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Viscoelastic model for analysing the behaviour of adhesivebonded FRP-to-steel joints in civil engineering applications

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14 Abstract

1

15 Adhesively bonding fibre reinforced polymer (FRP) plates have been a mainstream method for 16 strengthening civil engineering structures. The effectiveness of bonding strengthening is 17 dependent on the performance of the adhesive layer. This study examines the time-temperature 18 dependent viscoelasticity of a typical structural epoxy adhesive and develops a linear viscoelastic material model that can be used for numerical analysis. An accelerated test method 19 20 involving dynamic mechanical analysis (DMA) and time-temperature superposition principle 21 (TTSP) were used to characterise the structural adhesive. The corresponding limitations of this method were discussed, including decomposition effect, nonlinear viscoelasticity, applicability 22 23 of TTSP, and further curing of the sample. The numerical study of the long-term behaviour of 24 the single lap-shear FRP-to-steel joint found that a slightly warmer temperature (30 °C for this 25 study) is beneficial for the bonded joint as it can reduce the concentrated shear stress with negligible increase in the shear strain. However, a higher temperature (50 °C for this study) 26 27 that near the adhesive's glass transition temperature (T_g) could be detrimental as it can lead to 28 significant shear strain in the adhesive joint within the first year of service life.

30 Keywords: Structural epoxy adhesive; Viscoelasticity; Elevated temperature; DMA; FRP-to31 steel joint.

32

33 1 Introduction

Adhesives have been widely used in various industries including automotive, aircraft, and aerospace, since their advent [1–3]. Also for civil engineering, with the rapidly growing need of rehabilitating and strengthening existing structures, adhesively bonding an FRP plate has currently been a mainstream strengthening method [4–6]. Compared to traditional strengthening methods involve welding, bolting, or riveting, the use of adhesively bonding can bring advantages such as lower additional weight, lower time and labour intensity, and more uniformly distributed stresses in the bonding area [7–13].

41 The effectiveness and reliability of bonding strengthening are significantly dependent on 42 the quality, integrity, and durability of the adhesive layer which acts as a load transfer medium 43 between the composite and structure. Especially for FRP-to-steel joints, the adhesive layer 44 inevitably becomes the weakest section due to its relatively lower strength and stiffness [3,14-45 16]. Commonly used ambient-cured adhesives are typically made of viscoelastic polymers. 46 Thermal exposure can soften the adhesive and induce temperature- and time- dependent 47 viscoelastic creep along bonded joints, which may be detrimental and hence affect the 48 performance of strengthened structures [17–22]. In current design guidelines for strengthening 49 civil engineering metallic structures using FRP, this issue is avoided by simply limiting the 50 maximum operating temperature to a minimum of 15 °C below the peak Tan δ glass transition 51 temperature (T_g) of the bonding adhesive and using a multiple combination of large safety 52 factors, which reduces the design efficiency and hampers the extensive use of this advanced strengthening method [7,10,23-25]. A more profound understanding of how the time-53

54 temperature related viscoelasticity of adhesives affects the performance of bonded joints is 55 desired.

56

57 **1.1** Literature survey

58 The existing studies on FRP-to-steel bonded joints in structures have been mainly focusing 59 on their short-term performance [18,20,26,27]. While Heshmati et al. [5] examined the 60 performance of FRP-to-steel double lap-shear joints after immersed in distilled water at 20 °C 61 and 45 °C for up to three years and found that there was a noticeable greater strength reduction 62 in those joints immersed at 45 °C. De Zeeuw et al. [19] investigated the effects of 40 °C air (or 63 distilled water) and constant load on the behaviour of single lap-shear joints. Within 14 days 64 of sustained loading test, they observed the time-dependent viscoelastic creep behaviour. 65 However, the design service life of civil engineering constructions is more than decades. The 66 results obtained by these conventional experimental methods are usually limited.

67 Another research strategy is to use an accelerated test method to directly characterise the 68 viscoelasticity of adhesives, as it is the major factor responsible for the deterioration of joints' 69 performance. Dynamical mechanical analysis (DMA) is a common technique used to 70 characterise the viscoelasticity of polymers as a function of temperature and frequency (or 71 time). The time-temperature superposition principle (TTSP) is classically applied along with 72 the DMA test, which allows the viscoelastic behaviour at a lower frequency (or over a longer time) to be estimated from the behaviour at a higher frequency (or over a shorter time), but at 73 74 another higher temperature, thus avoiding the limitations of the measurement instruments [16,28–30]. This method has been used in several studies in other industries [16,21,29,30], 75 76 however, relevant studies in civil engineering are limited, and the data and conclusions from 77 other studies may not be applicable due to differences in types of adhesives, curing conditions, 78 operating temperatures, and useful service life [3].

79 Houhou et al. [25] examined the viscoelasticity of a structural epoxy adhesive using 80 aforementioned accelerated test method (DMA and TTSP), but the further analysis was 81 conducted on the behaviour of FRP-to-concrete joints, in which concrete might be the weakest 82 part. Nevertheless, the study found that the viscoelastic creep can induce a redistribution of 83 interfacial stresses, leading to a reduction in the concentrated peak stress and an increase in 84 effective transfer length, which could be beneficial for the durability of bonded joints. However, 85 this work did not further consider the temperature effect on the joint, which may bring greater 86 creep deformation. The authors [31] previously examined a strengthening adhesive with a relatively low onset T_g (38.0 °C) for investigating the worst case of viscoelastic creep at 87 88 elevated temperatures. Whilst it was found that the creep reduced the performance of FRP-89 strengthened metallic beams during the long-term services, the details of stress redistribution 90 and increased strain within the bonded joints were not shown, and the authors were also aware 91 that the characteristic data of the adhesive with low T_g may be less representative. Besides, 92 none of these studies discussed the limitations of using the accelerated test method, leaving it 93 unclear whether the structural adhesive decomposed during the DMA test at high temperatures, 94 and whether applying the TTSP was feasible without introducing substantial errors.

Against this background, the objectives of this study are (a) charactering the viscoelasticity of a typical structural adhesive; (b) discussing the limitations of the applied accelerated test method and identifying the need for further research; (c) obtaining a profound understanding of the impact of elevated temperatures on the long-term behaviour of FRP-to-steel joints.

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- 100

2 Experimental characterisation of viscoelasticity

The structural adhesive characterised in the present study was a two-component epoxy resin,
 Sikadur[®]-330 [32], which has been widely used in FRP strengthening for civil engineering
 structures. The adhesive samples used in these tests were cured at ambient temperature (21 ± 4

104 2 °C) and humidity (50 \pm 10% RH) for 28 days, which could be consistent with the practical 105 strengthening application and ensured that the samples had reasonable T_g values. Prior to 106 characterising the viscoelasticity of the adhesive through DMA tests at high temperatures, the 107 thermogravimetric analysis (TGA) test was conducted to characterise its thermal stability in 108 order to determine the maximum applicable experimental temperature. Combining the DMA 109 test with TTSP, a complete modulus master curve was then constructed. The usage range of 100 applying TTSP was also discussed.

111

112

12 **2.1** Thermogravimetric analysis (TGA)

In many studies that apply DMA to conduct the accelerated characterisation of adhesives at high temperatures (> 100 °C), the thermal stability of the tested sample was not examined [16,30,31]. However, when the temperature rises above 100 °C, the loss of mass attributed to the evaporation of low molecules from epoxy adhesives may have started, as structural epoxy adhesives typically contain fillers, curing agents, and accelerators [32–35]. This can therefore affect the obtained modulus response data and thus introduce errors.

In this study, the TGA experiment was conducted to examine the thermal stability of the studied adhesive according to ISO 11358 [36] through a Mettler Toledo thermogravimetric instrument. The powdery adhesive sample (15.1 mg) was placed in the aluminium oxide crucible which was set in the sample holder of the analyser. The test was run from 25 °C to 900 °C in an air atmosphere with a heating rate of 10 °C/min. Figure 1 shows the resultant mass loss curve and mass derivative curve.



 $\frac{125}{126}$

Figure 1: Decomposition temperature (T_d) of the adhesive

127 Whilst the obtained onset decomposition temperature (T_d) of the structural adhesive was 128 373 °C [36,37], the thermal decomposition behaviour started to occurs when the temperature 129 exceeded approximately 100 °C. This suggests a maximum experimental temperature of 130 100 °C, below which the decomposition behaviour can be ignored.

131

132 2.2 Dynamic mechanical analysis (DMA)

The DMA characterising experiments were carried out according to ISO 6721 [38] through
a DMA 8000 instrument. The dimensions of the rectangular adhesive specimen and applied
configuration in each test are illustrated in Table 1.

136

Table 1: Dimensions of specimens and applied configurations in DMA tests

| Test | Configuration | Dimension (mm) | | |
|----------------|----------------------|----------------|-------|-----------|
| | | Length | Width | Thickness |
| T_g | Single cantilever | 35 | 7.45 | 1.59 |
| Frequency scan | | 35 | 7.43 | 1.60 |
| Strain scan | Dual cantilever | 45 | 7.50 | 1.48 |

137

138 2.2.1 Glass transition temperature (T_g)

As mentioned in Section 1, the maximum working temperature of the structural bonded joint stipulated in the current design guidelines is based on the T_g of the applied adhesive [7,23]. T_g , 141 as an important material property, indicates the transition temperature range of a thermoset142 polymer from a stiff glassy state to a rubbery state [39].

143 The DMA time/temperature scan test was carried out to determine the T_g of the examined 144 adhesive, using a sinusoidal displacement of 0.05 mm at 1.0 Hz, and 2 °C/min temperature 145 ramp [28,38]. The measured temperature-dependent storage modulus and Tan δ were shown in 146 Figure 2.



 $147 \\ 148$

A significant reduction in the storage modulus occurred between the onset T_g (56.2 °C) and the peak Tan δ T_g (66.4 °C) [28,38]. In order to fully characterise the viscoelasticity of the adhesive at elevated temperatures, the average T_g (61.3 °C) was used as the benchmark intermediate temperature so that the applied temperature range of the subsequent viscoelastic characterising tests was set between 25 °C to 100 °C.

154

155 2.2.2 Modulus response in DMA strain scans

156 Strain scans measurements were performed by applying an oscillatory displacement of 1.0 157 Hz with amplitudes ranging from 0.006 mm to 0.100 mm, corresponding to applied strains of 158 between approximately 0.012% and 0.197%, to examine whether the adhesive would exhibit 159 nonlinear viscoelasticity. A dual cantilever configuration was utilised to guarantee that the

- 160 sample was subjected to a generally uniform strain when large displacements were applied [28].
- 161 These scans were repeated at 5 °C intervals over a temperature range of 25 °C to 100 °C. Figure
- 162 3 illustrates the resultant storage modulus responses.





165 The examined adhesive exhibits very limited strain (stress) dependent behaviour. It is only 166 when the temperature rises close to the T_g , resulting in a sharp drop in the storage modulus, 167 that the behaviour of the adhesive shows a relatively noticeable nonlinearity. Consequently, in 168 the current paper, the behaviour of the adhesive was discussed solely in terms of linear 169 viscoelasticity.

Note that the strain (stress) applied in the DMA test is relatively minimal to prevent fatigue damage to the sample [28]. The adhesive may exhibit significant nonlinear viscoelastic behaviour when subjected to higher strain (stress). However, the stress redistribution behaviour of bonded joints at elevated temperatures can drastically reduce the stresses carried by the adhesive layer [25,27,40], and thus limit the effect of nonlinear response. Detailed investigation of the nonlinear viscoelasticity of adhesives was not, however, the focus of this paper.

177 2.2.3 Modulus response in DMA frequency scans

The viscoelastic response of the structural adhesive was examined using DMA multifrequency scans test [28]. The specimen was exposed to a sinusoidal displacement of 0.01 mm at frequencies ranging from 0.01 Hz to 100 Hz, logarithmically spaced to cover five measurements pre decade. The test was carried out over a temperature range of 25 °C to 100 °C with a temperature step of 5 °C. The obtained storage modulus that increased with frequency but decreased with temperature is shown in Figure 4 (a), and the corresponding loss modulus is shown in Figure 4 (b).



188

189 2.2.4 *Time-temperature superposition principle*

As mentioned in the literature review, TTSP has been used in a number of studies to build modulus master curves of adhesives based on DMA viscoelastic characterisation results [16,21,25,29–31]. However, most of studies did not consider the applicability of TTSP, which raised concerns about the accuracy of developed master curves.

194 TTSP is applicable to thermorheologically simple materials whose viscoelastic response 195 depend equally on temperature [41,42]. The wicket plot (log Tan δ versus log storage modulus) is one way to identify the thermorheologically simplicity of materials [28,29,41]. Based on the
results of the DMA multi-frequency scans test (Figure 4), the wicket plot for the adhesive
investigated in this study is shown in Figure 5. The near-arch shape (near symmetrical curve)
indicates the thermorheologically simplicity, so as to confirm the applicability of TTSP
[28,29,41].



According to TTSP, as shown in Figure 6, the horizontal shift was applied to the individual measurements (Figure 4) to form the master curve at the reference temperature of 55 °C (Figure 6 (b)), which is the closest temperature step to the adhesive's onset T_g , where the modulus begins to drop significantly. The determined shift factors (α_T) are fitted to the well-known Williams-Landel-Ferry (WLF) equation (Figure 6 (c)) [28,30]:

208
$$log(\alpha_T) = \frac{-C_1(T - T_{ref})}{C_2 + (T - T_{ref})}$$
(1)

where *T* is the temperature of interest, $C_1 = 30.48$ and $C_2 = 151.14$ (°C) are obtained calibration constants. The corresponding shifted loss modulus using the same shift factors are illustrated in Figure 6 (d), showing a good continuity as well.

 $\frac{281}{282}$



A Prony series, expressing the generalised Maxwell model, was used for the linear viscoelastic model, which is an in-built material model in the Abaqus analysis software. The shifted storage modulus and loss modulus shown in Figure 6 were fitted to a Prony series in the frequency domain [43,44]:

224
$$E'(\omega) = E_0 \left(1 - \sum_{i=1}^n e_i \right) + E_0 \sum_{i=1}^n \frac{\omega^2 \tau_i^2 e_i}{\omega^2 \tau_i^2 + 1}$$
(2)

225
$$E''(\omega) = E_0 \sum_{i=1}^{n} \frac{\omega \tau_i e_i}{\omega^2 \tau_i^2 + 1}$$
(3)

where E_0 is the instantons elastic modulus, $E'(\omega)$ is the storage modulus, $E''(\omega)$ is the loss modulus, ω is the angular frequency, and *i* presents the number of terms in Prony series. τ_i and *e_i* are the parameters indicating the relaxation times and relaxation modulus respectively. In addition, in the Abaqus, a Prony series in time domain using the same parameters can be used to calculate the time-dependent modulus response [43,44]:

231
$$E(t) = E_0 \left[1 - \sum_{i=1}^n e_i \left(1 - e^{-t/\tau_i} \right) \right]$$
(4)

The fitting results are illustrated in Figure 7 and the obtained parameters are presented in Table 2. In general, the fitting is satisfactory, and it is typical for the Prony series that the fitting to the storage modulus is more accurate than the fitting to the loss modulus [16,28,42]. However, in many studies, the fitting to the loss modulus is ignored or not shown [21,25,30,31], which makes the accuracy of fitting unclear.



239

Figure 7: Comparison of Prony series fitting versus shifted experimental values

Table 2: Parameters of the obtained viscoelastic material model

| | Prony | y series | |
|---------------------|--------|-------------------|--------|
| $	au_i(\mathbf{s})$ | e_i | $	au_i$ (s) | e_i |
| 1×10 ⁻¹⁰ | 0.0224 | 1×10^{1} | 0.1295 |

| 1×10-9 | 0.0180 | 1×10^{2} | 0.1333 |
|--------------------|-----------|------------------------|---------------------|
| 1×10 ⁻⁸ | 0.0057 | 1×10^{3} | 0.1213 |
| 1×10-7 | 0.0137 | 1×10^{4} | 0.0724 |
| 1×10-6 | 0.0124 | 1×10 ⁵ | 0.0147 |
| 1×10-5 | 0.0084 | 1×10^{6} | 0.0043 |
| 1×10-4 | 0.0186 | 1×10^{7} | 0.0008 |
| 1×10-3 | 0.0196 | 1×10^{8} | 0.0007 |
| 1×10 ⁻² | 0.0136 | $\sum e_i =$ | 0.9836 |
| 1×10 ⁻¹ | 0.2141 | $E_0 = 3730 \text{ N}$ | 1Pa; $\mu_0 = 0.45$ |
| 1×10^{0} | 0.1601 | k_i | = 0 |
| WLF equation | | | |
| $T_{ref} = 55$ ° | C $C_l =$ | - 30.48 C ₂ | = 151.14 (°C) |

240

For simplicity, the bulk modulus (k_i) and Poisson's ratio (μ_0) of the adhesive are considered to be independent of time (Table 2), which is usually acceptable for polymers [16,43,44]. In Section 4, the above parameters are applied into the Abaqus FE software to analyse the joint creep behaviour caused by the thermal-viscoelastic response of the adhesive layer. The viscoelasticity of the adhesive is defined by entering each term of the Prony series (τ_i , e_i , and k_i) through the Abaqus *material-viscoelastic* option, while the temperature dependence is defined through the sub-option, *Trs*, using the WLF equation parameters (T_{ref} , C_1 , and C_2).

248

249 **3.2** Accuracy investigation of the obtained material model

Figure 8 illustrates a comparison between the predicted values of the viscoelastic material model (Prony series) and the original experimental values illustrated in Section 2.2.





254

Figure 8: Comparison of viscoelastic model predictions versus raw experimental values

As shown in Figure 8 (a), the material model developed through the TTSP shifting and curve fitting processes has a high accuracy in expressing the raw viscoelastic modulus response of the adhesive.

258 However, as shown in Figure 8 (b), compared to the modulus response in the DMA T_g test (Section 2.2.1), the predicted modulus decreases more slowly around the T_g . This could be due 259 to the further curing of the exaimed adhesive sample during DMA multi-frequency scans test 260 261 at high temperatures, which has received little attention in literatures. In real life applications, 262 the adhesive may also have further curing at elevated ambient temperatures during the longterm service, which to some extent counteracts the effect of further curing of the sample in the 263 264 accelerated chacatersing test. Further investigation of this issue was not possible within the current study, but nevertheless the comparison shown in here identifying the need for further 265 266 research.

267

268

8 4 Numerical study of the behaviour of the FRP-to-steel joint

This section presents an FE analysis of an adhesive-bonded single lap-shear joint to investigate the effect of elevated temperatures on the behaviour of bonded joints in civil 271 engineering applications. The constitutive material model developed in Section 3 was used to

determine the time-temperature dependent viscoelasticity of the adhesive layer.

273 Three temperature levels were applied:

- 30 °C, being a slightly warmer temperature compared to the ambient condition;
- 40 °C, which is a potential elevated temperature in civil engineering conditions;
- 50 °C, which is consistent with the maximum operating temperature stipulated in current
- 277 design guidelines, which should be a minimum of 15 °C below the peak Tan δT_g , giving

278 a 51.4 °C (66.4 °C - 15 °C) for the studied adhesive [7,23,24].

279

280 4.1 FE model for the adhesive-bonded single lap-shear joint

Figure 9 shows the geometry of the studied lap-shear joint, which was developed according
to BS 5350-C5 [45].



283 284

285 The joint was modelled in Abaqus using CPS4R planar stress elements with reduced 286 integration and hourglass control to limit the stress singularity effect [44,46]. Finer meshes 287 (0.1 mm) were applied to the area close to the overlap. The CFRP substrate was modelled as 288 elastic with a Yong's modulus of 170 GPa and a Poisson's ratio of 0.3, while the steel substrate 289 was modelled as elasto-plastic with a Young's modulus of 205 GPa, a yield strength of 355 290 MPa and a Poisson's ratio of 0.3. The adhesive layer was defined as viscoelastic material. The 291 temperature was uniformly applied to all parts of the model. The load of 1 kN was applied to 292 the CFRP substrate, which can result in a high instantaneous stress in the adhesive joint (but 293 below the strength of the adhesive [32]) to investigate the behaviour of stress redistribution.

Note that this study focuses on the impact of viscoelastic creep of the structural adhesive on the response of the FRP-to-steel bonded joint. The joint damage related to the strength of the joint is not within the scope of this study, as a result, perfect bond was assumed between different sections.

298

299 **4.2** The effect of viscoelasticity of the adhesive on the bonded joint

Figure 10 illustrates the creep compliances of the studied adhesive at elevated temperatures, which were obtained from a single cube element (C3D8) model, whilst it can also be calculated from equation (1) and (4).



303 304

Figure 10: Creep compliances of the viscoelastic adhesive at various temperatures

The constitutive material model is sufficient to predict creep over the temperature range examined up to 50 years. After 50 years at 30 °C, the creep compliance of the adhesive increases almost as much as that after only 1 month at 40 °C, and even less than that after 1 day at 50 °C. Temperature has a great influence on the viscoelastic creep response of the adhesive, which will therefore affect the long-term behaviour of the adhesive bonded joint.

Figure 11 shows the effect of adhesive creep at 50 °C on the distribution of shear strain (LE12) and shear stress (S12) of the single lap-shear bonded joint. After 50 years (Figure 11 (b) and (d)), compared to the instantaneous behaviour of the joint (Figure 11 (a) and (c)), creep

- 313 of the adhesive leads to a significant increase in the shear strain, however, it is accompanied
- 314 by a redistribution behaviour of the shear stress, which reduces the concentrated shear stress at
- 315 the edges of the joint and increases the effective stress transfer length.





327 328

Figure 12: The shear strain and shear stress at the edge of the adhesive joint varying with time at different temperatures At 30 °C, the maximum shear stress starts to decrease significantly after about 1 day due to the redistribution of stresses resulting from the viscoelastic behaviour of the adhesive layer. After 50 years, the maximum shear stress decreases by almost 37 % from about 23 MPa to 14.5 MPa, and the shear strain only increases by less than 0.001. This could be beneficial for the long-term performance of the bonded joint.

At 40 °C, the maximum shear stress begins to decrease within a few minutes, however, it is accompanied by a sharp rise in the rate of shear strain increase. Whilst the maximum shear stress decreases to about 6.5 MPa after 50 years, the maximum shear strain increased by more than three times compared to the results at 30 °C.

At 50 °C, the maximum shear stress and shear strain change significantly within the first year of use. This may indicate that the first year of service life of the bonded structure is critical under high operating temperature conditions, as creep in the joint may introduce significant shear strain, resulting in limited effectiveness of the bonded strengthening and the potential of early debonding failure.

344

345 **5** Conclusions

| 346 | In this study, a viscoelastic material model was developed for a typical structural epoxy |
|-----|---|
| 347 | adhesive (Sikadur®-330 [32]) by applying an accelerated test method involving dynamic |
| 348 | mechanical analysis (DMA) and time-temperature superposition principle (TTSP), which was |
| 349 | utilised to investigate the effects of elevated temperatures on the response of the adhesive- |
| 350 | bonded FRP-to-steel joint. |
| 351 | The process of the accelerated tests has been illustrated and the limitations of applying this |
| 352 | method have been discussed: |
| 353 | • The thermal stability of studied samples should be examined prior to performing DMA |
| 354 | characterisation tests at high temperatures (> 100 °C). |
| 355 | • Materials that obey TTSP should be thermorheologically simple, which can be |
| 356 | confirmed using the wicket plot. |
| 357 | • DMA tests may be challenging to characterise the nonlinear viscoelasticity of the |
| 358 | sample, and the sample can be further cured in high temperature tests. |
| 359 | Whilst the accelerated test method has some limitations and requires further research, it |
| 360 | remains a viable option for estimating the long-term behaviour of bonded joints, as it is |
| 361 | impractical to carry out the conventional sustained load creep tests for decades to fully cover |
| 362 | the design life of civil engineering structures. |
| 363 | The numerical study results of the single lap-shear FRP-to-steel joint indicate that: |
| 364 | • Elevated temperatures have a great influence on the creep response of the adhesive layer. |
| 365 | The creep can lead to a significant increase in the shear strain of the adhesive joint |
| 366 | accompanied by a redistribution behaviour of the shear stress, which reduces the |
| 367 | concentrated shear stress at the edges of the joint and increases the effective stress |
| 368 | transfer length. |
| | |

A slightly warmer temperature (30 °C for this study) than ambient environments is
considered to be beneficial for the bonded joint as it can reduce the concentrated shear
stress (by 37 % after 50 years) with negligible increase in the shear strain (by 0.001
after 50).

However, higher temperatures (≥40 ° C in this study) significantly increase the creep
rate of the adhesive, resulting in a multifold increase in shear strain of the joint with
limited reduction in the shear stress, which can be detrimental.

For cyclic (oscillating) temperature conditions, when the maximum temperature exceeds 40 °C, attention should also be paid to the detrimental effects of viscoelastic creep on bonded joints [47].

The available characterisation data and material models for adhesives used in civil engineering are limited. The developed viscoelastic model can be used for further numerical analysis as long as the applied adhesive is the same and has a similar T_g . For other structural adhesives, it will be necessary to carry out similar tests and modelling work (described in Section 2 and 3) to develop the corresponding material models.

The long-term creep failure of the joints tends to lead to interface fracture in FRPstrengthened structures, which has so far lacked satisfactory explanations and predictions. The constitutive material model developed in this paper focuses on analysing the viscoelasticity of the structural adhesive, whereas for the future study, the plasticity and the damage of the FRPto-steel bonded joint will also be significant to investigate.

389

390 CRediT authorship contribution statement

391 S. Wang: Methodology, Formal analysis, Investigation, Writing - original draft,
 392 Visualization. T. Stratford: Conceptualization, Resources, Writing - review & editing. T.P.S.
 393 Reynolds: Writing - review & editing.

394

395 Declaration of competing interest

396 The authors declare that they have no known competing financial interests or personal 397 relationships that could have appeared to influence the work reported in this paper.

398

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