

1 Fabrication of Endothelial Cell-Laden Carrageenan Microfibers for 2 Microvascularized Bone Tissue Engineering Applications

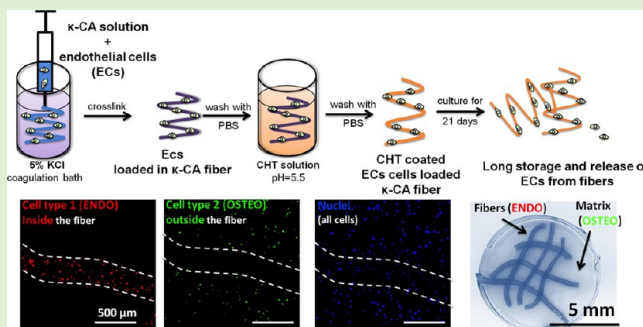
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7 **S** Supporting Information

8 **ABSTRACT:** Recent achievements in the area of tissue
9 engineering (TE) have enabled the development of three-
10 dimensional (3D) cell-laden hydrogels as in vitro platforms
11 that closely mimic the 3D scenario found in native tissues.
12 These platforms are extensively used to evaluate cellular
13 behavior, cell–cell interactions, and tissue-like formation in
14 highly defined settings. In this study, we propose a scalable and
15 flexible 3D system based on micro-sized hydrogel fibers that
16 might be used as building blocks for the establishment of 3D
17 hydrogel constructs for vascularized bone TE applications. For
18 this purpose, chitosan (CHT) coated κ -carrageenan (κ -CA)
19 microfibers were developed using a two-step procedure
20 involving ionotropic gelation (for the fiber formation) of κ -CA and its polyelectrolyte complexation with CHT (for the
21 enhancement of fiber stability). The performance of the obtained fibers was assessed regarding their swelling and stability
22 profiles, as well as their ability to carry and, subsequently, promote the outward release of microvascular-like endothelial cells
23 (ECs), without compromising their viability and phenotype. Finally, the possibility of assembling and integrating these cell-laden
24 fibers within a 3D hydrogel matrix containing osteoblast-like cells was evaluated. Overall, the obtained results demonstrate the
25 suitability of the micro-sized κ -CA fibers to carry and deliver phenotypically apt microvascular-like ECs. Furthermore, it is shown
26 that it is possible to assemble these cell-laden micro-sized fibers into 3D heterotypic hydrogels constructs. This in vitro 3D
27 platform provides a versatile approach to investigate the interactions between multiple cell types in controlled settings, which
28 may open up novel 3D in vitro culture techniques to better mimic the complexity of tissues.



1. INTRODUCTION

29 In the past decade, tissue engineering (TE) has emerged as a
30 multidisciplinary field at the interface of medicine, biology, and
31 engineering, aiming at fabricating tissue-like biological con-
32 structs.¹ However, the lack of a vasculature that can sustain the
33 nutrients and oxygen demands within the tissue-engineered
34 construct is a major limiting factor in creating thick artificial
35 tissues.² Thus, developing vessel-like networks based on
36 endothelial cells (ECs), as integrated templates within TE
37 constructs,³ would be essential for creating real-size replicas of
38 tissues or organs.^{4,5}

39 Hydrogels have been proven to provide ideal mimics of cellular
40 matrices, as their hydrated state resembles that of native
41 extracellular matrix (ECM),⁶ and their high permeability to
42 oxygen, nutrients, and metabolites diffusion⁷ enables cell
43 encapsulation. Additionally, natural-origin hydrogels showed
44 high potential in the TE field, due to their chemistry and
45 properties similar to ECM.⁸ Several processing methodologies
46 such as prototyping/printing,⁹ microfluidics,¹⁰ and photo-
47 lithography¹¹ have been used to develop vessel-like architectures
48 based on ECs-loaded hydrogels that exhibit micro-sized features
49 (~50 μm¹¹ to 1 mm⁹). However, these methods often require

sophisticated equipment and elaborated experimental settings 50
and reagents that frequently raise concerns over their 51
cytotoxicity.^{12,13} 52

Wet spinning of hydrogels is a very simple and straightforward 53
method, requiring minimal laboratorial utensils and short 54
processing time. The wet spinning of hydrogel fibers involves 55
the extrusion of a polymer solution through a needle into a 56
coagulation bath that triggers the cross-linking of the polymer 57
into a fiber-like shape. This methodology has been successfully 58
used to develop hydrogel fibers based on several natural-origin 59
polymers, such as alginate,¹⁴ collagen,¹⁵ gellan gum,¹⁶ chitosan,¹⁷
60 or a combination of carrageenan and alginate.¹⁸ 61

Carrageenans (CAs) stand out, among other natural-origin 62
polymers, as potential candidates for TE applications, due to 63
their mild gelation properties and resemblance to glycosamino- 64
glycans (GAGs), main components of the ECM of biological 65
systems. CAs are highly sulfated linear polysaccharides with 66
alternating repeating 3,6-anhydro-D-galactose and β-D-galactose- 67

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68 4-sulfated units, that provide structural support in several species
69 of marine red algae (class of Rhodophyceae). Due to the half-
70 ester sulfate moieties present on their backbone (their amount
71 determines the type of carrageenan that can be iota, lambda, or
72 kappa), CAs are strong anionic polymers. As a consequence, their
73 gelation occurs by ionic interactions with appropriate counter-
74 ions, such as K^+ , Na^+ , or Ca^{2+} . Although these hydrophilic
75 polysaccharides have been widely used as emulsifiers, gelling,
76 thickening, or stabilizing agents in the food or pharmaceutical
77 industry,¹⁹ the intrinsic thixotropic behavior of κ -carrageenan (κ -
78 CA), in particular, has justified its exploitation as an injectable
79 matrix for the delivery of living cells^{18,20,21} and biomacromole-
80 cules.^{22,23} However, κ -CA hydrogels exhibit high swelling ratios
81 and mechanical instability in physiological conditions²⁴ which
82 has motivated the development of several approaches to increase
83 their stability, such as chemical modifications with photo-cross-
84 linkable moieties,²⁵ blending with other biopolymers,^{18,26}
85 addition of nanocomposites to the polymer solution,²³ and
86 formation of interpenetrating networks²⁷ or polyelectrolyte
87 complexation (PEC) with polycations, such as chitosan
88 (CHT).²⁸ Numerous studies have reported CHT-based PEC
89 systems with positive outcomes, in terms of stability as well as
90 cellular viability and behavior.^{29,30} Electrostatic interactions
91 between κ -CA and CHT have also been used for the
92 development of nanoparticles,³¹ beads,³² and layer-by-layer
93 systems.³³

94 The present work reports on the development of κ -CA
95 microfibers coated with CHT using a two-step-procedure, and
96 on their potential as cell carriers and building blocks of 3D
97 constructs for vascularized bone tissue engineering. Therefore, in
98 a first step, κ -CA fibers of various diameters were obtained by wet
99 spinning. In a second step, in order to reinforce the fibers and
100 enhance their stability in physiological relevant microenviron-
101 ment, κ -CA fibers were coated with CHT, through the
102 electrostatic interaction between the polymers. The obtained
103 fibers were then loaded with microvascular-like ECs obtained
104 from the SSEA-4⁺ subpopulation of adipose tissue stromal
105 vascular fraction, as described in a previous study.³⁴ The
106 phenotype of the ECs was evaluated prior to and after
107 entrapment within the fibers. Ultimately, envisioning a
108 heterotypic 3D bone TE construct with independent and
109 defined microarchitectures, we proposed an innovative 3D
110 buildup of ECs loaded fibers within a hydrogel matrix containing
111 osteoblast-like cells. In summary, this study showed that it is
112 possible to developed cell-laden κ -CA-based hydrogel fibers that
113 exhibit high stability in prolonged culture conditions. Fur-
114 thermore, these fibers could be stacked in a desired pattern and
115 successfully integrated into a 3D construct.

2. MATERIALS AND METHODS

116 **2.1. Materials.** κ -Carrageenan (κ -CA), potassium chloride (KCl),
117 and β -glycerophosphate disodium salt hydrate (β GP) were purchased
118 from Sigma, Germany. Reagent grade medium molecular weight
119 chitosan (CHT) (Sigma, Germany) with a 90% degree of acetylation
120 was used. Prior to use, CHT was purified using a precipitation method.³⁵
121 All other reagents were used as received.

122 **2.2. Development of CHT Coated κ -CA Fibers.** **2.2.1. Production**
123 **of κ -CA Fibers through Ionotropic Gelation.** The κ -CA hydrogel fibers
124 were obtained by wet spinning technique, which consisted of the
125 extrusion of the polymer solution through a needle immersed in a
126 coagulation bath, as previously described elsewhere.¹⁸ Briefly, a 1.5%
127 (wt/v) κ -CA solution was prepared by dissolving the polymer in distilled
128 water at 50 °C and under constant stirring until complete dissolution
129 was achieved. Subsequently, the κ -CA solution was loaded into 5 mL

130 syringes and κ -CA fibers with different diameters were obtained by
131 extruding the polymeric solution through needles with different gauges
132 (18G to 27G), directly into a 5% (wt/v) KCl solution prepared in
133 distilled water. The presence of K^+ ions initiated the ionotropic gelation
134 by counterbalancing the negative charges of κ -CA. The fibers were
135 allowed to harden in the coagulation bath for about 10 min, sufficient
136 time for them to retain the shape. Finally, the fibers were washed with
137 phosphate buffered saline (PBS) in order to remove the excess of salt.
138 Fibers obtained with needles of 25G and 27G were selected for all the
139 subsequent assays.

2.2.2. Optimization of the pH of the CHT Working Solution.

140 Chitosan dissolves in acid solutions, which limits its use in the presence
141 of living cells. Previous studies³⁶ have shown that β GP, a weak base,
142 increases the pH of CHT solutions, without jeopardizing its solubility.
143 Therefore, in order to establish optimal conditions for the incorporation
144 of cells, while enabling PEC, a curve of variation of pH with the addition
145 of β GP to the CHT solution was obtained. For this purpose, a CHT
146 solution was prepared by dissolving CHT in a 1% (v/v) acetic acid
147 solution at a final concentration of 0.5% (wt/v). In order to determine
148 the degree to which the addition of β GP affects the overall charge of
149 CHT solution, zeta potential measurements were performed using a
150 Malvern Zeta Sizer Nano ZS (Malvern Instruments, UK). Each sample
151 was diluted in ultrapure water at a concentration of 0.1% (wt/v) and
152 analyzed at 25 °C for 60 s.

2.2.3. Coating of κ -CA Fibers through Polyelectrolyte Complexation with CHT.

154 The κ -CA fibers, previously obtained by ionotropic
155 gelation, were immersed in the optimized CHT solution (0.5% (wt/v)
156 and pH = 5.5) for 20 min, followed by several washing steps with PBS to
157 remove the excess of CHT. The presence of the CHT coating was
158 evaluated by staining the fibers with Eosin Y (Sigma, Germany). Fibers
159 without coating were used as negative control.

2.3. Physico-Chemical Characterization of the Developed

Fibers. 2.3.1. Swelling Kinetics.

161 The influence of CHT coating on the
162 swelling and stability of the developed κ -CA fibers was determined by
163 evaluating the fibers' water absorption kinetics and diameter variation
164 upon immersion in culture medium for up to 21 days. The culture
165 medium used for this purpose was Dulbecco's Modified Eagle Medium
166 (DMEM, Gibco, USA) supplemented with 10% (v/v) HiFBS (Gibco,
167 USA) and 1% (v/v) antibiotic/antimycotic (penicillin/streptomycin,
168 100 U/100 μ g/mL, Gibco, UK). Envisioning the use of fibers for cell
169 encapsulation/culturing, the experimental parameters were set at
170 physiological temperature (37 °C) and humidified atmosphere with
171 5% of CO_2 . Medium was replenished every 3–4 days. Fibers ($n = 3$)
172 were retrieved at days 7, 14, and 21, blotted with KimWipe paper to
173 remove the excess of liquid, and weighed (M_w). Sample's final weight
174 (M_{DF}) was determined upon lyophilization. The swelling kinetics was
175 defined as the ratio between the liquid uptake ($M_w - M_{DF}$) and the final
176 dry mass of polymer (M_{DF}), according to eq 1.

$$\text{mass swelling ratio} = (M_w - M_{DF})/M_{DF} \times 100 \quad (1)$$

177 The final diameter of the hydrogel fibers was also measured applying
178 software-measuring tools (NIH ImageJ software, <http://rsbweb.nih.gov/ij/>) to at least three micrographs of each sample.

2.3.2. Morphological and Chemical Characterization.

181 Concomitant with the swelling behavior analysis, samples were retrieved for
182 surface morphological evaluation and elemental analysis under a
183 scanning electron microscope (SEM) coupled with energy-dispersive
184 X-ray spectroscopy (EDX/EDS). Briefly, fibers were snap frozen in
185 liquid nitrogen, transferred to microcentrifuge tubes, and freeze-dried
186 overnight. The dried samples were carefully mounted on sample holders
187 using double-sided carbon tape. Before being analyzed by SEM (Nano-
188 SEM FEI Nova 200), the samples were gold sputter coated (Fisons
189 Instruments, sputter coater SC502, UK). Elemental analysis was carried
190 out with an energy dispersive spectrometer (EDAX-Pegasus X4M) on
191 the same samples used for SEM. All observations/image acquisitions
192 and measurements were made at an acceleration voltage of 15 kV.

2.4. Isolation and Endothelial Differentiation of SSEA-4⁺hASCs.

196 Human lipoaspirate samples from healthy donors were
197 kindly provided by Hospital de Prelada (Porto, Portugal), under
198 previously established protocols and with the informed consent of the

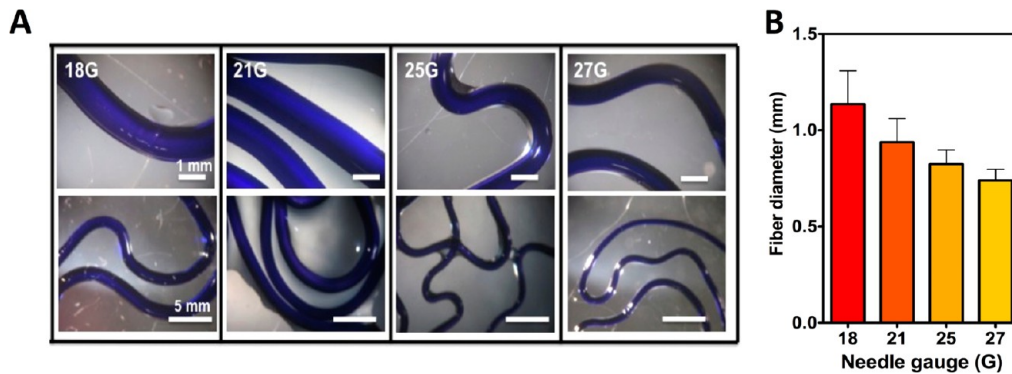


Figure 1. K-CA fibers formed by ionotropic gelation. Needles with different gauges, from 18G to 27G, were used to form the hydrogel fibers. (A) Aspect of fibers using different needle gauges. Staining with methylene blue was performed for contrast purposes. (B) Depending on the needle gauge, fibers diameter spanned from 1.25 mm down to 0.5 mm.

199 patients. The selection of SSEA-4 positive cells (SSEA-4⁺hASCs)
 200 residing within the stromal vascular fraction of the adipose tissue and
 201 differentiation toward the endothelial lineage were performed according
 202 to our previously published protocols.³⁴ Briefly, the SSEA-4⁺hASCs
 203 were selected using immunomagnetic beads (DynaM-450 Epoxy beads
 204 from Dynal Biotech, Carlsbad, CA, USA) coated with SSEA-4 antibody
 205 (Abcam, Cambridge, UK). Then, the selected SSEA-4⁺hASCs cells were
 206 differentiated toward endothelial lineage by culturing them for 2 weeks
 207 in endothelial cell growth medium, EGM-2 MV (Lonza, Switzerland),
 208 containing 5% FBS and supplemental growth factors: hydrocortisone,
 209 hFGF-B, ascorbic acid, gentamicin/amphotericin, VEGF, long R3-IGF-
 210 1, hEGF, and heparin, at concentrations established by the
 211 manufacturer.

212 **2.5. SSEA-4⁺hASCs-Derived Endothelial Cells Encapsulation**
 213 **within κ -CA Fibers.** The κ -CA, CHT, and KCl solutions were
 214 sterilized at 120 °C for 30 min. SSEA-4⁺hASCs-derived ECs at passage 3
 215 were trypsinized and centrifuged, and further suspended in the κ -CA
 216 solution at a final cell density of 2×10^6 cells/mL. ECs loaded fibers
 217 (with and without CHT coating) were produced as described above.
 218 The cell-loaded fibers were then transferred to 24-well plates and
 219 maintained in culture in EGM-2 MV for 21 days, at 37 °C in a humidified
 220 atmosphere with 5% CO₂. Culture medium was replenished every 3–4
 221 days.

222 In order to assess the maintenance of the phenotype of the SSEA-
 223 4⁺hASCs derived ECs at 21 days of culture cell-loaded fibers were
 224 treated with 0.1% proteinase K (vWR, Portugal) in 1 mM EDTA
 225 (Sigma, Germany), 50 mM TrisHCl (Sigma, Germany), and 1 mM
 226 iodoacetamide buffer (Sigma, Germany), for 1 h at 37 °C, under
 227 constant agitation, to release the cells from the fibers. The cellular pellet
 228 recovered after centrifugation (10 min, 400×g) was resuspended in
 229 EGM-2 MV medium and plated into tissue culture flasks. Cells were
 230 cultured until reaching confluence for further analysis, as described
 231 below.

232 **2.6. Characterization of the Constructs.** **2.6.1. Microscopic**
 233 **Analysis.** Variations in the shape and diameter of the developed cell-
 234 loaded fibers (with or without CHT coating), as well as the potential
 235 outward migration of the cells to the culture well, were examined using a
 236 stereomicroscope (Stemi 1000, Zeiss, Germany) along the time of
 237 culture.

238 **2.6.2. Live–Dead Assay.** At preselected time culturing points (1, 7,
 239 14, and 21 days), cell-loaded fibers were washed with PBS and incubated
 240 with 4 μ M calcein-AM (Invitrogen, USA) for 40 min followed by 10 min
 241 incubation with 1 μ M propidium iodide (PI, Invitrogen, USA). Samples
 242 ($n = 3$) were then washed and fixed for 40 min in 10% formalin. After
 243 fixation, samples were washed with PBS and cell nuclei were
 244 counterstained for 20 min with 4,6-diamidino-2-phenylindole dilactate
 245 (DAPI, Sigma, Germany) and then washed three times with PBS.
 246 Representative fluorescent micrographs were acquired using the
 247 Axioplan Imager Z1 fluorescence microscope (Zeiss, Germany) and
 248 the AxioVision 4.8 software (Zeiss, Germany). The quantification of
 249 cellular viability was measured by applying software-measuring tools

(NIH ImageJ software, <http://rsbweb.nih.gov/ij/>) to at least three
 250 micrographs of each sample. 251

252 **2.6.3. Flow Cytometry.** SSEA-4⁺hASCs-derived ECs, prior to and
 253 after encapsulation, were retrieved from cell culture flasks using TrypLE
 254 Express (Invitrogen, USA). 5×10^5 cells were incubated for 30 min at 4
 255 °C with the following markers: CD45-FITC, CD34-PE, CD73-PE,
 256 CD31-APC (all from BD Pharmingen, USA), and CD105-FITC and
 257 CD90-APC (eBiosciences, USA) at a concentration of 6 μ g/mL, as
 258 recommended by the manufacturer. After washing with PBS, the cells
 259 were resuspended in acquisition buffer (PBS containing 1% form-
 260 aldehyde and 0.1% sodium azide) and analyzed with a BD FACS-Calibur
 261 flow cytometer (BD Biosciences, USA). A minimum of 2×10^5 events
 262 was acquired and gated in a forward versus side-scatter dot plot with a
 263 linear scale. Results were displayed in histogram plots created using
 264 CellQuest software (BD Biosciences, USA). The number of positive
 265 events for each cell-specific marker was expressed as a percentage of the
 266 total cell number.

267 **2.6.4. Matrigel Assay.** SSEA-4⁺hASCs-derived ECs, at passage 2,
 268 were trypsinized and plated at a density of 3×10^4 cells/well in 48-well
 269 plates coated with 64 μ L of Matrigel (BD Biosciences, USA). Cells were
 270 incubated for 4 h at 37 °C. One hour prior fixation, 4 μ M calcein-AM
 271 (Invitrogen, USA) was added to the wells. Upon fixation, cell nuclei
 272 were counterstained with DAPI for 10 min, and then washed three times
 273 with PBS. Three representative images were acquired under a
 274 fluorescence microscope.

275 **2.7. Assembling the κ -CA Fibers into 3D Hydrogel Discs.** Cell-
 276 laden fibers containing SSEA-4⁺hASCs-derived ECs labeled with a GFP
 277 tag were transferred to a Petri dish and allowed to settle randomly. A
 278 freshly prepared κ -CA solution containing osteogenic differentiated
 279 SSEA-4⁺hASCs (obtained according to a method described else-
 280 where³⁴) and labeled with a rhodamine tag was poured onto the fibers
 281 until full coverage. The cross-linking of the κ -CA solution was achieved
 282 with a 5% (wt/v) KCl solution. Immediately after cross-linking, the
 283 hydrogel disc containing the fibers was fixed with 10% formalin for 40
 284 min. After fixation, the cell nuclei were stained with DAPI for 20 min and
 285 observed under a confocal laser scanning microscope (CLSM, Olympus,
 286 Fluoview 1000). YZ and XY projections were performed in order to
 287 evaluate the cellular distribution throughout the 3D structures.

288 **2.8. Statistical Analysis.** For the swelling ratio and fiber diameter
 289 data, statistical analysis was performed using GraphPad Prism 5.00
 290 software (San Diego, USA). Statistical differences ($p < 0.05$) were
 291 determined using one-way ANOVA, followed by a Tukey post test.

3. RESULTS AND DISCUSSION

292 **3.1. Different Diameter κ -CA Fibers Formation through**
 293 **Ionotropic Gelation.** Cell encapsulation techniques using
 294 hydrogels enable the formation of 3D cell-culture models that
 295 can potentially replicate tissue organization, which cannot be
 296 achieved in conventional 2D cultures.³⁷ Since the architecture
 297 and chemical composition of hydrogels can be easily engineered,

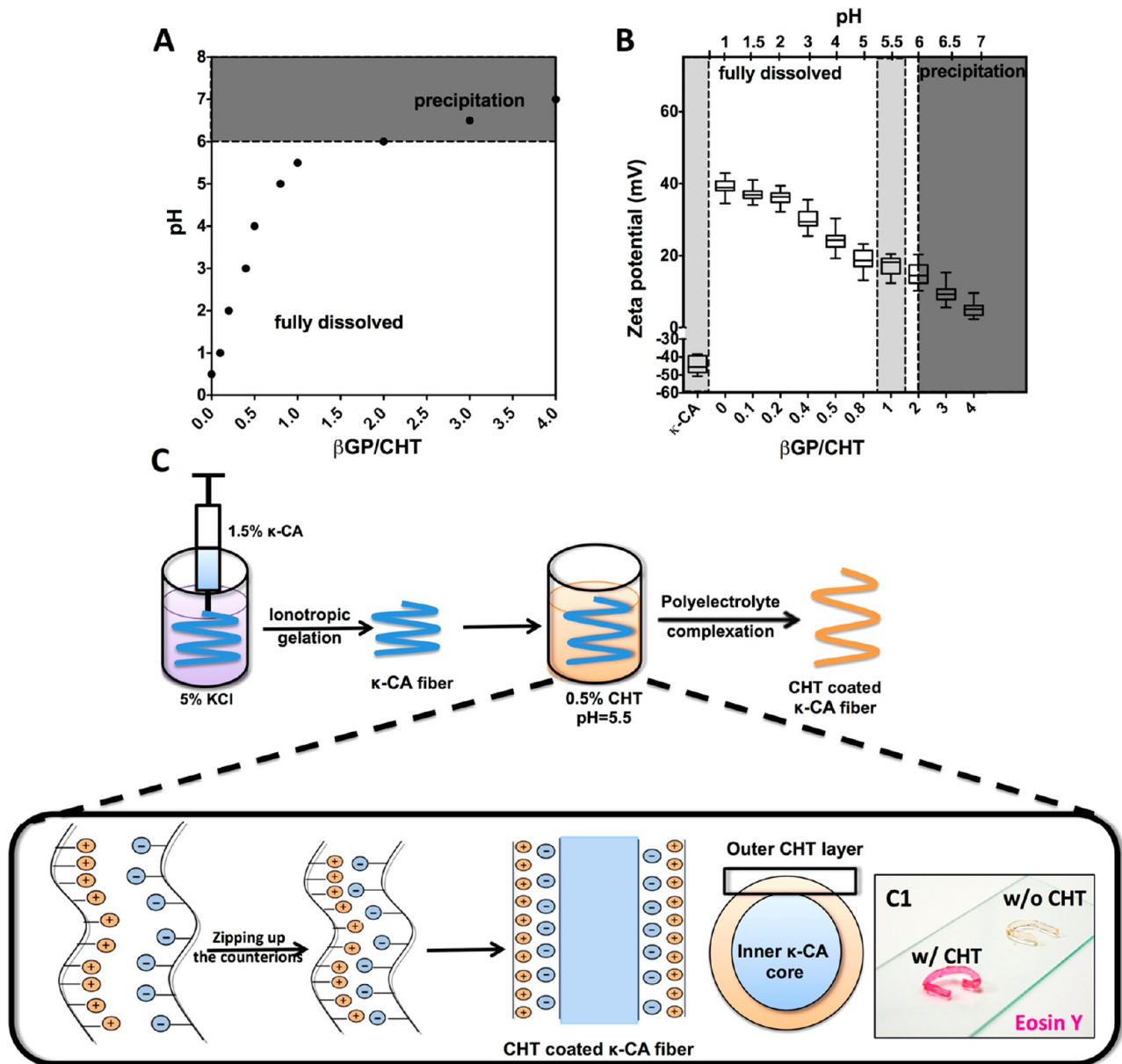


Figure 2. Production of CHT coated κ -CA hydrogel fibers. (A) Variation of the pH relative to the β -glycerophosphate (β GP)/chitosan (CHT) ratio. With the addition of β GP, the CHT solutions reach physiologically relevant conditions (pH 5.5–6). Above this point the precipitation of CHT occurs (dark gray area). (B) With the addition of β GP, the zeta potential of CHT solution is dramatically affected. The zeta potential of κ -CA solutions, measured at pH = 5.5, was used as reference (approximately -40 mV). Values reported correspond to $n = 10$. (C) Schematics of the production of CHT coated κ -CA hydrogel fibers. The process involves the formation of fibers by ionotropic gelation in a KCl coagulation bath, followed by their immersion in an optimized CHT solution. (C1) The presence of the CHT coating was confirmed by staining the fibers with Eosin Y which specifically binds to CHT.

298 they can be used as tools and platforms to design capture/release
299 of cells in/from 3D tissue-engineered constructs.³⁸

300 Thus, we aimed to develop hydrogel microfibers by making
301 use of a simple and straightforward procedure that enables cell
302 encapsulation in a controlled distribution pattern, without
303 jeopardizing their phenotype and functionality. Among the
304 natural origin polymers, such as alginate,^{14,13} collagen,¹⁵ gellan
305 gum,^{16,39} or hyaluronic acid⁴⁰ that have been used to obtain
306 fibers through wet spinning technique, κ -CA stands out due to its
307 GAG-like features, reversible temperature and ionic gelation
308 properties and intrinsic shear thinning behavior, particularly
309 thixotropic. This means that upon applied stress the organization
310 of κ -CA chains is disrupted, but it will reset once the deformation

is removed.²⁵ This property renders the use of κ -CA hydrogels as
injectable matrixes, as they can be easily extruded through narrow
needles without affecting the structure or the entrapped
biomacromolecules or cells.^{41,42} Taken together, the gelation
properties of κ -CA enable the formation of gels in different
shapes, including fibers, highlighting therefore the versatility of κ -
CA processability.⁸

In the present study, the possibility of producing κ -CA fibers
with different diameters was first investigated. To obtain the fiber
shape, κ -CA was extruded directly into the coagulation bath,
where the cross-linking of the polymer solution occurred.
Variation of the diameter depended on the needle gauge used to
extrude the κ -CA solution into the coagulation bath (Figure 1A).

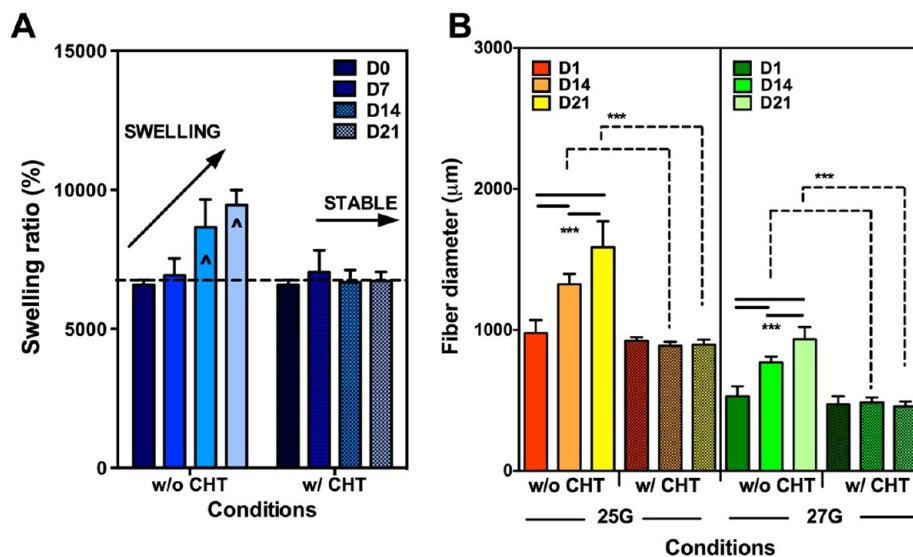


Figure 3. Swelling behavior of κ -CA fibers, without (w/o) and with (w/) CHT coating, in culture medium (DMEM). (A) Swelling ratio increases over time for the fibers without CHT coating. On the other side, the CHT coating stabilizes the hydrogel fibers, hampering their swelling. Values correspond to average ($n = 3$) \pm standard deviation. $\hat{\wedge}$ corresponds to a statistical difference ($p < 0.05$) when compared to day 0 and day 7 values of the κ -CA fibers without coating. (B) The diameter of the developed fibers is a direct consequence of the swelling behavior. Thus, the uncoated κ -CA fibers show a significant increase in the fiber diameter along time ($***p < 0.001$), while the diameter of coated fibers remains constant.

324 Using needle gauges ranging from 18 to 27G, it was possible to
 325 obtain fibers with different diameters, directly proportional to
 326 the internal diameter of the needles (838–210 μm). The fibers
 327 produced within these settings had a diameter ranging from 500
 328 to 1250 μm (Figure 1B), making them appealing for further
 329 applications.^{13,14,43} These trends were in agreement with
 330 previous reports for hydrogel fibers obtained via wet spinning¹⁸
 331 or other techniques.¹⁴

332 Taking into consideration that a micro-sized architecture
 333 enables nutrients and oxygen diffusion, crucial for the cellular
 334 performance, we decided to further explore the potential κ -CA
 335 fibers with the smallest diameters, obtained by the extrusion of κ -
 336 CA solutions through the 25G and 27G needles.

337 **3.2. κ -CA Fibers Can Be Coated with CHT through**
 338 **Polyelectrolyte Complexation.** Although encouraging results
 339 have shown the potential of using κ -CA in TE applications, the
 340 ionically cross-linked κ -CA hydrogels exhibit high swelling ratios.
 341 It is possible that the culture medium destabilizes the physically
 342 cross-linked network, as a result of the uncontrollable and
 343 permanent exchange of K^+ ions with other positive ions present
 344 in the physiological environment.²⁴ This allows the inner
 345 network to continuously expand, and consequently become
 346 loose and mechanically weak.^{25,44} If one envisions the prolonged
 347 culture periods or the maintenance of initial microfeatures of the
 348 hydrogels, such as shape, size, and stability, the reinforcement of
 349 the structure is an impetus.

350 Several methodologies have been employed,^{18,25,27} including
 351 the formation of polyelectrolyte complexes formed by the ionic
 352 interaction between oppositely charged polyelectrolytes. We
 353 hypothesized that, by coating the κ -CA hydrogels fibers with
 354 CHT, we would be able to stabilize the fibers structure. CHT, a
 355 β -1,4-linked D-glucosamine and N-acetyl-D-glucosamine, is one
 356 of the few positively charged natural-origin polysaccharides and,
 357 thus, is widely used as a polycation for the formation of
 358 polyelectrolyte complexes. Additionally, its appealing properties
 359 such as biocompatibility, biodegradability, low toxicity, and
 360 relatively low production cost from abundant sources endorse its
 361 use for TE strategies.⁴⁵

362 Since the amino groups of CHT have a pK_a value of 6.5–6.8,⁴⁶
 363 their protonation occurs in acidic solutions with a charge density
 364 dependent on pH and degree of acetylation. Besides causing
 365 CHT to become water-soluble, the protonated groups can
 366 readily bind to the negatively charged groups (carboxylic acid,
 367 hydroxyl, or sulfate) of other polymers, to form polyelectrolyte
 368 complexes. Thus, our hypothesis was that the protonated amino
 369 groups of CHT would bind to the available negatively charged
 370 sulfate groups of κ -CA and form an acid–base type of
 371 polyelectrolyte complex. Therefore, we have deliberately chosen
 372 shorter gelation times for the formation of κ -CA fibers (10 min
 373 over standard 30 min^{18,41}), so that the fiber shape was retained,
 374 without neutralizing all sulfate groups.

375 However, since the growth of cells is usually optimal when the
 376 microenvironment is buffered at a pH in the range 7.2–7.4,⁴⁷ the
 377 low pH values at which CHT dissolves might compromise their
 378 viability. Thus, we explored the possibility of increasing the pH of
 379 CHT solutions without jeopardizing their solubility, by adding a
 380 week base, βGP ,³⁶ to the solutions. We noticed that, by gradually
 381 adding βGP to the CHT solutions, the pH slowly increased
 382 without evident signs of precipitation (Figure 2A). Nevertheless,
 383 upon reaching the pK_a value, precipitation occurred.

384 One concern associated with the increase of pH was that the
 385 amino groups of CHT would be deionized, which would weaken
 386 their binding affinity to the oppositely charged κ -CA, jeopardizing
 387 the stability of the layer. To determine whether this was the
 388 case, we evaluated the electrical charge of the CHT solution upon
 389 addition of βGP , and consecutively upon increase of pH, by
 390 measuring the zeta potential. As expected, with the increase of
 391 the pH, the overall charge of the CHT solution decreased,
 392 though still remaining within the positive range (approximately
 393 +10 eV) (Figure 2B). These findings indicate that with the pH
 394 modification, the protonation of amino groups still occurs, and
 395 consequently CHT would continue to act as a polycation, being
 396 suitable for PEC with κ -CA. In the perspective of coating cell-
 397 loaded κ -CA fibers, pH of 5.5, where no signs of precipitation
 398 were observed and the overall charge of CHT was positive, was
 399 considered the most suitable condition.

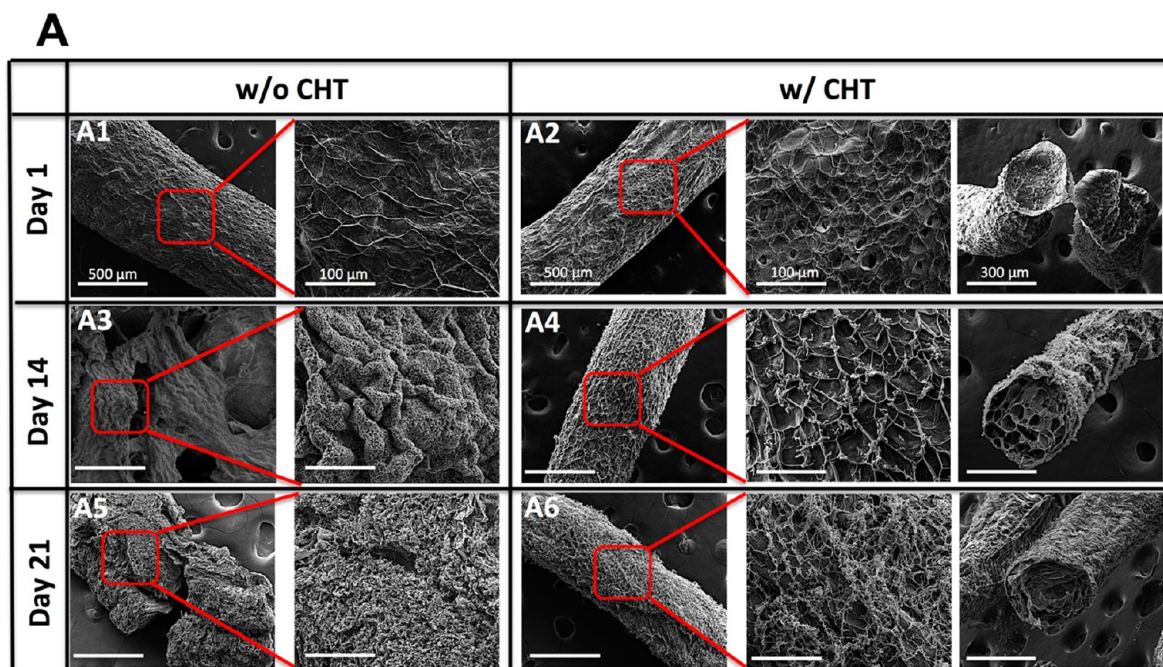


Table 1. Elemental composition of κ -CA microfibers with or without CHT coating

Time (days)	Condition	EDX/EDS elemental analysis (%wt)					
		C	O	S	K	Cl	N
1	w/o CHT	26.01	25.82	4.49	9.64	22.32	-
	w/ CHT	35.40	31.67	4.04	1.33	13.19	5.79
14	w/o CHT	31.27	18.8	3.85	7.90	25.15	-
	w/ CHT	41.62	22.38	5.15	1.81	12.04	7.01
21	w/o CHT	41.54	13.5	2.58	1.71	26.67	-
	w/ CHT	24.79	12.88	3.64	1.28	35.25	4.47

Abbreviations: w/o = without; w/ = with.

Figure 4. Physicochemical characterization of the freeze-dried fibers. (A) SEM micrographs depict the alteration of fiber morphology in culture medium (DMEM) over time. The structure of the uncoated fibers is not stable and disintegrates after 21 days (A1-A3-A5), while the CHT coating exhibits a protective role that hinders the disintegration of the fibers and therefore enhances their stability (A2-A4-A6). (B) The presence of CHT was evaluated along time by tracing the amounts of nitrogen (from the $-\text{NH}_2$ groups of CHT) present on the surface of the fibers. The element was detected until day 21, suggesting that CHT was still present on the fibers. The defined squares in the SEM micrographs represent the area where the magnification and elemental analysis were performed.

400 The formation of the polyelectrolyte complexes relies on the
401 interaction between the available negatively charged sulfate
402 groups of κ -CA and the protonated amino groups of CHT
403 (Figure 2C). Thus, the immersion of κ -CA fibers in the
404 optimized CHT solution led to the formation of a localized CHT
405 nanosized layer, which was in agreement with previously
406 published data.^{29–33} The presence of the CHT coating was
407 confirmed by the intense pink coloration observed only on the
408 coated fibers after staining with Eosin Y, which specifically stains
409 CHT⁴⁸ (Figure 2C–C1 inset). From a practical standpoint,
410 CHT coated fibers, when compared with the uncoated fibers,
411 were easier to separate and handle despite their small diameters.

412 3.3. CHT Coated κ -CA Fibers Depict Improved Stability.

413 In the context of TE, it is important to evaluate the behavior of
414 these fibers in conditions similar to the in vivo microenviron-
415 ment. Swelling properties of hydrogels directly affect the overall
416 features of the polymeric network (shape, size, and mechanical

417 stability). Thus, we aimed to evaluate to which extent the
418 swelling ratio of the developed fibers affects their overall
419 diameter, and whether the presence of CHT provides stability
420 to the fibers. We performed swelling studies by maintaining the
421 coated and uncoated fibers in culture medium, which consists of
422 a collection of ions and proteins that could destabilize the κ -CA
423 hydrogel network.⁴¹

424 Our findings showed that the swelling ratios of ionically cross-
425 linked κ -CA fibers significantly increased with time ($p < 0.001$)
426 (Figure 3A). The destabilization of the ionically cross-linked
427 κ -CA network is likely to occur due to the continuous exchange of
428 K^+ ions entrapped within the network with other ions present in
429 the culture medium. In fact, similar results have been reported for
430 ionically cross-linked alginate²⁴ and gellan gum,⁴⁴ where
431 prolonged immersion in culture medium led to the weakening
432 of the network and, consequently, to its collapse.

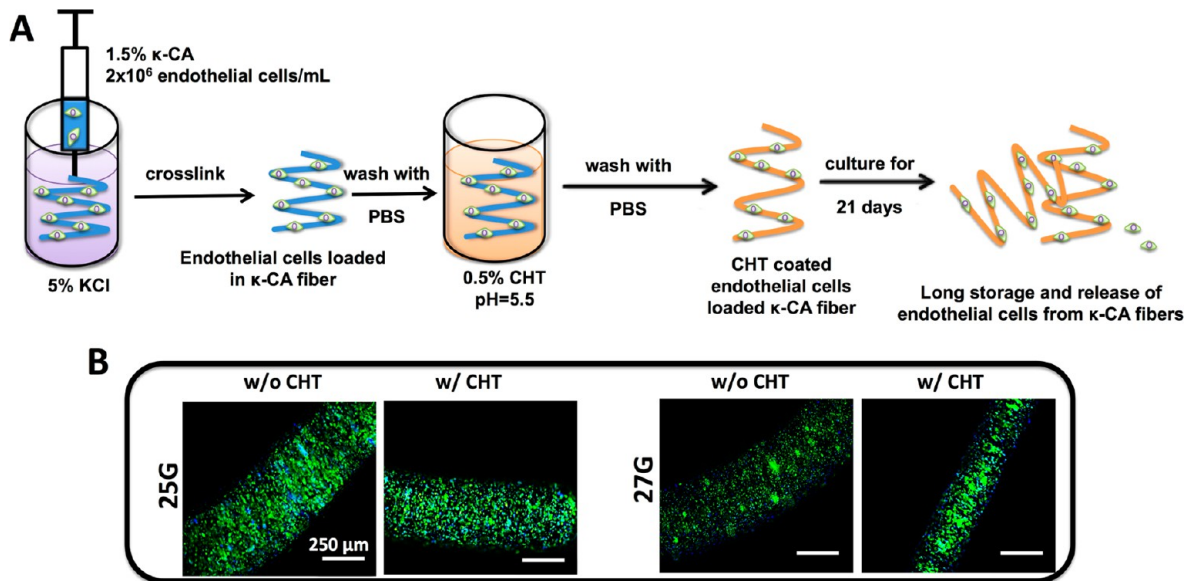


Figure 5. Encapsulation of SSEA-4⁺hASCs-derived ECs into κ -CA fibers. (A) Schematics of the encapsulation procedure and experimental setup. (B) The assessment of cell viability (live (green)/dead (red)) 24 h after encapsulation shows that the cells were not affected by the processing method, nor by the size and CHT coating of the fibers. Cell nuclei were counterstained with DAPI (blue).

433 However, CHT coated fibers exhibited a stable swelling ratio
434 over time, significantly inferior to that of the uncoated fibers ($p <$
435 0.001). This behavior can be due to the strong protective coating
436 layer, preventing the fiber swelling. A similar behavior was
437 described in other studies, concerning the use of CHT in
438 combination with alginate to form a layer of wrapping alginate
439 beads for the controlled release of drugs.²⁹

440 As a consequence, the diameter of CHT coated fibers was
441 maintained constant over time, while the uncoated fibers
442 exhibited an increase in their diameter (Figure 3B). Even
443 more, in terms of manipulation, the uncoated fibers became soft
444 and easy to break, while the coated ones were stable over the long
445 term and easy to handle, without obvious damage to their
446 integrity.

447 The closer evaluation of the fiber morphology by SEM
448 revealed that the uncoated fibers started to lose their integrity
449 around day 14 (Figure 4A1–3–5), as predicted by the swelling
450 ratio results (Figure 3A). By contrast, the CHT coated fibers
451 exhibited a rougher surface, displaying a mesh-like wrapping
452 around an open-pore κ -CA core. These fibers maintained their
453 shape, size, and integrity until day 21, when small cracks become
454 visible on their surface (Figure 4A2–4–6). The presence of the
455 CHT coating was further demonstrated by tracing the N content
456 on the surface of the CHT-coated κ -CA fibers by EDS/EDX
457 (Figures 4B) over time. Its detection up to 21 days suggested that
458 the CHT- κ -CA electrostatic interactions are strong enough to
459 withstand the presence of other ions and proteins present in the
460 culture medium, leading to a robust and stable structure.

461 **3.4. CHT Coating of the Cell-Loaded κ -CA Fibers Does**
462 **Not Affect the Endothelial Differentiated SSEA-4⁺hASCs**
463 **Phenotype in the Long Term.** Recently, several cell types,
464 such as human adipose derived stem cells, human nasal
465 chondrocytes, ATDC5 chondrocytic cell line, and L929
466 fibroblast cell line, were encapsulated within κ -CA hydrogels,
467 processed in different shapes (fibers, beads, discs), with positive
468 results in terms of viability, cellular metabolism, proliferation,
469 and differentiation.^{41,49,50} Herein, for the first time, we proposed
470 the encapsulation of microvascular-like ECs within κ -CA

471 hydrogel fibers as building blocks of vascularized bone tissue
472 engineered constructs. Pre-vascularization of engineered con-
473 structs by using progenitor/endothelial cells has been considered
474 the most promising strategy to address the vascularization of
475 large constructs.⁵¹ However, the use of human cells has been
476 hampered by the limited sources and reduced yields.⁵² Our
477 earlier work³⁴ has shown that the SSEA-4⁺ subpopulation,
478 selected among the heterogeneous stromal vascular fraction of
479 human adipose tissue, can be differentiated toward both
480 endothelial and osteogenic lineages. Moreover, SSEA-
481 4⁺hASCs-derived ECs hold features of microvascular ECs that
482 trigger the formation of 3D vascular networks within engineered
483 constructs. Therefore, we have taken advantage of our
484 knowledge to obtain ECs from SSEA-4⁺ subpopulation
485 (Supporting Information (SI) Figure 1) to develop cell-laden
486 κ -CA-based hydrogel fibers to be used as 3D cellular carriers.
487 SSEA-4⁺hASCs-derived ECs were encapsulated within the κ -CA
488 and CHT coated κ -CA fibers with 2 different diameters and
489 cultured for a period of 21 days (Figure 5A).

490 As the encapsulation procedure requires several steps that
491 could lead to reduced cellular performance,^{53,54} the direct effect
492 of the experimental conditions over the viability of the
493 encapsulated cells was assessed. Independently of the size and
494 whether fibers were coated or not, the majority of the
495 encapsulated cells were viable (average >65%, SI Figure 2) and
496 homogeneously distributed within the fibers (Figure 5B),
497 suggesting that the production of fibers (ionotropic gelation
498 and PEC) was mild enough.

499 Additionally, along the culture time and independently of the
500 conditions, viable cells were predominantly observed within the
501 fibers (Figure 6). A slight decrease in cellular viability was
502 noticed, so that by day 21, viable cells comprised 60% of the total
503 cells (SI Figure 2). However, this drop was not as significant as
504 reported in previous studies that associated this outcome with
505 the slow diffusion of nutrients caused by the sample thickness
506 and/or cross-linking mechanism.^{25,55,56} It might be due to the
507 fact that fiber-shaped hydrogels were characterized by a larger
508 surface area than disc shaped units, and as a consequence, a faster

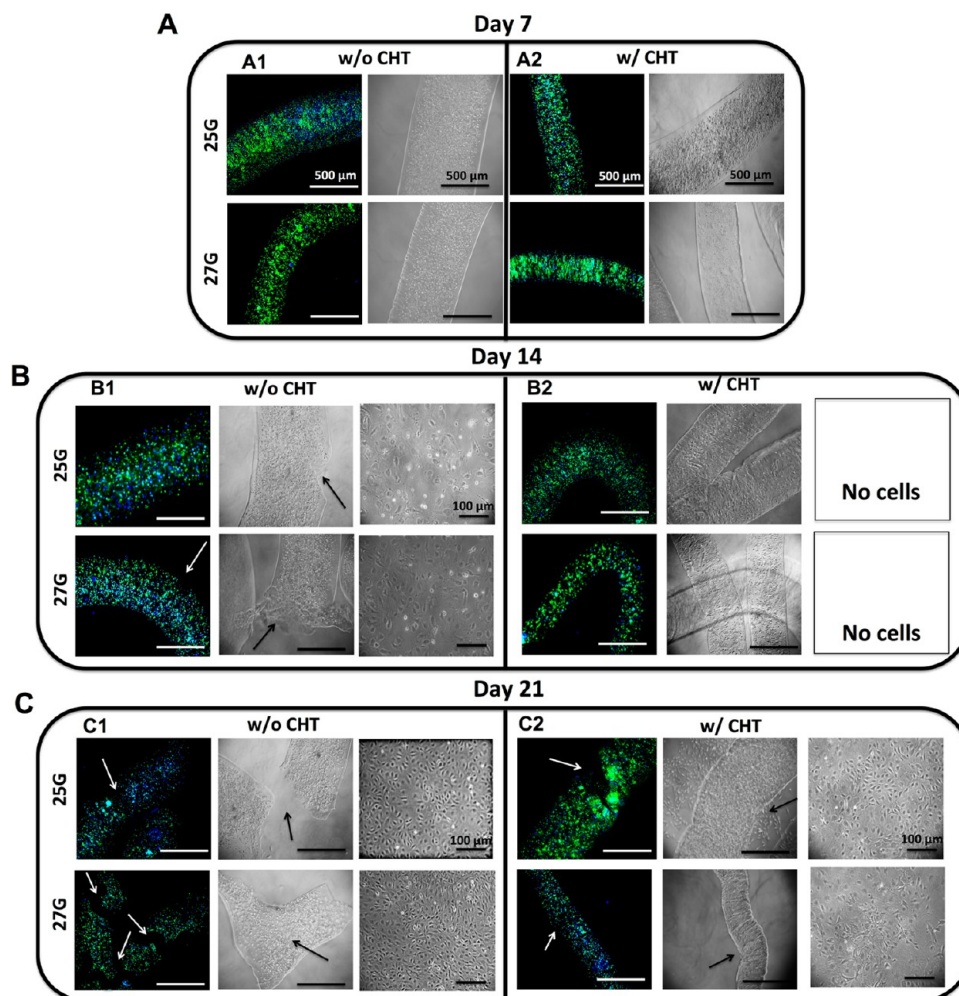


Figure 6. Evaluation of the cell-laden κ -CA hydrogel fibers along 21 days of culture. The viability of encapsulated cells (live/dead/) was maintained during the culture time in all conditions. (A) At day 7, independently of the conditions, fibers were intact, without displaying signs of disintegration. (B1) At day 14, cells escaped the uncoated fibers and small endothelial-like colonies could be observed at the bottom of the well. By day 21 (C1), fibers lose their integrity (arrows), allowing further release of cells into the well plate. Contrarily, the CHT coating (w/CHT) delayed the disintegration of the fibers, hence the release of the cells (B2 and C2). At day 21, the coated fiber diameter did not alter; however, small cracks (arrows) were observed along the fibers allowing the release of the cells. Cell nuclei were counterstained with DAPI (blue).

diffusion of nutrients and oxygen to the core of the sample could occur.⁵⁷ It has also been suggested that the ions needed for hydrogel formation may influence the cellular viability; nevertheless, the fibers reported herein are formed via a partial cross-linking process, by shorter exposure times to K^+ treatment than previously reported.^{18,58} The additional CHT coating, that was shown to prevent the fibers from swelling, did not affect viability. From the SEM images it was possible to notice that the CHT layer was displayed as a mesh wrapped around the fiber, preventing it from swelling, but enabling diffusion.

The optical analysis of the cell-loaded uncoated fibers showed an increase in their diameter and subsequent disintegration, in agreement with the results obtained with acellular fibers. Concisely, at day 7, uncoated fibers were intact, with a well-delimited smooth surface; however, at day 14, they presented signs of disintegration. This allowed the entrapped cells to “escape” from the fibers and adhere to the well and proliferate (Figure 6A1–C1). In opposition, the CHT coating delayed the disintegration of fibers, as they maintained their initial diameter and their surface did not present disruptions up to day 14 (Figure 6A2–B2). Only at the last time point did small cracks become

visible, and consequently, EC colonies were observed on the bottom of the well (Figure 6C2).

Beside potentially affecting cellular viability,⁷ cell encapsulation procedures and prolonged culture might also compromise the cell’s phenotype or functionality.^{20,59} In order to address this question, cells were retrieved from the fibers, after culturing for 21 days, and seeded into tissue culture flasks. Cells were able to adhere and organize into small endothelial-like colonies (Figure 7A) and, more importantly, maintained their ability to form tubular-like structures when seeded on Matrigel (Figure 7B). Moreover, when screened for the initial cell surface marker panel, cells were shown to be $CD105^+/CD73^+/CD31^+/CD34^+/CD45^-/CD90^-$, a signature that matches the one registered prior encapsulation. Taken together, these results show that the encapsulation process, followed by the prolonged culture of cells within fibers and their retrieval, did not affect the phenotype and in vitro functionality of ECs.

Overall, as κ -CA fibers exhibit high swelling ratios, the weakening of the network occurred and, consequently, cells were able to escape from the fibers. CHT coating maintained the fiber diameter, delaying their disintegration and the consequent release of cells into the surrounding environment. Nonetheless,

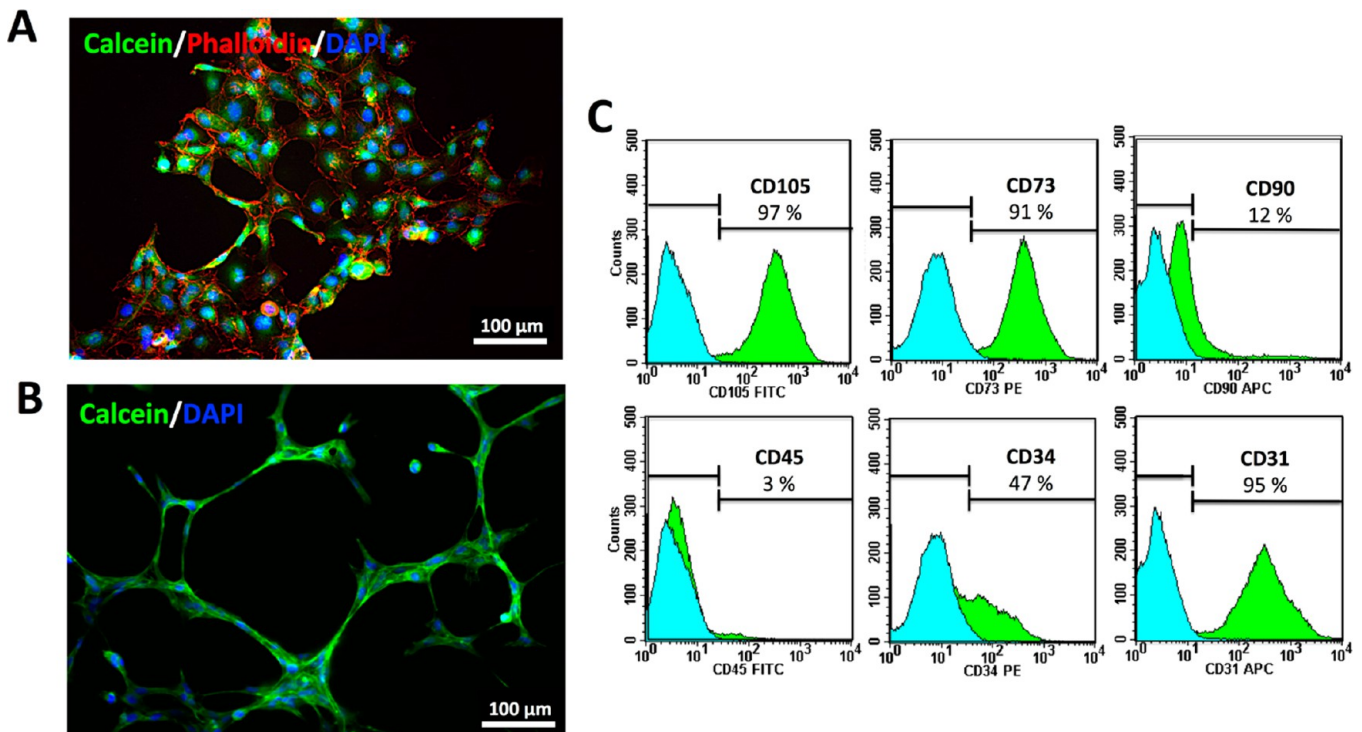


Figure 7. Endothelial phenotype of cells retrieved from the fibers after 21 days of culture. (A) After retrieval of cells through degradation of the fibers, and their subsequent seeding, cells adhered and formed small colonies, characterized by close contact between cells (live/cytoskeleton/nuclei). (B) Cells maintained their ability to form tubular-like structures (green) when seeded on Matrigel (C) and the expression of the initial endothelial phenotype-associated panel of markers. Cell nuclei were counterstained with DAPI (blue).

552 the CHT layer was still permissive to culture medium, allowing
 553 the diffusion of its components, as suggested by the subsequent
 554 remarkable results concerning the viability and phenotype of
 555 encapsulated cells. These findings further demonstrate the
 556 potential of using these fibers as EC carriers within different TE
 557 constructs.

558 **3.5. Cell-Loaded κ -CA-Based Fibers Can Act as Building**
 559 **Blocks within 3D Hydrogel Constructs.** In vitro 3D co-
 560 culture platforms provide a scalable and flexible approach to
 561 evaluate the indirect interactions between multiple-cell types in
 562 highly defined settings, enabling us to address specific cellular
 563 behaviors and the formation of complex tissue-like analogs.
 564 Simultaneously, the integration and assembly of cell-laden
 565 templates as building blocks within 3D constructs may allow
 566 the development of replicas of complex tissues to use in TE
 567 approaches. Herein, we proposed a straightforward method to
 568 integrate the κ -CA-based fibers enclosing SSEA-4⁺hASCs-
 569 derived ECs, within a 3D hydrogel containing SSEA-4⁺hASCs-
 570 derived osteoblast-like cells, as heterotypic constructs of interest
 571 for bone TE. Within this approach, we took advantage of the ease
 572 preparation of cell-loaded κ -CA fibers, to recreate the cellular
 573 indirect communication at a 3D scale. Thus, cell-loaded fibers,
 574 used as building blocks, were assembled within a κ -CA hydrogel
 575 disc by simply laying them in a Petri dish (Figure 8). They could
 576 be singled out or stacked in a random or organized manner
 577 (Figure 8A). When the cell-laden fibers were integrated in a
 578 hydrogel disc containing osteoblast-like cells, the formation of a
 579 3D heterotypic platform with a defined cellular topography
 580 suitable for bone-vascularization strategies was achieved (Figure
 581 8B–C). The proposed system was entirely formed by a κ -CA
 582 matrix, with a controlled spatial distribution of the two cell types
 583 derived from the same source. The fibers are expected to act both

584 as cell-reservoirs and structurally defined arrangements in the
 585 surrounding matrix. Within the context of bone TE, further
 586 experiments aim to deeply assess the cellular interactions within
 587 the 3D structure and the consequent outcome, both concerning
 588 the outflow of ECs triggered by the crosstalk with osteoblast
 589 cells,^{60,61} as well as osteogenic matrix deposition and
 590 mineralization. This could eventually lead to the organization
 591 of ECs within the osteoblast-containing matrix, leading to an
 592 improvement in the osteogenic outcome.

593 For the immediate integration of fibers in a secondary matrix,
 594 the CHT coating does not provide additional advantages, as the
 595 secondary matrix will already hold the fibers in place.
 596 Nevertheless, the CHT coating proves to be crucial for
 597 prolonged culture periods that are relevant when using this
 598 approach to study indirect interactions between multiple-cell
 599 types and/or for those 3D systems in which the accuracy of the
 600 initial parameters (shape, size, swelling, mechanical stability) is
 601 crucial. For instance, it could be possible to create a 3D co-culture
 602 system based on coated cell-laden κ -CA fibers by seeding another
 603 cell type on top of the fibers.

4. CONCLUSIONS

604 The development of chitosan (CHT) reinforced κ -carrageenan
 605 (κ -CA) fibers using a two-step procedure, under cell-friendly
 606 experimental settings, aiming at homogeneous immobilization of
 607 cells is herein reported. The diameter of the fibers could be easily
 608 tuned by selecting the appropriate needle gauge during
 609 processing. The presence of the CHT coating enhanced the
 610 stability of the fibers and their diameter was maintained constant
 611 by restraining their swelling. Moreover, these fibers support the
 612 viability and functionality of encapsulated cells during long-term
 613 culturing, enabling their use as cell-delivery systems within 3D

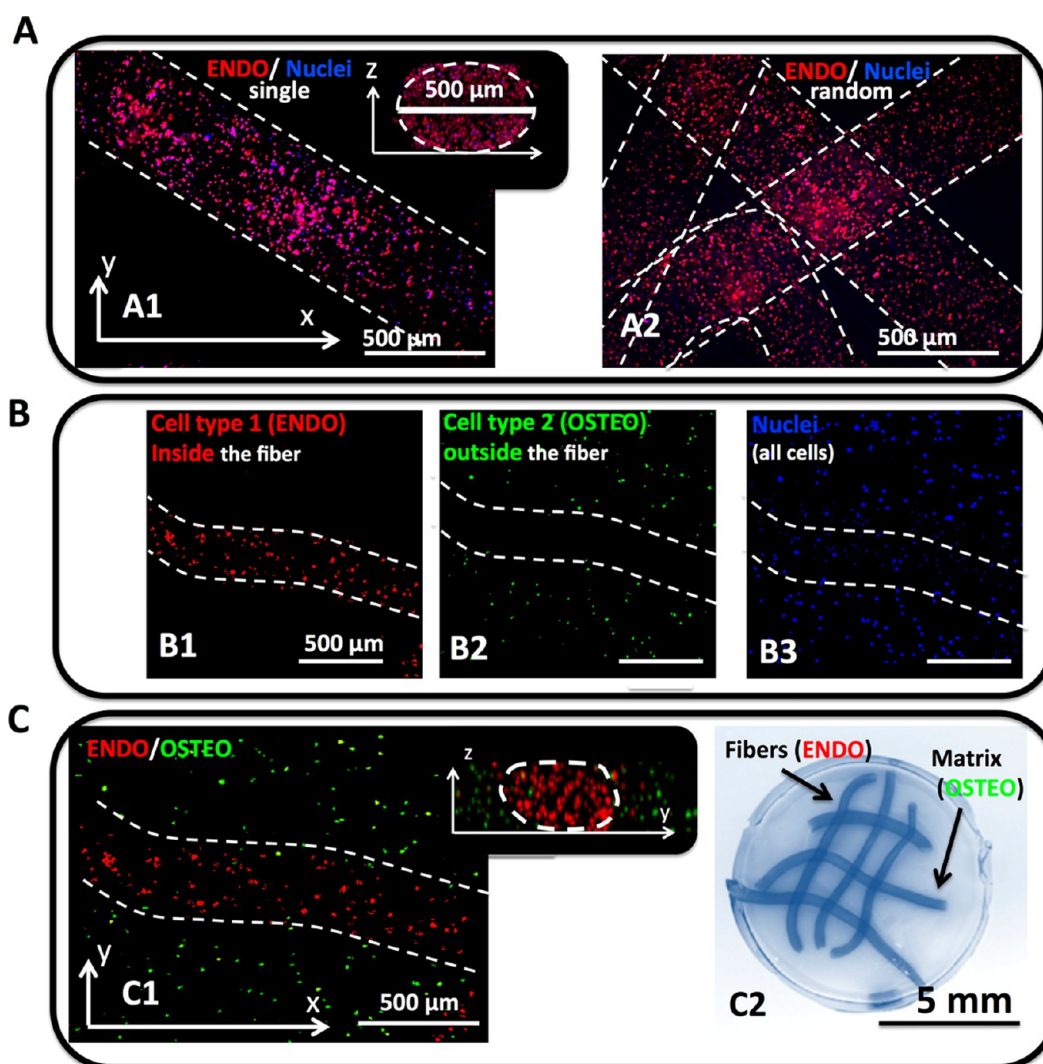


Figure 8. Proof of concept of the use of cell-loaded κ -CA-based fibers as building blocks within 3D κ -CA hydrogel constructs. Confocal laser scanning micrographs of (A1–2) encapsulated SSEA-4+hASCs-derived ECs (ENDO) within κ -CA fibers. Fibers can be (A1) singled out or (A2) randomly stacked. (B,C) Heterotypic 3D κ -CA-based structure. (B) The encapsulation of ENDO preceded the integration of the fibers within the κ -CA containing the SSEA-4+hASCs-derived osteoblast-like cells (OSTEO) localized outside the fiber. All cells (blue) were evenly distributed within the hydrogel. (C) 3D buildup of fiber stacks within a hydrogel disc consisted of a controlled spatial localization of two cell types within a single hydrogel matrix. (C1) The colocalization of ENDO and OSTEO is relevant for developing spatial controlled heterotypic systems aimed at vascularized bone TE approaches. Cell nuclei were counterstained with DAPI (blue). (C2) A macroscopic view of the 3D construct. The discs were manually manipulated without damage. κ -CA fibers were stained with methylene blue (dark blue).

614 tissue engineering constructs. Furthermore, using a bottom-up
615 approach, these fibers can be used as building blocks for the
616 development of suitable 3D platforms of independently
617 organized heterotypic cell-containing hydrogels, relevant for
618 TE approaches.

619 ■ ASSOCIATED CONTENT

620 ● Supporting Information

621 Supplemental figures depict the endothelial phenotype of SSEA-
622 4+hASCs-derived ECs phenotype prior encapsulation and the
623 propidium iodide/DAPI images of fibers along time as well as the
624 quantification of the cellular viability. This material is available
625 free of charge via the Internet at <http://pubs.acs.org>.

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Notes

The authors declare no competing financial interest.

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