

Influence of alkaline-earth substitution on structure and oxygen transport of perovskite-type oxides $\text{La}_{0.6}\text{A}_{0.4}\text{FeO}_{3-\delta}$ ($\text{A} = \text{Ca}, \text{Sr}, \text{Ba}$)

Jia Song^{(a)}, De Ning^(c), Bernard A. Boukamp^(a), Yang Yi^(b), Changrong Xia^(b), Henny J.M. Bouwmeester^(a,b)*

✉ j.song-2@utwente.nl

Background & Aims

- Alkaline earth substituted lanthanum ferrites are good candidate cathode materials for solid oxide fuel cells (SOFCs) due to its high mixed ionic-electronic conductivity.
- Limited research has been conducted to systematically distinguish the influence of Ca, Sr and Ba substitution, while $\text{La}_{1-x}\text{Sr}_x\text{FeO}_{3-\delta}$ has been extensively used as a cathode material.
- To investigate the influence of alkaline earth substitution on crystal structure, electrical conductivity and oxygen transport properties of $\text{La}_{0.6}\text{A}_{0.4}\text{FeO}_{3-\delta}$ ($\text{A} = \text{Ca}, \text{Sr}$ and Ba).

Crystal structure

- Thermally stable up to 1000 °C
- Phase transition → loss of BO_6 octahedra tilt system
- According to Price *et al.*¹ LCF64 shows a sequence of $a^0a^0a^0 \rightarrow a^0a^0a^* \rightarrow a^0a^0a^*$ during cooling, in Glazer's notation.

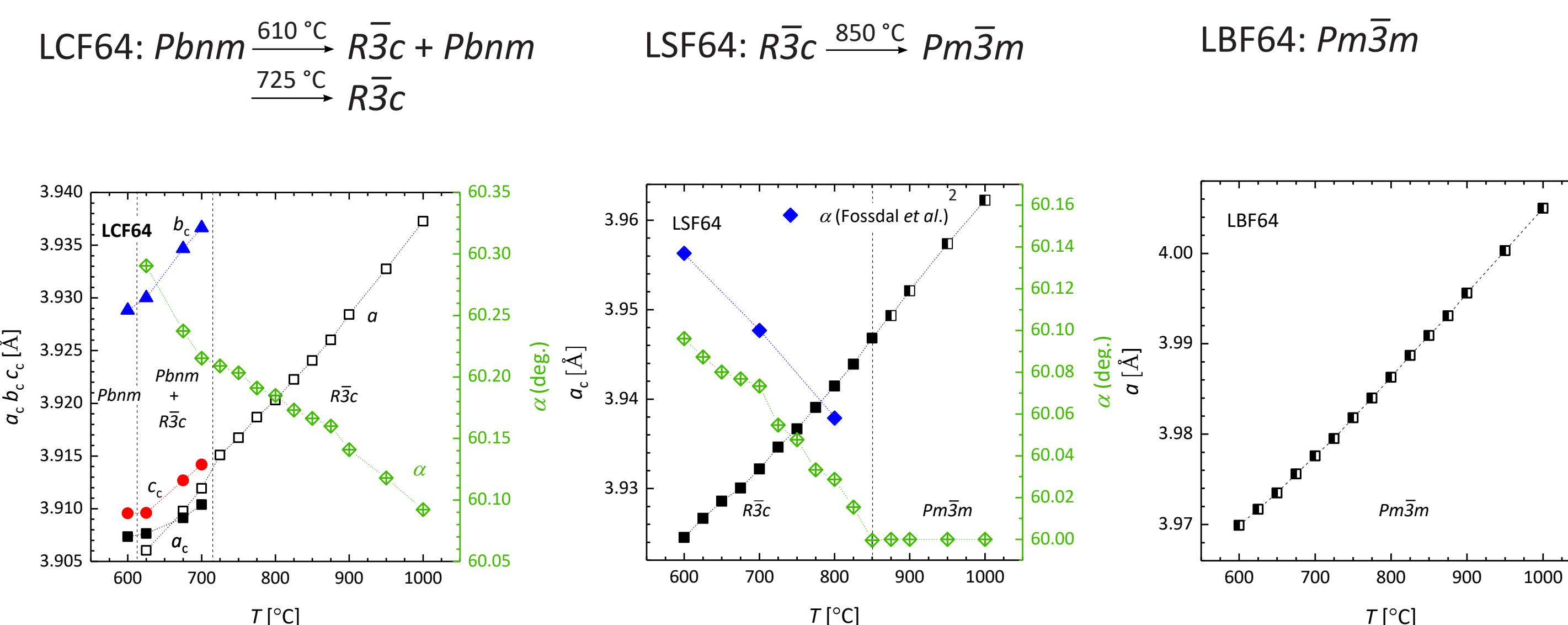
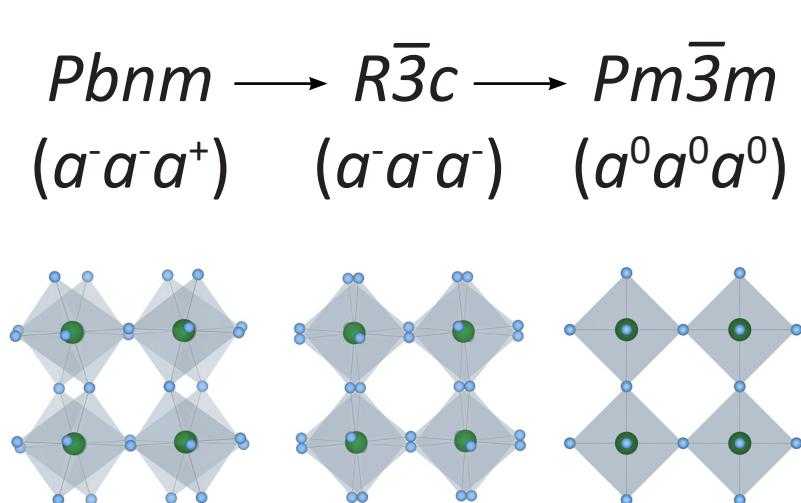


Fig. 1 Pseudo-cubic lattice parameters for LCF64, LSF64, and LBF64 as a function of temperature from Rietveld refinement of HT-XRD data recorded in the range 600-1000 °C in air.

Electrical conductivity

- In the temperature range 650-1000 °C and $p\text{O}_2 = 0.21$ bar:
- $\sigma_{\text{LSF64}} > \sigma_{\text{LCF64}} \approx \sigma_{\text{LBF64}}$
- $\mu_p = \sigma / (N_{\text{ABO}_3} e [\text{Fe}^{3+}/\text{Fe}^{2+}])$
 N_{ABO_3} the number of unit cell per unit volume
- Mobility of the electron hole (μ_p)

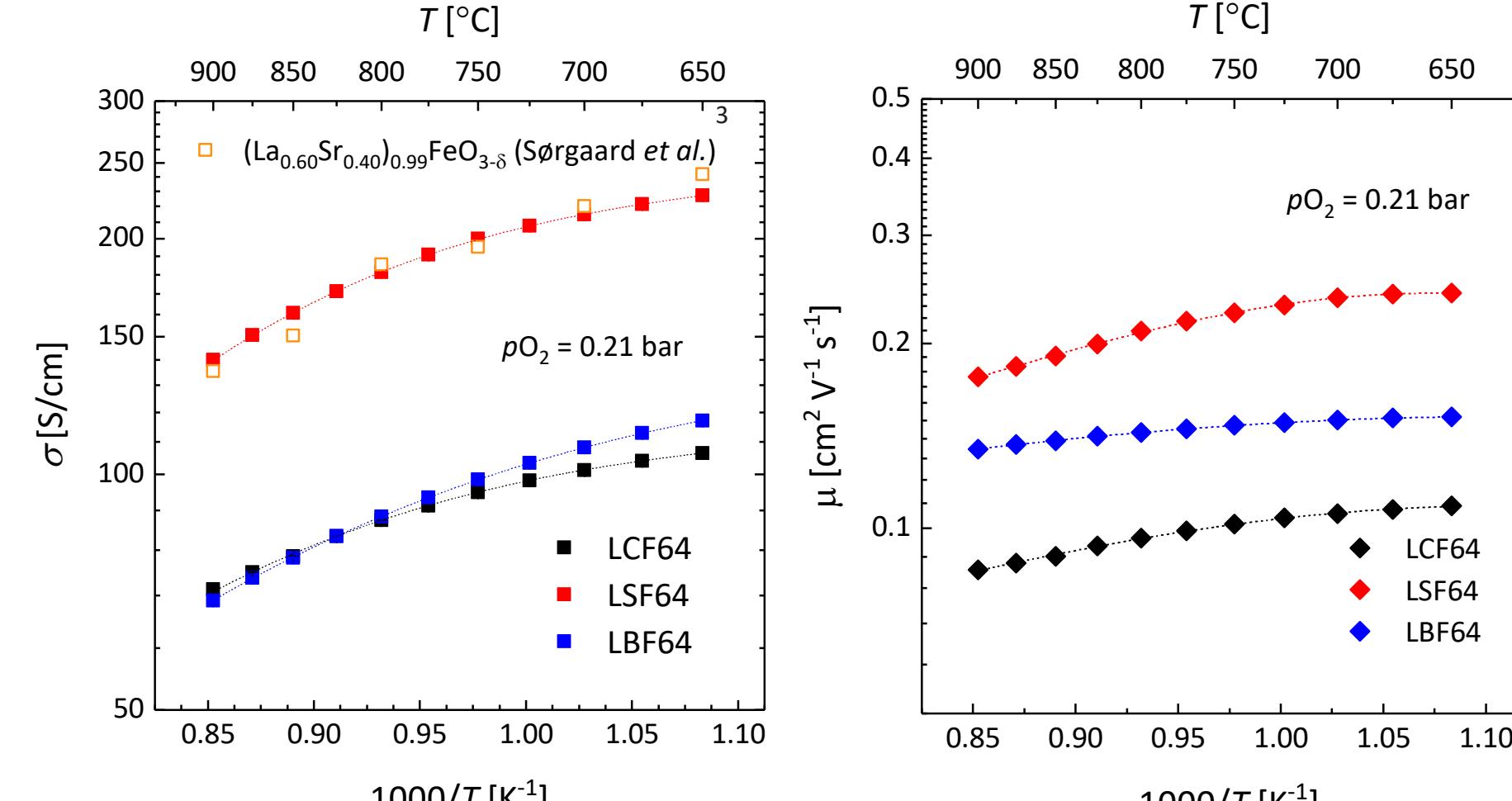


Fig. 2 Reciprocal temperature dependence of electrical conductivity (left) and charge carrier ($[\text{Fe}^{3+}/\text{Fe}^{2+}]$) mobility (right) from 650°C to 900 °C at $p\text{O}_2 = 0.21$ bar. LSF64: good agreement is found with data from literature.²

Oxygen nonstoichiometry

- Oxygen nonstoichiometry (δ) was studied using thermogravimetric analysis (TGA) in the temperature range of 650-900 °C and $p\text{O}_2$ range of 0.045-0.9 bar

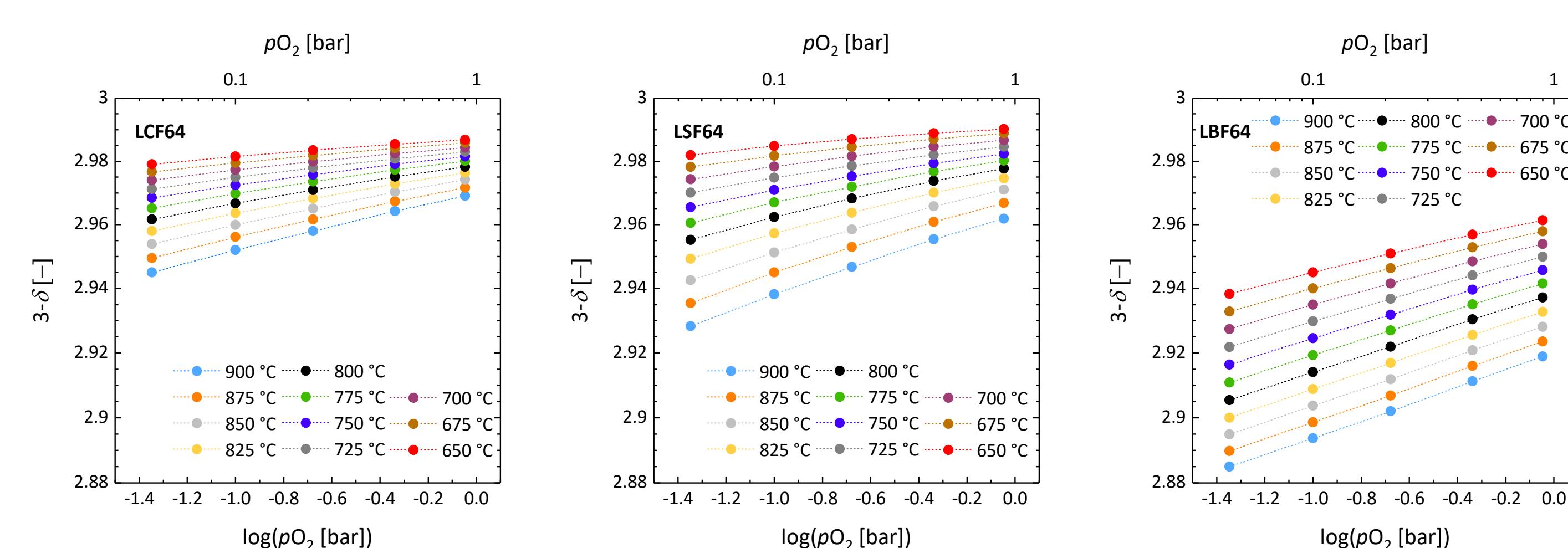


Fig. 3 Oxygen stoichiometry ($3-\delta$) of LCF64, LSF64 and LBF64 in response of $p\text{O}_2$ at different temperature.

Electrical Conductivity Relaxation (ECR)

Thermal History



- Surface exchange coefficient (k_{chem}) and chemical diffusion coefficient (D_{chem}) were derived from the nonlinear least-square fitting of ECR data.

- Significant similarities between D_{chem} values of LCF64, LSF64 and LBF64

- Activation energy of k_{chem} :

$$E_a^{\text{LCF64}} = 1.51 \pm 0.02 \text{ eV}$$

$$E_a^{\text{LSF64}} \approx E_a^{\text{LBF64}} = 0.89 \pm 0.01 \text{ eV}$$

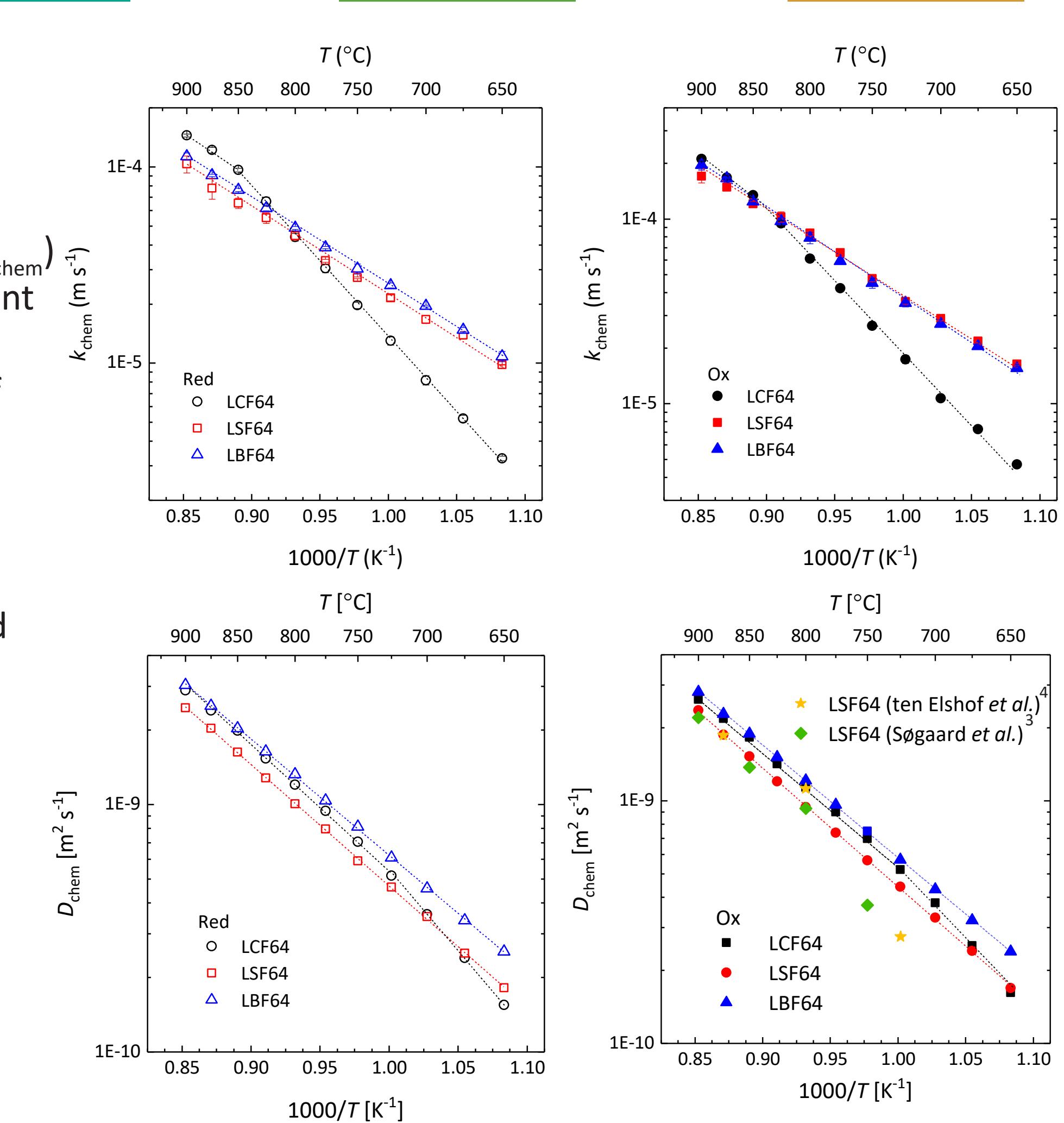


Fig. 4 Inverse temperature dependence of the surface exchange coefficient (k_{chem}) and chemical diffusion coefficient (D_{chem}) upon reduction and oxidation run for LCF64, LSF64 and LBF64 in the temperature range 650-900 °C between 0.1 and 0.21 bar.

Self-diffusion coefficient (D_s) & vacancy diffusion coefficient (D_v)

- D_s and D_v was derived by: $D_{\text{chem}} = \gamma_O D_s = \gamma_V D_v$ where γ_O, γ_V thermodynamic factors of oxygen ion, oxygen vacancy in the crystal, respectively.

$$\gamma_O = \frac{1}{2} \cdot \frac{\partial \ln(p\text{O}_2)}{\partial \ln(C_O)} = \frac{3-\delta}{2} \cdot \frac{\partial \ln(p\text{O}_2)}{\partial(3-\delta)}$$

$$\gamma_V = -\frac{1}{2} \cdot \frac{\partial \ln(p\text{O}_2)}{\partial \ln(C_V)} = -\frac{\delta}{2} \cdot \frac{\partial \ln(p\text{O}_2)}{\partial \delta} = \frac{3-\delta}{\delta} \gamma_O$$

- Significant similarities in D_s and D_v value between LCF64, LSF64 and LBF64

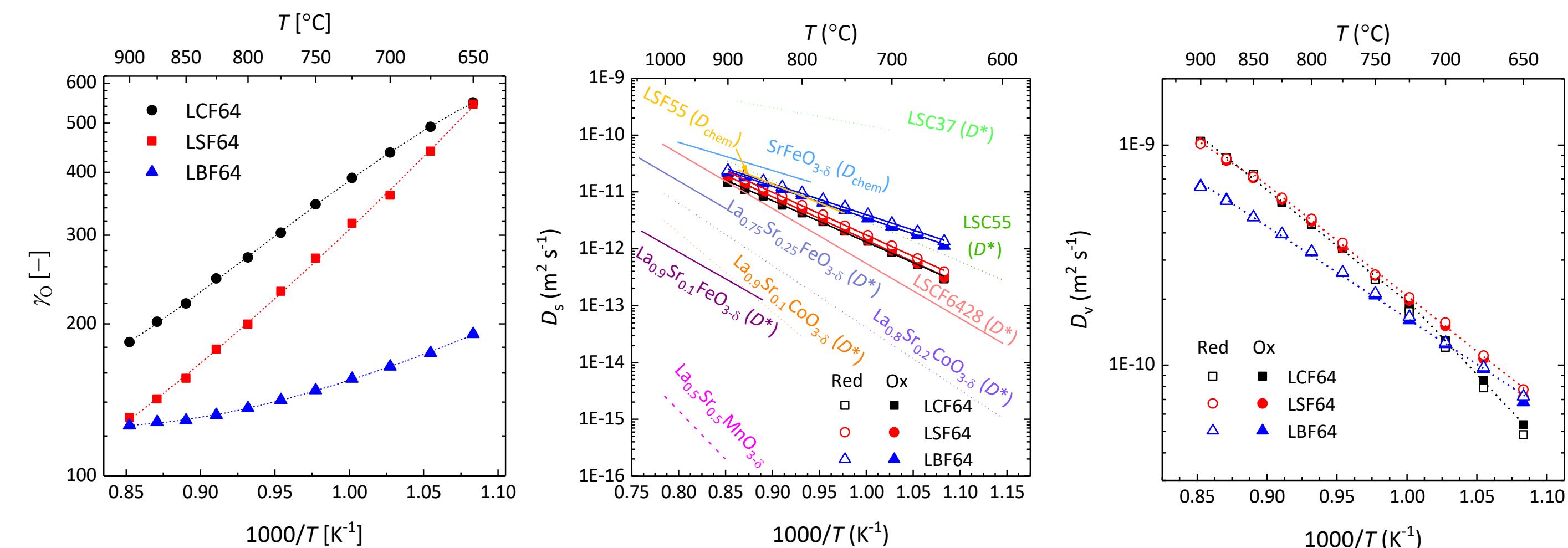


Fig. 5 Inverse temperature dependence of the oxygen ion thermodynamic factor (γ_O), self-diffusion coefficient (D_s) and vacancy diffusion coefficient (D_v) upon reduction and oxidation run for LCF64, LSF64 and LBF64. D_s values of LCF64, LSF64 and LBF64 were compared with several other compositions.^{5,6,7,8,9} D^* denotes the D_s value was obtained from the isotope exchange depth profiling technique and D_{chem} denotes the D_s value was derived from the data of ECR measurements.

Conclusions

- Thermal evolution of crystal structures of pure-phase LCF64, LSF64 and LBF64 were characterized by HT-XRD.
- LSF64 shows a significantly higher electrical conductivity and electron hole mobility than that of LCF64 and LBF64.
- Great similarity in the value of D_{chem} , D_s and D_v was found between LCF64, LSF64 and LBF64.

References

- P. M. Price, E. Rabenberg, D. Thomsen, S. T. Misture and D. P. Butt, *J. Am. Ceram. Soc.*, 2014, 97, 2241-2248.
- A. Fossdal, M. Menon, I. Waernhus, K. Wiik, M.-A. Einarsrud and T. Grande, *J. Am. Ceram. Soc.*, 2010, 87, 1952-1958.
- M. Søgaard, P. Vang Hendriksen and M. Mogensen, *J. Solid State Chem.*, 2007, 180, 1489-1503.
- J. E. ten Elshof, *J. Electrochem. Soc.*, 1997, 144, 1060.
- T. Ishigaki, S. Yamauchi, K. Kishio, J. Mizusaki and K. Fukui, *J. Solid State Chem.*, 1988, 73, 179-187.
- J. Yoo, A. Verma, S. Wang, A.J. Jacobson, *J. Electrochem. Soc.* 152 (3) (2005) A497.
- R. H. E. van Doorn, I. C. Fullarton, R. A. DeSouza, J. A. Kilner, H. J. M. Bouwmeester and A. J. Burggraaf, *Solid State Ionics*, 1997, 96, 1-7.
- R. De Souza and J. A. Kilner, *Solid State Ionics*, 1998, 106, 175-187.
- H. Bae, J. Hong, B. Singh, A. K. Srivastava, J. H. Joo and S.-J. Song, *J. Electrochem. Soc.*, 2019, 166, F180-F189.

^(a) Electrochemistry Research group, Department of Science and Technology, University of Twente, Enschede, The Netherlands

^(b) CAS Key Laboratory of Materials for Energy Conversion, Department of Materials Science and Engineering, University of Science and Technology of China, Hefei, 230026, P. R. China

^(c) Helmholtz-Zentrum Berlin für Materialien und Energie, Hahn-Meitner-Platz 1, 14109 Berlin, Germany