# $\mathrm{CaCO}_{3}$ FROM SEASHELLS AS A REINFORCING FILLER FOR NATURAL RUBBER 

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#### Abstract

Bio-composites, due to their sustainably renewable nature, have attracted considerable attention due to their low cost and bio-degradability. $\mathrm{CaCO}_{3}$ from seashells is a 'green' calcium carbonate that could enhance the mechanical and thermal properties of composites. The mixing of natural rubber (NR) with $\mathrm{CaCO}_{3}$ using melt blending enhanced the hardness and tensile strength of the composites. The glass transition temperature did not have a significant effect on the composites, while the composites with filler of $355 \mu \mathrm{~m}$ in size showed the highest melting point in differential scanning calorimetry (DSC) analysis.


Keywords: Natural rubber; calcium carbonate; seashells; mechanical properties; thermal properties.

## INTRODUCTION

Natural rubber (NR) is an important elastomers and widely utilized in a variety of rubber products. NR is often reinforced by incorporating a filler to improve its mechanical properties, such as modulus, hardness and tensile strength [1]. Generally, the reinforcing phase, in the form of particles, fibers or flakes, is embedded within the rubbery matrix phase. The newly obtained material exhibits unique properties with the overall properties being better than the individual properties [2-4]. The efficiency of the reinforcing fillers depends on several factors, such as the shape of the filler, the surface area and the particle size. The most often used reinforcing fillers are carbon black and silica $\left(\mathrm{SiO}_{2}\right)$ [5-7]. Renewable resources, such cornstarch, wood, and chitin, were previously the fillers of choice in polymers. The benefits of these fillers include their low cost, light weight and biodegradability [8-12]. Shells or exoskeletons from marine products, such as shrimp and cuttlebone, which are composed of different chemicals depending on their source, are used as fillers in NR. The exoskeletons of most crustaceans principally comprise calcium carbonate $\left(\mathrm{CaCO}_{3}\right)$, being typically more than $80 \% \mathrm{CaCO}_{3}$ by weight [13]. Therefore, no complex extraction process is necessary to use the product. Sometimes, an extraction process is used to determine the chemicals or eliminate other contaminating substances as in the case of chitin/chitosan. The protein content and pigments can cause problems in further utilization, especially in pharmaceutical and biomedical applications [14]. For this reason it should be removed. Chitin or $\mathrm{CaCO}_{3}$ is known as a biocompatible material because of its low antigenicity, low toxicity and biodegradability. Several studies have reported using chitin as a filler. $\mathrm{CaCO}_{3}$ is used in rubber as a de-tackifying agent to improve the rheology and
processability, and impact strength, and also to reduce the cost of the formulation. It is also used as inert filler for tablets and dietary calcium supplement or antacid. An important use of $\mathrm{CaCO}_{3}$ for environmental applications is to neutralize acidic conditions in water and to de-sulfurize flue gas and waste-water treatments [15]. This study aims to assess the possibility of using biodegradable $\mathrm{CaCO}_{3}$ from product waste as a substitute for commercialized $\mathrm{CaCO}_{3}$ to be the filler in NR by studying the mechanical and thermal properties. The mechanical studies focus on the tensile properties and hardness, while differential scanning calorimetry (DSC) analysis will study the $T_{g}$ and $T_{m}$ behavior of the composites.

## EXPERIMENTAL SET-UP

## Experimental Procedure

Standard Malaysia Rubber 10 (SMR-10) was used as the raw NR and supplied by Lee Rubber Sdn. Bhd., Kelantan. The NR was cut into small pieces with a weight of 100 phr for each sample using a cutter machine. The seashells were collected from Teluk Kalong Beach, Kemaman. Other chemicals were obtained commercially. The three major steps were preparation of the samples, mixing and characterization.

## Preparation of Samples

In preparing the samples, the seashells were washed with distilled water several times to remove sand and dirt and were then dried in an oven overnight. Then, they were crushed into a powder and the powder foam was sieved using a $355 \mu \mathrm{~m}$ size sieve. The effects of $\mathrm{CaCO}_{3}$ loading between 10 phr to 60 in 100 phr of NR were studied using the tensile test to obtain the optimum percentage.


Figure 1. Initial study to find the optimum percentage of $\mathrm{CaCO}_{3}$ in the NR blend.
After obtaining the optimum amount of $\mathrm{CaCO}_{3}\left(40 \mathrm{phr} \mathrm{CaCO}_{3}\right)$, the research was furthered to form NR/40 phr $\mathrm{CaCO}_{3}$ composites with different grain sizes of the $\mathrm{CaCO}_{3}: 112 \mu \mathrm{~m}, 355 \mu \mathrm{~m}, 800 \mu \mathrm{~m}$ and $1000 \mu \mathrm{~m}$ with optimum NR loading measured in the initial process.


Figure 2. Flow chart shows preparation of NR / $40 \mathrm{phr} \mathrm{CaCO}_{3}$ blends with different grain sizes of the filler.

The samples were compounded via melt blending using a two-roll mill at 35$40^{\circ} \mathrm{C}$ with four chemicals of fixed amounts added continuously during mixing to assist in acceleration and crosslinking. Initially, the NR was masticated in the mill for 15 min . Thereafter, the rubber was blended with zinc oxide ( 5 phr ), stearic acid ( 2 phr ), cyclohexylbenzothiazole-sulfenamide (CBS) ( 2.5 phr ), $\mathrm{CaCO}_{3}$ and sulfur ( 0.5 phr ). The mixture was plasticized for $10-15 \mathrm{~min}$ and thin-passed several times at $70-80^{\circ} \mathrm{C}$. Last, the obtained samples were hot-pressed using a $1-\mathrm{mm}$ thick mold to form sheet samples.

## Characterization Process

All prepared samples were tested using Shore durometer hardness testing, in accordance with ASTM D2240 [16], as the measure of a resilient material's resistance to indentation. Each sample was tested three times to achieve measurement accuracy. During the hardness testing, the sample was put on a flat surface. The Shore durometer hardness tester was pressed onto the sample. After 15 seconds, the reading was shown by the durometer. Tensile testing measures the force required to break a plastic sample and the extent to which the sample stretches or elongates to that breaking point. The samples were cut into dumbbell shapes by using the ASTM D412 standard cutter [17]. The tensile test was undertaken using an Instron Universal Testing Machine.


Figure 3. Instron Universal Testing Machine for tensile testing and the dumbbell shaped samples.

Thermal analysis was undertaken using DSC employing a Mettler Toledo to determine the temperature of a sample to glass transition $\left(T_{g}\right)$ and the melting point $\left(T_{m}\right)$. During DSC characterization, the weighed sample was put into an aluminum pan. The pan was clamped using the clamp pan in the DSC equipment and the experiment was run at temperatures of $-80^{\circ} \mathrm{C}$ to $500^{\circ} \mathrm{C}$ at a heating rate of $10^{\circ} \mathrm{C} / \mathrm{min}$.

$$
\begin{equation*}
\text { Tensilte strenth }(\mathrm{MPa})=\frac{\text { Pulling force }(\mathrm{N})}{\text { Sample width }(\mathrm{mm}) \times \text { Sample thickness }(\mathrm{mm})} \tag{1}
\end{equation*}
$$

$$
\begin{gather*}
\text { Tensilte modulus }(\mathrm{MPa})=\frac{\text { Stress }(\mathrm{Pa})}{\text { Strin difference }}  \tag{2}\\
\text { Sample heat flow }(q)=\frac{\text { Temperature difference between samples }}{\text { Resistance of the thermoelectric disk }} \tag{3}
\end{gather*}
$$

## RESULTS AND DISCUSSION

## Mechanical Properties of Samples

All samples were successfully prepared using a two-roll mill. In the initial stage, the $\mathrm{NR} / \mathrm{CaCO}_{3}$ composites were prepared with different filler loadings using the $355 \mu \mathrm{~m}$ filler size. This step was undertaken to find the optimum amount of $\mathrm{CaCO}_{3}$ to give the highest tensile strength. Figure 4 shows the tensile strength of the samples. Using $100 \%$ NR, the tensile strength was 20 MPa but with increasing filler loading, a uniform increase was seen, due to reinforcement effect of $\mathrm{CaCO}_{3}$ filler to NR, until 40 phr $\mathrm{CaCO}_{3}$ was loaded [18]. It can be seen that the $\mathrm{CaCO}_{3}$ particles could be fully wetted by the NR matrix, especially at low loading. The rubber chain could penetrate into the pores of the $\mathrm{CaCO}_{3}$ and become a part of the filler (bound rubber). The presence of an organic component, such as chitin, is speculated to give a good reinforcement effect to the seashell particles during mixing with NR [19]. The further addition of $\mathrm{CaCO}_{3}$ filler decreased the tensile strength due to agglomeration in the filler particles, thereafter being less wetted and less penetrated by natural rubber chains, which had a detrimental effect on the tensile strength [19]. According to these results, further studies should be undertaken using a 40 phr load of $\mathrm{CaCO}_{3}$ in NR composites to study the effect of different sizes of filler on the mechanical and thermal properties.


Figure 4. Effect of $\mathrm{CaCO}_{3}$ loading on tensile strength of $\mathrm{NR} / \mathrm{CaCO}_{3}$ composites.
Table 1 shows the results from the hardness and tensile tests of $100 \%$ NR and $\mathrm{NR} / 40 \mathrm{phr} \mathrm{CaCO}_{3}$ with different sizes of filler: $112 \mu \mathrm{~m}, 355 \mu \mathrm{~m}, 800 \mu \mathrm{~m}$ and $1000 \mu \mathrm{~m}$. It is clearly seen that the increase in size of the filler enhanced the hardness and tensile modulus properties of the composites due to the decrease in elasticity of the NR. From
the hardness test, the composites are seen to be able to resist deformation during indentation. The increment in modulus is due to the degree of crosslinking and the more homogenous $\mathrm{CaCO}_{3}$ filler distribution in the NR [20]. Increases in modulus often bring about decreases in elongation in filled-rubber materials [15]. This was proved by the results of elongation at break in Table 1. Only the addition of the $112 \mu \mathrm{~m}$ filler showed a significant effect on elongation, but for the next replenishment, a declining trend was shown. This is because the small sized filler would not cause disruption to the entangled polymer chains sliding freely when subjected to extension. The enhancement to the tensile strength was seen with the increasing filler size up to $355 \mu \mathrm{~m}$ and decreased for further loading with composites containing $800 \mu \mathrm{~m}$ and $1000 \mu \mathrm{~m}$ sizes of $\mathrm{CaCO}_{3}$. This can be explained in that the smaller size of filler has a higher surface area which interacts well with the polymer matrix. The presence of the higher size filler highly affects the dispersed phase which finally affects the mechanical properties. In fact, as widely reported, filler aggregations, acting as local defects and reducing the interfacial area, could adversely affect the functional properties of final composites, hence, their reduction in the obtained analysis [21].

Table 1. Mechanical properties of $\mathrm{NR} / 40$ phr $\mathrm{CaCO}_{3}$ with different $\mathrm{CaCO}_{3}$ filler sizes.

| Size of $\mathrm{CaCO}_{3}$ in <br> $\mathrm{NR} / \mathrm{CaCO}_{3}$ <br> composite $(\mu \mathrm{m})$ | Hardness <br> $($ Shore A) | Elongation <br> at break <br> $(\mathrm{mm})$ | Tensile <br> strength <br> $(\mathrm{MPa})$ | Tensile <br> modulus <br> $(\mathrm{MPa})$ |
| :---: | :---: | :---: | :---: | :---: |
| 0 | 10.8 | 349 | 20 | 1.9 |
| 112 | 10.83 | 380 | 24 | 2.0 |
| 355 | 11.37 | 375 | 25.2 | 2.2 |
| 800 | 13.57 | 358 | 21 | 2.6 |
| 1000 | 15.67 | 342 | 18.9 | 2.9 |

## Thermal Analysis of Samples

Figure 5 shows $T_{g}$ and $T_{m}$ from the DSC analysis. With regard to the $T_{g}$, the addition of $\mathrm{CaCO}_{3}$ seashells as the filler produced no significant variation in $T_{g}$ in the NR/40 phr $\mathrm{CaCO}_{3}$ composites. The NR amorphous region was not influenced by the filler loading as has been seen in other articles [22] regardless of the nature of the polymeric matrix [23]. The melting point increased due to the higher size of the $\mathrm{CaCO}_{3}$. The maximum $T_{m}, 401.120{ }^{\circ} \mathrm{C}$, was reached for the $355 \mu \mathrm{~m}$ size filler. For the fourth and fifth samples, with the $800 \mu \mathrm{~m}$ and $1000 \mu \mathrm{~m}$ filler sizes, a slight decrease in $T_{m}$ was shown. This can be explained by the aggregation in the blending decreasing the compatibility of the mixture. The viscosity of the blends increased because of the poor flowability of neat NR ; hence, the diffusion rate of the $\mathrm{CaCO}_{3}$ filler chains was reduced. The rigidity of the composites was enhanced with the increments in the filler size; hence, the increased melting point.


Temperature $\left({ }^{\circ} \mathrm{C}\right)$
Figure 5. DSC analysis results for NR/40 phr $\mathrm{CaCO}_{3}$ composites with different $\mathrm{CaCO}_{3}$ sizes.

## CONCLUSIONs

NR-biocomposites were produced by blending NR with filler obtained from nature. In the resulting composites, initial studies showed the highest tensile strength in the 40 phr loading of $\mathrm{CaCO}_{3}$ in NR. From the tensile strength and melting point analysis using different filler sizes, the optimum properties were given by the $355 \mu \mathrm{~m} \mathrm{NR} / 40 \mathrm{phr}$ $\mathrm{CaCO}_{3}$ sample and the melting point at $401.12{ }^{\circ} \mathrm{C}$. The hardness reading increased with the increasing filler size to a maximum 15.67 Shore A shown by filler size $1000 \mu \mathrm{~m}$. The tensile modulus showed the same trend. There was no significant difference in the glass transition for all composites. The author suggests that further studies should be undertaken using different nanometer sizes of filler that it is believed can enhance the properties of the composites.

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