



Superficies y vacío

ISSN: 1665-3521

[alex@fis.cinvestav.mx](mailto:alex@fis.cinvestav.mx)

Sociedad Mexicana de Ciencia y  
Tecnología de Superficies y Materiales  
A.C.  
México

Hernández - Molina, R.; Hernández - Márquez, J.A.; Enríquez - Carrejo, J.L.; Farias -  
Mancilla, J.R.; Mani - González, P.G.; Viguera Santiago, E.; Rodríguez - Aranda, M.C.;  
Vargas - Ortiz, A.; Yáñez - Limón, J.M.

Synthesis by wet chemistry and characterization of LiNbO<sub>3</sub> nanoparticles

Superficies y vacío, vol. 28, núm. 4, diciembre, 2015, pp. 115-118

Sociedad Mexicana de Ciencia y Tecnología de Superficies y Materiales A.C.  
Distrito Federal, México

Available in: <http://www.redalyc.org/articulo.oa?id=94245280004>

- How to cite
- Complete issue
- More information about this article
- Journal's homepage in [redalyc.org](http://redalyc.org)



Scientific Information System

Network of Scientific Journals from Latin America, the Caribbean, Spain and Portugal

Non-profit academic project, developed under the open access initiative

## Synthesis by wet chemistry and characterization of LiNbO<sub>3</sub> nanoparticles

Hernández-Molina R., Hernández-Márquez J.A., Enríquez-Carrejo J.L., Farias-Mancilla J.R., Mani-González P.G.\*  
*Instituto de Ingeniería y Tecnología, Departamento de Física y Matemáticas, Universidad Autónoma de Ciudad Juárez,  
 Ave. Del Charro 450, Cd. Juárez C.P. 32310, Chihuahua. México.*

Vigueras Santiago E.

*Laboratorio de Investigación y Desarrollo de Materiales Avanzados (LIDMA), Facultad de Química, Universidad  
 Autónoma del Estado de México, Paseo Colón esquina Paseo Tollocan, Toluca Estado de México CP 50000, México.*

Rodríguez-Aranda M.C.

*Coordinación para la Innovación y Aplicación de la Ciencia y la Tecnología, Univ. Autónoma de San Luis Potosí  
 Sierra Leona #550, Lomas 2a. Sección, San Luis Potosí, S.L.P. México*

Vargas-Ortíz A.

*Facultad de Ingeniería Mochis, Universidad Autónoma de Sinaloa,  
 Ciudad Universitaria, C.P. 81223, Los Mochis, Sinaloa, México.*

Yáñez-Limón J.M.

*Centro de Investigación y Estudios Avanzados del IPN., Unidad Querétaro,  
 Apdo. Postal 1-798 76010, Querétaro Qro. México  
 (Recibido: 8 de mayo de 2015; Aceptado: 27 de noviembre de 2015)*

Actually, lithium niobate (LiNbO<sub>3</sub>) has been used for optical wavelength conversion and ultrafast optical signal processing because of its outstanding rapid nonlinear optical response behavior, low switching power and broad conversion bandwidth. LiNbO<sub>3</sub> nanoparticles, which belong to the ferroelectric oxide class, were synthesized by chemical reaction with wet chemistry. Their size distribution was centered around 200 nm. X-ray diffraction (XRD) and scanning electron microscopy (SEM) were used to further investigate the quality of the obtained LiNbO<sub>3</sub> powders. The present work shows that by employing this chemical method the correct stoichiometric phase was obtained. This was corroborated by XPS (X-Ray Photoelectron Spectroscopy) results. Also, the nanoparticles showed a defined crystallinity and uniform morphology. This way of obtaining nanoparticles is innovative because of its low cost and simple way to reproduce it. It is an important method of increasing the surface area, controlling the phase purity and reducing the particle size distribution. The samples were obtained under low temperature annealing at 500, 650 and 800 °C. Those features can be controlled using variables such temperature, time of synthesis, and calcination. In previous works it was found that hydrothermal methods offer many advantages over conventional ceramic synthesis methods.

**Keywords:** LiNbO<sub>3</sub>; wet chemistry; XRD; XPS; SEM; nanoparticles.

### 1. Introduction

Nowadays there is an interest in obtaining LiNbO<sub>3</sub> nanoparticles due to its ferroelectric properties and because they present a wide variety of applications in non-linear optics, thin film capacitors, pyroelectric detectors, optical memories, electro-optics modulators, and others [1-5]. As already known, there are several ways to produce LiNbO<sub>3</sub>, however wet chemical methods are a promising alternative because of its low time-consuming processing steps, moreover the ability to regulate the uniformity of nucleation and improvement in the control of size and morphology of the material [6].

When LiNbO<sub>3</sub> is formed by high energy milling, it needs high temperatures and a long calcination time to complete the reaction. This brings up disadvantages like contamination and enlargement of the grain during the material processing period [7]; at the same time it tends to generate a lack of homogeneity in the stoichiometry composition and size of the particle. This is why a great need to develop an alternate synthesis method with a higher

quality and an easier reproduction process arises [8]. In general, the wet chemistry method has some benefits with respect to the traditional method, and even compared to other materials, such as Pb and Ti nanoparticles [9,10]. The use of a wet process synthesis methodology is recommended to be performed, taking into consideration that Villegas *et al.* and Tabata *et al.* point out that the chemical process has an advantage over the traditional method because it works with lower temperatures and there is more control over the composition and the particle size [11,12].

The processing of LiNbO<sub>3</sub> by the co-precipitation method uses H<sub>2</sub>O as a dispersion medium (solvent). First, the Li<sub>2</sub>CO<sub>3</sub> reacts with the H<sub>2</sub>O producing LiOH and then we are able to produce solvation with the Nb<sub>2</sub>O<sub>5</sub>, and finally obtain the phase of LiNbO<sub>3</sub> [13,14].

The synthesis and calcination temperatures were low to avoid phenomena like nucleation and synthetization. This led us to obtain LiNbO<sub>3</sub> nanoparticles.

\* pierre.mani@uacj.mx

## 2. Experimental Details.

The precursors employed were  $\text{Li}_2\text{CO}_3$  99.999% SigmaAldrich,  $\text{Nb}_2\text{O}_5$  99.999% SigmaAldrich and DI  $\text{H}_2\text{O}$ .  $\text{Li}_2\text{CO}_3$  was mixed with DI  $\text{H}_2\text{O}$  until dissolved for about 30 minutes at 85 °C forming  $\text{LiOH}$ . After that, the  $\text{Nb}_2\text{O}_5$  is added with vigorous agitation for 1 hour, then the powders are dried at room temperature. Finally, the samples are exposed to a thermal treatment at 650 °C for 12 hours. Time and pH during the reaction were common parameters. The samples were characterized by using PANanalytical XpertPRO diffractometer with  $\text{Cu K}\alpha$  radiation and 2 $\theta$  diffraction angle from 10 to 100 degrees. TGA and DSC analysis of powders were also carried out. Crystal morphology and size were analyzed by SEM. For XPS analysis, a JEOL JPS 9200 with monochromatic  $\text{Al K}\alpha$  source with resolution of 0.65 eV was employed.

## 3. Results

### A. TGA, DTA and DSC analysis

In Figure 1, the weight change can be visualized (identified with green), showing a weight loss in the 500-700 °C range possibly because of some remaining water and carbonate vaporization, which is also corroborated by the peak in the temperature difference graph (identified with purple). Heat flow graph (identified with blue) shows the exothermic peak due to the physical structure change [15,16]. Those changes suggest the formation of  $\text{LiNbO}_3$  phase that will be supported with XRD data below. After 650 °C the suggested formation of  $\text{LiNb}_3\text{O}_8$  is observed.

### B. XRD

The diffractogram shown in Figure 2 reveals that the reaction was successful and it was possible to obtain the  $\text{LiNbO}_3$  phase when the powders were treated at temperatures of 500, 650 and 800 °C. At 500 °C, the carbonate removal is not complete and signals of the precursor jet are present.

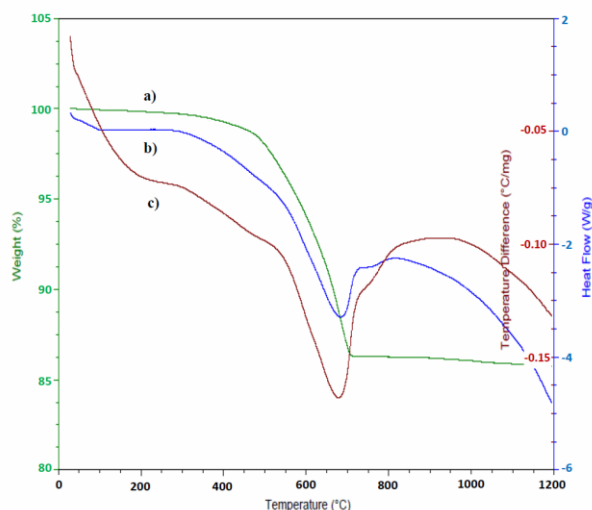


Figure 1.  $\text{LiNbO}_3$  Nanoparticles analysis a) TGA b) DSC c) DTA

At 650 °C, the phase is formed and it should be noted that crystallite size is about 37.6 nm according to the Scherrer equation [17]. At 800 °C, the diffractogram related to  $\text{LiNb}_3\text{O}_8$  appears and further increasing the temperature results in a decrease of the phase. Then, with high or low temperature it is possible to have lack or excess of lithium. In addition, the lithium excess is due to possible rearrangements of the structure as residual stress [18]. If there is lithium loss it is due to volatilization during annealing treatment while if there is lithium excess it is due to changes in the atomic structure. This result was observed by Bouquet *et al.* [19].

### C. XPS

The core level signals obtained from the measured  $\text{LiNbO}_3$  particles were the Nb 3d and O 1s, which verify that the nanoparticles are stoichiometric and are formed without structural defects at the surface.

The Nb 3d signal at 206.8 eV allows the identification of the chemical environment and it was associated with chemically stable niobium. Therefore, it is bound to oxygen and neither oxidation nor reduction is present. The Figure 3 shows a small signal to larger binding energies that is associated with charge density of Nb-O-Li bond at the structure.

For O 1s, it is possible to see in Figure 4 that the signal associated to niobium-lithium bonds at 534.4 eV is a contribution of residual oxygen in the particles surface. For 532.4 eV signal, it is associated to  $\text{LiNbO}_3$  and the 530.5 eV signal is associated to lithium-oxygen bond due to the low binding energy [20].

These results for the sample treated at 650 °C for 12 h are evidence that with this method the correct phase is obtained and the results are satisfactorily supported by XRD. Therefore, the particles had the required structure and stoichiometry. Then, the stoichiometry is directly related with the loss or excess of lithium observed by XRD. The nanoparticles were stable and were formed without defects at the structure.

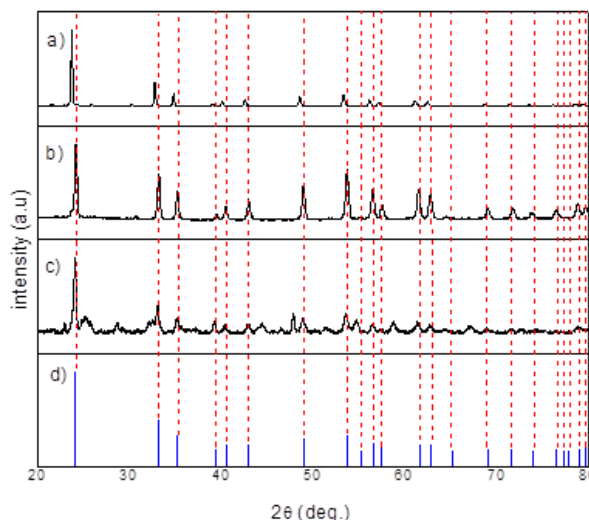
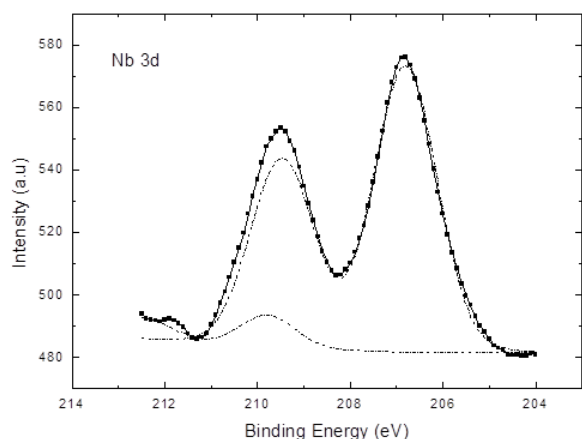
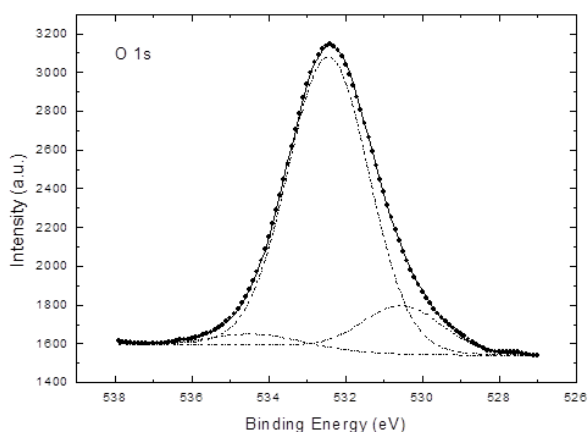


Figure 2. XRD a) 500 °C, b) 650 ° C, c) 800 ° C for 12 hr and d) PDF of  $\text{LiNbO}_3$  [21], obtained  $\text{LiNbO}_3$  nanoparticle.



**Figure 3.** Nb 3d XPS signal is associated to niobium in  $\text{LiNbO}_3$  with an evidence of signal of low density charge due to Niobium-Oxygen-Lithium.



**Figure 4.** O 1s XPS signal. Identifiable peaks associated to surface oxygen, oxygen in  $\text{LiNbO}_3$  and oxygen with lithium bond.

#### D. SEM

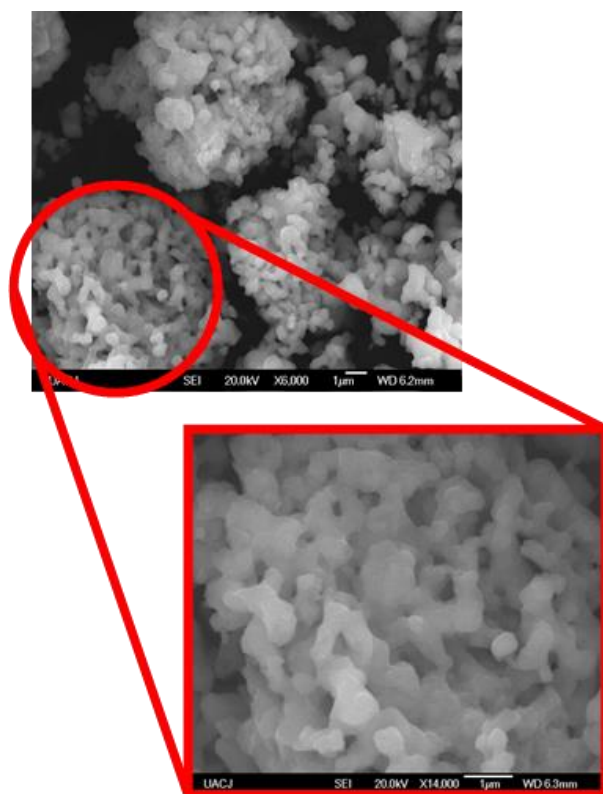
The morphology of the samples was qualitatively analyzed using SEM. Figure 5 shows the agglomerated spherical particles.

The size of the  $\text{LiNbO}_3$  nanoparticles was very uniform; it ranged from 100 to 400 nm. The size distribution of the nanoparticles is shown in Figure 6. The control over the particle size is an evidence since the majority are around 200 nm.

#### 4. Discussion

Under low temperature annealing it was possible to achieve the phase of  $\text{LiNbO}_3$  nanoparticles. The  $\text{LiNbO}_3$  nanoparticles were obtained by the chemical method at low temperature-energy. This method allows a convenient mechanism in the synthesis of the nanoparticles. Compared to the traditional ceramic method, we can control the phase and particle size. XPS supported the method and there was evidence of the correct phase when compared to XRD results.

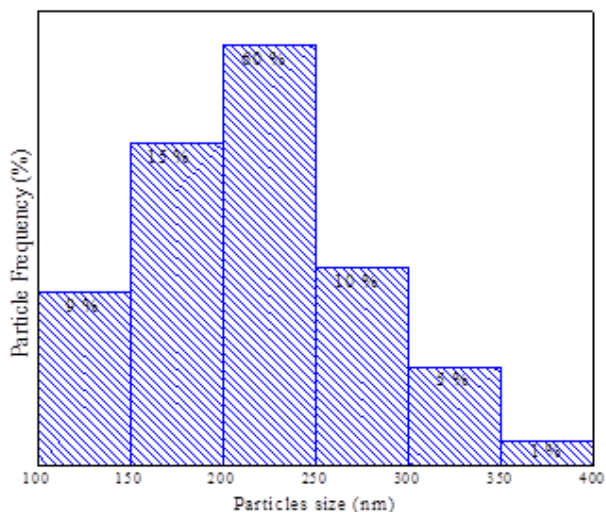
The size of the  $\text{LiNbO}_3$  nanoparticles was uniform (around 200 nm) by using this method. Samples with 85 °C and



**Figure 5.** SEM micrographic to show the size of  $\text{LiNbO}_3$  nanoparticles.

650 °C thermal treatment at synthesis and sintering, respectively, are the best ways to obtain the desired phase.

Contribution of  $\text{Li}_2\text{CO}_3$  dissolution is a principal factor for the reaction. This method has advantages compared with high energy thermal milling (traditional ceramic method), as it consumes less energy, has more control over the temperature, particle size, and the final phase of the nanoparticles. The results of this work could be useful for other  $\text{LiNbO}_3$  processing techniques to obtain thin films.



**Figure 6.** The particle size histogram shows the size distribution of  $\text{LiNbO}_3$  nanoparticles.

## Acknowledgments

One of the authors (P.G Mani-Gonzalez) acknowledges the support from NANOTECH-CIMAV. Thanks for the collaboration of Enrique Torres, Pedro Piza, Daniel Lardizabal for XRD, Raman and TGA, DSC measurements. Also thanks to Gustavo Lopez Tellez from UAEM-UNAM for XPS measurements. This work was support by F-PROMEP-38/Rev-03, SEP-23-005.

## References

- [1]. M. Aufray, S. Menuel, Y. Fort, J. Eschbach, D. Rouxel, B. Vincent, *J. Nanoscience & Nanotechnology* **9**, 4780 (2009).
- [2]. M. Kadota, Y. Ida, T. Kimura, *Japanese Journal of Applied Physics* **51**, 07GC14 (2012).
- [3]. J.D. Brownridge, S.M. Shafroth, arXiv preprint physics/0303040, (2003).
- [4]. Chang, C. C., Russell, K. L., & Hu, G. W. *Applied Physics B* **72**, 307 (2001).
- [5]. G.K. Gopalakrishnan, C.H. Bulmer, W.K. Burns, R.W. McElhanon, A.S. Greenblatt, *Electronics Letters* **28**, 826 (1992).
- [6]. W.L. Suchanek, R.E. Riman, *Advances in Science and Technology* **45**, 184 (2006).
- [7]. M.N. Palatnikov, N.V. Sidorov, V.T. Kalinnikov, *Journal of Inorganic Materials* **47**, 768 (2011).
- [8]. D. Mohanty, G.S. Chaubey, A. Yourdkhani, S. Adireddy, G. Caruntu, J.B. Wiley, *RSC Advances* **2**, 1913 (2012).
- [9]. L.P. Ramírez-Rodríguez, M. Cortez-Valadez, J.G. Bocarando-Chacon, H. Arizpe-Chávez, M. Flores-Acosta, S. Velumani, R. Ramírez-Bon, *Nano* **9**, 1450070 (2014).
- [10]. R. Britto-Hurtado, M. Cortez-Valadez, Ramón A. B. Alvarez, P. Horta-Fraijo, J.-G. Bocarando-Chacon, R. Gámez-Corrales, A. Pérez-Rodríguez, F. Martínez-Suárez, F. Rodríguez-Melgarejo, H. Arizpe-Chavez, M. Flores-Acosta, *Nano* **10**, 1550069 (2015).
- [11]. M. Villegas, C. Moure, J.R. Jurado, P. Duran, *Boletín Soc. Esp. Ceram. Vidr.* **30**, 357 (1991).
- [12]. K. Tabata, T. Choso, Y. Nagasawa, *Surface Science* **408**, 137 (1998).
- [13]. Y. Lin, H. Yang, J. Zhu, F. Wang, H. Luo, *Materials and Manufacturing Processes* **23**, 791 (2008).
- [14]. B. Prakash, B.S. Jaya, *Indian Journal of Pure and Applied Physics* **50**, 320 (2012).
- [15]. X.M. Chen, *Journal of materials science: materials in electronics* **7**, 51 (1996).
- [16]. T. Hatakeyama, L. Zhenhai, *Handbook of thermal analysis* (Wiley, Chichester 1998).
- [17]. A.L. Patterson, *Phys. Rev.* **56**, 978 (1939).
- [18]. Y. Repelin, E. Husson, F. Bennani, C. Proust, *Journal of Physics and Chemistry of Solids* **60**, 819 (1999).
- [19]. V. Bouquet, E. R. Leite, E. Longo, J. A. Varela, *J. Eur. Ceram. Soc.* **19**, 1447(1999).
- [20]. M.A.B Gomes, L.O. de S. Bulhões, S.C. de Castro, A.J. Damião, *J. Electrochem. Soc.* **137**, 3067 (1990).
- [21]. H.E. Swanson, H.F. McMurdie, M.C. Morris, E.H. Evans, *National Bureau of Standards, monograph* 25, Section 6 (1968).