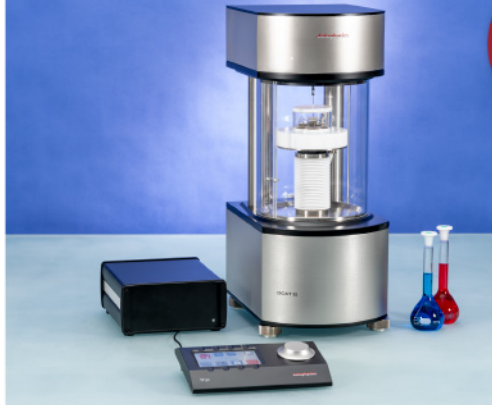




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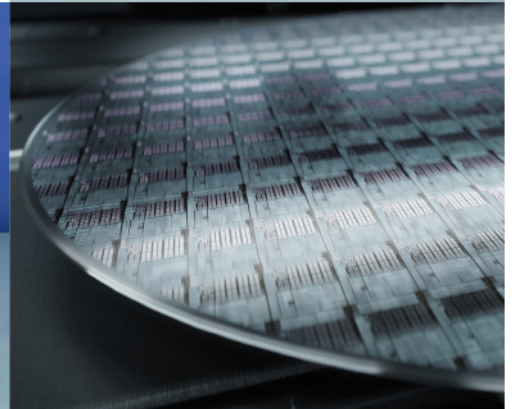


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# Modular Type III Porous Liquids Based on Porous Organic Cage Microparticles

Aiting Kai, Benjamin D. Egleston, Andrew Tarzia, Rob Clowes, Michael E. Briggs, Kim E. Jelfs, Andrew I. Cooper,\* and Rebecca L. Greenaway\*

The dispersion of particulate porous solids in size-excluded liquids has emerged as a method to create Type III porous liquids, mostly using insoluble microporous materials such as metal–organic frameworks and zeolites. Here, the first examples of Type III porous liquids based on porous organic cages (POCs) are presented. By exploiting the solution processability of the POCs, racemic and quasiracemic cage microparticles are formed by chiral recognition. Dispersion of these porous microparticles in a range of size-excluded liquids, including oils and ionic liquids, forms stable POC-based Type III porous liquids. The flexible pairing between the solid POC particles and a carrier liquid allows the formation of a range of compositions, pore sizes, and other physicochemical properties to suit different applications and operating conditions. For example, it is shown that porous liquids with relatively low viscosities or high thermal stability can be produced. A 12.5 wt% Type III porous liquid comprising racemic POC microparticles and an ionic liquid, [BPy][NTf<sub>2</sub>], shows a CO<sub>2</sub> working capacity (104.30 μmol g<sup>-1</sup>) that is significantly higher than the neat ionic liquid (37.27 μmol g<sup>-1</sup>) between 25 and 100 °C. This liquid is colloidal stable and can be recycled at least ten times without loss of CO<sub>2</sub> capacity.

in 2007 by O'Reilly et al.<sup>[1]</sup> Three types of porous liquids were initially proposed (Types I, II, and III; Figure 1a) and after more than a decade, examples of each type have now been reported.<sup>[2]</sup> Type I porous liquids are neat liquid substances with shape-persistent molecular porosity; Type II porous liquids are solutions of natively porous molecules dissolved in a non-porous solvent, where the pores remain unoccupied because the solvent molecules are size excluded; Type III porous liquids are multiphase fluids, typically porous solid particles dispersed in size-excluded liquids. Type III porous liquids are typically simpler to prepare than the other two types, and this approach might prove to be more generalizable in the future. For Type I porous liquids, it is necessary to devise strategies for lowering the melting points of shape-persistent porous molecules, such as cages and macrocycles, and the viscosity of such liquids may also be

## 1. Introduction

Porous liquids are a novel kind of porous material that are distinguished by their ability to flow. These materials have seen considerable progress since the concept was first introduced

limiting. Type II porous liquids are somewhat easier to access, though this requires strategies for increasing the solubility of the pore carriers in the size-excluded solvents, and again viscosity may be an issue at high concentrations of the pore carrier. By contrast, the formation of Type III porous liquids can be approached in a modular way, and it is feasible to screen a wide range of porous materials and bulky liquids to find suitable combinations.<sup>[3]</sup> This modular approach might prove more industrially applicable, not least because one can draw on a large range of existing materials as building blocks, some of which have already been scaled up.<sup>[4]</sup> So far, Type III porous liquids have mostly been constructed from metal–organic frameworks (MOFs) and zeolites. For example, ZIF-8 was found to be dispersible in several large hindered ionic liquids,<sup>[5–7]</sup> or in polymeric glycols,<sup>[8]</sup> while Cahir et al. reported the formation of a range of Type III porous liquids using various microporous solids, including MOFs and zeolites, in nonionic oils.<sup>[3]</sup> Until now, Type III porous liquids have been built from extended porous materials that are insoluble in common liquids.

Porous organic cages (POCs) are a type of molecular organic material that contains shape-persistent internal voids in the molecules that are accessible through windows.<sup>[9–11]</sup> Imine-derived POCs can be synthesized in solution through a reversible condensation reaction, and many derivatives can be produced based on the selection of different synthetic

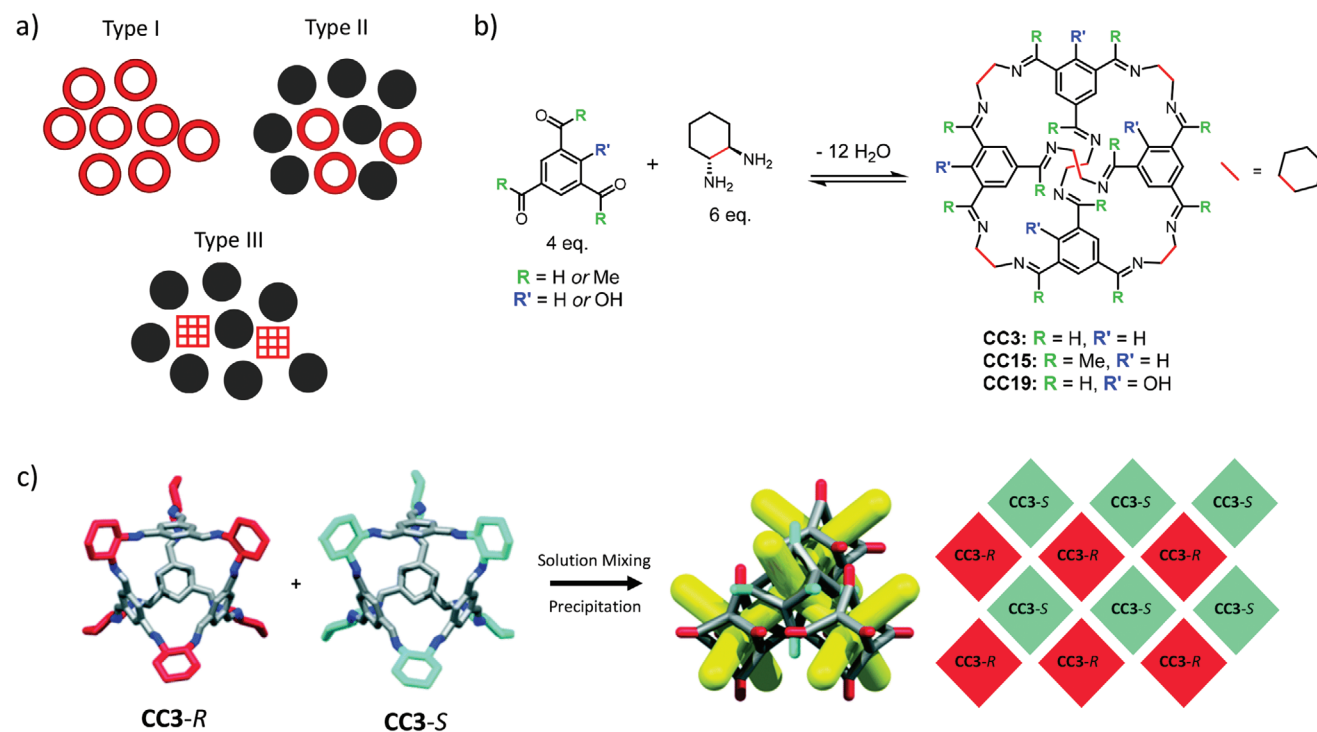
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**Figure 1.** a) Scheme showing the three classifications of porous liquids: Type I, Type II, and Type III. b) General reaction scheme for the synthesis of **CC3**, **CC15**, and **CC19**, which are the POCs used to form microparticles in this study. Two cage enantiomers can be formed for each species; for example, **CC3-R** and **CC3-S** can be formed by using the different enantiomers of *trans*-1,2-cyclohexyldiamine (CHDA), (*R,R*)-CHDA, and (*S,S*)-CHDA. c) POCs such as **CC3** can pack together window-to-window to form a 3D diamondoid pore network in the solid state by chiral recognition between the two cage enantiomers, which can be exploited to form POC microparticles for processing into Type III porous liquids.<sup>[19]</sup>

precursors: a representative scheme for preparing the POCs used in this study is shown in Figure 1b. The window sizes and chemical environment of POCs can be tuned by introducing different functional groups into the precursors.<sup>[12]</sup> POCs can also have good solubility in organic solvents, and are thus a candidate for preparing Type II porous liquids.<sup>[13–15]</sup> There have also been attempts to decorate cages with long alkyl chain functionality to lower the melting points in the preparation of Type I porous liquids.<sup>[16,17]</sup> Some of the properties of POCs can be retained when transferred into the liquid state, such as size-selective guest uptake,<sup>[18]</sup> while others, such as chiral recognition,<sup>[14]</sup> are lost. One potential way to retain the solid-state properties of POCs is to form Type III porous liquids, rather than Type I or Type II systems.

As a multiphase system, the colloidal stability of Type III porous liquids is an important consideration. For example, Dai and co-workers used hydrogen bonding between H-ZSM-5 zeolites and bulky phosphonium ionic liquids to construct a colloidal stable Type III system,<sup>[20]</sup> whereas Li and co-workers and Liu et al. used surface functionalization to coat particles of UiO-66 or silicalite-1, respectively, with solvent-similar polymeric groups to stabilize these as Type III porous liquids.<sup>[21,22]</sup> More recently, the surface functionalization method was also used to form a stabilized ZIF-67 dispersion, which can be used directly as a Type III porous liquid or conveniently processed further to form mixed matrix membranes with excellent selectivity between propane and propylene.<sup>[23]</sup> While a 3D imine-linked covalent organic framework (COF) was functionalized with ionic liquid functionality on its outer shell, forming COF

colloids that were extremely stable toward flocculation (>1 year) in an ionic liquid.<sup>[24]</sup> Alternatively, in ZIF-8-based Type III porous liquids, a rapid synthetic method can be employed to produce nanocrystalline particles that allow for stable emulsions when the solid ZIF-8 is dispersed.<sup>[25]</sup>

POC nano- and microparticles can also be easily fabricated from solution by either direct synthesis of a cage in a binary solvent mixture,<sup>[26]</sup> or by exploiting the chiral recognition between two prefabricated homochiral precursor cages, producing cocrystals in a modular way without the use of any surfactants or templating agents.<sup>[19,27]</sup> These can either be prepared from enantiomeric variants of the same cage species to form a racemate (Figure 1c), or formed from isostructural cage derivatives of opposite chirality to produce quasiracemic cocrystals.<sup>[28–30]</sup> By using two chemically different cages to form quasiracemic cocrystals, it is possible to tune the properties of the pore carrier, such as the pore size or the chemical environment in the pores—for example, to produce materials with superior properties for quantum sieving of hydrogen isotopes.<sup>[31]</sup> Furthermore, upon cocrystallization, the solubility of these racemic and quasiracemic cage materials is often greatly reduced compared to the solution-processable enantiopure cage precursors, thus making them good candidates for preparing Type III porous liquids.

Here, we report the first use of POC assembled microparticles to form stable Type III porous liquids. Unlike POC-based Type II porous liquids whose increase in gas uptake arises solely from the intrinsic cavity of the POC, the porosity in POC microparticle-based Type III porous liquids arises from

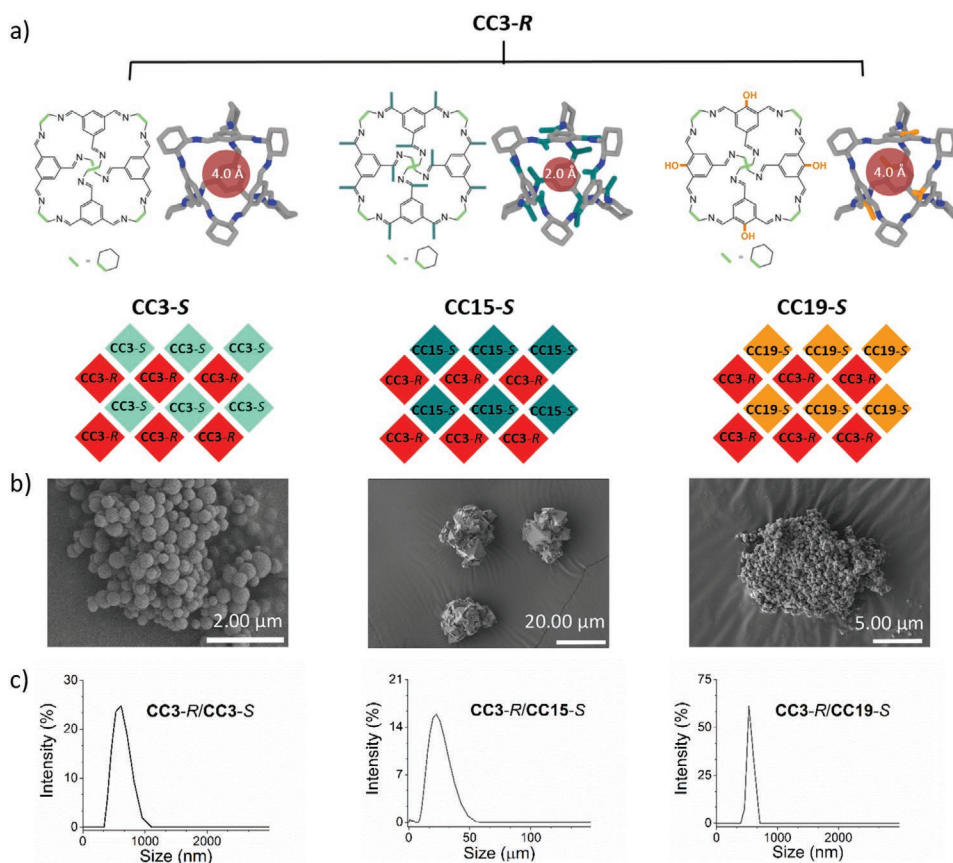
both the intrinsic cage cavities and the extrinsic connections between pores. However, all porous liquids must contain permanent intrinsic porosity from a pore carrier, rather than just the transient extrinsic porosity, which is present in all liquids,<sup>[1]</sup> and it is within this context that we present these systems. Different compositions of cage microparticles (CC3-R/CC3-S, CC3-R/CC15-S, and CC3-R/CC19-S) were dispersed in a selection of size-excluded liquids including a range of oils and ionic liquids, and their gas uptakes were investigated. Overall, a versatile Type III porous liquid platform can be built using different POCs in neat liquids, without the need for any surfactants or postsynthetic modification, with physical properties that can be tailored according to the application.

## 2. Results and Discussion

### 2.1. Synthesis and Characterization of POC Microparticles

First, the formation of three different POC microparticles was investigated: racemic CC3-R/CC3-S, quasiracemic CC3-R/CC15-S, and quasiracemic CC3-R/CC19-S (Figure 2a). The synthesis of CC3-R/CC3-S microparticles by chiral recognition was reported previously, where homochiral solutions of CC3-R and CC3-S were mixed and uniformly fine microparticles

were formed.<sup>[19]</sup> Using a slight modification of that method, multiple large-scale batches of CC3-R/CC3-S microparticles were formed by rapidly mixing dilute solutions of both CC3-R and CC3-S in dichloromethane at room temperature with overhead stirring (Table 1, entries 1–5). Rapid precipitation of the POC microparticles was observed upon mixing, which were isolated by filtration—due to the uncontrolled nature of this rapid-mixing technique, this led to some variation in the average size of the obtained particles. However, the particle size can be tuned by finely controlling either the addition rate or temperature,<sup>[19]</sup> and therefore an additional batch of CC3-R/CC3-S was formed using a carefully controlled addition rate for comparison (Table 1, entry 6). We also investigated the formation of CC3-R/CC15-S, and CC3-R/CC19-S, with crystallization of CC3-R/CC19-S occurring rapidly using the same method as with CC3-R/CC3-S (Table 1, entry 8). However, for CC3-R/CC15-S, rapid precipitation did not occur under these conditions; this is thought to be due to the methyl groups in CC15, which reduces the favorable interaction between the cage windows that promotes rapid cocrystallization. Slater et al. generated cocrystals of CC3-S/CC15-R from dichloromethane using fivefold more concentrated solutions of the two cages, although small particle sizes were not targeted in that study.<sup>[12]</sup> We found that microcrystalline particles of CC3-R/CC15-S could be prepared at the same dilute concentration by changing the solvent



**Figure 2.** a) Scheme for POC microparticle formation by combining CC3-R with either CC3-S, CC15-S, or CC19-S with the window diameter of each POC highlighted with a red circle. b) Scanning electron micrographs for CC3-R/CC3-S, CC3-R/CC15-S, and CC3-R/CC19-S microparticles. c) Particle size distributions of CC3-R/CC3-S (by DLS in dichloromethane), CC3-R/CC15-S (by SLS in EtOH), and CC3-R/CC19-S (DLS in dichloromethane).

**Table 1.** Summary of the POC microparticle compositions investigated, along with the measured particle sizes and observed particle morphology for each sample.

Entry	Microparticle components	Z-average particle size by DLS [nm]	Particle size range by SEM [nm]	Average particle size by SEM [nm]	Particle morphology by SEM
1	CC3-R/CC3-S	540	200–700	374	Spherical
2	CC3-R/CC3-S	600	300–800	492	Spherical
3	CC3-R/CC3-S	1200	300–1000	594	Spherical
4	CC3-R/CC3-S	720	300–1000	613	Spherical
5	CC3-R/CC3-S	630	200–800	599	Spherical
6	CC3-R/CC3-S	940	200–1500	799	Octahedral
7	CC3-R/CC15-S	17 000 <sup>a)</sup>	8000–31 000	16 337	Octahedral
8	CC3-R/CC19-S	300	200–800	478	Octahedral

<sup>a)</sup> Particle size determined with static light scattering (SLS) measurements.

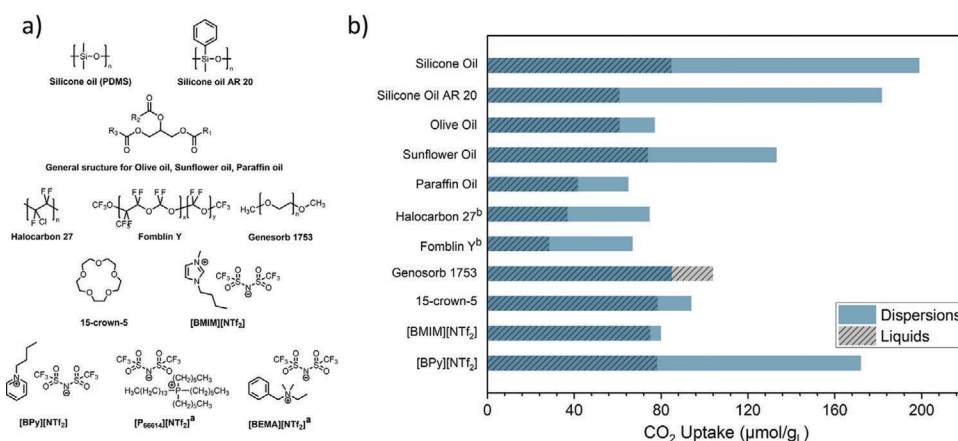
to a 1:1 mixture of dichloromethane:methanol (Table 1, entry 7). For full details of the formation and characterization of the different batches of CC3-R/CC3-S, CC3-R/CC15-S, and CC3-R/CC19-S microparticles (Table 1), please see Section S2 in the Supporting Information.

Analysis of these cocrystals confirmed they were the desired microparticles (Table 1), as determined by dynamic light scattering (DLS) measurements and scanning electron microscopy (SEM), with diameters for CC3-R/CC3-S and CC3-R/CC19-S ranging from ≈200 nm to 1.5 μm (Figure 2b,c). The CC3-R/CC15-S particle size could not be measured using the solvents that we deployed for DLS because sedimentation occurred in MeOH, while the cocrystal dissolved in dichloromethane. However, by using static light scattering (SLS) measurements, these particles were found to be significantly larger (17 μm). This reflects the slower precipitation of these particles, which gives them the chance to increase in size. SEM also gave an indication of the particle morphology; when CC3-R/CC3-S was isolated after rapid mixing, the microparticles were spherical, whereas on slow addition, octahedral

particles were formed (Figure S4, Supporting Information). For CC3-R/CC19-S, SEM images also confirmed an octahedral morphology when using the rapid addition technique, whereas for CC3-R/CC15-S, larger blocks of crystals with blended octahedral morphology were apparent, which corresponded to the larger particles observed by SLS (Figure S8, Supporting Information). While the particle size ranges were broadly comparable for the CC3-R/CC3-S particles formed using the same method (Table 1, entries 1–5), due to the variance in average particle sizes as measured by DLS, the average sizes were also calculated from the acquired SEM images, resulting in much more comparable sizes across the different batches (Table 1)—the variance between the two average particle sizes for each batch of microparticles is likely due to the z-average by DLS being the intensity weighted harmonic mean size while the average by SEM is the arithmetic mean. Analysis by powder x-ray diffraction (PXRD) showed results that were consistent with previous studies; the cocrystals had powder patterns that were similar to CC3α, but with peak shifts that reflected smaller unit cell sizes (Figures S3 and S7, Supporting Information). This indicates that all cocrystals adopt a window-to-window arrangement, with tighter packing due to the use of opposite cage enantiomers.

## 2.2. Screening for Type III Porous Liquids Using POC Microparticles

We first carried out a screen for suitable size-excluded liquids to construct Type III porous liquids using the CC3-R/CC3-S microparticles (Table 1, entries 3 and 5). A range of liquids was selected, including different oils and ionic liquids (Figure 3a), some of which were reported previously to be size-excluded for MOFs and zeolites, such as silicone oil, pyridinium bis(trifluoromethanesulfonyl)imide ([BPy][NTf<sub>2</sub>]), and trihexyltetradecylphosphonium bis(trifluoromethanesulfonyl)imide ([P<sub>66614</sub>][NTf<sub>2</sub>]).<sup>3,6,7</sup> A 12.5 wt% dispersion of CC3-R/CC3-S microparticles was prepared in these liquids by mixing and sonication. For oils that were more viscous, such as Halocarbon



**Figure 3.** a) Bulky liquids screened in this study for the formation of Type III porous liquids with CC3-R/CC3-S POC microparticles. b) Comparison of the CO<sub>2</sub> uptakes for the neat bulky liquids and corresponding 12.5 wt% dispersions of CC3-R/CC3-S. <sup>a)</sup> Ionic liquids screened with homochiral cage CC3 instead of CC3-R/CC3-S POC microparticles to determine if homochiral POCs can be used as a rapid indicator of size-exclusivity and porosity. <sup>b)</sup> A 5 wt% dispersion was studied to reduce viscosity.

27 and Fomblin Y, 5 wt% loadings were prepared to obtain a flowable dispersion. These dispersions were then screened for porosity to allow comparison of the gas uptakes with the neat, unadulterated liquids. Previously, analytical techniques such as IR and NMR spectroscopy, or gas displacement measurements, were used to probe the gas solubility in POC-derived Type II porous liquids.<sup>[13–15,18]</sup> Here, the use of bulky liquids with high boiling points and low vapor pressures allowed us to study the gas solubility using more standard sorption measurements that are used for porous solids. Specifically, the use of a Quantachrome Nova instrument, benchmarked to known uptakes in liquids and with stirring to ensure that the liquids reached saturation uptake, allowed different gases to be screened and adsorption isotherms to be obtained (see Section S1 in the Supporting Information for more information).

Comparison of the CO<sub>2</sub> uptakes at 1 bar in the neat liquids and the corresponding CC3-R/CC3-S dispersions (Table S3 and Figure S11a–k, Supporting Information) indicated that most of the dispersions showed a marked improvement in CO<sub>2</sub> solubility over the neat liquid (Figure 3b). An exception was a 12.5 wt% dispersion in Genosorb 1753, where the neat liquid absorbed more CO<sub>2</sub> than the dispersion, suggesting perhaps that the liquid molecules can occupy the pores in the CC3-R/CC3-S microparticles. It should also be noted that Genosorb liquids are designed specifically to absorb CO<sub>2</sub>, and hence have a relatively high native CO<sub>2</sub> solubility in any case. However, the use of a Type III porous liquid which incorporates permanent micropores with tunable properties may offer certain advantages over conventional liquid sorbents, including Genosorb 1753, which is used to remove CO<sub>2</sub> from natural gas (CH<sub>4</sub>) and syngas (CO/H<sub>2</sub>),<sup>[32]</sup> by exceeding both uptake and selectivity while requiring less energy for regeneration, offering a potentially more efficient alternative for carbon capture. The most marked increases in CO<sub>2</sub> solubility were found with silicone oil (5 cSt), silicone oil AR 20, and [BPy][NTf<sub>2</sub>]. Comparison of the measured uptakes with theoretical maximum uptakes for each dispersion, calculated by taking into account the proportions of the cage solid and the liquid (Table S4, Supporting Information), found that 59–66% of the overall theoretical maximum porosity was obtained for these three liquids. The 5 wt% CC3-R/CC3-S dispersions in Halocarbon 27 and Fomblin Y also displayed 57–60% of the maximum theoretical porosity, but higher solid loadings led to highly viscous systems that did not flow. The remaining dispersions ranged from 25% to 46% of the maximum theoretical porosity. These calculations took into account both the contribution from the porous solid and the solubility of CO<sub>2</sub> in the neat liquid, but on calculating the percentage porosity maintained on the microparticles being dispersed (Table S4, Supporting Information), this ranged from 0% in Genosorb 1753 up to 42% in [BPy][NTf<sub>2</sub>], Halocarbon 27 and Fomblin Y, and 54% in silicone oil (5 cSt) and silicone oil AR 20. This suggests that some porosity is lost on moving from the solid to dispersed state, even for the systems which demonstrate the highest CO<sub>2</sub> uptakes. The observed reductions in uptake are likely due to a number of factors, including the potential loss of some extrinsic porosity that may be present between the POC microparticles in the solid state when CC3-R/CC3-S is dispersed, differing degrees of preferential binding or interpenetration of the liquid used to form the

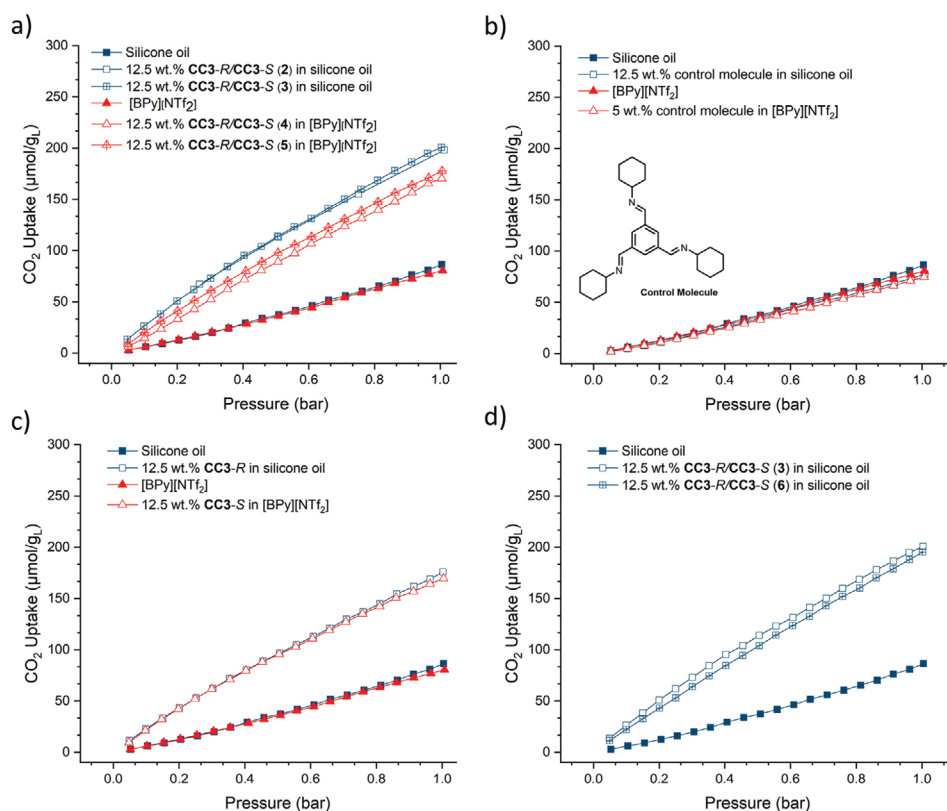
dispersion indicating that not all of the selected liquids are excluded from the porous network and are instead occupying the cavities within the POC microparticles, and the potential viscosities of each liquid and dispersion affecting the diffusion of gas through the system. However, the increased uptakes indicated the successful formation of some Type III porous liquids. Additionally, all of these dispersions were visually stable after standing at room temperature for at least 24 h.

Interestingly, the use of the ionic liquid [BMIM][NTf<sub>2</sub>] did not lead to an improved gas uptake, whereas the use of [BPy][NTf<sub>2</sub>] did—this suggests that the anion is size-excluded but the cation is having an effect on the gas uptake capability. In particular, it initially suggested that the 5-membered imidazolium ring in the [BMIM] cation is not size-excluded, whereas the 6-membered pyridyl ring in the [BPy] cation is, even with the same *n*butyl appended chains. Therefore, to investigate the relationship between the structure of the ionic liquid (in terms of the effect of the cation) and the overall gas uptake performance further, a series of additional ionic liquids were also investigated (Figure S11n, Supporting Information). On addition of a *para*-methyl group to [BPy][NTf<sub>2</sub>] to form the 6-membered analogue of [BMIM][NTf<sub>2</sub>] (i.e., [MBPy][NTf<sub>2</sub>]) the porosity was shut-off (Figure S11n(b), Supporting Information), whereas when this methyl is replaced with a bulky *t*butyl group ([TBPy][NTf<sub>2</sub>]; Figure S11n(c), Supporting Information), the porosity is restored. Additionally, on replacing the methyl group in [BMIM][NTf<sub>2</sub>] with another *n*butyl group ([BBIM][NTf<sub>2</sub>]; Figure S11n(d), Supporting Information), the resulting dispersion was also partially porous. This suggests that it is not solely due to the size of the ring system in the ionic liquid cation that certain combinations exhibit low or zero improvement in gas uptake, but instead is primarily due to the functionality on the rings, with the methyl groups potentially having a preference to penetrate and occupy the outer cage windows and thus block the porosity of the POC microparticles.

In order to gain further understanding of these systems, the CC3-R/CC3-S dispersions in both silicone oil and [BPy][NTf<sub>2</sub>] were selected for further studies. These 12.5 wt% dispersions illustrated the greatest enhancement in CO<sub>2</sub> uptake over the neat liquids, for example, in silicone oil the dispersion illustrated an approximately twofold increase in capacity (198 ± 3.8 μmol g<sub>L</sub><sup>-1</sup> vs 84.8 ± 2.4 μmol g<sub>L</sub><sup>-1</sup> for pure silicone oil). A similar increase was also observed for the dispersion in [BPy][NTf<sub>2</sub>] (172.1 ± 3.6 μmol g<sub>L</sub><sup>-1</sup> vs 78.3 ± 4.5 μmol g<sub>L</sub><sup>-1</sup> for pure [BPy][NTf<sub>2</sub>]). In addition, both of these Type III porous liquids exceeded the CO<sub>2</sub> uptake in Genosorb 1753 at 1 bar (198 μmol g<sub>L</sub><sup>-1</sup> and 172.1 μmol g<sub>L</sub><sup>-1</sup> vs 104 μmol g<sub>L</sub><sup>-1</sup>), and also had the lowest viscosities in each class out of the oils and ionic liquids, allowing higher weight loadings to be studied.

### 2.3. Comparison of Type III POC Microparticle Porous Liquids with Control Liquids

In our preliminary screen, CC3-R/CC3-S dispersions led to similar gas uptake improvements in different liquids (e.g., silicone oil vs [BPy][NTf<sub>2</sub>]; Figure 4a), indicating that the enhanced gas uptake was related to the CC3-R/CC3-S microparticles introducing extra pore volume into the liquid. To further



**Figure 4.** a) CO<sub>2</sub> adsorption isotherms of 12.5 wt% dispersions of CC3-R/CC3-S with varying particle sizes in silicone oil and [BPy][NTf<sub>2</sub>]. b) CO<sub>2</sub> adsorption isotherms of nonporous control molecule dispersions in silicone oil and [BPy][NTf<sub>2</sub>], with inset showing the structure of the control molecule. c) CO<sub>2</sub> adsorption isotherms of homochiral cage CC3 dispersions in silicone oil and [BPy][NTf<sub>2</sub>]. d) CO<sub>2</sub> adsorption isotherms of 12.5 wt% dispersions of CC3-R/CC3-S with different particle morphologies in silicone oil. The numbers in brackets indicate the batch of microparticles used from Table 1.

confirm that the observed improvement in CO<sub>2</sub> uptake was due to the internal pore space of the CC3-R/CC3-S microparticles, rather than, say, surface adsorption on the microparticles, a control molecule that mimics a fragment of the CC3 cage was synthesized (Figure 4b inset).<sup>[14]</sup> This nonporous control molecule does not dissolve in silicone oil or [BPy][NTf<sub>2</sub>], and thus can also be dispersed, allowing us to evaluate the effect of surface adsorption on the particles in the system. CO<sub>2</sub> isotherms were measured on dispersions of the control molecule in both silicone oil and [BPy][NTf<sub>2</sub>] for direct comparison with the CC3-R/CC3-S dispersions. No improvement in CO<sub>2</sub> uptake was observed in either system, confirming that surface adsorption does not contribute measurably to the observed improvements in gas uptake (Figure 4b). Additionally, in contrast to the microparticle dispersions in the same liquids, dispersions of the control molecule were not stable, with sedimentation rapidly occurring (<1 h) on standing.

Dispersions of homochiral cage CC3-R in silicone oil, and CC3-S in [BPy][NTf<sub>2</sub>], were also investigated for their gas solubility—while similar gas uptakes were obtained compared to the CC3-R/CC3-S microparticle systems (Figure 4c), the homochiral dispersions were not as colloiddally stable, with sedimentation occurring when stirring was not maintained at room temperature. However, this further confirms that these liquids are size-excluded from CC3, and that homochiral cages could be used to rapidly determine whether new liquids are size-excluded prior to

forming POC microparticles with controlled particle sizes and narrow distributions. For example, measurement of the CO<sub>2</sub> uptake in dispersions of CC3-S in [P<sub>66614</sub>][NTf<sub>2</sub>] and benzyl(ethyl) dimethylammonium bis(trifluoromethanesulfonyl)imide ([BEMA][NTf<sub>2</sub>]) demonstrated that these two ionic liquids were size-excluded (Figure 3a; Figure S11,m, Supporting Information).

Finally, different batches of CC3-R/CC3-S microparticles were investigated to evaluate the effects of particle size and surface morphology on the gas uptake. Dispersions of CC3-R/CC3-S with particle sizes of 600 and 720 nm (Table 1, entries 2 and 4) show similar CO<sub>2</sub> solubility compared to dispersions with particle sizes of 1200 and 630 nm (Table 1, entries 3 and 5), in silicone oil and [BPy][NTf<sub>2</sub>], respectively (Figure 4a). When CC3-R/CC3-S with an octahedral morphology (Table 1, entry 6) was used, its dispersion presented similar gas uptakes to spherical CC3-R/CC3-S microparticles (Figure 4d). Overall, this shows that the particle size and morphology has little if any effect on the gas uptake of the dispersions, but that controlling the particle size is important for ensuring good, long-term dispersion stability.

#### 2.4. Gas Uptake, Properties, and Stability of Type III CC3-R/CC3-S Microparticle Porous Liquids

To be put into practical use, porous liquids should offer favorable characteristics, such as being recyclable and thermally

**Table 2.** Summary of viscosity measurements on the liquids and different concentration dispersions investigated.

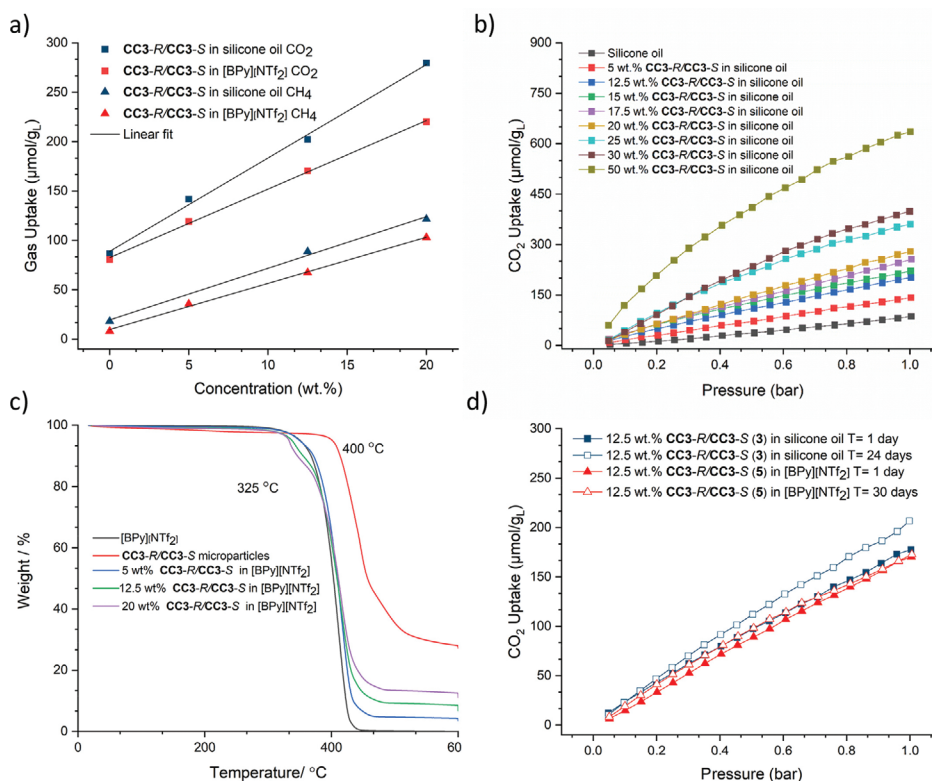
Liquids	Viscosity [mPa s]			
	0 wt% dispersion	5 wt% dispersion	12.5 wt% dispersion	20 wt% dispersion
Silicone oil	5.01 ± 0.01	7.41 ± 0.09	13.22 ± 0.55	19.09 ± 0.60
[BPy][NTf <sub>2</sub> ]	67.34 ± 0.06	86.18 ± 0.16	142.2 ± 0.14	303.5 ± 0.29

stable. We selected two systems from the initial screen—CC3-R/CC3-S in silicone oil and in [BPy][NTf<sub>2</sub>]<sup>+</sup>—to more fully study their physicochemical properties. Different loadings of CC3-R/CC3-S microparticles (5, 12.5, and 20 wt%) were prepared in both liquids to evaluate the effect of microparticle concentration on both the gas uptake and the dispersion viscosity.

The viscosity of the dispersions in [BPy][NTf<sub>2</sub>]<sup>+</sup> was higher than those in silicone oil (Table 2), which is due to the general viscous nature of ionic liquids.<sup>[28]</sup> However pure [BPy][NTf<sub>2</sub>]<sup>+</sup> has a viscosity of 67.34 mPa s, which is substantially lower than the viscosity of other size-excluded ionic liquids; for example, [P<sub>66614</sub>][NTf<sub>2</sub>]<sup>+</sup> and [BEMA][NTf<sub>2</sub>]<sup>+</sup> have viscosities of 395.7 and 496.6 mPa s, respectively. Viscosity studies on the 5, 12.5, and 20 wt% dispersions of CC3-R/CC3-S microparticles in [BPy][NTf<sub>2</sub>]<sup>+</sup> and in silicone oil showed that the viscosity of the dispersions increased with increasing particle concentration (Table 2). However, while the ionic liquid dispersions had relatively higher viscosities, which could slow down gas adsorption

kinetics, the saturated gas solubilities appeared to be unaffected: both CO<sub>2</sub> and CH<sub>4</sub> uptakes increased linearly with increasing CC3-R/CC3-S content from 0 to 20 wt% for both the [BPy][NTf<sub>2</sub>]<sup>+</sup> and silicone oil-based dispersions (Figure 5a; Figures S12–S15, Supporting Information). At 20 wt%, the uptakes were fairly comparable across both systems, although the overall uptakes for both gases were slightly higher in the silicone oil dispersion (CO<sub>2</sub>: 280 μmol g<sup>-1</sup> vs 210 μmol g<sup>-1</sup>; CH<sub>4</sub>: 121 μmol g<sup>-1</sup> vs 103 μmol g<sup>-1</sup>). Dispersions of CC3-R/CC3-S in silicone oil were also investigated up to 50 wt%, with a linear trend for CO<sub>2</sub> adsorption again being observed up to 635 μmol g<sup>-1</sup> at 50 wt% (Figure 5b; Figure S16, Supporting Information). Interestingly, on calculating the CO<sub>2</sub>/CH<sub>4</sub> selectivity based on the measured uptakes at 1 bar for each of the different wt% dispersions in both silicone oil and [BPy][NTf<sub>2</sub>]<sup>+</sup> (Table S7, Supporting Information), the selectivity increased at higher loadings of CC3-R/CC3-S in silicone oil, but decreased at higher loadings when dispersed in [BPy][NTf<sub>2</sub>]<sup>+</sup>. This indicates that the choice of size-excluded liquid could not only have an impact on the overall gas uptake in a system, but could also lead to improved gas selectivity in a Type III porous liquid containing the same pore carrier.

The dispersion in [BPy][NTf<sub>2</sub>]<sup>+</sup> demonstrates good thermal stability according to thermal gravimetric analysis (TGA; Figure 5c), with no significant mass loss observed until 325 °C. However, the low viscosity silicone oil used in these studies (PDMS, 5 mPa s) has a significantly higher vapor pressure (<700 Pa at 298.15 K)<sup>[33]</sup> compared to [BPy][NTf<sub>2</sub>]<sup>+</sup> (0.0519 Pa at



**Figure 5.** a) Measured gas uptakes for a range of CC3-R/CC3-S dispersions at different concentrations (1 bar, 298–300 K). b) CO<sub>2</sub> adsorption isotherms of silicone oil and the corresponding 5, 12.5, 15, 17.5, 20, 30, and 50 wt% dispersions of CC3-R/CC3-S microparticles. c) TGA of [BPy][NTf<sub>2</sub>]<sup>+</sup> and 5, 12.5, and 20 wt% CC3-R/CC3-S dispersions in [BPy][NTf<sub>2</sub>]<sup>+</sup>. d) CO<sub>2</sub> adsorption study of aged dispersion samples.



498.07 K),<sup>[34]</sup> and the corresponding porous liquid therefore demonstrates a much lower thermal stability, with mass loss occurring from 80 °C based on TGA analysis (Figure S17a, Supporting Information). This does not cause significant problems for the gas sorption measurements carried out here, with the degassing and uptake measurements performed immediately after sample preparation at ambient temperature, but this increased volatility might account for the slightly elevated CO<sub>2</sub> adsorption isotherm of a 12.5 wt% silicone oil dispersion after ≈1 month compared to after 1 day, possibly due to slow evaporation of the liquid at ambient temperature over time (Figure S17b, Supporting Information) and increased evaporation when put under repeated dynamic vacuum to degas the sample prior to gas uptake measurements (overall 6% mass loss observed after 2 degas cycles), making the porous liquid more concentrated, while the gas uptake remained nearly unchanged for the aged [BPy][NTf<sub>2</sub>] sample (Figure 5d).

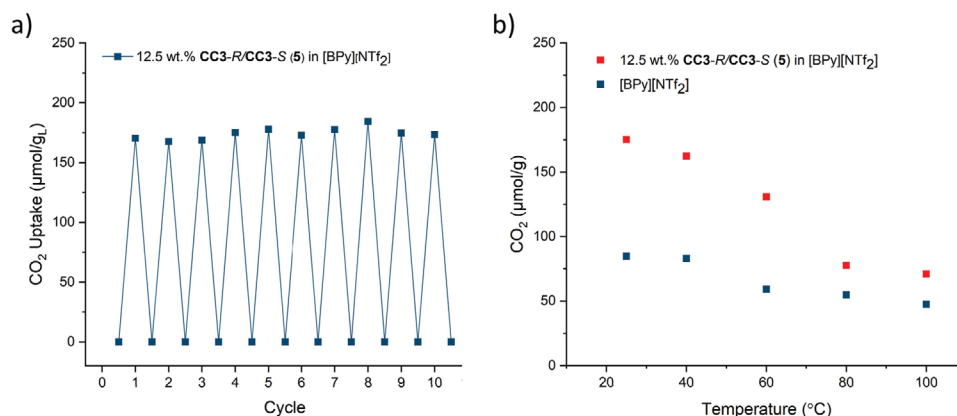
For these two systems, the viscosity and thermal stability are essentially trade-offs; ideally, one would like a high thermal stability liquid with low viscosity to allow the broadest range of potential applications. We note that the thermal stability limit is imposed by the liquid, rather than the POC.

While there is a trade-off between the viscosity and thermal stability for dispersions in silicone oil (PDMS, 5 mPa s) versus [BPy][NTf<sub>2</sub>], the other physical properties for these two porous liquids are broadly similar. Both systems retain their porosity over time (Figure 5d). To evaluate the recyclability of these liquids as gas sorbents, the CO<sub>2</sub> uptake was measured over 10 cycles in a 12.5 wt% dispersion of CC3-R/CC3-S in [BPy][NTf<sub>2</sub>] (Figure 6a; Figure S18, Supporting Information), with reactivation by vacuum degassing for 100 min carried out between measurements. The CO<sub>2</sub> uptakes were reproducible, with no loss in uptake apparent over repeated cycles (average CO<sub>2</sub> uptake over 10 cycles: 174.3 ± 4.6 μmol g<sup>-1</sup>). Due to the slight volatility of the silicone oil, the corresponding 12.5 wt% dispersion was only cycled twice, but again, the CO<sub>2</sub> solubility was completely reproducible over two cycles (Figure S19, Supporting Information).

The cage microparticle and phase stability of both dispersions was also investigated in more detail. The crystallinity of

the CC3-R/CC3-S microparticles was retained upon formation of dispersions in silicone oil and [BPy][NTf<sub>2</sub>] (Figures S20 and S21, Supporting Information), and no phase change was observed postsorption (Figure S22, Supporting Information). In addition, the morphology and average size of the particles was maintained postdispersion as confirmed by SEM of recovered CC3-R/CC3-S particles from the [BPy][NTf<sub>2</sub>] dispersion by filtration (Figure S23, Supporting Information). Both the silicone oil and [BPy][NTf<sub>2</sub>] dispersions of CC3-R/CC3-S microparticles appeared to be visually stable over extended periods. To quantify the dispersion stability, a LUMiSizer was used to accelerate the creaming or sedimentation process for 12.5 wt% dispersions of CC3-R/CC3-S (particle size: 500–700 nm) in both silicone oil and [BPy][NTf<sub>2</sub>].<sup>[35]</sup> According to Stoke's law, particle size plays an important role in determining the ratio of gravitational to Brownian forces applied to the particles in a dispersion, and hence reducing the particle size can help to control the sedimentation velocity and thus stabilize multiphase systems.<sup>[36]</sup> At a relative centrifugal force of 532 × g, the CC3-R/CC3-S microparticles separate by creaming to the surface of the silicone oil at a sedimentation velocity of 0.24 mm day<sup>-1</sup>, indicating that the dispersion is stable (Table S8 and Figure S24, Supporting Information). The [BPy][NTf<sub>2</sub>] system also showed creaming but had a reduced sedimentation velocity of 0.04 mm day<sup>-1</sup>, indicating that this dispersion has improved stability over the one in silicone oil (Table S8 and Figures S25 and S26, Supporting Information). This could be due to the increased viscosity of the ionic liquid, or potentially because the cation and anion of [BPy][NTf<sub>2</sub>] provide electrostatic stabilization for the POC particles.<sup>[37]</sup>

Finally, the increased thermal stability of [BPy][NTf<sub>2</sub>] allowed for further adsorption studies to be carried out at higher temperatures to investigate the potential of a temperature-swing process for gas uptake and release. The CO<sub>2</sub> uptakes of neat [BPy][NTf<sub>2</sub>] and its corresponding 12.5 wt% CC3-R/CC3-S dispersion were measured at a range of temperatures: 25, 40, 60, 80, and 100 °C (Figure 6b; Figures S27 and S28, Supporting Information). Both systems showed measurable uptakes, even when the temperature was increased to 100 °C, which might arise from the strong interaction of CO<sub>2</sub> with the ionic liquid, but the working absorption capacity of CC3-R/CC3-S



**Figure 6.** a) Recycling study for CO<sub>2</sub> uptake (1 bar, 298–300 K) in a 12.5 wt% dispersion of CC3-R/CC3-S in [BPy][NTf<sub>2</sub>]. b) CO<sub>2</sub> solubilities in [BPy][NTf<sub>2</sub>] and the corresponding 12.5 wt% CC3-R/CC3-S dispersion at 25, 40, 60, 80, and 100 °C. The Type III porous liquid has a higher working capacity (104 μmol g<sup>-1</sup>) than the neat ionic liquid (37 μmol g<sup>-1</sup>) between 25 and 100 °C.

in [BPy][NTf<sub>2</sub>] was significantly larger than for neat [BPy][NTf<sub>2</sub>] (104.30 μmol g<sub>L</sub><sup>-1</sup> vs 37.27 μmol g<sub>L</sub><sup>-1</sup> between 25 and 100 °C). The increased working capacity in the porous liquid is due to the presence of the POC microparticles, with the significant capacity decrease at higher temperatures predominantly dictated by the gas uptake behavior in the dispersed microporous material, whereas the difference in capacity at increasing temperatures in the ionic liquid is much smaller and likely due to the strong interaction between the CO<sub>2</sub> molecules and the anion, and the absence of permanent intrinsic porosity with CO<sub>2</sub> absorption occurring in the extrinsic “free space” that is available.

### 2.5. Modification of Porosity in Type III PLs by the Modular Assembly of Two POCs into Microparticles

Controlling POC pore structure to influence the gas uptake has been studied widely in the solid state, for example by introducing bulky groups to decrease the window or cavity sizes, as applied in Xe or hydrogen isotope separation.<sup>[12,31]</sup> Likewise, specific functional groups have been introduced into POCs to modulate the chemical environment for CO<sub>2</sub>/CH<sub>4</sub> separations.<sup>[27]</sup> This concept was also recently translated to the liquid state in Type II porous liquids with POCs that dissolve in bulky organic solvents.<sup>[18]</sup> Here, we sought to transfer this concept to Type III porous liquids by using quasiracemic cocrystals.

Two systems were studied: CC3-*R*/CC15-*S* and CC3-*R*/CC19-*S* (Figure 2a). CC15 is isostructural with CC3, but methyl groups occupy the four windows and reduce the window size to 2.0 Å (from 4 Å in CC3).<sup>[18]</sup> The reduced cage window size in CC15 affects gas sorption in the internal cavities and in the portal sites, with the methyl groups both excluding larger guests and interfering with electrostatic interactions of guest molecules with the imine bonds. For example, CO<sub>2</sub> and Xe uptakes in CC15 were approximately half as large as for CC3 in the solid state,<sup>[12]</sup> and this occlusion was even more pronounced in a CC15-derived Type II porous liquid, where the Xe uptake was almost shut off entirely.<sup>[18]</sup> CC19 has an analogous molecular structure, but with one hydroxyl group attached to each of its four aromatic rings. The incorporation of these polar hydroxyl groups has been reported to improve the CO<sub>2</sub> uptake in the solid state compared to CC3, while lowering CH<sub>4</sub> uptakes.<sup>[27]</sup> Hence, by incorporating either CC15 or CC19 into the microparticles, there is an opportunity to tune the gas selectivity in the corresponding Type III porous liquids.

Previously, we have demonstrated that single molecule analysis of POCs gives an indication of the overall properties of the solid-state material.<sup>[38]</sup> On comparison of the window and cavity sizes of CC3-*S*, CC15-*S*, and CC19-*S* (Figure S32, Supporting Information), it can be observed that the presence of the methyl groups in CC15 significantly impacts the window size compared to CC3, while the hydroxy-functionality in CC19 does not have the same impact (Figure 2a). Additionally, because all three systems pack in the CC3α arrangement based on the measured PXRD patterns, where window-to-window interactions dominate the pore network, the decreased window size in CC15 leads to an expected decrease in porosity for CC3/CC15 compared to CC3 (Figure S33, Supporting Information). This corresponds

with the observed CO<sub>2</sub> and CH<sub>4</sub> uptake in the solid microparticles (Figures S5, S9, and S10, Supporting Information), with the uptake of both gases being reduced in CC3-*R*/CC15-*S* compared to that in CC3-*R*/CC3-*S* due to the restricted window portals, and an increased CO<sub>2</sub> uptake and a comparable CH<sub>4</sub> uptake observed in CC3-*R*/CC19-*S* compared to CC3-*R*/CC3-*S* due to the similar interconnected pore network and incorporation of polar hydroxy functionality in the structure.

As might be expected, both the CO<sub>2</sub> and CH<sub>4</sub> uptakes in the 12.5 wt% dispersion of CC3-*R*/CC15-*S* in silicone oil were significantly reduced compared to the CC3-*R*/CC3-*S* liquid (CO<sub>2</sub>: 68.5 ± 1.3 μmol g<sub>L</sub><sup>-1</sup> vs 198.0 ± 3.8 μmol g<sub>L</sub><sup>-1</sup>; CH<sub>4</sub>: 30.2 ± 0.3 μmol g<sub>L</sub><sup>-1</sup> vs 105.2 ± 16.5 μmol g<sub>L</sub><sup>-1</sup>). By contrast, a slightly higher CO<sub>2</sub> uptake, and broadly comparable CH<sub>4</sub> uptake, were observed in the 12.5 wt% dispersion of CC3-*R*/CC19-*S* in silicone oil (CO<sub>2</sub>: 210.3 ± 3.3 μmol g<sub>L</sub><sup>-1</sup>; CH<sub>4</sub>: 117.4 ± 21.4 μmol g<sub>L</sub><sup>-1</sup>) (Figures S29–S31, Supporting Information). The reduction in porosity upon introducing CC15-*S* is likely due to the methyl groups blocking the cage windows. While incorporation of CC19 into the microparticles improved gas uptakes in the Type III porous liquids, no significant enhancement in CO<sub>2</sub>/CH<sub>4</sub> selectivity was observed (Table S7, Supporting Information), unlike in the solid state for pure CC19. This is probably due to its homogeneous distribution in the CC3-*R*/CC19-*S* microparticles, because previous studies<sup>[27]</sup> found that CO<sub>2</sub>/CH<sub>4</sub> selectivity only occurred when CC19 was included in a core-shell morphology. The use of such core-shell cocrystals to form selective Type III porous liquids is a subject for future studies.

## 3. Conclusions

We present here the first Type III porous liquids based on the dispersion of POC microparticles prepared by chiral recognition between chirally opposed isostructural cages. A selection of size-excluded carrier liquids was investigated, including oils and ionic liquids, which can be used to tailor the porous liquids to meet application requirements. We were able to access liquids with relatively low viscosity, high thermal stability, or good recyclability, while maintaining properties such as gas solubility and colloidal stability. These liquids also demonstrate increased gas uptakes compared to previously reported Type II porous liquids based on dissolving POCs in size-excluded solvents, predominantly due to the much higher accessible loadings. A 12.5 wt% CC3-*R*/CC3-*S* dispersion in silicone oil or [BPy][NTf<sub>2</sub>] showed an approximately twofold increase in CO<sub>2</sub> uptake over the neat silicone oil and [BPy][NTf<sub>2</sub>], and up to eight times the CH<sub>4</sub> uptake. These POC microparticle dispersions maintained their gas adsorption properties for at least one month and showed good colloidal stability. CC3-*R*/CC3-*S* dispersions in [BPy][NTf<sub>2</sub>] suffered no mass loss until 325 °C, and demonstrated a working capacity of 104.30 μmol g<sub>L</sub><sup>-1</sup> by temperature swing (25–100 °C). These liquids were recyclable over at least ten CO<sub>2</sub> adsorption/desorption cycles. The incorporation of CC15 into the microparticles reduced the CO<sub>2</sub> and CH<sub>4</sub> uptakes significantly, while inclusion of CC19 increased the uptake of these gases, demonstrating the feasibility of fine-tuning gas sorption properties in Type III porous liquids by changing the

microparticle components. Other potential applications for the future include the separation of volatile organic compounds or chiral separations using POC porous liquids.<sup>[39,40]</sup>

## Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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## Conflict of Interest

The authors declare no conflict of interest.

## Author Contributions

A.K., B.D.E., and R.L.G. carried out the cage synthesis, characterization, thermal studies, and property investigation of the cage microparticles and Type III porous liquids. A.T. and K.E.J. carried out the computational porosity analysis. A.I.C. and R.L.G. conceived the project, and the paper was written by A.K., B.D.E., and R.L.G. with input from all authors.

## Data Availability Statement

The data that supports the findings of this study are available in the supplementary material of this article.

## Keywords

gas uptake, porosity, porous liquids, porous organic cages

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[1] N. O'Reilly, N. Giri, S. L. James, *Chem. - Eur. J.* **2007**, *13*, 3020.

[2] A. Bavykina, A. Cadiou, J. Gascon, *Coord. Chem. Rev.* **2019**, *386*, 85.

[3] J. Cahir, M. Y. Tsang, B. Lai, D. Hughes, M. A. Alam, J. Jacquemin, D. Rooney, S. L. James, *Chem. Sci.* **2020**, *11*, 2077.

[4] H. Li, B. Liu, M. Yang, D. Zhu, Z. Huang, W. Chen, L. Yang, G. Chen, *Ind. Eng. Chem. Res.* **2020**, *59*, 6154.

[5] W. Shan, P. F. Fulvio, L. Kong, J. A. Schott, C. L. Do-Thanh, T. Tian, X. Hu, S. M. Mahurin, H. Xing, S. Dai, *ACS Appl. Mater. Interfaces* **2018**, *10*, 32.

[6] M. Costa Gomes, L. Pison, C. Červinka, A. Padua, *Angew. Chem., Int. Ed.* **2018**, *57*, 11909.

[7] S. Liu, J. Liu, X. Hou, T. Xu, J. Tong, J. Zhang, B. Ye, B. Liu, *Langmuir* **2018**, *34*, 3654.

[8] H. Liu, B. Liu, L. C. Lin, G. Chen, Y. Wu, J. Wang, X. Gao, Y. Lv, Y. Pan, X. Zhang, X. Zhang, L. Yang, C. Sun, B. Smit, W. Wang, *Nat. Commun.* **2014**, *5*, 5147.

[9] T. Tozawa, J. T. A. Jones, S. I. Swamy, S. Jiang, D. J. Adams, S. Shakespeare, R. Clowes, D. Bradshaw, T. Hasell, S. Y. Chong, C. Tang, S. Thompson, J. Parker, A. Trewin, J. Bacsa, A. M. Z. Slawin, A. Steiner, A. I. Cooper, *Nat. Mater.* **2009**, *8*, 973.

[10] F. Beuerle, B. Gole, *Angew. Chem., Int. Ed.* **2018**, *57*, 4850.

[11] M. Mastalerz, *Acc. Chem. Res.* **2018**, *51*, 2411.

[12] A. G. Slater, P. S. Reiss, A. Pulido, M. A. Little, D. L. Holden, L. Chen, S. Y. Chong, B. M. Alston, R. Clowes, M. Haranczyk, M. E. Briggs, T. Hasell, G. M. Day, A. I. Cooper, *ACS Cent. Sci.* **2017**, *3*, 734.

[13] N. Giri, M. G. Del Pópolo, G. Melaugh, R. L. Greenaway, K. Rätzke, T. Koschine, L. Pison, M. F. C. Gomes, A. I. Cooper, S. L. James, *Nature* **2015**, *527*, 216.

[14] R. L. Greenaway, D. Holden, E. G. B. Eden, A. Stephenson, C. W. Yong, M. J. Bennison, T. Hasell, M. E. Briggs, S. L. James, A. I. Cooper, *Chem. Sci.* **2017**, *8*, 2640.

[15] R. J. Kearsley, B. M. Alston, M. E. Briggs, R. L. Greenaway, A. I. Cooper, *Chem. Sci.* **2019**, *10*, 9454.

[16] N. Giri, C. E. Davidson, G. Melaugh, M. G. Del Pópolo, J. T. A. Jones, T. Hasell, A. I. Cooper, P. N. Horton, M. B. Hursthouse, S. L. James, *Chem. Sci.* **2012**, *3*, 2153.

[17] G. Melaugh, N. Giri, C. E. Davidson, S. L. James, M. G. Del Pópolo, *Phys. Chem. Chem. Phys.* **2014**, *16*, 9422.

[18] B. D. Egleston, K. V. Luzyanin, M. C. Brand, R. Clowes, M. E. Briggs, R. L. Greenaway, A. I. Cooper, *Angew. Chem.* **2020**, *132*, 7432.

[19] T. Hasell, S. Y. Chong, K. E. Jelfs, D. J. Adams, A. I. Cooper, *J. Am. Chem. Soc.* **2012**, *134*, 588.

[20] P. Li, H. Chen, J. A. Schott, B. Li, Y. Zheng, S. M. Mahurin, D. E. Jiang, G. Cui, X. Hu, Y. Wang, L. Li, S. Dai, *Nanoscale* **2019**, *11*, 1515.

[21] S. He, L. Chen, J. Cui, B. Yuan, H. Wang, F. Wang, Y. Yu, Y. Lee, T. Li, *J. Am. Chem. Soc.* **2019**, *141*, 19708.

[22] Y. Liu, Y. Bai, T. Tian, *Materials* **2019**, *12*, 3984.

[23] A. Knebel, A. Bavykina, S. J. Datta, L. Sundermann, L. Garzon-Tovar, Y. Lebedev, S. Durini, R. Ahmad, S. M. Kozlov, G. Shterk, M. Karunakaran, I. D. Carja, D. Simic, I. Weilert, M. Klüppel, U. Giese, L. Cavallo, M. Rueping, M. Eddaoudi, J. Caro, J. Gascon, *Nat. Mater.* **2020**, *19*, 1346.

[24] R. E. Mow, A. S. Lipton, S. Shulda, E. A. Gauding, T. Gennett, W. A. Braunecker, *J. Mater. Chem. A* **2020**, *8*, 23455.

[25] J. Cravillon, S. Münzer, S. J. Lohmeier, A. Feldhoff, K. Huber, M. Wiebcke, *Chem. Mater.* **2009**, *21*, 1410.

[26] M. W. Schneider, L. G. Lechner, M. Mastalerz, *J. Mater. Chem.* **2012**, *22*, 7113.

[27] S. Jiang, Y. Du, M. Marcello, E. W. Corcoran, D. C. Calabro, S. Y. Chong, L. Chen, R. Clowes, T. Hasell, A. I. Cooper, *Angew. Chem., Int. Ed.* **2018**, *57*, 11228.

[28] J. T. A. Jones, T. Hasell, X. Wu, J. Bacsa, K. E. Jelfs, M. Schmidtman, S. Y. Chong, D. J. Adams, A. Trewin, F. Schiffrman, F. Cora, B. Slater, A. Steiner, G. M. Day, A. I. Cooper, *Nature* **2011**, *474*, 367.

[29] T. Hasell, S. Y. Chong, M. Schmidtman, D. J. Adams, A. I. Cooper, *Angew. Chem., Int. Ed.* **2012**, *51*, 7154.

[30] S. Tothadi, M. A. Little, T. Hasell, M. E. Briggs, S. Y. Chong, M. Liu, A. I. Cooper, *CrystEngComm* **2017**, *19*, 4933.

[31] M. Liu, L. Zhang, M. A. Little, V. Kapil, M. Ceriotti, S. Yang, L. Ding, D. L. Holden, R. Balderas-Xicohténcatl, D. He, R. Clowes,

- S. Y. Chong, G. Schütz, L. Chen, M. Hirscher, A. I. Cooper, *Science* **2019**, 366, 613.
- [32] A. V. Rayer, A. Henni, P. Tontiwachwuthikul, *Can. J. Chem. Eng.* **2012**, 90, 576.
- [33] MSDS - 317667, <https://www.sigmaaldrich.com/MSDS/MSDS/DisplayMSDSPage.do?country=GB&language=en&productNumber=317667&brand=ALDRICH&PageToGoToURL=https%3A%2F%2Fwww.sigmaaldrich.com%2Fcatalog%2Fproduct%2Faldrich%2F317667%3Flang%3Den> (accessed: June 2020).
- [34] M. A. A. Rocha, L. M. N. B. F. Santos, *Chem. Phys. Lett.* **2013**, 585, 59.
- [35] D. Lerche, *J. Dispersion Sci. Technol.* **2002**, 23, 699.
- [36] G. T. Hermanson, *Bioconjugate Techniques*, Elsevier, Amsterdam **2013**, pp. 549–587.
- [37] F. Yu, Y. Chen, X. Liang, J. Xu, C. Lee, Q. Liang, P. Tao, T. Deng, *Prog. Nat. Sci.: Mater. Int.* **2017**, 27, 531.
- [38] M. Miklitz, S. Jiang, R. Clowes, M. E. Briggs, A. I. Cooper, K. E. Jelfs, *J. Phys. Chem. C* **2017**, 121, 15211.
- [39] A. Kewley, A. Stephenson, L. Chen, M. E. Briggs, T. Hasell, A. I. Cooper, *Chem. Mater.* **2015**, 27, 3207.
- [40] L. Chen, P. S. Reiss, S. Y. Chong, D. Holden, K. E. Jelfs, T. Hasell, M. A. Little, A. Kewley, M. E. Briggs, A. Stephenson, K. M. Thomas, J. A. Armstrong, J. Bell, J. Busto, R. Noel, J. Liu, D. M. Strachan, P. K. Thallapally, A. I. Cooper, *Nat. Mater.* **2014**, 13, 954.