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Prediction of yield stress development using structural relaxation

T.A.P.Engels, L.E.Govaert, G.W.M.Peters and H.E.H.Meijer

Introduction

In previous work [1] we showed that by extrapolating the results of the development of yield stress due to annealing treatments on a glassy polymer below the glass transition temperature, T_g , towards the development of properties during processing, we were able to predict the resulting distribution of yield stresses. In this previous approach, however, T_g was used as an input parameter. In the approach presented here [2], structural relaxation kinetics are used to describe the relaxation kinetics of the solidifying glass, and thus describing T_g .

TNM-Model

Structural relaxation has already been extensively used to describe a number of relaxation phenomena observed in polymers, e.g. volume and enthalpy. In figure 1 (left) relaxation of a property P (volume) with temperature is shown. Figure 1 (right) shows how this can be translated to the relaxation of the zero-viscosity.



Figure 1 Relaxation kinetics: annealing temperature, T_a , fictive temperature, T_f and glass transition temperature, T_g

In our approach we apply the Tool-Narayanswamy-Moynihan (TNM) model to the retardation of the zero-viscosity. The zero-viscosity is used rather than the yield stress itself, since the zero-viscosity is loading rate and -geometry independent. The equations describing the yield stress by means of the TNM model can be summarized as follows:

$$T_{f}(T,\xi) = T - \int_{0}^{\xi} M_{P}(\xi - \xi') \frac{dT}{d\xi'} d\xi'$$
 (1)

$$M_{P}(\xi) = \exp\left(-\left(\frac{\xi}{\tau_{Pr}}\right)^{\beta}\right) \qquad \xi(t) = \tau_{Pr} \int_{0}^{t} \frac{dt'}{\tau_{P}(T, T_{f})}$$
(2)

$$\tau_{\mathsf{P}}(\mathsf{T},\mathsf{T}_{\mathsf{f}}) = \mathsf{A}\exp\left(\frac{\mathsf{x}\Delta\mathsf{H}}{\mathsf{R}\mathsf{T}} + \frac{(1-\mathsf{x})\Delta\mathsf{H}}{\mathsf{R}\mathsf{T}_{\mathsf{f}}}\right) \tag{3}$$

$$\begin{split} \log_{10}(\eta_{0}(\mathsf{T},\xi)) &= \log_{10}\left(\frac{\mathsf{A}_{0}(\mathsf{S}_{\mathsf{a}})\mathsf{R}}{\mathsf{V}^{*}}\right) + \log_{10}(\mathsf{T}) + \frac{1}{2.303}\left(\frac{\Delta\mathsf{U}_{\mathsf{I}}}{\mathsf{R}}\right)\frac{1}{\mathsf{T}} \\ &+ \frac{1}{2.303}\left(\frac{\Delta\mathsf{U}_{\mathsf{I}} - \Delta\mathsf{U}_{\mathsf{g}}}{\mathsf{R}}\right)\left(\frac{1}{\mathsf{T}_{\mathsf{f}}(\xi)} - \frac{1}{\mathsf{T}}\right) \quad \text{(4)} \end{split}$$

 $= \sigma_s + \sigma_r$

$$\sigma_{\rm s} = 3\eta \dot{\epsilon}_0$$
 ; $\sigma_{\rm r} = {\sf G}_{\sf r}(\lambda_{\sf y}^2 - \lambda_{\sf y}^{-1})$ (6)

/department of mechanical engineering

Experimental

From a commercial grade of polycarbonate, Lexan 141R, injection molded samples were made. Mold temperatures were varied from 30° C to 130° C. Subsequently tensile bars were machined from the injection molded samples to determine the resulting yield stress, see figure 2 below.



Figure 2 Injection molded part and tensile bars made thereof

Results

The results show that both annealing close to T_g , and prediction of the yield stress distribution as it develops due to processing conditions can be described accurately by our approach.



Figure 3 Annealing close to T_g : symbols are experimental results, lines are model predictions (left); development of zero viscosity versus temperature, T_r is a reference temperature, here equal to T_g (right)



Figure 4 Distribution of the yield stress over the normalized part thickness (left); experimental yield stresses versus model predictions (right)

Conclusions

With the use of structural relaxation kinetics yield stresses can be predicted from processing conditions. This makes it possible to design a product for performance without ever doing a single experiment.

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(5)

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