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Review on Engineering of Bone Scaffolds Using Conventional and Additive Manufacturing Technologies" for 3D Printing and Additive Manufacturing

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Document Version Peer reviewed version

Citation for published version (Harvard): Mohammed, AH, Jiménez, A, Bidare, P, Elshaer, A, Memić, A, Hassanin, H & Essa, K 2023, 'Review on Engineering of Bone Scaffolds Using Conventional and Additive Manufacturing Technologies" for 3D Printing and Additive Manufacturing', 3D Printing and Additive Manufacturing.

Link to publication on Research at Birmingham portal

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Review on Engineering of Bone Scaffolds Using Conventional and Additive Manufacturing Technologies

Journal:	3D Printing and Additive Manufacturing
Manuscript ID	3DP-2022-0360.R2
Manuscript Type:	Review Papers
Date Submitted by the Author:	25-Mar-2023
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Keywords:	3D printers, additive manufacturing, hybrid additive manufacturing, medical applications
Manuscript Keywords (Search Terms):	3D printing, additive manufacturing, conventional manufacturing, bone scaffolds
Abstract:	Bone is a complex connective tissue that serves as mechanical and structural support for the human body. Bones' fractures are common, and the healing process is physiologically complex and involves both mechanical and biological aspects. Tissue engineering of bone scaffolds holds great promise for the future treatment of bone injuries. However, conventional technologies to prepare bone scaffolds can not provide the required properties of human bones. Over the past decade, three- dimensional printing or additive manufacturing technologies have enabled the control over the creation of bone scaffolds with personalized geometries, appropriate materials and tailored pores. This paper aims to review the recent advances in the fabrication of bone scaffolds for bone repair and regeneration. A detailed review of bone fracture repair and an in-depth discussion on conventional manufacturing and three- dimensional printing techniques are introduced with an emphasis on novel studies concepts, potentials and limitations.



2	Manufacturing Technologies
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1. Introduction

Bone injuries have recently increased due to ageing, traumatic injuries, and congenital diseases. making them a global health issue. It's estimated that the number of people aged more than 65 years will increase from 323 million to 1.55 billion by 2050 worldwide. Age intensifies the risk of osteoporosis and consequently has dangerous effects on people's healthy life, disability, countries' healthcare systems, and loss of productivity. Globally, over 200 million people have osteoporosis, with an increased number of patients receiving hospital treatment every year due to fragility fractures and bone loss, accelerating the demand for bone tissue surgeries. Efficient and cost-effective strategies to treat bone injuries will help to improve people's quality of life and relief the economic burden on governments (1, 2).

A bone defect is generally defined as the lack of bone tissue in an area of the body, which results in pseudarthrosis. Usually, the human body is capable of self-repair, yet when a segmental bone fracture exceeds a size of 10 mm, the body fails to self-repair (3). Therefore, external interventions are essential to assist in the self-repairing process by creating bone scaffolds. These scaffolds act as bridges over bone defect sites and facilitate repair (4). The design of the bone substitutes must be controlled to avoid excessive bone tissue removal at defect sites and to allow cell activity and proliferation (5, 6). The latter is facilitated by designing a scaffold with a porous and linked pore structure. Thus, manufactured bone scaffolds are a promising solution for treating bone fractures, but this comes with some challenges.

Regarding bone scaffolds manufacturing techniques, several methods have been investigated to create porous scaffolds for bone repair, such as salt leaching, gas foaming, self-assembly, phase separation, electrospinning, and freezing drying methods. Although these approaches are capable of fabricating porous structures, they have certain drawbacks, such as restricted pore structure control and a limited ability to customise for particular defect sites (7). Additionally, many of these techniques leave behind organic residues of the pore-forming agent, impairing the scaffolds' biological characteristics and lowering the quality of bone healing. Thus, developing fabrications technique for scaffolds that are not restricted to obtaining the desired external shape but also precisely control the pore structure is critical for their future orthopaedic application.

Given this context, additive manufacturing (AM) technologies are becoming a good alternative
 for manufacturing scaffolds as they can create porous scaffolds with customised external design

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and a porous inner structure (8). The use of 3D printing technology for the generation of bone scaffolds has been gaining more attention from researchers and the biomedical industry in recent years. The near future of bone regeneration and healing is closely linked to developments in tissue engineering. Polytherapy, which combines scaffolds, stem cells, and healing promoters with new advances in tissue-engineered constructs in three-dimensional printing, may be capable of overcoming current challenges in treating bone injuries. In this review paper, we will focus on scaffolds as an established treatment for bone fracture using 3D printing technologies and compare them with conventional manufacturing techniques (Figure 1).

Bone Fracture Repair

Bone tissue can undergo biological remodelling as a function of a dynamic process that involves osteoclasts absorbing mature bone tissue and osteoblasts forming new bone tissue (9, 10). Bone is a complex connective tissue made up of osteoblasts, osteocytes, bone lining cells, and osteoclasts. The outer layers of bones are mineralised, giving them significant strength and rigidity to support the body structure and allow skeletal movement. Bones composition includes the inorganic phase (60% - 70% of the tissue), (22% - 35%) organic matrix and liquid (5% -8%), where collagen represents the majority of the organic matrix and only 10% non-collagenous proteins (11). Bones strength and stiffness are mainly provided by hydroxyapatite crystal $(Ca_{10}(PO_4)_6(OH)_2)$ with carbonate ions (12) which are found within and between collagen fibres in the form of needles, plates and rods with an average diameter of 20-80 nm and 2–5 nm in thickness [2]. Bone can also modify its structure according to body requirements, such as in repair, modelling, remodelling, and growth (13, 14).

Bone tissues can be classified into cortical and trabecular (Figure 2). Both have the same matrix composition; however, they vary in structure and function as well as in relative distribution between bones. Cortical bone (dense or compact) is composed of layers surrounded by lamellar bone and vascular channels. This arrangement is known as the Haversian or osteon (15). An osteon's central channel contains cells, vessels, nerves, and osteon-connecting Volkmann's channels (15). On the other hand, trabecular bone (spongy or cancellous) is located in the epiphysis and metaphysis of long bones and inside small or flat bones. Trabecular bone has a wide network of individual trabeculae, small and interconnected plates and rods guided by external loading (15). Typically, cortical and trabecular bones have Young's moduli of approximately 17 and 1 GPa, respectively (15). Cortical bone is a dense structure representing about 80% of the total skeletal tissue. Yet, cortical bones have some microscopic pores (about

10% of the total cortical bone volume) to allow vascular and neural supply and enable the
 2 delivery of nutrients (16). The porosity of cortical bone is critical as an increase in intracortical
 3 porosity can reduce the bone strength and consequently increase the chances of fracture (17). It
 4 is evident that cortical bone becomes more fragile at very high strain levels reflecting high 5 impact trauma (Figure 2). As the strain rate increases, cortical bone shows a ductile to brittle
 6 transition (18), and like any material, the cortical bone could be prone to fatigue failure.

On the other hand, trabecular bone has a lower mass and high porosity when compared to cortical bone. Pores represent 50%-90% of total trabecular bone volume (19). This makes it considered an open cell porous foam with a reduced compressive strength to about one-tenth of the cortical bone (20). It does, however, provide a large surface area that is necessary for red bone marrow, blood vessels and bone-connected tissues and facilitates hematopoiesis and homeostasis of minerals. The trabecular bone's physical and mechanical properties vary widely depending largely on the anatomical location, age and orientation of the cell structures (21, 22). Depending on the type and orientation of these basic cell structures, the mechanical properties can differ by a factor of 10. A comparison between the compressive properties of trabecular and cortical bones is shown in (Figure 3) (21). As shown, the cortical bone acts as a typical brittle material at which the stress steeply increases at a low strain in the elastic region, and the fracture occurs without a noticeable change in the strain.

On the other hand, trabecular undergoes a ductile behaviour under compression loading with a
 substantial increase in the plastic deformation before fracture. Individual trabeculae bone
 damage and repair is a physiological process that occurs throughout life and increases with age
 (23).

Bone fracture healing process can be enhanced using several techniques, such as grafts, which replace defected bone with another bone from the patient's own body (autografts) or from a donor, or by using healing growth materials in fabricating bone implants or scaffolds (24). Autografts are currently the bone regeneration golden standard (24). However, Autografts techniques have several disadvantages, such as surgical complications and the limited supply of natural tissue. Tissue-engineered bone scaffolds are a suitable alternative to autografts as they improve fracture healing and enhance the incorporation of grafts (25, 26). Requirments of tissue-engineered scaffolds are to have properties close to those of autografts irrespective of their limitations (27).

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Bone scaffolds need to have high porosity with sufficient sizes of pores across all sites of the
 scaffold in order to create an ideal environment for the formation of new tissue matrix and bone.
 Moreover, growth factors like the basic growth factor of fibroblasts (28, 29) can influence cell
 functions, proliferation, or differentiation; Promotive healing agents, for example, Human
 platelet-rich plasma (hPRP) (30-32); and also Tarantula cubensis extract (33) could be
 incorporated into the scaffolds to improve the damaged connective tissue's ability to repair.

The scaffold's vascularity is important as if not present, ischemia will occur in the scaffold, and hence the cells would die. Therefore, it would be useful to incorporate growth factors such as FGF, PDGF, and VEGF to promote angiogenesis in scaffolds and grafts (34, 35). A combination of stem cells and scaffolds with growth factors can be one possible approach providing all the required characteristics to enhance bone repair and regeneration. Currently, none of the grafts provided all the desirable requirements such as biocompatibility, size limitation, cost, osteogenic, osteoconductive, osteoinductive, and angiogenic properties (Table 1). Tissue engineering seeks to provide all or most of these characteristics (36, 37). Also, tissue engineering can cause bone defects to be repaired and reconstructed (27). Incorporating the basics of orthopaedic surgery with knowledge from various sciences such as biology, engineering, chemistry, materials science and physics could overcome current treatments' shortcomings (25). Advances in biomaterials and tissue engineering can provide more appropriate tools to support the differentiation and proliferation of bone cells and improve bone fracture healing. Although there are a large number of studies in the literature on the effects of different agents on bone healing, it is certain to investigate the best manufacturing techniques for fabricating the desired scaffolds (38-40).

3. Conventional manufacturing techniques

Conventional techniques used to prepare bone scaffolds are based on subtractive procedures to get the desired shape by removing sections of the material from an original block. The inability to manage complex shapes and geometries, as well as to incorporate interior architecture, cavities or curved channels, is a major disadvantage of these techniques (41). This is of special importance in the biomedical industry, where complex and organic shapes are usually needed for the implants to fit well. Additionally, cell viability and function can be affected by the use of organic solvents, even if only residues remain (42). In order to obtain those geometries, until now, several conventional manufacturing methods, such as salt leaching, gas formation, phase separation, freeze-drying, electrospinning, and self-assembly, have been employed in the

fabrication of porous bone scaffolds despite their limitations (Figure 4). The principles of each
 procedure are covered in the following sections. The prevalence of research on each technique
 is summarised in (Figure 5). It is evident that more phase separation technique has attracted the
 attention of researchers over the last 10 years.

3.1. Salt leaching

8 This process was widely used in the manufacture of tissue-based scaffolds. In this technique, 9 salt crystals or porogen (e.g. sodium chloride) are put in a mould, and the remaining gaps are 10 filled with a polymer. The polymer is then solidified, and salt crystals are dissolved in a suitable 11 solvent like alcohol or water by dissolution (43-45). After all the salt leaches out, a solidified 12 polymer with porosity is created, as illustrated in (Figure 4a) (46).

 β -chitin and collagen have been successfully used to prepare using salt leaching technique. The prepared membranes achieved a porosity of 77.81% and an average pore size of 260-330 μm. β -chitin membranes were prepared with NaCl salt-leaching, and then collagen solution crossed membranes by lyophilisation at -75 ° C (Figure 6a-b) shows the scaffold's cross-sectional and surface morphologies. In vitro cell culture demonstrated that the human fibroblasts attached to the collagen sites after 3 days, and proliferation took place within 14 days of cultivation (47).

The salt leaching technique enables the customisation of the pore size by adjusting the porogen size employed. It is also possible to control the porosity and pore size of the scaffold by manipulating the volume and size of the salt particles used, respectively (43, 48). Despite the mentioned benefits of salt leaching for scaffold fabrication, this process has some limitations. For instance, it is not possible to control the pore distribution or the shape of scaffolds created (43). Additionally, this technique requires scaffolding to be manufactured only in the form of tubes and flat sheets, which means that it is ideal for the manufacture of membrane scaffolds. Besides, the use of organic solvents can negatively impact the viability of cells and their biological functions (49). Although the residues of any cytotoxic solvents could be detected (50), they pose limitations for general applications of salt leaching scaffolds.

3.2. Gas foaming

Gas foaming is a manufacturing approach in which a polymeric material is filled with a foaming
agent such as carbon dioxide, water or nitrogen at high pressure (51-53). Solid polymer disks

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like polyglycolide (PGA) and poly-L-lactide (PLA) are created at high temperatures before spreading high-pressure carbon dioxide gas through the disks for a few days before decreasing the pressure down to the ambient level (Figure 4b) (54).

Gas foaming was used by Kim et al. (55) to fabricate porous biphasic calcium phosphate (BCP) scaffold by using gas-foamed polyurethane as a model achieving a porosity of 75% to 85% and pore size around 300-800 µm. The BCP scaffold was biocompatible and successfully differentiated and regenerated bone according to both the in vivo and in vitro experiments conducted in this study (55). In another study, a biodegradable poly (L-lactic acid) (PLLA) scaffold with high open porosity was fabricated using a gas-foaming technique along with salt leaching (56). The scaffold had a porosity of around 90% with pore sizes around 300-400 µm, which is ideal for high-density seeding of cells (Figure 7). Upon seeding rat hepatocytes into the scaffold, 40% viability and around 95% seeding efficiency were achieved within 24 hours (56).

The key advantage of using the gas foaming technique is that it does not require the use of chemical solvents, thus reducing the overall manufacturing time. Nonetheless, it is challenging to control the internal structure of the scaffolds in terms of pore size and high connectivity using this technique (57, 58). In addition, high temperatures during the creation of disks often prevent the use of bioactive molecules in scaffolds (59). Although this technique has the ability to fabricate scaffolds with 93 % porosity and pore sizes up to 100 µm (59), it has been noted that the scaffold interconnects only 10–30 % of the pores which may limit the proliferation of encapsulated scaffold cells (54).

3.3. Phase separation

In phase separation, a polymer is generally dissolved in an appropriate solution and then deposited in a mould that is gradually cooled till the solution freezes. The solvent is then removed by freezing, leaving behind a porous matrix, as illustrated in (Figure 4c). Various types of phase separation methods are available, including thermal-induced, solid-liquid and liquid-liquid phase separation (60, 61).

The study by Kim et al. employed thermally induced phase separation (TIPS) for the manufacture of polyethene-glycol (PEG) poly(l-lactic) acid (PLLA) scaffold to support MC3T3-E1 osteoblast-like cells (62). It was shown in this study that ageing times and temperature have a significant effect on the pores morphology of the fabricated scaffold,

quenching temperatures of 25°C, 30°C, and 35°C (Figure 8a-c). The TIPS technique allowed a
 simple control of the scaffold pore size between 100 – 300 μm. Authors noted that MC3T3-E1
 cells could proliferate successfully within 4 weeks after being seeded on the microporous
 scaffold of PEG-PLLA (62).

The main merit of using the phase separation technique is that it does not require extra leaching.
However, the use of organic solutions like ethanol or methanol during the scaffold
manufacturing process can prevent the integration of bioactive compounds or cells.
Furthermore, the small pore diameters generated are another constraint for phase-separation
scaffolds. (43, 59).

3.4. Freeze-drying

Freeze-drying technique is based on a frozen liquid that sublime directly into the gas phase leaving behind a porous hydrogel (Figure 4d) (63). The manufacturing approach was first explored by Whang et al. to produce PLGA scaffolds (64). The literature demonstrates that fabricated scaffolds' porosity and pore diameter are highly influenced by variables like the water-to-polymer mixture ratio and the viscosity of the emulsion (59). Also, altering the cooling temperature can control the scaffold's internal pore structure (65).

In a study conducted by Park et al. (66), freeze-drying was used to fabricate collagen and acid (HA) membranes and then crosslinked hyaluronic using 1-ethvl-3-(3-dimethylaminopropyl) carbodiimide (EDC). Porosity and pores' size were measured to assess the effect of freezing temperature and crosslinking on the internal structure of the scaffolds, freezes dried temperature used were at $-20 \circ C$, $-70 \circ C$, and $-196 \circ C$ (Figure 9a-c), (Table 2). The higher the freezing temperatures, the larger the pore size and the porosity percentage. Also, the use of EDC has significantly increased both the porosity and pore size. The prepared membranes were safe and did not exhibit significant toxicity to L929 fibroblast cells upon testing (66).

 The main benefit of the freeze-drying technique is that it eliminates many rinsing processes by
immediately removing scattered water and polymer solutions (64). In addition, polymer liquids
can be utilised directly instead of any monomer crosslinking. Nevertheless, in order to increase

scaffold homogeneity, the freeze-drying method must be managed to minimise heterogeneous
 freezing (65). Moreover, this approach is associated with high energy consumption, long
 timescales, small irregular pores, and cytotoxic solvents (67, 68).

3.5. Electrospinning

Electrospinning is an innovative electrochemical technology that utilises an electrical charge to create solid, nano-sized fibres from a liquid solution (69). As illustrated in (Figure 4e), the electrospinning process begins with a syringe filled with a solution containing a precursor for the nanofiber material and a connecting polymer being loaded onto a regulated syringe pump. A metallic needle is attached to the syringe and is connected to a high-voltage power source (70). As the solution flows through the metallic tip, it becomes electrified, generating a deformed conical shape known as a Taylor cone. The Taylor cone's tip releases an electrified fibre jet. As the solution travels to a grounded collector, the solvent evaporates, and the fibres harden (70).

Wutticharoenmongkol et al. used electrospinning to create a 12% w/v PCL fibrous scaffold with HA nanoparticle concentrations ranging between 0.5 and 1.0%. The porosity of the constructed fibre scaffold increased by 82 and 90%, and had pore sizes ranging from 4.3 to 5.6 µm. The prepared fibrous scaffolds had a tensile strength between 3.6 and 3.8 MPa (71). Another study by He et al. reported the fabrication of a PCL/HA scaffold with different ratios of PCL/0.3 HA, PCL/0.4 HA, and PCL/0.5 HA (Figure 10a-c) and an average pore size of 167 µm which is suitable for osteoblasts, by stacking meshes using near-field electrospinning (72).

Electrospinning has the advantage of being able to manipulate both the mechanical properties and the porous structure of the fibre by regulating the voltages and distance between the syringe and the collectors (73, 74). Pores generated by this approach, however, are often fewer than a few tens of micrometres in size, making them unsuitable for tissue growth and cell culturing (72). Also, fabricating complex geometry can be challenging using this technique.

3.6. Self-assembly

30 Self-assembly is the process by which the components of a system, whether molecules, 31 polymers, colloids, or macroscopic particles, arrange into ordered and/or functional structures 32 or patterns without external direction as a result of specific, local interactions among the 33 components (75). Collagen should be examined to better understand the origins and

significance of these structural features, as it is one of the most common proteins in the human
body. Collagen is formed within the cell as a triple helix structure by the assembly of three
distinct alpha strands (procollagen) (Figure 4f) (76). Procollagen is enzymatically broken to
generate tropocollagen, which combines and crosslinks with other tropocollagen molecules to
create the characteristic 67 nm banded fibrils after vesicle transit to the exterior of the cell (76).
This fibrillar structure is retained in collagen types I (skin, tendon, and bone), II (cartilage), and
III (skin, muscle) (76).

A recent study conducted by [88] demonstrated the self-assembly of a 3D porous Reduced graphene oxide (RGO) composite scaffold that is composed of graphene oxide (GO) and nano-hydroxyapatite (nHA) with pore sizes ranging from 20–100 µm (Figure 11) (77). The scaffold significantly improved the proliferation, alkaline phosphatase activity (ALP), and osteogenic gene expression of rat bone mesenchymal stem cells (rBMSCs) (77). Another study used a combination of self-assembly and electrospinning techniques to create a hybrid scaffold with a honeycomb using Polyhydroxybutyrate/poly(-caprolactone)/58S sol-gel bioactive glass (PHB/PCL/58S) (78). The scaffold was created by changing the solution composition and concentration during a single electrospinning process (78). The nanofiber contained pores as small as a few micrometres in diameter, while the structure had pores ranging from 200 µm to 1000 µm. This facilitated the cell ingrowth and infiltration of MG-63 osteoblast-like cells into the honeycomb-like scaffold (78).

The self-assembly mechanisms are frequently triggered by the mixing of two elements or by an external stimulus (pH, ionic strength, or temperature), allowing these materials to be injected or even used directly to encapsulate cells, compared to the complex processing needed for other conventional manufacturing methods to fabricate a scaffold (76). In comparison to other manufacturing processes, the mechanisms governing the development of self-assembled nanofibers are generally more complex, requiring more careful molecular design and more intricate synthesis. (Table 3) summarises the main applications, advantages and drawbacks of the manufacturing techniques presented in this section.

30 From the literature review presented above, it can be concluded that conventional 31 manufacturing methods such as gas formation, salt leaching, freeze-drying, and phase 32 separation do not allow for accurate regulation of the internal scaffolding design or the 33 manufacture of complicated structures, which can be accomplished through AM modelled with computer-aided design (CAD) (79). Besides, these conventional methods require good
 manufacturing skills to maintain a consistent architecture of scaffolds. In addition, special care
 must be taken to use toxic solvents that can lead to the death of cells if they are not removed
 completely (80). Another limitation is that scaffolds manufactured in accordance with these
 conventional processing methods have poor mechanical properties (81). Therefore, alternative
 techniques such as 3D printing offer a good opportunity to avoid these issues.

7 4. Three Dimensional Printers8

Industry 4.0, commonly known as digital technology, is revolutionising industries by making factories smarter and assisting manufacturers in increasing quality, productivity, and profitability. 3D printing is a manufacturing tool that has advanced over the last three decades and is an essential component of digital technology. Charles Hull invented the technology in 1986, employing UV-sensitive polymers and ultraviolet light (UV) to generate three-dimensional structures (82). Stereolithography Apparatus (SLA) was the name given to the technology later on. Scientists and engineers have since developed a variety of unique 3D printing techniques. The main advantage of these new technologies is that they enable the fabrication of complex organic shapes and internal features and cavities in components that were difficult or even impossible to fabricate with conventional techniques (83). Additional benefits of 3D printing can include; reduced lead time, elimination of extra processing required for mass customisation, develop supply chain expertise, printing systems and assemblies, fabricating complicated designs in functional components, lightweight production of cellular structures, material recycling and environmentally friendly production, scalable workflow, on-demand production and enhanced service quality (83, 84). (Figure 12) summarises the most employed 3D printing techniques. Binder Jetting, Fused Deposition Modelling, Selective Laser Sintering and Stereolithography are the most employed for the manufacturing of scaffolds and, therefore, will be discussed in detail in the following sections.

In regards to the bio-medical applications, the most significant benefit of 3D printing technologies is allowing the fabrication of completely customised components. In 3D printing, different materials such as polymers, metals, or ceramics can be created layer by layer to produce the desired shape according to a computerised model, in contrast to typical manufacturing or foaming procedures that demand removing and/or adding, such as cutting, bending, and drilling (85). 3D printing technologies have been used in many industries, such as biomedical, automotive, aerospace, defence, and many others. This is due to the capacity of AM technologies to rapidly build complicated structures with precision and accuracy, as well

as the ability to recycle materials. Numerous researchers and industrial organisations have
 focused their efforts on enabling the widespread implementation of 3D printing and
 investigating its potential and limitations. (86-89). Therefore, 3D printing can play a major role
 in the future of tissue engineering in general and bone scaffolds in particular. This is evident as
 the number of studies employing 3D printing technology for bone scaffolds has increased over
 the last decade (Figure 13).

4.1. Binder jetting

Binder jetting starts with a powder bed, the composition of which varies according to the materials employed, which is dispersed over the building platform and flattened with the aid of a roller system (90). Following that, the printer nozzle spreads binder solution in the powdery regions indicated by the CAD. The excess powder is extracted (blown off) after the binder solvent and powder are mixed. The building platform is then lowered, allowing for the deposit and levelling of a new powder sheet (91). Following that, the process will be repeated till the required design is fully fabricated (Figure 12a). After the layer deposition, the part generated that is known as the green part usually has high porosity. In order to reduce the number of pores and to improve its integrity, the component is subjected to cleaning and post-processing operations: depowdering, debinding and finally, a sintering process in a furnace with densifying and strengthening purposes.

In a study conducted by Zeltinger et al., chitosan and hydroxyapatite biocomposite scaffolds were printed using a Z-Corp, Z-510 3D printer to create dense (solid, nonporous, 37.1% porosity) and cylindrical scaffolds (92). These scaffolds were fabricated by applying a 40 wt% lactic acid binder solution to various chitosan/hydroxyapatite composites (20 wt%, 25 wt%, and 30 wt% chitosan) followed by a post-hardening process. The authors observed that the scaffolds printed with 25% chitosan had good mechanical properties, as evidenced by their compression strength of 16.32 MPa and 4.4 GPa Young's modulus (92). Nevertheless, only the fabrication of nonporous scaffolds has achieved the desired mechanical strength. In another study, CALPHAD (Ca) and biodegradable Fe-Mn alloy were used to achieve higher decomposition rates (Figure 14a-b) (93). The achieved ultimate tensile strength was 228.1 MPa for the Fe-Mn and 296.6 MPa for the Fe-Mn-1Ca (93). During tensile testing, a brittle fracture occurred in a porous Fe-Mn-1Ca scaffold with 52.9 % open porosity. Fe-Mn scaffolds with an open porosity of 39.3 % had higher ductility than Fe-Mn-1Ca, demonstrating that scaffold Fe-based alloys with less porosity have higher ductility (Figure 14c-d) (93). This is a concern since porosity is a crucial feature as it promotes the diffusion of oxygen, nutrients, and cellular waste. The

availability and diversity of the powder-binder solutions make the binder jetting attractive for manufacturing bone scaffold (90). On the other hand, a drawback of this technique is that it needs post-processing, which may include heat treatment to assure durability (94).

4.2. Materials jetting

Materials jetting Printing (Bioplotter) is one of the most used 3D printing technologies for cellular research due to the low temperature it requires and the low volume it uses (between 3 and 5 mL) (95). This technology is designed for high-precision printing of small objects using small nozzles with a minimum diameter of 250 µm and low volume (96). The process starts by loading the printing material in a semi-liquid or liquid form into the syringe. Then, pneumatic pressure is applied to extrude the material through the printing nozzle (Figure 12b). The materials are deposited in a layer-by-layer manner, and the process enables the combination of different materials in each layer.

In the study by Poldervaart et al. VEGF was incorporated into a 3D printed matrigel-alginate scaffold to promote vascularisation using BioScaffolder pneumatic dispensing system (97). The incorporation of gelatin microparticles (GMPs) to sustainably regulate the release of VEGF led to higher vascularisation compared to scaffolds with no growth factors and rapidly released VEGF scaffolds when implemented in murine models (Figure 15). In another recent study, a biphasic scaffold model was fabricated with the BioScaffolder by combining the unmodified calcium phosphate cement (CPC) paste with a highly concentrated alginate-based hydrogel paste that was embedded with VEGF by two-channel plotting within a single scaffold (98). The scaffold was designed and manufactured to be used for evaluating a femur defect of size in the range of 200 μ m with a macro porosity of 57%. The scaffolds' size and high porosity made them suitable for enhancing bone regeneration (98).

A unique feature of materials jetting processes is that it allows the printing of cell-laden gels to deliver viable and usable scaffolds, often including other polymeric materials like PCL (99, 100). Another advantage of materials jetting is that it enables the growth factors such as platelet-derived growth factor (PDGF) or vascular endothelial growth factor (VEGF) to be added to the bio-ink to improve cell proliferation and differentiation which promotes angiogenesis (101). Adding these growth factors will increase the tissue formation rate in scaffolds and generate robust tissue as a result of increased differentiation. On the other hand, the shear stress from the nozzles of various sizes can negatively impact cell viability (102).

4.3. Materials extrusion

In materials extrusion, thermoresponsive polymers are heated above their glass transition temperature and then placed on a solid surface. It uses a winding thermoplastic polymer filament that is unwound and extruded through a heated nozzle on a fabrication platform. The polymer solidifies and sets after contact with the platform (103). Upon depositing a layer, the process is repeated in a layer-by-layer process until the part is fully fabricated see (Figure 12c) (104).

Hong et al. employed a multi-head deposition method to combine PCL and PLGA to fabricate a multi-material scaffold with high compressive strength of 3.2 MPa and pore size of around 300 µm with 66.7% porosity (Figure 16). In combination with mussel adhesive proteins as a functional material, the fabricated scaffolds facilitated high cell attachment and proliferation of stem cells derived from human adipose tissue (105). It also yielded positive outcomes in vivo tests, where increased bone regeneration was observed in a calvarial defect of a rat model (105). Overall, FDM was mainly used in combination with other techniques or in indirect 3D printing for tissue engineering purposes.

The key disadvantages of FDM are that it enables multi-material and multi-colour fabrication processes within one component, and the accuracy can go down to ± 0.5 mm (106). However, it prevents any possible toxicity caused mainly by organic solvents that are required for the solubilisation of certain polymers, like dichloromethane, used to solubilise PLGA. The thermoplastic criterion for this technique restricts its application and adaptability in the production of scaffolds, as acrylonitrile butadiene styrene (ABS) is the most often utilised material. Other polymers have been used in FDM, like polycarbonate (PC), polyphenylsulfone (PPSF), and polyetherimide (PEI). However, these materials are not mainly used in tissue engineering applications. (107). Further investigation is needed to determine whether alternative thermoplastics, like polyesters, are suitable scaffolding materials for tissue engineering. Despite this drawback, FDM has been demonstrated to be a viable approach for manufacturing scaffolds for tissue engineering. Polyester, PLA, PCL, as well as PCL and PLA composites like PCL-TCP, HA - PCL and HA - PLA are the main option for FDM printed scaffolds (108-111). Tion

4.4. Powder bed fusion

The laser powder bed fusion (L-PBF) technology begins with the powder layer being smeared on the surface of the base plate, followed by melting powdered particles together using a laser beam (normally a CO_2 laser) in the desired pattern (112). The process is repeated after the first layer is deposited, and then another layer is added on top of the pre-existing one (Figure 12d) (113, 114). The L-PBF technique was used to make scaffolds from biocompatible and biodegradable polymers like poly(lactic acid), polyvinyl alcohol, polycaprolactone, and polyetheretherketone (115). With the development of metal 3D printing, this technology is also employed for the fabrication of metallic scaffolds that can be created out of biocompatible metal alloys such as Ti6Al4V for the fabrication of implants (116).

The use of L-PBF for the manufacturing of scaffolds has been studied by many researchers in the literature. Liu et al. utilised hydroxyapatite (HA), sodium tripolyphosphate and silica sol biocomposite slurry to manufacture scaffolds using L-PBF with different heat treatment temperatures at ambient temperature, 1200 °C, 1300 °C, and 1400 °C (Figure 17a-d). These scaffolds showed significant mechanical strength (up to 43.26 MPa) but had low porosity with a pores size of 5-25 µm. The in vitro research, however, suggested the possibility of using these scaffolds for osteoblast growth, such as cells (117). In another study by I. Gibson, the authors optimised the laser beam power, scan spacing and laser thickness to fabricate a nanocomposite scaffold made of poly(hydroxybutyrate-co-hydroxyvalerate) and calcium phosphate (118). The analysed parameters were found to have a substantial effect on the mechanical properties of the scaffold; compressive properties, precision, and durability (119). Nevertheless, the scaffold's efficiency and utility must be evaluated in vitro and in vivo. Other scholars have fabricated a scaffold using bioresorbable polycaprolactone (PCL) at high precision and high compression moduli ranging from 52-67 MPa. The scaffold was loaded with bone morphogenetic protein-7 (BMP7), and has demonstrated bone generation in vivo (120).

When low porosity and high mechanical strength are needed, the use of L-PBF processes can be beneficial; nonetheless, the need for powdered material to be able to withstand laser heat and resistant shrinking throughout the melting process are some limitations of this technique. Another drawback of L-PBF is the pre-heating and post-heating treatments of the powdered material among the crystallisation glass transition or melting temperatures to lower the shrinking of the scaffolds induced by the laser (117, 121).

59 33

4.5. Vat photopolymerization

Vat photopolymerisation or stereolithography (SLA) technique is based on the fabrication of components from a liquid polymer via a chemical reaction mediated by light. A photocurable polymer is placed on a surface medium and then subjected to UV radiation in the 300-400 nm wavelength range, forming the first layer (Figure 12e) (122). After the initial layer has been hardened, the process is repeated, overlaying the preceding layer until the part is completely fabricated (123, 124). SLA biomaterials include polypropylene fumarate (PPF) with photocross linkers and polyethylene glycol acrylate.

A study conducted by Cooke et al. used the (SLA 250/40) stereolithography for Printing (PPF) scaffolds together with Irgacure 819 photoinitiator. The manufactured scaffolds had a porosity of 90% and a pore size range of 150-800 µm (125, 126). Their study demonstrated the possibility of fabricating scaffolds using (PPF) material (Figure 18). However, in vitro and vivo studies must be carried out to determine the scaffold cytotoxicity and biocompatibility. Despite the magnificent results that SLA can achieve in terms of complex geometries fabrication, Various novel biodegradable and biocompatible photocurable polymers must be developed. In addition, designing and improving visible light-based STA systems is important in order to have a list of polymeric materials (125).

The advantage of the SLA technology is that it allows for precise control and fabrication of high-resolution detailed scaffold geometries that almost perfectly mimic the CAD model. Nonetheless, due to the use of an extra curing phase to enhance the properties of the prototype, the final resolution is affected by the shrinkage usually occurring in the post-processing phase (127, 128). However, the drawback of SLA techniques is that it uses only photopolymers that use photoinitiators (129). In addition, the majority of photoinitiators include radical photopolymerisation by photocleavage, hydrogen extraction, and cationic photopolymerisation, with cationic photoinitiators being incompatible with biomedical applications because of the formation of toxic byproducts. Also, the widely used ultraviolet light source for the polymerisation process poses another risk as reports indicate that this light source is harmful to our DNA cells and might be a potential cause of skin cancer (130, 131). (Table 4) summarises the various 3D printing techniques used in preparing bone scaffolds. Tion

1		
2 3	1	4.6 Directed energy denosition and Sheet lamination
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7 8	3	The directed energy deposition (DED) technology fabricates the required object by melting
9	4	materials using a laser beam while using a nozzle to deposit the material in specific locations,
10 11	5	as demonstrated in (Figure 12f) (132). It mostly uses metal types of materials, such as stainless
12 13	6	steel, aluminum, or copper, in the form of powder or wire (133). This technique usually require
14	7	post-processing due to distortions in the fabricated part (134). Due to the limited types of
15 16	8	materials that can be processed and the poor quality of the fabrication, this technique has not
17	9	been utilized much in biomedical applications
18 19		L.
20	10	In sheet lamination (SL), a sheet material is laminated in a layer-by-layer manner and cut using
21 22	11	a laser beam to fabricate the required object as demonstrated in (Figure 12g). It uses different
23 24	12	types of materials such as paper, metal, and plastic (135). Similarly to directed energy
25	13	deposition, sheet lamination has not been widely used in biomedical applications due to its poor
26 27	14	fabrication quality, the need for post-processing, and the difficulty in fabricating complex
28	15	shapes using this technique (136).
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3	1	In general, both manufacturing technologies have different advantages and disadvantages,
4 5	2	primarily depending on the application of the fabricated object. For example, when complex
6 7	3	geometries and designs are required, additive manufacturing technology has proven to be a
8	4	better option than conventional manufacturing as it allows for the creation of internal structures
9 10	5	(137). In contrast, conventional manufacturing is more precise, can handle a wider variety of
11 12	6	materials, and is better suited for large-scale production than additive manufacturing (138).
13	7	Additive manufacturing is a relatively new technology that is still being developed and has
14 15	8	limitations regarding the types of materials it can process and the size of the components it can
16 17	9	produce (138). In fact, researchers have combined conventional and additive manufacturing
18 10	10	technologies into a single machine for bone scaffold fabrication effectively combining their
20 21	11	respective advantages.
22 23	12	The research conducted by Jiankang He and his team, a new printing method was developed
24	13	that combines FDM and electrospinning technologies to create 3D tissue-engineered scaffolds
25 26	14	with intricate curved shapes and microscale fibrous structures (Figure 19a). The melting
27 28	15	temperature was optimized to print PCL filaments of around 10 um, which were stacked to
29 30	16	create 3D walls with smooth surface (139) By adjusting the stage movement speed and
31	17	direction they were able to print PCL scaffolds with curved outlines, predefined fiber spacing
32 33	18	and orientations at 90° and 45° (Figure 19b-g). Biological experiments demonstrated that the
34 35	19	printed microscale scaffolds were biocompatible and promoted in vitro cellular proliferation
36	20	and alignment (139). In another research conducted by H. Hassanin et al., they successfully
37 38	20	produced micro implantable components with the highest density and best surface quality
39 40	21	possible by utilizing a hybrid microfabrication technology that incorporates the design
41	22	flowibility of SLM and the executional surface quality of a EDM (140). Another group of
42 43	23 24	researchers has developed a new approach to grante three dimensional graphers (2DC)
44 45	24 25	compositor souffold by combining selective locar molting (SLM) and chamical years deversition
46	25	composites scattoid by combining selective faser menting (SEW) and chemical vapor deposition (CVD) (1.41). The of this (1.42) are selective faser menting (SEW) and chemical vapor deposition
47 48	26	(CVD) techniques (141). They fabricated a 3D porous copper template using SLM and grew
49 50	27	graphene in-situ via CVD on the template. This technique allowed for accurate control of the
51	28	design and regulation of 3DG, resulting in enhanced electromagnetic interference (EMI)
52 53	29	shielding and improved thermal diffusion (141).
54 55 56	30	
57 58	31	5. Conclusions

Conclusions 5.

> This paper presents a literature review of the most relevant works and recent advances concerning manufacturing bone scaffolds. Conventional manufacturing techniques have been

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reviewed, and their main benefits and shortcomings have been addressed. Additionally, 3D printing technologies that have emerged in the last years have proved to be a feasible alternative. In this context, the review demonstrated that 3D printing technologies enable the customisation of bone scaffolds to meet individual patients' unique needs and health situations. Progress in this area is facilitated by advancements in computer-aided design (CAD) and computer-aided manufacturing (CAM), which enable rapid and precise organ scanning and design. The scaffold's structural properties, such as pore size and porosity, have a direct influence on their functionality in both vitro and vivo. In general, interconnected porous scaffolding networks that allow nutrient transport and waste disposal and promote cell migration and proliferation are significant. Pore size and porosity affect the behaviour of the cells and determine the overall mechanical properties of the scaffold. Presently, the concept of fabricating scaffolds is concentrated on generating materials with suitable pore size, structure, and porosity for specific uses. Typically, scaffolds are 3D printed, and cells are grown in/on these scaffolds. One of the challenges of 3D printing is using non-biocompatible materials in several 3D printing techniques, such as binders or photoinhibitors, even after the high-temperature debinding or sintering process. These components cannot remove entirely after heating or sintering processes and may compromise the biocompatibility of the constructs. Also, applying the temperature in some of the technologies restricts the applicability of materials. Incompatibility of the cellular application with the scaffold would gradually cause the entire scaffolding system to fail.

In addition, 3D printing technology has altered the way bone fractures and has enabled the utilisation of drug-loaded implants and/or scaffolds of complicated geometries and high resolution to accelerate the healing procedure and recover bone structure and toughness. Bone scaffolds have been extensively manufactured using techniques like FDM and binder jetting. FDM has been demonstrated to be capable of processing a wide variety of scaffolds with complicated structures and a variety of polymeric materials. On the other hand, techniques like directed energy deposition and sheet lamination (Figure 12f-g) were not explored in this field due to their processing characteristics or the quality of their products or materials. Clinical trials conducted by academia or commerce on the developed systems demonstrate significant potential. However, challenges such as materials recycling, quality control, and the effect of inherited issues of 3D printing such as surface roughness, internal defects, and post-processing are still lacking.

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Figure 1. Flowchart of manufacturing technologies.

127x129mm (220 x 220 DPI)





Figure 2. A long bone's macroscopic structure.

160x113mm (96 x 96 DPI)



Figure 3. A comparison between the stress-strain properties of trabecular and cortical bones. Adapted from Damien Lacroix (21).

140x92mm (113 x 113 DPI)

Freeze- Drye

Electrified Jet

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Figure 4. Conventional manufacturing techniques of bone scaffolds, (a) Salt leaching, (b) Gas forming, (c) Phase separation, (d) Freeze-drying, (e) Electrospinning and (f) Self-assembly.

169x213mm (220 x 220 DPI)





Figure 5. Number of papers published on bone scaffold fabricated by each conventional manufacturing technique over the last 10 years.

150x97mm (330 x 330 DPI)





Figure 6. Collagen-coated chitin scaffold morphologies: (a) cross-section and (b) surface. Adopted from Sang Bong Lee (47).

160x63mm (113 x 113 DPI)





159x64mm (96 x 96 DPI)







Figure 9. Effect of freezing temperature on morphology of the matrix. Collagen-hyaluronic acid scanning electron micrograph freezes dried at -20 ° C (A), -70 ° C (B) and -196 ° C (C) (magnification ×200). Adopted from SN Park (66).

161x62mm (113 x 113 DPI)



Figure 10. Morphological characterisation of the PCL/HA composite scaffolds. SEM image of (a) PCL/0.3 HA scaffold, (b) PCL/0.4 HA scaffold and (c) PCL/0.5 HA scaffold. Adopted from Feng-LiHe (72).

166x48mm (113 x 113 DPI)



Figure 11. SEM of the nHA@RGO scaffold with the different nHA loading ratios. Reduced graphene oxide (RGO) and nano-hydroxyapatite (nHA). Adopted from WeiNie (77).

162x71mm (113 x 113 DPI)





Figure 12. 3D printing techniques, (a) Binder jetting, (b) Materials jetting, (c) Materials extrusion, (d) Powder bed fusion, (e) Vat photopolymerisation, (f) Directed energy deposition, (g) Sheet lamination.

164x219mm (220 x 220 DPI)





Figure 13. Number of papers published on bone scaffold fabricated by each 3D printing technique over the last 10 years.

-----Powder bed fusion

153x99mm (330 x 330 DPI)



Figure 14. Morphology of (a) Fe-Mn powders (b) Fe-Mn-1Ca powders, (c) 3D printed Fe-Mn sample, and (d) 3D printed Fe-Mn-1Ca sample. Adopted from Hong D (93).

119x82mm (113 x 113 DPI)





Figure 16. SFF-based 3-D PCL/PLGA scaffold. Adopted from J. M. Hong (105).

122x81mm (113 x 113 DPI)



Figure 17. SEM images of the scaffold at (a) ambient temperature, (b) 1,200 °C, (c) 1,300 °C, and (d) 1,400 °C. hydroxyapatite (HA) and Ceramic-matrix composites (CMCs). Adopted from F.-H. Liu (117).

114x104mm (216 x 216 DPI)



Figure 18. fabricated scaffold using SLA. Adopted from M. N. Cooke (124).

113x84mm (100 x 100 DPI)





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Figure 19. Fiber orientation manipulation during the melt electrohydrodynamic printing process. (a) Schematic illustration of manipulating fiber orientation through directing stage movement, (b-d) microscopic images of the printed scaffold, which have fibers spaced at 1 mm intervals and oriented at 90° and (e-g) oriented at 45°. Adopted from Jiankang He (139).

159x115mm (96 x 96 DPI)

Figure 1. Flowchart of manufacturing technologies.

Figure 2. A long bone's macroscopic structure.

Figure 3. A comparison between the stress-strain properties of trabecular and cortical bones. Adapted from Damien Lacroix (21).

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Figure 6. Collagen-coated chitin scaffold morphologies: (a) cross-section and (b) surface. Adopted from Sang Bong Lee (47).

Figure 7. SEM images of surface morphology of PLLA scaffolds. Adopted from Nam (56).

Figure 8. scanning electron micrograph of PLLA membranes as a function of aging time at quenching temperatures of 25°C (A), 30°C (B), and 35°C (C). Adopted from H Do Kim (62).

Figure 9. Effect of freezing temperature on morphology of the matrix. Collagen-hyaluronic acid scanning electron micrograph freezes dried at $-20 \degree C (A)$, $-70 \degree C (B)$ and $-196 \degree C (C)$ (magnification ×200). Adopted from SN Park (66).

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Figure 15. 3D printed matrigel-alginate scaffold the two regions (– without VEGF, + VEGFladen GMPs). Adopted from M. T. Poldervaart (97).

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Table 1. Terms,	definitions	and examples	of bone	repair	[40].
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Term	Definition	Example
Osteogenesis	The process by which new bone is synthesised	Stem cell, autografts transplants
	using donor cells taken either from the host or the	
) (graft donor.	
Osteoconduction	The passive ingrowth of host vasculature, tissue,	Phosphate cements or calcium
	and cells into an implanted scaffold.	sulphate resorption
Osteoinduction	Exogenous growth factors enable host	Proteins involved in bone
	mesenchymal stem cells (MSCs) to differentiate	morphogenesis
	into osteoblasts and chondroblasts capable of	
	producing new bone.	
	0	

Table 2. Morphological properties of collagen-HA membranes. Adopted from SN Park [66].

Freezing temperature	Porosity (%)	V	Pore size (µm)	
	Before crosslinking	After crosslinking	Before crosslinking	After crosslinking
-196°C	58.1±3.4	61.95±3.8	40±7	84±20
-70°C	59.28±4.9	62.3±4.8	90±16	186±29
-20°C	66.46±2.6	64.93±2.3	230±52	190±42
			<).
3 . summary of	conventional ma	nufacturing.		
Casturing				

Table 3. summary of conventional manufacturing.

Manufacturing technique	Main applications	Advantages	Disadvantages	Ref.
Salt leaching	Forming porosity in part by employing salt particles	 Controlled pore size Suitable for manufacturing of membranes 	 Lack of pores distribution control Lack of scaffold shape control 	[43, 48] [50]

Gas foaming	Forming porosity in part by applying high pressure	 It does not use chemical solvents It is capable of fabricating parts with very high porosity 	 Pore sizes are difficult to control It employs a high temperature Lack of scaffold shape control
Phase separation	Forming porosity in part by employing chemical solvents	 It does not require post-processing 	 Pore sizes are very small [4] Lack of scaffold shape 59 control
Freeze-drying	Forming porosity in part by freezing the liquid mixture	• Pore sizes can be adjusted by controlling the temperatures	 Lack of scaffold shape control high energy 68 consumption uses cytotoxic solvents
Electrospinning	Forming solid nano size fibres	 Mechanical and porosity properties of the fibre can be controlled by regulating the voltage and distance 	• Lack of scaffold shape 74 control
Self-assembly	Forming a pattern by the interactions of two components without external direction	 It does not require the use of cytotoxic solvents 	 Pore sizes are very small It necessitates greater attention to molecular design and intricate synthesis

Table 4. summary of the 3D printing techniques.

Binder jetting PLGA, PLLA, PEEK-HA, PCL, starch- based polymer • No support structure • Can require post- finish [142, Poccessing Materials jetting 10-1000 PCL, PLLA, TCP, Hydrogel, Organic ink • Uses a variety of materials • Trapped powder [144] Materials jetting 10-1000 PCL, PLLA, TCP, Hydrogel, Organic ink • Uses an enhanced range of materials • Low mechanical strength • Low mechanical strength • Low accuracy • Slow processing Materials extrusion 250 PCL, PP-TCP, PCL-HA, PCL- TCP, PLTA, TCP, PLTA, TCP, PLTA, TCP, PLTA • Good mechanical strength • High temperature filament material • Need to produce filament material [100] Powder bed fusion 500 PEEK-HA, PCL, titanium, stainless steel, cobalt- chromium alloys • Microporosity induced in the scaffold • Material must be in powder form [153] Vat photopolymerisation 366 Resin, PPF, polyethylene glycol acrylate, HA • Control of both external and internal morphology • Multistep involved structure deced strength • Multistep involved strength [156] Vat photopolymerisation 366 Resin, PPF, polyethylene glycol acrylate, HA • Control of both external and internal morphology • Multistep involved strength [157] Votageprolactore (PCL), Tricalcium phosph	Binder jetting 200-300 PLGA, PLLA, PEEK-HA, PEEK-HA, PEEK-HA, PCL, starth- based polymer • No support structure • Can require post- processing 142, Powdery surface finish Materials jetting 10-1000 PCL, PLLA, TCP, Hydrogel, Organic ink • Uses an enhanced range of materials • Low mechanical strength [142, Powdery surface finish Materials jetting 10-1000 PCL, PP-TCP, PCL-HA, PCL- TCP, PFTG- TCP, PFTG- TCP, PETG- TCP, PETG- TCP, PLA • Good mechanical strength • Low mechanical strength 10400 Materials extrusion 250 PCL, PP-TCP, PCL-HA, PCL- TCP, PLA • Good mechanical strength • High temperature filament material strength • Narrow processing window • Complex design require support structure • Narrow processing window • Narrow processing window • Narrow processing window • Complex design can require support structure • Narrow processing window • Trapped powder 152] Vat photopolymerisation 500 PEEK-HA, PCL, titanium, suitess steet, cobalt- chromium • Microporosity induced in the scafiold • Material must be in powder form structure needed Powder suites • Trapped powder • Trapped powder Vat photopolymerisation 366	AM technology	Resolutio	Material	Strength	Weakness	Refs
Materials jetting 10-1000 PCL, PLLA, TCP, Hydrogel, Organic ink • Uses an enhanced range of materials • Low mechanical strength [144- 149] Materials jetting 0 PCL, HA, PCL- TCP, PETG- PBT, PLLA- TCP, PETG- PBT, PLLA- TCP, PLA • Good mechanical strength • Smooth surface Low accuracy Slow processing [108, Vaterials extrusion Powder bed fusion 500 PEEK-HA, PCL, titanium, Stainless steel, chromium alloys • Microporosity induced in the scaffold • Material must be in powder form [153- topowder form Vat 366 Resin, PPF, photopolymerisation 366 Resin, PPF, polyethylene glycol acrylate, HA • Control of both external morphology • • Multistep involved range of materials • 156, 157] Vat 366 Resin, PPF, polyethylene glycol acrylate, HA • • • • Multistep involved range of materials • 157] Vat 366 Resin, PPF, polyethylene glycol acrylate, HA • • • • • • 157] • 100 • • • • • • 157] Vat 366	Materials jetting 10-1000 PCL, PLA, TCP, Hydrogd, Organic ink • Uses an enhanced range of materials • Low mechanical strength • Low mechanical strength • Low mechanical strength • Low accuracy Materials extrusion 250 PCL, PP-TCP, PCL-HA, PCL- TCP, PETC PBT, PLLA- TCP, PLA • Good mechanical strength • High temperature strength • High temperature strength • Narrow processing window • Narrow processing window • Complex design can require support structure • Microporosity induced in the scaffold • Microporosity induced in the scaffold • Microporosity induced in the scaffold • Material must be in powder form 153- • No support structure 155- • Trapped powder Vat 366 Resin, PPF, plotopolymerisation • Control of big lycol acrylate, HA • Control of big lycol acrylate, High accuracy • Multistep involved strength 156- • Thermal damage can occur during processing 157] Vat 366 Resin, PPF, plotopolymerisation • Control of High accuracy • Lises an enhanced range of materials • High accuracy • Fast processing • Multistep involved • Fast processing • Multistep involved • Damages cell during photo curing • UV blue light can be toxic to cells 157] voldputchete-co-glycolic acid/ (PLGA). Poly(L-lactide) (PLLA). (polyeterbeterketkerkety (PEK), Hydroxyaptife (HA) • Volvethylene terphthalate (PBT) and Paint protection film (PPF). <	Binder jetting	200-300	PLGA, PLLA, PEEK-HA, PCL, starch- based polymer	 No support structure is required Fast processing Uses a variety of materials 	 Can require post- processing Powdery surface finish Trapped powder 	[142, 143]
Materials extrusion250PCL, PP-TCP, PCL-HA, PCL- TCP, PETG- PBT, PLLA, TCP, PLAGood mechanical strengthHigh temperature Need to produce filament material Narrow processing window[108, 150]Powder bed fusion500PEEK-HA, PCL, titanium, Stainless steel, cobalt- chromium alloys• Microporosity induced in the scaffold• Microporosity induced in the scaffold• Material must be in powder form[153]Vat photopolymerisation366Resin, PPF, polyethylene glycol acrylate, HA• Control of both external ange of materials High accuracy • Fast processing• Multistep involved to cells[156, 157]Vat photopolymerisation366Resin, PPF, polyethylene glycol acrylate, HA• Control of both external ange of materials High accuracy • Fast processing• Multistep involved to cells[156, 157]Vat photopolymerisation366Resin, PPF, polyethylene (PLGA), Poly(L-lactide) (PLLA), (polyetheretherketone) (PEEK), Hydroxyapatie (HA), Volveaprolactone (PCL), Tricalcium phosphate (TCP), Polypropylene (PP), Polyethylene terephthalate glycol (PETG),	Materials extrusion 250 PCL, PP-TCP, PCL-HA, PCL- TCP, PETG- PBT, PLLA- TCP, PLA • Good mechanical strength • High temperature Need to produce filament material [108, 150- 152] Powder bed fusion 500 PEEK-HA, PCL, titanium, Stainless steel, cobalt- chromium alloys • Microporosity induced in the scaffold • Material must be in powder form [133- powder form Vat 500 PEEK-HA, PCL, titanium, Stainless steel, cobalt- chromium • Microporosity induced in the scaffold • Material must be in powder form [153- powder form Vat 366 Resin, PPF, polyethylene glycol acrylate, HA • Control of both external and internal morphology • Multistep involved structure and internal morphology • Multistep involved structure needed range of materials • Multistep involved processing [156, 157] Vat 366 Resin, PPF, polyethylene glycol acrylate, HA • Uses an enhanced range of materials • Multistep involved structure needed range of materials • Damages cell during photo curing [156, 157] • Uses an enhanced range of materials • Damages cell during photo curing • Damages cell during photo curing • Damages cell during photo curing • Fast processing • Wythoxyapatite (HA). • Polytehretherketone) (PEEK). Hydroxyapatite (HA). Polytaprolactone (PCL), Tricalcium phosphate (TCP). Polypropylene (PP). Polytet	Materials jetting	10-1000	PCL, PLLA, TCP, Hydrogel, Organic ink	 Uses an enhanced range of materials Can incorporate biomolecule 	 Low mechanical strength Smooth surface Low accuracy Slow processing Complex design requires support structure 	[144- 149]
Powder bed fusion500PEEK-HA, PCL, titanium, Stainless steel, cobalt- chromium alloys• Microporosity induced in the scaffold• Material must be in powder form[153- 155]Vat photopolymerisation366Resin, PPF, polyethylene glycol acrylate, HA• Control of both external and internal morphology• Material must be in powder form[153- total total total total total total total[153- total powder formVat photopolymerisation366Resin, PPF, polyethylene glycol acrylate, HA• Control of both external and internal morphology• Multistep involved total strength[156, 157]Voly(lactic-co-glycolic acid)(PLGA), Poly(L-lactide)(PLLA), (polyethylene (TCP), Polypropylene (PP), Polyethylene (PP), Polyethylene (PEEK), Hydroxyapatie (HA), Polycaprolactone (PCL), Tricalcium phosphate (TCP), Polypropylene (PP), Polyethylene (PP), Polyethylene <td>Powder bed fusion 500 PEEK-HA, PCL, titanium, Stainless steel, cobalt- chromium alloys • Microporosity induced in the scaffold • Material must be in powder form [153- 155] Vat No support structure needed • No support structure needed • Trapped powder • Trapped powder Vat 366 Resin, PPF, polyethylene glycol acrylate, HA • Control of both external and internal • Multistep involved [156, Poor mechanical strength Output 366 Resin, PPF, polyethylene glycol acrylate, HA • Control of both external and internal • Multistep involved [156, Poor mechanical strength Output 157] Polyclactic-co-glycolic acid) (PLGA), Poly(L-lactide) (PLLA), (polyetheretherketone) (PEEK), Hydroxyapatite (HA), Polycaprolactone (PCL), Tricalcium phosphate (TCP), Polypropylene (PP), Polyethylene terephthalate glycol (PETG), Polybutylene terephthalate (PBT) and Paint protection film (PPF). Fast processing Itemption film (PPF).</td> <td>Materials extrusion</td> <td>250</td> <td>PCL, PP-TCP, PCL-HA, PCL- TCP, PETG- PBT, PLLA- TCP, PLA</td> <td> Good mechanical strength Preparation time is reduced </td> <td> High temperature Need to produce filament material Narrow processing window Complex design can require support structure </td> <td>[108, 150- 152]</td>	Powder bed fusion 500 PEEK-HA, PCL, titanium, Stainless steel, cobalt- chromium alloys • Microporosity induced in the scaffold • Material must be in powder form [153- 155] Vat No support structure needed • No support structure needed • Trapped powder • Trapped powder Vat 366 Resin, PPF, polyethylene glycol acrylate, HA • Control of both external and internal • Multistep involved [156, Poor mechanical strength Output 366 Resin, PPF, polyethylene glycol acrylate, HA • Control of both external and internal • Multistep involved [156, Poor mechanical strength Output 157] Polyclactic-co-glycolic acid) (PLGA), Poly(L-lactide) (PLLA), (polyetheretherketone) (PEEK), Hydroxyapatite (HA), Polycaprolactone (PCL), Tricalcium phosphate (TCP), Polypropylene (PP), Polyethylene terephthalate glycol (PETG), Polybutylene terephthalate (PBT) and Paint protection film (PPF). Fast processing Itemption film (PPF).	Materials extrusion	250	PCL, PP-TCP, PCL-HA, PCL- TCP, PETG- PBT, PLLA- TCP, PLA	 Good mechanical strength Preparation time is reduced 	 High temperature Need to produce filament material Narrow processing window Complex design can require support structure 	[108, 150- 152]
Vat photopolymerisation366Resin, PPF, polyethylene glycol acrylate, HA• Control of both external and internal morphology• Multistep involved 157][156, 157]• Damages cell during photo curing Poly(lactic-co-glycolic acid) (PLGA), Poly(L-lactide) (PLLA), (polyetheretherketone) (PEEK), Hydroxyapatite (HA), Polycaprolactone (PCL), Tricalcium phosphate (TCP), Polypropylene (PP), Polyethylene terephthalate glycol (PETG),• Multistep involved • Multistep involved • Dor mechanical strength • Damages cell during photo curing • UV blue light can be toxic to cells[156, 157]	Vat photopolymerisation 366 Resin, PPF, polyethylene glycol acrylate, HA • Control of both external and internal morphology • Multistep involved Poor mechanical strength [156, 157] • Uses an enhanced range of materials • Uses an enhanced range of materials • Damages cell during photo curing 157] • Volution of both glycol acrylate, HA • High accuracy • Est processing • UV blue light can be toxic to cells 140, 157] • Oly(lactic-co-glycolic acid) (PLGA), Poly(L-lactide) (PLLA), (polyetheretherketone) (PEEK), Hydroxyapatite (HA), • Olybutylene terephthalate glycol (PETG), • Olybutylene terephthalate (PBT) and Paint protection film (PPF). • Olybutylene terephthalate (PBT)	Powder bed fusion	500	PEEK-HA, PCL, titanium, Stainless steel, cobalt- chromium alloys	 Microporosity induced in the scaffold Uses an enhanced range of materials No support structure needed Fast processing 	 Material must be in powder form High temperature Powdery surface finish Trapped powder Thermal damage can occur during processing 	[153- 155]
oly(lactic-co-glycolic acid) (PLGA), Poly(L-lactide) (PLLA), (polyetheretherketone) (PEEK), Hydroxyapatite (HA), olycaprolactone (PCL), Tricalcium phosphate (TCP), Polypropylene (PP), Polyethylene terephthalate glycol (PETG),	Poly(lactic-co-glycolic acid) (PLGA), Poly(L-lactide) (PLLA), (polyetheretherketone) (PEEK), Hydroxyapatite (HA), Polycaprolactone (PCL), Tricalcium phosphate (TCP), Polypropylene (PP), Polyethylene terephthalate glycol (PETG), Polybutylene terephthalate (PBT) and Paint protection film (PPF).	Vat photopolymerisation	366	Resin, PPF, polyethylene glycol acrylate, HA	 Control of both external and internal morphology Uses an enhanced range of materials High accuracy Fast processing 	 Multistep involved Poor mechanical strength Damages cell during photo curing UV blue light can be toxic to cells 	[156, 157]
	Polybutylene terephthalate (PBT) and Paint protection film (PPF).	oly(lactic-co-glycolic acid) olycaprolactone (PCL), Tri	(PLGA), Pol calcium phosp	y(L-lactide) (PLLA) phate (TCP), Polypro	• Fast processing , (polyetheretherketone) (PE pylene (PP), Polyethylene t	EEK), Hydroxyapatite (HA), erephthalate glycol (PETG),	