



### AperTO - Archivio Istituzionale Open Access dell'Università di Torino

### Fire behavior of polyamide 12 nanocomposites containing POSS and CNT

This is the author's manuscript					
Original Citation:					
Availability:					
This version is available http://hdl.handle.net/2318/1602331 since 2016-11-09T17:56:03Z					
Published version:					
DOI:10.1016/j.polymdegradstab.2016.10.005					
Terms of use:					
Open Access					
Anyone can freely access the full text of works made available as "Open Access". Works made available under a Creative Commons license can be used according to the terms and conditions of said license. Use of all other works requires consent of the right holder (author or publisher) if not exempted from copyright protection by the applicable law.					

(Article begins on next page)

This Accepted Author Manuscrip (AAM) is copyrighted and published by Elsevier. It is posted here by agreement between service and the University of Turi Changes resulting from the publiship of cess - such as editing, corrections, structural formatting, and other quality control mechanian so the reflected in this version of the text. The definitive version of the text was subsequently published in Polymer Degradation and Stabilit (24 (2016) 151156)

You may download, copy and otherwise use the AAM for-commercial purposes provided that your license is limited y the following restrictions:

(1) You may use this AAM for nonommercial purposes only under the terms of the BYCNC-ND license.

(2) The integrity of the work and identification of the author, copyright owner, and publisher must be preserved in any copy.

(3) You must attribute this AAM in the following format: Creative Commons NBCS-ND license (http://creativecommons.o/ligcenses/bync-nd/4.0/deed.en)doi: 10.1016/j.polymdegradstab.2016.10.005

# Fire behavior of polyamide 12 nanocomposites taining POSS and CNT

Silvia Gentiluomo, Antonino Domenico Veca Marco Monti, Marta Zaccone Marco Zanetti\*

<sup>1</sup>Dipartimento di Chimica, Centro Interdipartimentale NCSentro Interdipartimentale ICxTUniversità di Torino, Via P. Giuria 7, 10125, Torino, (Italy) <sup>2</sup>CRF S.C.p.A., Strada Torino 50, 10043, Orbassano (TO), (Italy)

<sup>3</sup>Proplast, Consorzio per la promozionella cultura plastica, Strada Savonesa 9, 15 85 Alta Scrivia (AL),(Italy)

### Abstract

Nanocomposites of polyamide 12 with multiwall carbon nanotubes (CNT) and octaisobutyl polyhedral oligomeric silsesquioxanes ROSS) were studied to assess the teardancy properties. The fire ehaviour was investigated with a cone calorimeter using 50k Were at fluxes, by means of the oxygen index and the UL 94 H classification. The fire residue was characterized using FTIR and SEM The best overall performance observed for the composites containing 3.3 wt% of POSS and 4 wt% of CNT. This composite achieved HB in 4 Jlar9 oxygen index of 27 and a 74% reduction in three ak heat release rate. A synergistic effect was exide using POSS and CNT together for the surface of the burning sample.

#### Keywords:

Carbon nanotubes, Polyhedral Oligomeric Silsesquioxanes, Flame retardancy, Cone Calorimeter, Nylon 12

\* Corresponding author. E-mail addressmarco.zanetti@unito.it

#### 1. Introduction

Fire safety is essential to our modern society what days the demand for flame retardants (FRts) polymer is increasing in industry, as the development of fire resistant materials has become a key factor for the pursuit of safet PA 12 is an important materials in the automotive field since is the most available material currently used for tankelfline and air brake tubing. Albf those applications require being flame retarded without sacrificing mechanical properties as well barrier properties. Thermal degradation of PA 12 was firstly extensively studied by V. Mailledis vreet al. using thermal analysis methods and combine CPMS [1]. They observed that lactam and nitriles are the most abundant products. Moreover, giving particular attention to the influence of a commercial halogenated flame retardant (decabromodipleting), they detected the particulation of the flame retardant in condensed phase reac Since thenhalogencompounds have long been used in polyamide usually with a synergisstuch as antimony trioxid[2]. However, the United States government and the European Community expressed concern about the use of FR especially for those that can be harmful or can release toxic compound dfuring such as halogenated FRIs a matter of fact that halogenate R are ubiquitous and persistent in the environment and pose potential health hazards to humans and values of these chemicals, such as tetrathrough heptabrominated diphenyl ethers (PBDEs) and hexabromocyclododecane (HBCD) have been included in the Stockhoton on Persistent Organic Pollutants for elimination[4].

Effective solutions have been found for derivatives containing nitrogen and phosphasmæsriewed by Weil et al [2]. However even phosphous compounds can bæonsidereddangerous for human health being capable of elease phosphoric acid in case of combust(**57**). Metal hydroxides demonstrated to be effective **§**Rowever, the fire properties of polynsefilled with aluminum trihydrate[6] or hydromagnesit(**§**7] are only interesting at high loading/elsthat usually leads to the lowering of the mechanical propertiesexcluding them from automotive application of PA12 Recently, in the effort to avoid the use of halogenated componunch attention was given to the use of nanoparticles such as carbon nanot(**(DbbsT)** [8€10], layered silicate\$11,12], nanoscopic silica particles [13,14] and polyhedral oligomeric silsesquioxane(**POSS**) [8,15,16]. Nanocomposites of PA12 and **ang**clay was firstly prepared by Reichart et **a**[17] via in situ polymerization, but the effect of nanofilteon the fire behavioof PA12 and PA11was firstly observed byLao et el[18] on nanocomposites obtained **tim**-screw extrusiondispersing low concentrations of nanoparticlesamely nanoclays, carbon nanofibers, and nanosilicas. Only nanoclays demonstrated to be effective decreasing the peak of heat release rate (pHRR) of 24%

4

with a load of 5wt%. However, they needed to add 20 wf% onventional intumescent flame retardant to obtain a V0 rating in UL94 test done on PA 11 samblers recently Lecouvet et al. [19] studied the fire behavior of PA12/halloysite nanotubes nanocomposites prepared in a semi industrial scale extruder using a nuraditional one step, wate assisted extrusion process. During mass obss calorimeter experiments performed with a heat flux of 50 kW/meny observed a reduction of 21% of the pHRR in nanocomposites containing 4 wt% of halloysite nanotubes. The mode of action of the halloysite nanotubes as flame retardant additives was next pas a combination of physical mechanisms including barrier, heat sink and fuel dilution effects. The FR nanocomposites developed so far revealed a good efficiency in reducing the heat release rate in the orcedflaming combustion of conecalorimeter however they donft allow the material to be selfextinguishing in other fire test such as UL 94 and With this aim previous studies have focused on the synergistic effects between nanofillers and conventional flame retardant. Zanetti et al. observed syergy between organically modified montmorillonite and conventional vapor phase fire retardants, such as the combination of decabromodiphenyloxide and antimony oxide in PP nanocomposite[20]. Gallo et al. studied the flametardant synergism between phosphorased additives and metal oxides observing improvements in the UL 94 classification of poly(butylene terephthalate)[21] and polyhydroxyalkanoate[22]. Pappalardo etal. demonstrated that the combination of 0.5 wt% of premodified sepiolite with 12 wt% of a commercial intumescent flame retardant showed a clear synergy (M), UL-94 ranking and peak heat release rate of PP composites [23].

Despite the large amount of literature considering the fire behavior of polymonetaining POSS and CNT either alone or as synergic additive with conventional FR, dirating et. alconsiderthe flammability of composites containing at the same time CNT and F[24]SThey prepared thin sheets of MWCNT functionalized with POSS, containing about 25 wt.% of POSS. Those sheets, called bukypapers by the authors, were then impregnavited an unsaturated polyester resin subsequently cured in oven. The amount of MWQNFTOSS wasevaluated to be around 170 wt% via TGA experiments. The flarmetardant performance of the buqtagper composites was evaluated by the micro cone calorimettes, where a significant reduction in HRR by approximately 72% was observed for the MWCNFPOSS buckypaper/resin composite sample. The MWCNT-g-POSS buckypaper exhibited better efficiency in fire retardancy compared to the MWCNT buckypaper. SEM observati of the residue after the combustion revealed the formation of a more compact residue containing Sparticles, suggesting that this structure could effectively act as barrier to limit the diffusion of flammable gases to the surface and slow dowrnthestion and degradation of the resin. No data are reported regardingextiset jush behavior in tests such

5

as UL 94 and OI. Despite these interesting results, the procedure repolyted angrequire a prepreg approach not applicable to the most communication polymers further, the POSS and CNT require to be chemically bonded.

This paper makes a contribution by investigating the synergistic effect betweetiewrall CNT and octaisobutyl polyhedral oligomeric silsesquioxaires fire behavior of polyamide12 composites A comprehensive characterization of fibe havior was performed with cone calorimeter, oxygen index and UL 94 test.

# 2. Materials and methods

An amineterminated lowviscosity polyamide  $12(M_n = 25500 \text{ g mol}^1, M_w = 43800 \text{ g mol}^1, \text{ melt}$ index MVR  $230^{\circ}$ C/2.16kg = 2570 ml/10min), characterized blyigh-flow extrusion grademelting temperature of 178 °C and a tensile modulus of 1500, MeBpecially suitable for high speed extrusion of thin walled tubewas used as polymer matr(RA 12). It was supplied by EMS Grivory, under the trade name@filamid L 16 LM.

The nanocomposites were produced by melt compounding in a Thermo Prism Eurolabrewin extruder the PA 12 with masterbatch of the same polymeillers have been dispersed mixing the PA 12 with a masterbatch containing 100% of CNTs, which was supplied by Nanocyl under the trade name of Plastic PA1502 and a masterbatch containing 100% of POSS, which was supplied by Nova Res.

The CNTs was supplied by Nanocyl under the trade name of Narloc MC7000. The polyhedral oligomeric silses quioxanes contained within the masterbatch was actais obutyl polyhedral oligomeric silses quioxane (POSS), characterized by a closed cage structure with a sobutyl groups as substituents

A first extrusion was carried out by melt mixing the pure polymer and the masterbatch, followed by pellettization, and thera second extrusion was carried out on the obtained material, in order to get an higher homogezation. Table 1 represents the composition of the prepared composites:

Sample	PA 12(wt%)	CNTs (wt%)	POSS(wt%)
PA 12	100		
PA 12-4CNT	96	4	
PA 12-POSS	96.7		3,3
PA 12-2CNT-POSS	94.7	2	3,3
	I	l	

Table 1:Samples name and relative composition

The specimenseeded for their testing weremanufactured, after a drying process at 80°C for 4 hours, throughnijection moulding machines as, the Babyplast 6/10 P and the Sandretto Serie Micro 65t, depeding on the samples size

Oxygen index (OI) testing was performed on the instrument Stenton Redcroft following the standard ASTM D28687, while the horizontal combusting testing was performed following the procedure of UL 94 H, buton reduced dimension specimental speciment

Cone calorimetentests were carried out to evaluate the fire behavior of the produced materials. The tests were performed with a Fire Testing Technology expeript, at a heat flow of 50 kW/mThe reported results are the average of some materials of some sources of the produced materials.

The residue from UL 94 test where studied in ATR and SETIR infrared spectra were obtained with a PerkinElmer Spectrum 100 instrument equipped with a diamond single reflection ATR accessory. In the typical experiment, 32 scans **vecen**ulated in the range of 40000 cm<sup>1</sup> at 4cm<sup>1</sup> resolution. The residue morphology was observed by SEM (Leica Steßeran 410) applying the accelerating voltage during scanning was 20 kV. Samples were mounted on metallic stubs with doublesided conductive tape and ion coated with gold by a sputter coatert (BeaSCD 050) for 60 s under vacuum at a current intensity of 60 mA.

# 3 Results and discussion

# 3.1 Fire behavior under forced flaming conditions

Figure 1 displayresults obtained for the heat restearate of the PA2 compositescontaining 4wt% of CNT, 3wt % of POSS and both CNT and POSS, compared to PA2, with an external heat "ux of 50 kW/m<sup>2</sup> applied in the cone calorimetein table 2 are reported the main parameter measured during the test the total heateleased and the peak of heat release rate are two of the most important fire behavior characteristics, corresponding to total fire loadinendrowth, respectively

Table2: Cone calorimeter results (staard deviations in parentheses)

	pkHRR	THR	Residual	TTI	CO	CO <sub>2</sub>	TSR	FPI	FIGRA
	(kW/m²)	(MJ/m <sup>2</sup> )	weight	(s)	yield	yield	(m²/m²)	(m²s/kW)	(kW/m²s)
			(%)		(g/kg)	(kg/kg)			
PA 12	1635	1158	0.5	55	20	3.06	766	0,03	12.7
	(28)	(0.9)	(0.6)	(6)	(2)	(0.05)	(38)		(0.5)
PA 12-4CNT	604	101.1	2,8	34	18	2.72	1232	0,05	6.0
	(22)	(0.5)	(0.5)	(9)	(1)	(0.01)	(57)		(0.7)
PA 12-POSS	1521	112.4	0.0	36	16	2.80	610	0.02	11.5
	(100)	(1.3)	(0.0)	(4)	(0)	(0.05)	(35)		(1.0)
PA 12-4CNT-POSS	425	94.9	5.1	21	19	2.63	1536	0.05	4.7
	(10)	(0.2)	(0.3)	(1)	(1)	(0.02)	(69)		(0.2)

PA 12 shows aheat release rateHRR) curve which is typical of many thermoplastic materials, where the heateleaseshows a sudden acceleration that causes a peak in the heat release rate. Subsequently, there is a fast decrease in the HRR, as a result of a gradual reduction of material amount in the sample hold During combustion PA 12 melts and leaves no residue. Adding wt% of POSS the fire behavior of the 12 did not change, with the exception of the time to ignition that is decreased to 36 Octaisobuty POSS showed evaporation or sublimation in inert atmosphere, over a range 250 °C to 300 °C [25]. This means that probably in the cone calorimeter experiment the POSS will evaporate earlier than the thermal degradations portuble ct polymer, increasing the ignitabilit to significant change in tal heatevolved (THR) was found. The presence of 4% at of CNT changed the fire behavior at the theat rease rate curve typical for PA 12was changed intofatteredcurve shape imilar to those observed in averedsilicate polymer nanocomposite 26,27]. As observed before by other authorse fire behavior oCNT composites was influenced by the consolidation of an interconnected network structure gviscosity effects that hinder the release of pyrolysis produces]. Indeed, during the experiment was not observed an apparent meltingith bubbling on the sample surface. At the end of the cone experimentathe 12-4CNT leaves a black/grey solid residue evidencing that not all the sample has ubreed (Fig. 2 A). However, he residual weight for PA 12-4CNT is lower than the amount of CNTs contained in the composite indicating that almost all the polymer has been pyrolyzed and even a part of CNTs may be burned PA 12-4 CNT showed a time to ignition of 34s, which is lower than that of PA 12 Carbon nanoparticles are known to crucially change the thermal conductivity as well as the heat absorption and may thus have a significant influence on the TTI through these physica[29] ects An increase in thermal conductivity is proposed as a factor increasing the thermal inertia and thus delaying TTI, whereas a concentration of heat absorption in the top layer, in which the pyrolysis

and thus mass loss occurs, is believed into ance early ignition and it has been shown that for CNT a clear change in absorption can become the main parameter controlling ignation.

The simultaneous presence ROSS and CNT in the composites leads to a further diminishing in the HRR curve, with a peak of heat released of 425 kWamd a longer time to stop flaming. As in the case to the sample containing the sole CNT there is no apparent melting and bubbling bu residue left by PA 124CNT-POSS has a different aspect as showed in figuBe It seems more compact and presents a smoother surface with some dark grey area and other biteost w Previous studies on the thermal degradation of different POSSedhtbat during heating they can be subjected to two competing processes: evaporation/sublimation and the oxidation of organic substituents which leads to the formation of a silica resi25, 234]. As observed in the sample PA 12-POSS octaisobutyl POSS alone fails to produce a silica residue when exposed at here at: r it undergoes volatilizationHowever, in thePA 12-4CNT-POSS, it seems that CNTs favor the formation of a silica residue. This event can explain the increase in the residualithasespect to the compositePA 12-4CNT. This new barrier composed of silica and CNTs resulted more effective in the reducing the pkHRR. However, considering the time to ignition 4CMT-POSS shows a major ignitability with the lowest TTI recorded in this set of experiments: 21 s. The increase heat absorption due to CNT affected not only the pyrolysis of the polymer, but even the earlier evaporation/sublimation of octaisobutyl POSS causing it earlier than PAtherPOSS sample.

Considering exhaust emissions, there are no significant **ethtées** in theCO yield during combustion. Instead, as for QOt can be noticed that for all the composites the<sub>2</sub> Qield is slightly lower with respect to A 12 The total smoke releast (TSR) of PA 12-POSS is similar to that of PA 12, while in the case of composites containing CNThe TSR higher, indicating that the presence of CNT affects the production

A fire risk evaluation can be made by taking into consideration two parameters as the Fire Performance Index (FPI) and the Fire Growth RateRA) (table 2). The FPI is defined as the ratio TTI/pkHRR and it is a parameter that balances fire risks due to ignition and fire **(ge5)**/vth An increase in the FPI parameter indicates an increase in the time to flashover, which corresponds to the moment when the fire becomes uncontrollable, allowing more time for firefigRt/ng2 showed a FPI of 0.03<sup>2</sup>/s/kW. Adding the sole POSS to the polymer leads to the higher fire risk lowering the FPI to 0.02n<sup>2</sup>/s/kW. PA 12-4CNT andPA 12-4CNT-POSS have the higher FPI ( 0.05 m<sup>2</sup>/s/kW). This indicates that even with an increased ignitability due to a lower TTI, the fire risks are outbalanced by the reduction of HRR.

The FIGRA is defined as the growth rate of the burning intensity, HRR, during a test, and is calculated as the maximum value of the function (heat release rate)/ (elapsed te[\$16]jmTe))e lower is the FIGRA value the lower is the rate of fire growth. It can be noticed in 2atblet the compositePA 124CNT presents a FIRSA value which is lower than the A 12fs one, while the compositePA 124CNT-POSS shows the lowest FIGRA value among the analyzed materials. Again, the addition of the sole POSS did not lead to any improvement in the FIGRA as compared with thePA 12

### 3.1 Flammability

The investigation of flammability was performed by OI and UL 94 teadth tests generally are performed without an external heatux, since the ame ignition source is removed after some secondsThe OI are reported in table

Table 3:results of flammability tests

Sample	OI (O <sub>2</sub> ,Vol%)	UL94 H	Flaming Dripping
PA 12	23.0	-	Yes
PA 12-4CNT	22.0	-	No
PA 12-POSS	21.0	-	Yes
PA 12-2CNT-POSS	27.0	-	No
PA 12-4CNT-POSS	27.5	HB	No

The OI is the minimum concentration of oxygen expressed as volume percent, in a mixture of oxygen andnitrogen that will just support flaming combustion of a material initially at room temperature under the condition reported in the ASTM D **28763** tandardOI values above 27 normally indicate materials that are **self** tinguishing in accordance with the V classification in UL 94. The OI of PA 12 is 23,0 while the composite PA 12 4CNT shows a higher flamma bility with an OI of 22. In the case of PA 12 large amouts of melt escaped the pyrolysis zone by dripping o whereas the melt was stabilized by the interconnected network structure **CMT** hence asing the intensity of burning This behavior has been seen before in nanocomposite particular for PA 6 and CNT nanocomposite **S**. Moreover, a charred residue is formed.

Even in the case dPA 12-POSSit has been detected OI lower than that of the A 12 alone, being 21. During the combustion of a black layer the pyrolysis zone of the sample can be observed, however, this layer escapes together with the molten polymer Surprisingly, using at the same time POSS and CINETOI of PA 12 is increased to 27,0% FOA 12-2CNT-POSSand 27,5% for PA 12-4CNT-POSS During the combustion, the samples burned leaving a solid blackesidue with a specimen like shape

In the horizontal burning test, UL944, the HB classification is reached when the burning of a specimen, clamped horizontally, stops fore 100 mmPA 12, PA 12-4CNT and PA 12-POSS resulted not classifiable? A 12 resulted as nonclassifiable, because the flame arrived to the sample end, producing a remarkable amount of flaming dripping material being not classifiable, PA 12-4CNT showed a totally different behavior compared to the unfilled PA 12 etewasno dripping and theflame transit induced the formation of a self upporting residue, whose shape and size correspond to those of the specimen, as can be observe flight upport.

The fire behavior of PA 12-POSS is similar to those observed in OI test he sample produced thin carbon layenot sufficient to hinder the polymer dripping d leave do residue

On the other handhe composites containing both POSS and CRA (2-4CNT-POSS) cease to burn before the 100mm mark, reaching the HB classificalized, the flame has extinguished at 27 mm from the sample fis, with an average time of burning **d**D8 seconds The flame front transit induced the formation of a selfsupporting residue, whose shape and size correspondent to the of the specimen. As can be seenfigure 3 the residue appear to be composed of a mix of charred material and inorganic compound showing white/grey zones. This residuce has characterized in FTIR-ATR and the spectrum is reported in figure Being mainly composed of carbon the spectra do not possess absorption peak at conduct of a characteristic absorption peak at conduct typical of SiO-Si stretching ofsilica, belonging from POSS thermal decomposition specimen was then burned over a Bunsen flame in order to eliminate completely the carbonaceous phase. The ATR spectra of the remaining white residue correspond to those of amorphous silica and des repo in figure 4. The residue from UL 94 was also observed by SEM and the picture of a portion of the specimen corresponding to a surface crack is reported in figure 5. As can be seen the surface is covered by micro particles of silica, while the bulk love tspecimeris composed by a multicellular structure. The walls of these cells appear to be rather smooth and containing micro particles similar to those observed on the surface but in a less amount.

PA 12-2CNT-POSS, even if its value is very similard the composite A 12-4CNT-POSS s did not reach the HB classification showing a slightly different burbiet pavior. Indeed, during the test the sample withstands a heavy deformation, losing its original size and as the specimented.

These data indicate that both POSS and CNT used alone are not sufficient to achieve a flame retardant effect, even worsening the fire performance litest. However, the simultaneous use of the two nanofillers shows the possibility of obtaining a-eetinguishing composite thanks to a synergistic effect between the fillers. CNTs are able to produce a network during the combustion but are not able torpduce an effective protective shield. The PORSENE is not able to produce layer of silica to make effectively flame retarded the polymethese condition POSS undergoes to volatilization. On the other handlet action of the POSS supported by the voek of CNT is able to produce an effective shield capable etduce the flammability.

### 4. Conclusions

In this paper, a study on fine haviour of composites based on polyamide 12 containing CNTs, POSS or both of them, was carried out.

In the forced flame condition of the cone calorimeter experimen PAthe2 composites containing both CNT and POSS showed the best performances in term of HRR, FPI and FIGRA. Even shoving the lower TTI, the reduction of fire growth balanced the fired rights the other hand, adding the sole POSS to the polymer worsen the fire properties increasing both the fire risks and the fire growth. However, the addition of POSS to the composite containing CNT improves the fire performance strengthening the CNT effect

The investigation of flammability wasperformed by OI and UL94 H tests.PA 12-CNT and PA 12-POSS showed higher flammability than PA 12 showing lower OI, 22 and 21 respectively. However, the simultaneous presence of both additives in the composite reduced the flammability increasing the OI to 27. The same trend was observed the **ULtest** where all the samples were unclassifiable, with the exception of PA 12-4CNT-POSS that showed a set tinguishing behaviour In this latter case the residue from UL 94 H contains silica, as observed in ATR, accumulated on the surface, as observed in STEM se results indicated a synergistic effect between POSS and CNT during fire experiments since octaisobutyl POSS alongeroisse to evaporate/sublimate then, the formation of silica contribute to produce a shield together with the CNT network acting in a similar manner as observed for layered silicate.

Figure 1. Heat release rate of the PA12 and PA12 composites containing 4 wt% of CNT, 3.3 wt% of POSS and both CNT and POSS, with an external, breatf 50 kW/m<sup>2</sup>.

Figure 2. Residues of PA42CNT (A) and PA124CNT-POSS (B)

Figure 3.Specimens from UL 94 Hest after fire test.

Figure 4.FTIR-ATR spectrum of the residue obtained ine tbL 94 H test of PA12-4CNT-POSS and after oveburning it on a Bunsen flame.

Figure 5. SEM imagef a portion of the specimen from UL 94 H testresponding to a surface crack.

Bibliography

- [1] Mailhos-Lefievre V, Sallet D, Martel B. Thermal degradation of pure and flærtærded polyamides 11 and 12. Polym Degrad Stab 1989;233327
- [2] Weil ED, Levchik S. Current Practice and Recent Commercial Developments in Flame Retardancy of Polyamides. J Fire Sci 2004;226841
- [3] Covaci A, Harrad S, Abdallah MAE, Ali N, Law RJ, Herzke D, et al. Novel brominated flame retardants: A review of the analysis, environmental fate and behaviour. Environ Int 2011;37:53256.
- [4] Stockholm Convention

   (http://chm.pops.int/TheConvention/Overview/TextoftheConvention/tabid/2232/Default.aspx
   )
- [5] Laoutid F, Bonnaud L, Alexandre M, Lop Ezuesta JM, Dubois Rew prospects in flame retardant polymer materials: From fundamentals to nanocomposites. Mater Sci Eng R Reports 2009;63:16225.
- [6] Yen YY, Wang HT, Guo WJ. Synergistic flame retardant effect of metal hydroxide and nanoclay in EVA composites. Polym Degratab 2012;97:863.
- [7] Haurie L, Fernandez AI, Velasco JI, Chimenos JM, Cuesta JML, Espiell F. Synthetic hydromagnesite as flame retardant. Evaluation of the flame behaviour in a polyethylene matrix. Polym Degrad Stab 2006;91:**939**.
- [8] Bourbigot S,Duquesne S, Jama C. Polymer Nanocomposites: How to Reach Low Flammability? Macromol Symp 2006;233:16300.
- [9] Kashiwagi T, Du F, Winey KI, Groth KM, Shields JR, Bellayer SP, et al. Flammability properties of polymer nanocomposites with singleled carbon nanotubes: Effects of nanotube dispersion and ncentration. Polyme2005;46:47€81.
- [10] Kashiwagi T, Grulke E, Hilding J, Groth K, Harris R, Butler K, et al. Thermal and flammability properties of polypropylene / carbon nanotube nanocompositesymerPol 2004;45:422739.
- [11] Gilman JW, Harris RH, Sheilds JR, Kashiwagi T, Morgan AB. A study of the flammability reduction mechanism of polystyre have red silicate nanocomposite: Layered silicate reinforced carbonaceous char. Polym Adv Technol 2006;3772.
- [12] Zanetti M, Kashiwagi T, Falqui L, Camino G. Cone Calorimeter Combustion and Gasification Studies of Polymer Layered Silicate Nanocomposites. Chem Mater 2002;14:88€7.
- [13] Barus S, Zanetti M, Lazzari M, Costa L. Preparation of polymeric dynamiocomposites

based on PEnd nanosilica. Polyme2009;50:2595600.

- [14] Kashiwagi T, Morgan AB, Antonucci JM, VanLandingham MR, Harris RH, Awad WH, et al. Thermal and flammability properties of a silipaly(methylmethacrylate) nanocomposite.
   J Appl Polym Sci 2003;89:20728.
- [15] Liu L, Hu Y, Song L, Nazare S, He S, Hull Rombustion and thermal properties of OctaTMA-POSS/PS composites. J Mater Sci 2007;42:4325
- [16] Song L, He Q, Hu Y, Chen H, Liu L. Study on thermal degradation and combustion behaviors of PC/POSS hybrids. Polym Degrad Stab 2008;933027
- [17] Reichet P, Kressler J, Thomann R, Mulhaupt R, Stoppelmann G. Nanocomposites based on a synthetic layer silicate and polyamit@. Acta Polym 1998;49:1€23.
- [18] Lao SC, Wu C, Moon TJ, Koo JH, Morgan A, Pilato L, et al. flame retardant polyamide11 and 12. J Corpos Mater 2009;43:18 @ 8.
- [19] Lecouvet B, Sclavons M, Bourbigot S, Bailly C. Highly loaded nanocomposite films as fire protective coating for polymeric substrates: ire Sci 2014;32:14564.
- [20] Zanetti M, Camino G, Canavese D, Morgan AB, Lamelas FJ, WilkieF@A.Retardant Halogen..Antimony...Clay Synergism in Polypropylene Layered Silicate Nanocomposites †. Chem Mater 2002;14:1893.
- [21] Gallo E, Schartel B, Braun U, Russo P, Acierno D. Feterdant synergisms between nanometric Fe 2O 3 and aluminum phosphinate in poly(butylene terephthalate). Polym Adv Technol 2011;22:238291.
- [22] Gallo E, Schartel B, Acierno D, Russo P. Flame retardant biocomposites: Synergism between phosphinate and nametric metal oxides. Eur Polym J 2011;47:169001.
- [23] Pappalardo S, Russo P, Acierno D, Rabe S, Schartel B. The synergistic effect of organically modified sepiolite in intumescent flame retardant polypropylene. Eur Polym J 2016, €6:196 207.
- [24] Tang Y,Gou J, Hu Y. Covalent functionalization of carbon nanotubes with polyhedral oligomeric silsequioxane for superhydrophobicity and flame retardancy. Polym Eng Sci 2013;53:102€30.
- [25] Fina A, Tabuani D, Carniato F, Frache A, Boccaleri E, Camino G. Porlyheidgomeric silsesquioxanes (POSS) thermal degradation. Thermochim Acta 2006;4420:36
- [26] Kashiwagi T, Harris RH, Zhang X, Briber RM, Cipriano BH, Raghavan SR, et al. Flame retardant mechanism of polyamid€ctay nanocomposites. Polym2004;45:88€91.
- [27] Yu J, Tonpheng B, Grobner G, Andersson O. Thermal properties and transition studies of multi-wall carbon nanotube/nyle® composites. Carbon N Y 2011;49:4866.

- [28] Schartel B, Pötschke P, Knoll U, Abdebad M. Fire behaviour of polyamiden@/Itiwall carbon nanotube nanocomposites. Eur Polym J 2005;41€7061
- [29] Kashiwagi T, Fagan J, Douglas JF, Yamamoto K, Heckert AN, Leigh SD, et al. Relationship between dispersion metric and properties of PMMA/SWMMTocomposites. Polymer 2007;48:485666.
- [30] Dittrich B, Wartig KA, Hofmann D, Mülhaupt R, Schartel B. Flame retardancy through carbon nanomaterials: Carbon black, multiwall nanotubes, expanded graphite/ayeulti graphene and graphene in polypropylene. Polym Degrad Stab 2013;98504895
- [31] Dittrich B, Wartig KA, Hofmann D, Mülhaupt R, Schartel B. Carbon black, multiwall carbon nanotubes, expanded graphite and functionalized graphene flame retarded polypropylene nanocomposites. Polym Adv Technol 2013;242281.6
- [32] Fina A, AbbenhuisHCL, Tabuani D, Camino G. Metal functionalized POSS as fire retardants in polypropylene. Polym Degrad Stab 2006;91
- [33] Fina A, Bocchini S, Camino G. Catalytic fire retardant nanocomposites. Polym Degrad Stab 2008;93:164755.
- [34] Fina A, Tabuai D, Peijs T, Camino G. POSS grafting on PPgMA by-steep reative blending. Polymer2009;50:21 **6**26.
- [35] Hirschler MM. Heat release from plastic materials. In: Grayson VB and SJ, editor. Heat Release Fires, London: Elsevier; 1992, p.€4722.
- [36] Sundström B, Van Hees P, Thureson P. Results and analysis from fire tests of building products in ISO 9705, the room/corner test 1998.