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#### Advanced methods for analysis of mixed gas diffusion in polymeric membranes

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#### 18 ABSTRACT

19 The rapid advancement of membrane gas separation processes has spurred the development of new and more 20 efficient membrane materials, including polymers of intrinsic microporosity. The full exploitation of such 21 materials requires thorough understanding of their transport properties, which in turn necessitates the use of 22 powerful and reliable characterization methods. Most methods focus on the permeability, diffusivity and 23 solubility of single gases or only the permeability of mixed gases, while studies reporting the diffusion and 24 solubility of gas mixtures are extremely rare. In this paper we report the use of a mass-spectrometric residual 25 gas analyser to follow the transient phase of mixed gas transport through a benzotriptycene-based 26 ultrapermeable polymer of intrinsic microporosity (PIM-DTFM-BTrip) and a polydimethylsiloxane (PDMS) 27 membrane for comparison, via the continuous online analysis of the permeate. Computational analysis of the 28 entire permeation curve allows the calculation of the mixed gas diffusion coefficients for all individual gases 29 present in the mixture and the identification of non-Fickian diffusion or other anomalous behaviour. The mixed 30 gas transport parameters were analysed by three different approaches (integral, differential and pulse signal), 31 and compared with the results of the 'classical' time lag method for single gases. PDMS shows very similar 32 results in all cases, while the transport in the PIM gives different results depending on the specific method and 33 instrument used. This comparative study provides deep insight into the strengths and limitations of the different instruments and data elaboration methods to characterize the transport in rubbery and high free volume glassy 34 35 membranes with fundamentally different properties and will be of help in the development of novel membrane 36 materials.

#### 38 Keywords

Gas separation membrane, solution-diffusion model, mixed gas diffusion, time lag, transport properties,
 mathematics of diffusion.

41

#### 42 **1. Introduction**

In a variety of industrial gas separation processes, such as oxygen enrichment or pure nitrogen production from air  $(O_2/N_2)$  [1], hydrogen separation from ammonia tail gas  $(H_2/N_2)$  [2,3], natural gas sweetening  $(CO_2/CH_4)$  [4,5], membrane-based gas separation processes are a consolidated technology [6,7]. It is emerging or under study for other separations, such as biogas upgrading  $(CO_2/CH_4)$  [8,9] or carbon capture from flue gas  $(CO_2/N_2)$  [10–12]. The successful introduction of new applications requires the best possible materials and process configuration for that specific separation [13], and this, in turn, requires precise knowledge of their transport properties.

#### 50 1.1. Gas transport in dense membranes

It is well-known that the transport of gases in dense membranes takes place according to the so-called Solution-Diffusion mechanism [14,15], in which the gas is first absorbed in the membrane at the high-pressure side, to then diffuse across the membrane in the direction of decreasing concentration gradient, and finally desorb at the downstream side. In the simplest and most commonly used description, the gas solubility is constant and the equilibrium concentration follows Henry's law:

$$c = p \times S$$
 Eq. 1

where *c* is the gas concentration in the membrane in equilibrium with the feed gas, *p* is the feed pressure (or partial pressure of the gas) and *S* is the gas solubility coefficient. The gas diffusion coefficient, *D*, follows Fick's first and second law, defining the diffusion flow rate, *J*, in one dimension as:

$$J = -D \times A \frac{dc}{dx}$$
 Eq. 2  
$$\frac{dc}{dt} = D \times \frac{d^2c}{dx^2}$$
 Eq. 3

where dc/dx is the concentration gradient across the membrane, dc/dt its change in time, and A is the area of the membrane. Eq. 2 and Eq. 3 are completely true only when D is not a function of the concentration in the membrane, and the simplest theories used to describe the transport and to calculate the permeability, solubility and diffusion coefficients of the gases, rely on a series of assumptions such as a constant, concentration- and time-independent D and S. In this model and with these assumptions, the permeability, P, can be calculated as:  $P = D \times S$ 

Eq. 4

65 Most studies on the transport properties of new membranes rely on conceptually simple measurements, 66 often only concerning single gases that give, at best, an approximation of the membrane performance in their 67 final application. In practice, for many materials the situation is much more complex, and S and D are either 68 not constant with time or pressure or they depend on the presence of other gas species in the mixture. Therefore, 69 it is useful to analyse the behaviour of a membrane under different conditions and assess how much it differs 70 from ideality. Such studies require a careful experimental investigation of the process with the most suitable 71 methods. With that in mind, the present manuscript will discuss the advantages and limitations of the different 72 methods for the analysis of the gas transport properties of membranes, discussing several advanced 73 measurement techniques and data elaboration methods, with the aim of providing a deeper insight into the 74 transport properties of novel membrane materials. We will describe the transport in two very different 75 polymers: the rubbery polydimethylsiloxane (PDMS), and the glassy ultrapermeable polymer of intrinsic 76 microporosity PIM-DTFM-BTrip [16] (Figure 1).

77



Figure 1. Polymers used in the present work.

78

79 PDMS is a benchmark polymer for gas separation applications, and at room temperature, it is far above its 80 crystalline melting point, so it is in its amorphous rubbery thermodynamic equilibrium state. Instead, PIM-81 DTFM-BTrip is an amorphous glassy polymer in a thermodynamic non-equilibrium state, far below its glass 82 transition temperature, which is usually above the degradation temperature for PIMs [17]. PIM-DTFM-BTrip 83 contributed to the definition of the latest  $CO_2/CH_4$  and  $CO_2/N_2$  upper bounds [16] due its particular 2D chain 84 structure [18] and high rigidity [19]. The latter is common for PIMs [20] and, in combination with the contorted 85 backbone structure, prevents the efficient packing of the polymer chains in the solid state, providing a large 86 fractional free volume [21,22], which typically results in high permeability and high selectivity. This 87 combination makes PIMs responsible for all the main shifts in the Robeson Upper bounds since 2008 [16,23– 88 25]. The fluorinated groups in PIM-DTFM-BTrip are likely to decrease the cohesive forces in the polymer 89 matrix, reducing its tendency to undergo physical aging, a feature that makes glassy perfluoropolymers rather 90 unique, because they tend to age less rapidly [26] compared to other high free volume polymers such as 91 poly(trimethylsilylpropyne) PTMSP [27]. At the same time, their high hydrophobicity is expected to make the 92 permeability less influenced by the humidity in the gas stream [28] and its fluorinated nature provides unique sorption properties [29]. PIMs are furthermore known to be sensitive to pressure and to mixed gas composition
[24], the feature that makes them most interesting for this comparative study.

#### 95 *1.2. Methods for the analysis of transport parameters*

96 The most straightforward methods employed to analyse the transport properties in membranes, measure 97 directly the gas permeation rate in either a dead-end cell, in the case of pure gases or vapours, or in a cross-98 flow cell for mixtures [30]. The permeate flow rate can be measured directly, with a bubble flow meter or more 99 sophisticated electronic flow meters, or indirectly, via the concentration in a sweeping or carrier gas with 100 known flow rate [31]. In a fixed volume setup, it is calculated from the pressure increase rate of the permeate 101 volume. The gas solubility can be determined directly via gravimetric or volumetric sorption measurements, 102 even in complex systems [32], and the diffusion coefficient, i.e. the transport diffusivity of permeating gases, 103 can be assessed under transient conditions from either sorption kinetics [33] or permeation kinetics 104 measurements, the latter typically via the so-called time lag method. Other methods include more complex 105 analysis, such as NMR spectroscopy [34,35] or molecular modelling approaches [36,37]. While well-calibrated 106 instruments should be able to provide the same results, this is often not the case because of the materials 107 properties and the measurement principle used or because of the operational conditions [38,39].

- 108 The time lag method for the analysis of the diffusion coefficient in polymer films was first reported about 109 a century ago by Daynes [40]. Since then, it became by far the most commonly used technique for the analysis 110 of gas transport parameters in polymers and in porous materials [41]. Its use has been extended to the 111 quantification of hydrogen diffusion in metals [42] or even to salt diffusion in liquid phase membranes [43] 112 with the latter using not only the downstream concentration but also the upstream concentration [44]. In its 113 simplest form, the method consists in the measurement of the total amount of gas in the permeate, usually 114 determined as the pressure in a fixed permeate volume. The method showed some limitations, related to the 115 effect of the instrument itself on the gas transport [45-47], in combination with non-ideal properties of the 116 materials [48]. This might require minor adjustments in the calculations, but these are usually only important 117 in some extreme cases and they did not prevent this method becoming one of the most widely used techniques. 118 One of the most important limitations of the classical time lag technique is that the measurement of pressure
- 119 allows the analysis of a single species only. It is much more difficult to determine the transient behaviour and 120 to obtain information on diffusion and solubility of gas mixtures, because the most commonly used gas 121 chromatographic analysis of the permeate composition is too slow to follow the transient phase of permeation. 122 Thermal conductivity detectors can be used to follow changes in the permeate (sweep gas + permeate) 123 continuously [31], but they are unable to analyse the composition of complex mixtures, for which a 124 combination with for instance gas chromatography (GC) is needed. Several methods have been reported, based 125 on sorption measurements [49-52], on permeation measurements with selective condensation of the least volatile component [53,54], on computer simulations [55,56] or on alternative techniques such as NMR 126 127 spectroscopy [35]. Many of these suffer from complex experimental procedures or data elaboration methods. 128 Various studies have proposed online mass spectrometry to measure the permeation transient in pervaporation
- 129 [57,58] or gas permeation [59–61]. This requires a more sophisticated instrumentation (mass spectrometer),

130 but it facilitates the measurement procedure. Inspired by this earlier work, we designed a gas permeability 131 setup with fast online analysis of the permeate composition in a standard cross-flow cell via mass spectrometry [62], and we further optimized this method for the determination of the mixed gas diffusion coefficients 132 [63,64], that is also suitable for highly permeable polymeric membranes with a fast transient, unlike much 133 134 slower GC analysis. The method determines the time lag by the tangent-method in a plot of the cumulative permeate volume versus time, measured via online analysis of the gas composition in the permeate by mass 135 136 spectroscopy, followed by integration of the concentration over time to yield the total amount of permeate. 137 The time lag must be corrected for the instrumental delay, namely the time lag determined from the response 138 of the system when exposing an aluminium disk with pinhole to the gas, or when extrapolating the time lag of 139 membranes with different thicknesses to zero thickness. We assumed that the instrumental time lag 140 corresponded to average residence time of the gas in the system without membrane. The response of the system 141 was found to be slightly slower at higher pressure, because of the need to pressurize the feed line with the gas 142 [64]. For a somewhat analogous situation, where the Knudsen diffusion of the gas from the membrane to the 143 analyser causes an additional delay, Kruczek et al. showed that the correction factor should not be equal to the 144 time lag due to the Knudsen diffusion, but slightly smaller [45]. For the flow of a gas pulse in a cylindrical 145 tube, Evans and Kenney discussed that the average residence time of a species flowing in a laminar flow 146 corresponds to the peak maximum at a distance from the injection point [65], analogous to what Tailor 147 described for liquid flow [66]. This suggests that the time lag of the membrane-less system is indeed not the 148 best correction for the instrumental delay, as we have assumed previously [63], because it slightly deviates 149 from the average residence time. For diffusion in dense membranes, Beckmann et al. have discussed the 150 comparison between the classical pressure increase curve with its derivative, corresponding to the typical 151 permeate concentration curve or flow rate curve after a step change in a cross-flow cell, and with its second 152 derivative, corresponding to the signal after a pulse change in the feed concentration [67,68] (see Section 5).

Recently we have shown that we can also determine the permeation time lag directly from the measured permeate flow rate as a function of time, *i.e.* from the original signal, which is the mathematical derivative of the time lag curve [69]. The inflection point in this sigmoidal curve corresponds to the peak in the pulse signal, and the inflection point of the signal in a membrane-free test run can be used for the correction of the instrumental lag time.

158 In this work, that aims to develop even more versatile and powerful methods to characterize the transport 159 properties of membrane materials for gas separation, we compared each of the above methods to measure 160 mixed gas permeation and diffusion, via the integral or time lag curve, the derivative curve, and the pulse 161 curve, using two fundamentally different materials. Being glassy polymers with high free volume, PIMs are 162 known for their deviation from simple Fickian diffusion and for their pressure- and composition-dependent 163 transport properties. PDMS will therefore be used for the method setup and for comparison as a well-defined 164 benchmark membrane material, while PIM-DTFM-BTrip is used for validation and analysis of the sensitivity 165 and the strengths and limits of the methods.

#### 166 **2. Experimental**

#### 167 2.1. Materials and membrane preparation

A dense PDMS membrane was prepared from a two-component resin (SYLGARD® 184, Dow Corning Midland). The prepolymer and crosslinker were mixed in the ratio 10:1 according to the instructions of the supplier, and the resin was cured at room temperature over the weekend in a Teflon Petri dish. The final membrane thickness resulted  $1056 \pm 23 \mu m$ , measured with a Mitutoyo model IP65 digital micrometer as an average of 10 points. The exposed area inside the footprint of the sealing ring was 13.84 cm<sup>2</sup> for both the pure and mixed gas permeability measurements.

An aluminium sample with a pinhole was prepared by puncturing a 50  $\mu$ m thick aluminium disk (Ø 47 mm) with the extremity of a syringe needle, to leave an imperceptible pinhole that is only visible by the naked eye in backlight and that exhibits a nitrogen flow rate of 0.92 cm<sup>3</sup><sub>STP</sub> min<sup>-1</sup> at 1 bar. This is in a similar range as the CO<sub>2</sub> permeability of the most permeable membranes, and thus it allows the use of the same calibration data.

The PIM used in this study was PIM-DTFM-BTrip (Figure 1) and its preparation was described previously [16]. The membrane was prepared by solution casting from quinoline, solvent evaporation, and subsequent treatment with methanol to remove the residual solvent and reset the sample history. The sample was masked with aluminium adhesive tape to reduce its active area to 0.785 cm<sup>2</sup> and the average thickness in this area was 112 ± 6 µm. A long-term 1380 days aged sample, with well-known physical-chemical properties and ageing history [16], was used to guarantee maximum stability of the sample during the measurement campaign.

#### 184 *2.2. Gas permeation measurements*

Pure gas (fixed volume instrument) and mixed gas (fixed pressure instrument) permeation measurements were carried out with setups already described previously [63], and the detailed description of the standard experimental procedures are reported in the supporting information. The unique feature of the mixed gas permeation setup is that the gas composition is analysed continuously by means of a mass spectrometric residual gas analyser, which allows the simultaneous determination of multiple gas species in a mixture.

190 The nonstandard pulse measurements, with a short exposure of the membrane to the gas or gas mixture, 191 were carried out flushing the feed side with a dilute certified mixture, containing 3% CO<sub>2</sub> and 3% CH<sub>4</sub> in 192 argon, instead of pure argon before the measurement. This moderately increases the baseline signal for CH<sub>4</sub> 193 and CO<sub>2</sub> and allows a higher measurement frequency and more accurate analysis. The feed stream is then 194 substituted by the gas of interest by setting directly its flow rate at the maximum of 500 cm<sup>3</sup> min<sup>-1</sup> with the 195 mass flow controller, rather than switching the six-way valve as seen in the differential method. Thus, at fixed 196 moments, the membrane is exposed to pulses of the gas of interest for 1s, 2s, 4s, 6s, 8s and 10s at a flow rate 197 of 500 cm<sup>3</sup> min<sup>1</sup>. The use of the MFCs for the pulse control causes a slightly slower response of the system, 198 compared to the manual switching of the six-way valve, but it can be controlled entirely by the Flowplot 199 software. This procedure is repeated for at least three different pressures, both for the membrane samples and

for the control sample with pinhole. The raw MS signal is then elaborated by the procedure reported previously
[63] to calculate the volumetric permeate flow rate at STP conditions.

#### 202 2.2.1. Data elaboration

The fixed volume instrument provides the data of the permeate pressure (in mbar with 4 decimals) and the feed pressure as a function of time and these are elaborated as such [63]. The raw MS measurement signal (gas partial pressure in Torr as a function of time) is elaborated by the procedure reported by Fraga *et al.* [63], using argon as the internal standard to calculate the volumetric permeate flow rate (in STP) as a function of time as the basis for further elaborations. The background signal of the MS residual gas analyser is subtracted before calculation of the flow rates.

209 The experimental data are fitted with the appropriate equations described in Annex 1, namely: Eq. A1 for 210 the permeation curves obtained in the fixed volume setup; Eq. A5 for the cumulative permeate volume obtained 211 in the fixed pressure setup or after integration of the differential signal, as described in detail by Fraga et al. 212 [63]; Eq. A10 for the sigmoidal (derivative) curve obtained after calculation of the permeate flow rate from 213 the raw MS data; and Eq. A11 for the experimental data obtained via the pulse method. The fitting was 214 performed using least square method and the Excel nonlinear Generalized Reduced Gradient (GRG) solver 215 algorithm, after expansion of the Tailor series into 25 terms. The GRG algorithm finds local optimal solutions, 216 which means that the final solution could depend on the guessed starting points, and this is generally 217 recognized by the poor visual fit of the experimental data. For this reason, for all the fitting procedures the 218 initial starting points in the minimization procedure of D and S were set close to the values obtained by the 219 tangent method, and only if this did not lead to a satisfactory fit, the starting values were adjusted manually. 220 The terms related to starting permeate pressures and instrumental leak flow rate were set to zero since they 221 were found to be negligible for the tested membranes in these testing conditions.

#### 222 **3. Results and discussion**

#### 223 *3.1. Gas transport measured by the fixed volume setup*

224 The fixed volume setup is the instrument with the fastest response. The aluminium film with pinhole reaches 225 steady state permeation within a second (Figure SI2) and extrapolation of the steady-state pressure increase 226 curves yields an instrumental time lag of  $0.08\pm0.02$  s, independent of the gas type, feed pressure and permeate 227 volume. This extremely fast response confirms that there is no significant contribution of the instrument itself 228 to the time lag, and therefore no corrections related to the transport of the gas in the downstream side are 229 needed [46,47]. This very short instrumental time lag is indeed negligible compared to the time lag of the same 230 gases in the majority of thick dense polymeric membranes, with exception of extremely fast-diffusing gases 231 such as He and  $H_2$  in very thin membranes or in highly permeable polymers, such as PIMs. Details are given 232 in the supplementary information.

#### 233 3.1.1. Tangent method: PDMS vs PIM-DTFM-BTrip:

234 The original permeation curves of H<sub>2</sub>, He, O<sub>2</sub>, N<sub>2</sub>, CH<sub>4</sub> and CO<sub>2</sub> in PDMS and PIM-DTFM-BTrip are displayed in Figure 2. The curves do not show any perceptible anomalies in both materials, and the initial 235 236 pressure and the initial slope are completely negligible, excluding the presence of significant leaks in the 237 membranes or in the system. The time lag was determined by the tangent method, fitting a straight line through 238 the curve in the linear part from approximately 5x the value of the eventual time lag to the end of the 239 measurement. The time lag in the PDMS film falls in the range of 25-35 s for He and H<sub>2</sub>, and 100-150 s for 240 the other gases, O<sub>2</sub>, N<sub>2</sub>, CH<sub>4</sub> and CO<sub>2</sub>. The differences with the PIM are much larger and the permeation curves 241 show immediately some qualitatively interesting features. The time lag of O<sub>2</sub> is much shorter than that of CO<sub>2</sub>, 242 due to faster diffusion, but the permeability of CO<sub>2</sub> is higher, and thus the final slope of CO<sub>2</sub> is steeper. Instead, 243 N<sub>2</sub> and CH<sub>4</sub> have virtually the same permeability (and thus final slope), but the diffusion in CH<sub>4</sub> is significantly 244 slower. The time lag of He and H<sub>2</sub> is much shorter than that of all other gases

The quantitative data for P, D and S are listed in Table 2 and Table 3 for PDMS and PIM-DTFM-BTrip, 245 246 respectively. This large difference in the diffusion coefficients of the six gases in PIM-DTFM-BTrip, confirms 247 its high size selectivity compared to PDMS, and this is due to its highly rigid glassy nature [20]. The perfectly 248 linear pressure-increase rate in steady state and the qualitative shape of the curves give indications of evident 249 anomalies in the transport in both materials related, for instance, to a downstream pressure accumulation for 250 materials with strongly non-linear sorption isotherms [48], as observed previously for Amine-PIM-1 [70,71]. The enormous difference in size-selectivity between PDMS and PIM-DTFM-BTrip is shown by the very steep 251 252 correlation between the diffusion coefficient and the effective diameter of the gas in the PIM (Figure SI3).





**Figure 2.** Time lag curves of  $H_2$ , He (A),  $O_2$  and  $CO_2$  (B), and  $N_2$ ,  $CH_4$  (C) in PDMS and  $H_2$ , He (D),  $O_2$  and  $CO_2$  (E), and  $N_2$ ,  $CH_4$  (F) in PIM-DTFM-BTrip. The intersection of the tangents (dashed lines) with the time axis show the position of the time lag.

#### 255 3.1.2. Complete curve fit

The least squares fit of the permeation curve of CO<sub>2</sub> and CH<sub>4</sub> with Eq. A1 by the procedure reported in 256 257 section 2.2.1 is shown in Figure 3A,D. The thin lines show the extrapolated curve when fitting only part of the 258 transient zone up to the indicated time, and the calculated values of P, D, and S are plotted in Figure 3B,C, 259 E,F. Although the fitting procedure can be much more complex [72], these results highlight the advantage of 260 the fitting procedure compared to the tangent method. The PDMS curves already converge when the data are fitted only until  $t = 2\Theta$ , *i.e.* long before pseudo-steady state is reached, whereas for the tangent method the 261 262 permeability and time lag are usually determined by extrapolation of the data in the interval from  $t = 5\Theta$  to 263  $t = 10\Theta$ . Interestingly, for the PIM-DTFM-BTrip membrane the curves and the resulting values of P converge 264 only after  $t = 6\Theta$ , while the values of D and S keep changing slightly until  $t = 10\Theta$ . This suggests anomalous behaviour for the PIM, for instance due to a nonlinear sorption isotherm [48] and/or non-Fickian diffusion, 265 266 which are both very common in PIMs [73,74].



**Figure 3.** Least squares fit of the experimental permeation curve of pure CO<sub>2</sub> at a feed pressure of 1 bar in a 1056  $\mu$ m PDMS membrane (A) and a 112  $\mu$ m PIM-DTFM-BTrip membrane (D) according to Eq. A1. The thick green line indicates the closely spaced experimental points and the thin lines show the extrapolated curves upon a partial fit of the experimental data until the indicated time. The quantitative values of *P*, *D*, and *S* for CO<sub>2</sub> and CH<sub>4</sub> for the partial fit are plotted as a function of the total fit interval for PDMS until *t* = 6 $\Theta_{\infty}$  (B C) and for PIM-DTFM-BTrip until *t* = 10 $\Theta_{\infty}$  (E,F).  $\Theta_{\infty}$  is the time lag obtained when the entire curve is fitted. Lines are plotted as a guide to the eye.

269 The residual error between the experimental data and the fit of the entire experimental curve (see Figure 270 SI4A,C) shows very good agreement for PDMS, with less than 0.01 mbar spread for  $CO_2$  and 0.02 mbar for 271 CH4, during the entire measurement time interval. Nevertheless, the weakly undulating trend in especially CH4 272 is a systematic deviation shows that Eq. A1 cannot fit the data precisely. Very close examination of the 273 permeation curve (quantitative data not shown here) reveals that this is because the slope of the curve slightly 274 decreases after 700 s for  $CH_4$ . The situation is significantly different for the PIM. In spite of the apparently 275 smooth determination of the time lag by the tangent method, the integral fit of permeation curve with Eq. A1 shows a marked trend in the residual errors for both CO<sub>2</sub> and CH<sub>4</sub> in PIM-DTFM-BTrip (Figure SI4B,D). Eq. 276 277 A1 cannot describe the experimental permeation curve accurately and the fit underestimates the experimental data in the initial part of the curve, because the transient is broader than expected. This is generally due to pressure-dependence of the solubility and/or the diffusion coefficient and it highlights the main advantage of the fitting procedure. It pinpoints features that the tangent method does not reveal, and thus provides much deeper insight into the transport phenomena

282 The pressure dependence of the transport parameters, in turn, is a result of the dual mode sorption behaviour.

283 Not only solubility is pressure dependent, but also the diffusion of molecules in Henry's and Langmuir sites is

284 believed to be different [33]. Especially in the case of strong interactions, the gas molecules in Langmuir sites

may be partly immobilised [70,71] and the transport in PIMs or high free volume polymers in general may be

- even more complex than that in common glassy polymers due to possible surface diffusion [20,75]. Therefore,
- the traditional time lag method evaluates the transport as effective values of P, D and S, which may deviate

significantly from the real behaviour and is a strong limitation of this method, in spite of its simplicity.

#### 289 *3.2. Gas transport measured by the variable volume setup*

#### 290 *3.2.1. Differential and integral method:*

291 The fundamental difference between the variable volume setup and the fixed-volume setup, is the additional 292 delay due to the average residence time of the gas in the tubes, the valves, the membrane cell and the gas 293 analyser itself, before being recorded by the analyser. This time depends on the total dead volume of the 294 system, and on the flow rate and the pressure in the system, as shown in our previous work. We quantified this 295 delay by two independent methods, in the first with a set of membranes of different thickness and subsequent 296 extrapolating to zero thickness, and in the second with an aluminium disk with a pinhole that provided a 297 negligible time lag. We corrected for this time by subtracting the instrumental time lag  $\Theta_0$  from the actual 298 experimental time lag  $\tau_{TL}$  [63].

$$\Theta_{Mem} = \frac{l^2}{6D} = \tau_{TL} - \Theta_0$$
 Eq. 5

299 This correction, and subsequent calculation of the diffusion coefficient from the difference between the two 300 times, assumes that the time lag of the membrane-less system corresponds to the average residence time of a 301 gas in the system. Kruczek et al. proposed an analytical solution for a similar situation, where additional delay 302 in a fixed-volume setup is caused by Knudsen diffusion in the tube between the membrane and the 303 measurement point [45]. However, they showed that the calculation of the diffusion coefficient should not 304 simply be based on the difference of the effective time lag and the time lag of the instrument, and the correction 305 factor should not be equal to the time lag due to the Knudsen diffusion, but slightly smaller [45]. Indeed, in his 306 original work on the response of a chromatographic system to a pulse injection, Tailor confirmed that the peak 307 maximum corresponds to the average residence time of the solute in the system, which corresponds to a shorter 308 time than the time lag [66]. Thus, the average residence time corresponds to the maximum in the curve in 309 Figure A1C, which coincides with the inflection point in Figure A1B.

In this work, the system's response was determined by measuring the  $CO_2$  and  $CH_4$  flow rate through an aluminium disk with a pinhole, immediately after switching from argon purge gas to the feed gas. The resulting 312 sigmoidal differential signal was used as such, without integration. This signal shows a very abrupt step in the

313 permeate concentration (Figure SI5) and to be fitted correctly with a similar equation as Eq. A10, an additional

lag time,  $t_0$ , needs to be introduced. In the absence of leak flows or baseline signal, i.e. when  $\left(\frac{dV_t}{dt}\right)_{0,STP} = 0$ ,

the permeate flow rate is described by the following equation:

Б

117

$$\frac{dv_{t,STP}}{dt} = A \cdot p_f \cdot S \cdot \frac{D}{l}$$
$$\cdot \left(1 + 2\sum_{n=1}^{\infty} (-1)^n \exp\left(-\frac{D \cdot n^2 \cdot \pi^2 \cdot (t - t_0)}{l^2}\right)\right) \qquad \text{Eq. 6}$$

316 The numerical values of the fit results are given in the supporting information (Table SI1). The data for 317 CO<sub>2</sub> and CH<sub>4</sub> are very similar, and also the data of the pure gases and the mixtures. The average of the four 318 measurements gives the inflection point  $\tau_{INF}$  20.1  $\pm$  2.1 s. Using the above correction of the instrumental 319 inflection point, we are able to evaluate the transport parameters of the PDMS membrane and the PIM-DTFM-320 BTrip membrane directly from the differential signal with a 35/65 Vol% CO<sub>2</sub>/CH<sub>4</sub> mixture and with the pure 321 gases at 1 bar(a) total feed pressure. The measurement was carried out as in our previous work, switching from 322 argon purge gas to the feed gas with the 6-way valve. The normalized results are plotted in Figure 4, where 323 each gas is normalized for its steady state flow rate for immediate comparison of the curve shape, and thus the 324 diffusion behaviour. The permeation kinetics of CO<sub>2</sub> as a pure gas and as a mixture of 35 Vol% CO<sub>2</sub> in methane 325 are identical in the PDMS film, and Eq. A10 fits the data perfectly, without any perceptible difference between 326 the two fits (Figure 4A). Analogously, Eq. A10 fits the CH4 data very well in PDMS, but in this case the mixed 327 gas CH<sub>4</sub> curve slightly anticipates the pure gas curve, which means that CH<sub>4</sub> diffusion is slightly faster in the 328 mixture (Figure 4B). This situation is remarkably different in the PIM, where CO<sub>2</sub> diffusion in the mixture is 329 slightly slower than that of the pure gas, and CH<sub>4</sub> diffusion is much faster with the mixture than with the pure 330 gas. More importantly, Eq. A10 fails to fit the CH<sub>4</sub> data and to a lesser extent also the CO<sub>2</sub> experimental data 331 (Figure 4C,D). Thus, the fit parameters provide accurate transport parameters P, D and S of CO<sub>2</sub> and CH<sub>4</sub> in 332 PDMS but they give at best a rough estimation in PIM-DTFM-BTrip, i.e. the effective averaged values under 333 the given experimental conditions. The numerical values of the transport parameters are summarized in Table 334 2 and Table 3. The fundamentally different behaviour of PDMS and the PIM can be associated to the time 335 scale needed for the polymer segment rearrangement, which is fast in PDMS and slow in the superglassy PIM, 336 even slower with respect to glassy polycarbonates [76] where the more condensable permeant, i.e. CO<sub>2</sub>, acts 337 as dilating agent enhancing the diffusivity of bulkier molecules, i.e. CH<sub>4</sub>.[77,78]

338 One of the reasons for the poor fit of the PIM permeation curve is the nonlinear sorption and the 339 deviation from simple Fickian diffusion in the PIM. Therefore, the transient is very broad and it takes a 340 relatively long time to reach steady state, as discussed above in relation to Figure 3. The second reason is that 341 the diffusion in the PIM is so fast that the transient phase is significantly broadened by instrumental factors. 342 This is at the same time a weakness and a strength of this method, because it makes quantitative determination 343 of the transient phenomena (and thus the diffusion coefficient and the solubility) difficult, but its extreme 344 sensitivity makes this method very effective for the recognition of anomalies in the transport phenomena. It 345 must be noted that this strong deviation from the ideal behaviour is not easily detected when only the tangent method is used, as seen in section 3.1.1, and in the majority of the studies reported in the literature. Indeed, Figure 5 shows the integral of the signals in Figure 4, which take the form of classical time lag curves because they show the total amount of permeate gas as a function of time with the difference, compared to the classical time lag measurements in a fixed volume setup, that it can simultaneously analyse different components in gas mixture. Also in this case, the fit of the curves with Eq. A5 is nearly perfect for PDMS and shows deviations for the PIM, but these deviations seem much less significant than those in Figure 4.





**Figure 4.** Comparison of the normalized pure and mixed  $CO_2$  permeation curves by the **differential method** in (A) PDMS and (C) PIM-DTFM-BTrip, and pure and mixed CH<sub>4</sub> permeation curves in (B) PDMS and (D) PIM-DTFM-BTrip. Feed pressure 1 bar(a) of pure gases or of a 35/65 vol%  $CO_2/CH_4$  mixture. Filled symbols for the pure gases and empty symbols for the mixed gases. The continuous and dashed lines represent the least squares fit of the experimental data with Eq. A10 for pure and mixed gases, respectively. Note the different time scale only for  $CO_2$  permeation in PIM-DTFM-BTrip. The PDMS curves are horizontally shifted for 20.1 seconds to correct for the instrumental lag time, using the average time to reach the inflection point during permeation of the same gases in an aluminium disk with pinhole (Table SI1).

353

The integral method also allows the determination of the mixed gas diffusion coefficient, by calculation of the time lag from the tangent to the steady state volume increase curve and after subtraction of the instrumental time lag [63]. The numerical values of the transport parameters obtained by the fit of the complete curve and obtained by the tangent method are summarized in Table 2 and Table 3.



**Figure 5.** Comparison of the pure and mixed  $CO_2$  permeation curves by the integral method in (A) PDMS and (C) PIM-DTFM-BTrip, and pure and mixed  $CH_4$  permeation curves in (B) PDMS and (D) PIM-DTFM-BTrip. Feed pressure 1 bar(a) of pure gases or of a 35/65 vol%  $CO_2/CH_4$  mixture. Filled symbols for the pure gases and empty symbols for the mixed gases. The continuous and dashed lines represent the least squares fit of the experimental data with Eq. A5 for pure and mixed gases, respectively. Note the different time scale only for  $CO_2$  permeation in PIM-DTFM-BTrip. The PDMS curves are horizontally shifted for 20.1 seconds to correct for the instrumental lag time, using the average time to reach the inflection point during permeation of the same gases in an aluminium disk with pinhole (Table SI1).

#### 360 *3.2.2.* Pulse method

#### 361 3.2.2.1. Method development, instrumental delay time with pinhole

The instrumental residence time is calculated as the maximum in the signal for an aluminium disc with the pinhole after a short pulse with the gas of interest. Ideally, the pulse should be infinitely short, and thus very high to get a reasonably strong signal but, in our setup, the height is limited by the feed pressure, and very short pulses may therefore produce too weak signals. Therefore, some optimization was needed and the instrumental delay was studied with pulses of variable duration, and the experiments were carried out at different pressures. Examples of the response curves of  $CO_2$  and  $CH_4$  are given in Figure SI6 and Figure SI7, respectively. For a given volumetric flow rate, the size of the peak increases with pressure and with pulse 369 duration, because of the longer exposure and the larger amount of gas permeating through the pinhole at a 370 higher driving force. While the onset of permeation is relatively constant, also the time of the peak maximum increases both with the feed pressure and with the pulse duration. A plot of the time of the peak maximum as 371 372 a function of the pulse duration and the pressure is shown in Figure 6, showing a linear dependence of the peak 373 time on the pressure and on the pulse duration. Indeed, it should be expected that the response of the pulse 374 signal on changes in the feed gas stream depends on the average residence time of the gas in the system and should be directly related to the volumetric flow rate and the volume of the system. According to Tailor, the 375 376 average residence time corresponds to the peak maximum [66], and thus:

$$\tau_{Max} = \tau_{Max,0} + \frac{V_{Upstream}}{\Phi_{V,F}}$$
 Eq. 7

Where  $\tau_{Max,0}$  is the delay of the peak maximum after the pulse, due to the average residence time of the gas in the permeate side,  $V_{\text{Upstream}}$  is the volume in the upstream side and  $\phi_{\text{V,F}}$  is the volumetric flow rate of the feed stream. The latter is inversely proportional to the pressure:

$$\Phi_V = \frac{\Phi_{V,STP}}{p_F}$$
 Eq. 8

380

$$\tau_{Max} = \tau_{Max,0} + \frac{p_F \times V_{Upstream}}{\Phi_{V,STP}}$$
 Eq. 9

381 Besides some scatter, the trend in the data for each pulse duration fits reasonably well with a straight line 382 in Figure 6A and Figure 6C. The value of  $\tau_{Max,0}$  in Eq. 9 depends on the configuration of the gas analyser itself, 383 and on ratio between the volume of the connections at the permeate side and the sweep (+permeate) flow rate, but these parameters were not changed in this work and, therefore,  $\tau_{Max,0}$  can be considered constant. Since the 384 385 feed gas was changed by the mass-flow controller, and not instantaneously by switching the 6-way valve, some of the delay in the response may be due to the slow response of the MFC, causing an imperfect step in the 386 387 partial pressure of the feed stream, as described by Favre et al. for the time lag method [79]. In the case of 388  $CO_2$ , the peak maxima range from ca. 14 s to 21 s at 2 bar and from ca. 17 s to 24 s at 6 bar. This relatively 389 narrow range and the low standard deviations of 0.291 s for CO<sub>2</sub> and 0.725 s for CH<sub>4</sub> mean that once the 390 conditions of pressure and pulse duration are fixed, the signal delay due to the instrumental residence time can 391 be estimated accurately. Nevertheless, the determination of the peak maximum from the highest measurement 392 point may cause some scatter due to the low sampling frequency of the MS signal. The fit of the entire peak 393 would probably reduce the scatter in the calculation of the maximum, but this is much more laborious and 394 since the standard deviation is less than 1 second, we considered this accurate enough for the present work.

The original data in Figure SI6 and Figure SI7 show that the 4 s pulse and 6 s pulse show the best compromise between a sufficiently large but not too broad peak, modest peak deformation, and a short time to reach the peak maximum. Therefore, for our further work we decided to use the 6 seconds pulse duration and the average instrumental residence time is determined from the maximum in the permeate signal of the aluminium disc with a pinhole, after exposure to a pulse of this duration with the gas of interest. Since the peak 400 maximum depends on the pulse time, the same pulse length should be considered for the correction if the 401 membrane is also exposed to a pulse. Instead, if the membrane is exposed to a step-change in the feed, *i.e.* in 402 the case of the differential signal in the previous section (Section 3.2.1), the pulse length should be extrapolated 403 to zero to find the position of the inflection point. The latter is used for the correction of the time axis in Figure 404 4.

405



**Figure 6.** Time of peak maximum for  $CO_2$  permeation (A,B) and  $CH_4$  permeation (C,D) through the aluminium disk with pinhole as a function of pressure and pulse duration. The lines represent a least-squares fit with a linear trend in both time and pressure domain. The standard deviation of each individual point is far less than 0.5 s and the data can be described by the equations given in Table 1. The parity plot in Figure SI8 shows a good correlation between the measured and calculated values, especially for  $CO_2$ .

			Standard
			deviation <sup>a)</sup>
Gas	Equation		<b>(s)</b>
CO <sub>2</sub>	$\tau_{peak,C02} = 11.27 + 0.761 \times t_{pulse} + 0.730 \times p_F$	Eq. 10	0.250
$\mathrm{CH}_4$	$\tau_{peak,CH4} = 10.96 + 0.503 \times t_{pulse} + 0.805 \times p_F$	Eq. 11	0.426

Table 1. Results of the least squares fit of the average instrumental residence time of  $CO_2$  and  $CH_4$  as a function of the feed pressure and the pulse time

<sup>a</sup> Standard deviation of all absolute errors in Figure 6

407

#### 408 3.2.2.2. Analysis of the transport parameters of PDMS and PIM-DTFM-BTrip.

409 After identification of 6 s as a suitable pulse duration for the pinhole, an analogous test with the PDMS 410 membrane shows a much wider signal, due to the transient transport in the PDMS film itself (Figure SI9). Qualitatively there is no significant effect of the pulse duration on the peak position from 2-10 s, while only at 411 412 a pulse duration of 2 s, the peak intensity becomes rather low. This confirms the 6 s pulse to be a good choice. 413 The peak width of the pinhole is virtually negligible to that of the PDMS membrane, suggesting that the instrumental residence time is not expected to affect significantly the shape of the permeation curve for the 414 415 PDMS membrane. Figure 7 and Figure 8 show the permeation curves for a 6 s pulse of  $CO_2$ ,  $CH_4$  and their 416 35/65 vol.% mixture in the PDMS film and the PIM-DTFM-BTrip film, respectively at pressures from 1 to 6 417 bar(a).

418 The instrumental response is much faster than that of the PDMS film and the signal of the pinhole in the 419 aluminium film is almost negligible, both in time and in peak width. Therefore, the PDMS peak shape is not 420 significantly affected by the instrumental setup, and subtraction of the average instrumental residence time 421 (Eq. 10 and Eq. 11) from the time axis, should allow fitting of the entire curve with Eq. A11 to calculate all 422 transport parameters P, D and S. On the other hand, the response of the pinhole is only slightly faster than that of the PIM-DTFM-BTrip film, especially for CO<sub>2</sub>, and therefore the signal is likely so much deformed that a 423 424 fit of the entire curve is not possible if the peak broadening due to the instrument itself is not considered. 425 Qualitatively, there is no obvious difference between the peak shape and position in the pure and the mixed 426 gas permeation measurements, with exception of the higher noise for the mixtures due to the lower signal. 427 There is a weak increase in the peak position with increasing pressure, which is best visible for the PIM that 428 has the shortest time scale, but this is probably mostly due to the effect discussed in Figure 6. In all curves, the peak height increases more or less proportionally with the pressure, because upon substitution of  $t = \tau_{Max}$ 429  $\frac{l^2}{10.00}$ , Eq. A11 becomes independent of time and increases linearly with the feed pressure. 430



**Figure 7.** Permeate flow rate of pure CO<sub>2</sub> (A), pure CH<sub>4</sub> (B) and the same gases using a 35/65 vol% CO<sub>2</sub>/CH<sub>4</sub> mixture (C,D) after a 6 s pulse and a feed flow rate of 500 cm<sup>3</sup><sub>STP</sub> min<sup>-1</sup> through the PDMS membrane (membrane thickness 1056  $\mu$ m, effective area 13.84 cm<sup>2</sup>). The numbers indicate the feed pressure in bar(a). A background of 3 vol% CO<sub>2</sub> and 3 vol% CH<sub>4</sub> in argon is used to guarantee a slightly higher baseline signal and correspondingly higher sampling rate. The dotted grey curves in the graph for pure CO<sub>2</sub> and pure CH<sub>4</sub> show for comparison the corresponding response of a 6 s pulse at 6 bar through the pinhole, scaled vertically to fit in the same graph.





**Figure 8.** Permeate flow rate of pure CO<sub>2</sub> (A), pure CH<sub>4</sub> (B) and the same gases using a 35/65 vol% CO<sub>2</sub>/CH<sub>4</sub> mixture (C,D) after a 6 s pulse and a feed flow rate of 500 cm<sup>3</sup><sub>STP</sub> min<sup>-1</sup> through the PIM-DTFM-BTrip membrane (membrane thickness  $112 \pm 6 \mu m$ , effective area 0.785 cm<sup>2</sup>, age 1382 days + max 5 days to complete the measurement cycle). The numbers indicate the feed pressure in bar(a). A background of 3 vol% CO<sub>2</sub> and 3 vol% CH<sub>4</sub> in argon is used to guarantee a slightly higher baseline signal and correspondingly higher sampling rate. The dotted red curve in the graphs for pure CO<sub>2</sub> and pure CH<sub>4</sub> show for comparison the corresponding response of a 6 s pulse at 6 bar through the pinhole, scaled vertically to fit in the same graph.

The pulse peak position is very similar for the PIM and for the pinhole, but the PIM signal is clearly much wider. This suggests that the estimation of the diffusion coefficient can still be relatively accurate, because the membrane has the largest influence on the overall signal. Since the complete fit of the permeation curve is not possible for the PIM, only the peak maximum was determined by a partial fit of the peak apex, and the maximum due to the membrane transport,  $\tau_{Max,Mem}$  was determined as follows:

 $\tau_{Max,Mem} = \tau_{Max} - \tau_{Max,0}$  Eq. 12

439 Where  $\tau_{Max,0}$  is the peak position of system response, *i.e.* of the pulse on the aluminium sample with pinhole 440 given by Eq. 10 and Eq. 11. The diffusion coefficient is than calculated by rearrangement of Eq. A12 as:

$$D = \frac{l^2}{10.9 \,\tau_{Max,Mem}} \qquad \qquad \text{Eq. 13}$$

441 The results are plotted in Figure 9. There is some scatter in the diffusion data, which does not allow to 442 identify an unambiguous trend as a function of the feed pressure or a difference between pure and mixed gas diffusion coefficients in PDMS. Instead, in the PIM, both CO<sub>2</sub> and CH<sub>4</sub> have a higher diffusion coefficient in 443 the mixed gas experiment than in the pure gas experiment. The standard mixture (Ar with 3 vol.% of  $CO_2$  and 444 445 3 vol.% of CH<sub>4</sub>) that is used as the background has apparently little effect on the diffusion coefficient because 446 of the low CO<sub>2</sub> partial pressure. The average values of the diffusion coefficient taken at all pressures lie 447 remarkably close to the values determined by the other methods (Table 2 and Table 3), confirming the validity 448 of the procedures. For a more sensitive determination of the effect of the gas pressure or composition,  $CO_2$ 

449 could be run at the background when making  $CH_4$  pulses and vice versa, or <sup>13</sup>C labelled  $CO_2$  and  $CH_4$  could 450 be used for the pulse while the unlabelled mixture is permeating.



**Figure 9.** Diffusion coefficients for pure CO<sub>2</sub> and CH<sub>4</sub> (A,C) and for a 35/65 vol.% CO<sub>2</sub>/CH<sub>4</sub> mixture (B,D) in a 1056  $\mu$ m thick PDMS film (top) and a 112  $\mu$ m thick PIM-DTFM-BTrip film (bottom) as a function of the feed gas pressure. Feed gas mixture of 3 vol.% CO<sub>2</sub> and 3 vol.% CH<sub>4</sub> in Argon, followed by a 6 second pulse of the gas or gas mixture of interest. The lines are plotted as a guide to the eye. PIM-DTFM-BTrip membrane age 1382 days (+ max. 5 to complete the measurement cycle).

452

### 453 *3.3.* Comparison of methods

An overview of the results for all different methods is given in Table 2 for PDMS and Table 3 for PIM-DTFM-BTrip, while a selected number of data is also plotted in the Robeson diagram (Figure 10). Despite the different instruments used and despite the different measurement and evaluation modes when using the same instrument, the transport parameters of  $CO_2$  and  $CH_4$  in PDMS are strikingly similar, with a maximum of around 10% spread in both the permeability and the diffusion coefficient. Regardless the method used, the

## mixed gas diffusion coefficient of methane is systematically higher than the single gas diffusion coefficient,

460 but there is no systematic trend for CO<sub>2</sub>.

	$P_a [Barrer = 10^{-10} \text{ cm}^3_{STP} \text{ cm} \text{ cm}^{-2} \text{ s}^{-1} \text{ cmHg}^{-1}]$					Selectivity $\alpha_{\rm P} \left( P_a / P_b \right)$				
	N <sub>2</sub>	O <sub>2</sub>	CO <sub>2</sub>	CH4	H <sub>2</sub>	Не	H <sub>2</sub> /N <sub>2</sub>	CO <sub>2</sub> /N <sub>2</sub>	O <sub>2</sub> /N <sub>2</sub>	CO <sub>2</sub> /CH <sub>4</sub>
Fixed Vol. Tangent	235	501	2628	747	508	273	2.16	11.2	2.14	3.52
Fixed V Complete fit	235	499	2632	747	509	274	2.16	11.2	2.14	3.52
Variable Vol. Tangent			2802	955						2.9
(mixed gas)			(2815)	(894)						(3.1)
Complete fit tangent			3028	894						3.4
(mixed gas)			(2271)	(1051)						(2.2)
Complete fit sigmoidal			3028	894						3.4
(mixed gas)			(2275)	(1051)						(2.2)
	D <sub>a</sub> [10 <sup>-12</sup> m <sup>2</sup> s <sup>-1</sup> ]					Selectivity $\alpha_{\rm D} (D_a/D_b)$				
	N <sub>2</sub>	O <sub>2</sub>	CO <sub>2</sub>	CH <sub>4</sub>	H <sub>2</sub>	Не	$H_2/N_2$	CO <sub>2</sub> /N <sub>2</sub>	O <sub>2</sub> /N <sub>2</sub>	CO <sub>2</sub> /CH <sub>4</sub>
Fixed Vol. Tangent	1461	1777	1441	1274	5050	6889	3.46	0.99	1.22	1.13
Fixed V Complete fit	1469	1784	1423	1260	5107	6954	3.48	0.97	1.21	1.13
Variable Vol. Tangent			1652	1391						1.19
(mixed gas)			(1516)	(1671)						(0.91)
Complete fit tangent			1429	1290						1.11
(mixed gas)			(1396)	(1443)						(0.97)
Complete fit sigmoidal			1432	1293						1.11
(mixed gas)			(1399)	(1428)						(0.98)
Pulse method			$1393 \pm \!\! 31$	$1279\ {\pm}80$						1.08
(mixed gas)			(1422±68)	(1282 ±39)						(1.11)
	$S_a [cm^3 c$	cm <sup>-3</sup> bar <sup>-1</sup>	]				Selectivit	y as $(S_a/S_b)$		
	N <sub>2</sub>	O2	CO <sub>2</sub>	CH4	H2	He	H <sub>2</sub> /N <sub>2</sub>	CO <sub>2</sub> /N <sub>2</sub>	$O_2/N_2$	CO <sub>2</sub> /CH <sub>4</sub>
Fixed Vol. Tangent	0.12	0.21	1.37	0.44	0.08	0.03	0.66	11.4	1.75	3.11
Fixed V Complete fit	0.12	0.21	1.39	0.44	0.07	0.03	0.58	11.6	1.75	3.16
Variable Vol. Tangent			1.27	0.51						2.49
(mixed gas)			(1.39)	(0.40)						(3.47)
Complete fit tangent			1.59	0.52						3.06
(mixed gas)			(1.22)	(0.55)						(2.22)
Complete fit sigmoidal			1.58	0.52						3.04
(mixed gas)			(1.22)	(0.55)						(2.22)

	Age <sup>a)</sup>	Pa [Barr	$er = 10^{-10}$	cm <sup>3</sup> sTP cm cm <sup>-2</sup>	<sup>2</sup> s <sup>-1</sup> cmHg <sup>-1</sup> ]			Selectivit	$x \alpha_P (P_a/P_b)$	)	
	(days)	N <sub>2</sub>	O <sub>2</sub>	CO <sub>2</sub>	CH4	H <sub>2</sub>	Не	H <sub>2</sub> /N <sub>2</sub>	CO <sub>2</sub> /N <sub>2</sub>	O <sub>2</sub> /N <sub>2</sub>	CO <sub>2</sub> /CH <sub>4</sub>
Fixed Vol. Tangent	274	729	3048	14376	719	9654	4483	13.2	19.7	4.18	20.0
Fixed V Complete fit	274	728	3097	14497	722	9648	4531	13.2	19.9	4.28	20.1
Variable Vol. Tangent	1204			18049	1141						15.1
(mixed gas)	1384			(19297)	(1167)						(16.5)
Complete fit tangent	1384			17854	1120						15.9
(mixed gas)	1304			(18673)	(1220)						(15.3)
Complete fit sigmoidal	1384			17940	1115						16.1
(mixed gas)	1504			(18601)	(1214)						(15.3)
	Age <sup>a)</sup>	Da [10-12	$^{2} m^{2} s^{-1}$ ]					Selectivit	ty $α_D (D_a/D)$	b)	
	(days)	N <sub>2</sub>	O <sub>2</sub>	CO <sub>2</sub>	CH <sub>4</sub>	H <sub>2</sub>	Не	$H_2/N_2$	CO <sub>2</sub> /N <sub>2</sub>	O <sub>2</sub> /N <sub>2</sub>	CO <sub>2</sub> /CH <sub>4</sub>
Fixed Vol. Tangent	274	114	474	171	32.1	8635	9465	75.7	1.50	4.16	5.33
Fixed V Complete fit	274	125	488	179	34.2	9078	12210	72.6	1.43	3.90	5.23
Variable Vol. Tangent	1204			92.7	54.5						1.70
(mixed gas)	1384			(170)	(55.0)						(3.09)
Complete fit tangent	1294			94.9	58.6						1.61
(mixed gas)	1364			(193)	(58.4)						(3.30)
Complete fit sigmoidal	1384			91.5	62.2						1.47
(mixed gas)	1304			(193)	(60.2)						(3.20)
Pulse method	1384			$170{\pm}18$	$57\pm9$						2.98
(mixed gas)	1504			$(226\pm 61)$	(66 ±4)						(3.43)
	Age <sup>a)</sup>	S <sub>a</sub> [cm <sup>3</sup> s	TP cm <sup>-3</sup> ba	ar <sup>-1</sup> ]				Selectivit	$xy \alpha s (S_a/S_b)$		
	(days)	N <sub>2</sub>	O <sub>2</sub>	CO <sub>2</sub>	CH4	H <sub>2</sub>	He	H <sub>2</sub> /N <sub>2</sub>	CO <sub>2</sub> /N <sub>2</sub>	O <sub>2</sub> /N <sub>2</sub>	CO <sub>2</sub> /CH <sub>4</sub>
Fixed Vol. Tangent	274	4.80	4.82	63.2	16.8	0.84	0.36	0.17	13.2	1.00	3.76
Fixed V Complete fit	274	4.36	4.75	60.6	15.8	0.79	0.27	0.18	13.9	1.09	3.83
Variable Vol. Tangent	1294			167	18.8						8.88
(mixed gas)	1364			(84.7)	(15.6)						(5.43)
Complete fit tangent	1384			141	14.3						9.86
(mixed gas)	1304			(88.8)	(15.6)						(5.69)
Complete fit sigmoidal	1384			147	13.4						10.9
(mixed gas)	1501			(72.4)	(15.1)						(4.79)

**Table 3.** Transport properties P, D and S with the corresponding selectivities  $P_a/P_b$ ,  $D_a/D_b$ ,  $S_a/S_b$  for PIM-DTFM-BTrip.

<sup>a)</sup> Membrane age after MeOH treatment, plus up to 5 days after the indicated age to complete the measurement cycle.

For PIM-DTFM-Btrip, there is somewhat more spread in the data. In terms of permeability, the variable volume setup gives higher permeabilities than the fixed volume setup for both  $CO_2$  and  $CH_4$ , for all the methods used. However, the variable volume setup gives lower pure gas diffusion coefficients for  $CO_2$  than the fixed volume setup, but very similar mixed gas diffusion coefficients, while the diffusion coefficients are substantially higher in the variable volume setup with all three measurement modes, both for the pure gases

<sup>463</sup> 

and for the mixed gases. The latter demonstrates the presence of a positive coupling effect of CO<sub>2</sub> for CH<sub>4</sub>,

- 470 while the lower pure gas diffusivity of CO<sub>2</sub> indicates a negative coupling by CH<sub>4</sub>.
- 471



**Figure 10.** Robeson plot with an overview of the results for the CO<sub>2</sub>/CH<sub>4</sub> gas pair in PDMS (Table 2) and PIM-DTFM-BTrip (Table 3). Data for PDMS are reported in purple for single gas and orange for mixed gas, while data for PIM-DTFM-BTrip are reported in blue for single gases and green for mixed gas. Symbols shapes indicate the different instrument or method used for their analysis: Fixed Volume Tangent ( $\bullet$ ), Fixed Volume Complete fit ( $\blacksquare$ ), Variable Volume Tangent ( $\blacklozenge$ ), Variable Volume, Complete fit time lag curve ( $\blacklozenge$ ), Variable Volume, Complete fit sigmoidal curve (X). Empty blue symbols indicate change in separation performance as a function of aging time for the PIM-DTFM-BTrip (data from Ref. [16])

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#### 473 *3.4. Gas separation process evaluation*

In real separation processes, the membranes normally operate under steady state conditions, where the most important variables are the feed pressure and the feed gas composition. While the transient measurements proved useful for the analysis of the transport parameters, yielding detailed information on the transport mechanism and its anomalies, Figure 11 shows an overview of the results under stationary conditions with the comparison of pure and mixed gas permeability and selectivity for both polymers as a function of the total feed pressure (A,B and D,E) and feed composition (C,F). PDMS shows negligible pressure dependence in the range of 1-6 bar absolute pressure at 35 vol% of  $CO_2$  in methane, and also very low composition dependence in the range of 10-50 vol% of  $CO_2$  in methane at 6 bar(a) feed pressure. While it was shown in the previous sections that the diffusion of  $CH_4$  is slightly faster in the mixture in both PDMS and the PIM (Figure 4), the  $CH_4$  mixed gas permeability is somewhat lower than the pure gas permeability, and  $CO_2$  is unaffected, and thus the  $CO_2/CH_4$  selectivity results slightly higher than the ideal selectivity. This is apparently due to competitive sorption of  $CO_2$  and  $CH_4$ , which reduces the solubility of the  $CH_4$ .



**Figure 11.** Permeability (in Barrer) of pure CO<sub>2</sub> and CH<sub>4</sub> as a function of the feed pressure in (A) PDMS and (D) PIM-DTFM-BTrip. Corresponding curves (B,E) for mixed gas permeation with a 35/65 vol% CO<sub>2</sub>/CH<sub>4</sub> and (C,F) plot of the composition-dependence at 6 bar(a) feed pressure. The lines serve as a guide to the eye only. The symbols at integer values of pressure and at 10, 20, 30, 40 and 50% CO<sub>2</sub> represent the increasing pressure steps or CO<sub>2</sub> concentration in the feed, while the points at half-integer value of pressure or the points at 15, 25, 35 and 45 % CO<sub>2</sub> represent the subsequent decreasing pressure steps or CO<sub>2</sub> concentration. The arrowheads on the lines point towards the axis where to read the data. Membrane age 274 days (+ max. 5 days to complete the measurement cycle).

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There is no evidence of plasticization by  $CO_2$ , which generally occurs at higher  $CO_2$  partial pressure, and is more evident in glassy polymers than in rubbers. Opposite to PDMS, the sample of PIM-DTFM-BTrip shows a distinct pressure dependence for  $CO_2$  permeation, while the  $CH_4$  permeability is nearly constant as a function of the feed pressure (Figure 11 D,E). As a result, the ideal selectivity and the mixed gas selectivity 492 decreases with increasing pressure, according to a similar trend. At a total pressure of 6 bar(a), instead, the 493 composition-dependence is almost negligible. In all three experiments there is also some hysteresis between 494 the increasing and decreasing pressure (Figure 11D,E) or CO<sub>2</sub> concentration (Figure 11F), with the highest 495 values being recorded in the (partial) pressure decrease run. The hysteresis in the PIM sample suggests that 496 dilation occurs at the highest absolute pressure or  $CO_2$  partial pressure, which does not relax back to the original 497 volume within the duration of a measurement cycle and therefore the permeability increases. Indeed, the effect 498 is strongest for the pure gas permeation measurements, where the CO<sub>2</sub> partial pressure reaches 6 bar(a), while 499 it reaches a maximum of 2.1 bar for the measurements at variable pressure (Figure 11E) and 3.0 bar for the 500 measurements at variable composition (Figure 11F). The pressure-dependence of the permeability in the PIM 501 is in line with the results above and confirms that it is strictly not possible to describe the transport in the PIM 502 with simple Fickian diffusion. Under the same conditions, PIM-DTFM-BTrip shows systematic much higher 503 permeability at different feed pressures with respect to PIM-2 [28] and PIM-SBF-1 [64], and also higher 504 permeability with respect to the ultrapermeable PIM-SBI-Trip [81], even if the latter as a higher selectivity 505 positioning both of them close to the most recent upper bounds. Given its fluorinated nature, it might be 506 expected that the performance of PIM-DTFM-Trip is less affected by the presence of humidity, similarly to 507 what was observed for PIM-2 [28]. This makes this PIM of interest for further studies regarding industrial 508 separations where large quantities of humid gases must be treated.

509

#### 510 **4.** Conclusions

511 The present paper describes the detailed analysis of the transient phase of mixed gas transport through two 512 fundamentally different membrane materials, namely the glassy benzotriptycene-based ultrapermeable 513 polymer of intrinsic microporosity (PIM-DTFM-BTrip) and the rubbery polydimethylsiloxane (PDMS), via 514 the online analysis of the permeate gas composition and flow rate by means of a mass-spectrometric residual 515 gas analyser. This analyser offers the unique advantage that it allows the calculation of the mixed gas diffusion 516 coefficients of all individual gases present in the mixture, thus providing novel insight into the transport in 517 these materials. The use of three different approaches (integral, differential and pulse signal) to determine the 518 mixed gas transport parameters, and the comparison with the 'classical' time lag method in a fixed volume 519 setup, provides further insight into the behaviour of the materials and in the strengths and limitations of the 520 different instruments and elaboration methods. The computational analysis of the entire permeation curve is 521 laborious, but provides the best insight in the gas transport mechanism, and unequivocally reveals non-Fickian 522 behaviour, if present, via the deviation of the experimental results from the theoretical permeation curve. It 523 also confirms the well-known but generally ignored limitations of the traditional time lag measurements, which 524 measures the effective transport parameters, but does not consider anomalies such as non-Fickian diffusion. 525 On the other hand, the time lag and other singular points provide the simplest and fastest methods for 526 determining the effective diffusion coefficient of gases in membranes.

527 All three approaches used with the variable volume instrument showed very similar results for PDMS, with 528 a maximum of around 10 % spread in both the permeability and the diffusion coefficient compared to the 529 'standard' fixed volume method. The generally good fit of the permeation curves confirms both the reliability 530 of the methods, and the 'normal' behaviour of PDMS, ascribed to its thermodynamic equilibrium state in the 531 rubber phase, which is not subject to physical aging. The PIM shows strongly non-Fickian transport, unlike 532 that usually observed in rubbery or common glassy polymer membranes, with a very poor fit of the 533 experimental data with the theoretical permeation curves for simple Fickian diffusion and Henry type sorption, 534 indicating the presence of strongly pressure-dependent permeability and diffusion coefficients. PIM-DTFM-BTrip further shows an evident coupling effect between CO<sub>2</sub> and CH<sub>4</sub> in mixed gas permeation experiments, 535 536 with a strong increase of the diffusion coefficient of  $CO_2$  in the presence of  $CH_4$  but a slightly weaker decrease 537 of its solubility. Interestingly, this PIM seems to have stopped aging after the 636 days reported in our previous study [16] and was not affected by the measurements with gas mixtures at elevated CO<sub>2</sub> partial pressures up 538 539 to 6 bar(a).

All methods confirm to be suitable for the analysis of the diffusion coefficient of mixed gases. For future work, the addition of an automatic gas switch after the mass flow controllers, could increase the reaction rate of the system, and thus the accuracy of the results, by the generation of an instantaneous pulse or step change in the feed gas. In order to fit the permeation curve correctly, especially for samples with a very short transient phase, a correction for the signal broadening due to the instrument would be needed in addition to the correction for the total response time.

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#### 547 5. Appendix A: Mathematical models describing different permeation modes

Numerous articles, books and book chapters have been dedicated to the mathematical description of the gas transport through membranes, e.g. [14,82,83]. The models have an analytical solution for flat sheet membranes, provided that a number of conditions are satisfied, such as a constant solubility and diffusivity of the gas in the membrane material. The amount of gas permeating through the membrane can be described by a Tailor series and the precise equation depends on whether we look at the permeation rate every moment or at the cumulative amount of gas permeated. The sections below will describe the models for the instruments and the specific procedures that we will use in this work.

#### 555 5.1. Fixed volume setup

A typical time lag curve of the permeate pressure or total gas volume in a fixed volume is shown in Figure A1A. Under the boundary conditions of concentration-independent solubility and diffusion coefficient, constant feed pressure, the absence of any gas dissolved inside the membrane before the experiment, and negligible permeate pressure compared to the feed, the trend of the total amount of gas in the permeate has an analytical solution. For the fixed-volume setup used in this work, the pressure  $p_t$  as a function of time *t* through a membrane with thickness *l* can be described as [63,84]:

Eq. A1

$$\cdot \left(\frac{D \cdot t}{l^2} - \frac{1}{6} - \frac{2}{\pi^2} \sum_{n=1}^{\infty} \frac{(-1)^n}{n^2} \exp\left(-\frac{D \cdot n^2 \cdot \pi^2 \cdot t}{l^2}\right)\right)$$

where  $p_0$  and  $(dp/dt)_0$  are the starting pressure and the baseline slope, respectively, which should be negligible in a well-evacuated and leak free membrane and permeability instrument. *R* is the universal gas constant [8.314·10<sup>-5</sup> m<sup>3</sup> bar mol<sup>-1</sup>·K<sup>-1</sup>], *T* is the absolute temperature [K], *A* is the exposed membrane area [m<sup>2</sup>], *V<sub>P</sub>* is the permeate volume [m<sup>3</sup>], *V<sub>m</sub>* the molar volume of a gas at standard temperature and pressure [22.41·10<sup>-3</sup> m<sup>3</sup><sub>STP</sub> mol<sup>-1</sup> at 0°C and 1 atm], *p<sub>f</sub>* the feed pressure [bar], *S* the gas solubility [m<sup>3</sup><sub>STP</sub> m<sup>-3</sup> bar<sup>-1</sup>] and *D* the diffusion coefficient [m<sup>2</sup> s<sup>-1</sup>]. Converting the pressure in the fixed volume permeate side to a permeate volume at standard temperature and pressure, *V*<sub>STP</sub>, the following terms must be substituted in Eq. A1:

$$p_t = V_{t,STP} \frac{p_{STP} \times T}{V_P T_{STP}}$$
 Eq. A2

$$p_0 = V_{0,STP} \frac{p_{STP} \times T}{V_P T_{STP}}$$
 Eq. A3

$$\left(\frac{dp_t}{dt}\right)_0 = \left(\frac{dV_t}{dt}\right)_{0,STP} \times \frac{p_{STP} \times T}{V_P T_{STP}}$$
 Eq. A4

569 Thus,  $V_{\text{STP}}$  at any time becomes:

$$V_{t,STP} = V_{0,STP} + \left(\frac{dV_t}{dt}\right)_{0,STP} \cdot t + \frac{RT_{STP}}{p_{STP}V_m} \cdot A \cdot l \cdot p_f \cdot S$$
$$\cdot \left(\frac{D \cdot t}{l^2} - \frac{1}{6} - \frac{2}{\pi^2} \sum_{n=1}^{\infty} \frac{(-1)^n}{n^2} exp\left(-\frac{D \cdot n^2 \cdot \pi^2 \cdot t}{l^2}\right)\right) \quad \text{Eq. A5}$$

570 Where  $\frac{RT_{STP}}{p_{STP}V_m} = 1$  and can be ignored.

#### 571 5.1.1. Tangent time lag method

572 The tangent method is the most commonly used method to calculate the diffusion coefficient from the steady 573 state gas permeation curve. For long times, the pressure-increase rate or the volumetric flow rate become 574 constant, and Eq. A1 and Eq. A5 reduce to:

$$p_t = \frac{RT}{V_p V_m} \frac{A \cdot p_f \cdot S \cdot D}{l} \left( t - \frac{l^2}{6D} \right)$$
 Eq. A6

$$V_{t,STP} = \frac{A \cdot p_f \cdot S \cdot D}{l} \left( t - \frac{l^2}{6D} \right)$$
 Eq. A7

which both describe a straight line (shown for  $p_t$  in Figure A1), that intersects the horizontal axis at the time defined as the time lag  $\Theta$  or  $\tau_L$ :

$$t = \frac{l^2}{6D} \equiv \Theta \text{ or } \tau_{TL}$$
 Eq. A8

577 Measurement of the time lag allows for the calculation of the experimental diffusion coefficient via Eq. A8, if

578 the membrane thickness is known.

### 579 5.1.2. Integral fit of the pressure-increase curve (fixed volume setup)

580 The tangent method requires that the permeation reaches a pseudo-steady state, from where the tangent can be extrapolated to the time axis. If the steady state is not reached, for instance in relatively thick membranes 581 582 or for permeants with a low diffusion coefficient and high solubility, for which the measurement time is 583 extremely long, then a least-squares fit of the entire permeation curve may offer a solution [72]. In this case, 584 Eq. A1 must be expanded in an appropriate number of terms that fits the experimental points and yields the values of P, D and S directly. If the curve shape deviates from the experimental points, this is an indication of 585 586 anomalous transport, such as (partial) immobilization [33] or clustering of the permeating species [84], or 587 plasticization of the polymer.

#### 588 *5.2. Cross flow permeation cell (variable volume setup)*

The most common method to measure the permeation of gas mixtures uses a variable volume setup with a cross-flow permeation cell, and measures the volumetric permeate flow rate. Usually, the steady state flow rate  $J_{\infty}$  is determined directly by a flow meter, or indirectly via the concentration of the permeating gas in a sweeping gas stream with known flow rate. The steady state flow rate is given by the equation:

$$J_{\infty} = \frac{A \cdot (p_f - p_P) \cdot S \cdot D}{l}$$
 Eq. A9

If the partial pressure of the gas in the permeate is negligible (i.e.  $p_P \ll p_f$ ), this corresponds to the slope of the time lag curve under steady state conditions (Eq. A7). Under the same boundary conditions (no gas present in the membrane before the experiment, permeate concentration negligible compared to the feed concentration), the transient gas flow rate from the moment when the membrane is first exposed to the gas is mathematically described by the derivative of the time lag curve in Eq. A5. It takes the form of a sigmoidal curve, described by the following equation:

$$\frac{dV_{t,STP}}{dt} = \left(\frac{dV_t}{dt}\right)_{0,STP} + A \cdot p_f \cdot S \cdot \frac{D}{l}$$
$$\cdot \left(1 + 2\sum_{n=1}^{\infty} (-1)^n \exp\left(-\frac{D \cdot n^2 \cdot \pi^2 \cdot t}{l^2}\right)\right)$$
Eq. A10

599 Under normal conditions and in the absence of leaks in the system and in the membrane, the term  $\left(\frac{dv_t}{dt}\right)_{0,STP}$ 600 is negligible. Most cross-flow setups typically use periodic analysis of the gas composition by GC or micro 601 GC, but with a sufficiently quick analyser one could also monitor the permeate flow rate continuously. It was 602 previously reported that an online mass-spectrometric residual gas analyser can evaluate multiple gases at the 603 same time, and integration of the signal allows the calculation of the mixed gas time lag, and thus mixed gas 604 diffusion coefficients [60,63].

605 If the membrane is exposed to a short pulse of a gas in the feed (ideally a delta function with area = 1 and 606  $t \ll 1000$  t to a step-change, the flow rate of this gas in the permeate is described by the derivative 607 of Eq. A10:

$$\left(\frac{dV_{t,STP}}{dt}\right)' = \left(\frac{dV_t}{dt}\right)'_{0,STP} - \frac{2D^2 \cdot \pi^2}{l^3} \cdot A \cdot p_f \cdot S$$

$$\cdot \left(\sum_{n=1}^{\infty} n^2 \cdot (-1)^n \exp\left(-\frac{D \cdot n^2 \cdot \pi^2 \cdot t}{l^2}\right)\right)$$
Eq. A11

This kind of experiment can be performed by flushing both sides of the cross-flow cell with a sweeping gas, briefly replacing it at the feed side with the gas or gas mixture of interest, and then following the flow rate of the gases in the permeate.

A plot of the signals according to Eq. A5, Eq. A10 and Eq. A11 is shown in Figure A1. The time lag in the integral curve, the inflection point in the differential curve and the peak maximum in the curve of the pulse, are characteristic times that depend on the diffusion coefficient and the membrane thickness. The analysis of these curves thus allows the calculation of D and S.

The inflection point in the differential curve corresponds to the peak maximum of the pulse and they are given by [68]:

$$\tau_{INF} = \tau_{Max} = \frac{l^2}{10.9D}$$
 Eq. A12

617 Another characteristic time is the time needed in the differential method to reach half of the steady state 618 permeate flow rate [68]:

$$\tau_{1/2} = \frac{l^2}{7.2D}$$
 Eq. A13

Both times, also defined as 'singular points', can be used to calculate di diffusion coefficient if the membrane thickness is known. The area under the pulse corresponds to the steady state flow rate in the differential curve and should be proportional to the permeability. Thus, upon appropriate calibration, integration of the pulse signal should allow the calculation of the permeability.





Figure A1. Plot of the permeate curves defined by (A) Eq. A5, (B) Eq. A10 and (C) Eq. A11 for a hypothetical membrane (area 10 cm<sup>2</sup>, thickness 1000 µm) with a permeability of 3000 Barrer, a diffusion coefficient of  $10^{-9}$  m<sup>2</sup> s<sup>-1</sup> and a solubility of 2 cm<sup>3</sup><sub>STP</sub> cm<sup>-3</sup> bar<sup>-1</sup> at a feed pressure of 1 bar.

#### 624 5.3. Singular points method

625

In general, any of the singular points can be used to determine the diffusion coefficient of the penetrant:

$$D = \frac{l^2}{10.9 \tau_{Max}} = \frac{l^2}{7.2 \tau_{1/2}} = \frac{l^2}{6 \tau_L}$$
 Eq. A14

626

Moreover, via the precise shape of the curves in Figure A1, defined by Eq. A5, Eq. A10 and Eq. A11, each 627 of these singular points is directly related to the flow rate through the membrane, and thus to its permeability:

$$J_{\infty} = \frac{J(\tau_{Max})}{0.2442} = \frac{J(\tau_{1/2})}{0.5} = \frac{J(\tau_L)}{0.6266}$$
 Eq. A15

628 If the flow  $J_{\infty}$  is known and normalized, then the singular points can be easily determined via these relations and vice versa, measurement of the singular points allows calculation of  $J_{\infty}$ . In the case of the pulse version, 629 assuming the area of the peak S = 1, the flow rate can also be correlated with its height, h: 630

$$h = 0.5922 J_{\infty} = 0.5922 \frac{\pi^2 D}{l^2}$$
 Eq. A16

and for the peak width at half height,  $\Delta_{(h/2)}$ :

$$\Delta_{(h/2)} = 0.14025 \frac{l^2}{\pi^2 D}$$
 Eq. A17

632 which allows simple determination of *D*. The singular points can be used as a criterion for the homogeneity 633 of the diffusion medium. In a homogeneous membrane, the diffusion coefficients calculated by any of formulas 634 in Eq. 5 will be the same, but in the presence of "facilitated" diffusion paths, the *D* values will decrease in the 635 order of time (see Eq. 5):  $\tau_{Max} < \tau_{1/2} < \theta = \tau_{TL}$ . Thus, the singular points provide the simplest and fastest method 636 for determining the diffusion coefficient of gases in membranes as a first approximation.

637

#### 638 Symbols and abbreviations

Symbol	Parameter (unit)
A	Membrane area (cm <sup>2</sup> )
С	Concentration (cm <sup>3</sup> <sub>STP</sub> cm <sup>-3</sup> )
D	Diffusion coefficient (cm <sup>2</sup> s <sup>-1</sup> ) or (m <sup>2</sup> s <sup>-1</sup> )
h	Peak height (cm <sup>3</sup> <sub>STP</sub> s <sup>-1</sup> )
J	Flow rate (cm <sup>3</sup> <sub>STP</sub> s <sup>-1</sup> )
l	Thickness (µm) or (cm) or (m)
р	Pressure (bar)
Р	Permeability (Barrer = $10^{-10}$ cm <sup>3</sup> <sub>STP</sub> cm cm <sup>-2</sup> s <sup>-1</sup>
	cmHg <sup>-1</sup> )
R	Universal gas constant (8.314·10 <sup>-5</sup> m <sup>3</sup> bar
	mol <sup>-1</sup> ·K <sup>-1</sup> )
S	Solubility ( $cm^{3}_{STP} cm^{-3} bar^{-1}$ ) or ( $m^{3}_{STP} m^{-3} bar^{-1}$ )
t	Time (s or min)
tpulse	Pulse duration (s)
Т	Absolute temperature (K)
V	Volume (cm <sup>3</sup> ) or (m <sup>3</sup> )
$V_{\rm m}$	Molar volume (22.41 $\cdot$ 10 <sup>-3</sup> m <sup>3</sup> <sub>STP</sub> mol <sup>-1</sup> at 0°C and
	1 atm)
x	Coordinate (m) or (cm)
Greek symbol	
α	Selectivity (-)
Δ	Peak width (s)
Θ	Time lag (s)
${\Phi}$	Flow rate (cm <sup>3</sup> <sub>STP</sub> s <sup>-1</sup> )
$ au_{TL}$	Time lag (s)
Subscript,	
index	
а	Gas species a
b	Gas species b
D	Diffusion
f, F	feed

GC	gas chromatography
h/2	at half height
i	Gas species i
INF	Inflection point
m	molar
Max	Maximum
Mem	Membrane
Р	Permeate, permeability
peak	at peak maximum
pulse	pulse (for pulse duration)
S	Solubility
STP	standard temperature and pressure (here 0°C and
	1 atm)
t	at time t
TL	at time lag
V	Volumetric
0	at reference time $t=0$ or pressure $p=0$
1/2	at half height
œ	at infinite
Abbreviation	
AMU	Atomic mass unit
GC	Gas chromatography
GRG	Generalized Reduced Gradient
MFC	Mass flow controller
MS	Mass spectrometry, mass spectrometric
NMR	Nuclear magnetic resonance
PDMS	Polydimethylsiloxane
PIM	Polymer of intrinsic microporosity
PIM-DTFM-	PIM with ditrifluoromethyl benzotriptycene side
BTrip	groups (the specific PIM in this work)
PTMSP	poly(trimethylsilylpropyne)

#### 640 **Declaration of competing interest**

Residual gas analyzer

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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646 Metal Organic frameworks and organic CAges for highly selective gas separation membranes and heavy metal

647 capture devices.

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#### 649 **References**

- R.S. Murali, T. Sankarshana, S. Sridhar, Air Separation by Polymer-based Membrane Technology, Sep.
  Purif. Rev. 42 (2013) 130–186. doi:10.1080/15422119.2012.686000.
- [2] N.W. Ockwig, T.M. Nenoff, Membranes for hydrogen separation, Chem. Rev. 107 (2007) 4078–4110.
   doi:10.1021/cr0501792.
- Air Products, Advanced Prism 
   Membrane Systems For Cost Effective Gas Separations, 1999.

   http://www.airproducts.ru/~/media/Files/PDF/industries/membranes-supply-options-brochure advanced-prism-membrane-systems.pdf.
- 657 [4] G. George, N. Bhoria, S. Alhallaq, A. Abdala, V. Mittal, Polymer membranes for acid gas removal
  658 from natural gas, Sep. Purif. Technol. 158 (2016) 333–356. doi:10.1016/j.seppur.2015.12.033.
- B. Kraftschik, W.J. Koros, J.R. Johnson, O. Karvan, Dense film polyimide membranes for aggressive
  sour gas feed separations, J. Membr. Sci. 428 (2013) 608–619. doi:10.1016/j.memsci.2012.10.025.
- [6] P. Bernardo, E. Drioli, G. Golemme, Membrane Gas Separation: A Review/State of the Art, Ind. Eng.
  Chem. Res. 48 (2009) 4638–4663. doi:10.1021/ie8019032.
- M. Galizia, W.S. Chi, Z.P. Smith, T.C. Merkel, R.W. Baker, B.D. Freeman, 50th Anniversary
  Perspective: Polymers and Mixed Matrix Membranes for Gas and Vapor Separation: A Review and
  Prospective Opportunities, Macromolecules. 50 (2017) 7809–7843.
  doi:10.1021/acs.macromol.7b01718.
- E. Esposito, L. Dellamuzia, U. Moretti, A. Fuoco, L. Giorno, J.C. Jansen, Simultaneous production of
  biomethane and food grade CO2 from biogas: an industrial case study, Energy Environ. Sci. 12 (2019)
  281–289. doi:10.1039/C8EE02897D.
- 670 [9] C.Y. Chuah, K. Goh, Y. Yang, H. Gong, W. Li, H.E. Karahan, M.D. Guiver, R. Wang, T.H. Bae,
  671 Harnessing filler materials for enhancing biogas separation membranes, Chem. Rev. 118 (2018) 8655–
  672 8769. doi:10.1021/acs.chemrev.8b00091.
- [10] R. Khalilpour, K. Mumford, H. Zhai, A. Abbas, G. Stevens, E.S. Rubin, Membrane-based carbon
  capture from flue gas: A review, J. Clean. Prod. 103 (2015). doi:10.1016/j.jclepro.2014.10.050.
- A.M. Arias, M.C. Mussati, P.L. Mores, N.J. Scenna, J.A. Caballero, S.F. Mussati, Optimization of
  multi-stage membrane systems for CO2 capture from flue gas, Int. J. Greenh. Gas Control. 53 (2016)
  371–390. doi:10.1016/J.IJGGC.2016.08.005.
- 678 [12] S. E. Kentish, 110th Anniversary: Process Developments in Carbon Dioxide Capture Using Membrane
  679 Technology, Ind. & amp; Eng. Chem. Res. 0 (2019) null-null. doi:10.1021/acs.iecr.9b02013.
- 680 [13] J. Deng, Z. Huang, B.J. Sundell, D.J. Harrigan, S.A. Sharber, K. Zhang, R. Guo, M. Galizia, State of

- the art and prospects of chemically and thermally aggressive membrane gas separations: Insights from
  polymer science, Polymer (Guildf). (2021) 123988. doi:10.1016/j.polymer.2021.123988.
- [14] J.G. Wijmans, R.W. Baker, The solution-diffusion model: a review, J. Membr. Sci. 107 (1995) 1–21.
   doi:10.1016/0376-7388(95)00102-I.
- [15] J.G. Wijmans, R.W. Baker, The Solution-Diffusion Model: A Unified Approach to Membrane
  Permeation, 2006. doi:10.1002/047002903X.ch5.
- [16] B. Comesaña-Gándara, J. Chen, C.G. Bezzu, M. Carta, I. Rose, M.-C. Ferrari, E. Esposito, A. Fuoco,
  J.C. Jansen, N.B. McKeown, Redefining the Robeson upper bounds for CO2 /CH4 and CO2 /N2
  separations using a series of ultrapermeable benzotriptycene-based polymers of intrinsic microporosity,
  Energy Environ. Sci. 12 (2019) 2733–2740. doi:10.1039/C9EE01384A.
- [17] H. Yin, Y.Z. Chua, B. Yang, C. Schick, W.J. Harrison, P.M. Budd, M. Böhning, A. Schönhals, First
  Clear-Cut Experimental Evidence of a Glass Transition in a Polymer with Intrinsic Microporosity:
  PIM-1, J. Phys. Chem. Lett. 9 (2018) 2003–2008. doi:10.1021/acs.jpclett.8b00422.
- I. Rose, C.G. Bezzu, M. Carta, B. Comesaña-Gándara, E. Lasseuguette, M.C. Ferrari, P. Bernardo, G.
  Clarizia, A. Fuoco, J.C. Jansen, K.E. Hart, T.P. Liyana-Arachchi, C.M. Colina, N.B. McKeown,
  Polymer ultrapermeability from the inefficient packing of 2D chains, Nat. Mater. 16 (2017) 932–937.
  doi:10.1038/nmat4939.
- M. Longo, M.P. De Santo, E. Esposito, A. Fuoco, M. Monteleone, L. Giorno, B. Comesaña-Gándara,
  J. Chen, C.G. Bezzu, M. Carta, I. Rose, N.B. McKeown, J.C. Jansen, Correlating Gas Permeability and
  Young's Modulus during the Physical Aging of Polymers of Intrinsic Microporosity Using Atomic
  Force Microscopy, Ind. Eng. Chem. Res. 59 (2020) 5381–5391. doi:10.1021/acs.iecr.9b04881.
- A. Fuoco, C. Rizzuto, E. Tocci, M. Monteleone, E. Esposito, P.M. Budd, M. Carta, B. ComesañaGándara, N.B. McKeown, J.C. Jansen, The origin of size-selective gas transport through polymers of
  intrinsic microporosity, J. Mater. Chem. A. 7 (2019) 20121–20126. doi:10.1039/C9TA07159H.
- [21] N.B. McKeown, P.M. Budd, Exploitation of Intrinsic Microporosity in Polymer-Based Materials,
   Macromolecules. 43 (2010) 5163–5176. doi:10.1021/ma1006396.
- Z.-X. Low, P.M. Budd, N.B. McKeown, D.A. Patterson, Gas Permeation Properties, Physical Aging,
  and Its Mitigation in High Free Volume Glassy Polymers, Chem. Rev. 118 (2018) 5871–5911.
  doi:10.1021/acs.chemrev.7b00629.
- 710
   [23]
   L.M. Robeson, The upper bound revisited, J. Membr. Sci. 320 (2008) 390–400.

   711
   doi:10.1016/j.memsci.2008.04.030.
- Y. Wang, X. Ma, B.S. Ghanem, F. Alghunaimi, I. Pinnau, Y. Han, Polymers of intrinsic microporosity
  for energy-intensive membrane-based gas separations, Mater. Today Nano. 3 (2018) 69–95.
  doi:10.1016/J.MTNANO.2018.11.003.

- R. Swaidan, B. Ghanem, I. Pinnau, Fine-Tuned Intrinsically Ultramicroporous Polymers Redefine the
   Permeability/Selectivity Upper Bounds of Membrane-Based Air and Hydrogen Separations, ACS
   Macro Lett. 4 (2015) 947–951. doi:10.1021/acsmacrolett.5b00512.
- M. Yavari, T. Le, H. Lin, Physical aging of glassy perfluoropolymers in thin film composite
  membranes. Part I. Gas transport properties, J. Membr. Sci. 525 (2017) 387–398.
  doi:10.1016/j.memsci.2016.07.002.
- [27] K. Nagai, T. Masuda, T. Nakagawa, B.D. Freeman, I. Pinnau, Poly[1-(trimethylsilyl)-1-propyne] and
  related polymers: synthesis, properties and functions, Prog. Polym. Sci. 26 (2001) 721–798.
  doi:10.1016/S0079-6700(01)00008-9.
- 724 A. Fuoco, B. Satilmis, T. Uyar, M. Monteleone, E. Esposito, C. Muzzi, E. Tocci, M. Longo, M.P. De [28] 725 Santo, M. Lanč, K. Friess, O. Vopička, P. Izák, J.C. Jansen, Comparison of pure and mixed gas 726 permeation of the highly fluorinated polymer of intrinsic microporosity PIM-2 under dry and humid 727 conditions: Experiment and modelling, J. Membr. Sci. 594 (2020)117460. 728 doi:10.1016/j.memsci.2019.117460.
- A.X. Wu, J.A. Drayton, X. Ren, K. Mizrahi Rodriguez, A.F. Grosz, J.-W. Lee, Z.P. Smith, Non equilibrium Lattice Fluid Modeling of Gas Sorption for Fluorinated Poly(ether imide)s,
   Macromolecules. 54 (2021) 6628–6638. doi:10.1021/acs.macromol.1c00950.
- J.C. Jansen, M. Lanč, Gas/Vapor Transport, in: E. Drioli, L. Giorno (Eds.), Encycl. Membr., Springer
  Berlin Heidelberg, Berlin, Heidelberg, 2016: pp. 859–871. doi:10.1007/978-3-662-44324-8\_263.
- 734 [31] S.Y. Markova, T. Gries, V. V. Teplyakov, Poly(4-methyl-1-pentene) as a semicrystalline polymeric 735 J. Sci. 598 for separating membranes, Membr. (2020)117754. matrix gas 736 doi:10.1016/J.MEMSCI.2019.117754.
- T.T. Moore, W.J. Koros, Gas sorption in polymers, molecular sieves, and mixed matrix membranes, J.
  Appl. Polym. Sci. 104 (2007) 4053–4059. doi:https://doi.org/10.1002/app.25653.
- [33] L. Wang, J.-P. Corriou, C. Castel, E. Favre, Transport of Gases in Glassy Polymers under Transient
  Conditions: Limit-Behavior Investigations of Dual-Mode Sorption Theory, Ind. Eng. Chem. Res. 52
  (2013) 1089–1101. doi:10.1021/ie2027102.
- J. Guzmán, L. Garrido, Determination of carbon dioxide transport coefficients in liquids and polymers
  by NMR spectroscopy, J. Phys. Chem. B. 116 (2012) 6050–6058. doi:10.1021/jp302037w.
- [35] L. Garrido, C. García, M. López-González, B. Comesaña-Gándara, Á.E. Lozano, J. Guzmán,
   Determination of Gas Transport Coefficients of Mixed Gases in 6FDA-TMPDA Polyimide by NMR
   Spectroscopy, Macromolecules. 50 (2017) 3590–3597. doi:10.1021/acs.macromol.7b00384.
- [36] M. Minelli, G.C. Sarti, Gas transport in glassy polymers: Prediction of diffusional time lag, Membranes
  (Basel). 8 (2018). doi:10.3390/membranes8010008.

- [37] E. Tocci, D. Hofmann, D. Paul, N. Russo, E. Drioli, A molecular simulation study on gas diffusion in
  a dense poly(ether–ether–ketone) membrane, Polymer (Guildf). 42 (2001) 521–533.
  doi:10.1016/S0032-3861(00)00102-6.
- [38] B. Kruczek, T. Matsuura, Limitations of a constant pressure-type testing system in determination of
  gas transport properties of hydrophilic films, J. Membr. Sci. 177 (2000) 129–142. doi:10.1016/S0376754 7388(00)00454-3.
- [39] K. Friess, J.C. Jansen, O. Vopička, A. Randová, V. Hynek, M. Šípek, L. Bartovská, P. Izák, M. Dingemans, J. Dewulf, H. Van Langenhove, E. Drioli, Comparative study of sorption and permeation techniques for the determination of heptane and toluene transport in polyethylene membranes, J. Membr. Sci. 338 (2009) 161–174. doi:10.1016/j.memsci.2009.04.030.
- [40] H.A. Daynes, The Process of Diffusion through a Rubber Membrane, Proc. R. Soc. A Math. Phys. Eng.
  Sci. 97 (1920) 286–307. doi:10.1098/rspa.1920.0034.
- [41] S.W. Rutherford, D.D. Do, Review of time lag permeation technique as a method for characterisation
  of porous media and membranes, Adsorption. 3 (1997) 283–312. doi:10.1007/BF01653631.
- Y.-S. Lee, J.-H. Shim, J.-Y. Suh, A finite outlet volume correction to the time lag method: The case of
  hydrogen permeation through V-alloy and Pd membranes, J. Membr. Sci. 585 (2019) 253–259.
  doi:10.1016/j.memsci.2019.05.048.
- [43] D. Bai, F. Asempour, B. Kruczek, Can the time-lag method be used for the characterization of liquid
   permeation membranes?, Chem. Eng. Res. Des. 162 (2020) 228–237. doi:10.1016/j.cherd.2020.08.012.
- [44] N. Al-Qasas, J. Thibault, B. Kruczek, A new characterization method of membranes with nonlinear
  sorption isotherm systems based on continuous upstream and downstream time-lag measurements, J.
  Membr. Sci. 542 (2017) 91–101. doi:10.1016/j.memsci.2017.07.039.
- [45] B. KRUCZEK, H. FRISCH, R. CHAPANIAN, Analytical solution for the effective time lag of a membrane in a permeate tube collector in which Knudsen flow regime exists, J. Membr. Sci. 256 (2005)
  57–63. doi:10.1016/j.memsci.2005.02.006.
- [46] S. Lashkari, B. Kruczek, H.L. Frisch, General solution for the time lag of a single-tank receiver in the
  Knudsen flow regime and its implications for the receiver's configuration, J. Membr. Sci. 283 (2006)
  88–101. doi:10.1016/j.memsci.2006.06.015.
- [47] S. Lashkari, B. Kruczek, Effect of resistance to gas accumulation in multi-tank receivers on membrane
  characterization by the time lag method. Analytical approach for optimization of the receiver, J. Membr.
  Sci. 360 (2010) 442–453. doi:10.1016/j.memsci.2010.05.043.
- [48] N. Al-Qasas, J. Thibault, B. Kruczek, The effect of the downstream pressure accumulation on the timelag accuracy for membranes with non-linear isotherms, J. Membr. Sci. 511 (2016) 119–129.
  doi:10.1016/j.memsci.2016.03.047.

- [49] G. Genduso, B.S. Ghanem, I. Pinnau, Experimental Mixed-Gas Permeability, Sorption and Diffusion
   of CO2-CH4 Mixtures in 6FDA-mPDA Polyimide Membrane: Unveiling the Effect of Competitive
   Sorption on Permeability Selectivity, Membranes (Basel). 9 (2019). doi:10.3390/membranes9010010.
- [50] G. Genduso, Y. Wang, B.S. Ghanem, I. Pinnau, Permeation, sorption, and diffusion of CO2-CH4
  mixtures in polymers of intrinsic microporosity: The effect of intrachain rigidity on plasticization
  resistance, J. Membr. Sci. 584 (2019) 100–109. doi:10.1016/j.memsci.2019.05.014.
- [51] G. Genduso, E. Litwiller, X. Ma, S. Zampini, I. Pinnau, Mixed-gas sorption in polymers via a new barometric test system: sorption and diffusion of CO2-CH4 mixtures in polydimethylsiloxane (PDMS),
  J. Membr. Sci. 577 (2019) 195–204. doi:10.1016/J.MEMSCI.2019.01.046.
- [52] E. Ricci, F.M. Benedetti, M.E. Dose, M.G. De Angelis, B.D. Freeman, D.R. Paul, Competitive sorption
  in CO2/CH4 separations: the case of HAB-6FDA polyimide and its TR derivative and a general
  analysis of its impact on the selectivity of glassy polymers at multicomponent conditions, J. Membr.
  Sci. 612 (2020) 118374. doi:https://doi.org/10.1016/j.memsci.2020.118374.
- [53] J. Chen, L.S. Loo, K. Wang, A Novel Time Lag Method to Measure the Permeation of Vapor-Gas
  Mixtures, J. Membr. Sep. Technol. 1 (2012) 94–99. doi:10.6000/1929-6037.2012.01.02.3.
- [54] H. Zhou, K.L. Goh, X. Feng, K. Wang, Measuring the permeabilities of binary gas mixtures with a novel time-lag technique, Can. J. Chem. Eng. n/a (n.d.). doi:https://doi.org/10.1002/cjce.24040.
- 800 E. Ricci, E. Di Maio, M. Degli Esposti, L. Liu, G. Mensitieri, P. Fabbri, S.E. Kentish, M.G. De Angelis, [55] 801 Towards a systematic determination of multicomponent gas separation with membranes: the case of acetates, 802 CO2/CH4 in cellulose J. Membr. Sci. 628 (2021)119226. 803 doi:10.1016/j.memsci.2021.119226.
- [56] D. Hofmann, L. Fritz, J. Ulbrich, D. Paul, Molecular simulation of small molecule diffusion and
  solution in dense amorphous polysiloxanes and polyimides, Comput. Theor. Polym. Sci. 10 (2000)
  419–436.
- [57] T. Schäfer, J. Vital, J.G. Crespo, Coupled pervaporation/mass spectrometry for investigating membrane
   mass transport phenomena, J. Membr. Sci. 241 (2004) 197–205. doi:10.1016/j.memsci.2004.05.014.
- [58] C. Brazinha, A.P. Fonseca, O.M.N.D. Teodoro, J.G. Crespo, On-line and real-time monitoring of
  organophilic pervaporation by mass spectrometry, J. Membr. Sci. 347 (2010) 83–92.
  doi:10.1016/j.memsci.2009.10.009.
- 812 [59] P. Tremblay, M.M. Savard, J. Vermette, R. Paquin, Gas permeability, diffusivity and solubility of 813 nitrogen, helium, methane, carbon dioxide and formaldehyde in dense polymeric membranes using a 814 on-line J. Membr. Sci. 282 (2006)245-256. new permeation apparatus, 815 doi:10.1016/j.memsci.2006.05.030.
- 816 [60] S.C. Fraga, M.A. Azevedo, I.M. Coelhoso, C. Brazinha, J.G. Crespo, Steady-state and transient

- transport studies of gas permeation through dense membranes using on-line mass spectrometry, Sep.
  Purif. Technol. 197 (2018) 18–26. doi:10.1016/j.seppur.2017.12.026.
- [61] S.C. Fraga, L. Trabucho, C. Brazinha, J.G. Crespo, Characterisation and modelling of transient transport through dense membranes using on-line mass spectrometry, J. Membr. Sci. 479 (2015) 213–222. doi:10.1016/j.memsci.2014.12.016.
- K. Pilnacek, J.C. Jansen, P. Bernardo, G. Clarizia, F. Bazzarelli, F. Tasselli, Determination of Mixed
  Gas Permeability of High Free Volume Polymers Using Direct Mass Spectrometric Analysis of the Gas
  Compositions, Procedia Eng. 44 (2012) 1027–1029. doi:10.1016/j.proeng.2012.08.664.
- S.C. Fraga, M. Monteleone, M. Lanč, E. Esposito, A. Fuoco, L. Giorno, K. Pilnáček, K. Friess, M. 825 [63] 826 Carta, N.B. McKeown, P. Izák, Z. Petrusová, J.G. Crespo, C. Brazinha, J.C. Jansen, A novel time lag 827 method for the analysis of mixed gas diffusion in polymeric membranes by on-line mass spectrometry: 828 Method development and validation. J. Membr. Sci. 561 (2018)39-58. 829 doi:10.1016/j.memsci.2018.04.029.
- [64] M. Monteleone, E. Esposito, A. Fuoco, M. Lanč, K. Pilnáček, K. Friess, C. Bezzu, M. Carta, N.
  McKeown, J.C. Jansen, A Novel Time Lag Method for the Analysis of Mixed Gas Diffusion in
  Polymeric Membranes by On-Line Mass Spectrometry: Pressure Dependence of Transport Parameters,
  Membranes (Basel). 8 (2018) 73. doi:10.3390/membranes8030073.
- E. V Evans, C.N. Kenney, Gaseous dispersion in laminar flow through a circular tube, Proc. R. Soc.
  London. Ser. A. Math. Phys. Sci. 284 (1965) 540–550. doi:10.1098/rspa.1965.0079.
- [66] G.I. Taylor, Dispersion of soluble matter in solvent flowing slowly through a tube, Proc. R. Soc.
  London. Ser. A. Math. Phys. Sci. 219 (1953) 186–203. doi:10.1098/rspa.1953.0139.
- [67] I.N. Beckman, M.G. Shalygin, V. V. Teplyakov, Particularities of Membrane Gas Separation Under
  Unsteady State Conditions, in: J. Markoš (Ed.), Mass Transf. Chem. Eng. Process., InTech, Rijeka,
  2011: pp. 205–232. doi:10.5772/24112.
- [68] I.N. Beckman, D.A. Syrtsova, M.G. Shalygin, P. Kandasamy, V.V. Teplyakov, Transmembrane gas
  transfer: Mathematics of diffusion and experimental practice, J. Membr. Sci. 601 (2020) 117737.
  doi:10.1016/J.MEMSCI.2019.117737.
- [69] A. Fuoco, M. Monteleone, E. Esposito, R. Bruno, J. Ferrando-Soria, E. Pardo, D. Armentano, J.C.
  Jansen, Gas Transport in Mixed Matrix Membranes: Two Methods for Time Lag Determination,
  Computation. 8 (2020) 28. doi:10.3390/computation8020028.
- C.R. Mason, L. Maynard-Atem, K.W.J. Heard, B. Satilmis, P.M. Budd, K. Friess, M. Lanč, P. 847 [70] 848 Bernardo, G. Clarizia, J.C. Jansen, Enhancement of CO2 Affinity in a Polymer of Intrinsic 849 Microporosity Amine Modification, Macromolecules. 47 (2014)1021-1029. by doi:10.1021/ma401869p. 850

- [71] B. Satilmis, M. Lanč, A. Fuoco, C. Rizzuto, E. Tocci, P. Bernardo, G. Clarizia, E. Esposito, M.
  Monteleone, M. Dendisová, K. Friess, P.M. Budd, J.C. Jansen, Temperature and pressure dependence
  of gas permeation in amine-modified PIM-1, J. Membr. Sci. 555 (2018) 483–496.
  doi:10.1016/j.memsci.2018.03.039.
- R. Scheichl, M.H. Klopffer, Z. Benjelloun-Dabaghi, B. Flaconnèche, Permeation of gases in polymers:
  Parameter identification and nonlinear regression analysis, J. Membr. Sci. 254 (2005) 275–293.
  doi:10.1016/j.memsci.2005.01.019.
- O. Vopička, K. Friess, V. Hynek, P. Sysel, M. Zgažar, M. Šípek, K. Pilnáček, M. Lanč, J.C. Jansen, 858 [73] 859 C.R. Mason, P.M. Budd, Equilibrium and transient sorption of vapours and gases in the polymer of Sci. 434 860 intrinsic microporosity PIM-1, J. Membr. (2013)148-160. 861 doi:10.1016/j.memsci.2013.01.040.
- [74] O. Vopička, M. Lanč, K. Friess, Phenomenology of vapour sorption in polymers of intrinsic
  microporosity PIM-1 and PIM-EA-TB: envelopment of sorption isotherms, Curr. Opin. Chem. Eng. 35
  (2022) 100786. doi:10.1016/j.coche.2021.100786.
- [75] A. Fuoco, B. Comesaña-Gándara, M. Longo, E. Esposito, M. Monteleone, I. Rose, C.G. Bezzu, M.
  Carta, N.B. McKeown, J.C. Jansen, Temperature Dependence of Gas Permeation and Diffusion in
  Triptycene-Based Ultrapermeable Polymers of Intrinsic Microporosity, ACS Appl. Mater. Interfaces.
  10 (2018) 36475–36482. doi:10.1021/acsami.8b13634.
- 869 [76] S.M. Jordan, W.J. Koros, G.K. Fleming, The effects of CO2 exposure on pure and mixed gas
  870 permeation behavior: comparison of glassy polycarbonate and silicone rubber, J. Membr. Sci. 30 (1987)
  871 191–212. doi:https://doi.org/10.1016/S0376-7388(00)81351-4.
- [77] D.R. Paul, W.J. Koros, Effect of partially immobilizing sorption on permeability and the diffusion time
  lag, J. Polym. Sci. Polym. Phys. Ed. 14 (1976) 675–685. doi:10.1002/pol.1976.180140409.
- 874 [78] S.M. Jordan, W.J. Koros, Permeability of pure and mixed gases in silicone rubber at elevated pressures, J. 28 (1990)795-809. 875 Polym. Sci. Part В Polym. Phys. 876 doi:https://doi.org/10.1002/polb.1990.090280602.
- [79] E. Favre, N. Morliere, D. Roizard, Experimental evidence and implications of an imperfect upstream
  pressure step for the time-lag technique, J. Membr. Sci. 207 (2002) 59–72. doi:10.1016/S03767388(02)00039-X.
- P. Stanovsky, M. Karaszova, Z. Petrusova, M. Monteleone, J.C. Jansen, B. Comesaña-Gándara, N.B. 880 [80] McKeown, P. Izak, Upgrading of raw biogas using membranes based on the ultrapermeable polymer 881 PIM-TMN-Trip, 882 of intrinsic microporosity J. Membr. Sci. 618 (2021)118694. 883 doi:10.1016/j.memsci.2020.118694.
- 884 [81] C.G. Bezzu, A. Fuoco, E. Esposito, M. Monteleone, M. Longo, J.C. Jansen, G.S. Nichol, N.B.

- McKeown, Ultrapermeable Polymers of Intrinsic Microporosity Containing Spirocyclic Units with
  Fused Triptycenes, Adv. Funct. Mater. 31 (2021) 2104474. doi:10.1002/adfm.202104474.
- [82] J. Crank, G.S. Park, Diffusion in Polymers, Academic Press, London & New York, 1968.
   https://books.google.it/books?id=TJkpcAAACAAJ.
- [83] J. Crank, The mathematics of diffusion, 2nd ed., Clarendon Press, Oxford, 1975.
  https://books.google.it/books/about/The\_mathematics\_of\_diffusion.html?id=7QjwAAAAMAAJ&red
  ir\_esc=y.
- [84] J.C. Jansen, K. Friess, E. Drioli, Organic vapour transport in glassy perfluoropolymer membranes: A
  simple semi-quantitative approach to analyze clustering phenomena by time lag measurements, J.
  Membr. Sci. 367 (2011) 141–151. doi:10.1016/j.memsci.2010.10.063.
- 895 [85] L.B. Loeb, The Kinetic Theory of Gases, Dover Publications, Inc., New York, 1961.
- 896 [86] V. Teplyakov, P. Meares, Correlation aspects of the selective gas permeabilities of polymeric materials
  897 and membranes, Gas Sep. Purif. 4 (1990) 66–74. doi:10.1016/0950-4214(90)80030-O.