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Version of record available online at
<https://doi.org/10.1080/15376494.2021.1963020>



Energy absorption analysis of aramid composite during blunt projectile impact.

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Abstract

In this work, a numerical model for aramid composites is experimentally validated in terms of permanent deformation, energy absorption and damage mechanisms for ballistic applications. Novel experimental and numerical results of non-perforating ballistic impacts with blunt projectiles are presented. The resistance forces and absorption energy by the specimen are measured for different impact velocities. A post-mortem analysis of the failure mechanisms is performed using computed tomography and a profilometer device. The numerical model is used to analyse the influence of impactor mass and impact velocity below the ballistic limit.

Keywords: Aramid composite, blunt projectile, resistance force, delamination, impact, finite element, energy absorption

1. Introduction

The woven aramid prepreg materials are widely used to construct protective structures, such as combat helmets and vehicle armor, due to their combination of high specific stiffness and strength [1]. The analysis of these structures under ballistic impact at high impact velocities involves perforation and/or non-perforation tests. Generally, tests performed by manufacturers and developed according to standards for the design of protections are focused on non-perforation. The main standard for non-perforation analysis of armour, like combat helmets, is NIJ 0101.06 [2], which involves using 9×19 mm FMJ (Full Metal Jacketed) projectiles. The trauma generated during the impact event is measured using Roma n°1 ballistic plasticine. However, the main studies related to the analysis of laminated aramid's mechanical behaviour have been carried out on perforation [3–6]. The lack of studies on non-perforation testing difficult the development of numerical tools for protections designing, since validation should be carried out in these conditions.

44 Tan et al. [3] developed experimental tests and numerical simulations with projectiles of different nose
45 shape to obtain the ballistic limit, the absorption energy and the penetration failure mechanisms. Similarly,
46 Bresciani et al. [4] developed an experimental and numerical study of perforation impacts with blunt
47 projectiles in which the impact angle and the subsequent delamination after impact were analysed. Post-
48 mortem analysis was performed by cross-sectioning the specimens. Li et al.[5] studied the influence of the
49 blunt projectile material on the deflection of aramid composite plates. They used steel and aluminium foam
50 blunt projectiles. The main mode of impact kinetic energy dissipation was by deforming the aluminium
51 foam projectile. However, energy dissipation was carried out in the aramid plate using the steel projectile.
52 Wang et al. [6] experimentally analysed the ballistic behaviour of Kevlar/epoxy specimens of different
53 thicknesses using projectiles of different geometries, including the blunt-type projectile. From each test, the
54 influence of the projectile geometry in terms of absorbed energy was analysed and related to the degree of
55 damage induced to the specimens. Sikarwar et al [7] in their experimental work on the impact of 9 mm FMJ
56 projectiles on laminates of different Kevlar/epoxy stacking orientations performed an analytical study of the
57 energy balance absorbed during impact.

58 Due to the high porosity of aramid laminates, the most commonly used methods for damage detection in
59 this type of composite material are the visual inspection of the specimen and cutting it in the half to evaluate
60 the morphology of the different failure modes [8-15]. However, Computed Tomography (CT) is a technique
61 that does not require the destruction of the specimen and allows knowing the extension of the delaminated
62 area or its qualitative study [16-20].

63 In a previous work [6], different technologies effectiveness to assess aramid composites after ballistic
64 impact was evaluated. Computed Tomography (CT) and force sensors were used to record the delamination
65 and impact force; the permanent residual deformation was measured with a profilometer device. The
66 method's effectiveness was evaluated for the impact of conical and blunt projectiles above the ballistic limit.

67 The present work focuses on the analysis of the aramid composite under high energy impact loads in non-
68 perforating conditions. This analysis is relevant for manufacturers because these conditions are common
69 during the service life of combat helmets. Non-perforation velocities can cause stiffness losses and reduce
70 its structural performance. For this study, an analysis of the resistant force of the material for different
71 impact energies was carried out. Moreover, a study of the remaining deformation in the material after
72 impact is performed by profilometry. Computerized tomography was used to analyse the damage in the
73 specimen after impact. This novel experimental methodology constituted the basis for the
74 calibration/validation of the numerical model. The numerical model allowed to analyse the internal
75 energies, the influence of projectile velocity on resistance force and the influence of the projectile mass for
76 impact velocities not achieved experimentally.

77 **2. Experimental set-up**

78 *2.1. Material plates*

79 The specimen consists of flat plates of aramid composite, in particular K129 fibres embedded in
80 Polyvinyl Butyral Phenolic matrix (PVB). Each layer presents a plain wave woven configuration of fibres.
81 The manufacturing of the plates consists of hand-made stacking of the different layers (woven fibres and
82 matrix), and then hot-pressing is applied in order to provide the material with the necessary cohesion to stiff
83 the plate. The dimensions of the plate are $130 \times 130 \text{ mm}^2$ with an areal density of 8.86 kg/m^2 , units
84 typically used in personal protection design to define the thickness of the sample.

85 2.2. Ballistic impact device

86 Ballistic impact tests are carried out using a pneumatic gas gun to launch a blunt projectile orthogonal
 87 to the composite plate. The diameter of the gas gun barrel is 13 mm, roughly equal to the projectile
 88 diameter, to avoid sabot and not include extra damage by the sabot in the specimens. The initial impact
 89 velocity (V_0) can be adjusted by changing the gas pressure in the tank, P_0 .

90 A set-up of laser sensor is used to measure the initial impact velocity of the projectile (V_0). They are
 91 fixed at the exit of the barrel, perpendicular to the projectile trajectory. The impact velocity is defined as
 92 $V_0 = \Delta X^{laser} / \Delta time$, where $\Delta time$ is the interval rise time when the projectile crosses from the first to the
 93 second laser beam, and ΔX^{laser} is the distance between both laser beams, $\Delta X^{laser} = 5\text{ cm}$.

94 A schematic description of the set-up and sensors position is described in Figure 1.

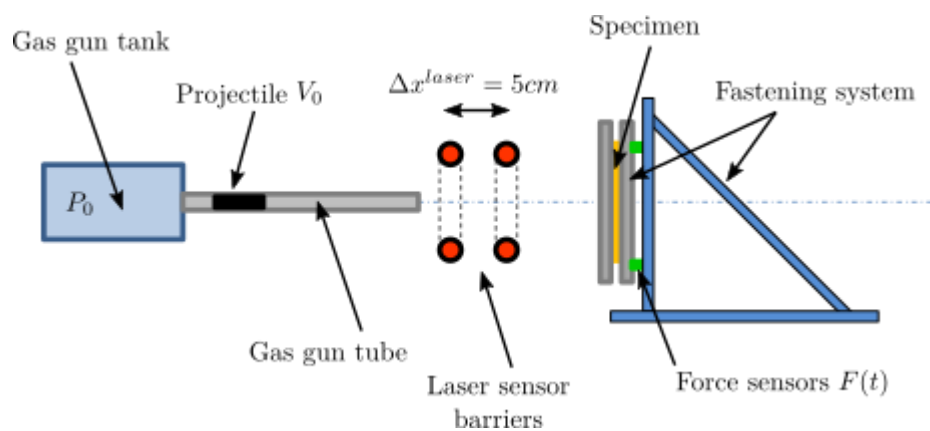


Figure 1. Experimental set-up of devices used on the impact test.

95 The composite plates ($130 \times 130\text{ mm}^2$) are embedded by two steel frames forming a fastening system,
 96 resulting in an effective area of $100 \times 100\text{ mm}^2$ (Figure 2a).

97 The projectile is made of heat-treated *maraging* steel with a hardness close to 640HV and a yield
 98 stress of $\sigma_y \approx 2\text{ GPa}$. Blunt nose shape projectile with a diameter of 13 mm, a length of 29 mm and a mass
 99 of 28.9 g. is used during all the tests performed [21] (Figure 2b).

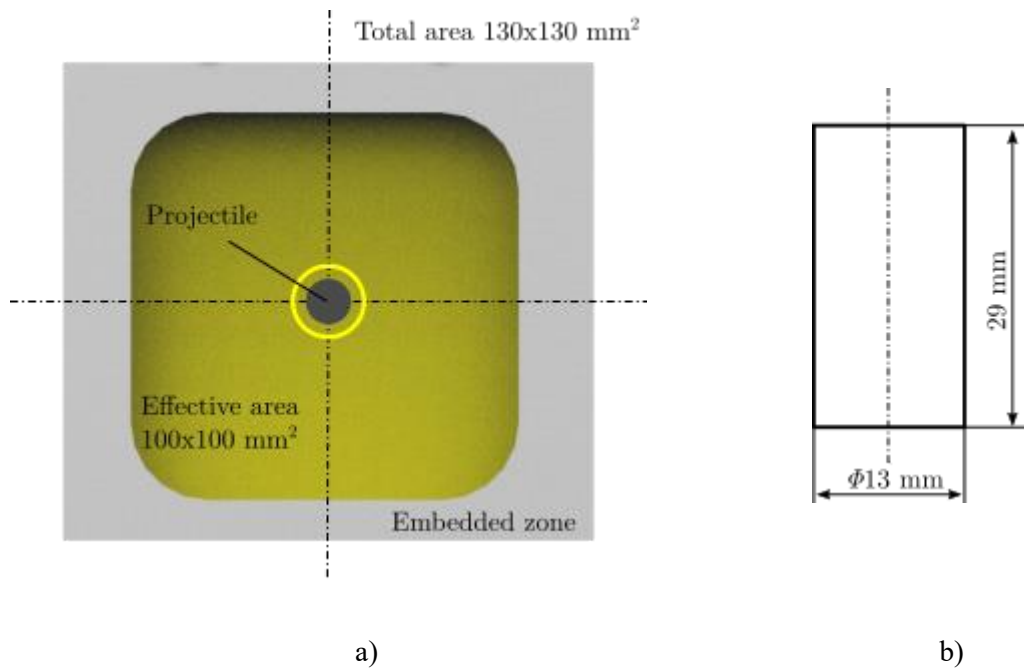


Figure 2. Dimensions of the target and the projectile used in tests. (a) Target geometry. (b) Projectile shape and dimensions

100

101 A force sensor set composed of four individual piezoelectric sensors is installed to measure the
 102 overall resistance force history $F(t)$. The sensors are fixed on the back face of the target holder and the
 103 fastening plate device, Fig. 1. The sensors (9011A Kistler) are used to measure only the uniaxial dynamic
 104 load along the impact direction. The maximum force on the specimen that can be measured is the sum of
 105 each sensor's individual and varies from 60 to 80 kN [20,21]. Each sensor has a natural frequency of 65 kHz,
 106 and it can measure a wide range of forces from 0 to 15 kN with a tolerance is ± 5 kN. Figure 3 shows the
 107 sensor location on the impact device. Similar studies about resistance force using a similar experimental
 108 device have been described in the literature when testing other materials [22,23]

109

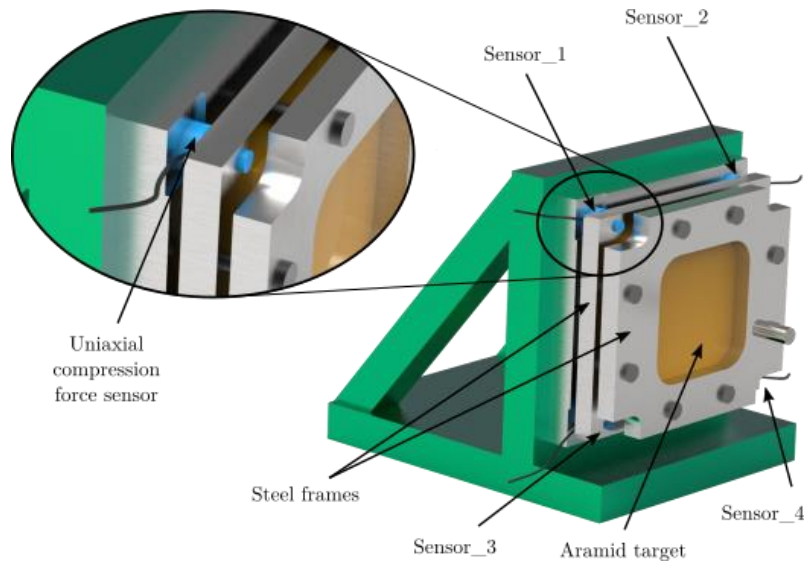


Figure 3. Fastening target plate and measurement system location.

110 In addition to the analysis of the mechanical behavior of the composite, a systematic study of the impacted
 111 plates has been performed to estimate the damage in the samples due to the impact event.

112 2.3. Post-mortem deformation (PBFD)

113 A *MahrSurf* model *CD 120* digital profilometer is used to obtain the permanent back-face
 114 deformation (PFBF) on plates (Figure 4). The profilometer uses a needle with a radius of $20\ \mu\text{m}$ and an arm
 115 of $350\ \text{mm}$. The precision obtained during measurements is related to the radius of the touch probe
 116 needle. The needle goes through the rear profile of the plate in the zenith plane. A set of coordinates of the profile
 117 points is obtained using *Mahr Easycontour* software, which allows to measure the maximum value of back
 118 face deformation of each tested sample. This method allows to obtain the permanent deformation (PFBF)
 119 induced in the specimen in a precise way for further comparison with numerical predictions.

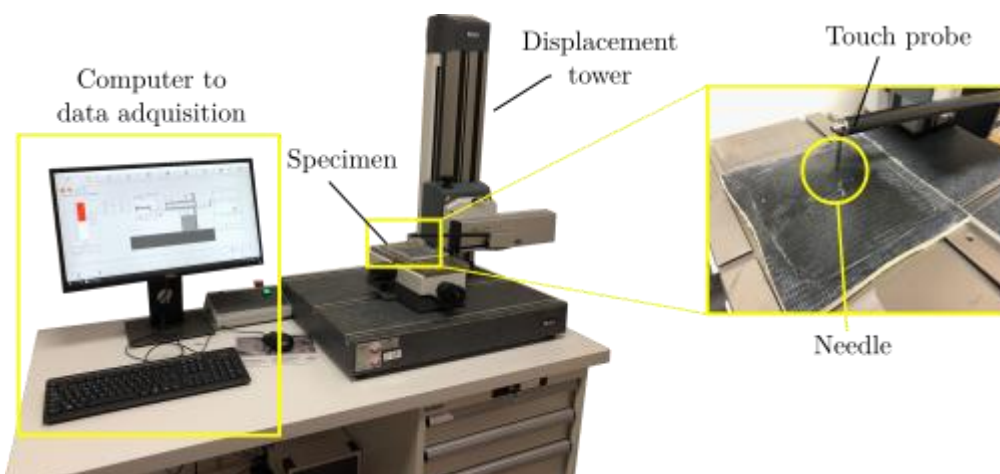


Figure 4. Profilometer device to measure the permanent back face deformation.

120

121 In addition to induced damage observed externally on specimens, internal damage occurs due to
122 delamination, shear failure or compression layers. In order to avoid introducing additional damage in the
123 specimen by cutting process that may affect the morphology of the failure modes observed, a non-
124 destructive damage analysis technique (X-ray-based computed tomography) is used. The following section
125 describes this technique, which allows qualitative analysis of internal damage in specimens.

126 *2.4. CT-Scan for tomography*

127 An analysis of the internal damage of the plates after impact is carried out by tomography of post-mortem
128 specimens. All Images are obtained by X-ray microtomography device (EasyTom Nano, Rx Solutions) with
129 a minimum X-ray beam power of 160 kW and a precision of 0.5 μ m. The analysis of each plate requires
130 around 4 hours. AVIZO and X-Act 2.0 software for processing images from tomography are used. With this
131 technique, it is possible to analyse internal damage induced in the plates due to impact without the necessity
132 of destroying the specimen, i.e., with middle cutting.

133 **3. Numerical simulation**

134 The numerical model is developed in the FEM software Abaqus/Explicit 6.17. Aramid composite
135 material behaviour is defined through a user subroutine VUMAT that considers the anisotropy of the
136 composite material.

137 The numerical model is composed of different parts: the aramid plate, the blunt projectile, the
138 auxiliary steel frames where the specimens are embedded, cylinder-shaped sensors, and different screws to
139 fasten and fix all the parts [22]. The material models, geometries, boundary conditions and mesh, are
140 defined below. A quarter-symmetry mesh is developed to reduce the computational time.

141 *3.1. Aramid composite plates*

142 *3.1.1. Mechanical behaviour of aramid/PVB laminates*

143 The mechanical behaviour of the aramid composite plate is assumed as an orthotropic elastic material
144 up to failure. This approach is widely used for composite models using impact problems [9,10,16,22-24].
145 The mechanical properties of the aramid composite used in this study are shown in Table 2.

146 The failure model distinguishes between intra-laminar failure and inter-laminar failure or
147 delamination.

148 • Intra-laminar failure

149 The mechanical behaviour of each layer of aramid composite is assumed as linear elastic until failure.
150 The failure is predicted using a modification of the Hou et al. failure criteria [15], equations 1-3, through a
151 VUMAT user subroutine. The criterion is formulated based on in-plane and out-of-plane stress distribution.
152 This model has been previously successfully applied in previous work [17]. The stress distribution for in-
153 plane and out-of-plane failure modes is shown in Figure 5.

154

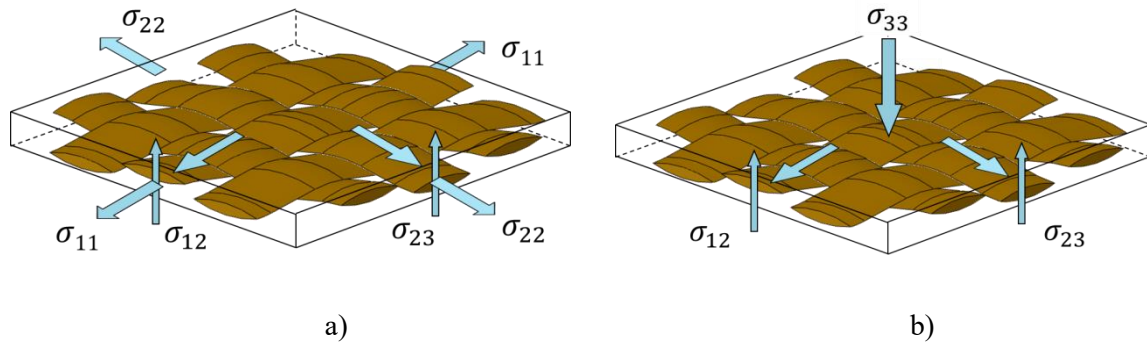


Figure 5. Stress components distribution involved in each failure mode. a) In plane tension distribution. b) Out of plane tension distribution.

155 The failure model is described in the following part depending on the loading direction.

156 - In-Plane failure modes

157 ***Fibre failure in direction 1***

$$d_{f1} = \left(\frac{\sigma_{11}}{X_{1T}}\right)^2 + \left(\frac{\sigma_{12}}{S_{12}}\right)^2 + \left(\frac{\sigma_{13}}{S_{13}}\right)^2 \quad 1)$$

158

159 ***Fibre failure in direction 2***

$$d_{f2} = \left(\frac{\sigma_{22}}{X_{2T}}\right)^2 + \left(\frac{\sigma_{12}}{S_{12}}\right)^2 + \left(\frac{\sigma_{23}}{S_{23}}\right)^2 \quad 2)$$

160 Where σ_{11} , σ_{22} correspond with stresses in fibre direction (1 and 2) respectively; σ_{12} , σ_{23} , and σ_{13}
 161 are the shear stresses; X_{1T} , X_{2T} threshold stresses in the fibre direction and, finally, S_{12} , S_{23} and S_{13} are the
 162 transverse shear strengths.

163 - Out-of-Plane failure modes

164 ***Failure in direction 3***

$$d_{f3} = \left(\frac{\sigma_{33}}{Z_c}\right)^2 + \left(\frac{\sigma_{13}}{S_{13}}\right)^2 + \left(\frac{\sigma_{23}}{S_{23}}\right)^2 \quad 3)$$

165 Where σ_{33} and Z_c are the stress and strength in through-thickness direction, σ_{23} , and σ_{13} are the shear
 166 stresses and S_{23} and S_{13} are the transverse shear strengths.

167 Matrix failure is not considered due to the low resin content in the composite, about 18%. The main
 168 failure mode is the fibre failure along each direction (eqs. 1-3). All mechanical properties of aramid
 169 composite are presented in table 2.

E_1 (GPa)	E_2 (GPa)	E_3 (GPa)	G_{12} (GPa)	G_{13} (GPa)	G_{23} (GPa)	ν_{12} (-)	ν_{13} (-)	ν_{23} (-)	ρ (kg/m ³)
22.0	22.0	9.0	0.77	5.34	5.34	0.25	0.33	0.33	1230
S_{1t} (MPa)	S_{1c} (MPa)	S_{2t} (MPa)	S_{2c} (MPa)	S_{3t} (MPa)	S_{3c} (MPa)	S_{12} (MPa)	S_{13} (MPa)	S_{23} (MPa)	
800	80	800	80	1200	1200	77	898	898	

Table 2. Mechanical properties of aramid composite [26].

170

171 Failure is considered when any damage variable " d_i " ($i = f1, f2, f3$) is equal to 1. At this time, the
 172 stresses involved in damage criteria (eqs. 1-3) are set to zero. Furthermore, it produces large deformations
 173 and distorted elements in the model, which do not contribute to the strength and stiffness of the laminate,
 174 but can produce loss of convergence during simulations and instability. Thus, it is necessary to include
 175 deletion criteria of distorted elements based on maximum deformation using a VUMAT subroutine. Before
 176 calculating each time increment, the strains in different directions were evaluated by comparing them to a
 177 critical value. This method for simulating failure has been reported in previous works [13,17,24,29].

178 • Inter-laminar failure

179 Interlaminar failure of the composite is modelled using cohesive surfaces. The damage initiation
 180 criteria and its evolution define cohesive interlaminar behaviour. The use of cohesive surfaces instead of
 181 cohesive elements presents some advantages: it improves the computational cost without adding mass or
 182 stiffness to the model.

183 The model used in this work is based on the traction-separation law, and it assumes a linear elastic
 184 behaviour, followed by damage initiation criteria and a damage evolution law describing the cohesive
 185 stiffness degradation mode. The damage evolution is described as a non-linear evolution law once the
 186 initiation criteria are reached.

187 The damage initiation criterion is based on stresses in a quadratic form (eq. 4)

$$\left(\frac{t_n}{t_n^0}\right)^2 + \left(\frac{t_s}{t_s^0}\right)^2 + \left(\frac{t_t}{t_t^0}\right)^2 \geq 1 \quad (4)$$

188 Where (t_n^0) is the normal threshold stress and t_s^0, t_t^0 are the shear strengths, respectively, and their
 189 values are listed in table 3.

190

G_n^c (J/mm ²)	$G_t^c = G_s^c$ (J/mm ²)	t_n^0 (MPa)	$t_t^0 = t_s^0$ (MPa)	α
0.24	0.47	34.5	9.0	1

191

Table 3. Cohesive properties used in the numerical model.

192 The damage evolution law is based on mix-mode fracture energy. That energy is the area under the traction-
 193 separation curve [30]. The damage evolution criterion implemented in this work is based on a potential law
 194 based on energies (eq. 5)

$$\left(\frac{G_n}{G_n^c}\right)^\alpha + \left(\frac{G_s}{G_s^c}\right)^\alpha + \left(\frac{G_t}{G_t^c}\right)^\alpha = 1 \quad 5)$$

195 where G_n , G_s and G_t are, respectively, the released rate energies in normal and shear directions, G_n^c , G_s^c and
 196 G_t^c are the corresponding critical values and, α is a parameter model.

197 3.1.2. Modelling of aramid/PVB laminates

198
 199 The plate is meshed with 8-node hexahedral elements of with reduced integration (C3D8R). Each plate
 200 contains 8952 elements with 18234 nodes. One element per layer through-thickness is used to mesh the
 201 plate. The mesh structure is divided into a central area and the rest of the plate. The central area dimensions
 202 are $15 \times 15 \text{ mm}^2$ -twice the projectile diameter-; the mesh is refined and structured in this area to define the
 203 damage zone with precision. The element size with an aspect ratio of 1.62 is used. Out of this region, the
 204 element size increases up to 1 mm, with an unstructured mesh. (Figure 6). A mesh sensitivity analysis is
 205 carried out to minimise the error in the results obtained and the computational time.

206 3.2. Projectile

207 The projectile is modelled as a linear elastic solid because no plastic deformation is observed after impact
 208 tests. This assessment allows to reduce the computational time during simulations. The dimensions of the
 209 projectile are the same as used during experiments. An equivalent density is defined to provide the projectile
 210 with the exact mass of 28.9 g. The projectile density is set to $\rho_{eqv} = 7744 \text{ kg/m}^3$. The mesh of the
 211 projectile consists of 2400 elements and 2944 nodes for the quarter of geometry simulated.

212 3.3. Target holder

213 According to the experimental set-up, the target holder is modelled as two steel frames (front and rear
 214 holders). Each frame dimensions are $190 \times 180 \times 15 \text{ mm}^3$ with an effective area of $100 \times 100 \text{ mm}^2$. The
 215 target holder is modelled as a linear elastic solid with 8-nodes hexahedral elements and reduced integration.
 216 The mesh density varied from fine to coarser from the sensor zone to the outer periphery of the for accurate
 217 impact force measurement results.

218 To ensure that both steel frames do not separate and obtain a correct record of the evolution of the resistant
 219 force during the impact, the steel frames were joined using two screws modelled as linear elastic. The
 220 screws were fastened to the frames, modelled by TIE type joints. The dimensions of the screw are $\Phi =$
 221 10 mm and $h = 13 \text{ mm}$, and are meshed with 8-nodes hexahedral elements with reduced interaction.

222 3.4. Sensors modelling and description.

223 The sensor is modelled as linear elastic behaviour because no plastic deformations are observed after
 224 impacts. The dimensions of the sensor are $\Phi = 5 \text{ mm}$ and $h = 5 \text{ mm}$. They are meshed with 8-nodes
 225 hexahedral elements with reduced integration.

226 In the experimental case, the cylinder-shaped sensors (in blue in Figure 3) are fasten to the supporting rig
 227 (in green in Figure 3). Therefore, these sensors define 4 point boundary conditions for the steel frame.
 228 According to the impact energy, steel frame stiffness, etc., it could affect the reaction forces appearing
 229 during the impact in a general case, so they modelled according to the experience depicted in previous
 230 papers [22,23]

231 Numerically, sensors are modelled in a simplified way as a cylindrical tip on the steel frame at the sensor
 232 location. The end of the sensor connects to the green rig, where the reaction force is measured, is modelled
 233 as an encastred surface. The 4 reaction forces were experimentally measured and compared at the
 234 corresponding numerical at the same points.

235 Moreover, numerical screws are modelled as simplified steel cylinders of equivalent diameter and stiffness,
 236 just connecting rear and front frames to keep the specimen supported in the perimeter. A friction coefficient
 237 of 0.2 between specimen and frame has been considered. The qualitative comparison of experimental and
 238 numerical specimen after impact (see Figure 9) show a very similar deformed shape.

239 The numerical set-up for impact testing in ABAQUS/Explicit can be observed in Figure 6.

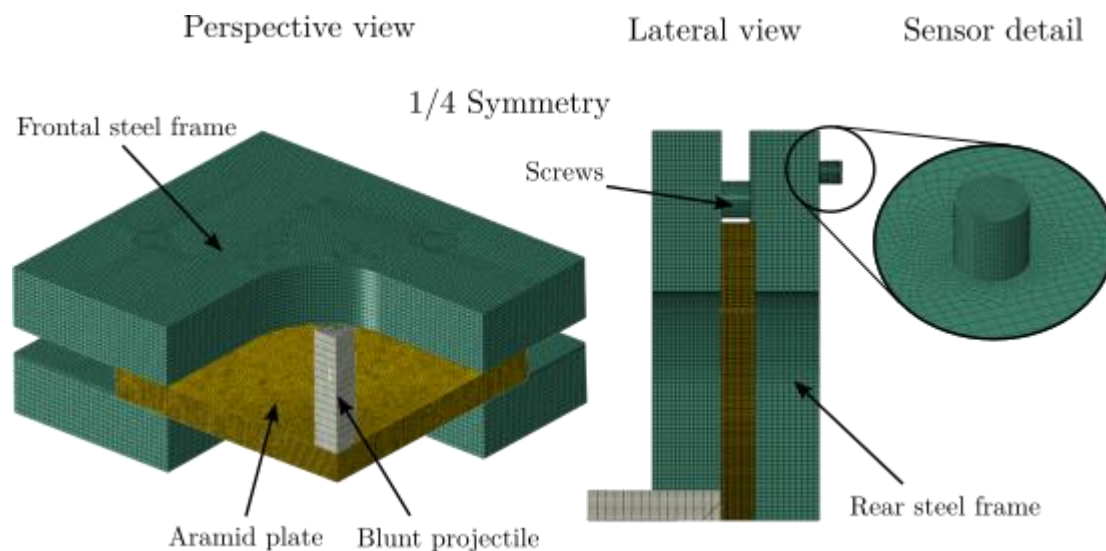


Figure 6. FEM model developed for impact test.

240 The contact between the plate and projectile and steel frames is defined with penalty contact algorithm and
 241 hard contact model [18]. The “hard contact” model allows adjusting the stiffness generated by the “penalty
 242 contact algorithm” to minimise penetration without adversely affecting the time increment. Concerning
 243 friction, a frictional coefficient equal to $\mu = 0.2$ is assumed. This is a typical value used in aramid impact
 244 problems [17,29].

245 **4. Experimental and numerical results**

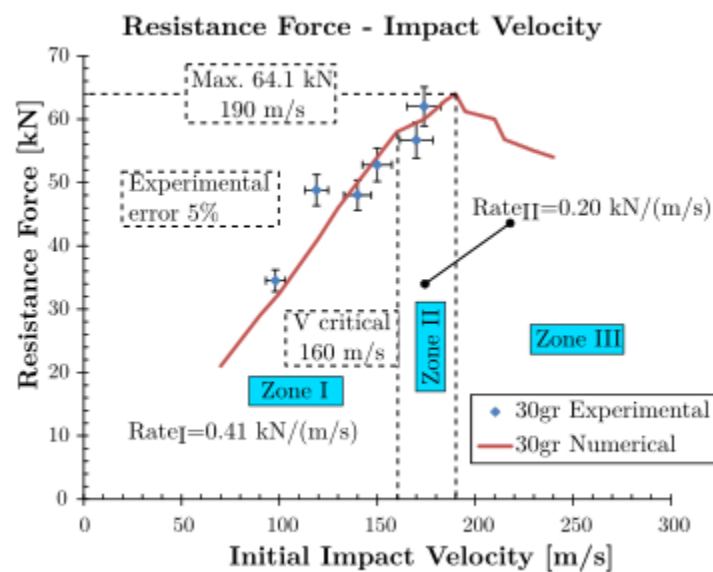
246
 247 The numerical model was validated comparing with experimental results. Subsequently, an analysis of the
 248 influence of the projectile mass (m_p) on the impact of aramid plates is carried out.

249 A total of 6 tests are performed at different initial impact velocities (V_0) in the range of 98 m/s to 174 m/s
 250 (*non-perforation configuration*) to validate the numerical model presented in this work. In each test, the
 251 resistance force is measured through the piezoelectric sensors. Then, remarkable damage is first observed
 252 by visual inspection of the specimens. Finally, permanent back-face deformation is measured using the
 253 profilometer to obtain the residual deformation generated due to different initial impact velocities.

254

255 4.1. Resistance force vs Impact velocity analysis

256 Figure 7 shows the results between the experimental test and numerical simulations for impact forces. Six
 257 experimental tests are performed within a range of impact velocities, from 98 m/s to 178 m/s.



258

259 **Figure 7.** Resistance force comparison between experimental and numerical results.

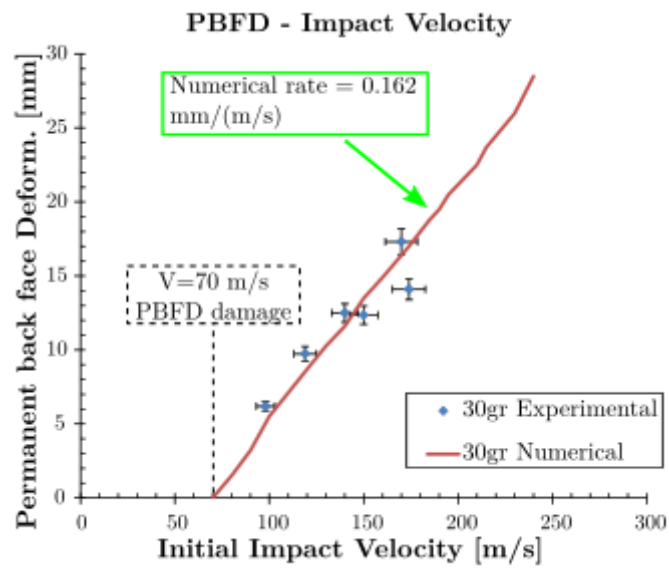
260

261 It can be observed that the numerical model provides a good correlation with experiments. Three zones are
 262 observed in Figure 7. In zone I, with a rate (Force/Impact velocity) of $0.41 \text{ kN}/(\text{m/s})$, both the experimental
 263 and numerical impact force show an increasing linear trend with impact velocity. A transition region (zone II)
 264 is found between 160 m/s and 190 m/s because the first few layers of the laminate are broken. In this zone
 265 the rate decreases by half, $0.20 \text{ kN}/(\text{m/s})$. This zone is delimited by the maximum peak force, 64.1 kN , for an
 266 impact velocity of 190 m/s . Over this impact velocity, a decreasing resistance force is observed (zone III).

267 4.2. Permanent Back Face Deformation (Pbfd)

268 The permanent Back Face Deformation (Pbfd), after the spring back effect, is brought along the plate's
 269 permanent deformation due to the absorption of elastic strain energy. Pbfd is obtained using the
 270 profilometer device described previously. A comparison of the Pbfd between experiments and numerical
 271 results is shown in Figure 8. A critical impact velocity is found at 70 m/s which corresponds to non-Pbfd

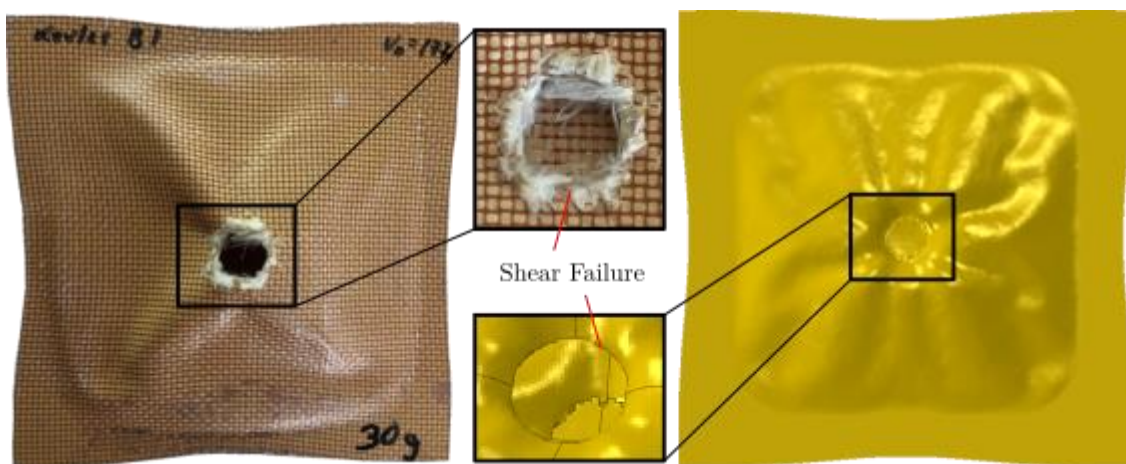
272 observation because the impact energy is dissipated by the cohesive interlayer damage. Above 70 m/s, a
 273 linear correlation between PBF D and the initial impact velocity is observed.



274

275 **Figure 8.** PBF D comparison between experimental and numerical results.

276 With a focus on analysing the external aspect of samples after impact, shrinkage phenomenon is caused,
 277 especially for high impact velocities. For illustrative purposes, experimental and numerical post-mortem
 278 specimens for 174 m/s are shown in Figure 9. The shrinkage, delamination and failure of the fibres are the
 279 main mechanisms that can be observed. The shrinkage mechanism is more noteworthy as the impact
 280 velocity increases.



281 **Figure 9.** Qualitative comparison (experimental and numerical) of shrinkage and damage
 282 induced on samples.

281

282 The experimental data are summarized in the Table 4.

<i>Impact velocity</i> [m/s]	<i>Impact energy</i> [J]	<i>Resistance Force</i> [kN]	<i>PBFD</i> [mm]
98	144.06	34.5	6.2
119	212.415	48.8	9.74
140	294	48	12.5
150	337.5	52.8	12.35
170	433.5	56.68	17.31
174	454.14	62	14.1

Table 4. Resume of experimental test carried out.

283

284 *4.3. Failure mechanisms and experimental observations*

285 Computerized tomography allows to observe internal damage in tested samples with high quality without
 286 the need to break them. The specimen is focused on the impact zone to observe the local damage produced.
 287 Quantifying the delaminated surface is complex using this technique if the complete specimen is not
 288 analysed. However, on visual inspection it has been observed that the most relevant damage occurs at high
 289 impact velocities where failure of the first layers of the material occurs. At low impact velocities (~ 100
 290 m/s), damage is limited to the local indentations in the impact zone.

291 For this reason, the computed tomography tool helps analysing the morphology of the damage induced in
 292 the specimen. The CT tests were carried out at the LEM3 of the University of Loraine. Figure 10 shows a
 293 comparison between the experimental specimen using this technique and the same numerical case for the
 294 most critical case analysed, impact velocity equal to 174 m/s. A good correlation is observed between
 295 results observed during experiments and numerical models, which correctly reproduces the failure modes
 296 and the final shape of plates after impact. In the impact zone, a through-thickness compaction layer occurs
 297 due to the nose shape projectile. Around this area, delamination increase being more notable at first layers,
 298 where shear failure is produced.

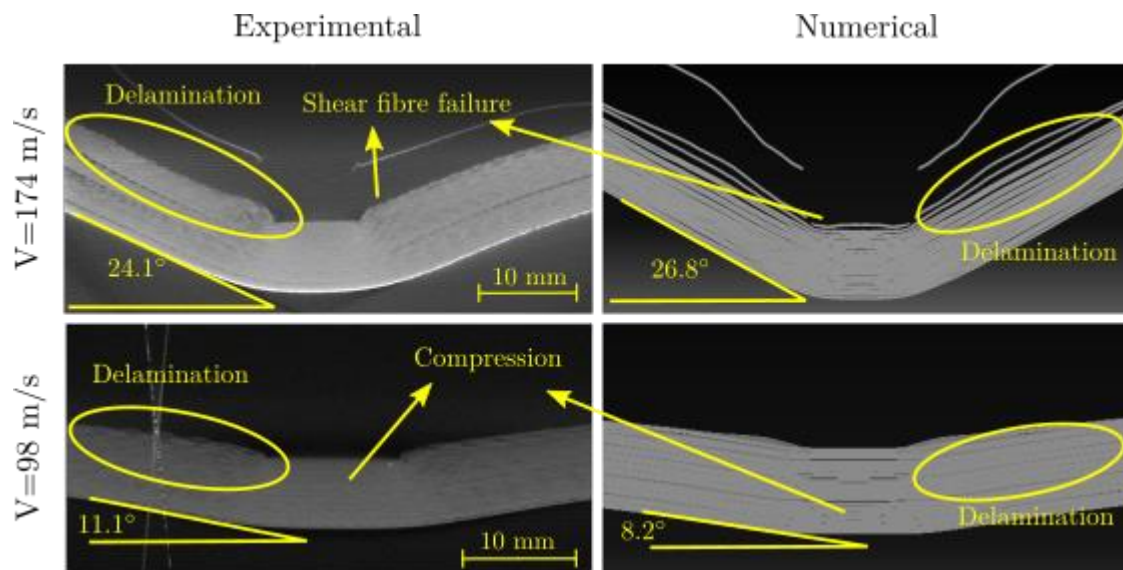


Figure 10. Failure mechanisms after impact using CT

299 *4.4. Energy balance based on numerical simulations.*

300 The internal mechanisms involve different absorption energies that can play relevant roles during the
 301 impact. For the sake of analysing the energy balance, only the numerical model can be used. Thus, the total
 302 balance energy total of the composite is based on the internal energy, the projectile kinetic energy, and other
 303 dissipated energies like frictions. The most important term of the composite plate is the internal energy,
 304 which is directly related to elastic energy, cohesive energy, and artificial energy. The energy balance is
 305 expressed as follows (Eq. 6-7):

$$E_{total} = E_{impact} = E_{inter} + E_{kin} + E_{dissip} \quad 6)$$

306 where

$$E_{inter} = E_{elastic} + E_{cohesiv} + E_{artif} \quad 7)$$

307

308 Figure 11a shows the energy balance for all impact cases where the energy absorbed by the cohesive
 309 damage, the internal energy, the elastic energy, and the artificial strain energy are plotted.

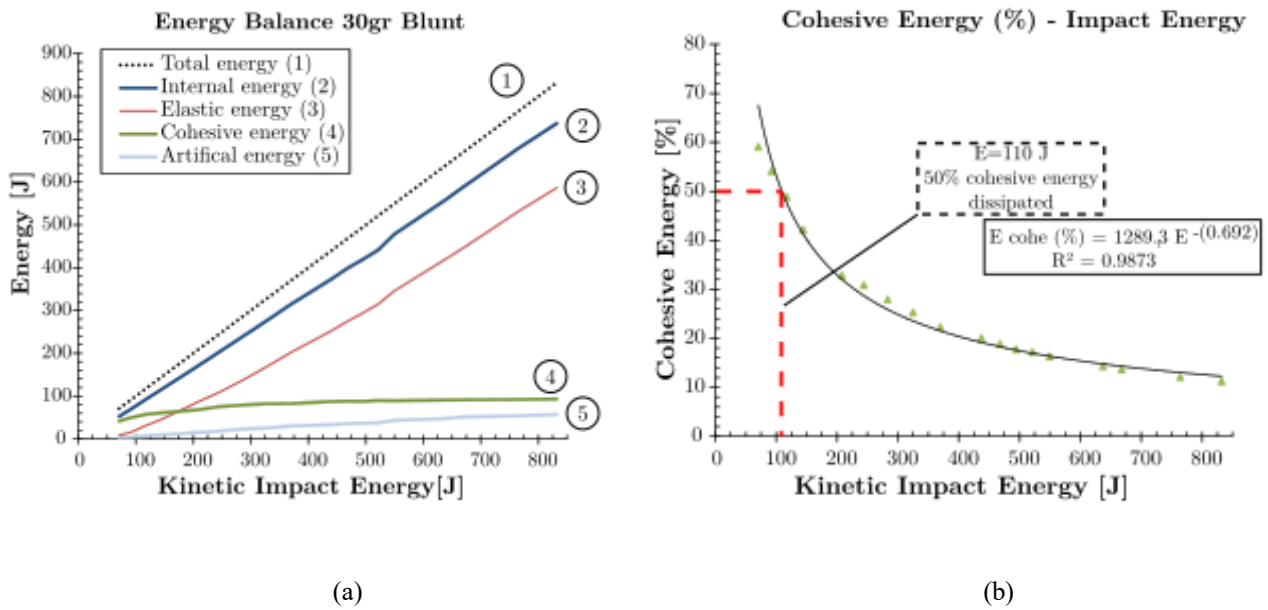


Figure 11. The energy dissipated by damage related to impact energy. a) Complete energy balance. b) Cohesive energy study.

310

311 Increasing energies are observed with the impact energy except for the energy absorbed by the cohesive and
 312 artificial energy, which remains steady. These energies are associated with the different global and local
 313 mechanisms that appear with impact velocity. Focusing on the cohesive energy, the energy dissipated
 314 through the plate delamination decreases asymptotically as the impact velocity increases, as shown in
 315 Figure 11b.

316 The data shown in the figures of the numerical model are summarized in Table 5.

<i>Impact Velocity [m/s]</i>	<i>Impact Energy [J]</i>	<i>Resistance Force [kN]</i>	<i>PBFD [mm]</i>	<i>Absorbed Energy [J]</i>	<i>Artificial Energy [J]</i>	<i>Elastic Energy [J]</i>	<i>Cohesive energy [J]</i>	<i>Damage area (mm²)</i>
26.00	10.00	14.00	0.00	5.20	0.27	0.32	4.68	178.38
70.00	70.81	21.00	0.00	52.58	2.76	8.01	41.81	6131.91
80.00	92.48	25.00	1.50	70.58	4.16	16.35	50.06	6694.50
90.00	117.05	29.00	3.20	92.00	6.80	31.00	57.20	7562.96
100.00	144.50	32.50	5.50	115.92	9.42	45.61	60.92	8342.97
120.00	208.08	41.20	8.70	170.54	15.40	86.68	68.30	9125.70
130.00	244.21	46.00	10.26	202.39	18.33	108.67	75.30	9280.84
140.00	283.22	50.00	11.60	236.95	22.79	135.20	78.94	9435.98
150.00	325.13	54.00	13.50	274.32	25.34	166.83	82.10	9713.82
160.00	369.92	58.00	14.90	315.01	30.16	202.41	82.40	9912.21
174.00	437.49	60.00	17.00	371.32	33.16	250.74	87.54	10386.82
180.00	468.18	61.60	18.00	398.72	35.57	275.11	87.99	10543.04
185.00	494.55	63.00	18.80	419.19	36.98	294.31	87.90	10530.07

190.00	521.65	64.00	19.50	441.84	37.33	315.16	89.55	10654.39
195.00	549.46	61.20	20.50	478.21	43.32	345.53	89.48	11179.28
210.00	637.25	60.00	22.50	557.42	47.32	419.11	91.08	11537.67
215.00	667.95	56.80	23.70	587.38	51.32	444.72	91.41	11822.54
230.00	764.41	55.00	26.00	678.00	54.20	531.00	92.30	11881.46
240.00	832.32	54.00	28.50	737.10	56.94	587.02	93.25	12083.63
250.00	903.13	53.00	31.00	806.86	61.35	649.64	95.97	12379.31

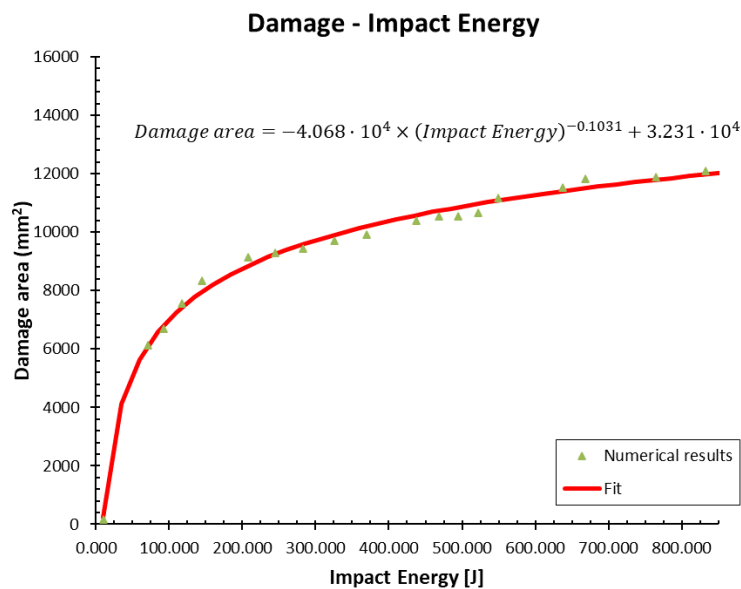
Table 5. Resume of numerical simulation developed.

317

318 According to the results of Table 5, the relationship between the damaged area and the impact energy can be
 319 expressed as:

$$Damage\ area\ (mm^2) = -4.068 \cdot 10^4 \times Impact\ Energy(J)^{-0.1031} + 3.231 \cdot 10^4 \quad (7)$$

320 the good agreement between numerical results and their fit is illustrated in Figure 12.



321

322

323

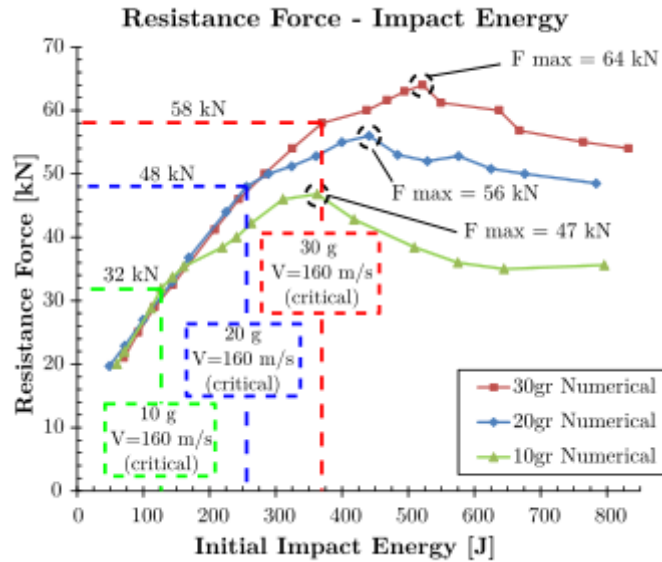
Figure 12. Damage area evolution with the impact energy.

324 *4.5. Analysis of the influence of projectile mass*

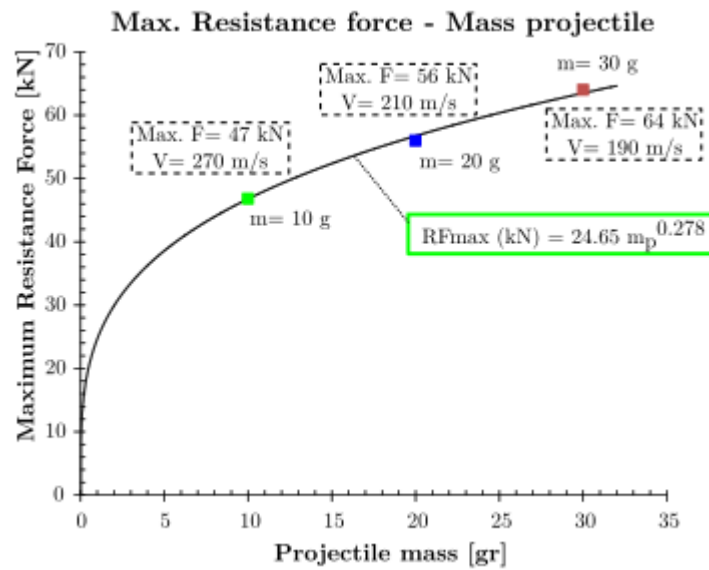
325 Since the numerical model provides accurate data with the experimental results because it has been
 326 calibrated and validated, the numerical model is used to analyse the effect of the mass impactor on the
 327 resistance force and PFBD, keeping the projectile diameter at different impact velocities. Two additional
 328 impactor masses are used: 20 g and 10 g.

329 Figure 13a provides information about the influence of impactor masses on the resistance force. All curves
 330 show a similar pattern over the impact energy. The change from zone I to zone II occurs at different impact

331 energies; however, it is fair to point out that it is found at the same impact velocity, 160 m/s. The peak
 332 resistance force is reached at different impact energy and velocities. The maximum values are 56 kN and 46
 333 kN for 20 g projectile and 10 g projectile, respectively. Figure 13b shows a linear trend between peak
 334 resistance force and projectile mass in the range of impact energy studied.



(a)



b)

Figure 13. Numerical mass influence analysis on resistance force. a) Resistance force vs Impact energy based on projectile mass influence b) Mass projectile influence on maximum resistance force.

335 The influence of impactor mass for the same projectile's diameters on Pbfd could be interesting for
 336 manufacturers and armour designers. According to our results, the impactor length/weight ratio has no
 337 influence on Pbfd for the same impact energy, Figure 14.

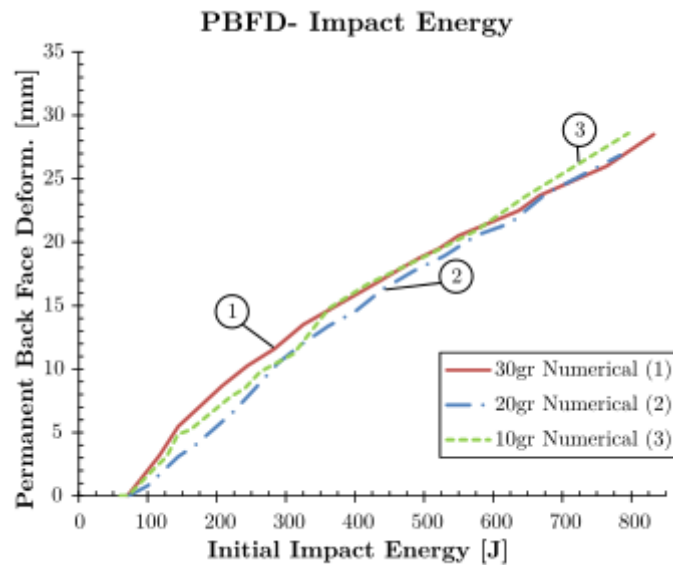


Figure 14. Permanent back face displacement for different projectile masses impacted.

338

339 5. Discussion

340

341 This paper presents a numerical model to predict the mechanical behaviour of aramid/PVB laminates for
 342 non-perforated impacts. The study has been carried out using a blunt projectile. The ballistic analysis of an
 343 impact structure should not be linked only to the ballistic limit, i.e. whether the projectile passes through the
 344 armour. Numerous parameters that are essential for the design of this type of structure for non-perforating
 345 operations should be analysed.

346 The relationship of the impact resistance to the failure mechanisms and the permanent deformation of the
 347 plate provide relevant information. It has been observed that before breaking the first layers of the material,
 348 delamination is the predominant energy absorption mechanism, leading to large deformation of the
 349 specimen. However, at high impact velocities, a more significant breakage of fibres is observed, and thus,
 350 the stiffness decreases and an increase of PFB D is found. This statement may be linked to the increase in
 351 plate shrinkage at high speeds. As it is shown in Figure 11a, at high impact velocities, the elastic energy
 352 (associated with the deformation of the plate and breakage of fibres) is the system's primary energy since
 353 the energy absorbed by the cohesive is almost constant. For impact velocities below the critical velocity
 354 ($v_o = 160 \text{ m/s}$ or $E_{impact} = 384 \text{ J}$), the importance of the energy absorbed by the cohesive is 22% of the
 355 total. However, when PFB D is measured ($v_o = 70 \text{ m/s}$), the energy absorbed by the cohesive is practically
 356 60%, according to Figure 11b, where fibre breakage does not occur. For impact velocities above 90 m/s, the
 357 elastic energy increases and becomes the system's main energy, leading to an increase in the PFB D.

358 Regarding the influence of mass, the maximum resistance is linearly related to the impact energy; the
 359 highest energy is given to the highest mass impactor. However, the breakage of the first layer of the
 360 laminate is linked to the same impact velocity. PFB D is independent of the mass of the projectile for the
 361 same impact energy.

362 **6. Conclusions**

363

364 This paper has developed a combined experimental and numerical methodology to analyse the impact force
365 and permanent back face deformation on aramid/PVB composite plates. The paper's main contribution is the
366 analysis of aramid composite behaviour and energy absorption mechanisms under high energy impact on a
367 non-perforation regime.

368

369 An experimental test campaign involving instrumentation of the process to obtain the impact velocity of the
370 projectile and impact force history on the target plate during the impact event was carried out. An
371 interesting parameter on the structures' design is the permanent back face deformation (Pbfd) which has
372 been measured using a digital profilometer, and a significant dependency with impact velocity was
observed.

373

374 The numerical model is based on the Hou modified criterion, and it has been implemented in a user
375 subroutine to be used with the finite element code ABAQUS. The laminate is modelled with the same
376 number of layers as specimens considering the cohesive surface between layers. The present numerical
377 model successfully predicts aramid/PVB laminates' mechanical behaviour using blunt projectiles for
impacts.

378

379 From the failure of the specimens, it has been observed that hardly any penetration occurs in the range of
380 the initial impact velocities considered and that most of the impact energy is absorbed through the
381 deformation of the specimen. The delamination originates this deformation between layers that allows its
382 relative displacement, causing a decrease in the specimen's stiffness and, therefore, greater ease of
383 deformation. From this deduction, it can be stated that by increasing the impact energy in the study range,
384 where the penetration is shallow, and no complete perforation is achieved, and affecting only the first layers
385 of the composite, the permanent deformation of the back-face deformation of the plates increases linearly.
386 This statement has been corroborated with the numerical model developed, yielding a good correlation
between both results.

387

388 Furthermore, it has been observed how the amount of energy dissipated by the delamination between layers
389 of the composite decreases as the impact energy increases, going from a more global response of the system
390 where this failure mechanism is the majority concerning others such as the breakage of fibres or crushing, to
more localised damage, where this type of failure is representative.

391

392 **Acknowledgements**

393

394 The authors acknowledge the Ministry of Economy and Competitiveness of Spain and FEDER
395 program under the Project DPI2017-88166-R for the financial support of the work. Prof. A. Rusinek
396 acknowledges the financial support from the program UC3M-Santander Chair of Excellence in
additive manufacturing.

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