

CaCO₃ FROM SEASHELLS AS A REINFORCING FILLER FOR NATURAL RUBBER

H. Norazlina^{*}, A.R.M. Fahmi and W.M. Hafizuddin

Faculty of Chemical Engineering Technology, TATI University College, Jln Panchor, Teluk Kalong, 24000, Kemaman, Terengganu, Malaysia ^{*}Email: norazlina@tatiuc.edu.my Phone: +6098601000; Fax: +6098635863

ABSTRACT

Bio-composites, due to their sustainably renewable nature, have attracted considerable attention due to their low cost and bio-degradability. $CaCO_3$ from seashells is a 'green' calcium carbonate that could enhance the mechanical and thermal properties of composites. The mixing of natural rubber (NR) with $CaCO_3$ using melt blending enhanced the hardness and tensile strength of the composites. The glass transition temperature did not have a significant effect on the composites, while the composites with filler of 355 µm in size showed the highest melting point in differential scanning calorimetry (DSC) analysis.

Keywords: Natural rubber; calcium carbonate; seashells; mechanical properties; thermal properties.

INTRODUCTION

Natural rubber (NR) is an important elastomers and widely utilized in a variety of rubber products. NR is often reinforced by incorporating a filler to improve its mechanical properties, such as modulus, hardness and tensile strength [1]. Generally, the reinforcing phase, in the form of particles, fibers or flakes, is embedded within the rubbery matrix phase. The newly obtained material exhibits unique properties with the overall properties being better than the individual properties [2-4]. The efficiency of the reinforcing fillers depends on several factors, such as the shape of the filler, the surface area and the particle size. The most often used reinforcing fillers are carbon black and silica (SiO₂) [5-7]. Renewable resources, such cornstarch, wood, and chitin, were previously the fillers of choice in polymers. The benefits of these fillers include their low cost, light weight and biodegradability [8-12]. Shells or exoskeletons from marine products, such as shrimp and cuttlebone, which are composed of different chemicals depending on their source, are used as fillers in NR. The exoskeletons of most crustaceans principally comprise calcium carbonate (CaCO₃), being typically more than 80% CaCO₃ by weight [13]. Therefore, no complex extraction process is necessary to use the product. Sometimes, an extraction process is used to determine the chemicals or eliminate other contaminating substances as in the case of chitin/chitosan. The protein content and pigments can cause problems in further utilization, especially in pharmaceutical and biomedical applications [14]. For this reason it should be removed. Chitin or CaCO₃ is known as a biocompatible material because of its low antigenicity, low toxicity and biodegradability. Several studies have reported using chitin as a filler. CaCO₃ is used in rubber as a de-tackifying agent to improve the rheology and processability, and impact strength, and also to reduce the cost of the formulation. It is also used as inert filler for tablets and dietary calcium supplement or antacid. An important use of CaCO₃ for environmental applications is to neutralize acidic conditions in water and to de-sulfurize flue gas and waste-water treatments [15]. This study aims to assess the possibility of using biodegradable CaCO₃ from product waste as a substitute for commercialized CaCO₃ to be the filler in NR by studying the mechanical and thermal properties. The mechanical studies focus on the tensile properties and hardness, while differential scanning calorimetry (DSC) analysis will study the T_g and T_m behavior of the composites.

EXPERIMENTAL SET-UP

Experimental Procedure

Standard Malaysia Rubber 10 (SMR-10) was used as the raw NR and supplied by Lee Rubber Sdn. Bhd., Kelantan. The NR was cut into small pieces with a weight of 100 phr for each sample using a cutter machine. The seashells were collected from Teluk Kalong Beach, Kemaman. Other chemicals were obtained commercially. The three major steps were preparation of the samples, mixing and characterization.

Preparation of Samples

In preparing the samples, the seashells were washed with distilled water several times to remove sand and dirt and were then dried in an oven overnight. Then, they were crushed into a powder and the powder foam was sieved using a 355 μ m size sieve. The effects of CaCO₃ loading between 10 phr to 60 in 100 phr of NR were studied using the tensile test to obtain the optimum percentage.



Figure 1. Initial study to find the optimum percentage of CaCO₃ in the NR blend.

After obtaining the optimum amount of $CaCO_3$ (40 phr $CaCO_3$), the research was furthered to form NR/40 phr $CaCO_3$ composites with different grain sizes of the $CaCO_3$: 112 µm, 355 µm, 800 µm and 1000 µm with optimum NR loading measured in the initial process.



Figure 2. Flow chart shows preparation of NR / 40 phr CaCO₃ blends with different grain sizes of the filler.

The samples were compounded via melt blending using a two-roll mill at 35-40 °C with four chemicals of fixed amounts added continuously during mixing to assist in acceleration and crosslinking. Initially, the NR was masticated in the mill for 15 min. Thereafter, the rubber was blended with zinc oxide (5 phr), stearic acid (2 phr), cyclohexylbenzothiazole-sulfenamide (CBS) (2.5 phr), CaCO₃ and sulfur (0.5 phr). The mixture was plasticized for 10–15 min and thin-passed several times at 70–80 °C. Last, the obtained samples were hot-pressed using a 1-mm thick mold to form sheet samples.

Characterization Process

All prepared samples were tested using Shore durometer hardness testing, in accordance with ASTM D2240 [16], as the measure of a resilient material's resistance to indentation. Each sample was tested three times to achieve measurement accuracy. During the hardness testing, the sample was put on a flat surface. The Shore durometer hardness tester was pressed onto the sample. After 15 seconds, the reading was shown by the durometer. Tensile testing measures the force required to break a plastic sample and the extent to which the sample stretches or elongates to that breaking point. The samples were cut into dumbbell shapes by using the ASTM D412 standard cutter [17]. The tensile test was undertaken using an Instron Universal Testing Machine.



Figure 3. Instron Universal Testing Machine for tensile testing and the dumbbell shaped samples.

Thermal analysis was undertaken using DSC employing a Mettler Toledo to determine the temperature of a sample to glass transition (T_g) and the melting point (T_m). During DSC characterization, the weighed sample was put into an aluminum pan. The pan was clamped using the clamp pan in the DSC equipment and the experiment was run at temperatures of -80 °C to 500 °C at a heating rate of 10 °C /min.

$$Tensiltestrenth(MPa) = \frac{Pulling force(N)}{Sample width(mm) \times Sample thickness(mm)}$$
(1)

Tensilte modulus (MPa) =
$$\frac{\text{Stress (Pa)}}{\text{Strin difference}}$$
 (2)
Sample heat flow (q) = $\frac{\text{Temperature difference between samples}}{\text{Resistance of the thermoelectric disk}}$ (3)

RESULTS AND DISCUSSION

Mechanical Properties of Samples

All samples were successfully prepared using a two-roll mill. In the initial stage, the NR/CaCO₃ composites were prepared with different filler loadings using the 355 µm filler size. This step was undertaken to find the optimum amount of CaCO₃ to give the highest tensile strength. Figure 4 shows the tensile strength of the samples. Using 100% NR, the tensile strength was 20 MPa but with increasing filler loading, a uniform increase was seen, due to reinforcement effect of CaCO₃ filler to NR, until 40 phr CaCO₃ was loaded [18]. It can be seen that the CaCO₃ particles could be fully wetted by the NR matrix, especially at low loading. The rubber chain could penetrate into the pores of the CaCO₃ and become a part of the filler (bound rubber). The presence of an organic component, such as chitin, is speculated to give a good reinforcement effect to the seashell particles during mixing with NR [19]. The further addition of CaCO₃ filler decreased the tensile strength due to agglomeration in the filler particles, thereafter being less wetted and less penetrated by natural rubber chains, which had a detrimental effect on the tensile strength [19]. According to these results, further studies should be undertaken using a 40 phr load of CaCO₃ in NR composites to study the effect of different sizes of filler on the mechanical and thermal properties.



Figure 4. Effect of CaCO₃ loading on tensile strength of NR/CaCO₃ composites.

Table 1 shows the results from the hardness and tensile tests of 100% NR and NR/40 phr CaCO₃ with different sizes of filler: 112 μ m, 355 μ m, 800 μ m and 1000 μ m. It is clearly seen that the increase in size of the filler enhanced the hardness and tensile modulus properties of the composites due to the decrease in elasticity of the NR. From

the hardness test, the composites are seen to be able to resist deformation during indentation. The increment in modulus is due to the degree of crosslinking and the more homogenous $CaCO_3$ filler distribution in the NR [20]. Increases in modulus often bring about decreases in elongation in filled-rubber materials [15]. This was proved by the results of elongation at break in Table 1. Only the addition of the 112 µm filler showed a significant effect on elongation, but for the next replenishment, a declining trend was shown. This is because the small sized filler would not cause disruption to the entangled polymer chains sliding freely when subjected to extension. The enhancement to the tensile strength was seen with the increasing filler size up to 355 µm and decreased for further loading with composites containing 800 μ m and 1000 μ m sizes of CaCO₃. This can be explained in that the smaller size of filler has a higher surface area which interacts well with the polymer matrix. The presence of the higher size filler highly affects the dispersed phase which finally affects the mechanical properties. In fact, as widely reported, filler aggregations, acting as local defects and reducing the interfacial area, could adversely affect the functional properties of final composites, hence, their reduction in the obtained analysis [21].

Size of CaCO ₃ in	Hardness	Elongation	Tensile	Tensile
NR/CaCO ₃	(Shore A)	at break	strength	modulus
composite (µm)		(mm)	(MPa)	(MPa)
0	10.8	349	20	1.9
112	10.83	380	24	2.0
355	11.37	375	25.2	2.2
800	13.57	358	21	2.6
1000	15.67	342	18.9	2.9

Table 1. Mechanical properties of NR/40 phr CaCO₃ with different CaCO₃ filler sizes.

Thermal Analysis of Samples

Figure 5 shows T_g and T_m from the DSC analysis. With regard to the T_g , the addition of CaCO₃ seashells as the filler produced no significant variation in T_g in the NR/40 phr CaCO₃ composites. The NR amorphous region was not influenced by the filler loading as has been seen in other articles [22] regardless of the nature of the polymeric matrix [23]. The melting point increased due to the higher size of the CaCO₃. The maximum T_m , 401.120 °C, was reached for the 355 µm size filler. For the fourth and fifth samples, with the 800 µm and 1000 µm filler sizes, a slight decrease in T_m was shown. This can be explained by the aggregation in the blending decreasing the compatibility of the mixture. The viscosity of the blends increased because of the poor flowability of neat NR; hence, the diffusion rate of the CaCO₃ filler chains was reduced. The rigidity of the composites was enhanced with the increments in the filler size; hence, the increased melting point.



Temperature (°C)

Figure 5. DSC analysis results for NR/40 phr CaCO₃ composites with different CaCO₃ sizes.

CONCLUSIONs

NR-biocomposites were produced by blending NR with filler obtained from nature. In the resulting composites, initial studies showed the highest tensile strength in the 40 phr loading of CaCO₃ in NR. From the tensile strength and melting point analysis using different filler sizes, the optimum properties were given by the 355 μ m NR/40 phr CaCO₃ sample and the melting point at 401.12 °C. The hardness reading increased with the increasing filler size to a maximum 15.67 Shore A shown by filler size 1000 μ m. The tensile modulus showed the same trend. There was no significant difference in the glass transition for all composites. The author suggests that further studies should be undertaken using different nanometer sizes of filler that it is believed can enhance the properties of the composites.

ACKNOWLEDGEMENTS

The authors gratefully acknowledge the Faculty of Chemical Engineering Technology, TATI University College for providing laboratory facilities and giving supports in this study.

REFERENCES

- [1] Byers J. Basic elastomer technology: fillers-B. Non-black. Akron: The Rubber Division American Chemical Society. 2001:82-111.
- [2] Ortiz-Serna P, Carsí M, Redondo-Foj B, Sanchis M. Electrical conductivity of natural rubber–cellulose II nanocomposites. Journal of Non-Crystalline Solids. 2014;405:180-7.

- [3] Mohamad N, Nur Sharafina Z, Ab Maulod HE, Yuhazri MY, Jeefferie AR. Morphological and mechanical properties of polypropylene/epoxidized natural rubber thermoplastic vulcanizates treated with maleic anhydride-grafted polypropylene. International Journal of Automotive and Mechanical Engineering. 2013;8:1305-15.
- [4] Ravi Sankar H, Srikant RR, Vamsi Krishna P, Bhujanga Rao V, Bangaru Babu P. Estimation of the dynamic properties of epoxy glass fabric composites with natural rubber particle inclusions. International Journal of Automotive and Mechanical Engineering. 2013;7:968-80.
- [5] Kanking S, Niltui P, Wimolmala E, Sombatsompop N. Use of bagasse fiber ash as secondary filler in silica or carbon black filled natural rubber compound. Materials & Design. 2012;41:74-82.
- [6] Ooi ZX, Ismail H, Bakar AA. Optimisation of oil palm ash as reinforcement in natural rubber vulcanisation: A comparison between silica and carbon black fillers. Polymer testing. 2013;32:625-30.
- [7] Rattanasom N, Saowapark T, Deeprasertkul C. Reinforcement of natural rubber with silica/carbon black hybrid filler. Polymer Testing. 2007;26:369-77.
- [8] Yu L, Dean K, Li L. Polymer blends and composites from renewable resources. Progress in polymer science. 2006;31:576-602.
- [9] Abdul Majid MS, Daud R, Afendi M, Amin NAM, Cheng EM, Gibson AG, et al. Stress-strain response modelling of glass fibre reinforced epoxy composite pipes under multiaxial loadings. Journal of Mechanical Engineering and Sciences. 2014;6:916-28.
- [10] Taufik RS, Adibah M NF, Muhamad MR, Hasib H. Feasibility study of natural fiber composite material for engineering application. Journal of Mechanical Engineering and Sciences. 2014;6:940-8.
- [11] Mat Hassan NN, M. Rus AZ. Acoustic performance of green polymer foam from renewable resources after UV exposure. International Journal of Automotive and Mechanical Engineering. 2014;9:1639-48.
- [12] Mei GS, Yuvaraj AR, Kabeb SMb, Yusoff MM, Chigrinov VG, Hegde G. Biocompatible Polymer Embedded in Light-Sensitive Materials: Investigation of Structural Properties. International Journal of Automotive and Mechanical Engineering. 2014;10:2025-33.
- [13] Kohjiya S, Ikeda Y. Chemistry, manufacture and applications of natural rubber: Elsevier; 2014.
- [14] Jing P, Ruan S-Y, Dong Y, Zhang X-G, Yue J, Kan J-Q, et al. Optimization of purification conditions of radish (Raphanus sativus L.) anthocyanin-rich extracts using chitosan. Lwt-Food Sci Technol. 2011;44:2097-103.
- [15] Poompradub S. Soft bio-composites from natural rubber (NR) and marine products. Chemistry, Manufacture and Applications of Natural Rubber. 2014:303.
- [16] Testing ASf, Materials. Standard Test Method for Rubber Property--Durometer Hardness. ASTM; 2005.
- [17] Materials ASfTa. D412—Standard Test Method for Vulcanized Rubber and thermoplastic Elastomers. 2006.
- [18] Fang Q, Song B, Tee T-T, Sin LT, Hui D, Bee S-T. Investigation of dynamic characteristics of nano-size calcium carbonate added in natural rubber vulcanizate. Composites Part B: Engineering. 2014;60:561-7.

- [19] Poompradub S, Ikeda Y, Kokubo Y, Shiono T. Cuttlebone as reinforcing filler for natural rubber. European Polymer Journal. 2008;44:4157-64.
- [20] Nabil H, Ismail H. Enhancing the thermal stability of natural rubber/recycled ethylene–propylene–diene rubber blends by means of introducing pre-vulcanised ethylene–propylene–diene rubber and electron beam irradiation. Materials & Design. 2014;56:1057-67.
- [21] Russo P, Acierno D, Rosa R, Leonelli C, Corradi A, Rizzuti A. Mechanical and dynamic-mechanical behavior and morphology of polystyrene/perovskite composites: Effects of filler size. Surface & Coatings Technology. 2014;243:65-70.
- [22] Bras J, Hassan ML, Bruzesse C, Hassan EA, El-Wakil NA, Dufresne A. Mechanical, barrier, and biodegradability properties of bagasse cellulose whiskers reinforced natural rubber nanocomposites. Industrial Crops and Products. 2010;32:627-33.
- [23] Angles MN, Dufresne A. Plasticized starch/tunicin whiskers nanocomposite materials. 2. Mechanical behavior. Macromolecules. 2001;34:2921-31.