



PAPER

Spectral features of Pb-related color centers in diamond – a systematic photoluminescence characterization

OPEN ACCESS

RECEIVED
2 February 2021REVISED
15 May 2021ACCEPTED FOR PUBLICATION
20 May 2021PUBLISHED
14 June 2021

Original content from
this work may be used
under the terms of the
[Creative Commons
Attribution 4.0 licence](#).

Any further distribution
of this work must
maintain attribution to
the author(s) and the
title of the work, journal
citation and DOI.



Sviatoslav Ditalia Tchernij^{1,2,3} , Emilio Corte^{1,2}, Tobias Lühmann⁴, Paolo Traina³, Sébastien Pezzagna⁴, Ivo Pietro Degiovanni^{2,3} , Georgios Provatas⁵ , Ekaterina Moreva³, Jan Meijer⁴, Paolo Olivero^{1,2,3} , Marco Genovese^{2,3} and Jacopo Forneris^{1,2,3,*}

¹ Physics Department, University of Torino, Torino 10125, Italy

² Istituto Nazionale di Fisica Nucleare (INFN), sezione di Torino, Torino 10125, Italy

³ Istituto Nazionale di Ricerca Metrologica (INRiM), Torino 10135, Italy

⁴ Applied Quantum Systems, Felix-Bloch Institute for Solid-State Physics, Universität Leipzig, Leipzig 04103, Germany

⁵ Laboratory for Ion Beam Interactions, Ruđer Bošković Institute, Zagreb 10000, Croatia

* Author to whom any correspondence should be addressed.

E-mail: jacopo.forneris@unito.it

Keywords: diamond, ion implantation, single photon emitters, luminescent centers, Pb-related color center

Abstract

We report on the systematic characterization of the optical properties of diamond color centers based on Pb impurities. An ensemble photoluminescence analysis of their spectral emission was performed at different excitation wavelengths in the 405–520 nm range and at different temperatures in the 4–300 K range. The series of observed spectral features consist of different emission lines associated with Pb-related defects. Finally, a room-temperature investigation of single-photon emitters under 490.5 nm laser excitation is reported, revealing different spectral signatures with respect to those already reported under 514 nm excitation. This work represents a substantial progress with respect to previous studies on Pb-related color centers, both in the attribution of an articulated series of spectral features and in the understanding of the formation process of this type of defect, thus clarifying the potential of this system for high-impact applications in quantum technologies.

1. Introduction

Diamond color centers have been extensively studied in the last decades as appealing systems for applications in quantum optics and quantum information processing, ranging from the use as single photon emitters to quantum sensing of magnetic and electric fields and of temperature, pressure and strain [1–7]. Despite the increasing interest of these systems among the scientific community, only a fairly limited number of emitters related to impurities incorporated in the diamond lattice has been demonstrated so far [6, 8, 9], and most of the preliminary practical demonstrations of diamond-based quantum technologies rely on the peculiar spin properties of the negatively-charged nitrogen-vacancy color center (NV) [10–15].

Indeed, even if NV centers present several interesting properties, they are also subject to significant drawbacks, such as broad spectral emission, charge state blinking, relatively low emission rate [16, 17], thus prompting for the search of alternative new centers overcoming this limits and/or presenting complementary properties.

Besides the already consolidated color centers related to group-IV impurities such as SiV, GeV, and SnV complexes [6, 18–27], two independent works have recently demonstrated the creation of optically-active defects in diamond upon Pb ion implantation and subsequent annealing, and have reported their characterization at the single-photon emitter level [28, 29].

The Pb-related defects exhibited appealing emission properties for quantum information processing applications, with room temperature photoluminescence (PL) emission rates exceeding 10^6 photons s^{-1} [28], an emission mainly concentrated in the zero phonon line (ZPL) due to the low phonon coupling and a

large ground-state splitting predicted by numerical simulations [29, 30], with potentially disruptive implications towards the achievement of long spin coherence for quantum memories and optical quantum networks [6].

On the other hand, the currently available reports evidenced the presence of several spectral lines of uncertain attribution. Furthermore, discrepancies in the observation of some emission peaks (namely: 520–525 nm doublet, 538 nm doublet, 715 nm line) emerged and were attributed to different experimental conditions such as excitation wavelength and operating temperatures [28, 29].

We report here on a systematic investigation of Pb-related centers under varying excitation wavelengths in the 405–520 nm range and temperature between 4 K and 300 K. Our results address and discuss the aforementioned discrepancies by providing systematic information on the relative emission intensities of these spectral features as a function of the measurement temperature, thus paving the way to the practical application of these centers. Finally, a comprehensive discussion summarizing the currently available results is presented, including the possible attribution of the observed emission lines and their relation with the theoretical models proposed in the literature.

2. Experimental

The experiments were performed on a single-crystal IIA diamond substrate produced by ElementSix. The sample has a ‘detector grade’ quality (i.e. substitutional N and B concentration: $[N_S] < 5$ ppb, $[B_S] < 5$ ppb). Pb-related centers were created upon the implantation of 35 keV PbO_2^- ions at different fluences, namely $5 \times 10^{11} \text{ cm}^{-2}$ and $2 \times 10^{13} \text{ cm}^{-2}$. The sample underwent a thermal annealing process at 1200 °C for 2 h in high vacuum ($< 5 \times 10^{-6}$ mbar) to promote the formation of the centers, as discussed in a previous report [28]. It is worth remarking that the implantation of PbO_2^- ions was motivated by the higher extraction efficiency and stability with respect to the technically challenging acceleration of a Pb^- beam [31]. At the same time, a control sample was implanted with 9 MeV O^{4+} ions at $4 \times 10^{13} \text{ cm}^{-2}$ fluence and annealed under the same experimental conditions, in order to explicitly assess the contribution of optically-active O-related defects to the PL spectra acquired in the considered wavelength range [28]. After annealing, an oxygen plasma cleaning process was performed to ensure the removal of the surface graphitic phases and optically-active contaminants.

The processed substrate was characterized using different PL single-photon sensitive confocal microscopy setups. A cryogenic setup operating based on a liquid helium closed-cycle optical cryostat [32] (100x air objective, 0.85 NA) operating in the 4–300 K temperature range was employed for the measurements reported in section 3.1. The latter apparatus was not equipped with nano-positioners, thus preventing the investigation of individual emitters at cryogenic temperatures. For this reason, an additional confocal microscope operating at room temperature (100x air objective, 0.95 NA), equipped with closed-loop nano-positioners was used for the identification of individual emitters (measurements reported in section 3.2). In both setups, the optical excitation was provided by a set of CW laser diodes with emission wavelengths of 405 nm, 490.5 nm and 520 nm. The PL spectra were acquired using a single-grating monochromator (1200 grooves mm^{-1} , 600 nm blaze, ~ 4 nm spectral resolution) that was fiber-coupled to a single-photon avalanche detector (SPAD). The room-temperature setup was equipped with a Hanbury-Brown & Twiss (HBT) interferometer, allowing the assessment of single-photon emitters via the measurement of the second-order autocorrelation function. The HBT interferometer was implemented by connecting two independent SPAD to the output of a multimode fiber-integrated 50:50 beam-splitter. A third confocal microscope (60x oil objective 1.35 N.A.) equipped with an SPAD outcoupled in open air was used to characterize the spectral emission from individual spots under 532 nm laser excitation (measurements reported in section 3.3). In this latter setup, the PL spectra were acquired using a Peltier-cooled CCD array camera with a spectral resolution of ~ 2 nm, as determined by the first-order Raman peak (572 nm) FWHM width.

3. Results

3.1. Ensemble-level PL characterization

Following preliminary reports on the formation and the optical properties of Pb-related color centers [28, 29], we performed the first extensive PL characterization of this class of defects at an ensemble level ($2 \times 10^{13} \text{ cm}^{-2}$ implantation fluence) under different excitation wavelengths in a temperature range of 4–300 K. Figure 1 shows the PL spectra acquired at 4 K, 77 K and 300 K under 405 nm (figures 1(a)), 490.5 nm (figures 1(b)) and 520 nm (figure 1(c)) excitations, respectively. The main results of this set of measurements can be summarized as follows.

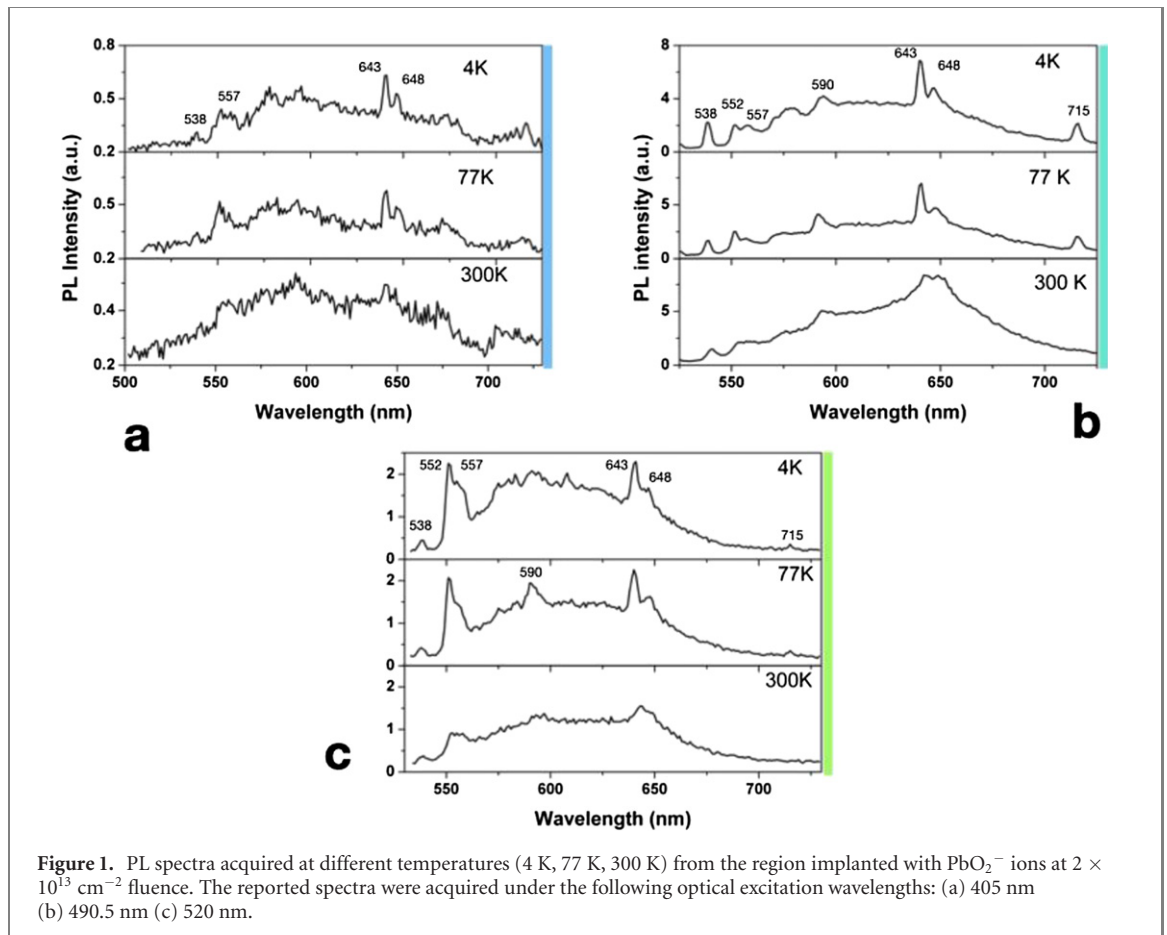


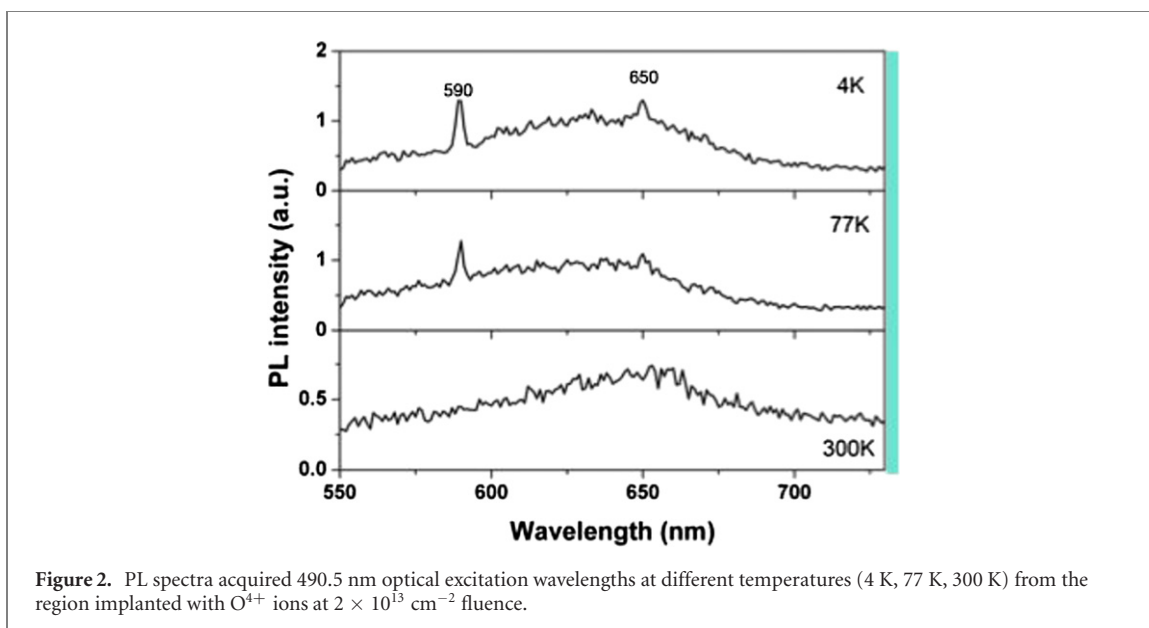
Figure 1. PL spectra acquired at different temperatures (4 K, 77 K, 300 K) from the region implanted with PbO_2^- ions at $2 \times 10^{13} \text{ cm}^{-2}$ fluence. The reported spectra were acquired under the following optical excitation wavelengths: (a) 405 nm (b) 490.5 nm (c) 520 nm.

405 nm excitation. The spectra acquired under 405 nm laser excitation (10 mW optical power) are reported in figure 1(a). All spectra were acquired using along-pass filter with 500 nm cutoff wavelength. The spectra were limited to wavelengths shorter than 720 nm due to the observation of fluorescence at larger wavelengths from the optical components of the confocal microscope. The most evident spectral features, mainly visible at lower temperatures, consist of a small peak at 538 nm, a band centered at ~ 557 nm, and a broad band in the 570–670 nm range. Additional sharp peaks at 643 nm and 648 nm are visible only at 4 K and 77 K temperatures. At 300 K temperature, the above-listed peaks convolve into a broader emission band centered at 645 nm. Considering the ~ 3 nm spectral resolution of the apparatus and the spectral blue-shifting at decreasing temperatures (already reported in reference [28]), these lines are compatible with the weak emission lines at 640.4 nm and 649.8 nm observed under the same excitation wavelength in reference [28].

490.5 nm excitation. Spectra were acquired at 3.5 mW excitation power, using a laser diode with 490.5 nm emission wavelength. The spectra were acquired using a 500 nm long-pass filter and a (514 ± 15) nm notch filter to prevent the observation of the first-order Raman line (expected at 524.8 nm based on the 1332 cm^{-1} Raman shift of diamond). The main PL emission features (figure 1(b)) consist of peaks located respectively at 538 nm, 552 nm and 557 nm, together with an additional wide band at ~ 590 nm, consistently with what reported in reference [28] under 532 nm excitation. The emission lines at 643 nm and 648 nm are also visible under 490.5 nm laser excitation. Finally, a PL peak is observed at 715 nm at 4 K and 77 K temperatures, while this spectral line could not be resolved at room temperature.

A PL spectrum (figure 2) of the O-implanted control sample was acquired under the same experimental conditions to address the contribution of oxygen-related defects to the emission properties observed in reference [28]. Notably, the low-temperature O-related emission exhibits a ZPL at 590 nm, which is qualitatively in line with what reported in reference [8], although its prominence with respect to the baseline is non-negligible only at low temperatures. Based on this observation, we cannot rule out that the 590 nm band could be attributed to O-related impurities, although it was observed on a sample implanted with elemental Pb only in reference [28].

Furthermore, a band centered at 650 nm was visible at low temperatures, although it did not display a doublet and exhibited a limited prominence with respect to the baseline. The band broadened at higher wavelengths emitting in the whole 650–700 nm range. The limited spectral prominence and the difference



in the spectral position (643 nm vs 650 nm) suggest that the 643 nm line observed in Pb (references [28, 29]) and PbO_2 (reference [28] and present work) implanted samples could not be attributed to O-related impurities.

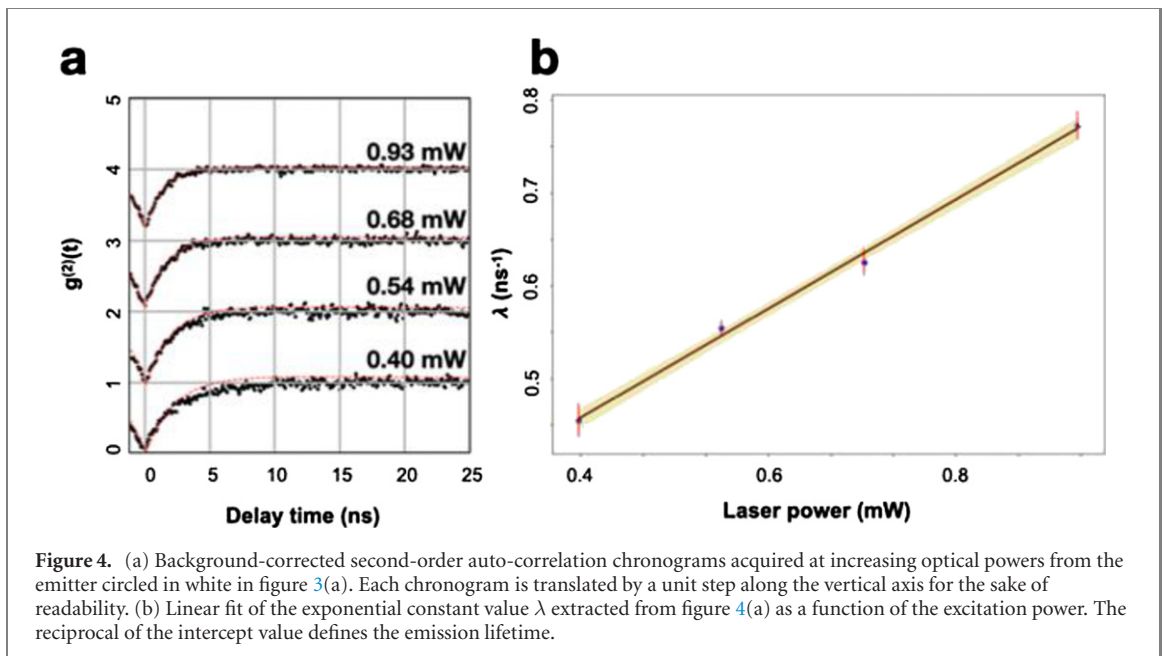
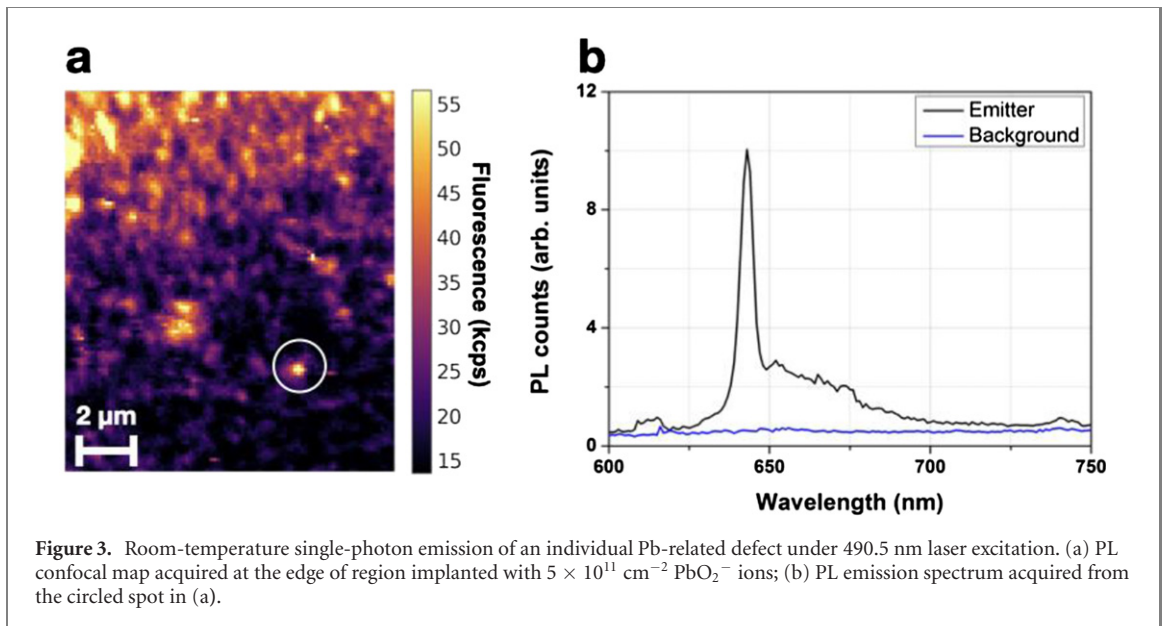
520 nm excitation. The PL spectra (figure 1(c)) were acquired using a 50:50 beam-splitter instead of a long-pass dichroic mirror, with the purpose of allowing the collection of the spectral features at wavelengths shorter than 567 nm, corresponding to the cutoff wavelength of the available dichroic mirror. Since the above-mentioned dichroic mirror was not employed, a (514 ± 15) nm notch filter and a long-pass filter with 530 nm cutoff wavelength were used instead, in order to filter out the laser excitation light.

The prominence of the PL peaks at 538 nm, 552 nm and 557 nm is significantly higher with respect to the other spectral features, if compared with the results obtained at shorter excitation wavelengths. The substantial intensity decrease of the 557 nm feature at increasing temperature suggests that the emission line should be attributed to PL from the Pb-related center, rather than to the first-order Raman emission of diamond (expected at 558.7 nm wavelength), more so considering that this emission line is also visible under different excitation wavelength. Therefore, in our interpretation the first-order Raman line is strongly convoluted with the PL emission feature, while being characterized by a significantly lower emission intensity. The 590 nm band is visible in the spectra, regardless of the measurement temperature. Similarly to what observed under 490.5 nm excitation, the 643 nm peak and the less intense 648 nm peak tend to convolve into a broad emission band at room temperature, although their intensities with respect to the emission doublet at 552 nm and 557 nm is weaker if compared with the PL spectra acquired at lower excitation wavelengths. The emission peak at 715 nm is also barely visible at 4 K temperature, differently from the case of 490.5 nm laser excitation.

3.2. Single-center-level photon emission characterization

We report on the room-temperature characterization of isolated quantum emitters resulting from ions scattered outside of the region implanted at $5 \times 10^{11} \text{ cm}^{-2}$ fluence, as reported in the confocal microscopy map [33] shown in figure 3(a). Differently from the data reported in references [28, 29] under 450 nm and 520 nm laser excitations, respectively, the characterization was performed at room temperature under 490.5 nm excitation (2.8 mW excitation power). In this case, an additional 520 nm notch filter was used to minimize the background emission due to the first-order Raman peak of diamond (expected at 521.8 nm). We discuss the properties of a typical emitter exhibiting a narrow ZPL at 643 nm and a low intensity phonon sideband, as reported in figure 3(b). The attribution of such spectral line to a Pb-related defect was justified by the observation of the 643 nm emission in both reference [27] as well as in section 3.1 of the present work at the ensemble level, and in reference [28] from individual spots, although without a systematic characterization at the single-photon emission level.

The emission properties of the center were studied by means of HBT interferometry [34, 35]. Figure 4(a) shows second-order auto-correlation chronograms collected under different excitation powers and background-subtracted following the procedure described in reference [28]. The <0.5 values observed in correspondence of zero delay time under all excitation power values is indicative of the fact that the



emission originates from an individual optical center [36]. A fitting procedure of the anti-correlation dip around zero delay time was carried on the basis of the two-level system model, using the $g^{(2)}(t) = 1 - \exp[-\lambda t]$ functional expression, as discussed in reference [24]. The time constant $\tau = \lambda^{-1}$ was estimated at each excitation power. The λ values obtained from the above-mentioned exponential fitting procedure as a function of excitation power P were linearly interpolated with the $\lambda(P) = \tau_0^{-1} + \alpha P$ function (figure 4(b)), thus allowing the estimation of the radiative lifetime of the center's excited state as $\tau_0 = (4.5 \pm 0.3) \text{ ns}$. This value is compatible with the value predicted by means of DFT calculations in reference [30], as well as comparable with what observed for both the 552 nm emission line under 520 nm laser excitation [28] and multiple emission lines (520, 552, 555, 575, 648 nm, see table 1) under 450 nm excitation [29]. The emission rate of the defect ($1.8 \times 10^5 \text{ s}^{-1}$ at $\sim 0.93 \text{ mW}$ excitation power) was not directly comparable with the previous results under 520 nm laser excitation (i.e. $>1 \text{ Mcps}$ [28]), since the confocal microscopy setup adopted there was coupled into open air, and was thus characterized by a higher photon detection efficiency. It is worth noting that the emission line discussed in this analysis could not be clearly detected under the same temperature conditions at the ensemble level, due to the convolution of the 643 nm peak with additional spectral feature at room-temperature (figure 3(b)). This observation suggests that the ensemble emission is composed of different types of optical transitions originating from different types of lattice defects. Particularly, this observation suggests that the room-temperature band centered at 650 nm is

Table 1. Summary of the emission lines related to Pb ion implantation in diamond. Comments: * might be superimposed to first-order Raman under 520 nm excitation; ** extremely weak emission under 520 nm excitation; ○ observed from the same emitter under 450 nm excitation; § observed from the same emitter under 490.5 nm excitation.

Emission wavelength (nm)	Excitation wavelength (nm)	Temperature (K)	Interpretation	Notes
517–520	450	4	ZPL doublet, DFT calculations [29, 30]	○
538	405, 490.5, 520, 532	4–300	Possible attribution as PbV ZPL [28]	
552	450, 490.5, 520, 532	4–300	Possible attribution as PbV ZPL [28]	○
557	405, 450, 490.5, 520*, 532	4–300	Phonon replica of the 538/552 nm ZPL [28]	○, *
565	450	4		
575	450, 520, 490.5	4–300	NV ⁰ center ZPL [28, 29]	○
590	405, 490.5, 520, 532	4–300	Phonon sideband of the 552 nm ZPL [28], O-related emission in PbO ₂ implanted diamond	
639–643	405, 450, 490.5, 520	4–300	NV ⁻ center under extreme strain [29], alternative PbV charge state ZPL [28, 29]	§
648	405, 450, 490.5, 520	4–300	Phonon replica of the 643 nm ZPL	§
655–659	450, 490.5	4–300		
715	490.5, 520, 532	4–77		**

the convolution of several different spectra of individual defects, each emitting at a specific set of wavelengths. To further qualify this point, a broader room-temperature investigation on the spectral emission properties of a set of individual luminescent spots is presented in the following section.

3.3. Additional emission lines from isolated emission spots

An additional study was performed to characterize individual emitters in Pb-implanted diamond. Figures 5(a)–(j) display a set of PL spectra acquired at room temperature under 532 nm excitation from the corresponding individual spots highlighted in the confocal map in figure 5(k), that was acquired in correspondence of the outer edge of a PbO₂⁻-implanted region (5×10^{11} cm⁻² fluence). For the sake of comparison, the ensemble PL spectrum acquired at the center of the implanted region under the same experimental conditions is reported in figure 5(l). In these measurements, the use of a 550 nm long-pass filter prevented the detection of the possible occurrence of the Pb-related 552 nm line.

It is worth noting that the individual spots investigated in this analysis were not assessed by means of HBT interferometry, and thus that the observed emission spectra do not necessarily correspond to PL emission at the single-photon level.

The emission features reported in figure 5 can be grouped into few well-defined emission lines. Among them, the PL peaks at 557 nm and 643 nm (figure 5(e)) were already observed at the ensemble level (figure 5(l)), and can thus be safely attributed to Pb-related defects. It is worth mentioning that these two lines concurrently appear in the same spectrum, thus not ruling out that they may originate from the very same defect. Secondly, these lines do not appear concurrently with other Pb-related features discussed in the previous works, such as the 575 nm line or the 590 nm band.

An additional emission line observed in figure 5 is located at 554 nm (figure 5(g)). In consideration of the spectral resolution of the apparatus, this line is compatible with both the 552 nm and 557 nm lines that were attributed to Pb-related defects, and thus its attribution is not straightforward. The line could result from a lattice strain-induced shift of either the afore-mentioned PL peaks, but it is also compatible with the 555 nm emission reported in previous works from proton and neutron irradiated diamond [37, 38].

In addition, a recurring set of spectral lines, not observed at the ensemble level, is visible in figure 5, namely: 564 nm (figures 5(c), (f) and (g)), 567 nm (figures 5(b), (i) and (j)), 581 nm (figures 5(a)), 605 nm (figure 5(a)), 651 nm (figures 5(d) and (i)), 655 nm (figure 5(d)) and 658 nm (figures 5(a), (b), (d), (g) and (j)). There is no apparent correlation between any of these peaks, whose occurrence in the spectra reported in figures 5(a)–(j) is independent from all of the other observed spectral features.

This observation suggests that these emission lines are associated to different types of luminescent defects. Furthermore, the absence of such spectral features in the ensemble PL spectrum in figure 5(l) could be compatible with two possible interpretations. On one hand, the lines might be associated either to radiation-induced defects which are not directly related to Pb impurities. This attribution may be justified by previous reports of emission lines in irradiated diamond at 563 nm [37, 39] and 656 nm [37]. On the other hand, some of the afore-mentioned PL peaks might be related to alternative Pb-containing defects, whose formation in high concentrations is not favored by the thermal processing adopted in the present work.

It is also worth mentioning that the 564 nm, 567 nm, 651 nm and 658 nm peaks have been observed from isolated spots at 4 K temperature (450 nm excitation wavelength) in reference [29]. In this work, their

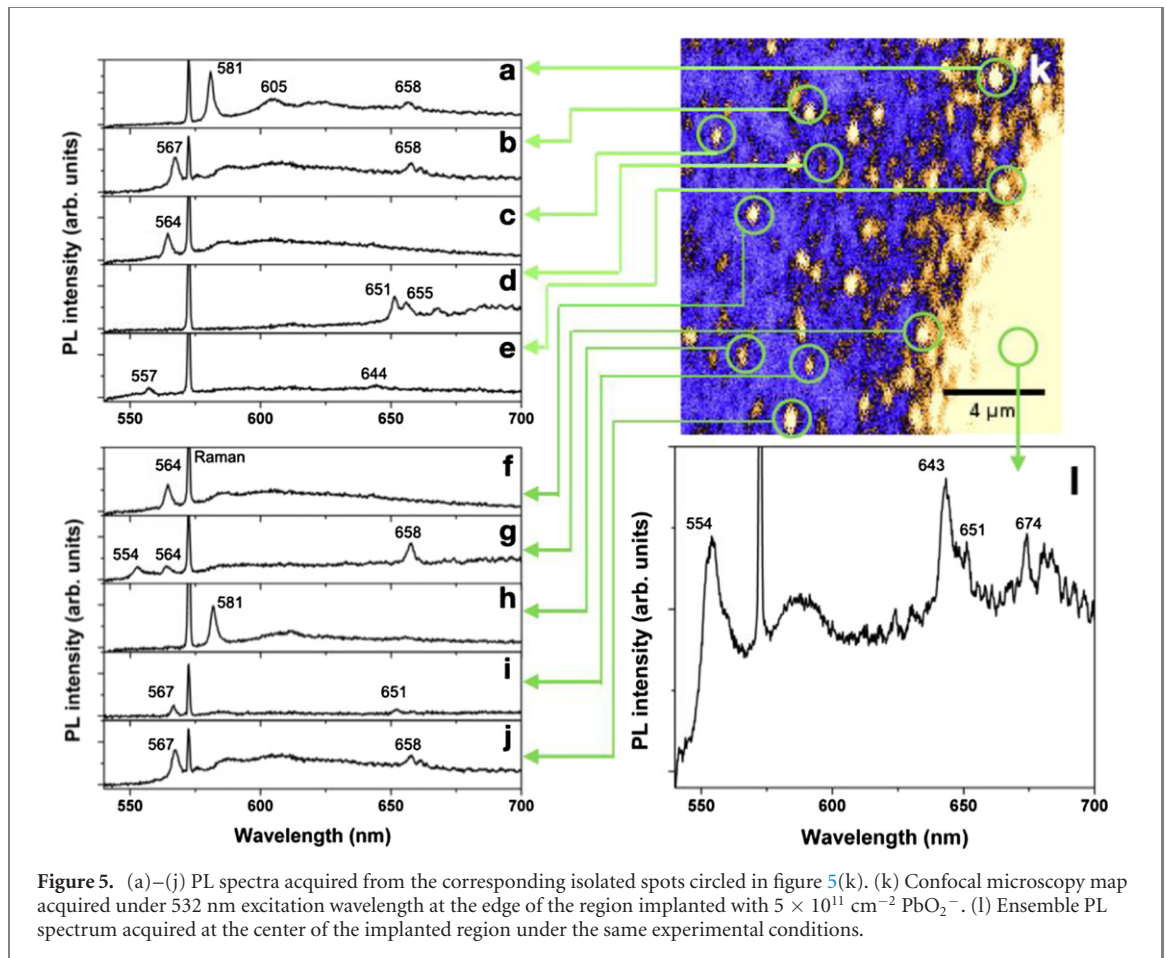


Figure 5. (a)–(j) PL spectra acquired from the corresponding isolated spots circled in figure 5(k). (k) Confocal microscopy map acquired under 532 nm excitation wavelength at the edge of the region implanted with $5 \times 10^{11} \text{ cm}^{-2} \text{ PbO}_2^-$. (l) Ensemble PL spectrum acquired at the center of the implanted region under the same experimental conditions.

occurrence was significantly less frequent with respect to the main emission lines (520 nm, 525 nm, 552 nm, 575 nm, 590 nm, 638 nm) that were attributed to Pb-related defects, therefore an attribution of these emission lines was not suggested. It is therefore worth remarking that our observation is in line with the results obtained in reference [29], despite the adoption of a different excitation wavelength (i.e. 450 nm).

4. Discussion

In order to clarify the compatibility of the reported results with respect to previous works, the results of PL characterization at ensemble level of PbO_2^- -implanted diamond is discussed more extensively in this section. The reported emission lines are summarized in table 1 and sorted by increasing emission wavelength, in the context of the relevant spectral features that have been reported so far. A direct comparison with the previous works [28, 29] is made possible by the fact that both of them relied on the fabrication of defects by ion implantation, as well as on the same annealing temperature (i.e. 1200 °C), implying the same conditions for the formation of optical centers in diamond. The findings are also discussed in the framework of the currently available DFT theoretical predictions [27, 30]. The following discussion is limited to the spectral features that were unambiguously observed at the ensemble level or directly attributed to Pb-related defects in the previous reports.

517–520 nm. This line was experimentally observed under 450 nm excitation at 4 K temperature in reference [28] from individual emission spots together with the 552 nm, 557 nm, 575 nm and 648 nm emission lines. A 2.4 eV emission is theoretically predicted to be the ZPL of the Pb-V center [29, 30]. However, in our experiment we were not able to observe it under any operating condition of excitation wavelength and measurement temperature. In figure 6 we report the PL spectra in the 510–600 nm range acquired under different excitation wavelengths (i.e. 405 nm and 488 nm) at 4 K temperature from the region implanted with PbO_2^- ions at $2 \times 10^{13} \text{ cm}^{-2} \text{ PbO}_2^-$ fluence. The fact that the peak was not visible under 405 nm laser excitation (figure 6, blue line) could be explained by assuming that the energy of the laser light might be too high to efficiently excite this emission or alternatively the occurrence of a partial ionization of the center. Furthermore, the 490.5 nm laser excitation determines the first-order 1332 cm^{-1}

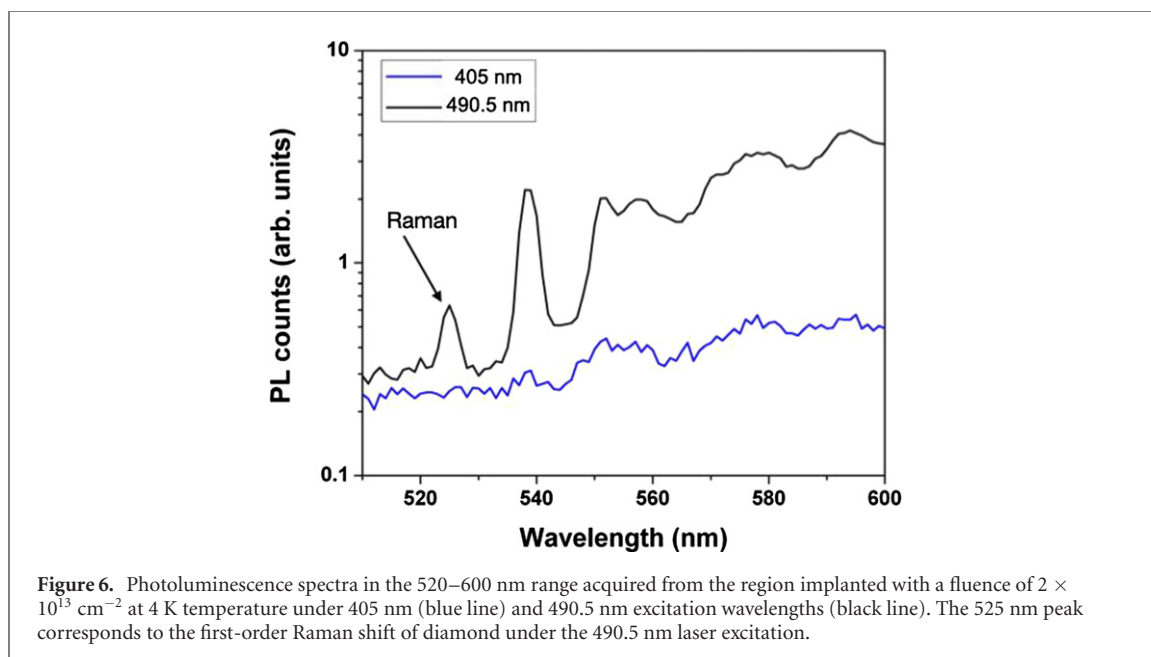


Figure 6. Photoluminescence spectra in the 520–600 nm range acquired from the region implanted with a fluence of $2 \times 10^{13} \text{ cm}^{-2}$ at 4 K temperature under 405 nm (blue line) and 490.5 nm excitation wavelengths (black line). The 525 nm peak corresponds to the first-order Raman shift of diamond under the 490.5 nm laser excitation.

Raman shift to be spectrally located at 524.8 nm: in this context, in this study a 490.5 nm laser excitation was adopted to allow the full deconvolution of Raman and PL spectral features across the 517–520 nm range. However, the PL spectrum collected under 409.5 nm excitation reported in figure 6 (black line) did not exhibit any spectral feature in the region of interest, with the exception of the 525 nm Raman line. Finally, the 520 nm laser could not be adopted to investigate this line under near-resonant excitation and therefore its possible occurrence under these excitation conditions could not be assessed.

525 nm. This line was observed under 450 nm excitation at 4 K from an individual spot together with the 552 nm, 557 nm, 575 nm and 648 nm lines [29]. However, it was not visible under 405 nm excitation in the 4–300 K temperature range and could not be assessed under 490.5 nm excitation due to the overlapping first-order Raman emission at 1332 cm^{-1} shift.

538 nm, 552 nm, 557 nm. These emission peaks were observed in reference [28] under 514 nm and 532 nm excitations, as well as in reference [29] under 450 nm excitation with the exception of the 538 nm line. Their observation is confirmed in the present work under 405 nm, 490.5 nm and 520 nm excitation. The highest prominence with respect to the spectral baseline was achieved under 520 nm excitation. These emission lines were concurrently observed from isolated spots under 450 nm excitation [28], although in combination with additional lines at 525 nm and 648 nm. Also, they were observed from single-photon emitters under 532 nm excitation concurrently with the 590 nm band [28]. These results strongly suggest that all of the above-listed emission lines originate from the same type of Pb-related impurity. The 552 nm and 557 nm lines were tentatively interpreted as possible ZPL candidates for the PbV center in reference [28], although without the support of a comprehensive theoretical modelling.

575 nm. This emission line is observed in this work as a minor peak under all operating conditions. The peak is visible in the whole 4–300 K temperature range under 405 nm, 450 nm, 490.5 nm, 514 nm, 520 nm and 532 nm laser excitation [28, 29]. Both of the above-cited works attributed the emission to the neutrally-charged NV center, although its systematic occurrence over a number of high-quality samples characterized by a low concentration of substitutional N (\sim ppb) is worth being remarked.

590 nm. This emission band was consistently observed in all experimental reports on the Pb-related defects in the 4–300 K temperature range under 514 nm [28] and 532 nm [28, 29] excitation. This work confirms its occurrence under 405 nm, 490.5 nm and 520 nm excitation in all of the considered measurement temperature ranges. The band was also observed concurrently with the 538 nm, 552 nm and 557 nm lines at the single-photon emitter level, and thus tentatively interpreted as an optical-photon-mediated emission process [28]. However, the identification in this work of a 590 nm emission line in O-related ensembles suggests that this latter attribution could not be ruled out.

639 nm–660 nm. PL emission in this spectral region was observed in reference [29] at 4 K, under both 450 nm and 532 nm excitations, and it was tentatively attributed to NV centers. The observation of different lines (640 nm and 650 nm, in the 143–300 K temperature range) in reference [28] under 405 nm laser excitation poses questions about this attribution, since it has been previously observed that the NV^- emission is quenched by ionization processes at excitation energies larger than 2.6 eV [40]. In this

sense, a possible, tentative attribution to a different charge state of the PbV center could not be entirely ruled out on the basis of the available data. This interpretation is mentioned in reference [29], where the spectral switching between the 520 nm ZPL and this set of lines could be attributed to the PbV charge state instability. It is also worth considering the recent attribution of a 639 nm PL line to interstitial-related lattice defects [41]. The plausibility of the latter interpretation is based on the large density of radiation-induced vacancies generated in diamond by the impact of an individual Pb ion.

In the present work an intense doublet was observed at the ensemble level at 643–648 nm under all experimental conditions (the highest prominence with respect to the baseline being observed under 490.5 nm), although at room temperature they were strongly convoluted into a broader emission band. The temperature-dependent results shown here offer an unambiguous link between the clear observation of the lines reported in reference [29] and their broadened, weak emission discussed in the reference [28]. The fact that the 643 nm line is more intense and narrower than the 648 nm line (figure 1) might support the interpretation of the former as a ZPL which is associated to its first phonon replica at a larger wavelength. Notably, the ~ 18 meV spectral separation between the two peaks is in line with the expected value for quasi-local vibrations involving a Pb atom in the diamond lattice (i.e. ~ 21.5 meV [28]). Moreover, room-temperature single-photon emission was observed with a ZPL of 643 nm, as reported in figure 3(b) under 490.5 nm laser excitation. In this case the possible phonon sideband (648 nm) is broadened at room temperature over a band in the 645–680 nm range.

The ~ 650 nm band observed from ensembles at room temperature does not directly correspond to the spectral features of a single-photon emitter, for which the emission line is confirmed to be narrow. The band could therefore be attributed to a broadening of the ensemble emission of a set of individual emitters as a function of temperature. Conversely, a band centered at 650 nm was also observed in this work at low temperatures in O-implanted diamond, although it did not display a doublet and exhibited a limited prominence with respect to the baseline. The limited spectral prominence and the difference in the spectral position (643 nm vs 650 nm) suggest that the 643 nm line observed in Pb (references [28, 29]) and PbO₂ (reference [28] and present work) implanted samples could not be attributed to O-related impurities.

655–659 nm. Some emission lines in this spectral range are reported in reference [29] under 450 nm excitation at 4 K temperature, however their attribution is not discussed in details. PL peaks in the 651–658 nm range were also observed at room temperature in the present work from isolated spots under 532 nm excitation, although without displaying any correlation with spectral lines related to Pb-containing defects. Our tentative attribution of these lines, due to their absence from ensemble PL spectra, is therefore of radiation-induced defects. However, it cannot be ruled out that the emission originates from a Pb-containing defect characterized by a lower formation efficiency under the thermodynamic conditions adopted both in the present study and in the previous experimental efforts [28, 29].

670 nm. A relatively weak emission line at this wavelength is reported in reference [28] from room-temperature measurements carried under 405 nm laser excitation. Its observation however was not confirmed by the measurements performed in the present work under different experimental conditions (excitation wavelength, measurement temperature).

715 nm. A peak appearing at this wavelength was reported in reference [28] at a 4 K temperature. Particularly, it was observable under 532 nm laser excitation, while it could not be detected under a 450 nm excitation. Moreover, it was not observed at room temperature in reference [28] under 514 nm and 532 nm excitations. In the present work we observe that the emission line can be excited under 490.5 nm and 520 nm wavelengths at 4 K and 77 K (see figure 1), and that its intensity significantly drops at increasing temperature until it is not visible at room temperature. This observation harmonizes the results reported references [28, 29], indicating that the ionization of the involved emitter occurs at excitation wavelengths across the 450–490.5 nm range, and that the line visibility is suppressed in the 77–300 K temperature range. To the best of our knowledge, no apparent interpretation of this emission lines are currently available in literature.

5. Conclusions

We reported on the creation of Pb-related color centers in diamond by means of PbO₂⁻ ion implantation and subsequent thermal annealing, and their optical characterization. The measurements were performed by adopting three different excitation wavelengths, namely 405 nm, 490.5 nm and 520 nm, at different temperatures in the 4–300 K range. The results of the systematic investigation of an articulated set of PL emission peaks provide a solid interpretational framework that substantially expands the understanding of this defective complex. Moreover, both of the most prominent doublets observed at 552–557 nm and 643–648 nm [28, 29] were also observed at the single photon emitter level: while the former was already

reported in reference [28] under 532 nm excitation, the latter was characterized in the present work for the first time, under 490.5 nm excitation. A dedicated study on the photochromism [16] of individual defects is needed to clarify whether the striking resemblance of the two observed doublets could stem from their attribution to the ZPLs of Pb-related defects in different charge states.

The observation of several additional emission lines has also been studied under different experimental conditions. Particularly, the observation of the 715 nm line in reference [29] has been confirmed by our observations under 490.5 nm and 520 nm laser excitations at $T \leq 77$ K, while the emission could not be observed at room temperature.

Additionally, the ZPL at 517–520 nm observed in reference [29] and predicted in reference [30] was not observed in this work under any excitation wavelength and measurement temperature conditions.

Finally, a set of additional lines, not visible at the ensemble level, were observed from isolated spots at the edge of Pb-implanted regions. Their attribution was not conclusive and it cannot be ruled out that their lack of visibility in ensemble PL spectra might originate from a sub-optimal activation under the adopted thermal processing conditions. The high damage density introduced in the diamond lattice by the implantation of a heavy ion such as Pb might indeed require more effective processing methods to promote the formation of the PbV center, such as HPHT annealing [25], or alternative fabrication methods such as the incorporation of impurities during the crystal synthesis [26]. Future experimental studies on the defect formation based on such techniques could contribute to fully disambiguate the nature of the spectral features associated with Pb implantation in diamond.

These results clarify the potential of the PbV center for its high-impact application in quantum technologies. Besides the clear advantage offered by a bright and narrowband emitter for on-demand single-photon generation [42] or labeling in biological environments [43], the long spin coherence stemming from the predicted large ground-state splitting of the PbV^- center prospects a promising tool for the practical realization of multi-qubit quantum network nodes.

Acknowledgments

This work was supported by the following projects: Coordinated Research Project ‘F11020’ of the International Atomic Energy Agency (IAEA); Project ‘Piemonte Quantum Enabling Technologies’ (PiQuET), funded by the Piemonte Region within the ‘Infra-P’ scheme (POR-FESR 2014–2020 program of the European Union); ‘Departments of Excellence’ (L. 232/2016), funded by the Italian Ministry of Education, University and Research (MIUR); ‘Ex post funding of research’ of the University of Torino funded by the ‘Compagnia di San Paolo’; FET OPEN Pathos; ‘Intelligent fabrication of QUANTum devices in DIAMond by Laser and Ion Irradiation’ (QuantDia) project funded by the Italian Ministry for Instruction, University and Research within the ‘FISR 2019’ program; ‘Training on LASer fabrication and ION implantation of DEFects as quantum emitters’ (LasIonDef) project funded by the European Research Council under the ‘Marie Skłodowska-Curie Innovative Training Networks’ program. The projects 17FUN06 (SIQUST), 17FUN01 (BeCOMe) and 20FUN05 (SEQUME) leading to this publication have received funding from the EMPIR programme co-financed by the Participating States and from the European Union’s Horizon 2020 Research and Innovation Programme. TL, SP and JM acknowledge the support of the ASTERIQS program of the European Commission. SDT and JF gratefully acknowledge the EU RADIATE Project (proposal 19001744) for granting transnational access to the Laboratories of the Ruđer Bošović Institute.

Data availability statement

The data that support the findings of this study are available upon reasonable request from the authors.

ORCID iDs

Sviatoslav Ditalia Tchernij  <https://orcid.org/0000-0002-2639-5229>

Ivo Pietro Degiovanni  <https://orcid.org/0000-0003-0332-3115>

Georgios Provatas  <https://orcid.org/0000-0001-9471-1572>

Paolo Olivero  <https://orcid.org/0000-0002-7512-6295>

Marco Genovese  <https://orcid.org/0000-0001-9186-8849>

Jacopo Forneris  <https://orcid.org/0000-0003-2583-7424>

References

- [1] Doherty M W, Manson N B, Delaney P, Jelezko F, Wrachtrup J and Hollenberg L C L 2013 The nitrogen-vacancy colour centre in diamond *Phys. Rep.* **528** 1–45
- [2] Childress L and Hanson R 2013 Diamond NV centers for quantum computing and quantum networks *MRS Bull.* **38** 134–8
- [3] Nelz R et al 2019 Toward wafer-scale diamond nano- and quantum technologies *APL Mater.* **7** 011108
- [4] Metsch M H et al 2019 Initialization and readout of nuclear spins via negatively charged silicon-vacancy center in diamond *Phys. Rev. Lett.* **122** 190503
- [5] Schröder T, Mouradian S L, Zheng J, Trusheim M E, Walsh M, Chen E H, Li L, Bayn I and Englund D 2016 Quantum nanophotonics in diamond (Invited) *J. Opt. Soc. Am. B* **33** B65–83
- [6] Bradac C, Gao W, Forneris J, Trusheim M and Aharonovich I 2019 Quantum nanophotonics with group IV defects in diamond (arXiv:1906.10992)
- [7] Petrini G, Moreva E, Bernardi E, Traina P, Tomagra G, Carabelli V, Degiovanni I P and Genovese M 2020 Is a quantum biosensing revolution approaching? Perspectives in NV-assisted current and thermal biosensing in living cells *Adv. Quantum Technol.* **3** 2000066
- [8] Lühmann T et al 2018 Screening and engineering of colour centres in diamond *J. Phys. D: Appl. Phys.* **51** 483002
- [9] Eaton S M et al 2019 Quantum micro-nano devices fabricated in diamond by femtosecond laser and ion irradiation *Adv. Quantum Technol.* **2** 1900006
- [10] Barry J F, Turner M J, Schloss J M, Glenn D R, Song Y, Lukin M D, Park H and Walsworth R L 2016 Optical magnetic detection of single-neuron action potentials using quantum defects in diamond *Proc. Natl Acad. Sci.* **113** 14133–8
- [11] Hensen B et al 2015 Loophole-free Bell inequality violation using electron spins separated by 1.3 kilometres *Nature* **526** 682
- [12] Breeze J D, Salvadori E, Sathian J, Alford N M and Kay C W M 2018 Continuous-wave room-temperature diamond maser *Nature* **555** 493
- [13] Webb J L, Clement J D, Troise L, Ahmadi S, Johansen G J, Huck A and Andersen U L 2019 Nanotesla sensitivity magnetic field sensing using a compact diamond nitrogen-vacancy magnetometer *Appl. Phys. Lett.* **114** 231103
- [14] Zhou H et al 2020 Quantum metrology with strongly interacting spin systems *Phys. Rev. X* **10** 31003
- [15] Unden T et al 2018 Coherent control of solid state nuclear spin nano-ensembles *npj Quantum Inf.* **4** 39
- [16] Gaebel T et al 2006 Photochromism in single nitrogen-vacancy defect in diamond *Appl. Phys. B* **82** 243–6
- [17] Bradac C et al 2010 Observation and control of blinking nitrogen-vacancy centres in discrete nanodiamonds *Nat. Nanotechnol.* **5** 345–9
- [18] Becker J N et al 2016 Ultrafast all-optical coherent control of single silicon vacancy colour centres in diamond *Nat. Commun.* **7** 13512
- [19] Görlitz J et al 2020 Spectroscopic investigations of negatively charged tin-vacancy centres in diamond *New J. Phys.* **22** 013048
- [20] Sukachev D D et al 2017 Silicon-vacancy spin qubit in diamond: a quantum memory exceeding 10 ms with single-shot state readout *Phys. Rev. Lett.* **119** 223602
- [21] Iwasaki T et al 2015 Germanium-vacancy single color centers in diamond *Sci. Rep.* **5** 12882
- [22] Siyushev P et al 2017 Optical and microwave control of germanium-vacancy center spins in diamond *Phys. Rev. B* **96** 081201
- [23] Meng X, Liu S, Dadap J I and Osgood R M Jr 2017 Plasmonic enhancement of a silicon-vacancy center in a nanodiamond crystal *Phys. Rev. Mater.* **1** 015202
- [24] Ditalia Tchernij S et al 2017 Single-photon-emitting optical centers in diamond fabricated upon Sn implantation *ACS Photonics* **4** 2580
- [25] Iwasaki T, Miyamoto Y, Siyushev P, Jelezko F, Taniguchi T, Metsch M and Hatano M 2017 Tin-vacancy quantum emitters in diamond *Phys. Rev. Lett.* **119** 253601
- [26] Palyanov Y N, Kupriyanov I N and Borzdov Y M 2019 High-pressure synthesis and characterization of Sn-doped single crystal diamond *Carbon* **143** 769
- [27] Trusheim M E et al 2020 Transform-limited photons from a coherent tin-vacancy spin in diamond *Phys. Rev. Lett.* **124** 023602
- [28] Ditalia Tchernij S et al 2018 Single-photon emitters in lead-implanted single-crystal diamond *ACS Photonics* **5** 4864
- [29] Trusheim M E et al 2019 Lead-related quantum emitters in diamond *Phys. Rev. B* **99** 075430
- [30] Thiering G and Gali A 2018 *Ab initio* magneto-optical spectrum of group-IV vacancy color centers in diamond *Phys. Rev. X* **021063** 2–17
- [31] Middleton R A 1990 *Negative Ion Cookbook* (Philadelphia: University of Pennsylvania)
- [32] Gozzelino L et al 2011 Magnetic characterization of MgB₂ bulk superconductor for magnetic field mitigation solutions *J. Supercond. Nov. Magnetism* **24** 307–12
- [33] Binder J M et al 2017 Qudi: a modular python suite for experiment control and data processing *SoftwareX* **6** 85
- [34] Brouri R, Beveratos A, Poizat J-P and Grangier P 2000 Photon antibunching in the fluorescence of individual color centers in diamond *Opt. Lett.* **25** 1294
- [35] Chunnillal C J, Degiovanni I P, Kück S, Müller I and Sinclair A G 2014 Metrology of single-photon sources and detectors: a review *Opt. Eng.* **53** 081910
- [36] Eisaman M D, Fan J, Migdall A and Polyakov S V 2011 Invited review article: single-photon sources and detectors *Rev. Sci. Instrum.* **82** 071101
- [37] Zaitsev A M 2001 *Optical Properties of Diamond* (Berlin: Springer)
- [38] Field J E 1992 *The Properties of Natural and Synthetic Diamond* (New York: Academic)
- [39] Forneris J et al 2017 Electrical control of deep NV centers in diamond by means of sub-superficial graphitic micro-electrodes *Carbon* **113** 76–86
- [40] Aslam N, Waldherr G, Neumann P, Jelezko F and Wrachtrup J 2013 Photo-induced ionization dynamics of the nitrogen vacancy defect in diamond investigated by single-shot charge state detection *New J. Phys.* **15** 013064
- [41] Zhang Y, Wang K, Chang S, Tian Y and Li J 2020 Photoluminescence studies of the 640 nm center in electron irradiated IIa diamond *Spectrosc. Lett.* **53** 580
- [42] Sipahigil A et al 2016 An integrated diamond nanophotonics platform for quantum-optical networks *Science* **354** 847–50
- [43] Simpson D A, Morrisroe E, McCoe J M, Lombard A H, Mendis D C, Treussart F, Hall L T, Petrou S and Hollenberg L C L 2017 Non-neurotoxic nanodiamond probes for intraneuronal temperature mapping *ACS Nano* **11** 12077