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Exploiting the sequence of naturally occurring elastin: Construction, production and characterization of a recombinant thermoplastic proteinbased polymer

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Abstract. Genetic engineering was used to produce an elastin-like polymer (ELP) with precise amino acid composition, sequence and length, resulting in the absolute control of MW and stereochemistry. A synthetic monomer DNA sequence encoding for (VPAVG)₂₀, was used to build a library of concatemer genes with precise control on sequence and size. The higher molecular weight polymer with 220 repeats of VPAVG was biologically produced in *Escherichia coli* and purified by hot and cold centrifugation cycles, based on the reversible inverse temperature transition property of ELPs. The use of low cost carbon sources like lactose and glycerol for bacteria cells culture media was explored using Central Composite Design approach allowing optimization of fermentation conditions. Due to its self-assembling behaviour near 33 °C stable spherical microparticles with a size ~ 1 μ m were obtained, redissolving when a strong undercooling is achieved. The polymer produced showed hysteresis behaviour with thermal absorbing/releasing components depending on the salt concentration of the polymer solution.

Introduction

Elastin-like polymers (ELPs) are repetitive polypeptides inspired on the mammalian elastin structure that consists of a pentapeptide repeat, VPGXG, where X, termed the guest residue, is any natural amino acid except proline [1]. The most striking feature of the ELPs is their Inverse Temperature Transition (ITT) behavior. Below a certain critical temperature, the transition temperature (T_t), and in the presence of water they are soluble, with the polymer chains relatively extended in a disordered state and fully hydrated mainly by hydrophobic hydration [2]. Above T_t , the polymer chains hydrophobically fold and adopt a dynamic structure, called β -spiral, involving one type II β -turn per pentamer, stabilized by intraspiral inter-turn and inter-spiral hydrophobic contacts [2,3].

The polymer poly(VPAVG), an ELP where the central glycine (G) is substituted by a L-alanine (A), was described as exhibiting similar properties to synthetic thermoplastic elastomers [4,5] and was previously chemically synthesized by Rodríguez-Cabello and co-workers [3,6]. The same group reported its characterization, demonstrating its extreme biocompatibility both in vitro and in



vivo, as well as its ability to self-assemble, forming microparticles that can entrap active substances during the self-assembling process [3,6].

Advances in recombinant DNA technology has allowed the biological synthesis of large Molecular Weight (MW) engineered protein polymers containing repeating blocks of amino acids with precise composition, sequence and length, resulting on an absolute control of MW and stereochemistry. The present work describes the biological production of the poly(VPAVG) using standard molecular genetic tools. A higher molecular weight polymer containing 220 repeats of VPAVG sequence, with an estimated MW of 96kDa was expressed in *Escherichia coli* pET25 vector, followed by purification and characterization. The fermentation conditions were optimized using a Central Composite Design (CCD) approach to design the culture media.

Materials and Methods

Gene construction. The DNA coding for a peptide monomer containing 20 repetitions of VPAVG and flanked by Eam1104I recognition sites was subjected to concatenation in order to obtain a multimeric unit. The multimeric block genes were obtained by recursive directional ligation in the cloning vector and the cloning strategy used was previously described and reported as the "Gutenberg Method" [7,8]. The design of the monomeric DNA sequence took into account Escherichia coli codon usage and DNA-repetitive sequences were avoided. All the constructions/concatenation were performed in a modified cloning vector, pDrive (Qiagen) and transformed into E. coli strain XL1 Blue (Novagen). Constructions were confirmed with the restriction enzymes Eam1104I and EcoRI (Fermentas). Taq DNA polymerase, LguI restriction enzyme and GENERULERTM 1kb DNA ladder were obtained from Fermentas. 100bp DNA ladder was purchased from Invitrogen and high marker from Bioventures. Vector dephosphorylation by anthartic phosphatase (New England Biolab) was performed prior to ligation with T4 DNA ligase (Roche). The plasmid extraction was performed with GENELUTETM Plasmid Miniprep Kit (Sigma-Aldrich) and the DNA purification from agarose gel by NUCLEOSPINTM Extract (Macherey-Nagel). Plasmids were selected based on their insert length and subcloned into a modified pET25b (+) expression vector (Novagen) without any tags. The construction was confirmed by sequencing and restriction with endonucleases NdeI and XhoI (Fermentas). Agarose gel images were acquired with Eagle Eye II system and Eagle Sight software (Stratagene). The plasmid construct was verified by DNA sequencing. The sequencing was performed following the method of Sanger et al. [9], using an ABI PRISM 310 Genetic Analyzer. The vector carrying (VPAVG)₂₂₀ gene was used to transform E. coli strain XL1 Blue, according to the SEM method [10]. DNA cloning and manipulation were performed according to standard protocols [11]. The recombinant plasmids were then transformed into the expression strain E. coli BL21(DE3) (Novagen), followed by protein production with optimized medium.

Central composite design (CCD) and data analysis. The culture medium was formulated based on the recipes of several media, namely: Luria Bertani (LB), Terrific Broth with Phosphate Buffer and Glycerol (TB+PBG) [11] and salt medium from Teich and co-workers[12].

To analyze the production of $(VPAVG)_{220}$, a central composite design with five coded levels was used for exploring the sub-region of the response surface in the neighborhood of the optimum. The three tested factors, glycerol, lactose and yeast extract concentrations (Table 1) were submitted to a 2^3 factorial design, with values fluctuating from a central point (0) to a lower level (-1) and a higher level (+1). The three level combinations of the three factors define our experiment "nucleus" with three additional central points (where all the variables adopt the central value zero) in order to minimize the risk of losing the non-linear relations in the intervals and to allow the calculation of confidence intervals due to repetition [13,14]. The factors with negative values were considered as zero. The two responses considered were the concentrations of biomass and polymer obtained. The analysis of the obtained data, were performed using the software Design-Expert 6.0.6 (Stat-Ease, USA). The model adequacy was analyzed by the F value and the goodness of fit was expressed by



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the coefficient of determination R^2 . The significance levels were determined by values of "Prob > F" less than 0.05.

Coded factors used in the CCD

Run	Glycerol [g L ⁻¹]	Lactose [g L ⁻¹]	Yeast Extract [g L ⁻¹]
1	-1	1	1
2	0	0	-1,68
3	1	-1	-1
4	-1	-1	-1
5	0	0	0
6	0	0	0
7	0	0	1,68
8	0	0	0
9	1	-1	1
10	1	1	-1
11	0	-1,68	0
12	1	1	1
13	0	1,68	0
14	-1,68	0	0
15	-1	-1	1
16	-1	1	-1
17	1,68	0	0

Table 1. Experimental range in coded factors of the three independent variables used in the CCD.

Polymer production and purification. Bacterial cultures were grown at 37 °C in optimized medium, containing 100mg/ml ampicilin. After fermentation time of 16 hours, cells were harvested by centrifugation, washed with TBS (20mM Tris, 140mM NaCl at pH 8.0), ressuspended in TE buffer (50mM Tris, 1mM EDTA at pH 8.0) and lysed by ultrasonic disruption. The insoluble cell debris was removed by centrifugation at 4 °C and 10000xg for 20 minutes, preceded by a cooling step at 4 °C for 2-3 hours. As this polymer does not respond to pH, the solution was adjusted to 3.5 with HCl in order to precipitate most of *E. coli* endogenous proteins, followed by a clearing centrifugation of 20 minutes at 10000xg and 4 °C. The cleared supernatant was submitted to a heating stage (HS) of 30 minutes at 60 °C, followed by a centrifugation at 40 °C to collect the aggregated polymer. The pellet was then ressuspended in 10-15mL of cold deionised water and left at 4 °C for 2 hours, in a cooling stage (CS), in order to solubilize the aggregates. The HS and CS were repeated two more times followed by freeze-drying. Polymer fractions were analysed by SDS-PAGE and stained with 0.3M Copper Chloride. SDS-PAGE gel images were acquired with Molecular Imager ChemiDoc XRS system and Quantity One software from Bio-Rad.

Maldi MS and MS/MS analysis of (VPAVG)₂₂₀. The purity and amino acid sequence of (VPAVG)₂₂₀ was confirmed by mass spectrometry with previously digested polymer solutions. Possible cleavable enzymes were found by using the "PeptideCutter" application from ExPASy



website (www.expasy.ch/tools/peptidecutter/). The digestion was carried out using proteinase K and standard protocols.

Turbidity and Differential Scanning Calorimeter (DSC). The turbidity experiments were conducted in a UV-Visible Thermo Spectronic Helios gamma spectrophotometer with a thermostatised sample chamber. The turbidity was assessed by the change in absorbance at 300nm for a 3mg/mL polymer in water solution. The T_t was identified as the temperature at which 50% of turbidity had occurred. DSC experiments were performed on a Mettler Toledo DSC822e. Liquid nitrogen was used as cooler and the calibration of enthalpy and temperature was performed with Indium and Zinc. The analyses were carried in a solution of 25mg/mL of polymer, either on deionised water, NaCl (0.2M, 0.4M, 0.6M and 0.8M) and urea (1M, 3M and 6M). In a typical DSC run, 25µl of polymer solution were placed in a hermetically sealed aluminium pan and studied under a four-stage thermal program. First, the sample is treated 15 minutes at 4 °C, followed by a heating stage (HS) from 4 to 60 °C with a constant heating rate of +5 °C/min. The sample is maintained at 60 °C for 3 minutes and cooled down in a cooling stage (CS) to -10 °C with a constant cooling rate of -5 °C/min. The onset and enthalpy (Δ H) values were calculated with the provided STARe software.

Preparation and morphological characterization of (VPAVG)₂₂₀ **particles.** A solution of 6mg/ml of (VPAVG)₂₂₀ in water was treated at 40 °C over 40 minutes. Particles were formed by polymer self-aggregation from aqueous solution. The solution was then centrifuged at 11000xg and the supernatant discarded. The pellet was finally dried at 37 °C overnight. The mean size and distribution of (VPAVG)₂₂₀ was measured by Dynamic Light Scattering (DLS) with a Zetasizer Nano equipped with Dispersion Technology Software v5.02 from Malvern Industries. The morphological examination was performed on dried particles by Scanning Electron Microscopy (SEM).

Results and Discussion

Construction of poly VPAVG. The DNA monomer coding for 20 repetitions of the sequence VPAVG, including the Eam1104I recognition sites, with 300 nucleotides long was subjected to a concatenation reaction, creating a library of multimeric DNA units. The products of this reaction were cloned directly into pDrive vector, linearized with LguI. Fig. 1 represents the n-mer constructions obtained, n = 1, n = 2, n = 4, n = 7 and n = 11, where n represents the number of (VPAVG)₂₀ repetitions. The correct constructions were confirmed by restriction analysis with EcoRI and Eam1104I. Restriction with Eam1104I resulted on fragments with 300, 600, 1200, 2100 and 3300 bp, corresponding to the different multimeric blocks plus additional bands generated by Eam1104I vector digestion. EcoRI cuts pDrive in the flanking regions near the gene originating two fragments, one corresponding to the cloning vector (approximately 3900bp) and another one that varies depending on the size of the concatemer. A 220 repeats construct, coding for a polymer with an estimated molecular mass of 93kDa, was obtained and subcloned into a modified pET25b (+) expression vector. The correct construct (pET-220) was confirmed by DNA sequencing and restriction with NdeI and XhoI endonucleases (data not shown), and used to transform *E. coli* strain BL21(DE3).



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Fig. 1. A – Schematic representation of pDrive vector originating a library of n-mer VPAVG coding sequence. ••• corresponds to the restriction site for Eam1104I in the multiple cloning site. ••• corresponds to the restriction site for EcoRI. B – Agarose gel electrophoresis of DNA constructions containing n (lanes 1, 2), 2n (3, 4), 4n (5, 6), 7n (9, 10) and 11n (11, 12), being n a 300 nucleotides long molecule coding for 20 repetitions of the sequence VPAVG, including the Eam1104I recognition sites. Lanes 1, 3, 5, 9, 11 – DNA restriction with EcoRI; lanes 2, 4, 6, 10, 12 – DNA restriction with Eam1104I; lane 7, 8 and 13 – molecular weight marker. The arrow indicates the respective cloned fragments.

CCD optimization. Preliminary fermentations with *E. coli* transformed with pET-220 in LB medium (induction at $DO_{600}=0.6$ with 1mM IPTG and fermentation times of 5 and 24 hours) resulted in a production of ~2mg/L and showed that the biomass and polymer productions were closely related (data not shown), suggesting that an optimization of biomass production would be an approach to enhance the polymer production. In order to increase the production of (VPAVG)₂₂₀ and biomass formation, the levels of the variables glycerol (A), lactose (B) and yeast extract (C) were selected according to the results of previous experiments. Fig. 2 shows the response surface plots for the biomass production indicating that the higher of this parameter were achieved for glycerol and yeast extract concentrations at the middle and maximum of the experimental range, respectively and in the absence of lactose. The model for the polymer response was not significant statistically and therefore could not undergo validation. A first order coded model was established describing the biomass concentration as a function of glycerol, lactose and yeast extract concentrations:

Biomass = 3.38 + 0.056A - 0.17B + 0.63C. (1)

The goodness of fit was expressed by the coefficient of determination R^2 calculated to be 0.55, indicating that 55 % of the variability of the response biomass could be explained by this model. The tests of lack of fit were not significant indicating that the model fits to the experimental results. The statistical significance of the second order model function was evaluated by the F-test analysis of variance which revealed that this regression is statistically significant. Based on the CCD results, the optimum medium composition for growing *E. coli* consisted of: 7.39g.L⁻¹ of glycerol, 23.25g.L⁻¹ of yeast extract, 0.5g.L⁻¹ of glucose, 12g.L⁻¹ of tryptone, 2.31g.L⁻¹ of KH₂PO₄, 12.54g.L⁻¹ of K₂HPO₄.3H₂O, 0.5g.L⁻¹ of MgSO₄.7H₂O, 1.5g.L⁻¹ of NH₄Cl, 3.85g.L⁻¹ of(NH₄)₂SO₄, 0.5g.L⁻¹ NaCl







and $8g.L^{-1}$ of $NaH_2PO_4.H_2O$. This new medium recipe allowed the achievement of ~6mg.L⁻¹ of polymer.

Fig. 2 – Response surface plots showing the production of biomass from *E. coli* and the interaction between glycerol *vs* lactose (A), glycerol *vs* yeast extract (B) and yeast extract *vs* lactose (C).

Production and Purification of (VPAVG)_{220.} Several *E. coli* transformants with pET-220 were selected and screened for production (data not shown) prior to fermentation conditions. The cells were harvested, lysed and the polymer purified as described in Material and Methods, based in simple hot and cold centrifugation steps (Fig. 3A). The polymer self-aggregates when the temperature rises above its T_t , allowing its collection from solution by hot centrifugation. As the self-aggregation of the polymer is a reversible reaction, the polymer could be resolubilized in cold deionized water, while the denatured and aggregating contaminating biomolecules remain in solution, being further separated by cold centrifugation. Preparations of fully purified polymer (Fig. 3B) were lyophilized and used for characterization experiments.





Fig. 3 – Purification of $(VPAVG)_{220}$ polymer. A – Behaviour of $(VPAVG)_{220}$ polymer solution submitted to hot and cold stages. After 30 minutes of incubation at 60 °C the polymer selfassembles, forming particles that come out of solution and solubilized again when submitted at 4 °C. B – 10% SDS-PAGE analysis of purified $(VPAVG)_{220}$ polymer stained with copper chloride (lane 1) or with Coomassie staining (lane 2). As ELPs stain very poorly with Coomassie [24, 25], copper staining was used as routine protocol and Coomassie staining as negative control. The purified band appears with higher molecular weight than the expected, a phenomenon observed previously by McPherson *et al.* [24]. The molecular weight marker is the same for both gels.

Maldi MS and MS/MS analysis of (VPAVG)₂₂₀. The correct molecular weight, composition and sequence of the pure bioproduced polymer were assessed by MALDI-TOF/MS (data not shown) and MALDI-TOF MS/MS analysis. The polymer was digested with proteinase K in order to obtain small fragments of the protein followed by MALDI-TOF analysis. The spectra of the digested product revealed several peaks corresponding to the different fragments obtained (Fig. 4A). A fragment of 442.2Da corresponding to the theoretical molecular weight of a single VGVPA (proteinase K cuts between the A and V) unit was selected and submitted to tandem MS to inspect the constitution and sequence (Fig. 4B) that was determined as Val-Gly-Val-Pro-Ala. The obtained sequence correlates with the prediction using the peptide cutter website when digesting (VPAVG)₂₂₀ with proteinase K.

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Fig. 4 – Mass spectra (MALDI-TOF/MS and MS/MS) of $(VPAVG)_{220}$ digests. A – Spectrum obtained from MS analysis of products digestion. The different signals correspond to the different length products, e.g. the signal at 187.1 Da corresponds to the PA product, 442.2 Da, corresponds to the VPAVG block and the signal at 883.5 Da to a double VGVPA block. B – Spectrum obtained from tandem MS analysis of the 442.2 product. The spectrum analysis revealed the VGVPA sequence.

Turbidity and DSC analysis. Fig. 5 shows the results obtained in the DSC experiments, carried out with the cyclic program described in Materials and Methods, for a 25mg/mL polymer solution in deionised water and NaCl at different concentrations.



Fig. 5 – DSC run with 25 mg/mL polymer solution in different salt concentration solutions. A – DSC heating stage with +5 °C/min rate. B – DSC cooling stage with a rate of -5 °C/min.

During the heating stage (HS) (Fig. 5A), the DSC pattern clearly exhibits an endothermic peak that is associated with the characteristic endothermic process of chain folding, accompanied by the destruction of the ordered water structures of hydrophobic hydration [3,8,15-17]. Accordingly to



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Rodríguez-Cabello and co-workers [8], the chain folding consists in interchain hydrophobic contacts that cause the formation of aggregates that separate from solution. The temperature of self-assembling (T_{tS}) was calculated based on the onset and is identified with the T_t (table II). On the contrary, an exothermic depression (Figure 5B) must be attributed to the reverse process, the unfolding of the polymer chain and rehydration of the ordered structures, leading to its consequent dissolution (T_{tD}) [3,17]. However, there is a clear difference between the T_{tS} and T_{tD} of each condition, revealing hysteresis behaviour. As an example, the polymer self-assembles (folds) near 33 °C in deionized water but only redissolves when a strong undercooling is achieved.

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Conditions	Onset of T _{ts} [°C]	Enthalpy ∆H [J/g]
Deionised water	32.80	38.35
NaCl 0.2M	29.18	41.30
NaCl 0.4M	25.91	45.88
NaCl 0.6M	22.50	46.53
NaCl 0.8M	19.18	54.85
Urea 1M	37.55	31.14
Urea 3M	47.16	25.56
Urea 6M		

Values of T_{tS} and ΔH calculated from the DSC experiment.

Table 2 – Values of T_{tS} and ΔH calculated from the onsets of the DSC run during the heating stage. The enthalpy (ΔH) of the endothermic process was calculated from the area of the endothermic peak.

In order to support the results obtained by the dynamic calorimeter study, a statical determination of T_{tS} was carried out by turbidity assays. The plot in Fig. 6 represents the absorbance values obtained for a water solution of (VPAVG)₂₂₀. The T_{tS} or T_t was determined as the temperature at which the turbidity reaches a value of 50%, which was found to have a value of approximately 33 °C. The rapid increase in turbidity upon reaching T_{tS} shows that the transition is very sharp, occurring in a range less than 5 °C. The small variation between the statical determination of T_{tS} and the DSC results can be attributed to the associated thermal lags of the DSC experiment.





Fig. 6 – Temperature profile of aggregation during the heating process. The T_t was determined as the temperature at which the turbidty reaches a value of 50%. The images shows the polymer solution below (in the left) and above (in the right) T_t and the formation of particles (in the right).

The hysteresis behaviour has been described in a previous work [18] with poly(VPGVG) and poly(VPAVG), as deduced from FTIR and Raman spectroscopies. It was suggested that this behaviour is a combination of two factors, the presence of a methyl group in the Ala residue in poly(VPAVG) and a more perfectly stable folding state when compared to the conventional poly(VPGVG). The methyl group in the Ala3 residue between the Pro2 and Val4, strongly influences the physical structure of the folded and unfolded states. In aqueous solution and below the T_{tS}, the water shell around poly(VPAVG) is more closely structured and thus it is more dissolved than poly(VPGVG). Above the T_{tS}, poly(VPAVG) contains less water bound to the carbonyl of amide groups than poly(VPGVG) [3,18], leading to a more compact and rigid structure due to the decrease in the plasticizing effect of water and its role in disrupting the intramolecular hydrogen bonds [8]. This impedes coacervation and causes the formation of stable particle suspension³. The presence of the methyl group is also fundamental for the reversibility of temperature transition. During the cooling, the unfolding process of poly(VPAVG) is hampered by the strongly bound amide groups that stabilize the folded structure and only takes place after strong undercoolings [15,18].

It is well known that salts cause a significant, concentration-dependent, decrease in T_t . Previous studies clearly shown that the T_t of ELPs is depressed by the addition of salts [19,20]. This fact is well represented in Fig. 5. When comparing the thermograms of the different conditions, there is a shift of the endothermic and exothermic components into lower temperatures as the salt concentration rises (for the endothermic peak: 33 °C - H₂O, 29 °C - 0.2M NaCl, 26 °C - 0.4M NaCl, 22.5 °C - 0.6M and 19 °C - 0.8M). The difference between the onset of the endothermic peaks can be explained by the ionic contribution of salt ions to the entropy of the system. Reguera et al. [20] demonstrated that there is a linear decrease in T_t versus the salt concentration (NaCl), which is in agreement with our data. Accordingly, this behaviour can be explained by a complex process that involves not only the disruption of water structures and hydrophobic association, but also an increment in order in the main chain with a subsequent stabilization. The effect of NaCl concentration in the thermal parameter is equivalent to an increase in the hydrophobicity of the polymer and therefore, an increase in salt would cause an increase in the polarity of the solvent, causing more and more ordered structures surrounding the polymer chains. This fact provides an answer to the increasing values of enthalpy (Table 2) as the NaCl concentration raises and was previously observed by Reguera *et al.* [20]. The estimated enthalpy values (ΔH) for the ELP



transition were endothermic and strongly influenced by the NaCl concentration, explainable by the increase in the number of more ordered structures of hydrophobic hydration surrounding the polymer chains.

In parallel to the calorimeter studies of NaCl and H₂O polymer solutions, a DSC run was performed using polymer solutions with urea at different concentrations (1M, 3M and 6M). As expected, both endothermic and exothermic components shift to higher temperatures (33 $^{\circ}C - H_2O$, 37.5 $^{\circ}C - Urea$ 1M and 47 °C – Urea 3M), accompanied by a broadening of the endothermic peak (well represented between 1M and 3M urea polymer solutions). In the presence of urea 6M (data not shown) a straight line without any endothermic or exothermic components was observed, due to a flatted peak (broadening of the peak) or suggesting that a probable transition phase would occur outside the range of temperatures used (4 - 60 °C). Urea at high concentrations is commonly used as a denaturing agent. This denaturation occurs because it diminishes the hydrophobic effect by displacing water in the solvation shell and specifically binds to amide groups through hydrogen bonds [21]. According to Urry [22] the addition of salts like NaCl, lowers the T_t , while the addition of urea or guanidinium chloride increases this value. Another work by Luan and Urry [23], carried with a synthetic poly(VPAVG), shows a raise in the T_t as the concentration of urea increases (34.2 °C, 38.5 °C, 43 °C and 48.5 °C for H₂O, and urea 1M, 2M and 3M, respectively) as well as a decrease in the sharpness of the endothermic peak. These results are in agreement with our data and suggest that the increasing urea concentration follows an almost linear increase in the $T_{\rm t}$. The enthalpy values (Table 2) calculated for the endothermic peak during the HS support the conclusions previously stated for the NaCl solutions. In opposition, while the addition of NaCl in the system leads to an increase in number of ordered structures, the addition of Urea disrupts the water of hydrophobic hydration, leading to a decrease in the hydrophobic effect.

Morphological characterization and size distribution by SEM and DLS analysis. The selfassembled (VPAVG)₂₂₀ particles resulted in sizes lower than 2μ m as demonstrated by DLS analysis. Fig. 7 shows the volume diameter distribution (volume/size) of microparticles formed in water at 37 °C from a 6mg/mL polymer solution. These results correlate with previous work done with a chemically synthesised poly(VPAVG), that also demonstrated that there is a dependence between the heating rate and the mean size of microparticles [3].



Fig. 7 – DLS analysis of volume diameter distribution of microparticles formed at 37 °C.

Fig. 8 shows micrographs of lyophilized/dehydrated polymer below and above the T_t . Below the T_t the polymer appears in a disordered state without any structure conformation. On the other hand, in the situation above the T_t , the polymer self-aggregates forming spherical particles. The size of



particles appears to be smaller than those obtained by DLS, which can be explained by the drying step in the sample preparation procedure.



Fig. 8 – Micrographs of a lyophilized/dehydrated (VPAVG)₂₂₀ polymer below and above the T_t . A – SEM image showing the disordered state of a lyophilized polymer before the particle preparation (see Material and Methods) and below the T_t . B – SEM image showing the size and shape of aggregates formed during the particle preparation (see Material and Methods) above the T_t .

Conclusions

The use of genetic engineering tools allowed the obtention of protein-based polymers with an absolute control over their molecular weight and architecture. In this work and taking advantage of those tools, an ELP was designed and produced always keeping in mind maximal production with minimal low cost. Applying the CCD the media recipe was redesigned achieving a greater cell biomass minimizing the costs and increasing the final production from 2mg/L to 6mg/L of $(VPAVG)_{220}$. Results from the DSC analysis showed a bio-produced polymer behaving the same way as the chemical synthesized. The influence of NaCl and Urea in the ITT of a VPAVG polymer was followed by conventional DSC showing that it is possible to modulate the T_t by exploiting the NaCl and Urea concentrations. According to the results obtained by DSC assays, the addition of NaCl causes a concentration-dependent decrease in the T_t and a subsequent decrease in the enthalpy. While the addition of NaCl to the system suggests that it leads to a better organization of the polymer when in the folded state and an increase in the number of ordered structures, the addition of Urea disrupts the bulk water, decreasing the hydrophobic effect.

When above its T_t , the polymer self-assembles into spherical particles with a diameter distribution ranging from 700nm to 2µm. Further work will lay on drug delivery studies, focusing on hydrophobic compounds and the possibility of making biodegradable films.

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