probe the

precise optical path from the sample in the respective transmission function.

From spectral features present in the calibration spectrograms, datasets with distinct event timings, and data on spectrally distinct light sources (i.e. laser or LED), some warping of streak images is observed, dependent on the streak window; that is, shear distortions with respect to the wavelength and time axes are identified. We compensate for this warping by applying a shear distortion to both wavelength and time axes unique to each sweep window. The magnitude of the shear, per pixel, in each direction, dependent on the streak window used, is given in table II.

Collected spectrograms are also expected to exhibit 'pincushion' flat-field sensitivity variation in both axes, on the order of 10% sensitivity. This is corrected for by calibrating the experimental emission data locally (in time and wavelength) to a spectrogram of a calibration lamp at a set temperature (e.g. Fig. 6), using identical sweep window and gain settings.

#### D. Fluorescence behavior

Non-thermal luminescence is often observed in targets containing dielectric materials, including YAG:Ce scintillator crystals used as fluorescent screens for beam alignments, and diamond anvil targets adding multiple layers of dielectrics (Fig 7). On shorter sweeps, in the 5 – 50  $\mu$ s range, this often manifests in SOP images as short pulses of emission, coinciding in time with irradiation from the XFEL pulses, and can be attributed to X-ray fluorescence. Fluorescence emission pulses are thus discriminable from thermal emission as they are usually symmetric in time (unlike heat pulses<sup>29</sup>), appear at lower power (where direct heating is small) and only where transparent dielectrics are present in the beam (which also have low emissivity). X-ray fluorescence also increases quasilinearly with X-ray fluence (Sec. VD). Since fluorescence appears in the streak images as a symmetric peak in time, in most cases we treat it as instantaneous, with an apparent width due only to the PSF of the combined optical-detector system. That is, the width is not dependent on time since true emission duration is much smaller than the time resolution of the utilized streak windows. and instead the width represents the physical size and distribution of emission arriving on the photo-cathode of the streak camera. Femtosecond-excited fluorescence and decay would normally exhibit asymmetric behavior. Indeed, some minor deviations from symmetric emission are observed but are generally not well resolved in experiments to date. The fluorescence could be examined in more detail using faster sweeps but this was not attempted here. Fluorescence peak profiles are thus well modeled with a Lorentzian function. Peaks can be fit to determine peak location, amplitude, and full width at half maximum (FWHM) – which is also the FWHM of the PSF. Fluorescence can also be forward modeled if

![](_page_8_Figure_1.jpeg)

FIG. 6: Optical emission collected from the calibration lamp at 2500 K and 2900 K. a) Spectrogram of optical emission collected from a calibration lamp at 2900 K using a 5  $\mu$ s streak window, taken using enclosing trigger with a sweep repetition rate of 1 kHz and exposure time of 1 s. Region corresponding to 193 ns section binned at 2.8  $\mu$ s is shown between two dashed lines. b) Binned optical emission at time = 2.8  $\mu$ s of calibration lamp at 2900 K (black markers) and expected optical emission of thermal radiation at 2900 K based on Planck's distribution (red dashed line). c) Binned optical emission at time = 2.8  $\mu$ s of calibration lamp at 2500 K, black markers, and transmission function of full optical path at same time (red solid line) derived from (b). d) Corrected optical emission of calibration lamp at 2500 K, grey markers, at time = 2.8  $\mu$ s, with moving average intensity with respect to wavelength shown as a solid black line. The fitted Planck distribution (red dashed) to the corrected emission data at T = 2591 ± 50 K is also plotted.

these parameters are known.

Fig. 7 shows how fluorescence emission, collected when operating in the slitless configuration, can be commonly observed from DAC targets when irradiated by X-ray pulses (here a single pulse). Emission intensity is pronounced when using type Ia diamond anvils, and may vary with respect to the pressure medium. In contrast, no fluorescence emission is observed from Type IIa diamonds with an  $H_2O$  pressure medium. Other experiments targeting an interface of the diamond with a metal have confirmed significant fluorescence from Type I diamonds and total absence of fluorescence from Type II diamonds, limiting the emission to the diamond itself. For almost all of the targets, fluorescence emission follows a similar Lorentzian-like distribution in the time domain. This confirms the timescale of fluorescence emission is instantaneous in comparison to the streak window and that the data indicates the PSF of the optical system. However, in the case of an NaCl sample, a time delay and asymmetry of the emission peak is observed with the corresponding Lorentzian fit shifted significantly to later times. The results suggest that the fluorescence lifetime of NaCl is

longer than other materials, and so can be detected here (i.e. relaxation time is long enough that it affects the fluorescence peak shape and position).

The spectral shape of fluorescence emission from select samples (HIBEF24, HIBEF30 and D1, see Sec. IV) is compared to thermal emission from the calibration lamp at 2900 K in Fig. 8a, collected in slit-less operation. The raw emission spectral shape is determined by summing the emission from large numbers of pulses over many spectrograms and within spectrograms, where there is no thermal emission and only fluorescence emission is present at the time of the fluorescence pulse. Strong peaks in fluorescence emission at  $\sim 450$  nm and  $\sim~800$  nm, not seen in emission from the calibration lamp and located in lower-sensitivity spectral ranges of the SOP, are characteristic of the XFEL induced fluorescence. Spectral shape is fitted to five super-imposed Gaussian distributions which approximate the distribution shape for fluorescence forward modeling (see Secs. VD, VE and VF). For comparison, Fig. 8b shows the spectral shape of the fluorescence emission for both Type Ia and IIa diamonds under X-ray irradiation at 18 keV.

![](_page_9_Figure_0.jpeg)

FIG. 7: Fluorescence emission from DAC targets under single XFEL pulse irradiation, at 100% XFEL transmission (See Table III for energy on target). Emission is averaged in wavelength and plotted against time in pixels, as apparent emission timescale is independent of sweep rate and represents the detector point spread function in most cases. The targeted materials are low-Z pressure-transmitting media, indicated in the legend, which were usually placed around other samples of interest and which could be independently targeted by the XFEL. These have low degrees of direct heating and low emissivity (and hence, negligible thermal emission). Most DAC targets employ type Ia anvils, with one sample ( $H_2O$  medium) with type IIa anvils shown for reference. The inset shows Lorentzian fits to the fluorescence emission (arbitrarily scaled) for each of the emission profiles. All spatial filters and slits were removed during acquisition.

The spectrum for the Type Ia diamond was measured from the upstream diamond only with the X-ray beam targeting the Re gasket, to exclude emission from the sample or downstream diamond. The Type IIa spectrum was alternatively measured from emission from a freestanding diamond.

#### E. Experimental Timing, Time Resolution, and PSF

The time resolution of temperature measurements is set by the choice of time bin width. The measured PSF establishes the minimum time resolution possible for a certain configuration, below which it is not possible to measure dynamic processes. For example, noisy low temperature data requires a larger time resolution, whereas near continuous high temperature data can be analysed with a small time resolution, but not smaller than the PSF since this does not provide any benefits.

The PSF of the streak images is highly dependent on the optical configuration used, including entrance slit widths. During initial commissioning at EuXFEL, when

TABLE II: Streak window (SW) dependent corrections applied to SOP images. The streak window timings given by factory setting compared to measured streak window from fluorescence emission are shown. The magnitude of the time and wavelength shear (pixel shift in time/wavelength divided by vertical pixel step) applied to images, when using the 1 x 1 un-binned image size, is also given.

| 5       | Streak Wind | Shear    |      |            |
|---------|-------------|----------|------|------------|
|         | Factory     | XFEL     |      |            |
| Setting | measured    | measured | Time | Wavelength |
| 1       | 0.994       | —        | 1/65 | 1/80       |
| 5       | 4.90        | 4.73     | 1/65 | 1/80       |
| 10      | 9.76        | 9.27     | 1/65 | 1/80       |
| 20      | 19.60       | 18.57    | 1/80 | 1/80       |
| 50      | 49.0        | 45.64    | 1/80 | 1/80       |
| 100     | 97.4        | -        | 1/20 | 1/80       |
| 200     | 200         | -        | 1/10 | 1/80       |

all spatial filtering and slits were present, the PSF was 15 pixels when the SOP image was binned 2 x 2, i.e. 30 pixels for 1 x 1 binning, and 140 ns for a 5  $\mu$ s streak window. When the spatial filter and slits were removed, the PSF widened to 25 pixels FWHM, i.e. 50 pixels for 1 x 1 binning, and 242 ns for a 5  $\mu$ s streak window. There was also a shift in the peak locations (by ~ 2 pixels earlier). Improvements after the experiments reported here further reduced the PSF with spatial filtering and slits to 7 pixels FWHM in 1 x 1 (un-binned) images (i.e. 34 ns for a 5  $\mu$ s streak window), using an optical configuration similar to experiment 3 (Tables I and III).

Sweep rate, and the associated streak window, of the SOP spectrograms were initially set to factory calibrations<sup>66</sup>. We confirm streak window lengths by measuring the distance between neighbouring pulses of fluorescence emission. XFEL pulses that are incident on target are separated by 443 ns in this study, therefore we expect that the associated fluorescence emission is also separated 443 ns across all wavelengths. By comparing the measured pulse separation of XFEL pulses in pixels to the known pulse separation in seconds, the full length of the streak window is calculated. Absolute timings of the 5, 10, 20 and 50  $\mu$ s streak windows are determined to be 4.73, 9.27, 18.57 and 45.64  $\mu$ s, respectively (Table II). For streak windows longer than this we are unable to determine streak windows as there is insufficient resolution between neighbouring peaks in emission to determine pulse separation. For a 5  $\mu$ s streak window we determine that the factory calibration is acceptable whilst for longer windows the discrepancy between our measured values and the factory calibration is such that measured values are recommended. Further measurements are required to determine the absolute timings of the other streak windows. A linear sweep rate approximation is found to be accurate enough for most analyses (include all those presented here) however non-linear cor-

![](_page_10_Figure_1.jpeg)

FIG. 8: Wavelength dependence of emission intensity from fluorescent samples compared to thermal emission. a) Spectral shape of fluorescence emission from D1 (blue), HIBEF24 (green) and HIBEF30 (red) compared to the spectral shape of thermal emission from a calibration lamp (at 2900 K). Fitted emission spectral shape is shown in black. All three samples use Type Ia diamond anvils on the upstream side. See Fig. 5a for the transmission function of the optical system used. b) Spectral shape of fluorescence from type Ia and IIa diamond measured under 18 keV synchrotron X-ray irradiation at the GSECARS beamline, Advanced Photon Source. The transmission function of the optical system at GSECARS is shown for reference (grey).

rections are required in certain scenarios, for example, in modeling fluorescence at smaller PSF where the precise location of the fluorescence peak becomes important.

Fluorescence peak position can also provide absolute information on X-ray timing within the streak window. Thus, where it was possible, the fluorescence of a particular target was used to determine the global timings of streak images collected for that target, i.e. time zero is set to be the time at which the first pulse in any given train reaches the target. If insufficient or indistinct fluorescence emission was present for data on a particular target, then experimental timings were estimated from the timing of separate YAG:Ce screen fluorescence or from the onset of optical emission. Relative timings of the optical laser were estimated using emission observations and local photo-diodes which observed both radiation sources.

#### F. Emission Spectrogram Reduction

Background is first removed by subtracting a dark image from the data and is further corrected by subtracting the averaged emission intensity in the spectral 'dead' zone (850 - 955 nm).

During image processing, each spectrogram is binned with respect to the time axis, therefore reducing the time axis to discrete time steps at which we determine temperatures. The size of each bin is the time resolution,  $t_R$ , of a run and will vary based on the type and quality of emission. The time of each bin is calculated to be the time at the centre, with t = 0 denoting the time at which the sample is first irradiated. Generally, a  $t_R$  comparable to the FWHM of the PSF optimizes the time resolution.

The moving average of the intensity of optical emission with respect to wavelength was also computed in order to visually better compare noisy data with the Planck fit presented in the figures, shown later in the manuscript. Spectral intensity distribution is determined by a 50 pixel (or 37.5 nm) wide moving average, regardless of sample or streak window.

#### G. Temperature Measurement

The time-resolved history of sample temperature is determined through a two-parameter least-squares fit<sup>67–69</sup> of each time-binned emission spectrum to a modified Planck distribution

$$I_{Planck}(\lambda, T) = \frac{1}{F(\lambda)} \epsilon(\lambda) \frac{2hc^2}{\lambda^5} \frac{1}{\exp(\frac{hc}{\lambda k_B T}) - 1}$$
(1)

where

$$F(\lambda) = \frac{I_{cal}(\lambda)}{I_{Planck}(\lambda, 2900 \text{ K})}.$$
 (2)

Here,  $I_{Planck}(\lambda, T)$  is the raw intensity of emission at a given wavelength,  $\lambda$ , and temperature, T;  $\epsilon(\lambda)$  is emissivity (here, in arbitrary units) and  $1/F(\lambda)$  is the transmission function of the optical system and detector; h, c, and  $k_B$  are Planck's constant, the speed of light, and Boltzmann's constant, respectively. We assume a greybody approximation for emission and therefore emissivity is taken to be constant in wavelength ( $\epsilon(\lambda) = \epsilon$ )<sup>47,70,71</sup>. Temperature and emissivity are correlated fit parameters in Eq. 1 as the magnitude of the emission intensity can be increased either by increasing temperature or

emissivity, but at all wavelengths equally if emissivity is assumed constant. Eq. 2 shows how the transmission function is calculated based on the ratio of intensity of thermal emission from a W calibration lamp,  $I_{cal}(\lambda)$ , to the theoretical grey-body Planck spectrum for the lamp temperature (here, 2900 K, see Fig. 6). All optical emission recorded is normalised by this function (Fig. 6d). The transmission function is calculated for each time integration bin and streak window used, so that we account for any sensitivity variance over the streak spectrogram (i.e. flatfield correction). Collection of the calibration spectrogram is made using the enclosing trigger mode to ensure optimum signal; also, proper spatial filtering must be in place during collection of the extended light source from the calibration lamp, and the streak camera must be operated in gated mode to exclude background noise produced during reverse sweep on the CW lightsource.

As discussed above, certain samples may exhibit Xray fluorescence during XFEL experiments which can overlap with any thermal signal present. In the examples discussed here, fluorescent emission is clearly discriminated from thermal emission. It only occurs during XFEL pulses, often as distinct peaks (or as a continuum where pulses overlap in longer sweep windows), whereas thermal emission often persists after pulses due to the longer relaxation time of thermal processes. Fluorescence is particularly strong in the blue where thermal emission is usually insignificant. Spectra containing high fluorescence also exhibit clearly non-Planckian trends. Here we explore removal of fluorescence emission to improve temperature analysis by building an empirical model of the fluorescence spectral and temporal shape and fluence dependence, for any given sample. This is based on data where no thermal emission is observed (at low XFEL power) which then establishes fluorescent background for thermal emission data (at higher XFEL power) (Sec. VD). We find that when the fluorescent background is weak compared to thermal signal it has a negligible influence on the assessed temperatures. Additionally, fluorescent emission tends to be temporally localized, so that purely thermal emission can be detected at other times (e.g. between or after pulses).

A number of criteria were used for automatic data reliability assessment in temperature measurements. These criteria were based on a careful review of data accuracy and noise performance, and exclude spurious data but do not guarantee measurement quality. Data are excluded if:

1. There is significant (> 30%) signal at or below zero intensity (after background subtraction), within a time bin. This implies insufficient optical emission to determine sample temperature. It should be noted that in high noise datasets the standard deviation of intensity in a binned spectrum can exceed the average intensity even after this exclusion is applied, but there is sufficient signal above zero for reliable fitting<sup>72</sup>.

- 2. Least-squares fitting returns temperatures below  $\sim$  1500  ${\rm K}^{9,11,45,48,54,55}.$
- 3. The emissivity determined from fitting the data to Eq. 1 ( $\epsilon_{fit}$ ) is 1000 times smaller or 100 times larger than the emissivity determined from fitting the emission spectrum from the calibration lamp ( $\epsilon_{cal}$ ), i.e 100  $\cdot \epsilon_{cal} < \epsilon_{fit} < 0.001 \cdot \epsilon_{cal}$ . This criterion accounts for physical limits on emissivity variation and the possibilities for different size emitting areas observed.
- 4. The magnitude of temperature uncertainty (Sec. II H) obtained from least-squares fitting exceeds 50% of the absolute temperature.

Temperature is ideally uniform over the heated area which the measurement is restricted to using the slits and spatial filter. The wavelength dependent emission as a function of time can be used to infer local temperatures reached and the subsequent rate of temperature change unambiguously<sup>9</sup>. However temperature gradients are expected to be present in the observed surface areas even when using these filters due to: (a) imaged beam sizes similar to or smaller than spatial filtering elements, (b) Gaussian (or similar) beam spot profiles, (c) misalignment and beam drift, (d) thermal conduction in targets, (e) overlapping exposures in X-ray pulse trains, and (f) interplay of optical and X-ray beams of different sizes. Spatial filtering elements (slits and spatial filter) were removed in some earlier experiments to address XFEL beam drift, but this should not affect the observed temperature distribution significantly as the optical pathway and observed area is only weakly affected by this. We consider that the observed temperature in SOP represents the maximum achieved on the observed sample surface, considering the strong scaling of emission intensity with temperature in the visible range, and findings in comparable offline systems (e.g. the optical laser heated DAC)<sup>52</sup>. Our experiments combining large-spot optical radiation and small-spot X-rays bears out this expectation, with local hotspots tending to dominate apparent temperatures (see Sec. V).

#### H. Temperature Error

We use two independent ways to calculate the error on the temperature that is determined through least-squares fitting: a) from the in-built error from the *curve\_fit* function in the SciPy python package<sup>69</sup> that is used for fitting and b) from the curvature of  $\chi^2$  in temperature-emissivity space.

The build in python error estimate, which is based in the minimization of the goodness-of-fit parameter  $\chi^2$  and is calculated from the square-root of the diagonal of the co-variance matrix. This error is consistently the smaller of the two. We therefore take it as the smallest possible error that could be associated with a particular temperature measurement.

![](_page_12_Figure_0.jpeg)

FIG. 9: Example of a  $\chi^2$  contour map of the two-parameter least-squares fit of wavelength resolved thermal emission from calibration lamp (at 2500 K). A

1  $\sigma_T$  contour is shown in black about the minimum, with the associated error based on Eq. 4 shown as a black error bar; the error determined from the python *curve\_fit* fitting routine is also shown for comparison in red. Inset shows quadratic fit of  $\chi^2$  taken along the

minimum path across the contour map, used to establish  $\sigma_T$ .

The curvature of  $\chi^2$  space is assessed by plotting  $\chi^2$ , given by

$$\chi^{2} = \sum_{i} \frac{(I_{i}(\lambda) - I_{P}(\lambda, T))^{2}}{\sigma_{I}^{2}} = \frac{(T - T_{fit})^{2}}{\sigma_{T}^{2}} \qquad (3)$$

as a surface about the minimum, i.e as a function of the temperature and emissivity, then assessing the local curvature  ${\rm as}^{73}$ 

$$\sigma_T^2 = 2 \left(\frac{\partial^2 \chi^2}{\partial T^2}\right)^{-1}.$$
 (4)

In Eq. 3,  $I_i(\lambda)$  is the measured intensity as a function of wavelength where  $\sigma_I$  is the standard deviation of  $I_i(\lambda)$ .  $I_P(\lambda, T)$  is the ideal black-body intensity for any arbitrary temperature T (Eq. 1). We take  $\sigma_I$  as the standard deviation between the measured intensity  $I_i(\lambda)$  and the ideal black-body intensity  $I_P(\lambda, T_{fit})$  at the measured temperature  $T_{fit}$ , as determined by the python *curve\_fit* routine. The temperature at which  $\chi^2$  is a minimum is the best-fit temperature,  $T_{fit}$ , with the uncertainty in the measurement given as  $\sigma_T$  (Eq. 4).

The second derivative of  $\chi^2$  (Eq. 4) about the minimum determines  $\sigma_T$ . Assuming a quadratic relationship between  $\chi^2$  and the temperature of the form  $\chi^2 = \alpha \cdot T^2 + \beta \cdot T + \delta$  implies that  $\sigma_T = \sqrt{1/\alpha^{73}}$ . Considering that fit parameters are strongly correlated, we compute the second derivative in  $\chi^2$ , with respect to temperature, using the minimum magnitude of  $\chi^2$  at each temperature i.e., with emissivity allowed to vary. This is intended to represent a conservative error estimate. In some cases we see that the minimum on the contour map is slightly different to the temperature determined from the *curve\_fit* fitting routine – we therefore, when appropriate, include this temperature measurement, T', on the final figure showing sample temperature as a function of time.

Fig. 9 shows the  $\chi^2$  contour map from least-squares fitting of the calibration lamp at 2500 K, calibrated to emission from the same lamp at 2900 K. The standard deviation,  $\sigma$ , in a parameter is defined as the magnitude at which a parameter needs to change from the fitted value, i.e.  $\sigma_T = |T - T_{fit}|$ , in order to increase  $\chi^2$  by 1, i.e.  $\chi^2(T) - \chi^2_{min} = 1$ . Fig. 9 shows the variation in the goodness-of-fit parameter,  $\chi^2$ , as a function of temperature and relative emissivity about the minimum, with the  $1\sigma$  contour shown in black; inset shows the quadratic fit of the determined  $\chi^2$  to the temperature. Both the error associated with this method and that determined from the fitting routine are also shown as temperature error bars for comparison (black and red bar respectively). We observe that the uncertainty that we obtain using this method regularly exceeds the other estimate and therefore will set the upper limit on the temperature uncertainty.

The uncertainty in a single temperature measurement, from a single bin, is expected to be dependent on both the experimental settings, i.e streak window or MCP gain, and fitting parameters, i.e. time resolution  $t_R$ . To investigate how the fitted temperature and it's associated uncertainty is affected by changing parameters, a series of calibration spectrograms were taken, with the lamp temperature set to 2500 K or 2900 K, the MCP gain varied from 30 to 60, and using streak windows between 1 and 200  $\mu$ s. SOP spectrograms of the calibration lamp were collected in gated enclosing trigger mode with either 1, 10 or 100 integrated sweeps. Because each image will measure the same, known temperature at all times, we can examine how measurement accuracy and precision depends on time resolution (bin width), sweep window, gain, and integrated sweeps, and finally compare uncertainty obtained from individual spectrum fits as discussed above to the statistical error in fitted temperature across the image (i.e. standard deviation of fitted temperatures at many times) which represents the true measurement error (Fig. 10).

Fig. 10a shows how the fitted lamp temperature at 2900 K, collected from single sweep with 60 MCP gain and a 5  $\mu$ s streak window, varies across the time axis with changing time resolution. Regardless of time resolution we see that the average fitted temperature, across the image, is close to that of the expected lamp temperature and that the scatter in fitted temperature increases as the time resolution is reduced from 483 ns to 96 ns. Fig. 10b and c shows how the average temperature across the spectrogram and its statistical error (standard deviation, given by the error bars), varies as a function of time resolution from multiple calibration runs at both 2500 K,

![](_page_13_Figure_0.jpeg)

FIG. 10: Temperature uncertainty for changing experimental and fitting parameters for SOP data on standard lamps. Error is determined from fitting single binned spectra (least-squares or  $\chi^2$  map) and from averages of the fitted temperature over all bins in an image (statistical error, given by standard deviation). a) Temperature data on lamp set at 2900 K, over a 5  $\mu$ s streak window using a MCP gain of 60 with a time resolution of 96, 198 and 482 ns and single sweep data collection. The error bar shows the least squares error on each temperature measurement. The number of integrated sweeps are varied in further analysis (b-g) as 1, 10 or 100 sweeps (red, green and blue, respectively). b.c) Average temperature and statistical error for images of lamp at 2500 K (b) and 2900 K (c), as a function of time resolution used (bottom axis) and number of sweeps integrated (N). Dashed horizontal line indicates the set lamp temperature and the shaded region between 45 and 305 ns shows the region over which the time resolution equals that typical PSF values. Streak window is constant at 5  $\mu$ s and MCP gain at 60 (b) or 50 (c). d) Average temperature and statistical error as a function of MCP gain (bottom axis) and integrated sweeps (N) at 2500 K. Time resolution is fixed at 198 ns with a streak window of 5  $\mu$ s. At a MCP gain of 30 we were unable to measure lamp temperature under single shot conditions (N=1) because of weak signal (Sec. IIG). e) Average temperature and statistical error as a function of streak window length (bottom axis) and integrated sweeps (N) at 2900 K, with a fixed 20 pixel time resolution and using full width of the spectrogram in the time axis (1016 pixels). The MCP gain is 50. f, g) Comparison of statistical error (bottom axis) to single fit errors, based either on least-squares fitting (closed circles) or  $\chi^2$  map analysis (open squares), as time resolution, MCP gain and streak window are varied, at lamp temperatures of 2500 K (f) and 2900 K (g). Vertical lines of points indicate the individual fit error distributions for the bins used to construct a given statistical error. Dashed line indicates ideal

with MCP gain at 60, and 2900 K, with MCP gain at 50. The range of PSF values ( $PSF_{range}$ , Sec. II E) is shown for comparison. At both lamp temperatures we see that by increasing the time bin size (decreasing time resolution) we decrease the statistical error in the temperature measurement whilst maintaining average temperatures consistent with the set lamp temperature. Regardless of the number of integrated sweeps (1, 10 or 100) we still are able to extract the set lamp temperature within the statistical error. Fig. 10d demonstrates how varying the MCP gain affects average temperature, for time resolution fixed at 198 ns in a 5  $\mu$ s streak window. Generally, changing the gain has very little effect on the average fitted temperature and the associated statistical error as they remain consistent at all gain values. However, increasing the number of integrated images substantially reduces the statistical error. Therefore, in single shot mode, which is used in the majority of experimental runs, we can increase the MCP gain in order to the measure sample temperature without largely affecting the validity of the temperature measurement, which implies that additional noise associated with higher gain modes has no effect on measurements.

In all cases above, the streak window has been fixed at 5  $\mu$ s as it is the most common streak window used in the experiments discussed here. However, observing emission over a longer time window (i.e up to 200  $\mu$ s) is often needed to for longer events, such as irradiation by long XFEL pulse trains. With the lamp temperature fixed at 2900 K, MCP gain at 50 and a fixed time resolution of 20 pixels we are able to determine the effects that increasing streak window from 1  $\mu$ s to 200  $\mu$ s has on both fitted lamp temperature and its statistical error (Fig. 10e). The reduction in error as the streak window is increased, mirrors the effect of decreasing the time resolution (Fig. 10), as by fixing the time resolution at 20 pixels and increasing the streak window we are essentially increasing the time resolution. Therefore at longer streak windows it is possible to bin images by a smaller number of pixels and still maintain sufficient emission per bin in order to measure sample temperature.

It is expected that the statistical error in fitted temperature across the image should be equal to the uncertainty of single fits at a given time. By comparing the individual single fit error either from the least-squares python fitting routine or the  $\chi^2$  map to the statistical error in each image, we evaluate the two practical estimates for temperature uncertainty. Fig. 10f and g show results when the lamp is set to 2500 K and 2900 K respectively. The line at which the single fit error is exactly equal to the statistical error (1:1) is shown for reference. Generally, we see that the least-squares fitting underestimates the error by a factor of ~ 0.5 whilst the  $\chi^2$  map error can overestimate the error by up to five times the statistical error. We therefore take the least-squares fitting error as the standard estimate for the uncertainty of a single temperature measurement in our analysis; where only a single error bar is shown in a plot, it refers to this

estimate.

#### III. EXPERIMENTAL PROCEDURE

Irradiation of samples was performed during SOP observation using a nanosecond NIR optical laser pulse at 1070 nm (LH) and a femtosecond XFEL pulse train, with one, two or multiple pulses, at 17.8 keV (XH) (see Sec. IIA). Samples were free-standing or confined at high pressure in a DAC (Table III). The streak window was set to cover both sample heating and cooling. Fig. 11a and b illustrates the differences between the two different pump scenarios. For LH, a rise in temperature as the sample is irradiated is expected, after which the sample should start to cool. Peak temperatures are expected to be observed after the peak in laser power. For XH, each XFEL pulse incident on the sample will cause its temperature to rapidly increase upon exposure, after which the sample will cool. Subsequent XFEL pulses may lead to further increase in temperature if arriving before cooling is complete. An XFEL pulse train with high repetition rate results in a step-wise, saw-tooth like temperature change with time  $^{29,34}$ . When LH and XH are combined, the process is considerably more complex, as discussed below. In addition to time variation in temperature, spatial variation (and its time dependence) can be important. Integration of a single sweep of a single heating and cooling event (sequential mode), or integration of many sweeps over repeating, ideally identical events (enclosing trigger mode), are employed in different experiments, with the latter improving the signal to noise ratio<sup>64</sup>.

The SOP was first tested with the NIR pulsed laser heating system, at the Deutsches Elektronen-Synchrotron, Germany (Experiment 1) before being moved to the HED instrument hutch at the European XFEL. Results are reported for a free standing sample of  $5 \ \mu m$  thick Pt foil employing LH at a repetition rate of 1 kHz (see Sec. V A) with the laser waveform WF\_0 (Fig. 3). The streak camera was configured to operate using the enclosing trigger mode over a integration time of 1 s (collecting 1000 heating cycles) with the streak window set to 1  $\mu$ s. Sample temperature analysis is limited to 525 - 800 nm (Fig. 5).

Once installed at the HED instrument, we performed MHz repetition rate  $(1.13 - 2.26 \text{ MHz})^{30}$  pulsed Xray heating (Experiment 2)<sup>31,37,56,58,61</sup> on samples freestanding in vacuum or statically confined in a DAC (Table III). Beam energy was set to 17.8 keV, with a Gaussian profile with a 10 – 20  $\mu$ m FWHM and a pulse length of 20 fs. The number of pulses per train varied from 1 – 20 with a pulse separation of 443 or 886 ns. The maximum energy on target per pulse was ~ 240  $\mu$ J with a peak fluence of ~ 100 J cm<sup>-2</sup>. Unless otherwise stated, the wavelength range for fitting in this experiment was restricted to 575 – 775 nm and a 2 x 2 binned spectrogram image size, i.e. 672 (H) x 508 (V) pixels was used.

![](_page_15_Figure_0.jpeg)

FIG. 11: Schematic of pump-probe timings used in XFEL heating (a) and laser heating (b) experiments. Blue lines indicate the timing and pulse shape of pump pulses, with black lines showing the expected variation of sample temperature. LH pulses (b) are  $\sim 300$  ns in duration (Fig. 3) with the XH pulses (a) having a pulse length of  $\sim 20$  fs and a pulse separation of 443 ns at 2.26 MHz.

Due to the limited intensity of emission that we observed at the beginning of the experiment, the slits and pinhole in the optical path were removed mid-experiment,

increasing signal.

Further LH experiments were performed at the HED instrument, during a series of laser-heating commissioning runs in combination with XFEL radiation (experiment  $3^{60}$ ). The goal was to understand how to heat samples optically and probe noninvasively with X-rays. These experiments focused on assessing any combined effects of heating with the NIR and XFEL when both lasers are incident on the sample during collection of optical emission. The X-ray pulse train comprised of 10 - 30 pulses spaced by 443 ns, an X-ray energy of 17.8 keV and average energy per pulse on target of 126  $\mu$ J at 100% transmission, with a focal spot size of 4.5 x 5  $\mu$ m<sup>2</sup> (FWHM) and pulse length of 20 fs, and due to improvements in the beam alignment the NIR spot, the XFEL spot, and the full spatial filtering system (which was in place during this experiment) were all well aligned. The full spectrogram image size was used, i.e 1344 (H) x 1016 (V) pixels, and streak windows were set to 5  $\mu$ s. The

XFEL was focused on the upstream surface of the sample and the optical laser is aligned to the same location on the sample (and sometimes also downstream).

#### **IV. SAMPLE ENVIRONMENT**

For experiment 1, the target (DESY1) comprised a 5  $\mu$ m thick piece of Pt foil suspended in air.

For experiment 2, we report on data from a free standing foil target and foils enclosed and pressurized by a symmetric piston-cylinder type DACs (Fig. 12). For SS2, a 7.0(3)  $\mu$ m Ta foil, from GoodFellow, was mounted on a 100(20)  $\mu$ m piece of kapton tape substrate. HIBEF3 was prepared with 200  $\mu$ m culet type IIa diamonds, and a  $2 \ \mu m$  piece of Au foil was loaded inside the chamber with  $H_2O$  and compressed to 30.3 GPa. An additional DAC (D1) was prepared with 300  $\mu$ m culet type Ia diamonds, and loaded with a 3  $\mu$ m thick insulating layer of pressed MgO powder and 5  $\mu$ m Ta foil before gas loading it with Ne to 12.0 GPa. DAC HIBEF30 was prepared with  $\sim 2$ mm thick, 300  $\mu$ m culet diamonds, with type IIa downstream and type Ia upstream. Then loaded with three 20  $\mu m$  thick disks of polycrystalline San Carlos Olivine and a 5  $\mu$ m Pt foil then filled with Ne gas and compressed to 35 GPa. Some samples also contained a ruby (Al<sub>2</sub>O<sub>3</sub>:Cr) grain for use as a pressure marker<sup>74</sup>. Additional samples where only the pressure medium was targeted are discussed in Table III.

For experiment 3, two symmetric piston cylinder DACs were used. XFEL1 was prepared with 300  $\mu$ m culet type Ia diamonds and XFEL2 was prepared with 200  $\mu$ m culet type Ia diamonds. A 4  $\mu$ m thick piece of Pt foil was placed with an insulating layer of KCl on either side in both cells and compressed to 31 GPa (XFEL1) and 30 GPa (XFEL2).

Sample position was moved, where possible, between runs so that each heating cycle is on a fresh unheated sample location.

#### V. RESULTS

#### A. Pulsed Laser Heating of free-standing foil

Fig. 13 shows the collected spectrogram and the resulting temperature measurement from experiment 1, based on accumulation of 1000 heating cycles. A peak temperature of  $3701 \pm 71$  K was observed, closely corresponding to the peak in laser power whereas emission intensity peaks later in time. Good agreement of emission intensity with the expected Planck distribution is observed over the full spectral range. We expect that the sample cools to ambient conditions between kHz cycles and that each cycle reaches similar temperatures; the close agreement of the measured temperature with a Planck distribution supports this. The differences between the different fit error estimations are shown in Fig. 13b.

TABLE III: Table of target samples with conditions and streak camera setting shown. The sample, sample thickness, location (either free standing, FS, or contained at pressure in a Diamond-Anvil Cell, DAC), pressure and diamond type are tabulated alongside the source parameters and the streak window used and the number of integrated sweeps used in each experimental run. Samples (S) or Pressure Medium (M) are either irradiated by the optical laser, operating in either single-sided (SS) or double-sided (DS) mode, the XFEL or both. Sample are irradiated at vacuum unless indicated.

| Target/Cell                  | DESY1           | SS2  | HIBEF3           | D1   | 1VAS    | SB001 | HIBEF2         | HIBEF15 | VP002             | HIBEF30 | HIBEF24   | XFEL1  | XFEL2      |
|------------------------------|-----------------|------|------------------|------|---------|-------|----------------|---------|-------------------|---------|-----------|--------|------------|
| Experiment                   | 1               | 2    | 2                | 2    | 2       | 2     | 2              | 2       | 2                 | 2       | 2         | 3      | 3          |
| Prepared by                  | EuXFEL          | UoE  | DESY             | LLNL | UoE     | SBU   | UoE            | LLNL    | APS               | Lille   | IMPMC     | EuXFEL | EuXFEL     |
|                              |                 |      |                  |      |         | Samp  | ole            |         |                   |         |           |        |            |
| Sample                       | Pt              | Ta   | Au               | Ta   | Various | Fe    | S/Se           | Mo      | Au                | Olivine | Fe        | Pt     | Pt         |
| Thickness $(\mu m)$          | 5               | 7    | 2                | 5    | Various | 4     | 30/20          | 12      | ?                 | 20      | 10        | 4      | 4          |
| Location                     | FS <sup>a</sup> | FS   | DAC              | DAC  | DAC     | DAC   | DAC            | DAC     | DAC               | DAC     | DAC       | DAC    | DAC        |
| Pressure Medium              |                 |      | H <sub>2</sub> O | MgO/ | NaCl    | $N_2$ | H <sub>2</sub> | MgO/    | $H_2O/$           | Ne      | Ne        | KCl    | KCl        |
|                              |                 |      |                  | Ne   |         |       |                | Ar      | n-Au <sup>b</sup> |         |           |        |            |
| Pressure (GPa)               |                 |      | 30.3             | 12   | 0.8     | 5     | 0.5            | 50      | 14.4              | 35      | 60        | 31     | 30         |
| Diamond Type (US)            |                 |      | IIa              | Ia   | Ia      | Ia    | Ia             | Ia      | Ia                | Ia      | Ia        | Ia     | Ia         |
| Diamond Type (DS)            |                 |      | IIa              | Ia   | Ia      | Ia    | Ia             | Ia      | Ia                | IIa     | Ia        | Ia     | Ia         |
| Irradiated Material          | S               | S    | S/M              | S/M  | М       | Μ     | M              | М       | Μ                 | S/M     | S/M       | S/M    | S/M        |
| Source                       |                 |      |                  |      |         |       |                |         |                   |         |           |        |            |
| XFEL                         |                 |      |                  |      |         |       |                |         |                   |         |           |        |            |
| Photon Energy (keV)          |                 | 17.8 | 17.8             | 17.8 | 17.8    | 17.8  | 17.8           | 17.8    | 17.8              | 17.8    | 17.8      |        | 17.8       |
| Rep. Rate (MHz)              |                 | 1.13 | 2.26             | 2.26 | 2.26    | 2.26  | 2.26           | 2.26    | 2.26              | 2.26    | 2.26      |        | 2.26       |
| Pulses/Train                 |                 | 1    | 1 - 5            | 4    | 1       | 1     | 1              | 1       | 1                 | 20      | 1         |        | 10 - 30    |
| Energy/Pulse $(\mu J)^c$     |                 | 293  | 506              | 720  | 413     | 801   | 483            | 500     | 672               | 567     | 536 - 689 |        | 420        |
| Filter Transmission          |                 | 0.5  | 1                | 1    | 1       | 1     | 1              | 1       | 1                 | 1       | 0.1 - 1   |        | 0.12 - 0.5 |
| Energy on Target $(\mu J)^d$ |                 | 44   | 152              | 216  | 124     | 240   | 145            | 150     | 202               | 170     | 21 - 161  |        | 15 - 63    |
| Optical Laser                |                 |      |                  |      |         |       |                |         |                   |         |           |        |            |
| Waveform                     | WF_0            |      |                  |      |         |       |                |         |                   |         |           | WF_0   | WF_0       |
| Pulse Length                 | 250 ns          |      |                  |      |         |       |                |         |                   |         |           | 250 ns | 250 ns     |
| Single / Double Sided        | SS              |      |                  |      |         |       |                |         |                   |         |           | SS     | DS         |
| Streak Camera Setting        |                 |      |                  |      |         |       |                |         |                   |         |           |        |            |
| Streak Window                | 1               | 5    | 5                | 5    | 5       | 5     | 5              | 5       | 5                 | 200     | 5         | 5      | 5          |
| Integrated Sweeps            | 1000            | 1    | 1                | 1    | 1       | 1     | 1              | 1       | 1                 | 1       | 1         | 1000   | 1          |

<sup>a</sup> Experiment conducted in air

<sup>b</sup> Pressure media contains nanometer sized particles of Au

<sup>c</sup> SASE XGM (Upstream)

<sup>d</sup> Energy/Pulse \* Filter Transmission \* Beamline Loss Factor (0.3)

#### B. XFEL heating of free-standing foil

Target SS2 underwent irradiation of a single XFEL pulse at 50% X-ray transmission, which corresponded to a pulse energy of  $\sim 44 \,\mu\text{J}$  on target (experiment 2). Using two different time resolutions, 135 and 193 ns, we determine peak temperatures of  $\sim 6500$  K (Fig. 14). Emission was observed for  $\sim 1 \,\mu s$  after exposure (Fig. 14a). A large uncertainty ( $\sigma_{\rm T} > 10\%$ ) correlates with a weak and noisy emission in this case<sup>59</sup>. That is, despite reaching higher temperature than both the calibration lamp and laser-heated foil in preceding example, signal is lower due to only single-shot collection, small emitting area, and shorter emission timescale. Validity of the temperature estimation is confirmed by the damage imprinting observed during post-experiment analysis (Fig. 14d). The observed hole and drip features formed suggest that there was sufficient heating to melt a large area of the target consistent with high peak temperatures measured well in excess of the melting point. The short emission timescale can be explained by rapid target quenching (Sec. VI).

#### C. XFEL heating of a sample in a diamond anvil cell

HIBEF3 underwent irradiation by a single train of five XFEL pulses at 100% X-ray transmission and 2.26 MHz (experiment 2) with an average pulse energy on target of  $152\pm81 \ \mu$ J. Two different time resolutions, 96 ns and 135 ns, were used to evaluate the temperature from the SOP (Fig. 15). This cell was loaded with type IIa diamond anvils, and no additional background from fluorescence is observed (see Sec. IID). In Fig. 15a it can clearly be seen that as each XFEL pulse interacts with the target there is an increase in emission intensity on the SOP spectrogram, followed by a drop in intensity between pulses, as expected when the Au foil cools. We observe a peak temperature of  $\sim 4500$  K and minimum measurable temperatures of around  $\sim 2500$  K. Fig. 15c highlights how the wavelength dependence of emission intensity changes with both time and temperature.

The temperatures achieved indicate that the Au sample was partly melted during the experiment, as temperatures exceed the known melting point of Au at 30 GPa  $(T_{melt} \sim 2300 \text{K})^5$ . We can see from the temperature

![](_page_17_Figure_0.jpeg)

FIG. 12: Schematics and photo-micrographs of free-standing and DAC samples used during experiment 2. a) Schematic and photo-micrograph of free-standing Ta sample (SS2). b) Schematic and photo-micrographs of DAC sample configuration used during experiment 2. Photo-micrographs of samples in DACs (HIBEF3, VP002, D1 and HIBEF30) are shown, with the gasket hole highlighted by dashed circle. Sample thickness (t<sub>sample</sub>), sample chamber thickness (t<sub>indent</sub>), sample chamber diameter and diamond culet diameter are shown. Both schematics are not to scale. All photo-micrographs are viewed from the upstream side, with respect to the XFEL beam, and are illuminated by transmitted and reflected light.

profile that subsequent pulses don't step-wise heat the sample as expected<sup>29</sup>, with peak temperatures of  $\sim 4500$ K at the time of pulses 2-5. This could be due to the specific energies of each pulse in the train, disruption of the Au upon melting $^{27}$ , or increase in the opacity of the  $H_2O$  under high pressure and temperature<sup>21,75</sup> leading to limitations on apparent temperature<sup>9</sup>. It is also notable that peak emission intensity increases with time, despite no clear increase in peak temperature, which can be attributed to increased emitting area (i.e. due to lateral heat conduction) or emissivity (i.e. due to optical changes in pressure medium). Similar effects are detected in other targets. Compared to the freestanding foil, there is a more gradual decay of the emission intensity after heating, likely a result of better confinement and insulation of the sample during quenching.

#### D. Characterizing X-ray fluorescence

The preceding examples lacked discernible fluorescence, and emission was interpreted as purely thermal radiation. We now discuss targets having significant Xray fluorescence background (Sec. IID). For example, Fig. 16a and b compares SOP records for irradiation by two XFEL pulses, with an average energy on target per pulse of  $\sim 135 \ \mu$ J, where a difference in the observed optical emission between the similar samples and diamond types (Type Ia in VP002 and Type IIa in HIBEF3) can be seen. In both cases, we are observing emission from a Au sample in H<sub>2</sub>O pressure medium, at 37.0 and 30.3 GPa, respectively.

Whilst the SOP spectrogram from the sample in DAC VP002 shows emission across all wavelengths below 850 nm, there is very little or no emission detected for sample of HIBEF3. As the only difference between VP002 and HIBEF3 is the diamond type, we conclude that Fig. 16a shows the fluorescence signal from the diamonds in VP002. In terms of the spectral response we see that in the region where  $\lambda < 575$  nm and  $\sim 800$  nm there are peaks in the emission intensity, which can be characterized by fluorescence emission as seen from a variety of targets (Fig. 8). The optical emission from the sample in VP002 demonstrates a symmetrical shape with respect to time, which as discussed represents the PSF of the SOP.

The weak emission from the sample in HIBEF3 at low XFEL pulse energy (Fig. 16b) shows only emission from the second pulse and with a very different temporal and

![](_page_18_Figure_1.jpeg)

FIG. 13: Optical emission collected from laser-heated platinum foil (5  $\mu$ m thick, DESY1) using a 1  $\mu$ s streak window and enclosing trigger mode. Time resolution is set to 39 ns. a) SOP Spectrogram with wavelength axis restricted to 525 – 800nm. Relative intensity profile with respect to the time axis (averaged over this wavelength range) is shown in black. b) Final temperature history with both temperature error estimations shown. Error from least-squares fitting is shown as the red error bar and green is the error associated with the contour on the chi-squared map. Emission intensity is the black curve and laser pulse intensity is the filled grey profile (timing estimated from the onset of emission.). c) Binned optical emission at time = 0.25  $\mu$ s. Planck fit to data determined through the least squares fitting (red dashed line) at  $T_{\rm fit} = 3385 \pm 28$  K. d) Binned optical emission at time = 0.61  $\mu$ s. Planck fit to data determined through the least squares fitting (red dashed line) at  $T_{\rm fit} = 2032 \pm 39$  K. c) and d) show the original uncorrected emission data (grey triangles), the corrected emission after taking into account the transfer function of the system (dark grey filled circles) and the final smoothed emission which is binned in time and wavelength (solid black line). The average standard deviation in emission intensity during smoothing is shown as a single black error bar.

spectral shape than that observed from VP002 – suggestive of thermal emission from cumulative heating. That emission here is purely thermal in nature is further clarified by increasing the XFEL energy on this sample (Fig. 16c), i.e. to  $245\pm53 \ \mu$ J/pulse. The emission is asymmetric with respect to the time axis, with a tail in emission intensity after each XFEL pulse incident on the target, likewise the emission peaks at around ~ 700 nm in the wavelength axis, similar to the spectral shape of the calibration lamp (Fig. 8). Finally, a cumulative increase in emission intensity showing stronger emission from the second pulse is clearly resolved, following expectations of step-wise heating. The peak temperature, after the sample in HIBEF3 has interacted with the two XFEL pulses, is measured to be  $5282\pm650$  K.

Whilst fluorescence is often of some use, such as for alignment<sup>5</sup> and determining experimental timings (Sec. II E), it can potentially interfere with pyrometric tem-

perature measurements. This is because the measured spectrogram can become a combination of thermal and fluorescence emission at higher powers. Here we explore how, when there is a large fluorescence signal, e.g. when using type Ia diamond anvils, the fluorescence component can be removed before temperature measurement. We thus examine whether and how the fluorescence signal would distort the spectral shape and temperature being measured.

To remove the fluorescence signal, we have developed a method to create a synthetic SOP image that simulates the fluorescence emission for a particular sample configuration. By measuring the optical emission from the same target at low XFEL beam transmission, with the same optical configuration and sample containment, and varying numbers of XFEL pulses we can survey the fluorescence emission shape in both wavelength and time before the onset of any thermal emission. The streak window

![](_page_19_Figure_1.jpeg)

FIG. 14: Optical emission from single-pulse XFEL irradiation of tantalum foil (7  $\mu$ m) in vacuum (SS2) with streak window set to 5  $\mu$ s. Wavelength range for fitting is restricted to 575 nm – 775 nm. a) SOP spectrogram with integrated intensity for reference (black). b) Temperature history using two different time resolutions, 135 ns (green) and 193 ns (blue), with relative intensity of emission for reference. Both error estimations for every temperature are shown (red is the least-squares error and green/blue shows the error determined from the contour  $\chi^2$  goodness-of-fit map). The time at which the XFEL pulse is incident on the sample is shown as a single vertical dashed line (based on fluorescence timing observed in other targets). c) Grey-body spectral fits to optical emission. Red and blue

dashed lines are the Planck distribution at  $T_{fit} = 5990 \pm 530$  K and  $T_{fit} = 2710 \pm 240$  K, respectively, and the black solid line shows the wavelength-averaged optical emission. d) Post-beamtime SEM image of damaged target area.

used in each run is kept constant throughout and all runs are recorded in sequential (single sweep) mode in order to correlate observed fluorescence with the power of each individual XFEL pulse.

The time evolution of fluorescence emission under pulsed XFEL heating is determined by multi-parameter least-squares fitting  $^{67-69}$  of a superposition of Lorentzian distributions (one Lorentz distribution is used for every pulse in the XFEL pulse train) to the wavelengthaveraged spectrogram emission intensity in the UV region (451 nm  $< \lambda < 545$  nm; Fig. 17 c and d). No pincushion effect was accounted for prior to fitting. Peak separation in each fit is fixed and corresponds to the time between XFEL pulses in a train (here 443 ns). At low XFEL transmission there is minimal or no thermal emission so we observe a good fit between the Lorentzian curves and emission intensity with time (Fig. 17c). Whereas at higher XFEL beam intensity and number of pulses there is only good agreement between the Lorentzian distribution and emission intensity at the start of the record (time  $< 0.75 \ \mu s$ ) where only a few pulses have irradiated the sample, typical of a signal dominated by

fluorescence (Fig. 17d). That is, optical emission from later pulses (time > 0.75  $\mu$ s) do not exhibit a good fit to the expected Lorentz shape and are seen to reach peak intensity later than expected from the pulse separation and have a longer decay time, indicating thermal emission contribution. This gives us a qualitative measure to distinguish between different emission types. After the Lorentz fits from multiple runs have been manually filtered, to only include fluorescence pulses, we can determine the average FWHM, or PSF, and amplitude of fluorescence emission in the time axis. The relationship between emission intensity and XFEL pulse energy, as measured by the beamstop photo-diode "PD\_3" (Figs. 1b and  $17e)^{18,37,58,59,76}$ , is determined to be a power-law fit (Intensity =  $0.072 \cdot PD_{-3}^{0.83}$  for D1, Fig. 17f). We can therefore determine the expected fluorescent emission intensity from the photo-diode signal for any run using this sample. Spectral shape is determined through a least-squares fitting of time-averaged emission intensity from the accumulated spectral shape from multiple fluorescence-only runs, to five superimposed Lorentzian distributions (Figs. 8 and 17b). We expect the results

![](_page_20_Figure_0.jpeg)

FIG. 15: Optical emission from five-pulse X-ray irradiation of gold foil (2  $\mu$ m) in a DAC with a H<sub>2</sub>O pressure medium held at 30.3 GPa (HIBEF3). The streak window is set to 5 $\mu$ s and the time resolution of the temperature measurements are 96 ns and 135 ns. The wavelength range used for fitting is 575 – 775 nm.

a) SOP spectrogram image of HIBEF3, with the relative emission intensity, averaged over wavelength range used for fitting shown for reference (black). b)

Temperature history of Au foil measured with two different time resolutions (96 ns red; 135 ns blue), with relative intensity of emission for reference (black). The times at which the XFEL pulse train is incident are shown as dashed vertical lines. c) Grey-body spectral

fits to optical emission. Red and blue dashed lines are the Planck distribution at  $T_{\rm fit} = 5650 \pm 760$  K and  $T_{\rm fit} = 2780 \pm 250$  K respectively. The black solid lines are the moving wavelength-averaged optical emission for each time step.

![](_page_20_Figure_5.jpeg)

FIG. 16: Optical emissions observed from two-pulse XFEL irradiation of gold foil (2  $\mu$ m) in a DAC with a H<sub>2</sub>O pressure medium from VP002 and HIBEF3. The upstream diamond type (DT) varies. During collection of emission the spacial filtering was removed and the streak window was 5  $\mu$ s. a) SOP spectrogram of optical emission when cell is loaded with Type Ia diamonds (VP002, upstream diamond thickness = 2.201 mm) at 37.0 GPa. Average energy per pulse is 134 ± 8  $\mu$ J. b) SOP spectrogram of optical emission when cell is loaded with a Type IIa diamond (HIBEF3, upstream diamond thickness = 2.275 mm) at 30.3 GPa. Average energy per pulse is 135 ± 23  $\mu$ J. c) SOP spectrogram of optical

emission from HIBEF3 irradiated with two XFEL pulses with an average energy per pulse is  $245 \pm 53 \ \mu$ J.

of the above fitting routines to depend highly on sample configuration, window thickness, materials present in the beam, etc., and so should be analysed independently for each sample.

A synthetic fluorescence-only SOP image is then created using the average FWHM emission duration, amplitude of emission with respect to photo-diode intensity, and spectral shape of emission (Fig. 17g). Subtraction of the synthetic fluorescence emission from the raw SOP image yields a modified SOP image which is used for temperature determination (Fig. 17h).

### E. XFEL heating of a diamond anvil cell after fluorescence removal

An example of a target with considerable fluorescence emission is DAC D1 (5  $\mu$ m Ta foil in MgO/Ne medium at 12 GPa in a DAC with Type Ia diamonds). After removal of fluorescence emission (Fig. 17, see Sec. VD), sample temperature is determined for four pulse XFEL irradiation at 100% X-ray transmission (216 ± 26  $\mu$ J/pulse) (Fig. 18).

Here we can see that emission from the first pulse is entirely is due to fluorescence; prior to fluorescence removal, peak temperatures of 4000 – 6000 K were suggested, however after correction the temperature is too low to detect  $(\lesssim 2500 \text{ K})$ . The second pulse shows comparable levels of fluorescence and thermal emission; however, the apparent peak temperature rises from  $\sim 4000$  K to  $\sim 5000$ K upon fluorescence correction. Hence, the effect of fluorescence on apparent temperature is not straightforward. Subsequent pulses (3 and 4) show much higher thermal emission than fluorescence, with the correction having a negligible effect. Finally, there is a long tail of thermal emission after the pulses which is not affected by fluorescence background and is unaffected by correction. As in the prior example of a DAC under XFEL irradiation (Sec. VC), peak temperatures reached after each pulse remain relatively constant later in the pulse train, while emission intensity rises.

## F. XFEL heating with longer pulse trains and SOP windows

With large numbers of pulses available at high repetition rates at modern XFEL sources (e.g. 2700 pulses at 4.5 MHz at European XFEL), there is much potential in SOP in dynamic experiments on longer timescales. However, as the streak window length is increased, the spacing, in pixels, between successive XFEL pulses is reduced. This leads to poor definition between the optical emission of neighboring peaks and the expected saw-tooth like shape of emission intensity with time is smoothed out. In addition, it may be more difficult to discriminate and model fluorescence background. This means at longer streak windows, we cannot easily identify emission from individual XFEL pulses in the pulse train.

An example of this scenario is given here for an oxide mineral sample (olivine) with low absorbance and slow thermal transport dynamics at 35 GPa in a DAC (HI-BEF 30), undergoing cumulative heating over 20 XFEL pulses at 2.26 MHz with a streak window of 20  $\mu$ s (Fig. 19). Fluorescence emission is evident from additional datasets collected at lower power, which shows characteristic strong emission in the blue.

To determine the expected fluorescence emission in this case, we surveyed the optical emission from a short streak window, e.g 5  $\mu$ s, at varying XFEL pulse energies and converted it to the longer streak window during modelling, e.g 20  $\mu$ s. The spectral shape and time dependence of the expected fluorescence spectrum is determined the same way as described as above. To adapt from a 5  $\mu$ s window to a 20  $\mu$ s window, we assume the spectral and temporal shape (Fig. 19b), PSF (Sec. IIE) and amplitude of emission intensity dependence on photodiode values is independent of the streak window but the spacing, in pixels, between XFEL pulses is reduced. The resultant fluorescence emission in the time domain (Fig. 19a and d) shows a continuous distribution throughout irradiation.

After removal of the fluorescence emission the thermalonly signal is observed (Fig. 19e) and can be used to measure sample temperature (Fig. 19f). By comparing the SOP spectrogram measured before (Fig. 19c) and after (Fig. 19e) the fluorescence emission is removed we can clearly see that the emission on the thermal only image is only significantly changed during the first 5  $\mu$ s. This suggests that the emission in the first 5  $\mu$ s is dominated by fluorescence and that at later times thermal emission is dominant. Thus only temperature measurement obtained early in the record differs significantly before and after fluorescence emission is removed whilst there is minimal change in temperature at later times – and none at all after the XFEL pulse train is completed (Fig. 19f). A stepwise heating dynamic is clearly observed in both the temperature and emission records after correction.

#### G. Laser heating of a diamond anvil cell

We first discuss results for pulsed LH of DAC samples without any X-rays (Fig. 20). Optical emission collected from XFEL1 under pulsed LH with 1000 heating cycles (1 s exposure at 1 kHz streak repetition rate) using the enclosing trigger mode is shown in Fig. 20b with fits to Planck's Law (Eq. 1) shown in Fig. 20c. A peak temperature of  $7800 \pm 400$  K is observed at the peak of emission at  $0.12 \ \mu$ s after the start of the laser pulse. The timing of the laser pulse is estimated from the onset of optical emission. Upon repeating this LH run multiple times we observe that there is a large scatter in the measured temperature (Fig. 21), which can be attributed to changes in the sample coupling, which may be due to repeated melting. In all runs there is an unusual broadening of

![](_page_22_Figure_0.jpeg)

FIG. 17: Thermal and fluorescence optical emission from XFEL irradiation of D1 (Ta foil at 12 GPa). Streak window is set to 5  $\mu$ s throughout. a) SOP spectrogram image of four-pulse irradiation on Ta, at 100% transmission of the XFEL beam with relative intensity for reference (black). b) Lorentz least-squares fitting to the time-averaged optical fluorescence emission from DAC D1, grey markers (overall fit shown as black dashed, individual Lorentz fits are shown in solid color lines). c) Lorentz least-squares fitting to the wavelength-averaged optical emission from two pulse irradiation of D1 at 50% XFEL beam transmission (run 837), grey markers (overall fit shown as black dashed, individual Lorentz fits are shown in solid color lines). The timing of the XFEL pulses are shown as dashed vertical

lines. d) Lorentz least-squares fitting to the wavelength-averaged optical emission from a four pulse irradiation of DAC D1 at 100% XFEL beam transmission (run 845), grey markers (overall fit shown as black dashed, individual

Lorentz fits are shown in solid color lines). The timing of the XFEL pulses are shown as dashed vertical lines. e) Photo-diode signal showing beam intensity from the runs shown in (c) and (d) f) Photo-diode intensity (shown in e) against fitted Lorentz distribution amplitude of fluorescence emission (c) and (d), red markers, and fitted power-law distribution, red dashed line. Envelope indicates a 20% error on power-law fit. g) Synthetic SOP image constructed based on time evolution (c), spectral shape (b) and beam intensity (f). Relative emission intensity is shown in black. h) Modified SOP image showing thermal only emission from Ta foil with relative emission intensity is shown in black.

![](_page_23_Figure_0.jpeg)

FIG. 18: Pyrometric temperature estimation of a four bunch irradiation of 5  $\mu$ m Tantalum foil at 12 GPa (D1), run 845, after fluorescence removal (Fig. 17). a) SOP spectrogram after fluorescence removal showing region of interest used to determine sample temperature (575 – 775 nm). Relative intensity of emission shown in black. b) Temperature history before (blue) and after

(red) fluorescence removal. The relative emission intensity (post-fluorescence removal) is shown in black, with the fluorescence signal shown as a dashed line and original emission intensity shown in grey. The times at which the XFEL pulse train is incident on our sample is shown as dashed vertical lines. c) Least-squares fit to Planck's Law at time = 1.29, 1.39 and 1.49  $\mu$ s. Time resolution is 96 ns in this case. Systematic deviation from the Planck fit at lower temperature (upward curvature) may be related to increasing emissivity variation in the visible range at low temperature<sup>77</sup> compared to high temperatures<sup>22</sup>. emission in time, such that emission appears well before the start of the laser pulse with relatively high apparent temperatures. This is possibly due to a bleeding effect occurring due to the large number of sweeps collected in the single image<sup>64</sup>; more study is required to understand this issue. However, it does not appear to play a role in single-shot collections.

### H. Laser heating of a diamond anvil cell with XFEL irradiation

Installation of the laser-heating system at the HED hutch allowed LH of samples in conjunction with X-ray irradiation. Fig. 22 shows an example of LH combined with X-ray irradiation of a pre-compressed Pt foil in a DAC with KCl pressure medium at 30 GPa (XFEL2). The sample was irradiated by a single NIR laser pulse and 30 XFEL pulses at 12% X-ray transmission. The relative timing of the laser and XFEL is estimated from photo-diode traces where both X-ray and optical laser were detected (e.g. Fig. 22b), as well as from peak positions in SOP data. Here the first XFEL pulse is delayed by -510 ns from the start of the optical laser pulse. Instead of extending the streak window to encompass the whole heating event, i.e. all 30 XFEL pulses, the streak window only recorded the first few pulses.

With the X-ray transmission at a level where no optical emission is observed, it is desired that only the laser pulse will heat the sample whilst the XFEL pulses will only probe the induced states. From the intensity profile, we observe a steady decrease in emission with time suggesting only laser-heating is occurring. From the temperature profile, however, we observe oscillations in the temperature after the laser pulse – not unlike the behavior shown above for X-ray heated targets, and at higher X-ray power on this target (see below). While subtle, the timing of these excursions, and their rather large amplitude, is consistent with significant XH.

At higher XFEL transmission, combined XH and LH clearly occurs in the target. Fig. 23 shows an example of laser heating with X-ray pulse exposures at 50% X-ray transmission, on the sample in XFEL2. The XFEL pulse train consists of 10 pulses and the delay of the first XFEL pulse from the start of the optical laser pulse was +60 ns.

Several noteworthy features are seen in the data. High temperatures are rapidly achieved by the transient optical pulse at the beginning of the experiment, as well as high emission. This heat pulse decays with time after the laser shuts off. The contribution from X-ray heating becomes clearly visible after about 1  $\mu$ s. In comparison to lower power data on this sample, XH roughly maintains the temperature after it has decayed to a certain level after the optical laser pulse, but notably, does not sustain the peak temperature, while emission intensity drops off considerably. A good alignment of the NIR laser with the X-ray beam is confirmed by clear in-situ detection

![](_page_24_Figure_1.jpeg)

FIG. 19: Thermal and fluorescence optical emission from HIBEF30 (Olivine in Ne at 35 GPa) under irradiation by 20 XFEL pulses at 100% transmission with a 20  $\mu$ s streak window. The average energy on target across the 20 XFEL pulses was 170  $\pm$  68  $\mu$ J. a) Original SOP spectrogram of run 781 with the relative emission profile shown in black. b) Lorentz fit to fluorescence emission spectrum of HIBEF30. Dashed black line shows the best fit line, with five individual Lorentzian peaks shown in color. c) Synthetic SOP image constructed by combining spectral shape (b) and time dependence (d) of fluorescence emission; time-dependent fluorescence emission profile is shown in

(b) and this dependence (a) of indeceeded emission, once dependent indeceeded emission proves is brown in black. d) Model time dependent fluorescence emission for HIBEF30. The black solid line shows cumulative fluorescence emission, with individual pulses shown in color. e) Corrected SOP spectrogram from run 781 with fluorescence emission removed. f) Measured sample temperature of Olivine under X-ray heating. The measured sample temperature determined before (red) and after (blue) fluorescence emission is removed is shown. The times at which 20 XFEL pulses are incident on the sample are shown as vertical dashed lines with the emission intensity before (grey) and after (black) fluorescence removal.

![](_page_25_Figure_1.jpeg)

FIG. 20: Pyrometric temperature measurement of pre-compressed Pt with KCl pressure media at 31 GPa (XFEL1) under pulsed LH. The recorded SOP spectrogram shows the integrated optical radiation collected over 1s. a) SOP image of region of interest with relative intensity of emission shown in black. b) Temperature history with relative emission intensity shown in black.  $1\sigma$  uncertainties are shown. The estimated time at which the laser pulse is incident on Pt sample is shown. Error bars represent the uncertainty in temperature measurement determined from the least-squares fitting. c) Least-squares temperature fit of emission spectrum at time = 0.1,

0.29 and 0.39  $\mu$ s using a time resolution is 96 ns.

of laser heating by simultaneous X-ray diffraction. The large difference in emission amplitude in the LH sample compared to XH sample, at a similar temperature, is likely due to the difference in hotspot size. That is, if the LH spot is larger by a factor of  $\sim 3$  (Sec. II A),

![](_page_25_Figure_5.jpeg)

FIG. 21: Reproducibility of pyrometric temperature of of pre-compressed Pt with KCl pressure media at 31 GPa (XFEL1) from 16 heating runs using identical NIR laser intensity. The average temperature is shown, with the standard deviation in temperature measurements plotted as the error bar. Individual heating runs are plotted as solid colored lines with the maximum and minimum measured temperature (dashed black lines).

the emission intensity should be greater by a factor of 9, which is similar to what we observed. In this experiment one can observe the transition from emission over the larger optical laser heating spot to a smaller hotspot associated with the X-ray beam. Also, rather than sustain the high initial temperature achieved by the optical laser, X-ray heating seems to achieve lower temperatures. This could be due to the conditions adopting a limiting X-ray train heating temperature for this sample<sup>29</sup>, which is lower than that achievable by the optical laser – that is, pulse heating and inter-pulse cooling by X-rays reach a balance at longer timescales.

Finally, there is no detectable fluorescent emission despite using a Type Ia diamond in this experiment, with emission in both types of heating (LH and XH) equally well described as thermal only. This indicates that the spatial filtering system reduces the contribution of fluorescent emission from diamonds in the SOP spectrum.

#### VI. DATA MODELLING

Given that temperature measurements are deployed here in new experimental scenarios where they have been previously untested, we applied numerical finite element analysis (FEA) to representative cases. Examples of data

![](_page_26_Figure_0.jpeg)

FIG. 22: Optical emission from LH of Pt foil in a DAC with a KCl pressure medium held at 30 GPa (XFEL2) which is simultaneously probed by an XFEL pulse train, consisting of 30 XFEL pulses. Streak window is set to 5  $\mu$ s and time resolution of 48 and 96 ns. a) SOP spectrogram image in region of interest used for fitting the Planck spectrum with relative intensity for reference (black) b) Upper panel : Photo-diode signal showing the total signal (red) with the XFEL beam intensity (black). Lower panel : Temperature history at two different time resolutions (48 and 96 ns), with relative intensity of emission for reference (black solid line). Laser pulse used is shown in grey and the times at which the XFEL pulse train is incident on our sample is shown as dashed vertical lines. c) Grey-body spectral fits to optical emission. Red and blue dashed lines are the Planck distribution, black solid line is the wavelength-averaged optical emission in both cases.

modelling for more traditional experiments (with optical laser radiation only) are discussed elsewhere<sup>9,11,64,72</sup>. That is, while the observations presented here are consistent with qualitative expectations for the heating and

![](_page_26_Figure_3.jpeg)

FIG. 23: Optical emission from optical laser and X-ray heated Pt foil (4  $\mu$ m thick), in a DAC with a KCl pressure medium held at 30 GPa (XFEL2), at higher X-ray transmission than Fig. 22 (see this figure for additional details). a) SOP spectrogram image. b) Upper panel : Photo-diode record. Lower Panel: Temperature history. c) Least-squares temperature fits.

cooling behavior of samples under XFEL, optical, or combined radiations sources (Fig. 11), we further compare the observations quantitatively with the expected behavior<sup>11,29</sup>, firstly as predictions, and in cases where the prediction fails, by optimizing parameters to improve agreement between the model and the data.

In our models, we assume the model geometry is fixed and employ an isobaric approximation for thermal analysis, i.e. that the heating and cooling can be treated as if it occurred at constant pressure<sup>29</sup>. This is suitable for making comparisons with the presented datasets. Temperature changes in the sample after initial heating are due to heat diffusion through the target; radiative heat losses from the sample (by way of thermal emission) have a negligible effect on the results (confirmed by running models with and without radiative losses). Inclusion of

![](_page_27_Figure_0.jpeg)

FIG. 24: Geometric configuration of finite element models with mesh, and temperature distribution at peak achieved temperature in each experiment,  $T_{max}$ .

latent heat effects<sup>37,78</sup> had a minor effect on the models, but as this did not significantly affect first-order comparisons with the data, it was left out in the models reported here. Temperature dependencies of parameters were included where available, whereas pressure changes with temperature were expected to have only a minor effect and were not included.

The models account for the incident energy per pulse as measured in the beamline, with a factor accounting for transmission to the sample, which is nominally  $\sim 30\%$ of the upstream pulse energy monitor (SASE XGM<sup>58</sup>) as determined in separate analysis of damage imprinting on freestanding foils<sup>37,59</sup>. In this way, the distribution of conditions achieved are predicted from the known beam properties, geometry of samples, and sample properties at relevant pressures and temperatures (Table IV).

Modeled temperatures are reported for the peak upstream temperature for metallic samples and a range of axial temperatures for transparent samples. For heated metal layers, the SOP data will be generally dominated by the maximum temperature on the observed (upsteam) surface due to the strong scaling of emission intensity with temperature in the studied temperature range. For nonmetallic targets emission from a wider range of locations may contribute to observed values. We also report a bulk averaged temperature within the primary sample layer, weighted by beam intensity, to represent the apparent temperature as would be observed by X-rays probing the bulk through X-ray pulses. This is computed as

$$T_X = \frac{1}{2\pi\sigma_G^2 d} \int_0^R \int_{-\frac{d}{2}}^{\frac{d}{2}} 2\pi r \ T(r,z) \exp\left(-\frac{r^2}{2\sigma_G^2}\right) dz dr$$
(5)

where  $\sigma_G$  is the Gaussian width parameter of the beam<sup>29</sup>, d is the thickness of the sample, and the sample radius is R, with the normalizing prefactor assumes  $R >> \sigma_G$ . Where the latter requirement is not fulfilled the initial  $T_X$ falls below the initial (ambient) value i.e.  $T_0$ ; a correction factor of  $T_0/T_X$  from the start of the simulation can be globally applied to the  $T_X$  as a correction. This is only relevant to very narrow samples: e.g. for the 25  $\mu$ m diameter Olivine discs compressed in DAC HIBEF30, comparable in size to the X-ray beam, the correction factor is 300/245 = 1.23 while computed to unity for the other samples analysed with the finite element technique.

Model results for 4 datasets are shown in Fig. 25 with sample and model parameters given in Table IV. Model parameters are initially set based on the known geometry, pressure, and literature properties of samples under pressure and temperature; where the model fails to predict the data, certain parameters are optimised to improve the fit as shown in the table.

Comparison of the models with the data and the meaning of this analysis is discussed in Sec. VIIE.

#### VII. DISCUSSION

#### A. Low signal performance

The minimum detectable temperature of the SOP diagnostic outlined in this study ranges from 1500 – 3000 K for microsecond-duration events. Below this limit greybody Planck fits become dominated by background (for well subtracted background this is close to zero counts) resulting in a flat Planck fit determining apparent temperatures in the 3000 – 4000 K range. Screening of the data based on rigorous criteria is needed to identify and exclude such data, as minimum detectable temperatures can be even lower than the temperatures spuriously obtained when no signal is present. That is, the apparent temperature alone is not a reliable indicator of accuracy or a direct indicator of the detection limit. For example, while low (< 3000 K) and high (> 4000 K) temperatures can be judged as plausibly accurate on the basis of fitted temperature alone, these midrange temperatures can also be mimicked through the background (i.e. absence of signal).

Additional insight is gained by considering the character of the data as the detection limit is passed. Streak camera low intensity performance depends on intensifier gain and other factors, however some broad trends are observed for this Hamamatsu device. At low signal levels, signal is dominated by individual intensifier events, leading to a spotty appearance of the streak image, with spectral lineouts showing sharp peaks over a low (zero) background. Fits to the data at this condition are dominated by background. As signal level is raised, the number of individual intensifier events becomes sufficient to populate more of the image area and, depending on size of a time bin, the spectral lineout. Once a statistically sufficient number of intensified events are present in the binned spectrum, fits become dominated by the signal and begin to accurately represent temperature (as confirmed using our analysis on standard lamps, Fig. 10). A portion of the image area may still be reading background levels even in this case, but a sufficient number of events

![](_page_28_Figure_1.jpeg)

FIG. 25: Finite Element Analysis of conditions achieved in selected samples in comparison to SOP data. SOP measurements are coloured; temperatures are blue and green with horizontal error bar showing time resolution, and red line shows emission intensity with any fluorescence contribution subtracted. FEA results are in black, with the solid line showing the peak (on axis) temperature of the upstream (SOP facing) sample surface, and the dashed line the apparent sample temperature for X-ray probes,  $T_X$ . Sources are in grey, with vertical dashed lines for XFEL pulses, and solid line filled for optical laser pulse. a) Conditions for freestanding Ta foil for single XFEL pulse irradiation (Sec. V B). b) Conditions for Ta foil in a DAC at 12 GPa under 4-pulse 2.26 MHz XFEL irradiation (Sec. V E). c) Conditions for Olivine disc in a DAC at 35 GPa under 20-pulse 2.26 MHz irradiation (Sec. V F). The dot-dashed line and dot-dot-dashed lines show the temperature on-axis in the sample center and downstream surface, respectively, considering that emission can emerge from the whole sample in this example. d) Conditions for Pt foil at 30 GPa under combined heating with a single optical laser pulse and 10 XFEL pulses at 2.26 MHz (Sec. V H). The dotted line shows the peak temperature on the observed upstream surface without X-ray heating included.

TABLE IV: FEA Parameters. Material properties are thermal conductivity (k), heat capacity at constant pressure  $(C_P)$ , density  $(\rho)$ , and absorption coefficient  $(\alpha)$ . Units for temperature T, pressure P, and density  $\rho$  are K, GPa, and g/cm<sup>3</sup>, respectively, where these appear in formulae. Heat Capacities are often Shomate fits. Pulse energy is based on SASE XGM upstream energy average and the pulse factors, which represent pulse to pulse variations in

energy, are determined from the SASE XGM, HED XGM, local photodiode, or are left as unity where this information was not available. Where information is left blank it is not relevant to the model. Text in parenthesis and in bold indicates values optimised after initial estimates, to improve match to data; the optical laser energy is for the whole pulse and is only optimised to the data, assuming complete absorption in the upstream sample surface.

#### TABLE IV: Continued.

| Target                         | Foil Array  | D1  | HIBEF30                             | XFEL2                      |  |  |  |  |
|--------------------------------|---|---|-------------------------------------|----------------------------|--|--|--|--|
| Shot / Experiment              | 161 / Exp. 2  | 845 / Exp. 2  | 199 / Exp. 3                        |                            |  |  |  |  |
| Pressure (GPa)                 | 0   | 12  | 30                                  |                            |  |  |  |  |
|                                | •   | Geometry  |                                     |                            |  |  |  |  |
| Thickness                      |   |   |                                     |                            |  |  |  |  |
| Cavity (µm)                    |   | 24  | 26                                  | 20 <sup>a</sup>            |  |  |  |  |
| Sample $(\mu m)$               | 7   | 5   | 18                                  | 4                          |  |  |  |  |
| Medium US/DS ( $\mu$ m)        | / 2000  | 15 / 4  | 4 / 4                               | 8 / 8 <sup>a</sup>         |  |  |  |  |
| Diamonds $(\mu m)$             |   | 2350  | 2290                                | 2000 a                     |  |  |  |  |
| Diameter                       |   |   |                                     |                            |  |  |  |  |
| Cavity (µm)                    |   | 80  | 90                                  | 100 <sup>a</sup>           |  |  |  |  |
| Sample $(\mu m)$               | 120   | 40  | 25                                  | 50 <sup>a</sup>            |  |  |  |  |
| Simulation (µm)                | 120   | 150   | 150                                 | 150                        |  |  |  |  |
| XFEL spot (µm)                 | 16  | 16 ( <b>13</b> )                                      | 5                                   |                            |  |  |  |  |
| NIR Laser $(\mu m)$            |   |   | 14                                  |                            |  |  |  |  |
| Material                       |   |   |                                     |                            |  |  |  |  |
| Sample                         | Та  | Та  | Pt                                  |                            |  |  |  |  |
|                                |   |   | $(Mg_{0.9}Fe_{0.1})_2SiO_4$         |                            |  |  |  |  |
| h(W/m V)                       | 57 C  | 67 cd(100 F)  | (10,000,77,0,493,(1+0,020,00)) e    | 645+1.02D+0.0108T f        |  |  |  |  |
| $\kappa (W/III K)$             | 57 -  | 07 - (100.3)  | $4.10(298/1)^{-100}(1+0.052F)^{-1}$ | 04.5+1.03P+0.0198T         |  |  |  |  |
| $C_P (J/kg K)$                 | $1.346 \times 10^2 + 2.221 \times 10^{-2}T$           | $1.346 \times 10^2 + 2.221 \times 10^{-2}T$           | $1585 - 1.238 \times 10^4 T^{-0.5}$ | 133+0.00775T f,g           |  |  |  |  |
|                                | $-1.10 \times 10^{-6} T^2 + 2.50 \times 10^{-11} T^3$ | $-1.10 \times 10^{-6} T^2 + 2.50 \times 10^{-11} T^3$ |                                     |                            |  |  |  |  |
|                                | $-3.243 \times 10^5/T^2$ h                            | $-3.243 \times 10^5/T^{2}$ h                          | $+8.414 \times 10^{-2} P^{2}$       |                            |  |  |  |  |
| $\rho (g/cm^3)$                | 16.662 h  | 17.522 <sup>j</sup>                                   | 4.028 k                             | 23.408 <sup>f</sup>        |  |  |  |  |
| $\alpha$ (1/m)                 | 138696 1  | 145985 <sup>1</sup>                                   | 2130 <sup>1</sup>                   | 235707 <sup>1</sup>        |  |  |  |  |
| Madium [US/DS]                 | / Kapton  | No / MgO  | No / No                             |                            |  |  |  |  |
| k (W/m K)                      | 0.46 m  | $1_{-1} = (1) /$                                      | 1 1,0                               | 28 P                       |  |  |  |  |
|                                | 0.40  | 5.9(a/3.3)4.6(2000/T) gr(1)                           | 1                                   | 20 -                       |  |  |  |  |
|                                |   | 0.5(p/0.0) (2000/1) (1)                               |                                     | f                          |  |  |  |  |
| $C_P (J/\text{kg K})$          | 1095  | 1840 "/   | 1840 "                              | 690 '                      |  |  |  |  |
|                                |   | $1173+0.1417-2.17\times10^{-5}T^{-2}$                 |                                     |                            |  |  |  |  |
|                                |   | $+2.59\times10^{-9}T^{3}-2.62\times10^{7}/T^{2}$      |                                     |                            |  |  |  |  |
|                                |   | $-5.3P+4.79\times10^{-2}P^{2}$ r,s(1000)              |                                     |                            |  |  |  |  |
| $\rho ~(g/cm^3)$               | 1.42 <sup>m</sup>                                     | 3.074 <sup>t</sup> / 3.826 <sup>u</sup>               | 3.932 <sup>t</sup>                  | 3.644 <sup>v</sup>         |  |  |  |  |
| $\alpha$ (1/m)                 | 92 1  | $612 / 1001^{-1}$                                     | 783 1                               | 4770 <sup>1</sup>          |  |  |  |  |
| Gasket                         | ~_  |   | Be                                  |                            |  |  |  |  |
|                                |   |   | 100 m                               |                            |  |  |  |  |
| k (W/m K)                      |   | 48 <sup>m</sup>                                       |                                     |                            |  |  |  |  |
| $C_P (J/kg K)$                 |   | 140 <sup>m</sup>                                      |                                     |                            |  |  |  |  |
| $\rho (g/cm^3)$                |   | 21.020 <sup>m</sup>                                   |                                     |                            |  |  |  |  |
| (0, 1/m)                       |   | 104000  |                                     |                            |  |  |  |  |
|                                |   | 184000 ·  |                                     |                            |  |  |  |  |
| Anvus                          |   | Diamond   |                                     |                            |  |  |  |  |
| k (W/m K)                      |   | 1500 m  |                                     |                            |  |  |  |  |
| $C_P (J/kg K)$                 |   | 630 <sup>m</sup>                                      |                                     |                            |  |  |  |  |
| $\rho \left( g/cm^{3} \right)$ |   | 2 510 m   |                                     |                            |  |  |  |  |
| P (8/ CIII )                   |   | 3.310   |                                     |                            |  |  |  |  |
| α (1/m)                        |   |   | 196 "                               |                            |  |  |  |  |
| VEDI                           |   | Source  |                                     |                            |  |  |  |  |
| XFEL                           | 15-5  | 15-5  | 15-5                                | 45.5                       |  |  |  |  |
| Photon Energy (keV)            | 17.8  | 17.8  | 17.8                                | 17.8                       |  |  |  |  |
| Rep Rate. (MHz)                |   | 2.26  | 2.26                                | 2.26                       |  |  |  |  |
| Pulse delay (ns)               |   | 443   | 443                                 | 443                        |  |  |  |  |
| Pulses/Train                   |   | 4   | 20                                  | 10                         |  |  |  |  |
| Energy/Pulse "(µJ)             | 293   | 720   | 567                                 | 420                        |  |  |  |  |
| Filter Transmission            | 0.5   |   |                                     |                            |  |  |  |  |
| Beamline Loss Factor           | 0.3   | 0.3 (0.036)   | 0.3                                 | 0.3 (0.045)                |  |  |  |  |
| Energy on Target (µJ)          | 44  | 216 (25.92)   | 170                                 | 63 (9.45)                  |  |  |  |  |
| Pulse Power Factors            | 1   | .88,1.36,.81,.95                                      | 1.50, 1.19, 1.28, 1.51, 1.39, 1.41, | 1, 1, 1, 1, 1, 1,          |  |  |  |  |
|                                |   |   | 1.26, 1.09, 0.84, 1.06, 0.84, 0.95, | 1, 1, 1, 1, 1 <sup>a</sup> |  |  |  |  |
|                                |   |   | 0.94,0.86,0.76,0.72,0.53,0.79       |                            |  |  |  |  |
|                                |   |   | ,0.52,0.57                          |                            |  |  |  |  |
| Optical Laser                  |   |   |                                     |                            |  |  |  |  |
| Energy on Target $(\mu J)$     |   |   |                                     | 28 b                       |  |  |  |  |
| Delay From Pulse 1 (ns)        |   |   |                                     | -60                        |  |  |  |  |
| Pulseshape                     |   |   |                                     | 250 ns (WF_0)              |  |  |  |  |

<sup>a</sup> value estimated <sup>b</sup> value optimised to data <sup>c</sup> Ref. 79 <sup>d</sup> Ref. 80 <sup>e</sup> Ref. 81  ${}^{\rm f}$  Ref. 82 <sup>h</sup> Ref. 59 <sup>i</sup> Ref. 83 <sup>j</sup> Ref. 84 <sup>k</sup> Ref. 85 <sup>l</sup> Ref. 86  ${}^{\rm m}$  Ref. 87 <sup>o</sup> Ref. 88 <sup>p</sup> Ref. 89 <sup>q</sup> Ref. 90 <sup>r</sup> Ref. 91 <sup>s</sup> Ref. 92 <sup>t</sup> Ref. 93 <sup>u</sup> Ref. 94 <sup>v</sup> Ref. 95 <sup>w</sup> SASE XGM (Upstream).

are accumulated in the time bin to represent the intensity at a given wavelength. In other words, near the detection limit the measurement operates in a quasi-photon counting mode, where accumulating individual intensifier events is necessary to give a statistical picture of emission spectra. Noise emerging from this phenomenon additionally becomes a good indicator of temperature fit quality. We note that low signal performance varies considerably for streak cameras used in SOP applications (see e.g. Ref. 72).

#### B. Temperature accuracy

The uncertainty of temperature estimation in pyrometry is dependent on many factors, but here it is predominantly controlled by the intensity and noise characteristics of the emission. Planck fits to low emission intensity spectra results in fits that have a larger associated uncertainty. Intensity of the optical emission is intrinsically controlled by the temperature, as described by the Stefan–Boltzmann relationship between intensity and temperature  $(I \propto T^4)$ , approximately valid in the visible at the present temperatures), as well as the size of the emitting and detected area, the optical properties of the sample, emission duration, and other factors. Hence the lower bound of measurable temperature depends on a number of experimental details, though below  $\sim 1500$ K there is insufficient emission intensity to accurately fit a Planck distribution in any case.

Determining the appropriate time resolution used to fit sample temperature needs careful consideration. Error generally increases with increasing time resolution (decreasing time bin width) as a result of poorer statistics, but also by possible introduction of systematic error as discussed above. That is, by reducing the amount of emission in any single time bin we introduce effects similar to those observed in the low temperature regime, i.e a spotty appearance of spectra (Sec. VII A). However by using a significantly larger time resolution to reduce the error in temperature one must also consider the effects related to time-dependent temperature changes, from either the sample heating or cooling within a time bin, which could itself increase the fit uncertainty and systematic error. This may have a somewhat reduced affect due to the  $T^4$  scaling of intensity with temperature resulting in the emission spectrum being dominated by the peak temperature.

Additional sources of systematic error can arise due to perturbations in the spectral shape of emission from any non-thermal emission sources. In our current survey, a significant proportion of data is affected by the nonthermal signal resulting from target luminescence (fluorescence) during X-ray exposures in targets having dielectric components. The spectral shape of emission for a DAC sample is shown in Figs. 7 and 17b. The spectral shape here is obviously non-thermal and will negativity affect any temperature measurements' precision and accuracy. As shown by Fig. 18b, the temperature that is measured when there is no accounting for fluorescence emission leads to a temperature significantly different from the actual sample temperature. This effect is small enough to be neglected when the intensity of thermal emission is much larger than that of the fluorescence emission.

Ideally, spatial filtering will both exclude emission from above and below of the sample plane, to eliminate nonsample background signal, and exclude emission from outside a selected area of the sample hotspot, to minimize the effect of integrating signal over spatial temperature gradients or isolate detection to a probed area<sup>9</sup> (e.g. to the area probed by XFEL radiation in laser heating applications). Non-sample signal exclusion is here realized using a tight-focussing configuration similar to a confocal microscope, as commonly employed in optical Raman spectroscopy measurements in DACs where diamond anvil Raman and fluorescence signals need to be suppressed<sup>96</sup>. Meanwhile, use of higher magnification with smaller pinhole is better for isolating detection to a specific region of a hotspot, at the expense of signal intensity. From initial observations at European XFEL, thermal emission measurements are generally light starved, fluorescence background potentially swamps the data, and XFEL beam alignment can vary, so we currently use a filtering scheme with a larger pinhole (1:1 magnification, 50  $\mu$ m diameter pinhole). Combined lateral and axial exclusion can be achieved by moving to a smaller pinhole, or using a dual spatial filter in future designs.

It should be emphasized that due to the prevalence of spatio-temporal gradients in temperature during experiments, the SOP can be taken to represent an average of temperatures in a particular spatial area and over a particular range of time, usually weighted towards higher temperature. The spatial and temporal variations in temperature must be separately accounted for in assessing final uncertainty on measurements of quantities of interest. That is, the SOP temperature should usually, in detail, be considered a spatio-temporal constraint on temperature in an experiment as opposed to the automatically best value of the temperature for a given contemporaneous property measurement (e.g. Sec VI).

#### C. Target fluorescence behavior

Non-thermal, visible-band optical luminescence generated by XFEL pulses was commonly observed by SOP in targets containing dielectric materials. This luminescence often resembled the conventional fluorescence behavior of the dielectrics under optical excitation, and was consistent with a rapid decay (faster than the time resolution) and broadband emission in most cases. For example, the XFEL-induced luminescence characteristics of diamond and YAG:Ce crystals was effectively instantaneous for the  $\mu$ s-sweeps used here, consistent with conventional relaxation times < 100 ns<sup>97,98</sup>. When excited

![](_page_31_Figure_0.jpeg)

FIG. 26: Optical fluorescence emission from ruby in target SB001 (Fe/N2 at 5 GPa) following single pulse XFEL irradiation (136  $\mu$ J/pulse). Black curve is the fitted fluorescence peak, using a single Gaussian fit (20 pixel time average, 2 x 2 image binning), at 696(1) nm indicating a pressure of 5(3) GPa<sup>74</sup>.

using a similar X-ray photon energy, fluorescence spectra of diamond are broadly similar between high-intensity XFEL and low-intensity synchrotron irradiation (Fig. 8).

Some deviations from rapid emission were noted for certain sample materials (Fig. 7), with a notable case being that of ruby. In one DAC, a larger ruby piece (used as a pressure marker, see Sec. IV) was targeted independently with the XFEL (Fig. 26). The XFEL-excited ruby luminescence also exhibited conventional fluorescence characteristics, including line emission at  $\sim 696$ nm (under pressure of  $\sim 5$  GPa) and a relaxation time measured to be  $\sim 250 \ \mu s$ , comparing well with literature values using laser-induced optical fluorescence<sup>97</sup>. This suggests that XFEL irradiation with SOP can allow in situ ruby pressure measurements without additional laser excitation, including during dynamic pressure conditions. As ruby was localized in samples it was easily avoided in practice. Data is currently insufficient to establish the degree of sample contributions to observed fluorescence in most cases in the DAC, with observed variations in fluorescence amplitude likely to be due to different diamonds and stress states.

While the temporal and spectral structure of X-ray fluorescence itself may have interesting properties and applications, we here characterized it principally for the purpose of its removal from thermal emission measurements.

The effect of fluorescence on apparent temperature is not straightforward to predict. When not accounted for in temperature measurements, fluorescence can manifest itself by spurious high or low temperatures, possibly related to emission rising toward both red and blue ends of the visible range in representative cases. In most cases with such contamination, the fluorescence is most strong in the blue and spurious high temperatures may be inferred. While this often leads to clear deviations from an ideal Planck distribution, such as a concave upward spectrum, rising toward both red (thermal contribution) and blue (fluorescence contribution), it may be considerably more subtle. Given that fluorescence emission is unique for each target both in terms of spectral shape and intensity (e.g. due to different diamonds, Figs. 7 and 16), characterization of the fluorescence emission must be performed for each target, and, for DACs, at the specific pressure conditions and sample location of interest. Diamonds at Mbar pressures have exhibited unique fluorescence distributions in the SOP differing considerably to those in lower pressure anvils, consistent with pressure sensitivity of fluorescence<sup>99</sup>.

Fluorescence emission can be characterized and removed from SOP spectrograms, but this currently requires a sufficient number of reference experimental runs at low XFEL power without any contribution from thermal signal. Given the difficulty of the fluorescence correction even armed with this reference data, its a priori elimination is desireable. Assuming most fluorescence emerges from the diamond anvils in DAC targets in studied cases, this can be achieved by eliminating signal from the anvils (or more generally, dielectric matter surrounding the target material of interest). Use of spatial filtering can minimize such contributions, with preliminary observations suggesting it can be reduced considerably for DAC targets (Fig. 22). Use of type II low fluorescence anvils on one or both sides of the DAC may also be valuable if fluorescence minimization is desired (for nontransparent samples only observed anvils need to be of type II). However, some evidence of sample contributions are also identified (Figs 7 and 26) so these practices may not entirely be able to eliminate fluorescence backgrounds.

Since fluorescence removal may not be completely accurate, it is possible that the removal process leaves residual systematic errors in assessed temperatures. This is particularly an issue where the correction in temperature is larger. Thus where corrections for fluorescence are large, additional systematic error in temperature can be expected. However this represents a minority of the present examples.

#### D. X-ray Heating Observations

Heating is observed with pulse energies of  $10 - 300 \mu$ J at 17.8 keV on various targets, both freestanding and confined (and pressurized), for a focal size in the range of  $5 - 20 \mu$ m FWHM – conditions common for X-ray pump probe experiments at free electron lasers. X-ray heating of samples is a natural byproduct of intense XFEL radiation<sup>27,29,35,37</sup>, and has been previously observed to occur with detectable optical emission<sup>35</sup>. We here realize the longstanding objective<sup>33</sup> of detecting this radiation with pyrometric techniques in order to determine temperature, while also validating directly heating models.

X-ray heating with single pulses shows a peak and decay in the temperature-time profile which can be un-

derstood as rapid (effectively instantaneous) heating followed by a gradual decay as the sample cools. X-ray heating from pulse trains is meanwhile expected to induce step-wise heating<sup>29</sup>, as samples are pumped to higher temperatures while incompletely cooled, assuming constant energies within each pulse. While the SOP images show some evidence of stepwise heating, specifically, a rising thermal emission intensity, temperatures rarely show dramatic corresponding increase later in the pulse trains. A plateau in temperature is expected even when pulse energies are constant due to balance of cooling and heating in time. Moreover, the effect of variable pulse energies can be significant. These effects are considered quantitatively in Sec. VI using FEA models, which confirm that the limiting temperatures with time observed can be considered a direct result of the specific pulse train energy profiles. It should also be noted that sample disruption and internal movement during pulse trains (e.g. due to melting $^{27}$ ) could also affect achieved temperatures, though this is not required to interpret the present data.

Comparing free-standing samples (e.g. SS2, Sec. VB) to those in a DAC (e.g. D1, Sec. VE) we see shorter emission timescales for the freestanding foil, i.e the time for the emission intensity to drop to zero after initial irradiation is shorter. The longer emission for confined compared to freestanding foil is somewhat unexpected as the additional heat sink of the highly thermally conductive diamonds around the confined sample should lead to it cooling relatively faster. This may be related to rapid damage processes in the unconfined case (Fig. 14d).

#### E. Finite Element Modeling of X-ray Heating

In some cases, FEA closely predict the observed SOP temperatures (Fig. 25). This is not unexpected for the freestanding Ta foil (Fig. 25a) since the nominal beamline transmission (30%) was calibrated to ablation damage data on Ta foil during the same experiments<sup>37</sup>, however this still shows the consistency of the independent SOP measurement of temperature with model predictions<sup>59</sup>. Similarly, a completely independent experiment on olivine (Fig. 25c), which reaches only modest temperatures, is closely predicted (considering that emission likely emerges from the full volume of the sample in this case). For high-Z foils in the DAC (Fig. 25b,d), significantly lower radiation absorption than expected must be assumed in order to reproduce the observed temperatures, and in some cases other parameters, e.g. thermal conductivity, also require adjustment, including outside their plausible range (Table IV). This sets some limit on achieved temperatures, discussed further below.

The apparent X-ray temperature predicted through the FEA models (i.e.,  $T_X$ ) reveal that average temperature in the bulk detected by X-ray probes can differ substantially from that observed in SOP. In part this is because the X-ray temperature averages over the temperature gradient in the bulk, whereas SOP temperature will be dominated by peak temperature on a surface (i.e. it is weighted to maximum temperature due to  $\sim T^4$ scaling of intensity). Moreover, if  $T_X$  is dependent on structural phase (e.g. the volume or phase observed in XRD), it will correspond to conditions just before each pulse whereas SOP T at similar time is dominated by the elevated temperature just after the pulse, potentially leading to even more significant differences between temperature in X-ray and SOP data. Even in cases where individual pulse heating is small (e.g. Olivine, Fig. 25c), the difference in X-ray probe temperature and emission temperature can be significant, primarily due to the presence of lateral gradients; i.e. hotspot peak temperatures on the beam axis dominate the SOP data whereas these temperatures contribute little to the volumetric average represented by X-ray probes. Thus, where XH is significant, comparisons of XRD data and SOP require a careful consideration of both time and space dependence of the temperature in the sample and the specific sensitivities of each measurement.

#### F. Temperature of laser heated samples

We observe that the temperature of the sample in DACs DESY1 (Fig. 13) and XFEL1 (Fig. 20) using LH only follows the generally expected thermal evolution from prior studies. In this case, the peak in temperature coincides closely with the laser pulse and peak in emission intensity, though higher time resolution shows more detail in the temperature evolution (Fig. 13). Both the emission intensity and fitted sample temperature fall off after the laser pulse as expected. The spectral shape of emission also clearly demonstrates a grey-body distribution with respect to changing temperatures. Both of these validate the use of this SOP set-up as a reliable temperature diagnostic for laser-heated samples.

### G. Relationship between emission intensity and temperature

Temperature and emission intensity are observed to have limited correlation in these experiments. That is, emission intensity is not a direct, quantitative proxy for temperature. Though qualitative trends of emission intensity increasing with temperature are observed, its amplitude often changes irrespective of the temperature within the same experiment. For example, in the case of serially X-ray heated metal in a DAC (Secs V C and V E, Figs. 15 and 18), peak temperatures, coincident with X-ray pulses, remain roughly constant with increasing time but corresponding emission intensity increases. Similarly, in the case of optical heating, temperature initially decreases as emission intensity rises at the beginning of optical laser pulse (Sec. V A and Fig 13).

A variety of factors can contribute to the emission

![](_page_33_Figure_0.jpeg)

FIG. 27: Emission intensity behavior observed in Shot 845 (Sample D1, Fig. 18) together with finite element model results. (a) Experimental data compared to finite element model, with emission intensity in arbitrary units. Observed emission intensity is averaged over the wavelength range 575 - 775 nm. Predicted intensities in FEM are the total intensity at 550, 700 and 850 nm, integrated over the metal foil surface (units of calculation are W sr<sup>-1</sup> Hz<sup>-1</sup>) (b) Radial temperature gradient on the observed (upstream) metal foil surface just after each X-ray pulse. Dashed line shows approximate detection limit. Emission intensity is calculated in the FEM model by integrating the Planck

function over this surface.

intensity trends in the SOP images. Variations in the emitting surface area size and temperature gradient significantly controls the emission. For example, Fig. 27 compares emission intensity observed experimentally and predicted from FEM modeling, during 4-pulse irradiation of Ta foil in MgO and Ne in DAC D1, shot 845 (see Figs. 18 and 25b). Consistent with the observations, the first pulse exhibits very weak emission (Fig. 27a), and temperature is predicted to fall below the detection limit (Fig. 27b). The next 3 pulses show similar peak temperatures (Fig. 27a) and radial temperature distributions (Fig. 27b), however even small changes in the radial temperature distribution are seen to dramatically affect the emission amplitude. The results show that the dramatic rise in emission intensity is due to the heated area becoming wider with time, regardless of peak temperature.

Thus, where emission intensity is generally increasing with time, often without a clear corresponding increase in peak temperature, as in the above examples, it can often be attributed to changes in heated area size and gradients. A continuous broadening of the X-ray laser heated spot with time as heat conducts from the focal point laterally<sup>29</sup> and inconsistent pointing within pulse trains<sup>37</sup> can both lead to larger hot areas at later time. For example, in Fig. 27b, the FWHM of the radial temperature gradient broadens, just due to conduction processes, from 6.7  $\mu$ m at the time of the first pulse to 8.6

 $\mu m$  at the time of the fourth pulse, corresponding to the significant increase in total emission.

Other factors may also play a role in emission amplitude, and it's relationship to measured temperature. Changes to target optical properties with temperature or time may become significant, such as where target emissivity may increase due to onset of conductivity in pressure media at high temperature, or where optical properties change as a function of time due to phase transformation or reaction kinetics<sup>9,36,72</sup>. Instrument performance can also play a role, including by smearing due to the point spread function or integration of multiple sweeps, or pincushion sensitivity variations. Finally, fluorescence contributions can lead to deviations from true thermal emission intensity particularly where temperatures and/or thermal emission signal is low, in many cases expressing as apparently blue (and therefore, hot) emission spectra despite low signal.

Comparing X-ray and laser heating, differing sizes of heated areas correlate to differences in emission intensity, with X-ray experiments with tighter focused radiation generally showing lower emission amplitudes for comparable temperatures. When combined together, the broad laser heated spots at one time in an experiment is in contrast to smaller hotspots maintained by the X-ray pulse train at other times (Sec. V H, Fig. 23), and consequently emission intensity varies irrespective of temperature.

One consequence of these observations is that a constant emissivity approach to fitting the time domain temperature data, which can reduce fitting error<sup>64</sup>, is not generally applicable in these experiments. Similarly an intensity-based measure of temperature, as is commonly used in SOP systems for shock wave experiments<sup>48</sup> is not easily used in experiments of the current type.

#### H. Maximum Temperatures Achieved

Potentially very high temperatures are achievable with either pulsed laser or X-ray heating in these experiments<sup>9,27,29</sup>. Laser heating observations are consistent with prior expectations and experience from standard laser heating experiments, with temperatures not much beyond the melting point for unconfined foil, and significantly higher (in the eV range) when pulsed heating confined foils<sup>9</sup>. Values in the range of 3000 - 6000K are common in even continuous laser heating of confined foils. X-ray heating shows more unexpected behavior. For the X-ray energies available at the beamline and the high absorbances of some samples studied (e.g. Ta and Au), very high temperatures should be achievable in these samples, in the multi  $eV range^{27,29}$ . However, rarely did temperatures produced by X-ray heating and detected pyrometrically reach high values beyond  $\sim 5000$ - 6000 K. This is comparable or even less than the typical maximum temperatures achieved by continuous laser heating of DAC samples and well below those seen in

pulsed laser heating, including using this same experimental setup.

Temperatures in freestanding XH foils were also larger than those seen in the DAC under X-ray heating. For example, in freestanding Ta, values up to  $\sim 6000$  K were observed in a single exposure at 50% power equivalent to 25  $\mu$ J/pulse (Fig. 14). The required beamline transmission to the sample in this case was  $\sim 30\%$  (Table IV), which was validated using an energy scan over the the full range of energy using ablation as a diagnostic of temperature<sup>37,59</sup>. However, for Ta in a DAC, at even higher power levels (100%, 137  $\mu$ J/pulse), thermal emission was not detectable in a single shot at all, and even the cumulative heating provided by a series of pulses did not achieve the same temperature as a single pulse on a free-standing foil (Fig. 18). The effective beam line transmission to the target needed to explain this data is roughly an order of magnitude smaller than in the freestanding foil case (Table IV) and below any plausible real value for the transmission.

There could be a variety of reasons for these observations. This effect may be partly due to cumulative and in-situ damage to these samples, occurring at high temperatures when samples are melted over large regions and subjected to large thermal and hydrodynamic stresses. Such localized damage is frequently observed in postmortem observations on high-Z samples<sup>27</sup>. It should also be noted that optically it is difficult to observe very high temperatures in DAC experiments due to onset of opacity in transparent materials at high pressure and temperature<sup>7,9,75</sup>; in fact, screening of thermal radiation by hot but optically opaque pressure media $^{7,9}$  could be even more severe for X-ray heating due to embedding of directly heated layers within indirectly heated media. Relevant scenarios include heating of metallic layers cumulatively with efficient quenching at the metal surface, allowing very high temperature within the metal interior but undetectable temperature on the surface (with corresponding irregularities between X-ray and SOP temperature probing). Other effects that can increasingly act to reduce heating at higher temperature include increasing contributions of electronic heat capacity<sup>27</sup> and decompression cooling of isochorically heated states<sup>29</sup>. Rapid dissipation of heat on sub-ns timescales, such as by fast electron dynamics, could also limit observed temperatures. Still, temperatures of interest, e.g. near high pressure melting points, are readily obtained using XH methods especially when cumulative heating is employed.

#### I. Combining X-ray probing and other excitations

A key factor for planning XFEL experiments using separate excitations, such as optical lasers, is the extent to which the XFEL itself may affect sample state and observations. A single or serial pulse train could lead to perturbations in detected temperatures in SOP (Fig. 23). Meanwhile, temperatures detected by X-ray probes may deviate from those achieved through external excitation alone, if preceding probe pulses in a pulse train have altered the local temperatures via residual heat. Such Xray heating can have obvious (Fig. 23) or subtle (Fig. 22) effects on SOP data, hence the role of X-ray heating in measurements using other excitation may not be easily assessed from SOP alone. One possibility is that probed spots may be much smaller than the excited area, with any local heating having a small impact on total emission signal but a critical impact on pulse train observations. Another is that significant heating is possible below the detection limit of SOP. While heating can be mitigated by reducing the XFEL fluence or using samples of low absorption, our data show that considerable heating remains possible in such scenarios. Alternatively, many experiments can be conducted in a fashion such that the conditions following the XFEL probe are not of interest.

#### VIII. CONCLUSION

In this work we have optically measured high temperature states induced by X-ray and optical laser pulses in condensed matter targets at a free electron laser facility. Dynamic temperatures in the range of 1,000 to 10,000 K, with time resolutions in the range of tens to hundreds of nanoseconds, are studied. These results match expectations of sample heating and temperature evolution calculated by numerical models, with forward modeling of temperature by finite element analysis able to accurately predict the measured total temperature evolution in certain scenarios.

Spectrally- and time-resolved optical emission is measured using a spectrometer mounted on a streak camera coupled to an optical microscope equipped with a confocal spatial filter to isolate a 50  $\mu$ m diameter area on target and exclude signal away from the focal plane. XFEL irradiation comprised of 17.8 keV, 20-femtosecond XFEL pulses in single shot or MHz pulse trains (443 ns repetition rate), while the optical laser was NIR (1070 nm) with a pulse length of 250 ns. These are focused to spots of  $\sim 10 \ \mu m$  diameter on targets. The time-dependent thermal emission spectra are least-squares fitted to the Planck function, with the time integration bins, temporal point spread function and selected streak window setting the time resolution of the temperature measurement. Uncertainty is evaluated from robust checking of goodness of fit parameters against statistical uncertainties in reference-temperature spectrograms.

We study a suite of representative targets including freestanding foils and multi-layer targets in the form of a diamond anvil cell, used to contain heated samples and apply high pressures (GPa) prior to irradiation. Measured temperature is dominated by the peak temperature in the field of view, whereas emission intensity has a more complex development accurately predicted by numerical modeling of dynamic temperature gradients (Fig. 27). Where dielectric-bearing targets are irradiated by the XFEL (e.g. diamond anvil cells), background emission due to sample fluorescence is often detected. Thermal and fluorescent signals have distinct spectral-temporal appearance, and when similar in amplitude, fluorescence can perturb temperature measurements. Fluorescence appears at lower powers than required to produce detectable thermal signal, such that data where only fluorescence is recorded can be used to interpret and model fluorescence behavior for a particular target and extract the thermal component of emission before sample temperature is measured. Fluorescence contributions are mitigated through spatial filtering and use of low fluorescence materials (e.g. type II rather than type I diamond). For optical laser heating, results broadly follow previous work using streak optical pyrometry methods.

The spectral SOP discussed here has potentially broad applications for time-resolved measurements of temperature and other optical signals at European XFEL and similar facilities. In the current sample environment configuration at EuXFEL. it is compatible with a range of X-ray and optical laser experiments. Modification of the sample imaging optics (e.g. for different targets, environments and measurements) can allow compatibility with a wider range of experiments. For example, SOP is well demonstrated in conjunction with shock wave compression, and the streak time range (0.5 ns to 1 ms) covers what is typically required for shock wave techniques. The spectral SOP is particularly useful for lower temperature ranges (< 10,000 K) where the emission wavelength dependence varies strongly with temperature. Other optical spectroscopic applications shown possible here include fluorescence spectroscopy, relaxation time studies and pressure measurements.

#### ACKNOWLEDGMENTS

We thank H. Sinn for experimental assistance and T. Tschentscher and S. Pascarelli for fruitful discussions. We acknowledge European XFEL in Schenefeld, Germany, for provision of X-ray free-electron laser beamtime at Scientific Instrument HED (High Energy Density Science) and would like to thank the staff for their assistance. The authors are indebted to the Helmholtz International Beamline for Extreme Fields (HIBEF) user consortium for the provision of instrumentation and staff that enabled this experiment. We acknowledge DESY (Hamburg, Germany), a member of the Helmholtz Association HGF, for the provision of experimental facilities. Parts of this research were carried out at PETRA III (beamline P02.2) through proposal I-20190367 EC. The research leading to this result has been supported by the project CALIPSOplus under the Grant Agreement 730872 from the EU Framework Program for Research and Innovation HORIZON 2020. Part of this work was performed under the auspices of the U.S. Department of Energy by Lawrence Livermore National Laboratory under Contract No. DE-AC52-07NA27344, and

by the US Department of Energy through the Los Alamos National Laboratory, operated by Triad National Security, LLC, for the National Nuclear Security Administration (Contract No. 89233218CNA000001). Research presented in this article was supported by the by the Department of Energy, Laboratory Directed Research and Development program at Los Alamos National Laboratory under project number 20190643DR and at SLAC National Accelerator Laboratory, under contract DE-AC02-76SF00515. Support is acknowledged from the Panofsky Fellowship at SLAC (awarded to EEM); the DOE Office of Fusion Energy Science funding number FWP100182 (MF and EEM); EPSRC Grants EP/P024513/1 (RSM) and EP/R02927X/1 (EJP and MIM); the European Research Council (ERC) under the European Union's Horizon 2020 research and innovation Programme for grant agreements Nos. 670787 (GF, MAB, and GM), 864877 (HM), and 101002868 (RSM); the National Science Foundation - Earth Sciences (EAR - 1634415) (VBP); and the Leader Researcher program (NRF-2018R1A3B1052042) of the Korean Ministry of Science and ICT (MSIT) (YL, HH). KA, KG, RJH, ZK, HPL, and RR thank the DFG for support within the Research Unit FOR 2440. ZK and CP acknowledge support from the DFG project number KO-5262/1. SM and JC acknowledge support from the MetalCore project, supported by the French government through the Programme Investissement d'Avenir (I-SITE ULNE / ANR-16-IDEX-0004 ULNE) managed by the Agence Nationale de la Recherche. OBB acknowledges the support of the Scottish Doctoral Training Program in Condensed Matter Physics (CM-CDT). JDM acknowledges support from EPSRC and AWE CASE studentships, UK Ministry of Defence ©Crown Owned Copyright 2022 / AWE. AFG and EE are grateful to Carnegie Science for the support.

- <sup>1</sup>T. Duffy, N. Madhusudhan, and K. Lee, "Mineralogy of Super-Earth Planets," in *Treatise on Geophysics* (Elsevier, 2015) pp. 149–178.
- <sup>2</sup>T. Guillot, "Interiors of Giant Planets Inside and Outside the Solar System," Science (American Association for the Advancement of Science) **286**, 72–77 (1999).
- <sup>3</sup>J. D. Lindl, P. Amendt, R. L. Berger, S. G. Glendinning, S. H. Glenzer, S. W. Haan, R. L. Kauffman, O. L. Landen, and L. J. Suter, "The physics basis for ignition using indirect-drive targets on the National Ignition Facility," Physics of plasmas **11**, 339–491 (2004).
- <sup>4</sup>K. Falk, "Experimental methods for warm dense matter research," High Power Laser Science and Engineering 6 (2018), 10.1017/hpl.2018.53.
- <sup>5</sup>G. Weck, V. Recoules, J.-A. Queyroux, F. Datchi, J. Bouchet, S. Ninet, G. Garbarino, M. Mezouar, and P. Loubeyre, "Determination of the melting curve of gold up to 110 GPa," Phys. Rev. B **101**, 014106 (2020).
- <sup>6</sup>Z. Du, G. Amulele, L. R. Benedetti, and K. K. M. Lee, "Mapping temperatures and temperature gradients during flash heating in a diamond-anvil cell," Review of Scientific Instruments 84, 075111 (2013).
- <sup>7</sup>A. Dewaele, M. Mezouar, N. Guignot, and P. Loubeyre, "High Melting Points of Tantalum in a Laser-Heated Diamond Anvil Cell," Phys. Rev. Lett. **104**, 255701 (2010).
- <sup>8</sup>S. Petitgirard, A. Salamat, P. Beck, G. Weck, and P. Bouvier, "Strategies for in situ laser heating in the diamond anvil cell at

an X-ray diffraction beamline," Journal of Synchrotron Radiation **21**, 89–96 (2014).

- <sup>9</sup>R. S. McWilliams, D. A. Dalton, Z. Konôpková, M. F. Mahmood, and A. F. Goncharov, "Opacity and conductivity measurements in noble gases at conditions of planetary and stellar interiors," Proceedings of the National Academy of Sciences **112**, 7925–7930 (2015), https://www.pnas.org/content/112/26/7925.full.pdf.
- <sup>10</sup>S. Anzellini, A. Dewaele, M. Mezouar, P. Loubeyre, and G. Morard, "Melting of iron at Earth's inner core boundary based on fast X-ray diffraction," Science **340**, 464 (2013).
- <sup>11</sup>Z. Konôpková, R. S. McWilliams, N. Gómez-Pérez, and A. F. Goncharov, "Direct measurement of thermal conductivity in solid iron at planetary core conditions," Nature **534**, 99–101 (2016).
- <sup>12</sup>G. Morard, S. Boccato, A. D. Rosa, S. Anzellini, F. Miozzi, L. Henry, G. Garbarino, M. Mezouar, M. Harmand, F. Guyot, E. Boulard, I. Kantor, T. Irifune, and R. Torchio, "Solving Controversies on the Iron Phase Diagram Under High Pressure," Geophysical Research Letters **45** (2018), 10.1029/2018gl079950.
- <sup>13</sup>S. Boccato, R. Torchio, I. Kantor, G. Morard, S. Anzellini, R. Giampaoli, R. Briggs, A. Smareglia, T. Irifune, and S. Pascarelli, "The Melting Curve of Nickel Up to 100 GPa Explored by XAS," Journal of Geophysical Research: Solid Earth **122**, 9921–9930 (2017).
- <sup>14</sup>S. Boccato, R. Torchio, P. D'Angelo, A. Trapananti, I. Kantor, V. Recoules, S. Anzellini, G. Morard, T. Irifune, and S. Pascarelli, "Compression of liquid Ni and Co under extreme conditions explored by X-ray absorption spectroscopy," Physical Review B **100** (2019), 10.1103/physrevb.100.180101.
- <sup>15</sup>A. Lévy, P. Audebert, R. Shepherd, J. Dunn, M. Cammarata, O. Ciricosta, F. Deneuville, F. Dorchies, M. Fajardo, C. Fourment, D. Fritz, J. Fuchs, J. Gaudin, M. Gauthier, A. Graf, H. J. Lee, H. Lemke, B. Nagler, J. Park, O. Peyrusse, A. B. Steel, S. M. Vinko, J. S. Wark, G. O. Williams, D. Zhu, and R. W. Lee, "The creation of large-volume, gradient-free warm dense matter with an X-ray free-electron laser," Physics of Plasmas **22**, 030703 (2015).
- <sup>16</sup>H.-P. Liermann, Z. Konôpková, W. Morgenroth, K. Glazyrin, J. Bednarčik, E. E. McBride, S. Petitgirard, J. T. Delitz, M. Wendt, Y. Bican, A. Ehnes, I. Schwark, A. Rothkirch, M. Tischer, J. Heuer, H. Schulte-Schrepping, T. Kracht, and H. Franz, "The Extreme Conditions Beamline P02.2 and the Extreme Conditions Science Infrastructure at PETRA III," Journal of Synchrotron Radiation **22**, 908–924 (2015).
- <sup>17</sup>B. Nagler, U. Zastrau, R. R. Fäustlin, S. M. Vinko, T. Whitcher, A. J. Nelson, R. Sobierajski, J. Krzywinski, J. Chalupsky, E. Abreu, S. Bajt, T. Bornath, T. Burian, H. Chapman, J. Cihelka, T. Döppner, S. Düsterer, T. Dzelzainis, M. Fajardo, E. Förster, C. Fortmann, E. Galtier, S. H. Glenzer, S. Göde, G. Gregori, V. Hajkova, P. Heimann, L. Juha, M. Jurek, F. Y. Khattak, A. R. Khorsand, D. Klinger, M. Kozlova, T. Laarmann, H. J. Lee, R. W. Lee, K.-H. Meiwes-Broer, P. Mercere, W. J. Murphy, A. Przystawik, R. Redmer, H. Reinholz, D. Riley, G. Röpke, F. Rosmej, K. Saksl, R. Schott, R. Thiele, J. Tiggesbäumker, S. Toleikis, T. Tschentscher, I. Uschmann, H. J. Vollmer, J. S. Wark, and e. Bob Nagler, "Turning solid aluminium transparent by intense soft X-ray photoionization," Nature Physics 5, 693–696 (2009).
- <sup>18</sup>U. Zastrau, C. Fortmann, R. R. Fäustlin, L. F. Cao, T. Döppner, S. Düsterer, S. H. Glenzer, G. Gregori, T. Laarmann, H. J. Lee, A. Przystawik, P. Radcliffe, H. Reinholz, G. Röpke, R. Thiele, J. Tiggesbäumker, N. X. Truong, S. Toleikis, I. Uschmann, A. Wierling, T. Tschentscher, E. Förster, and R. Redmer, "Bremsstrahlung and line spectroscopy of warm dense aluminum plasma heated by XUV free-electron-laser radiation," Physical Review E **78** (2008), 10.1103/physreve.78.066406.
- <sup>19</sup>U. Zastrau, P. Sperling, A. Becker, T. Bornath, R. Bredow, T. Döppner, S. Dziarzhytski, T. Fennel, L. B. Fletcher, E. Förster, C. Fortmann, S. H. Glenzer, S. Göde, G. Gregori, M. Harmand, V. Hilbert, B. Holst, T. Laarmann, H. J. Lee, T. Ma, J. P. Mithen, R. Mitzner, C. D. Murphy, M. Nakat-

sutsumi, P. Neumayer, A. Przystawik, S. Roling, M. Schulz, B. Siemer, S. Skruszewicz, J. Tiggesbäumker, S. Toleikis, T. Tschentscher, T. White, M. Wöstmann, H. Zacharias, and R. Redmer, "Equilibration dynamics and conductivity of warm dense hydrogen," Physical Review E **90** (2014), 10.1103/physreve.90.013104.

- <sup>20</sup>M. B. Boslough and T. J. Ahrens, "A sensitive time-resolved radiation pyrometer for shock-temperature measurements above 1500 K," Review of Scientific Instruments **60**, 3711–3716 (1989).
- <sup>21</sup>K. K. M. Lee, L. R. Benedetti, R. Jeanloz, P. M. Celliers, J. H. Eggert, D. G. Hicks, S. J. Moon, A. Mackinnon, L. B. D. Silva, D. K. Bradley, W. Unites, G. W. Collins, E. Henry, M. Koenig, A. Benuzzi-Mounaix, J. Pasley, and D. Neely, "Laser-driven shock experiments on precompressed water: Implications for "icy" giant planets," The Journal of Chemical Physics **125**, 014701 (2006).
- <sup>22</sup>C. Dai, J. Hu, and H. Tan, "Hugoniot temperatures and melting of tantalum under shock compression determined by optical pyrometry," Journal of Applied Physics **106**, 043519 (2009), https://doi.org/10.1063/1.3204941.
- <sup>23</sup>S. B. Kormer, "Optical study of the characteristics of shockcompressed condensed dielectrics," Soviet Physics Uspekhi 11, 229–254 (1968).
- <sup>24</sup>R. Jeanloz, P. M. Celliers, G. W. Collins, J. H. Eggert, K. K. M. Lee, R. S. McWilliams, S. Brygoo, and P. Loubeyre, "Achieving high-density states through shock-wave loading of precompressed samples," Proceedings of the National Academy of Sciences of the United States of America **104**, 9172–9177 (2007).
- <sup>25</sup>H. Cynn and C.-S. Yoo, "Equation of state of tantalum to 174 GPa," Phys. Rev. B **59**, 8526–8529 (1999).
- <sup>26</sup>V. Prakapenka, G. Shen, and L. Dubrovinsky, "Carbon transport in diamond anvil cells," High Temperatures-High Pressures **35/36**, 237–249 (2003).
- <sup>27</sup>R. J. Husband, R. S. McWilliams, E. J. Pace, A. L. Coleman, H. Hwang, J. Choi, T. Kim, G. C. Hwang, O. B. Ball, S. H. Chun, D. Nam, S. Kim, H. Cynn, V. B. Prakapenka, S.-H. Shim, S. Toleikis, M. I. McMahon, Y. Lee, and H.-P. Liermann, "Xray free electron laser heating of water and gold at high static pressure," Communications Materials 2 (2021), 10.1038/s43246-021-00158-7.
- <sup>28</sup>Z. Jenei, H. P. Liermann, R. Husband, A. S. J. Méndez, D. Pennicard, H. Marquardt, E. F. O'Bannon, A. Pakhomova, Z. Konopkova, K. Glazyrin, M. Wendt, S. Wenz, E. E. McBride, W. Morgenroth, B. Winkler, A. Rothkirch, M. Hanfland, and W. J. Evans, "New dynamic diamond anvil cells for tera-pascal per second fast compression X-ray diffraction experiments," Review of Scientific Instruments **90**, 065114 (2019).
- <sup>29</sup>J. Meza-Galvez, N. Gomez-Perez, A. S. Marshall, A. L. Coleman, K. Appel, H. P. Liermann, M. I. McMahon, Z. Konôpková, and R. S. Mcwilliams, "Thermomechanical response of thickly tamped targets and diamond anvil cells under pulsed hard X-ray irradiation," Journal of Applied Physics **127** (2020).
- <sup>30</sup>W. Decking, S. Abeghyan, P. Abramian, A. Abramsky, A. Aguirre, C. Albrecht, P. Alou, M. Altarelli, P. Altmann, K. Amyan, V. Anashin, E. Apostolov, K. Appel, D. Auguste, V. Ayvazyan, S. Baark, F. Babies, N. Baboi, P. Bak, V. Balandin, R. Baldinger, B. Baranasic, S. Barbanotti, O. Belikov, V. Belokurov, L. Belova, V. Belyakov, S. Berry, M. Bertucci, B. Beutner, A. Block, M. Blöcher, T. Böckmann, C. Bohm, M. Böhnert, V. Bondar, E. Bondarchuk, M. Bonezzi, P. Borowiec, C. Bösch, U. Bösenberg, A. Bosotti, R. Böspflug, M. Bousonville, E. Boyd, Y. Bozhko, A. Brand, J. Branlard, S. Briechle, F. Brinker, S. Brinker, R. Brinkmann, S. Brockhauser, O. Brovko, H. Brück, A. Brüdgam, L. Butkowski, T. Büttner, J. Calero, E. Castro-Carballo, G. Cattalanotto, J. Charrier, J. Chen, A. Cherepenko, V. Cheskidov, M. Chiodini, A. Chong, S. Choroba, M. Chorowski, D. Churanov, W. Cichalewski, M. Clausen, W. Clement, C. Cloué, J. A. Cobos, N. Coppola, S. Cunis, K. Czuba, M. Czwalinna, B. D'Almagne, J. Dammann, H. Danared, A. de Zubiaurre Wagner, A. Delfs,

T. Delfs, F. Dietrich, T. Dietrich, M. Dohlus, M. Dommach, A. Donat, X. Dong, N. Doynikov, M. Dressel, M. Duda, P. Duda, Free-Electro

H. Eckoldt, W. Ehsan, J. Eidam, F. Eints, C. Engling, *et al.*, "A MHz-repetition-rate hard X-ray free-electron laser driven by a superconducting linear accelerator," Nature Photonics **14**, 391–397 (2020).

- <sup>31</sup>J. Grünert, M. P. Carbonell, F. Dietrich, T. Falk, W. Freund, A. Koch, N. Kujala, J. Laksman, J. Liu, T. Maltezopoulos, K. Tiedtke, U. F. Jastrow, A. Sorokin, E. Syresin, A. Grebentsov, and O. Brovko, "X-ray photon diagnostics at the European XFEL," Journal of Synchrotron Radiation **26**, 1422–1431 (2019).
- <sup>32</sup>M. Ishino, N. Hasegawa, M. Nishikino, T. Pikuz, I. Skobelev, A. Faenov, N. Inogamov, T. Kawachi, and M. Yamagiwa, "Very low electron temperature in warm dense matter formed by focused picosecond soft X-ray laser pulses," Journal of Applied Physics **116**, 183302 (2014).
- <sup>33</sup>E. Principi, C. Ferrante, A. Filipponi, F. Bencivenga, F. D'Amico, C. Masciovecchio, and A. D. Cicco, "A method for estimating the temperature in high energy density free electron laser experiments," Nuclear Instruments and Methods in Physics Research Section A: Accelerators, Spectrometers, Detectors and Associated Equipment **621**, 643–649 (2010).
- <sup>34</sup>H. Wallander and J. Wallentin, "Simulated sample heating from a nanofocused X-ray beam," Journal of Synchrotron Radiation 24, 925–933 (2017).
- <sup>35</sup>E. J. Pace, A. L. Coleman, R. J. Husband, H. Hwang, J. Choi, T. Kim, G. Hwang, S. H. Chun, D. Nam, S. Kim, O. B. Ball, H.-P. Liermann, M. I. McMahon, Y. Lee, and R. S. McWilliams, "Intense Reactivity in Sulfur–Hydrogen Mixtures at High Pressure under X-ray Irradiation," The Journal of Physical Chemistry Letters **11**, 1828–1834 (2020), pMID: 32048851, https://doi.org/10.1021/acs.jpclett.9b03797.
- <sup>36</sup>H. Hwang, T. Kim, H. Cynn, T. Vogt, R. J. Husband, K. Appel, C. Baehtz, O. B. Ball, M. A. Baron, R. Briggs, M. Bykov, E. Bykova, V. Cerantola, J. Chantel, A. L. Coleman, D. Dattlebaum, L. E. Dresselhaus-Marais, J. H. Eggert, L. Ehm, W. J. Evans, G. Fiquet, M. Frost, K. Glazyrin, A. F. Goncharov, Z. Jenei, J. Kim, Z. Konôpková, J. Mainberger, M. Makita, H. Marquardt, E. E. McBride, J. D. McHardy, S. Merkel, G. Morard, E. F. O'Bannon, C. Otzen, E. J. Pace, A. Pelka, C. M. Pépin, J. S. Pigott, V. B. Prakapenka, C. Prescher, R. Redmer, S. Speziale, G. Spiekermann, C. Strohm, B. T. Sturtevant, N. Velisavljevic, M. Wilke, C.-S. Yoo, U. Zastrau, H.-P. Liermann, M. I. McMahon, R. S. McWilliams, and Y. Lee, "X-ray Free Electron Laser-Induced Synthesis of -Iron Nitride at High Pressures," The Journal of Physical Chemistry Letters **12**, 3246–3252 (2021).
- <sup>37</sup>H. P. Liermann, Z. Konôpková, K. Appel, C. Prescher, A. Schropp, V. Cerantola, R. J. Husband, J. D. McHardy, M. I. McMahon, R. S. McWilliams, C. M. Pépin, J. Mainberger, M. Roeper, A. Berghäuser, H. Damker, P. Talkovski, M. Foese, N. Kujala, O. B. Ball, M. A. Baron, R. Briggs, M. Bykov, E. Bykova, J. Chantel, A. L. Coleman, H. Cynn, D. Dattelbaum, L. E. Dresselhaus-Marais, J. H. Eggert, L. Ehm, W. J. Evans, G. Fiquet, M. Frost, K. Glazyrin, A. F. Goncharov, H. Hwang, Z. Jenei, J.-Y. Kim, F. Langenhorst, Y. Lee, M. Makita, H. Marquardt, E. E. McBride, S. Merkel, G. Morard, E. F. O'Bannon, C. Otzen, E. J. Pace, A. Pelka, J. S. Pigott, V. B. Prakapenka, R. Redmer, C. Sanchez-Valle, M. Schoelmerich, S. Speziale, G. Spiekermann, B. T. Sturtevant, S. Toleikis, N. Velisavljevic, M. Wilke, C.-S. Yoo, C. Baehtz, U. Zastrau, and C. Strohm, "Novel experimental setup for megahertz X-ray diffraction in a diamond anvil cell at the High Energy Density (HED) instrument of the European X-ray Free-Electron Laser (EuXFEL)," Journal of Synchrotron Radiation 28 (2021), 10.1107/s1600577521002551.
- <sup>38</sup>K. R. Beyerlein, H. O. Jönsson, R. Alonso-Mori, A. Aquila, S. Bajt, A. Barty, R. Bean, J. E. Koglin, M. Messerschmidt, D. Ragazzon, D. Sokaras, G. J. Williams, S. Hau-Riege, S. Boutet, H. N. Chapman, N. Tîmneanu, and C. Caleman,

"Ultrafast nonthermal heating of water initiated by an X-ray Free-Electron Laser," Proceedings of the National Academy of Sciences **115**, 5652–5657 (2018).

- <sup>39</sup>A. M. Saunders, A. Jenei, T. Döppner, R. W. Falcone, D. Kraus, A. Kritcher, O. L. Landen, J. Nilsen, and D. Swift, "X-ray Thomson scattering measurements from hohlraum-driven spheres on the OMEGA laser," Review of scientific instruments 87, 11E724– 11E724 (2016).
- <sup>40</sup>A. M. Saunders, B. Lahmann, G. Sutcliffe, J. A. Frenje, R. W. Falcone, and T. Döppner, "Characterizing plasma conditions in radiatively heated solid-density samples with X-ray Thomson scattering," Physical Review E **98** (2018), 10.1103/Phys-RevE.98.063206.
- <sup>41</sup>K. Falk, C. L. Fryer, E. J. Gamboa, C. W. Greeff, H. M. Johns, D. W. Schmidt, M. Smíd, J. F. Benage, and D. S. Montgomery, "X-ray Thomson scattering measurement of temperature in warm dense carbon," (2016).
- <sup>42</sup>S. H. Glenzer and R. Redmer, "X-ray Thomson scattering in high energy density plasmas," Reviews of Modern Physics 81, 1625–1663 (2009).
- <sup>43</sup>L. Fletcher, A. Kritcher, A. Pak, T. Ma, T. Döppner, C. Fortmann, L. Divol, O. Jones, O. Landen, H. Scott, J. Vorberger, D. Chapman, D. Gericke, B. Mattern, G. Seidler, G. Gregori, R. Falcone, and S. Glenzer, "Observations of continuum depression in warm dense matter with X-ray thomson scattering," Physical review letters **112**, 145004–145004 (2014).
- <sup>44</sup>Z. Konôpková, W. Morgenroth, R. Husband, N. Giordano, A. Pakhomova, O. Gutowski, M. Wendt, K. Glazyrin, A. Ehnes, J. T. Delitz, A. F. Goncharov, V. B. Prakapenka, and H.-P. Liermann, "Laser heating system at the Extreme Conditions Beamline, P02.2, PETRA III," Journal of Synchrotron Radiation 28, 1747–1757 (2021).
- <sup>45</sup>P. Ni, M. Kulish, V. Mintsev, D. Nikolaev, V. Ternovoi, D. Hoffmann, S. Udrea, A. Hug, N. Tahir, and D. Varentsov, "Temperature measurement of warm-dense-matter generated by intense heavy-ion beams," Laser and Particle Beams 26, 583–589 (2008).
- <sup>46</sup>E. Principi, R. Cucini, A. Filipponi, A. Gessini, F. Bencivenga, F. D'Amico, A. D. Cicco, and C. Masciovecchio, "Determination of the Ion Temperature in a Stainless Steel Slab Exposed to Intense Ultrashort Laser Pulses," Physical Review Letters **109** (2012), 10.1103/physrevlett.109.025005.
- <sup>47</sup>A. Barducci and I. Pippi, "Temperature and emissivity retrieval from remotely sensed images using the "Grey body emissivity" method," IEEE Transactions on Geoscience and Remote Sensing **34**, 681–695 (1996).
- <sup>48</sup> J. E. Miller, T. R. Boehly, A. Melchior, D. D. Meyerhofer, P. M. Celliers, J. H. Eggert, D. G. Hicks, C. M. Sorce, J. A. Oertel, and P. M. Emmel, "Streaked optical pyrometer system for laser-driven shock-wave experiments on OMEGA," Review of Scientific Instruments **78**, 034903 (2007).
- <sup>49</sup>P. D. Asimow, "Melts Under Extreme Conditions From Shock Experiments," in *Magmas Under Pressure* (Elsevier, 2018) pp. 396–397.
- <sup>50</sup>E. Floyd, E. T. Gumbrell, J. Fyrth, J. D. Luis, J. W. Skidmore, S. Patankar, S. Giltrap, and R. Smith, "A high spatio-temporal resolution optical pyrometer at the ORION laser facility," Review of Scientific Instruments 87, 11E546 (2016).
- <sup>51</sup>L. R. Benedetti, N. Guignot, and D. L. Farber, "Achieving accuracy in spectroradiometric measurements of temperature in the laser-heated diamond anvil cell: Diamond is an optical component," Journal of Applied Physics **101** (2007), 10.1063/1.2402587.
- <sup>52</sup>S. Rekhi, J. Tempere, and I. F. Silvera, "Temperature determination for nanosecond pulsed laser heating," Review of Scientific Instruments **74**, 3820–3825 (2003).
- <sup>53</sup>J. Deng, Z. Du, L. R. Benedetti, and K. K. Lee, "The influence of wavelength-dependent absorption and temperature gradients on temperature determination in laser-heated diamond-anvil cells," Journal of Applied Physics **121** (2017), 10.1063/1.4973344.
- $^{54}\mathrm{G.}$  A. Lyzenga and T. J. Ahrens, "Multiwavelength optical py-

rometer for shock compression experiments," Review of Scientific Instruments **50**, 1421–1424 (1979).

- <sup>55</sup>A. Seifter, S. T. Stewart, M. R. Furlanetto, G. B. Kennedy, J. R. Payton, and A. W. Obst, "Post-shock temperature measurements of aluminum," (2006) pp. 139–142.
- <sup>56</sup>R. S. McWilliams, "Femtosecond Hard X-ray Studies at High Static Pressure," (2019), 10.22003/XFEL.EU-DATA-002292-00.
- <sup>57</sup>G. Feng, I. Zagorodnov, T. Limberg, H. Jin, Y. Kot, M. Dohlusand, and W. Decking, "Beam Dynamics Simulations for European XFEL," TESLA-FEL Report 2013-04. DESY, Hamburg, Germany (2013).
- <sup>58</sup>T. Maltezopoulos, F. Dietrich, W. Freund, U. F. Jastrow, A. Koch, J. Laksman, J. Liu, M. Planas, A. A. Sorokin, K. Tiedtke, and J. Grünert, "Operation of X-ray gas monitors at the European XFEL," Journal of Synchrotron Radiation 26, 1045–1051 (2019).
- $^{59}$  J. D. McHardy, under review (2022).
- <sup>60</sup>V. Cerantola, A. D. Rosa, Z. Konôpková, R. Torchio, E. Brambrink, A. Rack, U. Zastrau, and S. Pascarelli, "New frontiers in extreme conditions science at synchrotrons and free electron lasers," Journal of Physics: Condensed Matter **33**, 274003 (2021).
- <sup>61</sup>H.-P. Liermann, Z. Konopkova, K. Appel, H.-G. Damker, A. Schropp, R. S. Mcwilliams, A. Goncharov, and C. Beahtz, "Conceptual Design Report for Diamond Anvil Cell Setup (DAC) at the HED instrument of the European XFEL.; 2 – 5," Tech. Rep. (2016).
- <sup>62</sup>M. Nakatsutsumi and T. Tschentscher, "Conceptual Design Report: Scientific Instrument High Energy Density Physics (HED)," (2013), 10.3204/XFEL.EU/TR-2013-003.
- <sup>63</sup>U. Zastrau, K. Appel, C. Baehtz, O. Baehr, L. Batchelor, A. Berghäuser, M. Banjafar, E. Brambrink, V. Cerantola, T. E. Cowan, H. Damker, S. Dietrich, S. D. D. Cafiso, J. Dreyer, H.-O. Engel, T. Feldmann, S. Findeisen, M. Foese, D. Fulla-Marsa, S. Göde, M. Hassan, J. Hauser, T. Herrmannsdörfer, H. Höppner, J. Kaa, P. Kaever, K. Knöfel, Z. Konôpková, A. L. García, H.-P. Liermann, J. Mainberger, M. Makita, E.-C. Martens, E. E. McBride, D. Möller, M. Nakatsutsumi, A. Pelka, C. Plueckthun, C. Prescher, T. R. Preston, M. Röper, A. Schmidt, W. Seidel, J.-P. Schwinkendorf, M. O. Schoelmerich, U. Schramm, A. Schropp, C. Strohm, K. Sukharnikov, P. Talkovski, I. Thorpe, M. Toncian, T. Toncian, L. Wollenweber, S. Yamamoto, and T. Tschentscher, "The High Energy Density Scientific Instrument at the European XFEL," Journal of Synchrotron Radiation 28, 1393–1416 (2021).
- <sup>64</sup>R. S. McWilliams, Z. Konôpková, and A. F. Goncharov, "A flash heating method for measuring thermal conductivity at high pressure and temperature: application to Pt," Physics of the Earth and Planetary Interiors **247**, 17–26 (2015).
- <sup>65</sup> CRC handbook of chemistry and physics, 103rd ed. (CRC Press, Boca Raton, Fla, 1978).
- <sup>66</sup>K. Minuro, Test Report for high dynamic streak camera: Model C13410-01A, Hamamatsu Photonics K.K.
- <sup>67</sup>J. J. Moré, "The Levenberg-Marquardt algorithm: Implementation and theory," in *Numerical Analysis*, edited by G. A. Watson (Springer Berlin Heidelberg, Berlin, Heidelberg, 1978) pp. 105– 116.
- <sup>68</sup>C. R. Harris, K. J. Millman, S. J. van der Walt, R. Gommers, P. Virtanen, D. Cournapeau, E. Wieser, J. Taylor, S. Berg, N. J. Smith, R. Kern, M. Picus, S. Hoyer, M. H. van Kerkwijk, M. Brett, A. Haldane, J. Fernández del Río, M. Wiebe, P. Peterson, P. Gérard-Marchant, K. Sheppard, T. Reddy, W. Weckesser, H. Abbasi, C. Gohlke, and T. E. Oliphant, "Array programming with NumPy," Nature **585**, 357–362 (2020).
- <sup>69</sup>P. Virtanen, R. Gommers, T. E. Oliphant, M. Haberland, T. Reddy, D. Cournapeau, E. Burovski, P. Peterson, W. Weckesser, J. Bright, S. J. van der Walt, M. Brett, J. Wilson, K. J. Millman, N. Mayorov, A. R. J. Nelson, E. Jones, R. Kern, E. Larson, C. J. Carey, I. Polat, Y. Feng, E. W. Moore, J. VanderPlas, D. Laxalde, J. Perktold, R. Cimrman, I. Henriksen, E. A. Quintero, C. R. Harris, A. M. Archibald, A. H. Ribeiro, F. Pedregosa, P. van Mulbregt, and SciPy 1.0 Contributors,

"SciPy 1.0: Fundamental Algorithms for Scientific Computing in Python," Nature Methods **17**, 261–272 (2020).

- <sup>70</sup>J. M. Montgomery, M. J. Lipp, Z. Jenei, Y. Meng, and W. J. Evans, "A simple and portable multi-channel pyrometer allowing temperature measurements down to 800 K on the microsecond scale," Review of Scientific Instruments 89, 125117 (2018).
- <sup>71</sup>L. R. Benedetti and P. Loubeyre, "Temperature gradients, wavelength-dependent emissivity, and accuracy of high and veryhigh temperatures measured in the laser-heated diamond cell," High Pressure Research **24**, 423–445 (2004).
- <sup>72</sup>R. S. McWilliams, D. A. Dalton, M. F. Mahmood, and A. F. Goncharov, "Optical Properties of Fluid Hydrogen at the Transition to a Conducting State," Physical Review Letters **116** (2016), 10.1103/physrevlett.116.255501.
- <sup>73</sup>P. R. Bevington, Data reduction and error analysis for the physical sciences, third edition.. ed. (McGraw-Hill, New York ; London, 2003).
- <sup>74</sup>G. Shen, Y. Wang, A. Dewaele, C. Wu, D. E. Fratanduono, J. Eggert, S. Klotz, K. F. Dziubek, P. Loubeyre, O. V. Fatâyanov, P. D. Asimow, T. Mashimo, R. M. M. Wentzcovitch, and other members of the IPPS task group, "Toward an international practical pressure scale: A proposal for an IPPS ruby gauge (IPPS-Ruby2020)," High Pressure Research 40, 299–314 (2020), https://doi.org/10.1080/08957959.2020.1791107.
- <sup>75</sup>V. B. Prakapenka, N. Holtgrewe, S. S. Lobanov, and A. F. Goncharov, "Structure and properties of two superionic ice phases," Nature Physics **17**, 1233–1238 (2021).
- <sup>76</sup>L. B. Fletcher, H. J. Lee, B. Barbrel, M. Gauthier, E. Galtier, B. Nagler, T. Döppner, S. LePape, T. Ma, A. Pak, D. Turnbull, T. White, G. Gregori, M. Wei, R. W. Falcone, P. Heimann, U. Zastrau, J. B. Hastings, and S. H. Glenzer, "Exploring Mbar shock conditions and isochorically heated aluminum at the Matter in Extreme Conditions end station of the Linac Coherent Light Source (invited)," Review of Scientific Instruments **85**, 11E702 (2014).
- <sup>77</sup>W. Haynes, CRC Handbook of Chemistry and Physics, 91st Ed., CRC Handbook of Chemistry and Physics, 91st Ed. (National Institute of Standards and Technology, Boulder, C.O., 2011).
- <sup>78</sup>J. A. Montoya and A. F. Goncharov, "Finite element calculations of the time dependent thermal fluxes in the laser-heated diamond anvil cell," Journal of Applied Physics **111**, 112617 (2012).
- <sup>79</sup>I. V. Savchenko and S. V. Stankus, "Thermal conductivity and thermal diffusivity of tantalum in the temperature range from 293 to 1800 K," Thermophysics and Aeromechanics **15**, 679–682 (2008).
- <sup>80</sup>P. W. Bridgman, "The Resistance of 72 Elements, Alloys and Compounds to 100,000 Kg/Cm<sup>2</sup>," Proceedings of the American Academy of Arts and Sciences **81**, 165–251 (1952).
- <sup>81</sup>Y. Xu, T. J. Shankland, S. Linhardt, D. C. Rubie, F. Langenhorst, and K. Klasinski, "Thermal diffusivity and conductivity of olivine, wadsleyite and ringwoodite to 20 GPa and 1373 K," Physics of the Earth and Planetary Interiors **143-144**, 321–336 (2004), new Developments in High-Pressure Mineral Physics and Applications to the Earth's Interior.
- <sup>82</sup>Z. Lin, L. V. Zhigilei, and V. Celli, "Electron-phonon coupling and electron heat capacity of metals under conditions of strong electron-phonon nonequilibrium," Phys. Rev. B 77, 075133 (2008).
- <sup>83</sup>C. Su, Y. Liu, W. Song, D. Fan, Z. Wang, and H. Tang, "Thermodynamic properties of San Carlos olivine at high temperature and high pressure," Acta Geochimica **37**, 171–179 (2018).
- <sup>84</sup>A. Dewaele, P. Loubeyre, and M. Mezouar, "Equations of state of six metals above 94 GPa," Phys. Rev. B **70**, 094112 (2004).
- <sup>85</sup>W. Liu and B. Li, "Thermal equation of state of (Mg0.9Fe0.1)2SiO4 olivine," Physics of the Earth and Planetary Interiors 157, 188–195 (2006).
- <sup>86</sup>B. Henke, E. Gullikson, and J. Davis, "X-ray Interactions: Photoabsorption, Scattering, Transmission, and Reflection at E = 50-30,000 eV, Z = 1 – 92," Atomic Data and Nuclear Data Tables **54**, 181–342 (1993).

- <sup>88</sup> A. F. Goncharov, M. Wong, D. A. Dalton, J. G. O. Ojwang, V. V. Struzhkin, Z. Konôpková, and P. Lazor, "Thermal conductivity of argon at high pressures and high temperatures," Journal of Applied Physics **111**, 112609 (2012).
- <sup>89</sup>P. Andersson, "Thermal conductivity under pressure and through phase transitions in solid alkali halides. I. Experimental results for KCl, KBr, KI, RbCl, RbBr and RbI," Journal of Physics C: Solid State Physics 18, 3943 (1985).
- <sup>90</sup>S. Stackhouse, L. Stixrude, and B. B. Karki, "Thermal conductivity of periclase (MgO) from first principles," Physical review letters **104**, 208501 (2010).
- <sup>91</sup>D. A. Dalton, W.-P. Hsieh, G. T. Hohensee, D. G. Cahill, and A. F. Goncharov, "Effect of mass disorder on the lattice thermal conductivity of MgO periclase under pressure," Scientific Reports 3 (2013), 10.1038/srep02400.
- <sup>92</sup>M. W. Chase and N. I. S. O. (US), NIST-JANAF thermochemical tables, Vol. 9 (American Chemical Society Washington, DC, 1998).
- <sup>93</sup>A. Dewaele, F. Datchi, P. Loubeyre, and M. Mezouar, "High pressure-high temperature equations of state of neon and diamond," Physical Review B **77**, 094106 (2008).

- <sup>94</sup>P. I. Dorogokupets and A. Dewaele, "Equations of state of MgO, Au, Pt, NaCl-B1, and NaCl-B2: Internally consistent hightemperature pressure scales," High Pressure Research 27, 431– 446 (2007).
- <sup>95</sup>A. Dewaele, A. B. Belonoshko, G. Garbarino, F. Occelli, P. Bouvier, M. Hanfland, and M. Mezouar, "High-pressure-high-temperature equation of state of KCl and KBr," Physical Review B 85, 214105 (2012).
- <sup>96</sup>Y. Akahama and H. Kawamura, "Diamond anvil Raman gauge in multimegabar pressure range," High Pressure Research 27, 473–482 (2007).
- <sup>97</sup>J. H. Eggert, K. A. Goettel, and I. F. Silvera, "Elimination of pressure-induced fluorescence in diamond anvils," Applied Physics Letters **53**, 2489–2491 (1988).
- <sup>98</sup>M. Moszyński, T. Ludziejewski, D. Wolski, W. Klamra, and L. Norlin, "Properties of the YAG:Ce scintillator," Nuclear Instruments and Methods in Physics Research Section A: Accelerators, Spectrometers, Detectors and Associated Equipment **345**, 461–467 (1994).
- <sup>99</sup>H. K. Mao and R. J. Hemley, "Optical transitions in diamond at ultrahigh pressures," Nature **351**, 721–724 (1991).