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Improved properties of Pulsed Laser Deposited YBaCuO on NdGaO₃ using CeO₂ template layers

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Abstract

NdGaO₃ (NGO) is, due to its low loss characteristics, a favourite substrate for high frequency applications [1]. However, problems occur when trying to deposit high quality YBaCuO layers on top of it. The optimum deposition temperature during pulsed laser deposition to obtain high T_c , high critical current values and c -axis orientation on NGO is significantly higher than for standard substrates, like SrTiO₃ or YSZ. A disadvantage of this high deposition temperature is the quality of the film surface. The number of outgrowths and precipitates increases rapidly at higher temperatures. To improve the quality of the layers, the effect of a CeO₂ template layer was investigated, leading to remarkable and very promising results. In this paper we focus on the orientation as well as on the surface morphology with and without a CeO₂ template layer on NGO. The orientations were analysed with XRD, the surface morphology with SEM and STM/AFM.

1. Experimental

Thin YBaCuO films (80 nm) were deposited on NdGaO₃ (100) substrates with the pulsed laser deposition (PLD) technique, using an Excimer-laser in XeCl-mode. The parameters, which were kept constant during this study, are the following: laser energy density of 1.2 J/cm², spot size = 1.8 × 4.0 mm², repetition rate = 2 Hz, oxygen pressure of 30 Pa, distance target to substrate 45 mm, PLD-time = 10 min, post PLD pressure = 1 bar, and cooling

down procedure within 30 min. The deposition temperature has been varied in a range of 50 degrees around the optimum deposition temperature for SrTiO₃ substrates, being in our set-up $T_{dep} = 740^\circ\text{C}$ [2]. With the PLD technique it was possible to assemble a CeO₂ template layer prior to the deposition of the YBaCuO film. The PLD parameters of CeO₂ were similar to YBCO, except for the O₂-pressure during deposition (10 Pa), repetition rate of 1 Hz and a PLD-time of 5 min for a 25 nm thin film. In this study, also the deposition temperature of the CeO₂ layer has been varied in the same temperature window as given above. The growth of the CeO₂ layer was followed by the in situ YBaCuO deposition.

The crystallographic orientation of the CeO₂ tem-

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plate layer as well as the YBCO film was studied by XRD-analysis. The electrical properties were analysed using a standard four-probe technique. The surface morphology, which strongly depends on the deposition temperature *and* the use of a template layer, was investigated with scanning electron microscopy. In addition, we looked in more detail to the flatness of the surface, using scanning tunnelling microscopy.

2. Results and discussions

The films deposited on NGO at the optimum growth temperature of SrTiO₃ and YSZ, being in our case 740°C, were very smooth, but had a suppressed transition temperature with a T_{c0} of 80 K, or even less. Fig. 1(a) shows a SEM picture of the as-grown film. The surface morphology looks different compared to *c*-axis oriented films on SrTiO₃ or YSZ. The rectangular grains indicate *a*-axis growth. Moreover, from XRD-analysis of these films we observe mainly *a*-axis peaks. Fig. 1(b) shows the θ -scan of the (102)/(012)-reflections of YBCO, with $\psi = 56.6^\circ$ in the *c*-direction and 33.4° in the *a*-direction. From these data we can conclude that the films grown on NGO at a 'standard' deposition temperature consist of almost 90% of *a*-axis oriented YBCO. This also explains the reduced T_{c0} of these films [3].

To fulfil the requirements of high T_c values ($T_{c0} = 90$ K) as well as *c*-axis orientation of the deposited YBCO, the deposition temperature has to be increased by 35 to 50 degrees. However, the surface of the layers grown at these high temperatures, shows an increase of outgrowths and precipitates. These irregularities did not proceed from the target due to 'splashing'. By choosing the accurate spot-diameter, laser-intensity and target compactness, these particles density becomes less than 1 per 100 μm^2 [4]. Fig. 2(a) shows a SEM picture of an typical example of such a film, grown at $T_{\text{dep}} = 785^\circ\text{C}$. The XRD-analysis of such films shows only *c*-axis orientation and no *a*-axis oriented material is found. In Fig. 2(b) the θ scan of the (102)/(012)-reflections of YBCO is given.

To avoid outgrowths and precipitates at high deposition temperatures the effect of a CeO₂ template was investigated. The properties, like surface mor-

phology and crystal orientation as given by SEM, AFM and XRD-analysis, of the CeO₂ layer on NGO, were very good in a wide deposition temperature range ($790^\circ\text{C} > T_{\text{dep}} > 670^\circ\text{C}$). However, it appeared that the quality of the in situ deposited YBCO film on top of the CeO₂ layer was strongly dependent on the deposition temperature of this CeO₂ template.

Most remarkable was the fact that the morphology of the YBCO films grown at high deposition temperatures ($T_{\text{dep}} = 785^\circ\text{C}$), using a CeO₂ template that was deposited at a lower temperature ($T_{\text{dep}} = 670^\circ\text{C}$), was identical to the films without a template. Again, features, like outgrowth and precipitates, were clearly visible. In Fig. 3(a) a SEM picture is presented. Note the similarity between Fig. 2(a) and Fig. 3(a). The template layer has no effect on the surface morphology and electrical properties ($T_{c0} = 89$ K and $J_c = 2-3 \times 10^6$ A/cm², at 77 K).

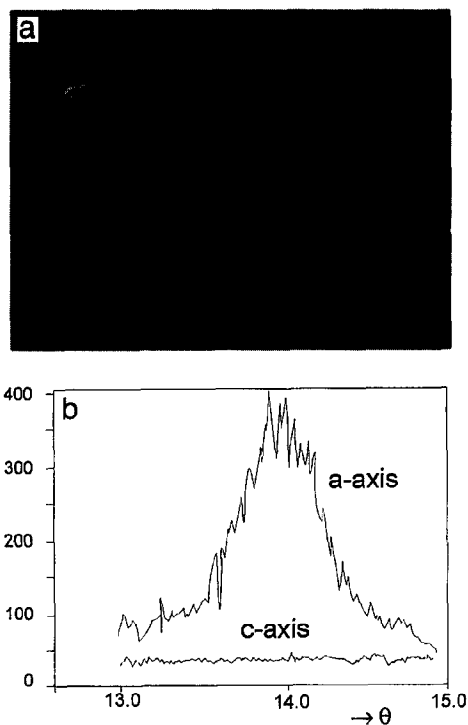


Fig. 1. (a) SEM micrograph (1.0 μm by 0.75 μm) of YBCO on NGO deposited at $T_{\text{dep}} = 740^\circ\text{C}$. (b) The θ scan of the (102)/(012)-reflections of YBCO, with $\psi = 56.6^\circ$ in *c*-direction and 33.4° in *a* direction.

The surface quality of the in situ YBCO layer improved significantly by increasing the deposition temperature of the CeO_2 template layer. There was a large correspondence between the results, using a deposition temperature between $T_{\text{dep}} = 740$ and 790°C . Fig. 3(b) shows a SEM picture of an YBCO film grown on top of a CeO_2 template, both deposited at $T_{\text{dep}} = 785^\circ\text{C}$. The superconductor has good superconducting properties ($T_{\text{c}0} = 90$ K and $J_{\text{c}} = 3 \times 10^6$ A/cm²) and a smooth surface.

The surface roughness has also been studied by AFM analysis. Typically height variations were found to be less than 5 nm, which is acceptable for multi-layer structures.

In addition to these results we lowered the deposition temperature of the YBCO layer to the standard optimum deposition temperature for SrTiO_3 substrates ($T_{\text{dep}} = 740^\circ\text{C}$) keeping the deposition temper-

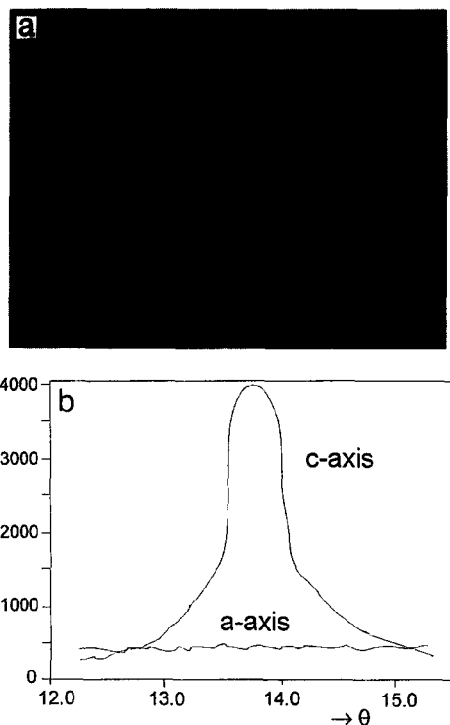


Fig. 2. (a) SEM micrograph ($5.0 \mu\text{m}$ by $3.0 \mu\text{m}$) of YBCO on NGO deposited at $T_{\text{dep}} = 785^\circ\text{C}$. (b) The θ scan of the (102)/(012)-reflections, with $\psi = 56.6^\circ$ in c -direction and 33.4° in a -direction, of YBCO on NGO deposited at $T_{\text{dep}} = 785^\circ\text{C}$.

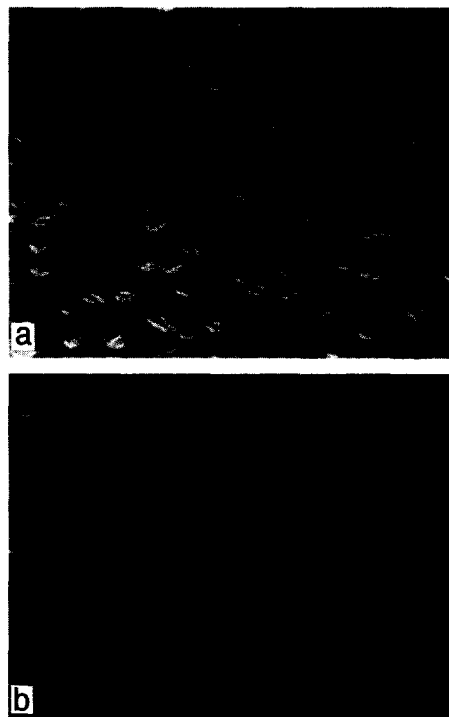


Fig. 3. (a) SEM micrograph ($5.0 \mu\text{m}$ by $3.0 \mu\text{m}$) of YBCO, deposited at $T_{\text{dep}} = 785^\circ\text{C}$, with a CeO_2 template layer, deposited at $T_{\text{dep}} \approx 670^\circ\text{C}$, on NGO (001). (b) SEM micrograph ($5.0 \mu\text{m}$ by $3.0 \mu\text{m}$) of YBCO with a CeO_2 template layer, both deposited at $T_{\text{dep}} = 785^\circ\text{C}$, on NGO (001).

ature of the CeO_2 template at $T_{\text{dep}} = 785^\circ\text{C}$. This reduced temperature had no effect on the superconducting properties or on the surface quality or crystal orientations.

From this we conclude that the growth of YBCO and the occurrence of precipitates is really affected by the surface morphology of the substrate. The effect is clearly visible by covering half of the substrate by a CeO_2 template, which is grown at higher temperatures, followed by the deposition of YBCO at lower temperatures. The covered part has only c -axis oriented YBCO, whereas the YBCO on top of the bare substrate is predominantly a -axis oriented. Furthermore, these results indicate that on NGO selective epitaxy can be obtained by using CeO_2 as a definition mask, especially in a -axis planar devices.

3. Conclusions

The optimum deposition temperature, in the sense of high T_c , high critical current and c -axis orientation, of YBCO on NGO substrates is significantly higher than with the use of standard substrates like SrTiO₃ or YSZ. The roughness of the surface, however, is much worse, due to an increase of outgrowths and precipitates at these higher deposition temperatures. The YBCO films deposited at standard temperatures are mainly a -axis oriented. Using a CeO₂ template, deposited at temperatures $T_{\text{dep}} > 740^\circ\text{C}$, the YBCO surface becomes very smooth and c -axis oriented, even at lower deposition temperatures. These results indicate that on NGO selective

epitaxy can be obtained by using CeO₂ as a definition mask.

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