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1 **Photocatalytic degradation of antibiotic and hydrogen production using**
2 **diatom-templated 3D WO_{3-x}@mesoporous carbon nano hybrid under**
3 **visible light irradiation**

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25 **Abstract**

26 Synthesis of highly efficient 3D photocatalysts offers unique abilities for hydrogen production
27 and chemical conversion to find a solution for energy shortage and environmental pollution
28 issues. However, current strategies for production of ordered nanohybrid photocatalysts usually
29 involve complex procedures and the use of expensive templates, which limit their practical
30 applications. In this work, 3D WO_{3-x} @mesoporous carbon photocatalyst was fabricated
31 through one-pot evaporation-induced self-assembly (EISA) process using *Cyclotella sp.* as
32 natural template. During heat-treatment, the precursor of carbon could partially reduce tungsten
33 oxide under N_2 atmosphere leading to the embedding of WO_{3-x} in conductive mesoporous
34 carbon structure. The diatom templated WO_{3-x} @mesoporous carbon (DT- WO_{3-x} @MC)
35 nanohybrid exhibited high surface area ($195.37 \text{ m}^2 \text{ g}^{-1}$) and narrowed band gap (2.67 eV).
36 Integration of tungsten oxide with mesoporous carbon and formation of oxygen vacancies
37 enhanced the absorption of visible light using DT- WO_{3-x} @MC and limited the recombination
38 of electron-hole pairs. 98.7% of cefazolin (CFZ) degradation efficiency and 85.5% of total
39 organic carbon (TOC) removal efficiency were observed within 90 and 180 min under visible
40 light irradiation, respectively. Scavenger quenching tests and electron spin resonance (ESR)
41 analysis demonstrated that $\text{O}_2^{\bullet-}$ played a main role in photocatalysis. CFZ degradation pathway
42 was proposed via identification of conversion intermediates using GC-MS analysis.
43 Photocatalytic hydrogen production rates of the pure WO_3 and the DT- WO_{3-x} @MC nanohybrid
44 were determined as 746 and $1851 \mu\text{mol g}^{-1} \text{ h}^{-1}$, respectively. This study presented a way to
45 develop a high-performance and stable photocatalyst using diatom frustules as natural template
46 which works under practical conditions for environmental remediation and energy production.
47

48 **Keywords:** Tungsten oxide; Diatom-templated photocatalyst; WO_{3-x} @mesoporous carbon;
49 Photocatalysis; Antibiotic degradation; Hydrogen production.

50 **1. Introduction**

51 Fast industrialization and urbanization has caused a remarkable increase in non-renewable
52 energy consumption depleting energy sources as well as discharging toxic contaminants in
53 environment (Kumar *et al.* 2018; Mir & Pandey 2019; Singh *et al.* 2019; Cha-Umpong *et al.*
54 2020; Kobya *et al.* 2020). To meet the environmental challenges and enhance the exploitation
55 of renewable energy resources, photocatalytic production of hydrogen (H₂) energy has gained
56 great attention (Liu *et al.* 2019). Furthermore, antibiotics residues from aquaculture, hospital
57 effluent, and livestock or humans medicines have been widely detected in water bodies
58 (Teodosiu *et al.* 2018; Patel *et al.* 2019). The misuse of antibiotic medications has resulted in
59 serious environmental pollution and human health hazards (Chen *et al.* 2019; Zhang *et al.*
60 2020a).

61 Various semiconductors such as metal oxides, sulfides, oxysulfide, heterojunctions and
62 layered double hydroxides have been widely investigated for the photocatalytic H₂ evolution
63 and photodegradation of different pollutants under light irradiation (Li *et al.* 2018b; Wang *et al.*
64 2019; Gholami *et al.* 2020b; Hu *et al.* 2020). But due to their large band gaps, they are only
65 excited by ultraviolet (UV) light (wavelengths less than 400 nm), that merely occupies about
66 3–5% in sunlight (Li *et al.* 2018a).

67 Tungsten trioxide (WO₃), an n-type photocatalyst with an energy band gap (E_g) of ~2.8 eV,
68 has gained great attention among the semiconductor photocatalyst materials owing to its
69 nontoxicity, unique physicochemical characteristics, low-cost and high stability (Žerjav *et al.*
70 2017; Cai *et al.* 2019). Nevertheless, pure WO₃ has low efficacy for energy conversion because
71 of insufficient reduction potential and fast recombination of charge carriers (Jamila *et al.* 2020).
72 According to literature, photocatalytic activity of WO₃ can be improved by incorporation of
73 carbon materials (Phang & Tan 2019; Jamila *et al.* 2020). Furthermore, O vacancy in WO_{3-x}

74 serves as shallow donor to increase the electrical conductivity, donor density, and the surface
75 species adsorption (Guo et al. 2019). For instance, Dong et al. have demonstrated that WO_{3-x}
76 $\times/2\text{D}$ g- C_3N_4 photocatalyst benefits from introduction of oxygen vacancy in WO_{3-x} and
77 exfoliation of g- C_3N_4 to show the superior photocatalytic performance for removing antibiotic,
78 organic dye and bacteria from water (Dong et al. 2020).

79 The synthesis of photocatalysts with the best performing morphology and nano-size pore
80 structure faces aggregation/agglomeration problems, which can reduce photocatalytic activity
81 by decreasing surface active sites and light adsorption and encouraging particles settling (Gora
82 & Andrews 2019). To overcome this complication and enhance the restricted characteristics of
83 synthesized nanomaterials, a feasible approach is required to produce the complex three-
84 dimensional (3D) structures with precisely controlled and well-defined morphologies (Lee et
85 al. 2018; Li et al. 2019a). Application of a natural template to fabricate inexpensive and ordered
86 nanostructures with controlled size and morphology is a particularly interesting realm. Single
87 cell diatoms could be considered as an attractive template to synthesize 3D structures (Ragni
88 *et al.* 2018; Li *et al.* 2019a; Sriram *et al.* 2020). One of the special properties of these
89 microorganisms is their 3D silica cell wall (frustule), which can provide ordered and complex
90 pore patterns. These unique 3D silica structures are not feasible to synthesize by existing
91 technologies (Ragni et al. 2018).

92 To the best of authors' knowledge, there has been no previous report on the fabrication of
93 3D WO_{3-x} /mesoporous carbon (WO_{3-x} @MC) nanohybrid using environmentally non-toxic and
94 abundantly available diatom template. The effective removal of the antibiotics is of great
95 significance to the environment due to their non-biodegradability and toxic effects (Chen *et al.*
96 2019; Acharya *et al.* 2020; Zhang *et al.* 2020a). Therefore, the aims of this study were to (a)
97 fabricate diatom templated WO_{3-x} @mesoporous carbon (DT- WO_{3-x} @MC) nanohybrid with
98 3D structure through one-pot evaporation-induced self-assembly (EISA) method, (b)

99 characterize as-synthesized nanohybrid photocatalyst, and (c) evaluate its activity in
100 photocatalytic hydrogen production and degradation of cefazolin (CFZ, a semi-synthetic β -
101 lactam antibiotic derived from *Cephalosporium acremonium*) under visible light irradiation.
102 Based on the results obtained from linear sweep voltammetry (LSV) response, electron spin
103 resonance (ESR) analysis and scavenger quenching tests a possible mechanism was predicted
104 for transportation of photo-generated charge carriers and formation of reactive species during
105 photocatalysis.

106

107 **2. Materials and Methods**

108 **2.1. Reagents and chemicals**

109 *Cyclotella sp.* cells were purchased from the Iranian Biological Resource Center (Iran).
110 Sulfuric acid (H_2SO_4 , $\geq 98\%$), sodium hydroxide (NaOH , $\geq 98\%$), ethanol ($\text{C}_2\text{H}_5\text{OH}$, 99%),
111 ferrous sulfate tetrahydrate ($\text{FeSO}_4 \cdot 4\text{H}_2\text{O}$, 99%), ammonium formate (HCO_2NH_4 , 99.9%),
112 sodium sulphate (Na_2SO_4 , 99.9%), diethyl ether ($\text{C}_4\text{H}_{10}\text{O}$, $\geq 99.7\%$), hydrochloric acid (HCl ,
113 38%), hydrogen peroxide (H_2O_2 , 30%), and N,O-bis-(trimethylsilyl)-acetamide ($\text{C}_8\text{H}_{21}\text{NOSi}_2$,
114 $\geq 95\%$) were obtained from Merck (Germany). Phenol ($\text{C}_6\text{H}_5\text{OH}$, $\geq 99\%$), formaldehyde
115 solution (CH_2O , 37 wt%), tetrahydrofuran (THF, $\text{C}_4\text{H}_8\text{O}$ $\geq 99\%$), benzoquinone ($\text{C}_6\text{H}_4\text{O}_2$, \geq
116 98%), isopropanol ($\text{C}_3\text{H}_8\text{O}$, 99%), acetone ($\text{C}_3\text{H}_6\text{O}$, 99.8%), tungsten chloride (WCl_6 , 99.9%),
117 sodium sulfide (Na_2S , 99%), sodium sulfite (Na_2SO_3 , 99%), acetonitrile ($\text{C}_2\text{H}_3\text{N}$, HPLC grade),
118 ethylenediaminetetraacetic acid disodium salt ($\text{C}_{10}\text{H}_{14}\text{N}_2\text{Na}_2\text{O}_8$, 99%) and disodium hydrogen
119 phosphate (Na_2HPO_4 , HPLC grade) were provided from Sigma-Aldrich (USA). Cefazolin was
120 obtained from Loghman Co. (Iran).

121

122 **2.2. Synthesis of materials**

123 Preparation of diatom cells: To prepare nanoporous silica from *Cyclotella sp.*, the organic
124 matrix covering the cell wall was removed using the method proposed in our previous work
125 (Gholami et al. 2020a). H₂O₂ (60 mM) and FeSO₄.4H₂O (15 mM) were added into 80 mL
126 deionized water containing *Cyclotella sp.* (4.8×10^5 cells mL⁻¹). The suspension pH was
127 adjusted to 3 using H₂SO₄ (0.1 M) and NaOH (0.1 M). The reaction mixture was then subjected
128 to ultrasound irradiation for 4 h at 40 °C using an ultrasonic bath (EP S3, 40 kHz, 300 W,
129 Sonica, Italy). The sonicated cells were separated from the liquid phase using centrifugation at
130 4500 rpm for 15 min. The collected white powder was washed with ammonium formate (0.5
131 M) and deionized water three times and dried at 10 °C for 24 h.

132 Resol precursor synthesis: A soluble resol was polymerized from phenol and formaldehyde
133 as described previously (Yang et al. 2017b). Typically, phenol (6.10 g) was melted at 42 °C
134 before addition of NaOH aqueous solution (1.30 g, 20%) with stirring for 10 min. Then,
135 formaldehyde solution (10.0 g, 37%) was dropwise added into the reaction mixture and the
136 obtained solution was mixed for 60 min at 75 °C. Subsequently, the system was cooled to
137 ambient temperature, followed by adding 2.0 M of HCl solution to reach the pH to 7.0. Water
138 was evaporated under vacuum condition at 45 °C and the final product was dispersed in THF
139 (20 wt%).

140 Synthesis of diatom templated WO_{3-x}@MC (DT-WO_{3-x}@MC): a suspension containing 20
141 mL acetone and 0.1 g treated diatom frustules was mixed with WCl₆ (0.13 g) and resol
142 precursor (0.1 g, 20 wt% in THF) and vigorously stirred for 10 min. After adding 0.25 mL
143 anhydrous ethanol, the suspension was sonicated for 2 h. The obtained mixture was
144 concentrated on a heater at 50 °C for 8 h, and then dried at 100 °C for additional 8 h. The solid
145 was calcined at 700 °C for 2 h under N₂ atmosphere. Finally, the as-synthesized nanohybrids
146 were emerged in 10 wt% HF solution for 24 h to remove silica cores. Pure WO₃ nanorods were

147 synthesized using the same procedure without adding resol precursor and diatom cells to the
148 solution.

149 **2.3. Characterization**

150 The phase purity and crystal structure of the materials were studied using a Bruker D8
151 (Germany) diffractometer with Cu radiation ($K\alpha$ line with a mean wavelength of 1.54184 Å)
152 operated at 40 kV. The morphologies were studied by a transmission electron microscope
153 (TEM, JEM-2100F, JEOL, Japan) and a scanning electron microscope (SEM, Zeiss Sigma HD
154 VP, Zeiss NTS, Cambridge, UK) equipped with an energy dispersive x-ray spectrometer (EDS,
155 Thermo Pathfinder v1.4, Madison, WI, USA). The Raman spectroscopy was performed using
156 a spectrometer (Ntegra Spectra NT-MDT, Russia) under a laser excitation at 532 nm. The
157 textural characteristics of the samples were evaluated by conducting N_2 adsorption-desorption
158 analysis at 77 K with Micromeritics 3Flex (USA) analyzer. Brunauer-Emmett-Teller (BET)
159 and Barrett–Joyner–Halenda (BJH) methods were applied to measure total pore volume,
160 surface area and pore size distribution. X-ray photoelectron spectroscopy (XPS) was conducted
161 using a Escalab 250Xi Thermo Fisher Scientific XPS spectrometer (USA). ESR spectra were
162 obtained on a Bruker ESR 300E electron paramagnetic resonance spectrometer to detect the
163 active species using 5,5-dimethyl-1-pyrroline N-oxide (DMPO) as a spin-trapping agent.

164

165 **2.4. Electrochemical and optical studies**

166 The electrochemical measurements were conducted using a PGSTAT 128 N potentiostat–
167 galvanostat (Metrohm Autolab, the Netherlands). To prepare working electrode, 40 mg of
168 photocatalyst was mixed with a solution of ethanol (800 μ L) and Nafion (5%, 200 μ L) and
169 dispersed using ultrasonication for 3 h. The resulted uniform suspension was dip-coated onto
170 a 1×1 cm² glassy carbon electrode with a geometric area of 0.07 cm² followed by overnight

171 drying at ambient temperature. A saturated calomel electrode and platinum were applied as the
172 reference and counter electrodes, respectively. LSV was performed at a scan rate of 10 mV s⁻¹.
173 ¹. Electrochemical impedance spectra (EIS) were recorded at an overpotential of 200 mV vs
174 normal hydrogen electrode (NHE) with an amplitude of 5.0 mV and a frequency range from
175 100 kHz to 0.01 Hz in Na₂SO₄ solution (0.5 M). Photoluminescence (PL) spectroscopy was
176 performed using a Perkin Elmer LS45 (USA) spectrometer. Photoluminescence (PL)
177 spectroscopy was performed using a Perkin Elmer LS45 (USA) spectrometer. To determine
178 the E_g of the photocatalysts, ultraviolet-visible diffuse reflectance spectra (UV-Vis DRS) were
179 collected using an S-250 Analytik Jena spectrophotometer (Germany).

180

181 **2.5. Photocatalytic activity test for H₂ evolution**

182 The photocatalytic H₂ evolution tests were carried out in 100 mL Pyrex flask at atmospheric
183 pressure and room temperature. Typically, the photocatalyst sample (50 mg) was added into
184 80 mL of Na₂SO₃ (0.25 M) and Na₂S (0.35 M) solution followed by sonication for 60 min.
185 Before starting the irradiation, the suspension was degassed by bubbling with highly pure N₂
186 gas for 0.5 h to ensure an anaerobic condition. A LED lamp ($\lambda > 420$ nm, 300 W, Shenzhen
187 StarVanq Technology, China) was applied as the irradiation source. During the experiment, the
188 suspension was mixed by a magnetic stirrer to keep the catalyst powder in suspension state.
189 The generated H₂ was analyzed by a gas chromatography system (Agilent Technologies
190 7890A, TCD).

191

192 **2.6. Photocatalytic activity test for CFZ degradation**

193 CFZ was chosen as target emerging contaminant to examine the photocatalytic activities of the
194 as-synthesized materials for treatment of antibiotic polluted water. Briefly, 60 mg of
195 photocatalyst was dispersed into CFZ (100 mL, 3 mM) solution and the suspension was stirred

216 for 30 min under the dark condition to obtain adsorption-desorption equilibrium. Consequently,
217 the light source was turned on and the suspension was exposed to visible irradiation using the
218 same LED lamp mentioned in section 2.5. During photocatalysis, 2.0 mL of the treated CFZ
219 solution was taken out at given time intervals followed by centrifugation for 5 min. The
220 degradation of CFZ was determined by a high-performance liquid chromatography system
201 (HPLC, HP-1090 Series II, USA), coupled with a C18 column and UV detector at 272 nm. The
202 temperature of the column was adjusted at 45 °C. The mobile phase composed of 17%
203 acetonitrile and 83% disodium hydrogen phosphate buffer was eluted at a flow-rate of 1.2 mL
204 min⁻¹. The detection limit was estimated to be 2.8 μmol L⁻¹ by plotting the calibration curves
205 from 10 μM to 100 μM. Intermediates were detected using an Agilent 6890 gas
206 chromatography (GC, Canada) instrument coupled to an Agilent 5973 mass spectrometer (MS,
207 Canada) with a TG-5MS capillary column. More detailed description has been given elsewhere
208 on the method used for preparation of the sample and the procedure of the identification of the
209 intermediates (Khataee et al. 2017). TOC concentration was determined by a multi N/C 2100S
210 TOC Analyzer. Anions concentrations were measured using an ion chromatography system
211 (Dionex ICS-2100, USA).

212

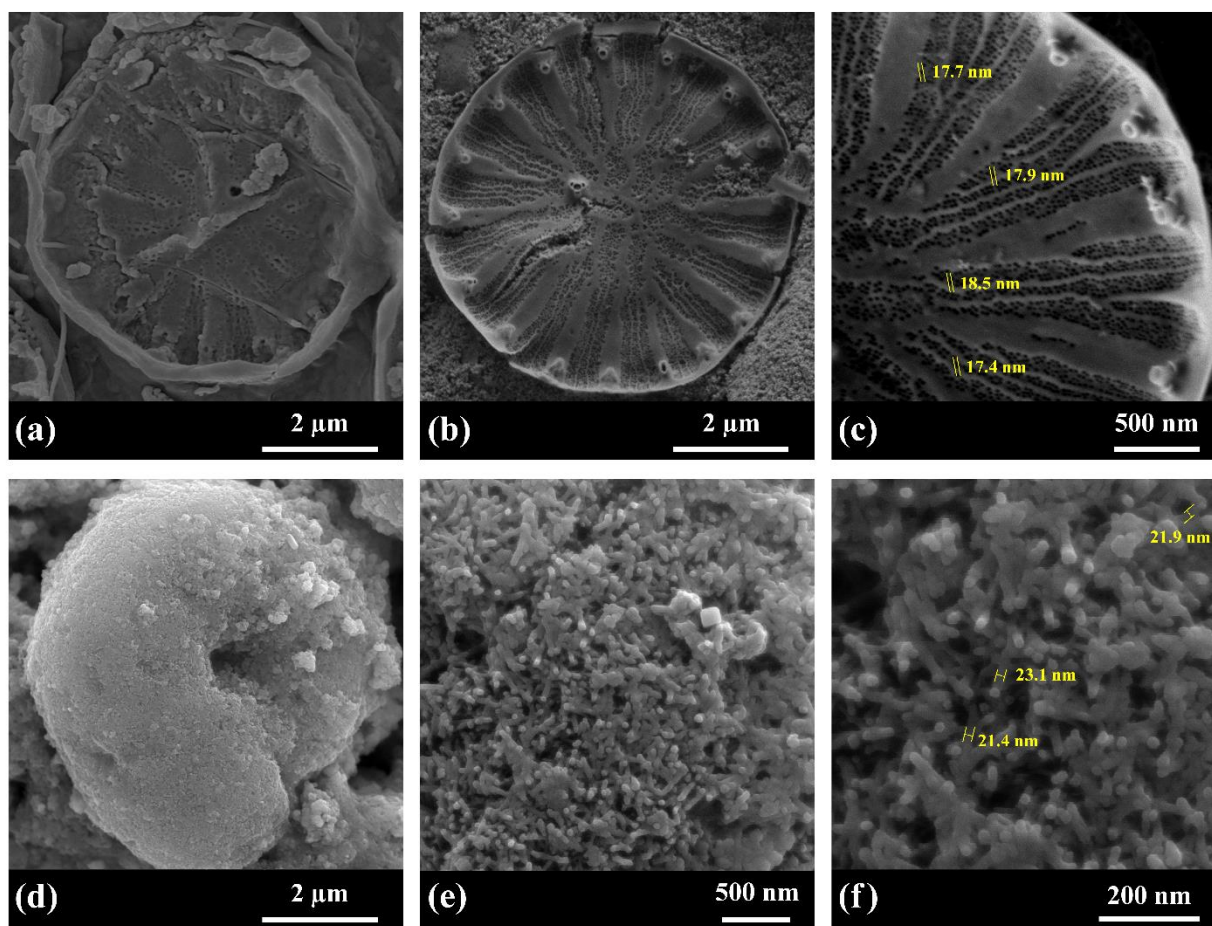
213 **3. Results and Discussion**

214 **3.1. Structural and morphological characteristics**

215 The SEM image of the untreated diatom cell is shown in Fig. 1 (a). The *Cyclotella sp.* frustules
216 are covered by some impurities and have a disk-like morphology with an outer diameter of
217 ~7 μm. To remove these materials, the diatom frustules were treated using ultrasound-assisted
218 H₂O₂/Fe²⁺ process. Fig. 1 (b) and (c) show that all the impurities and organic materials
219 surrounding the cell walls are removed during treatment of *Cyclotella sp.* cells. The treated
220 diatoms possess highly ordered nanoporous structure and the pores with controlled size

221 (diameter = 10-15 nm) are uniformly distributed on the cell walls. The SEM images of DT-
222 $\text{WO}_{3-x}@\text{MC}$ sample are displayed in Fig. 1 (d-f). As can be seen from Fig. 1 (d), the DT- $\text{WO}_{3-x}@\text{MC}$
223 nanohybrid shows a similar morphology with nearly the same diameter of diatom
224 template. It is observed that the surface of the diatom template is covered with $\text{WO}_{3-x}@\text{MC}$
225 nanohybrid and the 3D morphology of the fabricated nanostructures maintain even after
226 removing the template. The primary 3D architecture continues to grow in a homocentric growth
227 style, and subsequently, the hierarchical nanostructures with a hole in the center can be formed
228 during EISA process. Fig. 1 (e and f) exhibits that the entire hierarchical 3D nanostructures are
229 decorated by homogeneously dispersed nanorods with an average diameter of ~ 20 nm. Fig. A1
230 (a-c) shows the EDS spectra of untreated frustule, treated frustule and DT- $\text{WO}_{3-x}@\text{MC}$
231 samples, respectively. The presence of elements such as carbon, oxygen, silicon and calcium
232 atoms can be observed in spectra of both untreated and treated frustules. However, the intensity
233 of the carbon peak considerably decreases and the intensity of the peak ascribed to silicon
234 remarkably increases after removing organic materials from the surface of frustule. This can
235 be due to the purification of diatom frustule which is mainly composed of silica. Furthermore,
236 the EDS spectrum of DT- $\text{WO}_{3-x}@\text{MC}$ shows an intense peak assigned to tungsten beside
237 carbon and oxygen peaks which reveals the successful synthesis of $\text{WO}_{3-x}@\text{MC}$ nanohybrid
238 using diatom template with no impurities (Gholami et al. 2020a).

239



240

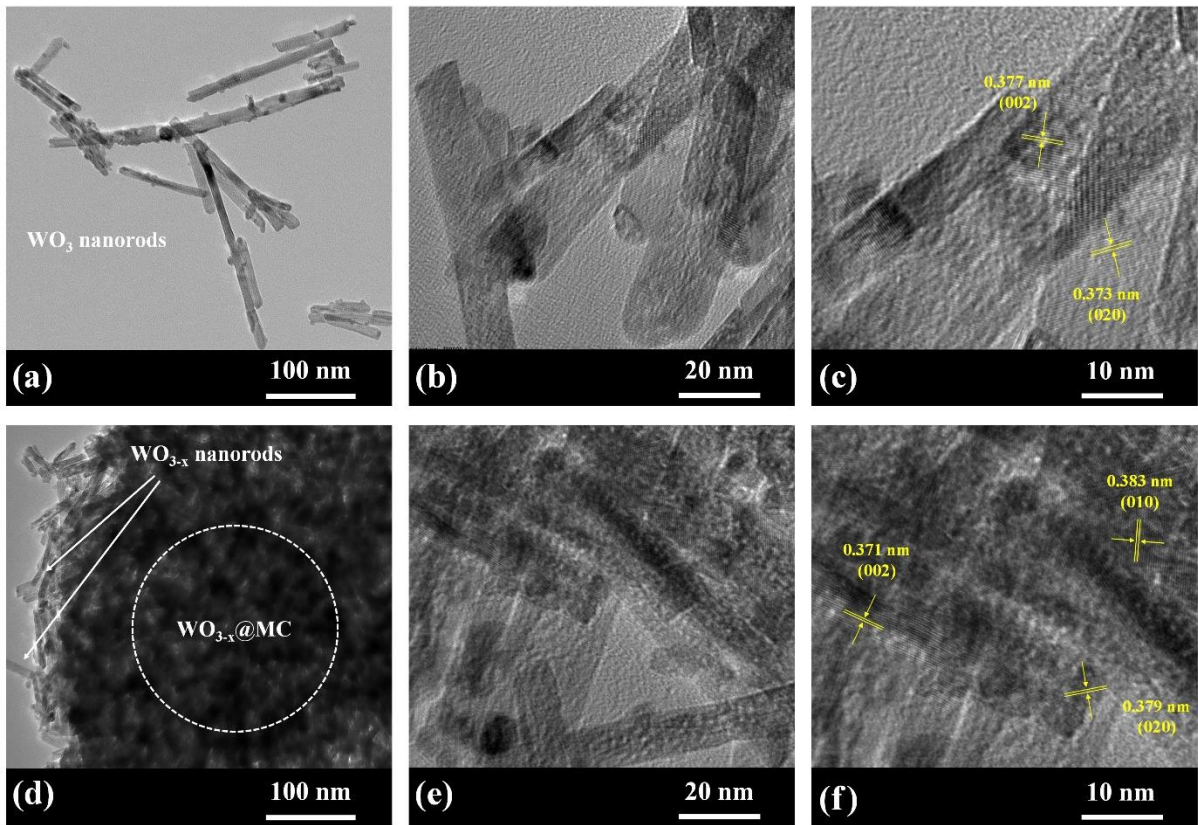
241 **Fig. 1.** SEM images of untreated frustule (a), treated frustule (b and c), and DT-WO_{3-x}@MC
 242 nanohybrid (d-f).

243

244 High-resolution TEM (HRTEM) analysis was performed to provide additional insights
 245 regarding morphological and structural properties of WO₃ nanorods and WO_{3-x}@MC
 246 nanohybrid. As can be seen from Fig. 2 (a and b), WO₃ nanorods with a diameter of ~20 nm
 247 and length of ~120 nm have a relatively uniform morphology. The interplaner distances of
 248 ~0.377 and ~0.373 nm between the regular lattice fringes can be corresponded to the (002)
 249 and (020) crystal planes of hexagonal tungsten trioxide [Fig. 2 (c)] (Chandra et al. 2019). TEM
 250 image of WO_{3-x}@MC nanohybrid [Fig. 2 (d)] reveals that tungsten oxide nanorods are entirely
 251 and homogeneously incorporated into mesoporous carbon structure. The representative
 252 HRTEM images [Fig. 2 (e and f)] demonstrate the interplanar spacing of ~0.383 nm, which

253 are related to (010) plane of $\text{WO}_{2.9}$, revealing the growth of the partially reduced tungsten
254 trioxide nanorods in the carbon matrix (Li et al. 2019b; Zhao et al. 2020).

255



256

257 **Fig. 2.** TEM and HRTEM images of WO_3 nanorods (a-c) and DT- WO_{3-x} @MC nanohybrid (d-f).

258

259 Crystallinity and structural properties of the synthesized materials were studied via XRD
260 analysis. The XRD patterns of the diatom frustules, WO_3 , and DT- WO_{3-x} @MC samples are
261 shown in Fig. 3 (a). XRD pattern diatom frustule displays no sharp diffraction, demonstrating
262 the amorphous feature of the sample. A wide peak expanded from 13.5° to 29.9° indicates the
263 existence of amorphous silica phase in diatom frustule (Shen & Zhang 2019). A similar pattern
264 has been reported in a large number of materials like diatoms (Nowak et al. 2019), opal-A (Sun
265 et al. 2018), and silica gel (Popkov et al. 2019) which contain amorphous silica. It can be

266 observed that the XRD pattern of pure WO₃ shows the diffraction peaks located at 2θ = 13.7°,
 267 22.8°, 24.1°, 26.6°, 28.1°, 33.3°, 36.4°, 46.4°, and 57.2°, corresponding to the (100), (001),
 268 (110), (101), (200), (111), (201), (002), (301), and (311) planes of WO₃ (JCPDS No. 75-2187).
 269 The unit cell parameters for the pure WO₃ were calculated to be a = 7.281 Å and c = 3.903 Å,
 270 which are in good accordance with those of hexagonal tungsten trioxide (Tu et al. 2018).
 271 Compared with pure WO₃, the XRD pattern of DT-WO_{3-x}@MC sample contains a broad peak
 272 at ~25° and three extra diffraction peaks centered at 42.5°, 48.2°, and 54.9°. The peaks
 273 observed at ~25° and 42.5° can be attributed to the (002) and (100) plans of graphite phase
 274 (JCPDS No. 08-0415) revealing the formation of mesoporous carbon through addition of resol
 275 precursor to the reaction medium during synthesis of DT-WO_{3-x}@MC (Zhou et al. 2019).
 276 Meanwhile, the additional peaks located at 48.2° and 54.9° can be assigned to (220) and (310)
 277 plans of WO_{2.9} (JCPDS No. 18-1417) which demonstrate the reducing role of carbon precursor
 278 during heat-treatment (Li et al. 2019c). The mean crystallite sizes of pure and MC modified
 279 tungsten oxide were calculated to be 24 and 18 nm, respectively, from the full width at half
 280 maximum (FWHM) of (200) plan using the Debye-Scherrer formula (Gu et al. 2019):

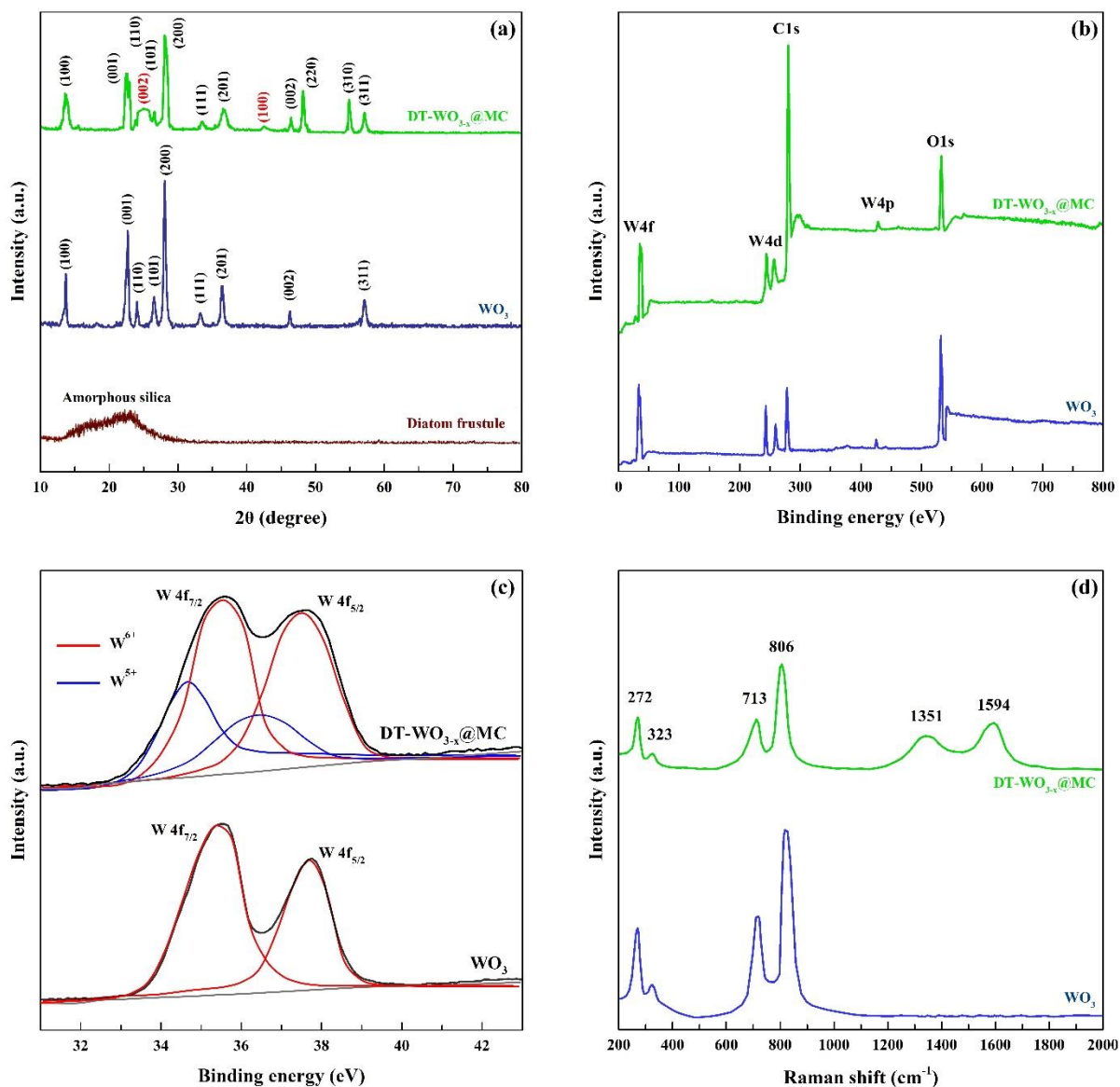
$$281 \quad D_{XRD} = \frac{K\lambda}{\beta \cos \theta} \quad (1)$$

282 where D_{XRD} is crystallite size in nm, β is the FWHM in radian, λ is the X-ray wavelength
 283 (1.5418 Å), K is the unitless Scherrer constant (0.89) and θ is the Bragg angle. These results
 284 show that the diatom surface can act as a nucleation agent for graphitization of carbon precursor
 285 and growth of tungsten oxide nanorods resulting in production of a nanostructured material
 286 with smaller crystallite size (Hu et al. 2017; de Moraes et al. 2020).

287 XPS spectroscopy was performed to investigate the elemental valence and surface
 288 electronic characteristics of the samples. Fig. 3 (b) displays the XPS spectra of the samples.
 289 Both WO₃ and DT-WO_{3-x}@MC display the existence of the same elements as confirmed from

290 the main peaks at 35.7, 284.8 and 533.7 eV in XPS full spectra attributing to W 4f, C 1s and O
291 1s, respectively. The elemental composition of WO₃ was determined to be C=15.3 at%,
292 W=18.6 at% and O=66.1 at%. XPS composition of DT-WO_{3-x}@MC shows higher carbon
293 content of 47.4 at% with W and O content of 14.9 at% and 37.7 at%, respectively. In
294 comparison to the pure WO₃, the C1s peak on the DT-WO_{3-x}@MC spectrum is considerably
295 intense, revealing the production of mesoporous carbon during synthesis of the nanohybrid
296 material. The W 4f high-resolution spectrum of WO₃ exhibits two peaks at 35.6 and 37.7 eV,
297 corresponding to 4f7/2 and 4f5/2, respectively, demonstrating that the main part of tungsten in
298 the structure of WO₃ is in 6+ oxidation state [Fig. 3 (c)] (Zhan et al. 2019a). However, the W4f
299 curve of DT-WO_{3-x}@MC shows new peaks at 34.3 and 36.6 eV which are ascribed to W⁵⁺.
300 These results are well in consistent with those achieved from XRD analysis and reveal that
301 tungsten oxide is partially reduced by resol polymer during EISA process (Tu et al. 2018; Cui
302 et al. 2019).

303 Raman spectroscopy provided more detailed structural and electronic characteristics of the
304 samples [Fig. 3 (d)]. The peaks 713 and 806 cm⁻¹ are assigned to the stretching modes of ν(O–
305 W–O), while 272 and 323 cm⁻¹ are attributed to the bending modes of δ(O–W–O) (Pervez et
306 al. 2015; Yan et al. 2015). In addition, there are two broad bands located at 1351 and 1594 cm⁻¹
307 in DT-WO_{3-x}@MC spectrum, respectively. The band at 1351 cm⁻¹ (D-band) is attributed to
308 the A_{1g} vibration of carbon and the band at 1594 cm⁻¹ (G-band) is associated with the E_{2g}
309 vibration of sp²-bonded carbon (Pervez et al. 2015; Yan et al. 2015). These results are in good
310 accordance with those reported by Pervez et al. for WO₃ mesosponge@Carbon (Pervez et al.
311 2015). Comparing the Raman spectrum of pure WO₃ with that of synthesized nanohybrid
312 material confirms successful fabrication of WO_{3-x}@MC.



313

314 **Fig. 3.** XRD patterns (a), XPS full spectra (b), W 4f high-resolution spectra (c) and Raman

315 spectra, (d) of WO_3 nanorods and $\text{DT-WO}_{3-x}\text{@MC}$ nanohybrid.

316

317 Fig. 4 (a) shows nitrogen adsorption-desorption isotherms of diatom frustules, WO_3 and

318 $\text{DT-WO}_{3-x}\text{@MC}$. All the developed materials show isotherms of type IV with a distinct type

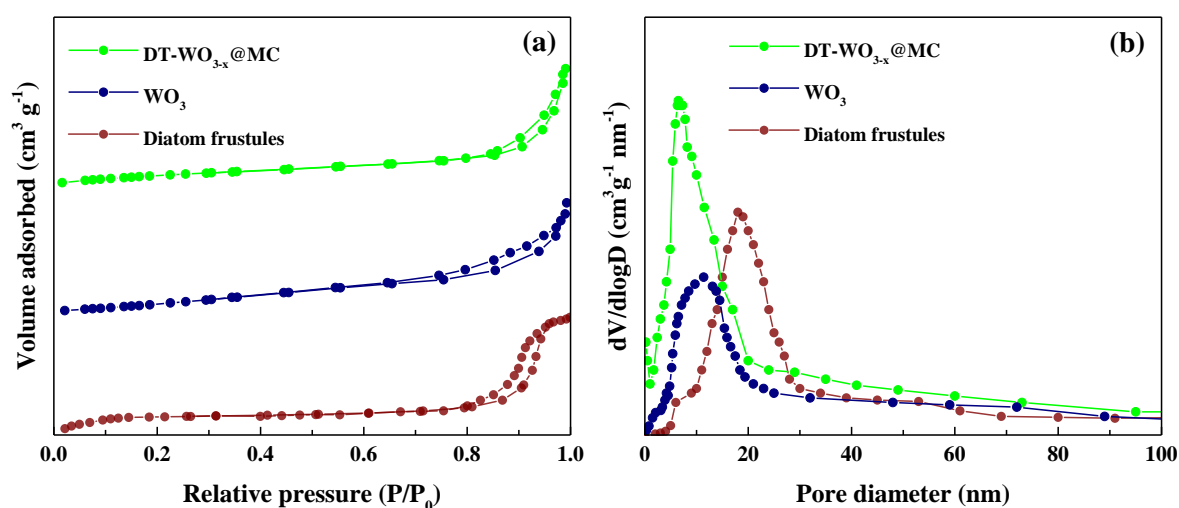
319 H3 hysteresis loops according to the IUPAC classification, which is the property of

320 mesoporous materials [Fig. 4 (a)] (Su et al. 2016). The obtained results were analyzed by

321 Brunauer–Emmett–Teller (BET) and Barrett–Joyner–Halenda (BJH) methods to gain

322 information regarding specific surface area, total pore volume, and pore size distribution of the
 323 samples, respectively (Table A1). The pore size distribution peaks [Fig. 4 (b)] are well
 324 concentrated at 18.13, 11.42, 6.59 nm for diatom frustules, WO_3 and DT- WO_{3-x} @MC,
 325 respectively. The smaller S_{BET} and total pore volume of WO_3 are due to its high density. It is
 326 assumed that reduced WO_3 in DT- WO_{3-x} @MC nanohybrid possesses the density close to WO_3
 327 density because it exhibits similar crystallinity and morphology as WO_3 , based on the results
 328 obtained from XRD and HRTEM analyses. However, when WO_{3-x} is integrated with the MC,
 329 the S_{BET} and total pore volume enhance from $87.52 \text{ m}^2 \text{ g}^{-1}$ and $0.10 \text{ cm}^3 \text{ g}^{-1}$ to $195.37 \text{ m}^2 \text{ g}^{-1}$
 330 and $0.393 \text{ cm}^3 \text{ g}^{-1}$, respectively. This is due to the formation of highly ordered mesoporous
 331 nanostructures resulted from effective carbonization at a relatively high temperature, leaving a
 332 lot of void spaces within the carbon structure (Kong et al. 2019).

333



334

335 **Fig. 4.** N_2 adsorption-desorption isotherms (a) and pore size distributions (b) of diatom

336 frustules, WO_3 and DT- WO_{3-x} @MC.

337

338 3.2. Optical and electrochemical properties

339 The UV–Vis DRS spectra were collected to evaluate the light absorption potential of the
340 as-synthesized materials. It is obviously observed in Fig. 5 (a) that WO_3 has high absorption in
341 the UV region with insignificant visible light harvesting. However, the DT- WO_{3-x} @MC shows
342 a wide and intense absorption across both UV and visible regions, demonstrating the generation
343 of further photo-produced electron-hole pairs under whole wavelength range, especially visible
344 region. Therefore, the nanohybrid photocatalyst containing MC can efficiently harvest visible
345 light to produce charge carriers. Moreover, the weak band at around 320 nm in absorption
346 spectrum of DT- WO_{3-x} @MC can be attributed to the polycyclic aromatic hydrocarbons (π -
347 conjugated species) which is consistent with previous works (Zhang et al. 2017).

348 The E_g of the WO_3 and DT- WO_{3-x} @MC were measured to be 2.85 and 2.67 eV as the
349 slopes of curves of $(\alpha h\nu)^2$ versus $h\nu$, where h , ν , and α are Planck constant, light frequency and
350 absorbance coefficient, respectively [Fig. 5 (b)] (Kumar et al. 2018). Integration of mesoporous
351 carbon could increase the light absorption efficiency of the photocatalyst by two reasons.
352 Firstly, the existence of mesoporous carbon can directly accelerate the separation of photo-
353 produced carrier and improve electron delocalization in the system, thus enhancing the
354 photocatalytic activity (Ding et al. 2018; Sun et al. 2019). Secondly, it is demonstrated using
355 XRD and XPS analyses that WO_3 is partially reduced to WO_{3-x} by resol polymer during EISA
356 synthesis; the presence of oxygen vacancies can produce filled impurity bands that
357 considerably decrease the E_g (Wang et al. 2018).

358 The PL spectroscopy is a useful analysis to investigate surface photochemical processes
359 involving photo-excited charge carriers from semiconductor materials. As illustrated in Fig. 5
360 (c), the PL intensity resulted from DT- WO_{3-x} @MC is lower than that of pure WO_3 nanorods,
361 indicating more effective prevention of electron-hole pairs recombination in the nanohybrid
362 photocatalyst. This is ascribed to the incorporation of high-conductive mesoporous carbon in
363 3D structure of DT- WO_{3-x} @MC which could act as an electron acceptor to receive and shuttle

364 the electrons generated from WO_3 through light excitation (Zhan et al. 2019b). As a
365 consequent, the separation and lifetime of charge carriers are enhanced, resulting in the
366 improved photocatalytic performance of the DT- WO_{3-x} @MC nanohybrid.

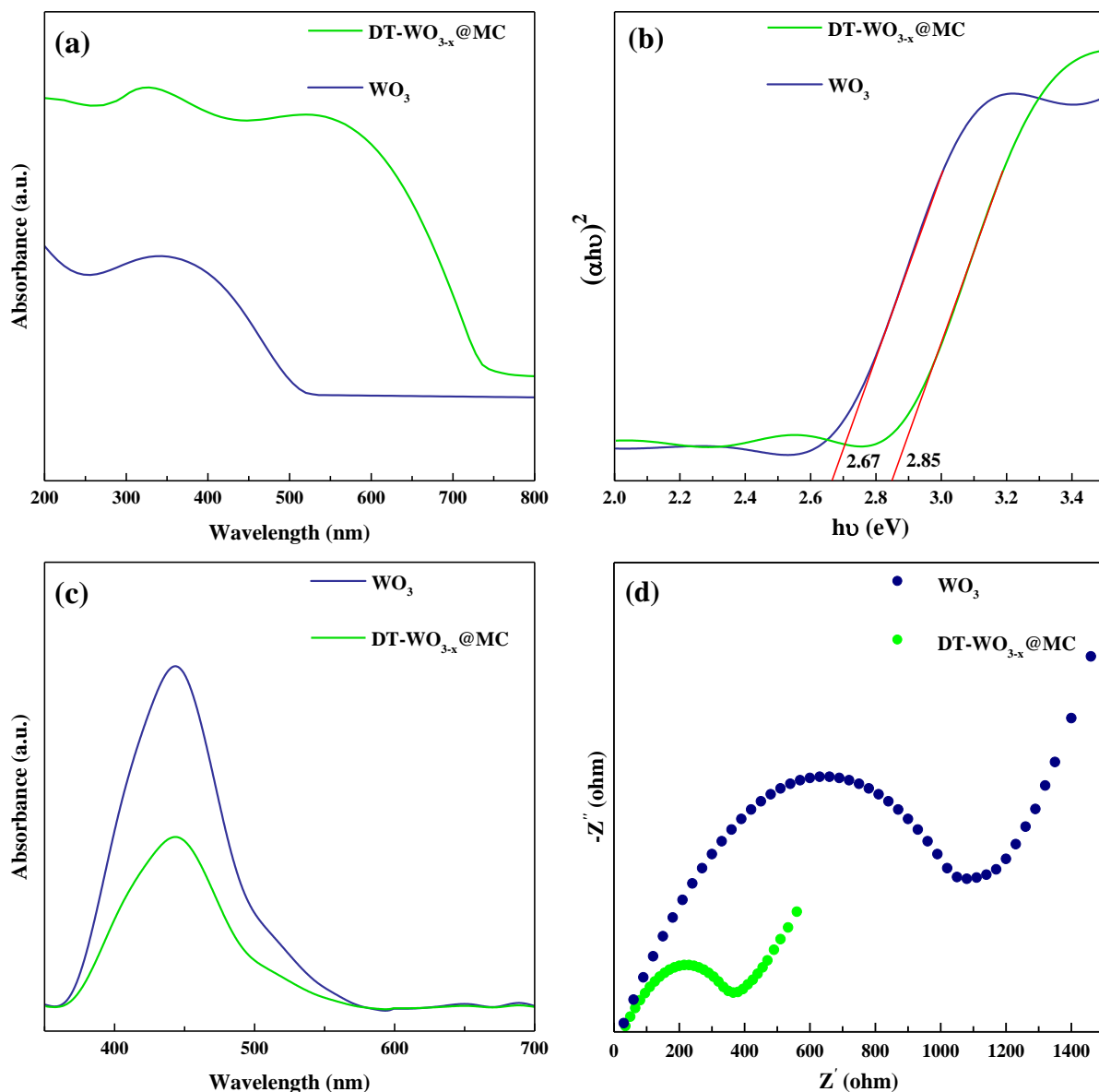
367 The electron-transport capacity of the samples was studied by electrochemical impedance
368 spectroscopy (EIS) test [Fig. 5 (d)]. Generally, in order to determine the variations in the
369 interfacial characteristics, the changes in the electrochemical impedance are essential
370 indicators. In semiconductors, a high interfacial transport facilitates the separation of the charge
371 carriers and weakens the recombination phenomenon, and this results in reduced impedance.
372 The arc radius of the EIS Nyquist plot of the DT- WO_{3-x} @MC nanohybrid is smaller than that
373 of pure tungsten oxide, revealing a quicker interfacial charge transport to the electron acceptor,
374 which leads to the effective electron-hole pairs separation (Zhang et al. 2017). In fact, following
375 the excitation of photoproducted electrons from the valence band (VB) to the conduction band
376 (CB) of WO_{3-x} , they are then transported to mesoporous carbon, preventing the fast
377 recombination of electron-hole pairs. The MC is a good electron acceptor due to its conjugated
378 structure, which makes it a separation center for the excited electrons and holes (Yang et al.
379 2017a). Zhang et al. have also reported a red-shift of the band edge absorption for WO_{3-x}/C
380 compared to pure WO_3 and demonstrated that carbon coating and oxygen vacancies bring
381 WO_{3-x}/C strong absorption at higher wavelengths and improve the charge carrier separation
382 performance (Zhang et al. 2020b).

383

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385

386



387

388 **Fig. 5.** UV-DRS (a), $(\alpha h\nu)^2$ - $h\nu$ plots (b), PL spectra (c) and EIS Nyquist plots (d) of WO_3

389

nanorods and $\text{DT-WO}_{3-x}@MC$ nano hybrid.

390

391 3.3. Photocatalytic H_2 production

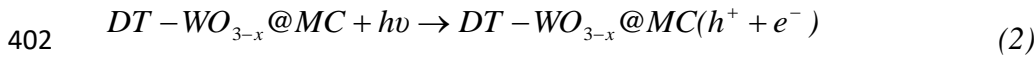
392 Fig. 6 (a) represents the photocatalytic H_2 evolution from water using WO_3 and $\text{DT-WO}_{3-x}@MC$

393 $\text{x}@MC$ photocatalysts under visible light irradiation. When the light is irradiated on the

394 photocatalyst surface, the electrons and holes are formed at the photocatalyst/solution interface

395 according to Eq. (2), and take part in reduction and oxidation processes, respectively

396 (Gomathisankar et al. 2013). The holes convert water molecules to H^+ ions and $\cdot OH$ radicals
 397 [Eq. (3)] . At the same time, the electrons react with H_2O and H^+ to produce hydrogen
 398 molecules [Eq. (4) and (5), respectively]. Na_2S/Na_2SO_3 is used as sacrificial reagents to
 399 scavenge the generated holes based on Eqs. (6-9). The generation of S_2^{2-} anions, which act as
 400 optical filters and compete with proton reduction, is inhibited by adding SO_3^{2-} anions resulting
 401 in production of thiosulfate anions (Gomathisankar et al. 2013).



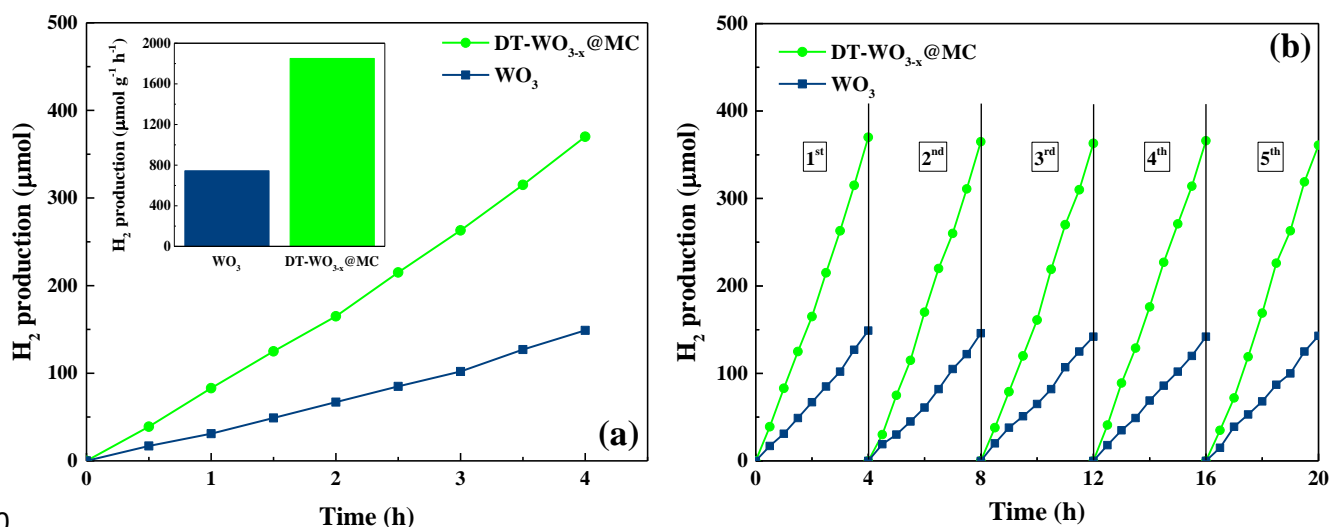
410

411 As can be seen in Fig. 6 (a), the relatively low H_2 production rate ($746 \mu mol g^{-1} h^{-1}$) is
 412 observed using WO_3 nanorods due to poor absorption in the visible region, large E_g , low
 413 specific surface area and limited charge transport. The 3D $WO_{3-x}@MC$ nanohybrid shows
 414 almost 2.5 times higher photocatalytic efficiency ($1851 \mu mol g^{-1} h^{-1}$) than the WO_3 nanorods
 415 mainly owing to the improved light absorption, lower recombination of electrons and holes,
 416 higher specific surface area and higher electron transfer ability. The synthesized nanohybrid
 417 material in this study possesses higher photocatalytic H_2 production rate than previously
 418 reported WO_3 -based photocatalysts (Zhang et al. 2019; Sun et al. 2020; Zhang et al. 2020c). In

419 DT-WO_{3-x}@MC system, initially, WO_{3-x} nanorods are excited to produce charge carriers under
420 visible light irradiation. Subsequently, the photo-produced electrons in the CB of WO_{3-x} can be
421 directly transported to MC via solid-solid interface because of its high electron transport ability
422 and participate in the surface reduction for hydrogen production (Kailasam et al. 2015). On the
423 other hand, the enhanced photocatalytic activity of synthesized nanohybrid should also be
424 ascribed to its 3D structural advantages which can considerably decrease the diffusion length
425 of charge migration and facilitate electron transfer on the surface terminal sites (Wan et al.
426 2018).

427 Fig. A2 represents the LSV curves of the photocatalysts. DT-WO_{3-x}@MC shows higher
428 current density than WO_{3-x} nanorods indicating stronger electron migration. The LSV curve of
429 a commercial Pt/C electrode was also evaluated, for comparison. The potential value for the
430 current density of 10 mA cm⁻² is a frequently used criterion to examine the electrochemical
431 properties of photocatalysts (Zhao et al. 2016). Pt/C electrode exhibits extremely high activity
432 with a very low overpotential (0.08 V vs NHE). The overpotential of DT-WO_{3-x}@MC at 10
433 mA cm⁻² is found to be 0.13 V vs NHE which is smaller than that of WO₃ (2.9 vs NHE) and
434 much closer to that of Pt/C electrode. The cycled photocatalytic runs were conducted to verify
435 the stability of WO₃ nanorods and DT-WO_{3-x}@MC nanohybrid during long-term application
436 as photocatalysts for H₂ production [Fig. 6 (b)]. The photocatalytic efficiencies of both
437 synthesized materials exhibit no remarkable decrease during five repeated cycles for 20 h,
438 confirming their high stability for visible-light-driven hydrogen production.

439



440

441 **Fig. 6.** Time course of hydrogen production and corresponding rate comparison (a) and cyclic

442 stability (b) for WO_3 nanorods and $\text{DT-WO}_{3-x}\text{@MC}$ nano hybrid under visible light;

443 Experimental conditions: [photocatalyst] = 0.625 g L^{-1} and temperature = $30 \pm 1 \text{ }^\circ\text{C}$.

444

445 3.4. Photocatalytic CFZ degradation

446 3.4.1. Comparison of WO_3 and $\text{DT-WO}_{3-x}\text{@MC}$

447 In order to further investigate the photocatalytic application of $\text{DT-WO}_{3-x}\text{@MC}$, a series of

448 degradation experiments were done using CFZ as the target emerging contaminant. Control

449 tests were carried out in the presence of catalysts and in the dark condition and it was found

450 that the adsorption of CFZ on WO_3 and $\text{DT-WO}_{3-x}\text{@MC}$ photocatalysts was negligible (9.4%

451 and 13.1%, respectively). According to the obtained results, the role of visible light irradiation

452 alone in the removal of CFZ is also inconsiderable (4.9%) within 90 min [Fig. 7 (a)]. The pure

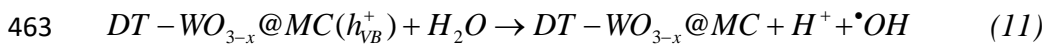
453 WO_3 displays a CFZ degradation efficiency of 35.4% within 90 min photocatalysis under

454 visible-light irradiation. It has already been proved that WO_3 is an excellent photocatalyst for

455 the degradation of different organic pollutants under UV light irradiation (Khan et al. 2018).

456 However, the poor absorption, relatively wide band gap and fast recombination of electrons

457 and holes generated through Eq. (2), restrict the application of pure WO₃ in photo-degradation
 458 of different contaminants under visible light irradiation. Interestingly, the integration of MC
 459 and WO_{3-x} fabricates an superior nanohybrid photocatalyst which leads to the almost complete
 460 degradation of CFZ (98.7%) under the same operational conditions according to the following
 461 equations (Darvishi Cheshmeh Soltani et al. 2016):



464

465 When WO_{3-x} nanorods are uniformly embedded in 3D mesoporous carbon framework,
 466 further active sites are exposed to produce more oxidizing species and improve the CFZ
 467 degradation efficiency because of higher specific surface area. Furthermore, according to the
 468 results obtained from UV-Vis DRS, DT-WO_{3-x}@MC shows the enhanced absorption potential
 469 in the visible region compared to the WO₃ nanorods. Corresponding band gap measurement
 470 also revealed that E_g of DT-WO_{3-x}@MC (2.67 eV) is considerably lower than that of WO₃
 471 nanorods (2.85 eV). In addition, integration of MC with WO_{3-x} as 3D structure developed by
 472 diatom template reduces the charge carriers recombination and increases the electron-transport
 473 capacity of the photocatalyst. The removal rate of CFZ by the studied systems was also
 474 investigated using pseudo-first order model [Eq. (12)] and the calculated values of rate
 475 constants (min⁻¹) were presented in Fig. 7 (b). As can be observed, the highest rate constant
 476 corresponds to the implementation of DT-WO_{3-x}@MC for degradation of CFZ. The apparent
 477 rate constant (k_{app}) value for the use of WO₃ under visible light irradiation is 62×10⁻⁴ min⁻¹,
 478 while the application of DT-WO_{3-x}@MC as photocatalyst results in the k_{app} value of 435×10⁻⁴
 479 min⁻¹. To demonstrate the enhancement in photocatalytic degradation made by replacing WO₃
 480 nanorods with DT-WO_{3-x}@MC nanohybrid, the synergy factors were determined using the

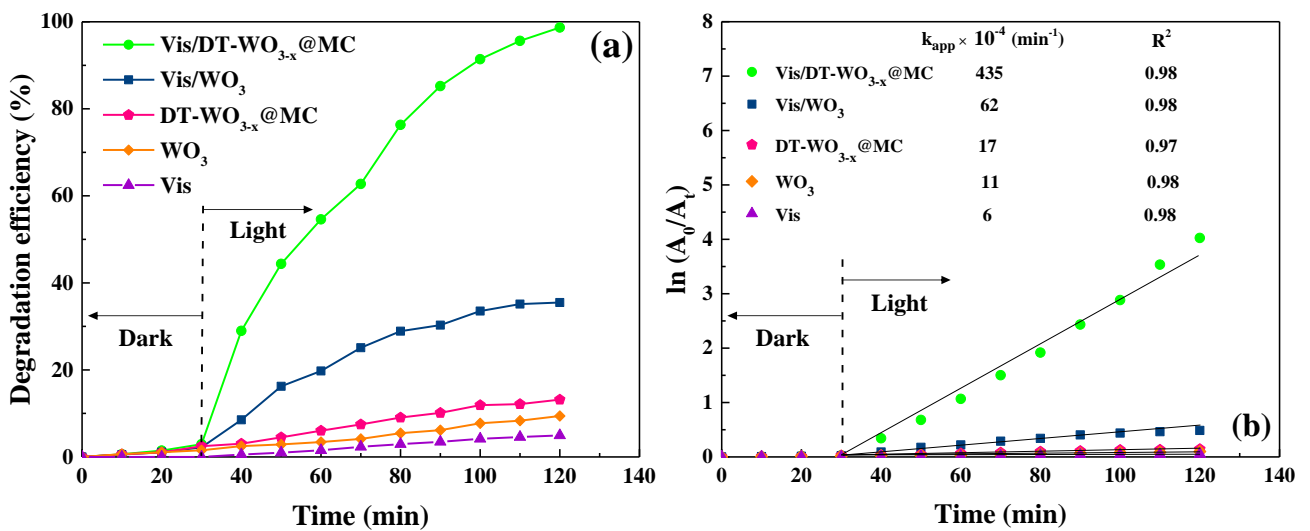
481 calculated k_{app} values for both WO_3 and $DT-WO_{3-x}@MC$ based on Eqs. (13 and 14) as 3.6 and
 482 18.9 (Gholami et al. 2019). The reusability of the nanohybrid was also studied by performing
 483 five successive runs. After each run, the photocatalyst was washed with distilled water and
 484 dried at room temperature. Then, a fresh pollutant solution was prepared and the next run was
 485 carried out under the identical operational conditions. As shown in Fig. A3, there is no
 486 considerable decrease in the photocatalytic activity of $DT-WO_{3-x}@MC$ after 5 runs.

$$487 \quad \ln\left(\frac{C_0}{C_t}\right) = k_{app}t \quad (12)$$

$$488 \quad \text{Synergy factor}_{Vis/WO_3} = \frac{k_{app(Vis/WO_3)}}{k_{app(Vis)} + k_{app(WO_3)}} \quad (13)$$

$$489 \quad \text{Synergy factor}_{Vis/DT-WO_{3-x}@MC} = \frac{k_{app(Vis/DT-WO_{3-x}@MC)}}{k_{app(Vis)} + k_{app(DT-WO_{3-x}@MC)}} \quad (14)$$

490
491



492

493 **Fig. 7.** Removal of CFZ using different processes (a) and corresponding kinetic analysis

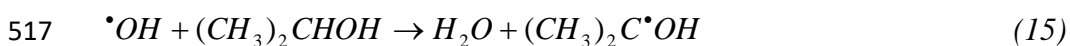
494 based on the pseudo-first-order model (b); Experimental conditions: [photocatalyst] = 0.6 g L⁻¹,

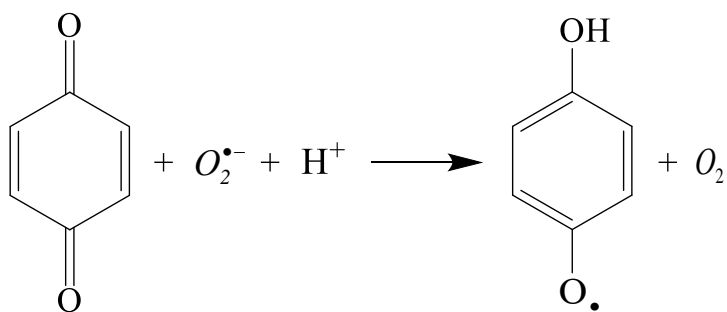
495 [CFZ] = 3 mM, pH = 6.3 and temperature = 30±1 °C.

496

497 3.4.2. CFZ degradation mechanism and pathway

498 To detect the major reactive species involved in photocatalytic degradation of CFZ,
499 scavenging experiments were carried out during photocatalysis using isopropanol (IPA), 1,4-
500 benzoquinone (BQ), and ethylenediaminetetraacetic acid disodium salt (EDTA-2Na) as $\cdot\text{OH}$,
501 $\text{O}_2^{\cdot-}$ and h^+ , scavengers, respectively (Kumar et al. 2018). The scavengers can inhibit the
502 activity of oxidizing species, leading to the limited degradation of the pollutant molecules. The
503 influence of scavenging species on photocatalytic degradation of CFZ is shown in Fig. 8 (a).
504 All the scavenging species negatively affect the DT-WO_{3-x}@MC catalyzed CFZ
505 photodegradation. EDTA-2Na (2 mM) has the lowest scavenging effect on the photocatalytic
506 performance of DT-WO_{3-x}@MC, which indicates that h^+ may not play the main role in the
507 degradation of CFZ. According to Eq. (15), $\cdot\text{OH}$ radicals are mainly scavenged by the IPA
508 molecules, generating isopropanol radicals (Wojnárovits et al. 2006). On the other hand, BQ
509 can react with $\text{O}_2^{\cdot-}$ to produce semiquinone radicals based on Eq. (16) (Schneider et al. 2020).
510 The addition of BQ led to a higher inhibiting effect than that of IPA, which reveals that $\text{O}_2^{\cdot-}$
511 acts as the most significant oxidizing species during photocatalysis. In the presence of 2 mM
512 EDTA-2Na, IPA and BQ, the degradation efficiency decreases from 98.7% to 79.5%, 66.9%
513 and 35.7%, respectively. ESR spin trapping technique was also applied to validate the
514 observations for production of reactive oxygen species. As can be observed from Fig. 8 (b), the
515 intensities of ESR signals of DMPO- $\text{O}_2^{\cdot-}$ are higher than those of DMPO- $\cdot\text{OH}$ which further
516 clarifies the key role of $\text{O}_2^{\cdot-}$ radicals during photodegradation of CFZ.

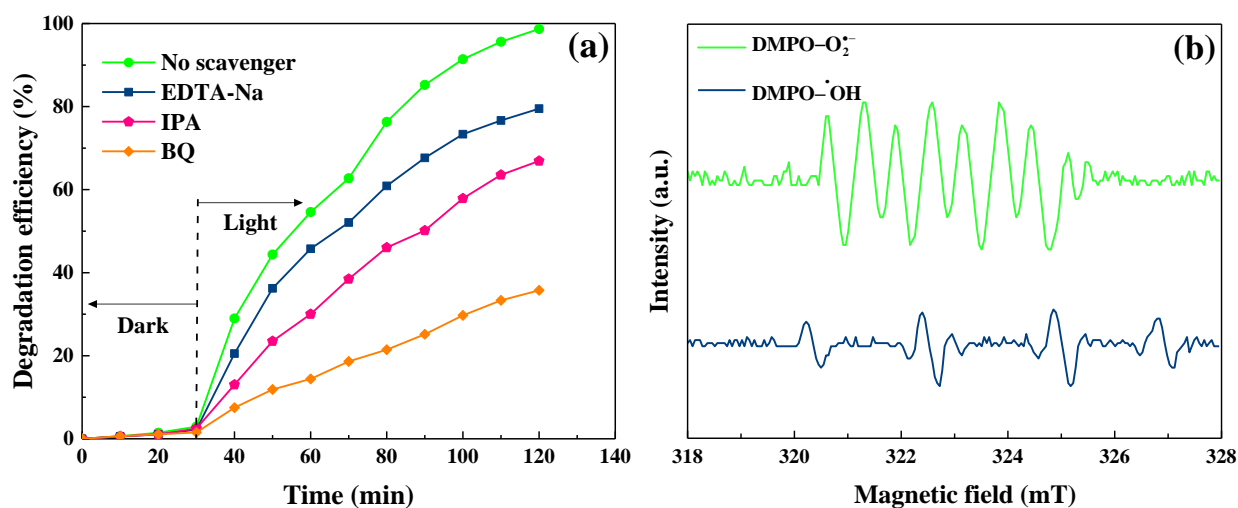




518

(16)

519



520

521 **Fig. 8.** (a) Effect of different scavengers on the CFZ degradation (Experimental conditions:
 522 [photocatalyst] = 0.6 g L⁻¹, [CFZ] = 3 mM, pH = 6.3, [scavenger] = 10 mM and temperature
 523 = 30±1 °C and (b) ESR spectra of DMPO-O₂^{•-} and DMPO-[•]OH.

524

525 The GC-MS was used to recognize the intermediates generated during the CFZ
 526 photodegradation. GC-MS spectra of detected intermediate compounds are shown in Fig. A4.
 527 Based on the identified by-products, three main competing pathways were suggested for the
 528 decomposition of CFZ (Fig. A5). In pathway I, the CFZ molecule is decomposed to 5-methyl-
 529 3H-1,3,4-thiadiazole-2-thiol. The -SH functional group is removed from 5-methyl-3H-1,3,4-
 530 thiadiazole-2-thiol and subsequently thiadiazole ring is broken to form 1-(hydroxymethyl)-3-

531 methylthiourea. In pathway II, the parent molecule is converted to 7-amino-3-methyl-8-oxo-5-
532 thia-1-azabicyclo[4.2.0]oct-2-ene-2-carboxylic acid. Then, oxidation of β -lactam ring by
533 reactive species precedes ring opening to generate aliphatic compounds like 2-
534 methylbutanedioic acid and 2,3-diaminopropanoic acid. In pathway III, 2-(tetrazol-1-yl) acetic
535 acid is firstly formed by the attack of reactive species. Subsequently, 2-(tetrazol-1-yl) acetic
536 acid is decomposed to 2H-tetrazole and oxalic acid. For all the pathways, the as-generated
537 compounds are finally oxidized to water, carbon dioxide and inorganic products. The TOC and
538 IC analyses were performed to examine the ability of DT-WO_{3-x}@MC photocatalyst in
539 mineralization of CFZ solution. The results obtained from TOC analysis demonstrates the TOC
540 removal efficiencies of 61.2% and 83.5% within 90 and 180 min, respectively. Moreover,
541 according to the results of IC analysis, the concentrations of NO₃⁻, NH₄⁺ and SO₄²⁻ increase
542 from 0.47, 0.13 and 0.39 mg L⁻¹ to 4.24, 1.93 and 1.37 mg L⁻¹ after 90 min reaction time. This
543 reveals the production of ionic products from the CFZ degradation. Reduction of TOC
544 concentration as well as the release of sulfur and nitrogen species during the degradation
545 process indicates the effective mineralization of CFZ solution.

546

547 **4. Conclusions**

548 In this work, *Cyclotella sp.* was utilized as a natural template to fabricate a highly ordered
549 mesoporous carbon embedding reduced tungsten oxide nanorods through a one-pot
550 evaporation-induced self-assembly method. The XRD and XPS analyses proved the successful
551 synthesis of WO₃ nanorods and DT-WO_{3-x}@MC nanohybrid and partial reduction of tungsten
552 oxide by resol precursor. The unique structure of diatom frustule could increase the specific
553 surface area and available active sites of DT-WO_{3-x}@MC by avoiding aggregation. The results
554 of EIS and PL analyses indicated that the recombination of electron-hole pairs was significantly

555 prevented by integrating reduced tungsten oxide and mesoporous carbon. The band gap value
556 was calculated to be 2.85 and 2.67 eV for DT-WO_{3-x}@MC nanohybrid and WO₃ nanorods,
557 respectively. DT-WO_{3-x}@MC exhibited a higher hydrogen production rate of 1851 μmol g⁻¹ h⁻¹
558 ¹ which was almost 2.5 times higher than that of WO₃ nanorods (746 μmol g⁻¹ h⁻¹). Moreover,
559 the rate constant of pseudo-first order model for photodegradation of CFZ using WO₃ and DT-
560 WO_{3-x}@MC was calculated as 62×10⁻⁴ and 435×10⁻⁴ min⁻¹, respectively. O₂^{•-} radicals were
561 found to be the most reactive species involved in degradation of CFZ by performing scavenging
562 experiments and ESR analysis. The synthesized photocatalyst could be advantageous for
563 degradation of target antibiotic pollutant, as EISA method for fabrication of DT-WO_{3-x}@MC
564 involves a simple one-pot synthetic strategy using inexpensive diatom frustule as a natural
565 template.

566

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571

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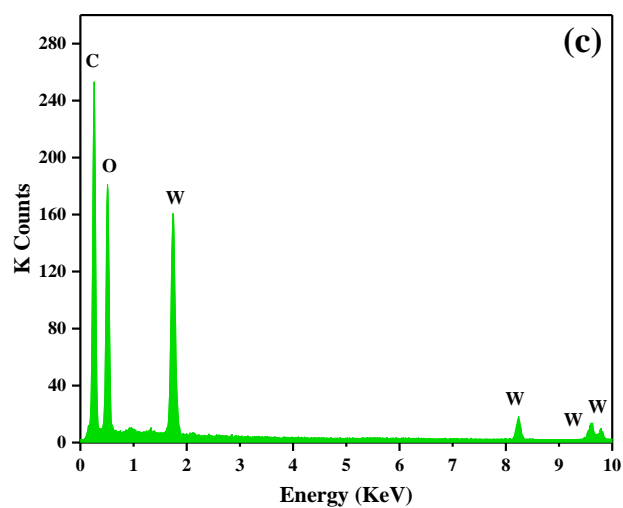
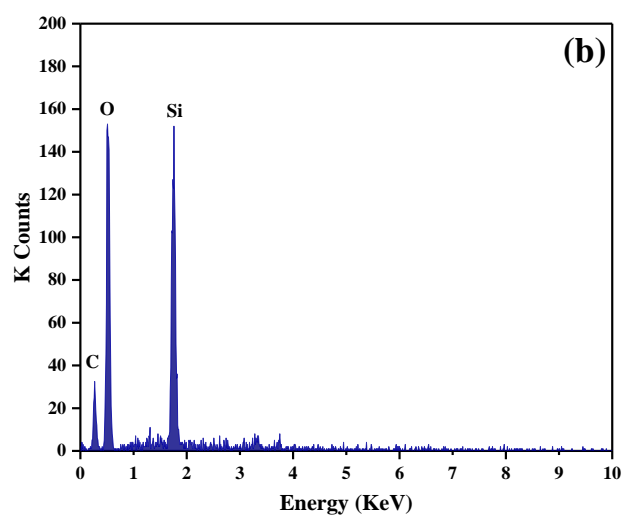
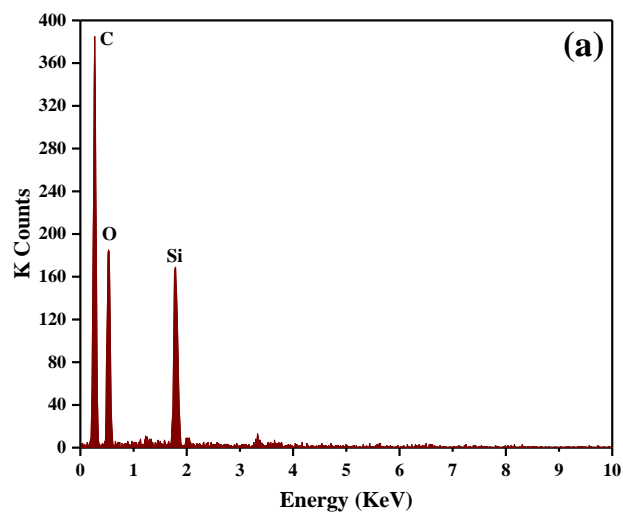
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Appendix



819 **Fig. A1.** EDS spectra of untreated frustule (a), treated frustule (b) and DT-WO_{3-x}@MC nanohybrid (c)

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821 **Table A1.** Textural characteristics of the samples.

Sample	Surface area (m ² g ⁻¹)	Total pore volume (cm ³ g ⁻¹)	Pore size (nm)
Diatom frustules	132.67	0.18	18.13
WO ₃	87.52	0.10	11.42
DT-WO _{3-x} @MC	195.37	0.39	6.59

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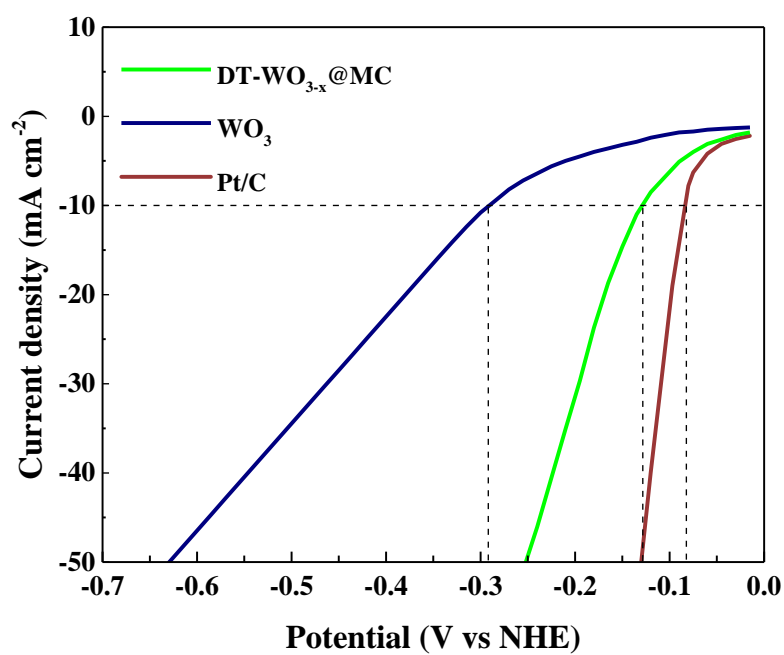
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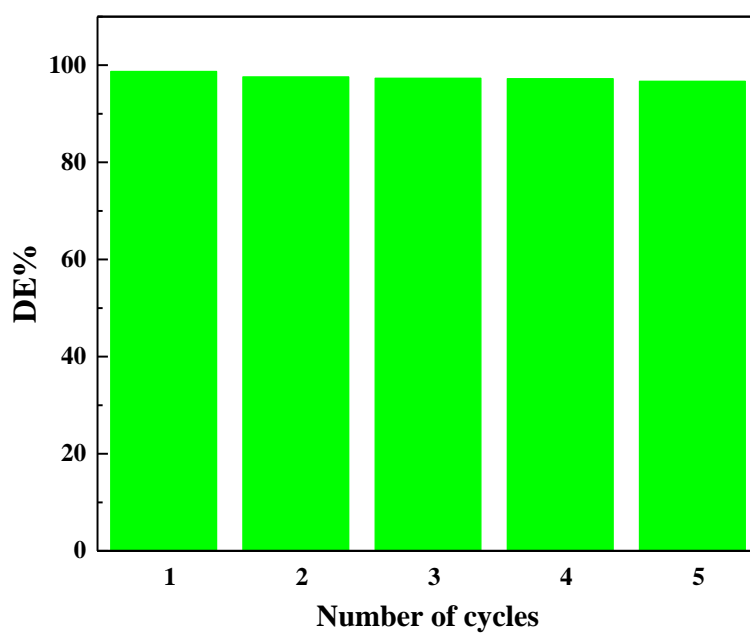
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Fig. A2. Comparison of LSV curves of WO₃ and DT-WO_{3-x}@MC coated GCE electrodes with that of commercial Pt/C electrode.

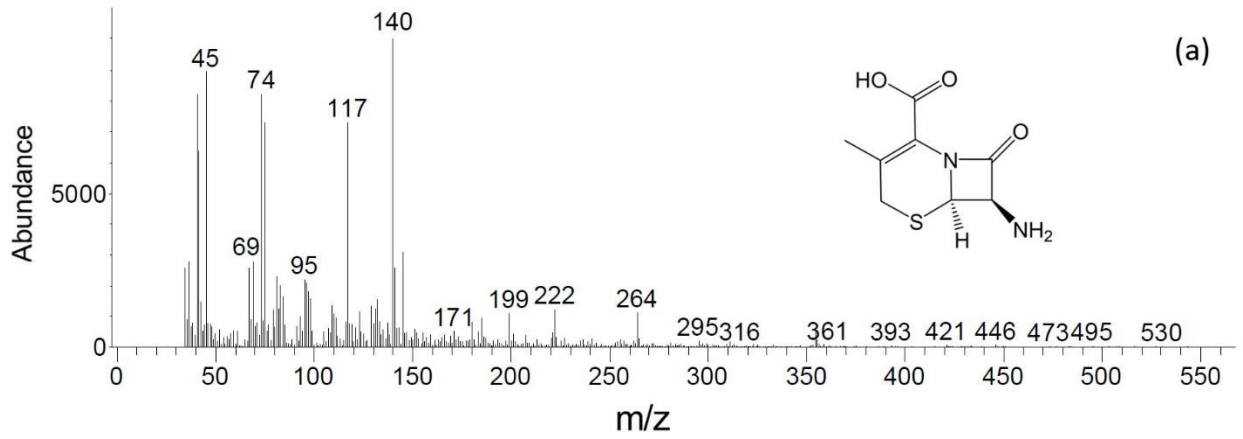
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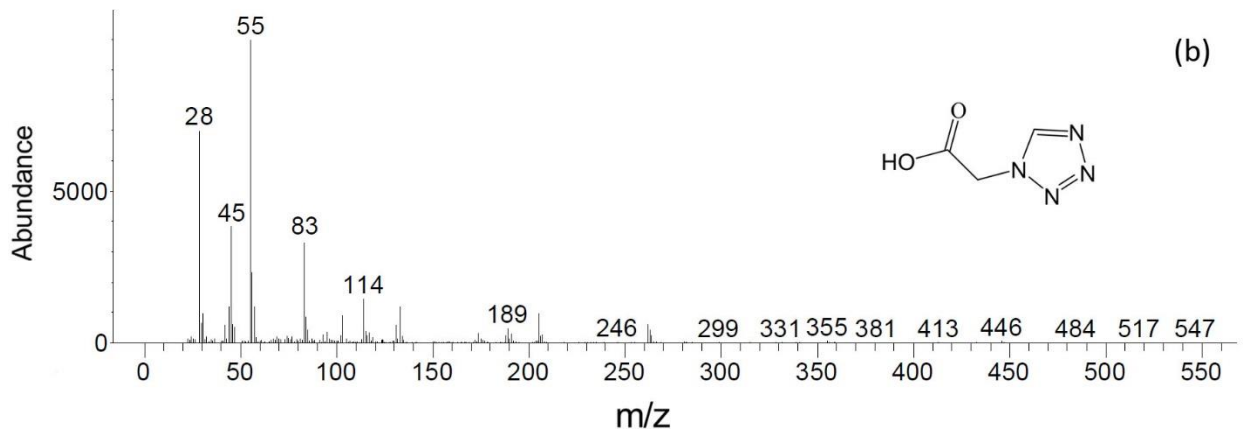
Fig. A3. Cyclic stability of DT-WO_{3-x}@MC nanohybrid for degradation of CFZ under visible light irradiation; Experimental condition: [photocatalyst] = 0.6 g L⁻¹, [CFZ] = 3 mM, pH = 6.3 and temperature = 30±1 °C.

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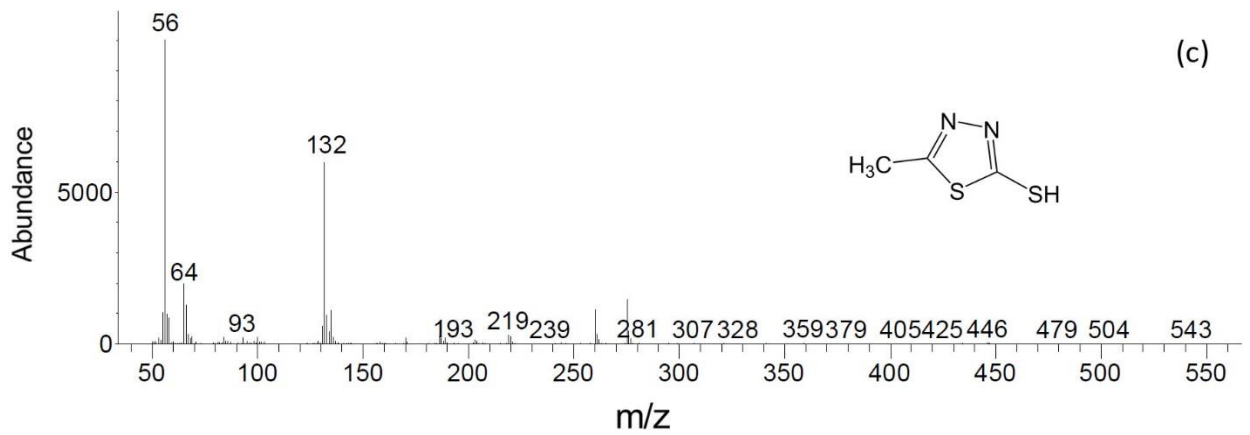
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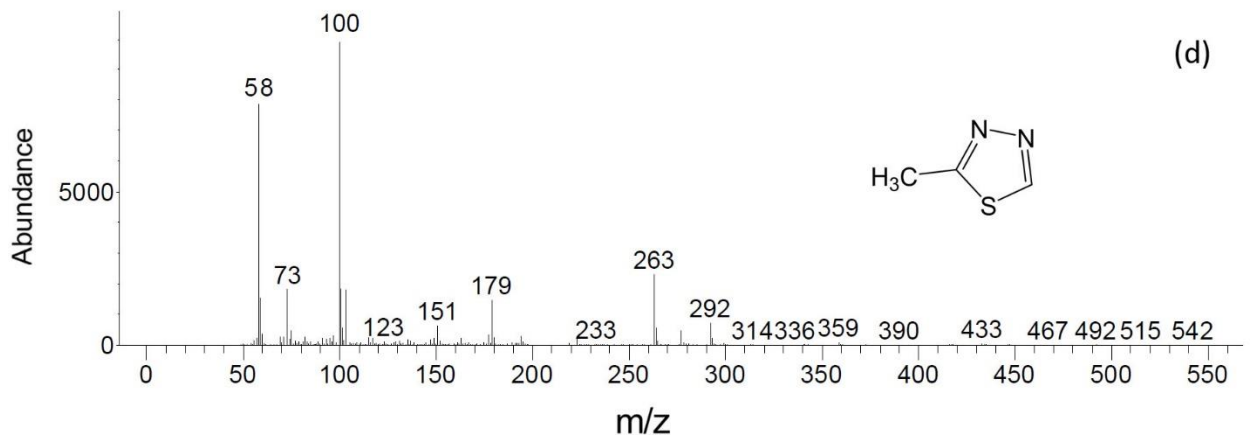


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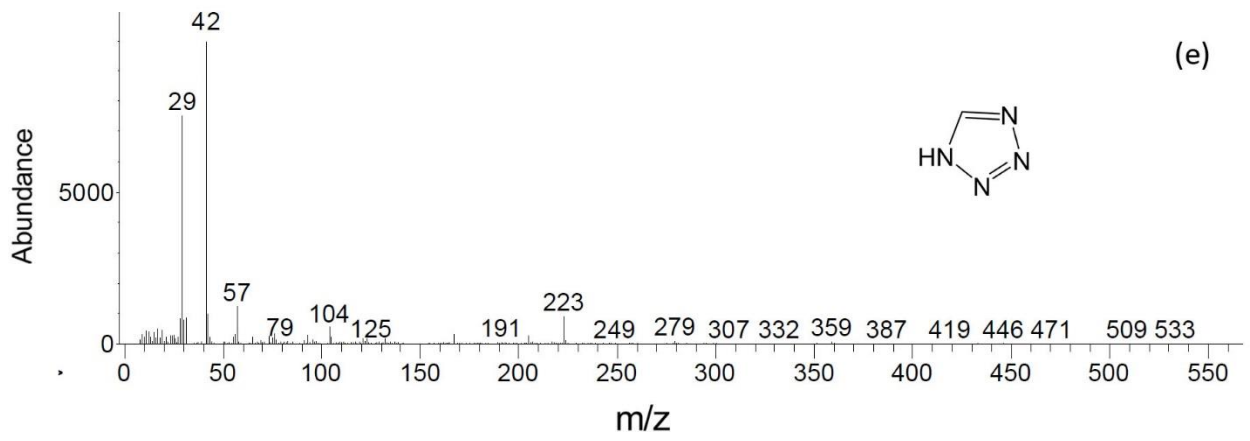
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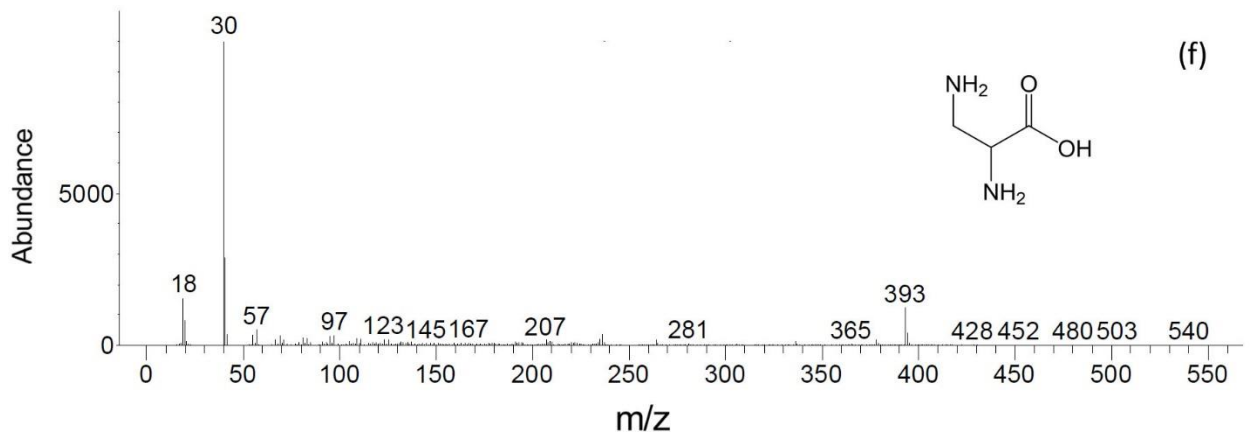
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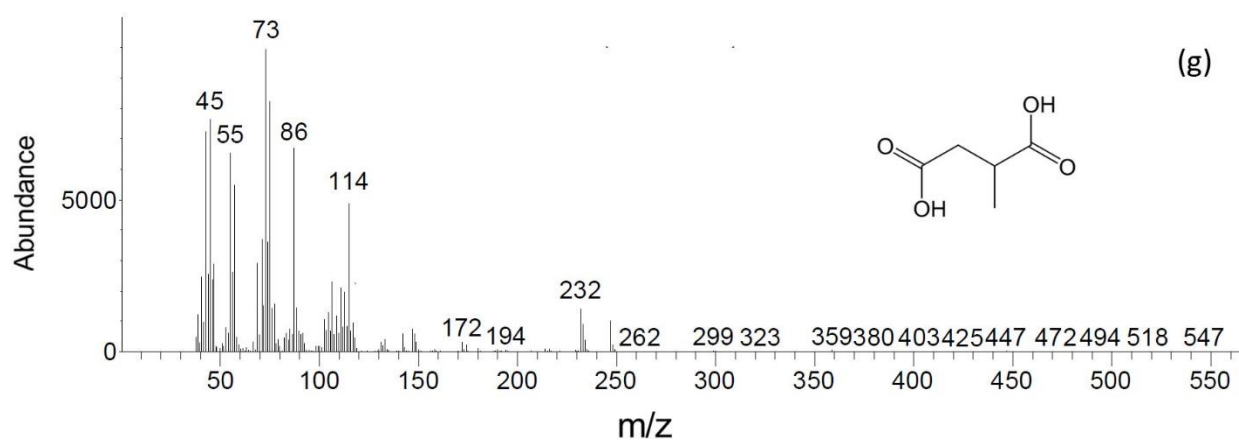
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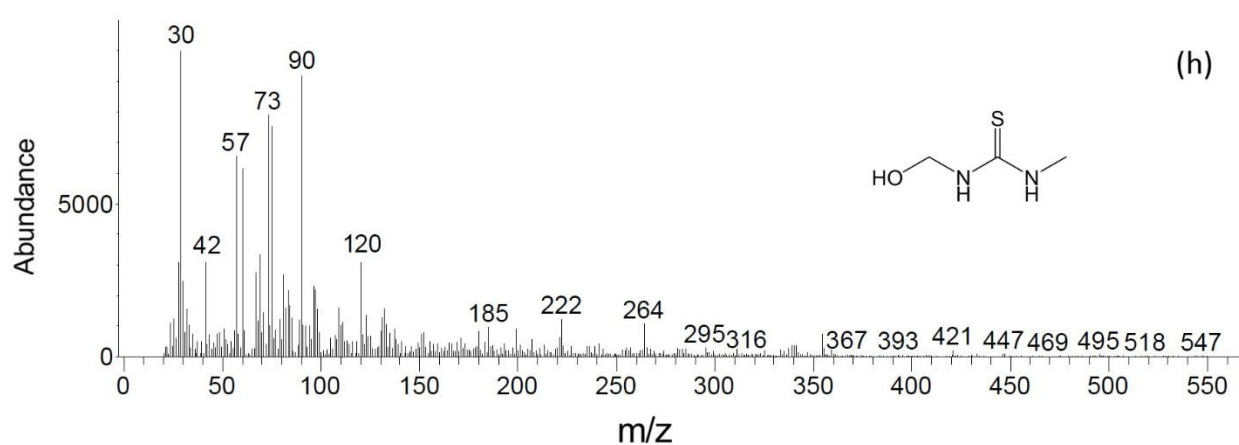
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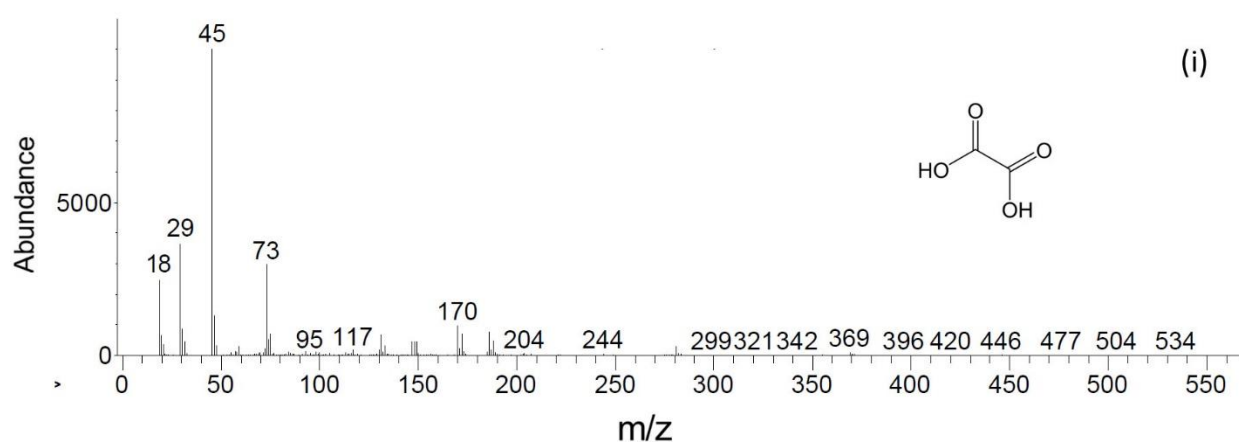
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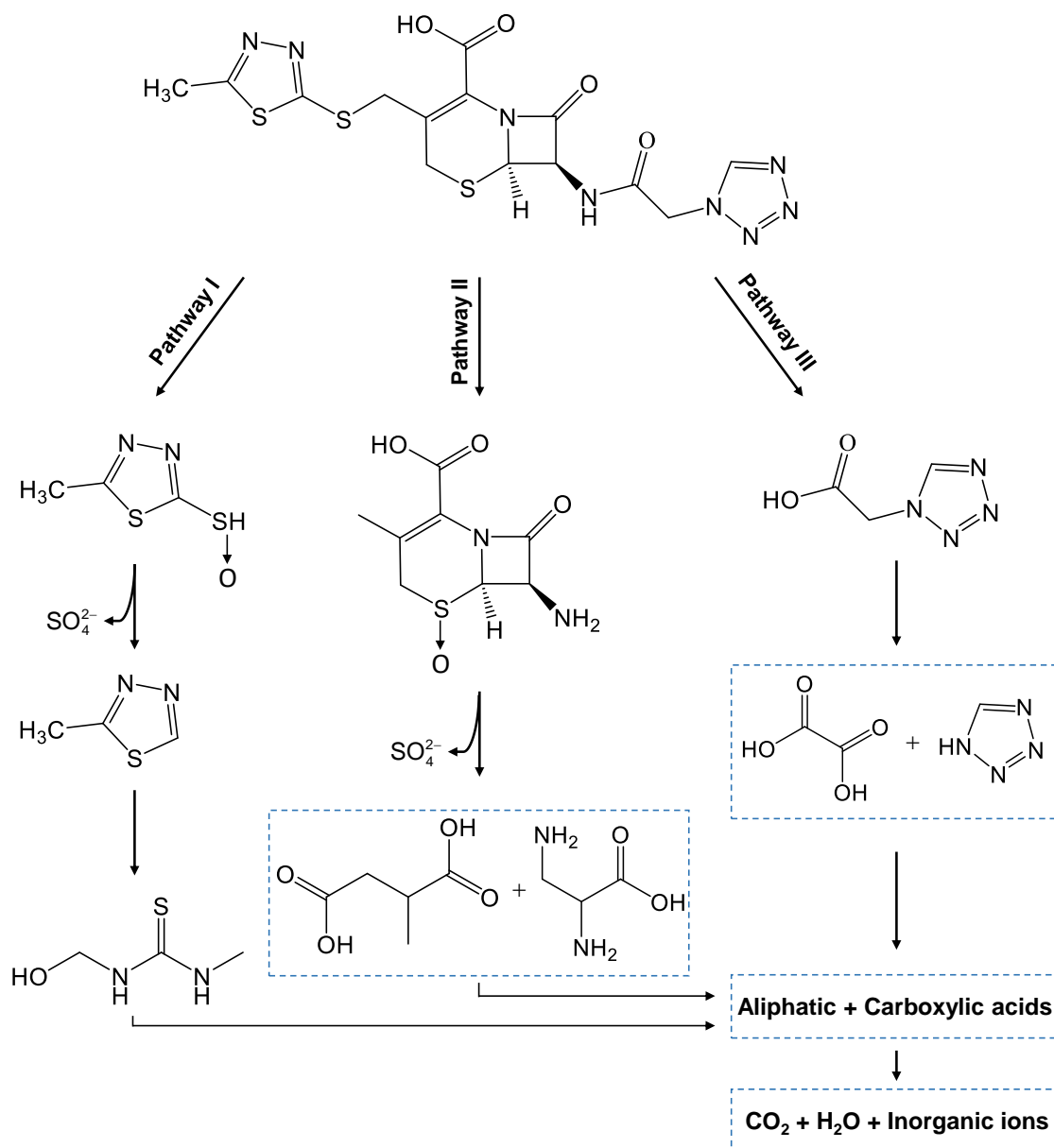
Fig. A4. GC-MS spectra of intermediates and their structures (a-i).

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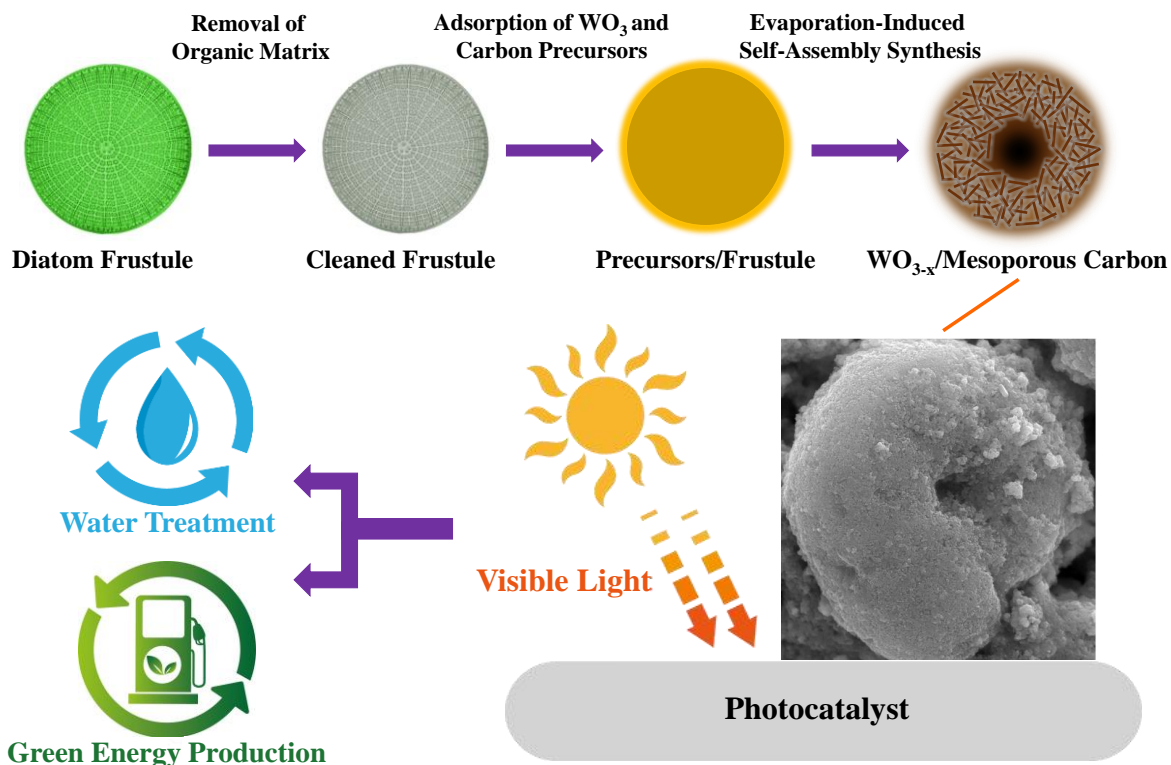


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Fig. A5. The proposed pathways for the photodegradation of CFZ using DT-WO_{3-x}@MC photocatalyst under visible light irradiation.

917 **Graphical Abstract**

918



920 Evaporation-induced self-assembly synthesis of $WO_{3-x}@MC$ nanohybrid using diatom
921 template for photocatalytic H_2 production and water treatment.

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924 **Research highlights**

- 925 • Synthesis of 3D reduced tungsten oxide@mesoporous carbon using diatom template.
- 926 • Superior photocatalytic properties of synthesized nanohybrid compared to pure WO_3 .
- 927 • Photocatalytic cefazolin degradation and H_2 production using synthesized nanohybrids.
- 928 • Reusability of tungsten oxide@mesoporous carbon photocatalyst after five cycles.
- 929 • A possible pathway for cefazolin photodegradation.

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