



PLASMA & MATERIALS PROCESSING

# Low Temperature Plasma-Enhanced ALD of Metal Oxide Thin Films

Stephen E. Potts, L. R. J. G. van den Elzen,  
G. Dingemans, E. Langereis, W. Keuning,  
J. C. Goverde, D. Hoogeland,  
M. C. M. van de Sanden and W. M. M. Kessels

216<sup>th</sup> ECS Meeting, Vienna, Austria  
6<sup>th</sup> October 2009



The research leading to these results has received funding from the European Community's Seventh Framework Programme (FP7/2007-2013) under grant agreement number CP-FP213996-1.

**TU** / **e**

Technische Universiteit  
**Eindhoven**  
University of Technology

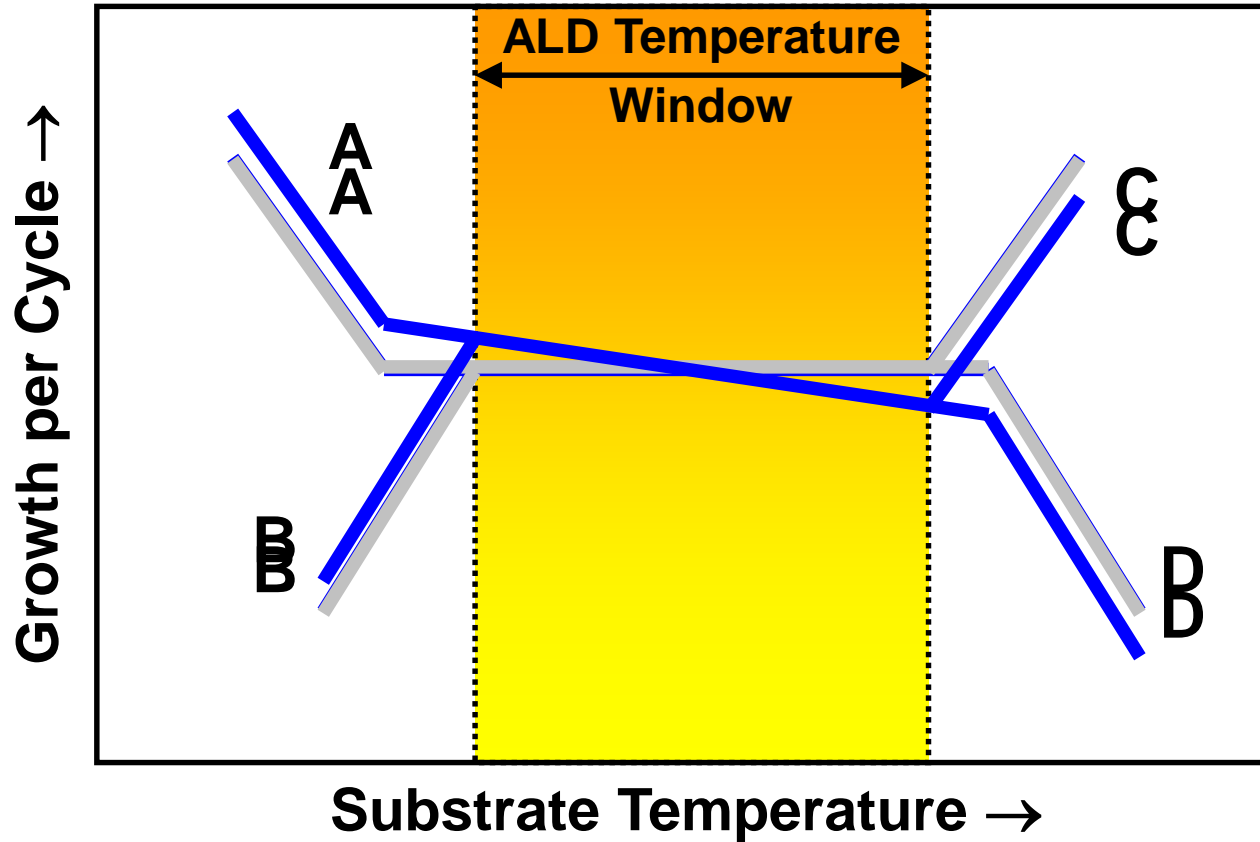
**Where innovation starts**

- **The ALD temperature window**
- **Why low temperature plasma-enhanced ALD?**
- **ALD Reactors**
- **Film characterisation**
  - spectroscopic ellipsometry and Rutherford backscattering
- **Overview of low temperature plasma-enhanced ALD of metal oxides**
  - Comparison with thermal procedures and literature
  - Growth per cycle
  - Atoms deposited per cycle
  - Film composition
- **Conclusions**

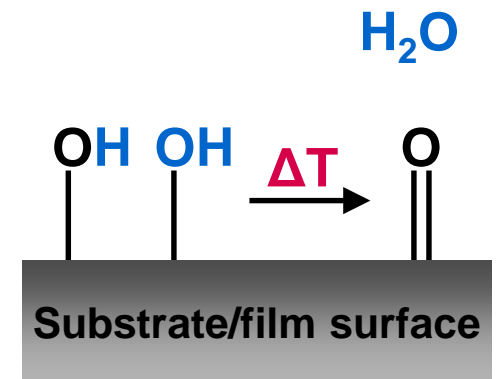


# The ALD Temperature Window

2



- A. Condensation
- B. Insufficient thermal energy
- C. CVD
- D. Evaporation



- Assumption: a sub-monolayer of material is deposited
- **Loss of surface groups** with increasing temperature

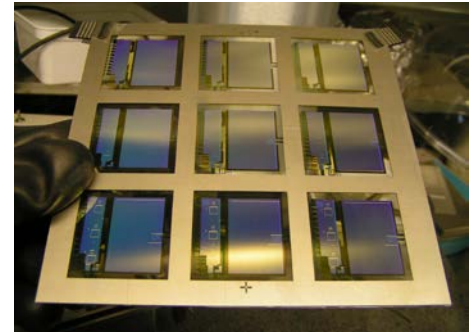
T. Suntola, *Mater. Sci. Rep.*, 4, 261 (1989).

/ Applied Physics / Plasma & Materials Processing

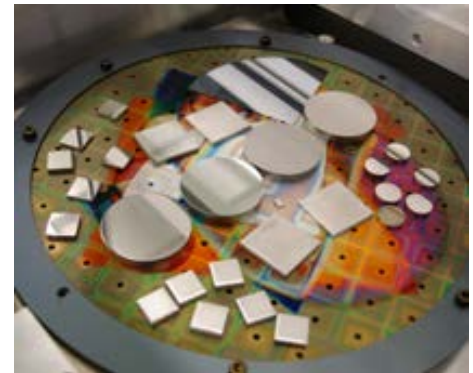
# Why Low Temperature ALD?

3

- **Some applications require high film quality but the substrates required are temperature-sensitive.**
- **Dense, defect-free films required.**
- **Organic substrates**
  - Moisture permeation barriers in OLEDs
  - Thin film transistors
- **Metals (or polymers) requiring a corrosion-resistant barrier layer**
  - Higher temperatures can alter the metal's mechanical properties



OLEDs at TU/e



Coating metal substrates at TU/e

# Low Temperature ALD in the Literature

4

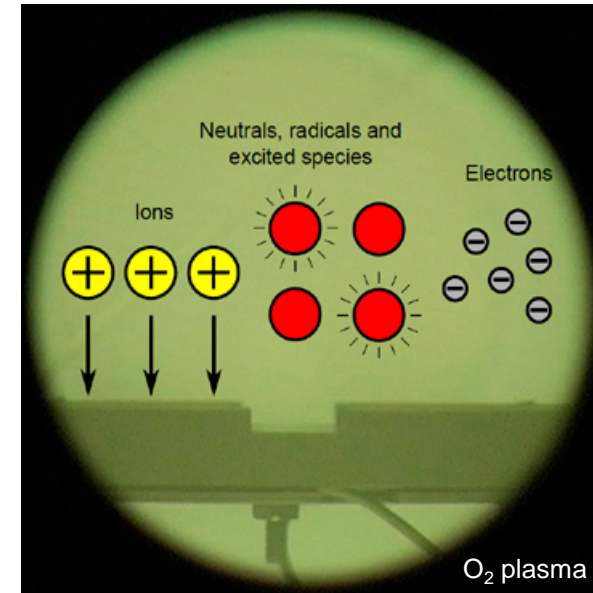
Material	Metal Precursor	Oxidant	Lowest $T_s$ (° C)	Reference
Al <sub>2</sub> O <sub>3</sub>	[Al(CH <sub>3</sub> ) <sub>3</sub> ]	H <sub>2</sub> O	33	Groner <i>et al.</i>
	[Al(CH <sub>3</sub> ) <sub>3</sub> ]	O <sub>3</sub>	25	Kim <i>et al.</i>
	[Al(CH <sub>3</sub> ) <sub>3</sub> ]	O <sub>2</sub> plasma	25	van Hemmen <i>et al.</i>
TiO <sub>2</sub>	TiCl <sub>4</sub>	H <sub>2</sub> O	100	Aarik <i>et al.</i>
	TiCl <sub>4</sub>	H <sub>2</sub> O <sub>2</sub>	100	King <i>et al.</i>
	[Ti(O <sup>i</sup> Pr) <sub>4</sub> ]	H <sub>2</sub> O	150	Ritala <i>et al.</i>
	[Ti(O <sup>i</sup> Pr) <sub>4</sub> ]	H <sub>2</sub> O <sub>2</sub>	77	Liang <i>et al.</i>
Ta <sub>2</sub> O <sub>5</sub>	TaCl <sub>5</sub>	H <sub>2</sub> O	80	Kukli <i>et al.</i>
	[Ta(NMe <sub>2</sub> ) <sub>5</sub> ]	H <sub>2</sub> O	150	Maeng <i>et al.</i>
	[Ta(NMe <sub>2</sub> ) <sub>5</sub> ]	O <sub>2</sub> plasma	100	Heil <i>et al.</i>
PtO <sub>x</sub>	[Pt(acac) <sub>2</sub> ]	O <sub>3</sub>	120	Hämäläinen <i>et al.</i>
	[Pt(Cp <sup>Me</sup> )Me <sub>3</sub> ]	O <sub>2</sub> plasma	100	Knoops <i>et al.</i>
ZnO	[Zn(CH <sub>2</sub> CH <sub>3</sub> ) <sub>2</sub> ]	H <sub>2</sub> O	60	Guziewicz <i>et al.</i>
	[Zn(CH <sub>2</sub> CH <sub>3</sub> ) <sub>2</sub> ]	H <sub>2</sub> O <sub>2</sub>	25	King <i>et al.</i>

For full references, see S. E. Potts *et al.*, *ECS Trans.*, **25**, 233 (2009).

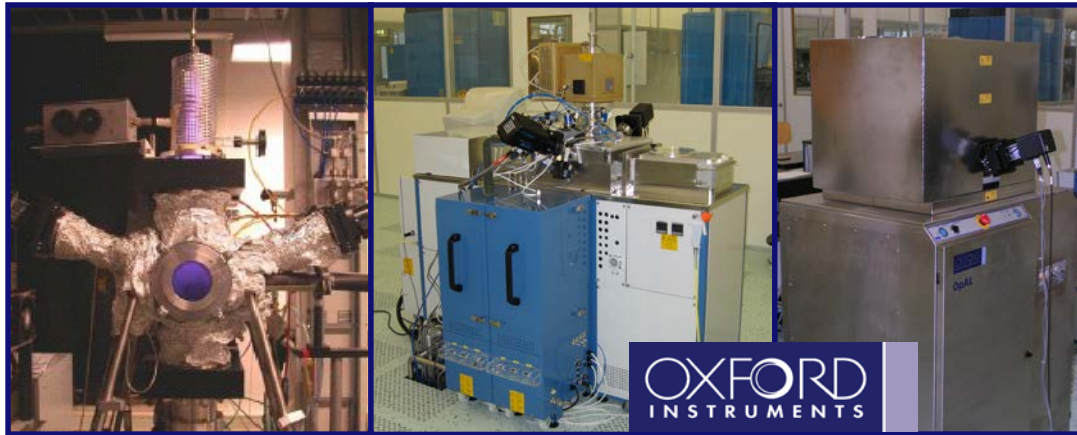
# Why Plasma-Enhanced ALD?

5

- **Gas ionised by electrical energy**
  - Ions
  - Electrons
  - Neutral species including radicals
- Radicals react with surface groups
- Ion energy and ion flux
  - surface ion bombardment
- Can lead to denser films
- Increased reactivity
- Extension of temperature window down to room temperature?



## Remote Plasma ALD Reactors

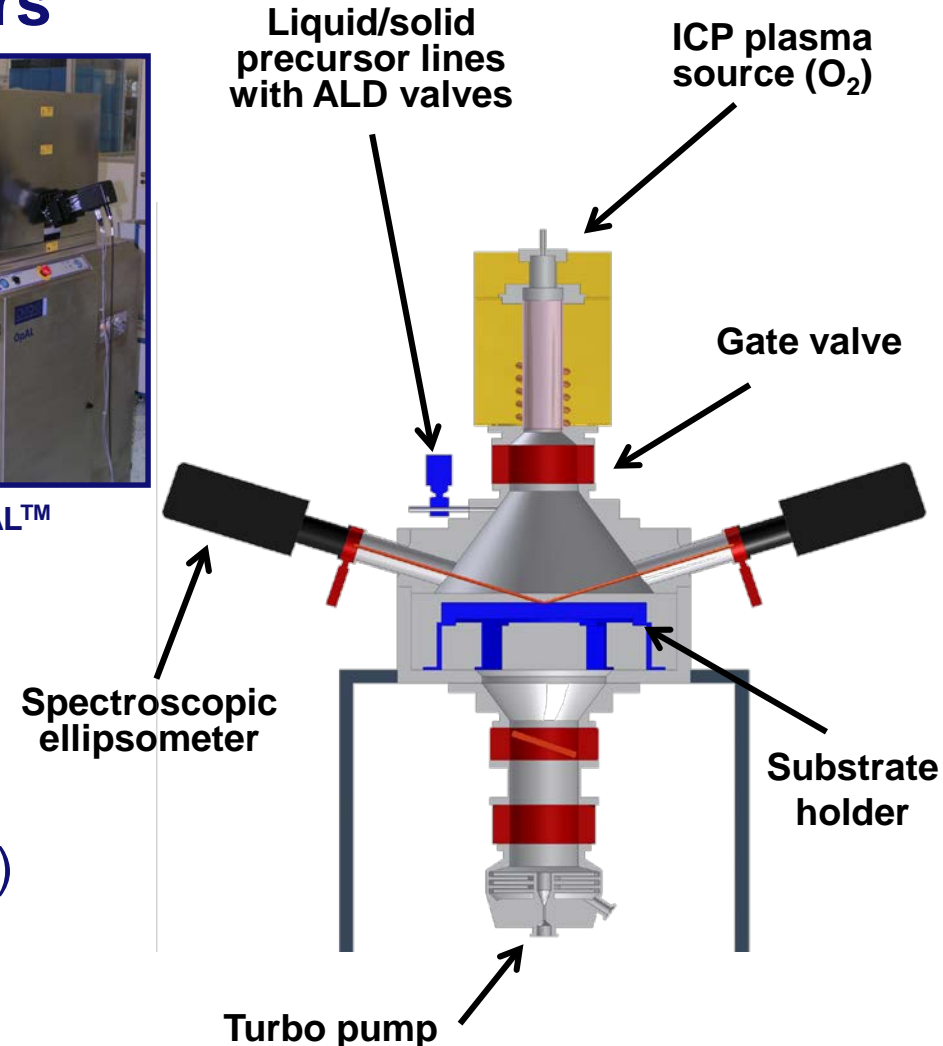


ALD-I  
(home-built)

FlexAL™

OpAL™

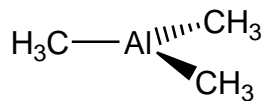
- p-type Si{100} substrates
- Diagnostics
  - Film thickness:
    - Spectroscopic ellipsometry (SE)
  - Film composition
    - RBS and ERD (H)



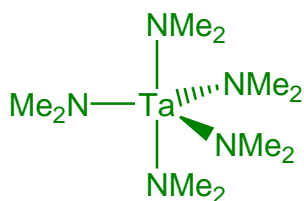


# Plasma-Enhanced ALD of Metal Oxides

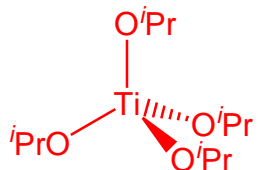
1.  $[Al(CH_3)_3]$   
TMA



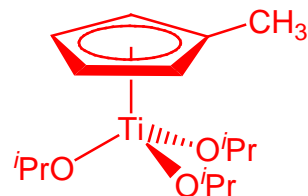
2.  $[Ta(NMe_2)_5]$   
PDMAT



3.  $[Ti(O^iPr)_4]$   
TTIP

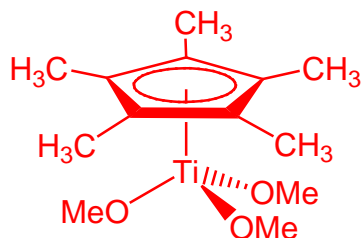


4.  $[Ti(Cp^Me)(O^iPr)_3]$

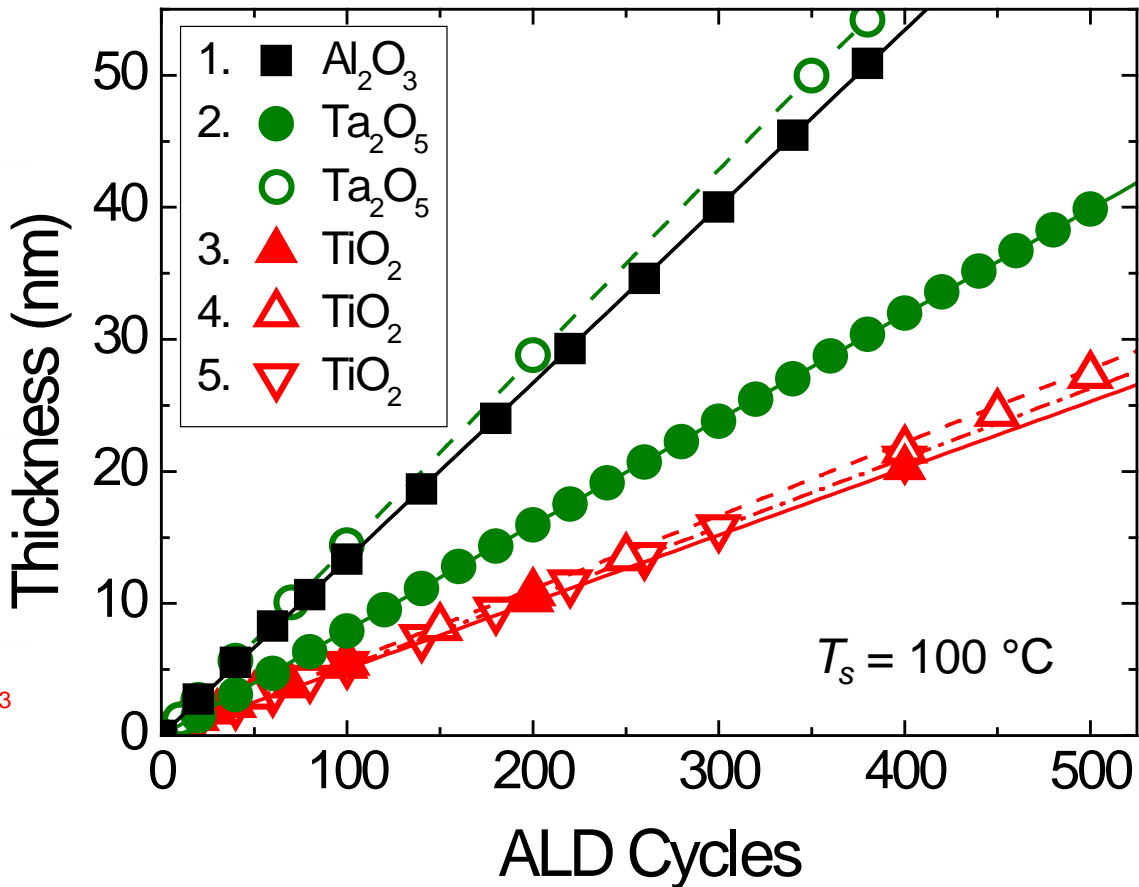


**SAFC** Hitech™  
Enabling Technology

5.  $[TiCp^*(OCH_3)_3]$



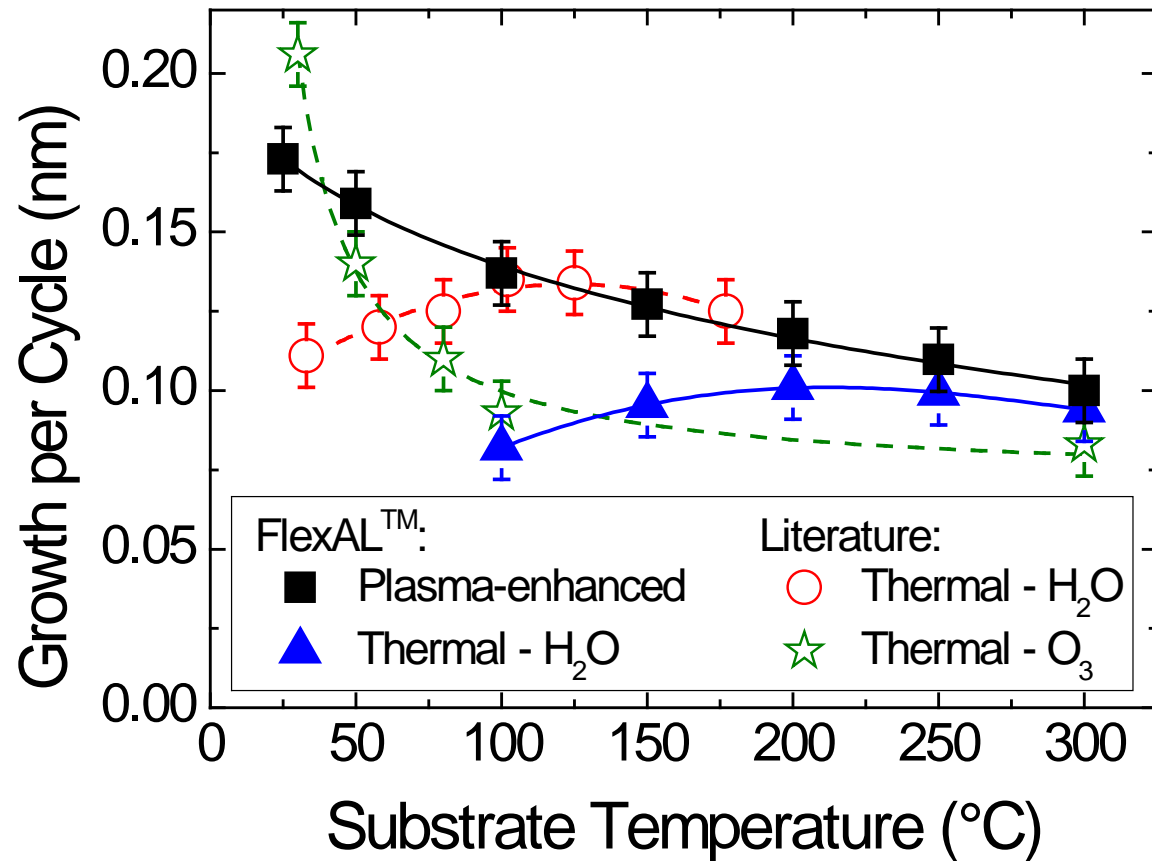
**AIR LIQUIDE**



- Measured using *in situ* SE
- No nucleation delay
- Slope gives growth per cycle for the process



# Al<sub>2</sub>O<sub>3</sub>: Growth per Cycle



- Water processes: lower growths per cycle at low temperatures
- Ozone process: many extra surface groups at  $T_s < 100$  °C.
- Reduction in growth per cycle with increasing  $T_s \rightarrow$  dehydroxylation.

**Plasma-enhanced ALD gives the higher growths per cycle at low deposition temperatures.**

[■], [▲] J. L. van Hemmen *et al.*, *J. Electrochem. Soc.* **154**, G165 (2007).

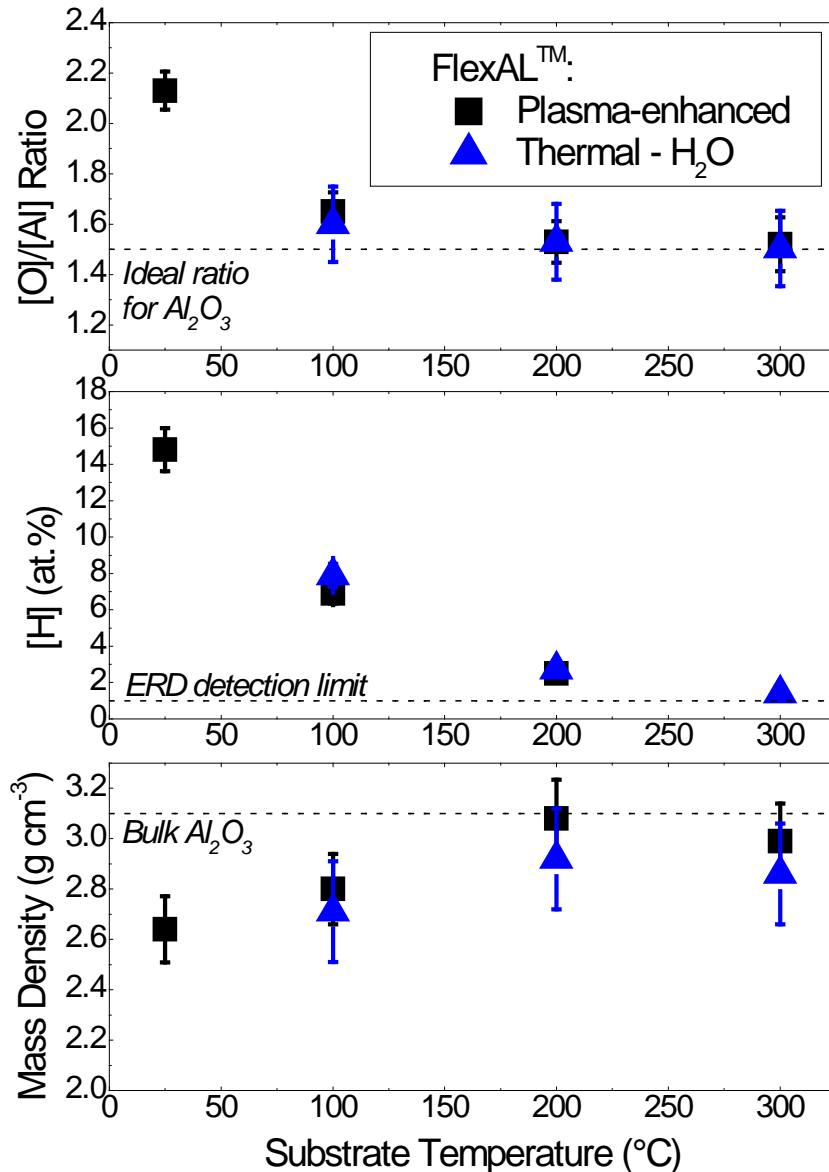
[○] M. D. Groner *et al.*, *Chem. Mater.*, **16**, 639 (2004).

[☆] S. K. Kim *et al.*, *J. Electrochem. Soc.*, **153**, F69 (2006).

/ Applied Physics / Plasma & Materials Processing

# Al<sub>2</sub>O<sub>3</sub>: Film Composition

9

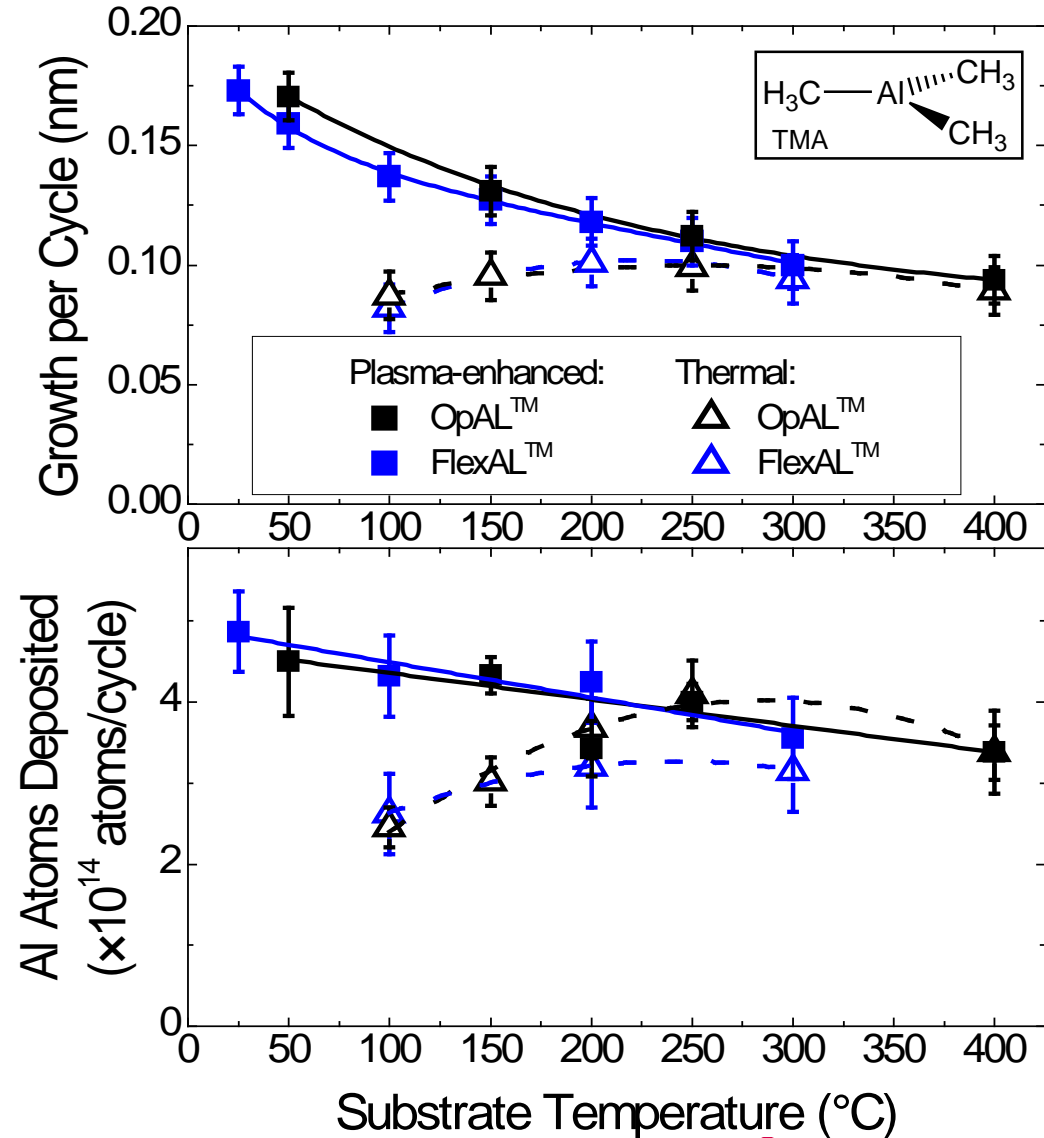


- [C] < 1 at.% in each case.
- –OH is prominent at lower temperatures.
- Leads to increasing mass density of the films with deposition temperature.
- No significant composition difference between plasma and thermal ALD.

# Al<sub>2</sub>O<sub>3</sub>: Al Atoms Deposited

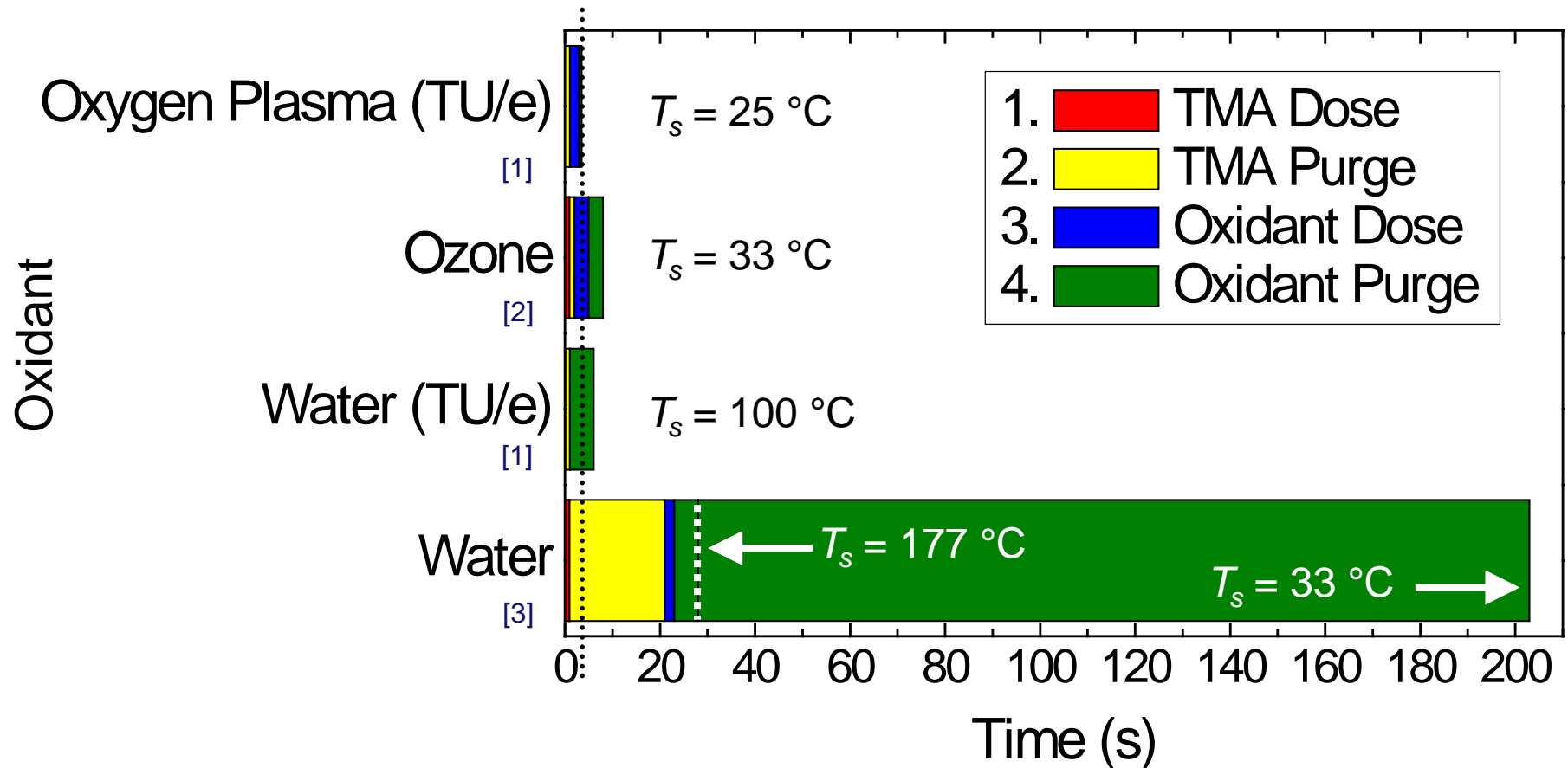
10

- GPC decreases with increasing substrate temperature for plasma ALD.
- This is less apparent when focussing on the Al atoms deposited per cycle.
- Thermal H<sub>2</sub>O process deposits fewer atoms below optimisation temperature.



# Al<sub>2</sub>O<sub>3</sub>: Cycle Time

11



**Lower deposition temperatures require a longer oxidant purge for thermal processes.**

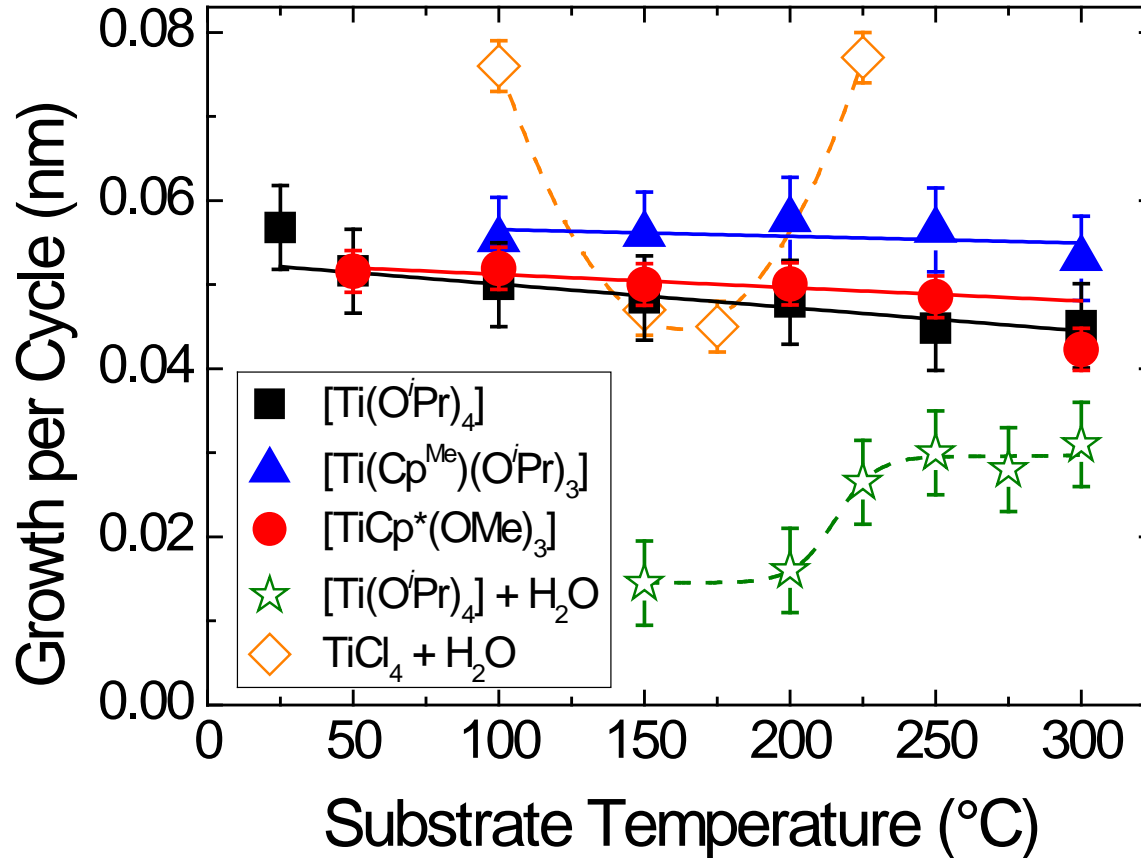
[1] L.R. J. G. van den Elzen, G. Dingemans *et al.*, work to be published (2009).

[2] S. K. Kim *et al.*, *J. Electrochem. Soc.*, **153**, F69 (2006).

[3] M. D. Groner *et al.*, *Chem. Mater.*, **16**, 639 (2004).

# TiO<sub>2</sub>: Growth per Cycle

12



- Dehydroxylation with increasing  $T_s$ .
- Use of alkoxy-based precursors → no chlorine in final film.
- [Ti(OiPr)<sub>4</sub>] + water process: very low growth per cycle
- TiCl<sub>4</sub> process: etching at  $T_s = 150-175$  °C.

**Plasma-enhanced ALD: higher growth per cycle**

[■] W. Keuning *et al.*, work to be published.

[▲] E. Langereis *et al.*, work to be published.

[●] S. E. Potts *et al.*, work to be published.

[☆] M. Ritala *et al.*, *Chem. Mater.*, **5**, 1174 (1993).

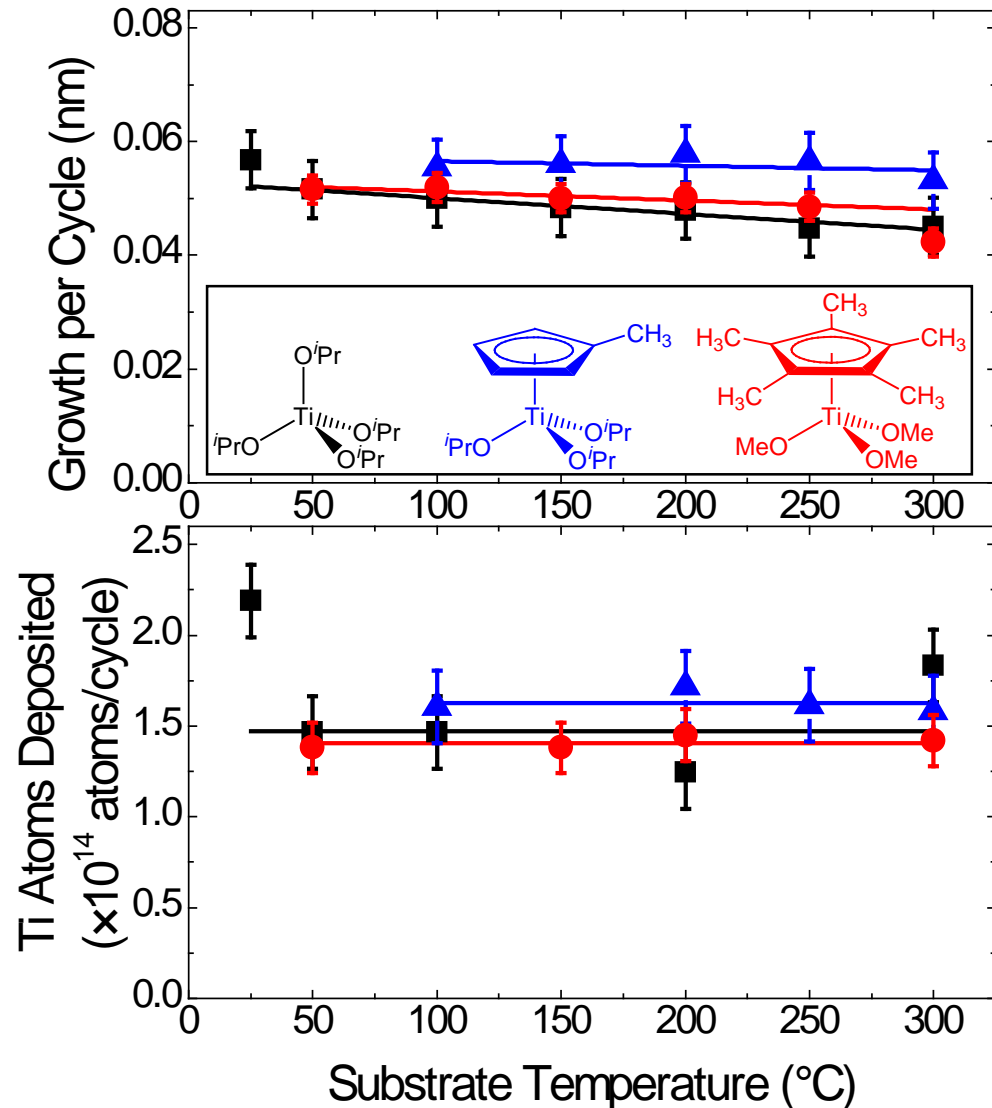
[◇] J. Aarik *et al.*, *J. Cryst. Growth*, **220**, 531 (2000).

# TiO<sub>2</sub>: Film Composition & Ti Atoms Deposited

13

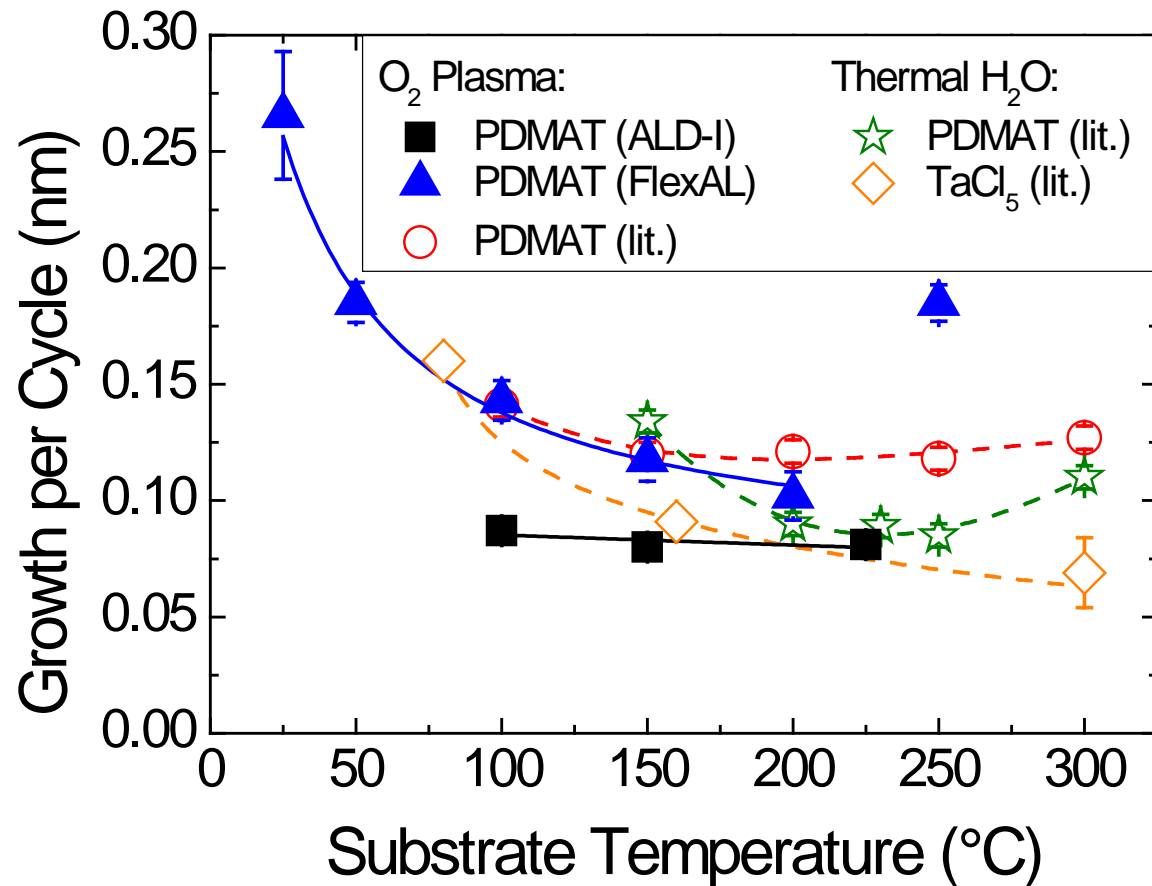
- All 3 Ti precursors: same film composition for  $T_s \geq 50$  °C
  - [O]/[Ti] = 2 (TiO<sub>2</sub>)
  - [C] < 1 at.%.
  - [H] below detection limit at  $T_s \geq 50$  °C (0.1-5%)
  - -OH groups only seen at  $T_s = 25$  °C for [Ti(O*i*Pr)<sub>4</sub>]
- Thermal route [H] ~0.3 at.%

**Reduction in GPC but Ti atoms deposited per cycle are almost constant for Cp precursors.**



# Ta<sub>2</sub>O<sub>5</sub>: Growth per Cycle

14



- [O]/[Ta] Ratio:
- **ALD-I = 2.5**
- **FlexAL = 2.5-3**
- **Lit. PDMAT = 2.6**
- **Lit. TaCl<sub>5</sub> = ~2 ± 0.1**
  
- [C] and [N] < 1 at.% in all cases for PDMAT
- [H] detected but < 5 at.% (ALD-I)
- **From TaCl<sub>5</sub> [Cl] up to 6 at.%**
  
- **Difference in GPC due to different reactors?**

[■] S. B. S. Heil *et al.*, *J. Vac. Sci. Technol. A*, **26**, 472 (2008).

[▲] S. E. Potts *et al.*, work to be published.

[○] W. J. Maeng *et al.*, *J. Vac. Sci. Technol. B*, **24**, 2276 (2008).

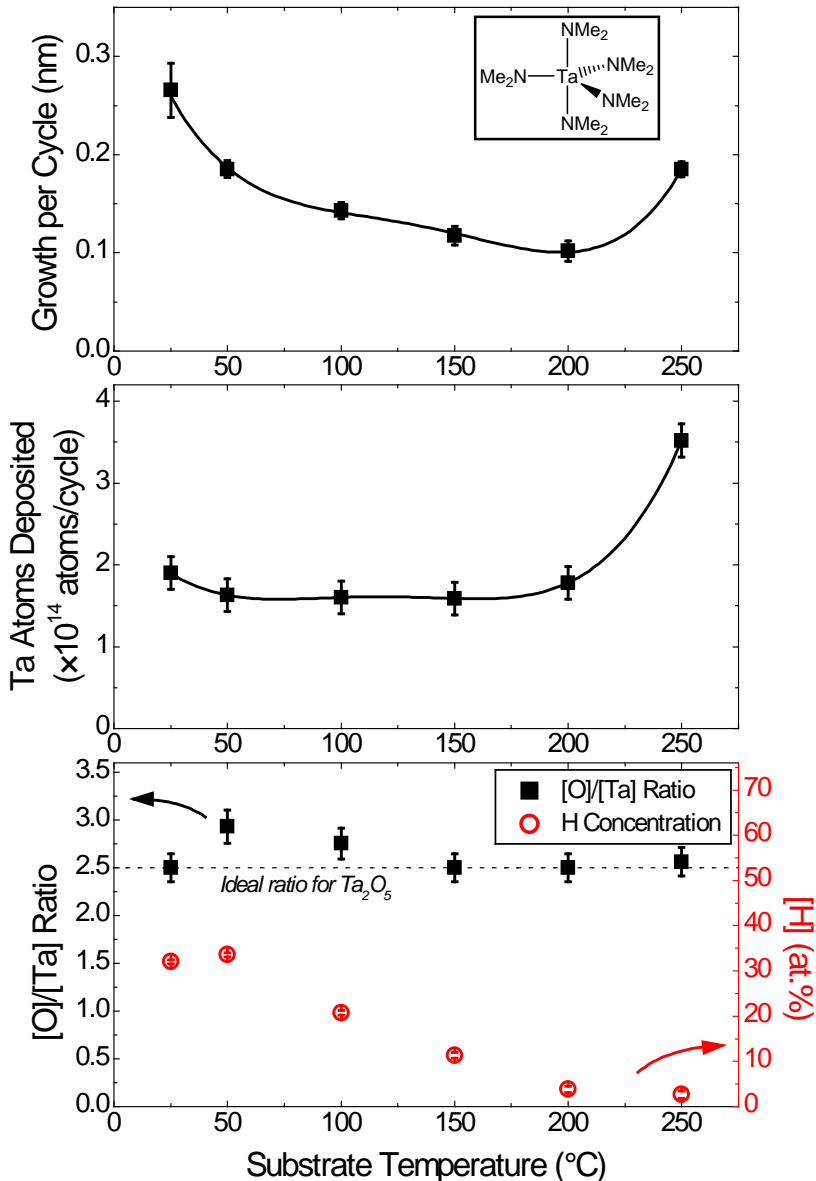
[☆] W. J. Maeng and H. Kim, *Electrochem. Solid-State Lett.*, **9**, G191 (2006).

[◇] K. Kukli *et al.*, *Thin Solid Films*, **260**, 135 (1995).



# Ta<sub>2</sub>O<sub>5</sub>: Film Composition & Ta Atoms Deposited

15



- Drop in growth per cycle with increasing temperature.
- Number of Ta atoms deposited reasonably constant for  $T_s = 50-200$  °C.
- A big increase in Ta deposited at 250 °C
- decomposition of PDMAT = CVD
- H decreases with increasing substrate temperature.

## $\text{Al}_2\text{O}_3$ from $[\text{Al}(\text{CH}_3)_3]$

- Higher growth per cycle down to room temperature for the plasma process.
- Higher quality films at low temperatures for the plasma process compared with the ozone thermal process.
- $\text{Al}_2\text{O}_3$  processes more sensitive to surface dehydroxylation compared with  $\text{TiO}_2$  and  $\text{Ta}_2\text{O}_5$ .
- Reduced cycle times for the plasma process.

## $\text{Ta}_2\text{O}_5$ from $[\text{Ta}(\text{NMe}_2)_5]$

- Stoichiometric  $\text{Ta}_2\text{O}_5$  down to 150 °C.
- CVD at 250 °C.

## $\text{TiO}_2$ from $[\text{Ti}(\text{O}^i\text{Pr})_4]$ , $[\text{Ti}(\text{Cp}^{\text{Me}})(\text{O}^i\text{Pr})_3]$ or $[\text{TiCp}^*(\text{OMe})_3]$

- Pure, stoichiometric  $\text{TiO}_2$  down to 50 °C.
- Higher growth per cycle than for  $[\text{Ti}(\text{O}^i\text{Pr})_4]$  with water.

- Benefits of plasma processes depend on the application.
- Plasma processes can give high quality films at lower temperatures.
- Analysis of atoms/cycle gives additional insight into the ALD temperature window.