



CHALMERS
UNIVERSITY OF TECHNOLOGY

Magnetic Properties of Multifunctional ⁷LiFePO<inf>4</inf> under Hydrostatic Pressure

Downloaded from: <https://research.chalmers.se>, 2023-07-15 08:15 UTC

Citation for the original published paper (version of record):

Miniotaite, U., Forslund, O., Nocerino, E. et al (2023). Magnetic Properties of Multifunctional ⁷LiFePO<inf>4</inf> under Hydrostatic Pressure. Journal of Physics: Conference Series, 2462(1). <http://dx.doi.org/10.1088/1742-6596/2462/1/012049>

N.B. When citing this work, cite the original published paper.

PAPER • OPEN ACCESS

Magnetic Properties of Multifunctional ${}^7\text{LiFePO}_4$ under Hydrostatic Pressure

To cite this article: Ugne Miniotaite *et al* 2023 *J. Phys.: Conf. Ser.* **2462** 012049

View the [article online](#) for updates and enhancements.

You may also like

- [Prediction of divertor heat flux width for ITER pre-fusion power operation using BOUT++ transport code](#)
X.X. He, X.Q. Xu, Z.Y. Li et al.
- [Pressure-induced superconductivity in CrAs and MnP](#)
Jinguang Cheng and Jianlin Luo
- [Nodeless superconductivity in the cage-type superconductor \$\text{Sc}_2\text{Ru}_6\text{Sn}_{10}\$ with preserved time-reversal symmetry](#)
D Kumar, C N Kuo, F Astuti et al.



The advertisement features a dark blue background on the left with white and orange text, and a photograph of a woman at a podium on the right. The woman is smiling and looking towards the camera, wearing a black top and a lanyard. The podium has a laptop on it. The background of the photo is a bright, modern interior.

ECS The Electrochemical Society
Advancing solid state & electrochemical science & technology

243rd Meeting with SOFC-XVIII

Boston, MA • May 28 – June 2, 2023

Accelerate scientific discovery!

Learn More & Register

Magnetic Properties of Multifunctional ${}^7\text{LiFePO}_4$ under Hydrostatic Pressure

Ugne Miniotaite^{1,*}, Ola Kenji Forslund², Elisabetta Nocerino¹, Frank Elson¹, Rasmus Palm¹, Nami Matsubara¹, Yuqing Ge², Rustem Khasanov³, Genki Kobayashi^{4,5}, Yasmine Sassa², Jonas Weissenrieder¹, Vladimir Pomjakushin⁶, Daniel Andreica⁷, Jun Sugiyama⁸, and Martin Månsson^{1,+}

¹Sustainable Materials Research & Technologies (SMaRT), Department of Applied Physics, KTH Royal Institute of Technology, SE-106 91 Stockholm, Sweden

²Department of Physics, Chalmers University of Technology, Göteborg, SE-412 96, Sweden

³Laboratory for Muon Spin Spectroscopy, Paul Scherrer Institute, CH-5232 Villigen PSI, Switzerland

⁴Solid State Chemistry Laboratory, Cluster for Pioneering Research (CPR), RIKEN, 2-1 Hirosawa, Wako, Saitama 351-0198, Japan

⁵Department of Materials Molecular Science, Institute for Molecular Science, National Institutes of Natural Sciences, 38 Nishigonaka, Myodaiji, Okazaki 444-8585, Japan

⁶Laboratory for Neutron Scattering & Imaging, Paul Scherrer Institute, CH-5232, Villigen PSI, Switzerland

⁷Faculty of Physics, Babes-Bolyai University, 400084 Cluj-Napoca, Romania

⁸Neutron Science and Technology Center, Comprehensive Research Organization for Science and Society (CROSS), Tokai, Ibaraki 319-1106, Japan

E-mail: *ugnem@kth.se, +condmat@kth.se

Abstract. LiFePO_4 (LFPO) is an archetypical and well-known cathode material for rechargeable Li-ion batteries. However, its quasi-one-dimensional (Q1D) structure along with the Fe ions, LFPO also displays interesting low-temperature magnetic properties. Our team has previously utilized the muon spin rotation (μ^+ SR) technique to investigate both magnetic spin order as well as Li-ion diffusion in LFPO. In this initial study we extend our investigation and make use of high-pressure μ^+ SR to investigate effects on the low- T magnetic order. Contrary to theoretical predictions we find that the magnetic ordering temperature as well as the ordered magnetic moment increase at high pressure (compressive strain).

1. Introduction

Multifunctional materials are defined as materials that have several interesting and applicable properties. For example, a material can both be a superconductor as well as an insulator, it is just a question of which temperature range the material is operated in. In the same way, the LiFePO_4 (LFPO) compound is both a cathode material for rechargeable batteries (1; 2; 3; 4; 5; 6; 7) in a room temperature regime, while displaying interesting magnetic properties at low temperatures (8; 4; 7; 9; 10; 11). Originally, the great interest for LFPO is founded in its electrochemical properties, where it is of high interest as a safe and highly stable cathode material in low



current conditions (c.f. other Li-ion battery cathode materials) (12). Further, LFPO is a 'Co-free' cathode material where the replacement of Co with Fe is very favorable due to its much higher natural abundance (Fe:Co = 2500:1). Consequently, both lower price as well as less friction on the geopolitical level makes LFPO attractive for industrial scale energy applications.

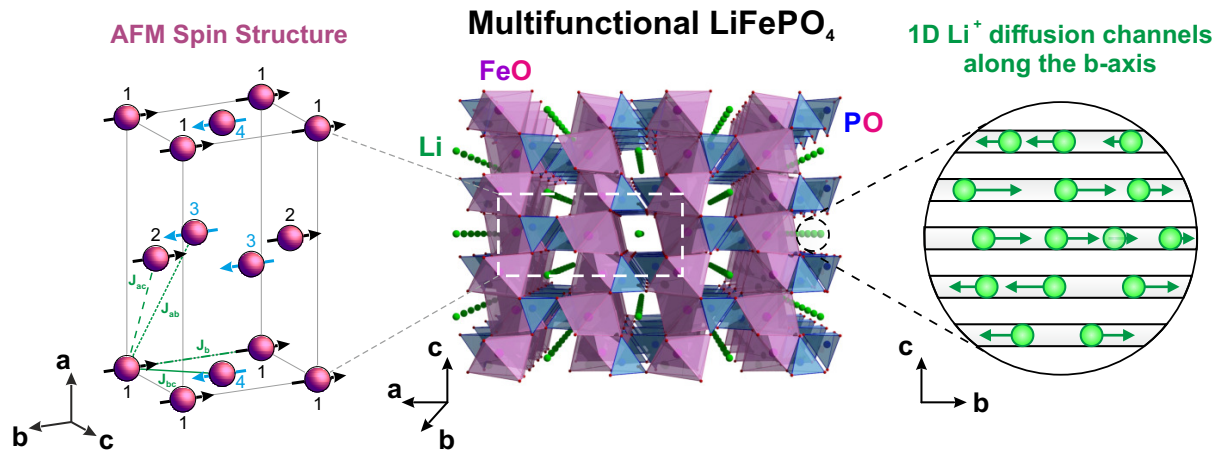


Figure 1. Atomic model of LFPO with zoomed in view on the 1D Li-ion diffusion channels along the b-axis (right) as well as the antiferromagnetic (AFM) spin structure of four different sublattices along with main magnetic interactions J (left, adopted from Refs. (11; 10) that contain further details).

LFPO adopts an olivine type crystal structure with the orthorhombic $Pnma$ space group (see Fig. 1). The Li ions are situated in one-dimensional (1D) channels running along the b-axis (13), which give rise to interesting effects on the lithium intercalation mechanism and dynamics (14). Such anisotropic intercalation/de-intercalation process during charge/discharge cycles is generally believed to occur topotactically, and thereby create a phase separated cathode material with LFPO and FPO phases, respectively. However, such two-phase separation mechanism has also been called into question by both theoretical (15) as well as experimental studies (16). At low temperature, LFPO enters into an antiferromagnetic (AFM) phase below $T_N \approx 50$ K (8). It has been shown that a rather complex canted spin order is present (10) (see also Fig. 1) resulting from competing in-plane and out of plane magnetic couplings (J).

A lot of effort has already been done in order to tune the ion-dynamics (i.e. battery performance) in LFPO. Here, nanostructuring as well as surface coatings (with carbon as well as with other elements and compounds) (17) are among the most common approaches. Such systems have also been studied systematically using both neutron scattering (18) and μ^+ SR (19). When it comes to the tuning of low-temperature magnetic properties, much less detailed studies are to be found in the published literature. Indeed, chemical substitution in LFPO is possible, yielding e.g. NaFePO₄, which display a slightly larger unit cell (20) and also an expected reduction in the magnetic ordering temperature (T_N) by a few Kelvin (Fe-Fe distance is larger). Of course, nanostructuring and surface coatings could also induce interesting effects on the magnetic properties, however, other more direct and 'cleaner' approaches could be employed first. In fact, a magnetic field dependence of the AFM magnons in LFPO has recently been presented (11). Another very common method is to apply hydrostatic pressure, thereby compressing the crystallographic lattice, and investigate how this affects the spin correlations in a material. Combining the penetrating properties of muons, the ultra-high sensitivity of μ^+ SR to detect changes in spin order/dynamics, along with hydrostatic pressure is a very powerful

approach (21; 22; 23; 24). In this study we have used a similar protocol to conduct the first investigation of the local magnetic properties of LFPO under hydrostatic pressure.

2. Experimental Setup

The ${}^7\text{LiFePO}_4$ powder sample was prepared using a solid-state reaction technique with ${}^7\text{Li}$ enriched and reagent grade ${}^7\text{Li}_2\text{CO}_3$, $\text{Fe(II)C}_2\text{O}_4 \cdot 2\text{H}_2\text{O}$, and $(\text{NH}_4)_2\text{HPO}_4$ as the starting materials. More details on the sample synthesis can be found in Ref. (4).

The neutron powder diffraction (NPD) experiments were performed using the HRPT instrument at the Swiss Spallation Neutron Source (SINQ) of Paul Scherrer Institute (PSI), Switzerland. Approximately 2 grams of sample inside a standard $\varnothing = 6$ mm vanadium sample container that was subsequently inserted into a cryofurnace ($1.5\text{K} \leq T \leq 550\text{K}$). High resolution data was acquired using the neutron wavelength $\lambda = 1.49$ Å. The NPD patterns were refined using the Fullprof software suite (25).

The $\mu^+\text{SR}$ experiments were performed at μE1 beamline within the $S\mu\text{S}$ muon source of PSI using the GPD instrument (26; 27). Hydrostatic pressures up to 20 kbar were applied utilizing a double-wall piston cylinder cell (MP35/CuBe alloy). Daphne oil was used as the pressure medium in order to achieve the hydrostatic pressure. Measurements were performed inside a ${}^4\text{He}$ flow cryostat, reaching temperatures down to 2 K. More details about the $\mu^+\text{SR}$ technique (28; 29) and experimental setup can be found in Refs. (26; 27; 30). Data was analyzed using the open analysis software suite *musrfit* (31).

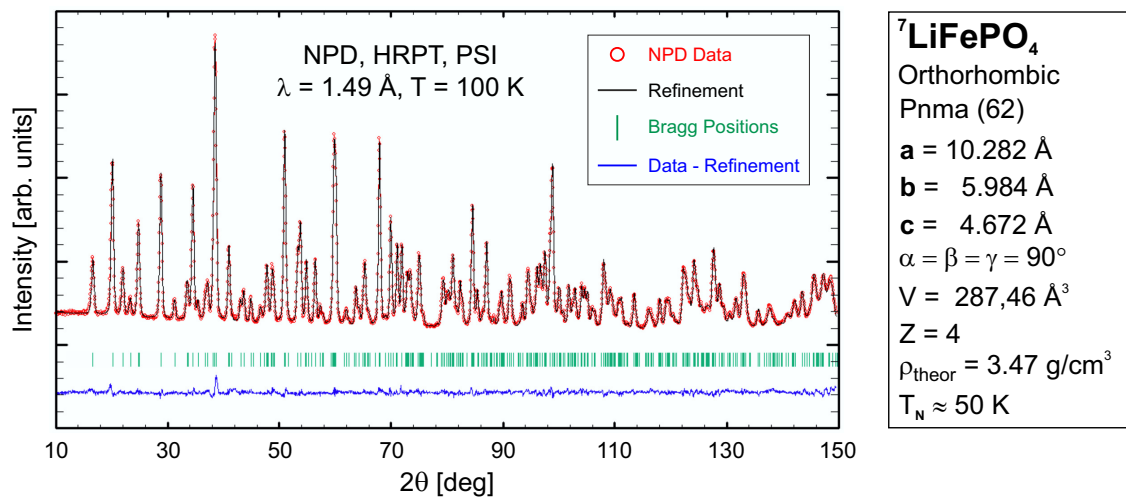


Figure 2. Neutron powder diffraction (NPD) pattern from ${}^7\text{LiFePO}_4$ acquired using a neutron wavelength $\lambda = 1.49$ Å. The right panel show crystallographic data obtained from the Rietveld refinement of our NPD data at $T = 100$ K. V describing the volume of the unit cell, Z the number of molecules per unit cell, ρ_{theor} the theoretical density of the compound, and T_N the Néel temperature.

3. Results & Discussion

Initial neutron powder diffraction (NPD) investigation show that the sample is of high quality. As shown in Fig. 2, the diffraction pattern acquired at $T = 100$ K is very well refined by the expected $Pnma$ space group and there are no signs of any impurity peaks. The right panel of Fig. 2 further display the basic crystallographic data obtained from the Rietveld refinement.

Weak transverse-field (wTF = 50 G) μ^+ SR time spectra were collected as a function of temperature for selected pressures ($p = 0, 10, 20$ kbar). Typical wTF time spectra at selected temperatures are shown in Fig. 3 for ambient and $p = 20$ kbar, respectively. A distinct oscillation is observed as well as an offset at lower temperatures. Therefore, wTF data was fitted using a combination of an exponentially relaxing cosine function and an exponentially relaxing signal:

$$A_0 P_{\text{TF}}(t) = A_{\text{TF}} \cos(2\pi f_{\text{TF}} t + \phi) e^{-\lambda_{\text{TF}} t} + A_{\text{S}} e^{-\lambda_{\text{S}} t}, \quad (1)$$

where A_0 is the initial asymmetry, which depends on the instrumental configuration, and $P_{\text{TF}}(t)$ is the muon spin depolarisation function under TF. A_{TF} , f_{TF} , ϕ and λ_{TF} are the asymmetry, frequency, phase and relaxation rate originating from the applied TF while A_{S} and λ_{S} are the asymmetry and relaxation rate originating from the internal magnetic field.

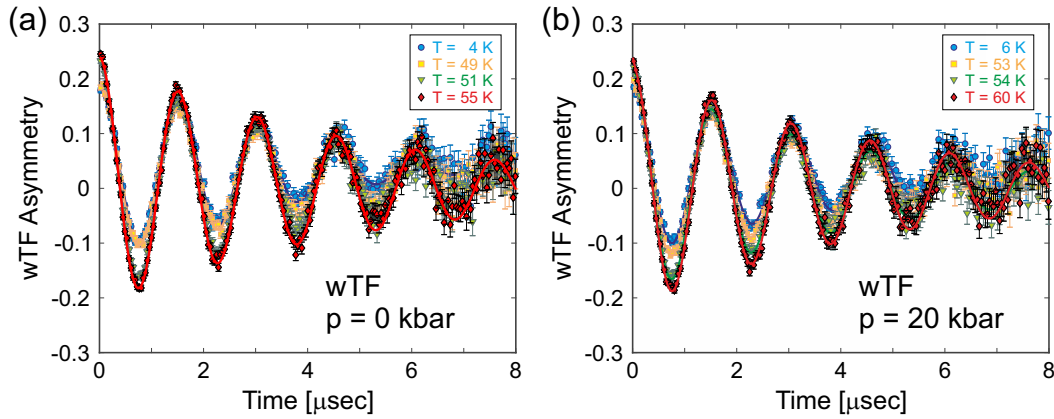


Figure 3. wTF μ^+ SR spectra of LFPO at selected temperatures for (a) ambient pressure and (b) $p = 20$ kbar, respectively. Solid lines are fits to the data using Eq. 1.

Fig. 4 shows the background subtracted and normalized A_{TF} as a function of temperature for the three different pressures. For all pressures, A_{TF} exhibits a temperature independent behaviour at low temperatures. A step-like increase is observed for each pressures as the temperature increases. Since A_{TF} corresponds roughly to the paramagnetic fraction, the step like increase is attributed to the magnetic transition. This critical temperature is obtained from a sigmoid fit and is defined as the point in which the paramagnetic fraction \simeq magnetic fraction function for each pressure [$T_{\text{N}}(p)$]. A clear pressure dependence is observed in T_{N} , which increases with applied hydrostatic pressure. Previous inelastic neutron scattering measurements determined most significant bonds with values: $J_1 = -1.08$ meV, $J_2 = -0.4$ meV and $J_{\perp} = 0.021$ meV (32) where J_1 and J_2 are in plane interactions along [001] and [010] while J_{\perp} is the perpendicular interaction along [100]. Moreover, high pressure XRD at room temperature reported a higher compressibility along the [100] direction (a-axis is the longest) (33). Therefore, the increase in $T_{\text{N}}(p)$ with pressure may be attributed to stronger J_{\perp} interactions due to shorter bond distance, at least as a first approximation. It is noted that J_{\perp} is a ferromagnetic interaction, unlike J_1 and J_2 . Therefore, a high pressure AF-FM transition may be present in LFPO at low temperatures.

Zero field (ZF) time spectra were collected at base temperature $T = 4$ K for ambient and highest (20 kbar) pressures. The spectra exhibit two distinct oscillations [see Fig. 5(a)], consistent with our previous study (4). Therefore, they were fitted using a combination of

exponentially relaxing cosine functions, an exponential and an exponentially relaxing static gaussian Kubo-Toyabe (G^{SGKT}):

$$A_0 P_{\text{ZF}}(t) = \sum_i^2 A_i^{\text{AF}} \cos(f_i^{\text{AF}} t + \phi_i^{\text{AF}}) e^{-\lambda_i^{\text{AF}} t} + A_{\text{tail}} e^{-\lambda_{\text{tail}} t} + A_{\text{PC}} G^{\text{SGKT}} e^{-\lambda_{\text{PC}} t} \quad (2)$$

where A_0 is the initial asymmetry, which depends on the instrumental configuration, and $P_{\text{ZF}}(t)$ is the muon spin depolarisation function under ZF. A_i^{AF} , f_i^{AF} , ϕ_i^{AF} and λ_i^{AF} are the asymmetry, frequency and phase resulting from internal field contributions that are non-parallel to the initial muon spin polarisation. A_{tail} and λ_{tail} are the asymmetry and relaxation rate of the tail component, that originates from internal field components that are parallel to the initial muon spin polarisation. A_{PC} and λ_{PC} are the asymmetry and relaxation rates, respectively, associated with the G^{SGKT} signal, attributed to contributions from the pressure cell (22).

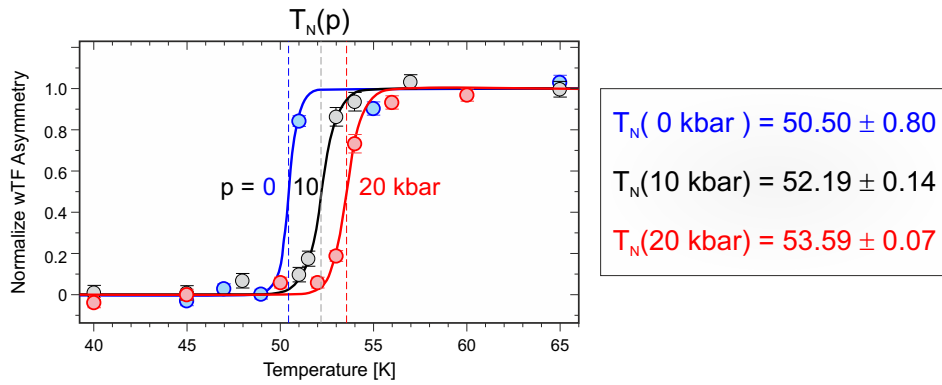


Figure 4. wTF asymmetry at $t = 0$ as a function of temperature and pressure. Solid line represent a fit to a sigmoid type function. Note that the asymmetry has been background subtracted and normalized to unity following the fit to the sigmoid function. The resulting values for $T_N(p)$ are shown in the right panel.

The precession frequencies at ambient pressure, $f_1^{\text{AF}} = 45.37$ MHz and $f_2^{\text{AF}} = 92.99$ MHz are fully consistent with our previous studies (4). When a hydrostatic pressure $p = 20$ kbar is applied (i.e. compression of the lattice), the value of both f_1^{AF} and f_2^{AF} increases (see Fig. 5), which is coherent with the TF results showing an increase in T_N . Such results are summarized in Fig. 6(a). An increase in the values of precession frequency, implies an increase in the ordered moment and supports a scenario in which J_{\perp} increases in strength with pressure application. This is also very reasonable when considering the NaFePO₄ sister compound mentioned above, which displays a larger Fe-Fe distance, and as a result, slightly lower T_N (20). Several high pressure studies of the atomic structure in LFPO have been published and there is some controversy in the results. An early high pressure XRD investigation reported a structural transition to a space group $Cmcm$ at a critical pressure below 65 kbar (34). However, other studies (35) indicate a much higher critical pressure. Further, theoretical calculations based on the $Cmcm$ structure predict a decrease in the ordered Fe magnetic moment (36) with applied pressure, as shown in Fig. 6(b). Our experimental result on the contrary show stronger magnetic correlations with applied pressure. Therefore, we may at least conclude that the supposed structural transition is not present up to 20 kbar. Our TF and ZF result also indicate that the spin structure at ambient pressure is most likely preserved up to at least $p = 20$ kbar. We can of course not fully exclude a combination of atomic lattice compression together with a subtle change in the spin

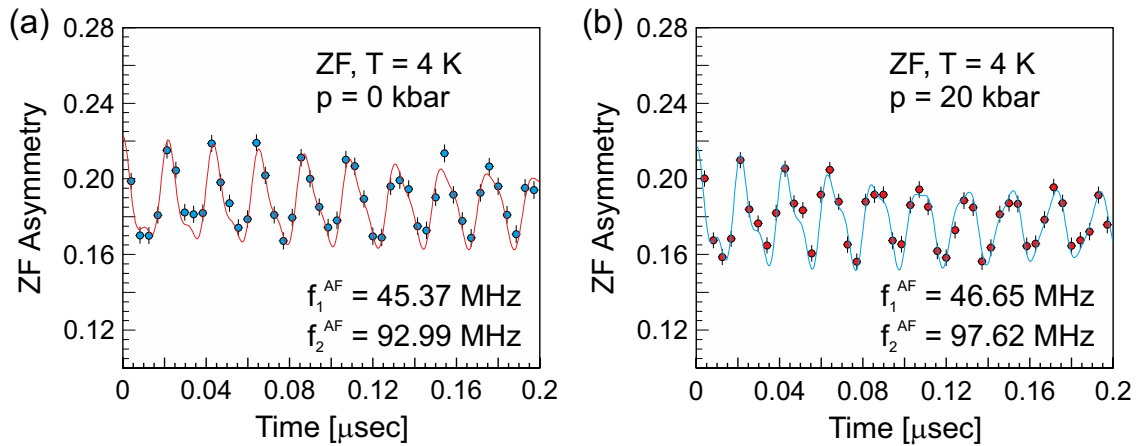


Figure 5. Zero field (ZF) μ^+ SR spectra acquired at $T = 4$ K and (a) ambient pressure, as well as (b) $p = 20$ kbar. Solid line is the fit to Eq. 2, resulting in the two frequencies f_1^{AF} and f_2^{AF} .

canting occurs. However, to discern such in-depth information, it would be necessary to conduct additional and detailed high-pressure neutron diffraction studies.

Finally, even if we only have acquired data at three different pressures, we could indeed fit both $T_N(p)$ and $f^{\text{AF}}(p)$ to a straight line. For $T_N(p)$ we can then extrapolate and acquire a critical pressure p_c for which $T_N = 0$ K, i.e. magnetic order in LFPO is fully suppressed. Such procedure yield $p_c \approx -325$ kbar, i.e. a very large tensile strain (larger lattice). This is clearly a very unreasonable phase space for practically conducting actual experiments, however, it gives an indication and guidance to how LFPO could be chemically tuned in order to suppress magnetic spin order.

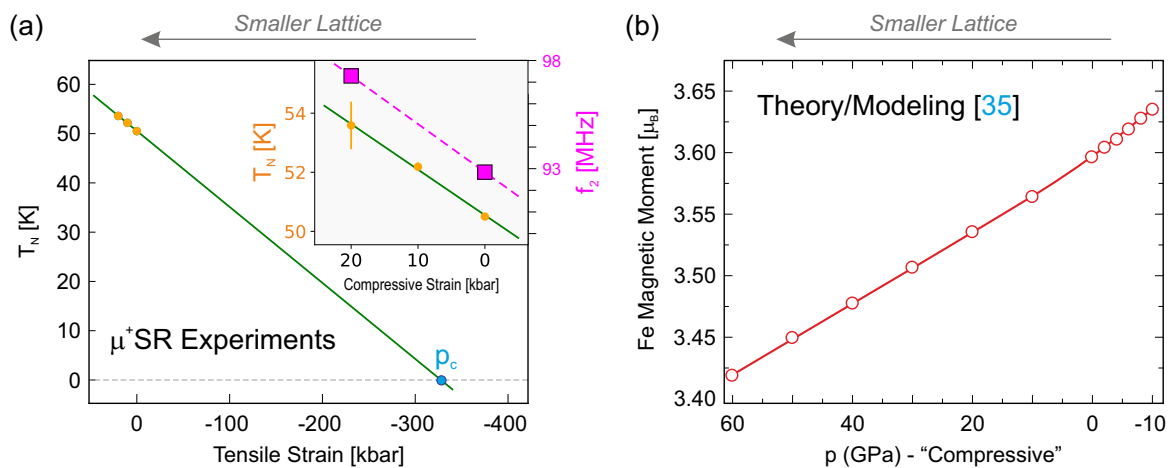


Figure 6. (a) Pressure dependence of T_N as well as the ZF μ^+ SR frequency f_2 (inset). A critical pressure can be estimated as $p_c \approx -325$ kbar. (b) Theoretical prediction for the Fe ordered magnetic moment in LiFePO_4 for the $Cmcm$ structure (adopted from Ref. (36)). Note that the x-axes have been selected to intuitively show a decrease in the lattice volume rather than as an increase in applied pressure.

4. Conclusion

We have performed the first μ^+ SR investigations of the magnetic properties in LiFePO_4 as a function of hydrostatic pressure and temperature. Measurements in transverse field (TF) and zero field (ZF) confirms the formation of long range antiferromagnetic order within the entire pressure range ($p = 0 - 20$ kbar). A clear increase in both magnetic ordering temperature (T_N) and ordered Fe magnetic moment is revealed, which we interpret as related to an increase of the out plane interaction strength. Moreover, our results also confirm the absence of any $Pnma$ to $Cmcm$ structural transition in LiFePO_4 , at least up to $p = 20$ kbar. Finally, we can estimate a critical pressure $p_c \approx -325$ kbar (i.e. tensile strain) for which magnetic spin order in LiFePO_4 is fully suppressed.

5. Acknowledgments

The authors wish to thank the staff of $S\mu S$ and SINQ (PSI) for their great support during the experiments. This research is funded by the School of Science and Engineering (SCI) of the KTH Royal Institute of Technology, the Ragnar Holm Foundation, the Swedish Foundation for Strategic Research (SSF) within the Swedish national graduate school in neutron scattering (SwedNess), as well as the Swedish Research Council VR (Dnr. 2021-06157 and Dnr. 2017-05078), and the Carl Tryggers Foundation for Scientific Research (CTS-18:272). J.S. was supported by the Japan Society for the Promotion Science (JSPS) KAKENHI Grant No.No. JP18H01863 and JP20K21149. D.A. acknowledges financial support from the Romanian UEFISCDI project PN-III-P4-ID-PCCF-2016-0112, Contract Nr. 6/2018. Y.S. and O.K.F. are also funded by the Chalmers Area of Advance - Materials Science.

References

- [1] Padhi A K, Nanjundaswamy K S and Goodenough J B 1997 *Journal of The Electrochemical Society* **144** 1188–1194
- [2] Yamada A, Chung S C and Hinokuma K 2001 *Journal of The Electrochemical Society* **148** A224
- [3] Kobayashi G, Nishimura S, Park M S, Kanno R, Yashima M, Ida T and Yamada A 2009 *Advanced Functional Materials* **19** 395–403
- [4] Sugiyama J, Nozaki H, Harada M, Kamazawa K, Ofer O, Månsson M, Brewer J H, Ansaldo E J, Chow K H, Ikedo Y, Miyake Y, Ohishi K, Watanabe I, Kobayashi G and Kanno R 2011 *Physical Review B* **84**
- [5] Zhang W J 2011 *Journal of Power Sources* **196** 2962–2970
- [6] Sugiyama J, Nozaki H, Harada M, Kamazawa K, Ikedo Y, Miyake Y, Ofer O, Månsson M, Ansaldo E J, Chow K H, Kobayashi G and Kanno R 2012 *Physical Review B* **85**
- [7] Sugiyama J, Nozaki H, Kamazawa K, Ofer O, Månsson M, Ansaldo E J, Brewer J H, Chow K H, Watanabe I, Ikedo Y, Miyake Y, Kobayashi G and Kanno R 2012 *Physics Procedia* **30** 190–193
- [8] Santoro R P and Newnham R E 1967 *Acta Crystallographica* **22** 344–347
- [9] Ofer O, Sugiyama J, Brewer J H, Månsson M, Prša K, Ansaldo E J, Kobayashi G and Kanno R 2012 *Physics Procedia* **30** 160–163
- [10] Toft-Petersen R, Reehuis M, Jensen T B S, Andersen N H, Li J, Le M D, Laver M, Niedermayer C, Klemke B, Lefmann K and Vaknin D 2015 *Physical Review B* **92**
- [11] Werner J, Neef C, Koo C, Ponomaryov A, Zvyagin S and Klingeler R 2021 *Physical Review B* **103**
- [12] Mohamed N and Allam N K 2020 *RSC Advances* **10** 21662–21685

- [13] Nishimura S, Kobayashi G, Ohoyama K, Kanno R, Yashima M and Yamada A 2008 *Nature Materials* **7** 707–711
- [14] Malik R, Abdellahi A and Ceder G 2013 *Journal of The Electrochemical Society* **160** A3179–A3197
- [15] Bai P, Cogswell D A and Bazant M Z 2011 *Nano Letters* **11** 4890–4896
- [16] Gu L, Zhu C, Li H, Yu Y, Li C, Tsukimoto S, Maier J and Ikuhara Y 2011 *Journal of the American Chemical Society* **133** 4661–4663
- [17] Dominko R, Bele M, Gaberscek M, Remskar M, Hanzel D, Pejovnik S and Jamnik J 2005 *Journal of The Electrochemical Society* **152** A607
- [18] Benedek P, Yazdani N, Chen H, Wenzler N, Juranyi F, Månsson M, Islam M S and Wood V C 2019 *Sustainable Energy & Fuels* **3** 508–513
- [19] Benedek P, Forslund O K, Nocerino E, Yazdani N, Matsubara N, Sassa Y, Juranyi F, Medarde M, Telling M, Månsson M and Wood V 2020 *ACS Applied Materials & Interfaces* **12** 16243–16249
- [20] Avdeev M, Mohamed Z, Ling C D, Lu J, Tamaru M, Yamada A and Barpanda P 2013 *Inorganic Chemistry* **52** 8685–8693
- [21] Thede M, Mannig A, Månsson M, Huvonen D, Khasanov R, Morenzoni E and Zheludev A 2014 *Physical Review Letters* **112**
- [22] Forslund O K, Andreica D, Sassa Y, Nozaki H, Umegaki I, Nocerino E, Jonsson V, Tjernberg O, Guguchia Z, Shermadini Z, Khasanov R, Isobe M, Takagi H, Ueda Y, Sugiyama J and Månsson M 2019 *Scientific Reports* **9**
- [23] Sugiyama J, Andreica D, Forslund O K, Nocerino E, Matsubara N, Sassa Y, Guguchia Z, Khasanov R, Pratt F L, Nakamura H and Månsson M 2020 *Physical Review B* **101**
- [24] Sugiyama J, Higemoto W, Andreica D, Forslund O K, Nocerino E, Månsson M, Sassa Y, Gupta R, Khasanov R, Ohta H and Nakamura H 2021 *Physical Review B* **103**
- [25] Rodríguez-Carvajal J 1993 *Physica B: Condensed Matter* **192** 55–69
- [26] Khasanov R, Guguchia Z, Maisuradze A, Andreica D, Elender M, Raselli A, Shermadini Z, Goko T, Knecht F, Morenzoni E and Amato A 2016 *High Pressure Research* **36** 140–166
- [27] Khasanov R arXiv:220811320 [physicsins-det], 2022 Perspective on the muon-spin rotation/relaxation under pressure
- [28] Yaouanc A and de Reotier P D 2010 *Muon spin rotation, relaxation, and resonance* International Series of Monographs on Physics (London, England: Oxford University Press)
- [29] Blundell S J, De Renzi R, Lancaster T and Pratt F L (eds) 2021 *Muon Spectroscopy* (London, England: Oxford University Press)
- [30] Andreica D 2001 Magnetic phase diagram in some kondo-lattice compounds: Microscopic and macroscopic studies ph.D. thesis, IPP/ETH-Zurich, 2001
- [31] A Suter and B M Wojek 2012 *Phys. Proc.* **30** 69
- [32] Li J, Garlea V O, Zarestky J L and Vaknin D 2006 *Phys. Rev. B* **73**(2) 024410
- [33] Dong H, Guo H, He Y, Gao J, Han W, Lu X, Yan S, Yang K, Li H, Chen D and Li H 2017 *Solid State Ionics* **301** 133–137 ISSN 0167-2738
- [34] García-Moreno O, Alvarez-Vega M, García-Jaca J, Gallardo-Amores J M, Sanjuán M L and Amador U 2001 *Chemistry of Materials* **13** 1570–1576
- [35] Valdez M N, Efthimiopoulos I, Taran M, Müller J, Bykova E, McCammon C, Koch-Müller M and Wilke M 2018 *Physical Review B* **97** 184405
- [36] Lin H and Zeng Z 2011 *IEEE Transactions on Magnetics* **47** 3817–3820