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Temperature Effects on Saturation of Flow-Enhanced Nucleation

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Introduction

In melt-processing of semicrystalline polymers, flowinduced crystallization is inevitable. Understanding of this phenomenon is required for process optimization. *Flow-enhanced nucleation (FEN)* is considered here.

Model of FEN kinetics

The nucleation rate depends on the number of quiescent and *flow-induced precursors (FIPs)* available:

$$\dot{N}_n = \dot{N}_{nq} + \dot{N}_{nf} = \frac{N_{pq} + N_{pf}}{\tau_{pn}}$$

Pointlike FIPs form due to stretch of long chains, Λ :

$$\dot{N}_{pf} = g_p \left(\Lambda^4 - 1\right) \left(1 - \frac{N_{pf} + N_{nf}}{N_{f,max}}\right) - \dot{N}_{nf}$$

The number of FIPs and flow-induced nuclei saturates at $N_{f,max}$. The reptation time and Rouse time of the slowest relaxation mode change proportional to the number of FIPs ('branching' of long chains). Crystallites, growing from nuclei, behave like viscoelastic particles.¹



Figure 1: FIP formation on a stretched chain and nucleation after flow.

Experiments (Linkam shear cell)

Isotactic poly-1-butene ($\bar{M}_w = 176 \text{kg mol}^{-1}$, $\bar{M}_w / \bar{M}_n = 5.7^2$) was sheared, varying the rate $\dot{\gamma}$ and strain $\dot{\gamma}t_s$, after cooling to different temperatures (Table 1, Figure 2).

Table 1: Experimental conditions and initial longest relaxation times.

$T_m \ [^{\circ}\mathrm{C}]^2$		$\dot{\gamma} \left[\mathrm{s}^{-1} \right]$			$\lambda_{\rm o}^{\rm rep} [{\rm s}]^2$	$\lambda_0^{\rm R} [{\rm s}]^2$
110.7		\triangle		\bigtriangledown	0 [~]	0 [.1]
T[°C]	93	0.1	1	5	4.92	0.17
	98	0.1	1	10	3.99	0.14
	103	1	5	10	3.26	0.11

In situ microscopy showed mainly spherulites, but none during flow, even after the time when they appeared in the quiescent melt. Hence $\tau_{pn} (\dot{\gamma} > 0) \gg t_s$: FIPs nucleate only if the melt relaxes (Figure 1). The distribution of spherulite radii was narrow, thus $\tau_{pn} (\dot{\gamma} = 0) \rightarrow 0$.

Results

Using the Rolie-Poly model³ to calculate Λ , FEN data are reproduced well, except for short fast flows (Figure 2). The parameter g_p scales with the rheological shift factor:

$$g_p(T) = a_T(T, T_{ref}) g_{p, ref}$$

The saturation limit, taken from the data, shows a quadratic dependence on $Wi = \dot{\gamma} \lambda_0$ at low temperature and a 4th-power dependence at high temperature.



Figure 2: FEN and its saturation at different temperatures.

Conclusion

Higher temperature lowers the stretch-sensitivity of FEN via a_T (chain mobility) but increases the saturation limit. This unexpected result is a subject of current research.

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