THE INTERPRETATION OF ELLIPSOMETRIC MEASUREMENTS OF ION BOMBARDMENT OF NOBLE GASES ON SEMICONDUCTOR SURFACES *

A.H.M. HOLTSLAG, U.C. SLAGER and A. VAN SILFHOUT

Department of Applied Physics, Twente University of Technology, P.O. Box 217, 7500 AE Enschede, The Netherlands

Received 1 April 1984

Low energy noble gas ion bombardment and thermal desorption studies were carried out on Si(111) and analysed, in situ, using spectroscopic ellipsometry. The amorphous layer thickness and implanted noble gas fraction were calculated.

1. Introduction

The influence of ion bombardment on the structure of the silicon lattice has been the subject of numerous investigations. The various effects induced by low energy ion bombardment in the outer layers of a solid – lattice damage, ion implantation, surface roughness, sputtering and surface state changes – are reflected in changes of the complex dielectric constant $\tilde{\epsilon}$, which can be measured by means of spectroscopic ellipsometry [1–3].

This paper reports noble gas ion bombardment and desorption studies carried out on Si(111) and analysed using spectroscopic ellipsometry.

2. Experimental

The experiments were carried out in an UHV system (base pressure 10^{-8} Pa), having facilities for sample temperature control, spectroscopic PSRA ellipsometry, AES, mass selective ion bombardment, sputtering and mass

0039-6028/85/\$03.30 © Elsevier Science Publishers B.V. (North-Holland Physics Publishing Division)

^{*} This work is part of the research program of the Stichting voor Fundamenteel Onderzoek der Materie (Foundation for Fundamental Research on Matter) and was made possible by financial support from the Nederlandse Organisatie voor Wetenschappelijk Onderzoek (Netherlands Organisation for the Advancement of Pure Research).

spectrometry. A minicomputer system has been constructed wich automatically controls the ellipsometer and data acquisition [4]. Initially the ellipsometer was calibrated [5] and alligned (angle of incidence 70.25°). Every ellipsometric measurement (Δ , ψ) was preceded by an offset measurement at each analyser setting, since during each desorption the silicon sample is heated up to 800°C, resulting in the optical measurements being disturbed by the partially polarised light emitted from the sample.

Before bombardment the silicon sample was cleaned by argon ion sputtering (45°, 800 eV, 2×10^{-6} A cm⁻²) followed by a 30 min anneal at 800°C and AES. The ion bombardments were carried out using a 2 keV mass selective ion beam, at normal incidence, for two different fluxes (flux 1: 15×10^{-9} A cm⁻²; flux 2: 115×10^{-9} A cm⁻²).

The intensity and profile of the beam was measured using a Faraday cup. After each bombardment a desorption measurement was performed. Using a controllable power supply the sample was directly heated in 50 reproducible increments from 25° to 800°C. For 100 s after each increment the sample temperature and the partial pressure, p(t), of the previously implanted noble gas were measured, followed by determination of Δ and ψ at four wavelengths (and p, T) (A, B, C and D: 695, 595, 495 and 395 nm), during 200 s.

For each desorption the total amount of desorbed particles per unit surface area, $N_{\rm D}(t)$, was calculated from:

$$\frac{N_{\rm D}(t)}{N_{\rm ref}} = \frac{p(t) - p(0)}{P_{\rm ref}} + \frac{1}{\tau} \int_0^t \frac{p(t') - p(0)}{P_{\rm ref}} \, \mathrm{d}t', \quad N_{\rm ref} = \frac{V P_{\rm ref}}{k T_{\rm v} A_{\rm eff}}, \tag{1}$$

where T_v is the vessel temperature, $A_{\rm eff}$ is the effective implanted area and τ is the quotient of the UHV vessel volume V and the effective pumping speed S ($\tau = 62.5$ s for argon).

3. Results and discussion

3.1. Optical effects of noble ion gas bombardments on Si(111)

The optical parameters $\delta\Delta$ and $\delta\psi$, as shown in the figures, are defined as $\delta\Delta = \Delta(N, T, \lambda) - \overline{\Delta}(T, \lambda)$ and $\delta\psi = \psi(N, T, \lambda) - \overline{\psi}(T, \lambda)$ where $\overline{\Delta}(T, \lambda)$ and $\overline{\psi}(T, \lambda)$ refer to the clean substrate and from which the dielectric constant $\tilde{\epsilon}_c(T, \lambda)$ of c-silicon can be calculated. $\Delta(N, T, \lambda)$ and $\psi(N, T, \lambda)$ refer to the bombardment substrate containing a dose N at temperature T. In figs. 1A and 1B the measured $\delta\psi$ and $\delta\Delta$ are given. In fig. 1B the effect of the bombardment can be divided into four different characteristic regions having divisions at Q, R, S and T (1.37 Å⁻²).

From S to T, $\delta\Delta$ changes slowly with increasing N; after dose T, $\delta\Delta$ and $\delta\psi$



nm, $\lambda_D = 395$ nm) and at two ion fluxes (flux 1: 15 nA cm⁻²; flux 2: 115 nA cm⁻²) for 2 keV normal incidence, ion bombardment (at 298 K). Points Q, R, S see text. (C) $\delta\Delta - \delta\psi$ plot for argon and krypton ion bombardment at fluxes 1 and 2; (+) and (×) are the calculated values for $\delta\Delta - \delta\psi$, which Fig. 1. (A, B) Ellipsometric parameter changes $\delta\Delta$ and $\delta\psi$ as a function of argon ion dose at four wavelengths ($\lambda_A = 695$ nm, $\lambda_B = 595$ nm, $\lambda_C = 495$ give a 10 Å increase in the amorphous layer thickness, for three different argon particle fractions (0, 0.05, 0.10).

remain constant. Upon termination of every bombardment $\delta\Delta$ increases and $\delta\psi$ decreases (see fig. 3). From point R the optical change in $\delta\Delta$ is scalable, meaning that $\delta\Delta(N, T_0, \lambda_k) = a_{kl} + b_{kl}\delta\Delta(N, T_0, \lambda_l)$ where λ_k and λ_l stand for two different wavelengths. However, this relationship does not hold for the region 0QR, suggesting that we can separate at least two optical processes caused by bombardment. In fig. 1C a $\delta\Delta - \delta\psi$ plot is given for the 2 keV argon and krypton bombardments.

3.2. Desorption measurements of implanted Si(111) surfaces

The desorption curves of implanted argon are shown in fig. 2, the steps appear in the order: ~ 1030 K, ~ 840 K (main step – which is also the case



Fig. 2. Argon desorption experiments. (A) The desorbed dose as function of temperature. (B) The maximum desorbed dose at 1080 K as function of the bombardment dose. (C, D) $\delta\Delta$ and $\delta\psi$ as function of the desorption temperature at two wavelengths (A, C). The numbers 0–7 correspond to the bombardment doses given at the top of fig. 1A. The insert in (B) gives the argon beam profile.

for krypton!) and ~ 780 K, with increasing ion dose. The step at ~ 1030 K appears after a very small dose, stays almost constant upon increasing dose, and is interpreted as being due to channelling. In fig. 2B the final desorbed doses $\hat{N}_{\rm D}$ are plotted as a function of the dose N; at ~ 0.38 Å⁻² the sample is saturated with argon. The increase of $\hat{N}_{\rm D}$ after point 6 is partially due to the boundary of the beam profile. In figs. 2C and 2D $\delta\Delta$ and $\delta\psi$ are given as a function of the sample temperature. Curves A0 and C0 exhibit monotonic damage recovery. At higher doses the large changes at ~840 K are due to damage recovery and argon desorption, only above 840 K is the change in $\delta\Delta$ always positive. In fig. 3 the bombardment (flux 2) and desorption experiments are shown. For low doses the desorption curve retraces the bombardment curve $(2 \rightarrow 1)$. In general the slopes of the desorption curves are quite similar to those of the bombardment curves. From the curves in figs. 2C, 2D and 3, four different effects can be distinguished: section $3 \rightarrow 4$ damage recovery at room temperature; section $4 \rightarrow 5$, damage recovery due to increasing temperature; section $5 \rightarrow 6$, desorption of argon and a large damage recovery at ~ 840 K; section $6 \rightarrow 7$, complete recovery of the crystal. At 1080 K the crystal regains the original crystalline state after ~ 30 min annealing.



Fig. 3. $\delta\Delta - \delta\psi$ plot of argon bombardment (flux 2) and desorption experiments at three wavelengths A, B, C. Symbols 1–7: divisions of characteristic regions, explained in the text.

3.3. Discussion

The complex dielectric constant, $\tilde{\epsilon}$, of a mixture of components in a solid can be calculated with the EMA equation [6]. We assume a simple model of a homogeneous amorphous layer of variable thickness (0-150 Å) containing a particle fraction, θ_1 , of implanted argon (0, 0.05 and 0.10) above a crystalline substrate. The dielectric constant of argon ($\tilde{\epsilon}_1 = 2.55 - 0i$) was calculated using the Clausius-Mossotti equation [6]. The problem now arises which of the published $\tilde{\epsilon}_a(\lambda)$ to use. Davis and Mott [11] propose that for amorphous silicon $(\alpha nE)^{1/3} = c(E - E_0)$, confirmed by other authors [1,12], where c and E_0 are constants, n is the index of refraction, α is the absorption coefficient and E is the photon energy. From our measured $\delta \Delta$ and $\delta \psi$ values in the E region 1.6-3.1 eV, after a dose N = 0.083 Å⁻², we calculate the amorphous layer thickness, d, under the condition that $(\alpha nE)^{1/3}$ is a linear function of E; from this calculation it follows that $E_0 = 1.05$ eV, c = 5.45 eV^{-2/3} μ m^{-1/3}, d = 100 ± 5 Å, obtaining at the same time $\tilde{\epsilon}_a(\lambda)$ (see fig. 4), where θ_1 is taken as 0.02.

The results of the calculation for $\delta\Delta$ and $\delta\psi$ are shown in fig. 1C. For all wavelengths the measured values of $\delta\Delta$ and $\delta\psi$ for argon, at flux 2, can be explained by a layer of ~ 85 Å which amorphosises until point Q is reached,



Fig. 4. The dielectric constant $\tilde{\epsilon}_a(\lambda)$. A comparison with regard to the expression of Davis and Mott is made for an Ar-bombarded sample (+ + +), a sputtered film (- -) sample 1 from ref. [12], and an undoped LPCVD film (- -) from ref. [1].

followed by a growth of the layer up to ~ 140 Å with $\theta_1 \approx 0.10$. For the similar situation of krypton ion bombardment the values are 70 Å and ~ 105 Å respectively.

At the dose of point Q (0.03 Å⁻²) we compare the calculated amorphous layer thickness with the projected range, χ_p , and range straggling, σ_p [7], of argon ($\chi_p = 34.3$ Å, $\sigma_p = 19.9$ Å) and krypton ($\chi_p = 38.3$ Å, $\sigma_p = 13.4$ Å) for 2 keV bombardment. One can expect an amorphous layer of $\chi_p + 3\sigma_p$, however, by using an layer of $\chi_p + 2.5\sigma_p$ the predicted thickness is 84 Å for argon and 72 Å for krypton, which is in agreement with the observed values.

Furthermore, from the ion collection model of Schulz and Wittmaack [8], using a sputter coefficient s = 1 [9,10], one can calculate a maximum implanted argon dose of 1.7 Å⁻², while the measured value from the desorption experiments (fig. 2B) is 0.38 Å⁻². This means that the total amount of argon is limited by the solubility of argon in the silicon lattice, or s = 4 finally.

By distributing the dose of 0.38 Å⁻² in a homogenous layer of 140 Å one can calculate an argon concentration of $\theta_1 = 0.05$ instead of $\theta_1 = 0.10$ in the final state. However for an accurate calculation one requires the actual distributions $\theta(z)$ and we have also omitted the surface roughness arising during bombardment. For the lower particle flux 1 it seems likely that the amorphous layer is not completely built up, as compared to flux 2.

References

- [1] D.E. Aspnes, A.A. Studna and E. Kinsbron, Phys. Rev. B29 (1984) 768.
- [2] J.R. Adams and N.M. Bashara, Surface Sci. 49 (1975) 441.
- [3] J.W.D. Martens, W.F. van den Bogert and A. van Silfhout, Surface Sci. 105 (1981) 275.
- [4] K.O. van der Werf, R.J.S. Sieverdink, W. Vonk, G. Meijer and L.J. Hanekamp, to be published.
- [5] A.H.M. Holtslag and A. van Silfhout, to be published.
- [6] D.E. Aspnes, Am. J. Phys. 50 (8) (1982) 704.
- [7] J.F. Gibbons, W.S. Johnson and S.W. Mylroie, Projected Range Statistics, Semiconductors and Related Material (Halsted, 1975).
- [8] F. Schulz and K. Wittmaack, Radiation Effects 29 (1975) 31.
- [9] K. Wittmaack and P. Blank, Appl. Phys. Letters 31 (1977) 21.
- [10] R. Behrisch, Topics in Applied Physics 47 (Springer, Berlin, 1981) p. 169.
- [11] E.A. Davis and N.F. Mott, Phil. Mag. 22 (1970) 903.
- [12] R.H. Klazes, M.H.L.M. van den Broek, J. Bezemer and S. Radelaar, Phil. Mag. B45 (1982) 377.