

1 **Investigation of material flow and thermomechanical behavior during friction stir welding**
2 **of an AZ31B alloy for threaded and unthreaded pin geometries using computational solid**
3 **mechanics simulation**

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17
18 **ABSTRACT**

19 In the friction stir welding process, the tool role in the material flow and its
20 thermomechanical behavior is still not entirely understood. Several modeling approaches
21 attempted to explain the material and tool relationship, but to this date, insufficient results were
22 provided in this matter. Regarding this issue and the urgent need for trustful friction stir welding
23 models, a computational solid mechanic's model capable of simulating material flow and defect
24 formation is presented. This model uses an Arbitrary Lagrangian-Eulerian code comparing a
25 threaded and unthread pin profile. The model was able to reproduce the tool's torque,
26 temperatures, and material flow along the entire process, including the underreported downward
27 flow effect promoted by threaded pin's. A point tracking analysis revealed that threads increase
28 the material velocity and strain rate to almost 30% compared to unthreaded conditions, promoting
29 a temperature increment during the process, which improved the material flow and avoided filling
30 defects. The presented results showed the model's capability to reproduce the defects observed in
31 real welded joints, material thermomechanical characteristics and high sensitivity to welding
32 parameters and tool geometries. Nevertheless, the outcomes of this work contribute to essential
33 guidelines for future friction stir welding modeling and development, tool design, and defect
34 prediction.

35
36 **Keywords:** computational solid mechanics; friction stir welding; material flow modeling;
37 design of tools; light alloys; magnesium alloys

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39 **1. INTRODUCTION**

40 Friction stir welding (FSW) is a solid-state technique in which severe plastic deformation
41 and heat generation results in the processing and joining of similar and dissimilar materials.
42 Features such as high-strength welds and small post-welding distortion made the FSW an
43 excellent alternative for fusion welding, capable to also avoid its problems related to melting and
44 solidification. Such a method has been applied to a wide range of materials in critical applications
45 [1–5], promoting an aggressive need for optimization and development in friction stir welding
46 scenarios.

47 During the process, the tool provides intense material deformation, which results in
48 bonding between materials. The understanding of tool/processed material relationships during
49 FSW is desirable for the process parameter optimization, microstructure control, and predictions
50 of welded joint properties. It is well known that the tool used at FSW impacts the microstructures
51 of welded joint and mechanical properties, but how it does it, is not fully comprehended [6].
52 Likewise, its effect on the flow of material is still one of the least understood phenomena due to
53 the complexity and difficulty of visualization and measurement restrictions. Threaded tools have
54 been considered efficient, avoiding the formation of defects, producing sound welds effectively
55 by improving the material flow in aluminum alloys [7], and magnesium alloys [8].

56 Considered a fast and cost-reduction analysis, the numerical simulation of the FSW
57 process analysis has become a powerful tool to support the real process. Nevertheless, the
58 numerical modeling of the friction stir welding process has some challenges due to simultaneous
59 high strain, strain rate, and temperature. Studies have been attempted to simulate temperature,
60 strain, strain rate, and material flow in FSW in the past years [9–13]. Despite being fast and
61 efficient methods for calculating temperatures of the friction welding process, computational fluid
62 dynamics (CFD) models need to assign viscosity to the material and generally shows an over-
63 prediction of the velocity fields and inability to predict defects generated by the process
64 parameters and tool response to the process [14]. Kadian et al., [15] work considered different
65 types of tool pin geometries and their effects on the material flow. The results showed that conical
66 threaded tools provide higher temperatures than smooth conical tools, although they provided
67 smaller velocities in the material. In most studies using CFD models, the material strain is based
68 on fluid shear stresses. Furthermore, the fluid assumption enables us to calculate some critical
69 process characteristics, such as tool torque, the mechanical response of the material, and defect
70 prediction.

71 Computational solid mechanics (CSM) based methods otherwise consider that the
72 welding material remains solid during the process and has been extensively used in research [16–
73 24]. The force equilibrium equation based on continuum mechanics and the resulting partial
74 derivative equations (PDE). Long et al. [17] presented a 3D thermal-mechanical coupled finite
75 element model and investigated the tool tilt angle effect during friction stir welding. This model
76 was able to predict the thermal-mechanical state variables during the welding and the post-
77 welding morphology of the weld. Gao et al. [16] reported the downward spiral movement affected
78 by the shoulder, whereas the lower material first moves in an upward spiral mode affected by the
79 pin and then moves downward. Pashazadeh et al. [19], explored FSW experiments and numerical
80 simulation for a pure copper alloy. The hardness and microstructure of a copper workpiece were
81 investigated. The numerical results for hardness showed good agreement with recorded
82 experimental data, suggesting the calculated deformation and temperatures were accurate.
83 However, recent CSM-based simulations had not included threaded geometries due to extensive
84 computational process-times and numerical robustness.

85 Due to the different results focused on individual issues, more studies including modeling
86 from a holistic approach are needed to obtain information about the effect of tool geometries in
87 material flow, thermomechanical behavior, and, consequently, defect prediction capabilities.

88 Here, based on the good results presented by CMS models, numerical simulation of the
89 friction stir process using CSM-based DEFORM 3D software was explored. Pin geometry effects
90 were evaluated using a new mesh approach to overcome the challenges related to large element
91 distortion. Moreover, thermal-mechanical conditions of the material for the welding cases with
92 and without threads are compared, the thread effects on strain rate, temperature, and flow paths
93 were explained.

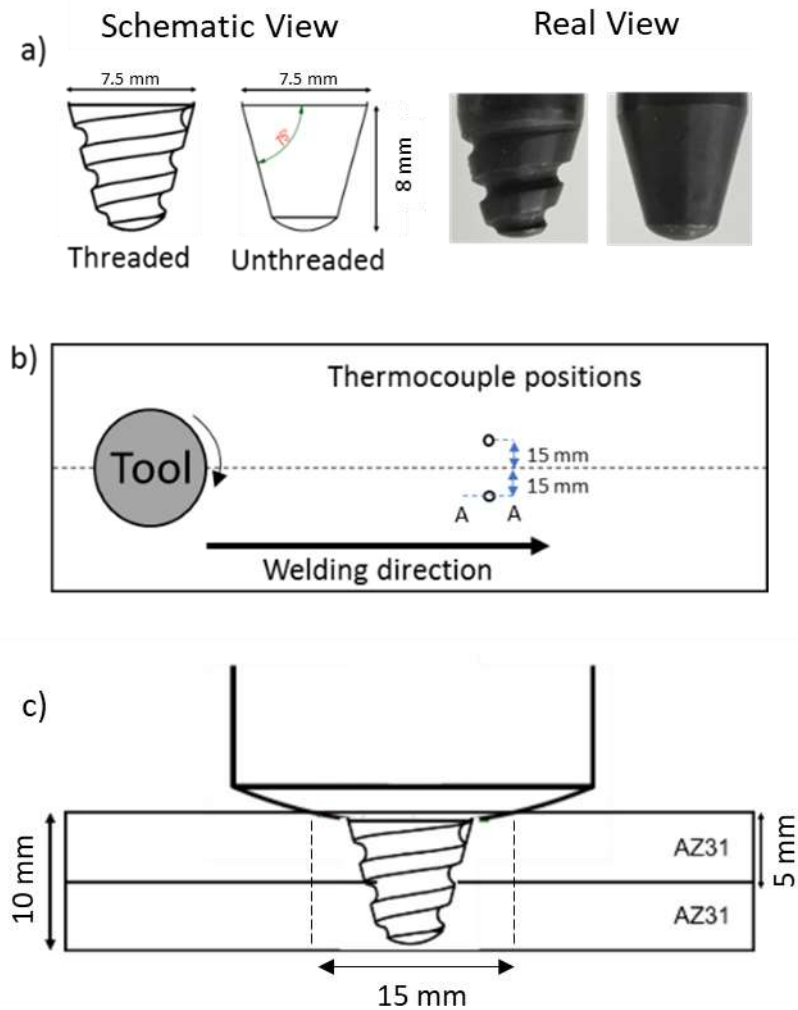
94 95 96 **2. EXPERIMENTAL SETUP**

97 98 **2.1. Tool design and welding parameters**

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100 The FSW tool material was H13 steel with a 36-mm concave shoulder. The tool design
101 was developed *in-house*. Figure 1 (a) shows the schematic and real pins' view, which considers
102 the basic conical geometries with and without threads. The threaded type has a pitch of 3.5 mm,
103 and the unthreaded is flat, without threads. The pins had a 7.5-mm diameter at the base and a total
104 length of 8-mm.

105 Sheets of AZ31 magnesium alloy with the dimensions of 200mm x 50mm x 5 mm were
106 used in lap position. Table 1 shows the FSW parameters employed in experimental and simulation
107 approaches. The experimental welding temperatures were measured using K-type thermocouples
108 placed 15 mm away from the weld center, 75 mm from the end of the plate and 2.5 mm depth
109 from the top surface, on both sides, as shown in Figure 1 (b). The tool torque was monitored and
110 recorded in real-time by the FSW machine spindle using the instantaneous power and current of

111 the motor. In Figure 1 (c), the weld schematic shows the maximum contact gap (20 mm) provided
 112 by the shoulder.
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Figure 1: (a) Schematic and real view of the pins; (b) Thermocouple positions on the plate
Table 1: FSW parameters for the experimental tests and numerical simulation

Spindle Speed (RPM)	Traverse Speed (mm.min ⁻¹)	Pin Geometry	Short identification
800	200	Threaded	T82
	200	Unthreaded	U82

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2.2. Material model

In order to simulate the material behavior during hot deformation, the material relationship between flow stress on temperature and strain rate was required. Empirical equations usually are applied to establish the deformation activation energy and hot deformation behavior of alloys. The Arrhenius relationship based on Sellars and McTegart [25] equations are one of the most frequently used, relating the strain rate, flow stress, and temperature. The selected parameters were acquired from an AZ31 hot deformation analysis performed by Giorjao et al., [26]. The temperatures (up to 500 °C), strain, and strain rate (up to 10 s⁻¹) range evaluated by the author fit well with the expected mechanical behavior observed in friction stir welding processes. A flow stress equation was obtained as shown in Equation 1,

131 where σ is the flow stress (MPa), $\dot{\epsilon}$ is the applied strain-rate (s^{-1}), R is the gas constant and T is
132 the temperature (K).
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$$135 \quad \dot{\epsilon} = 4.79 \times 10^9 [\sinh(0.014\sigma)]^{3.88} e^{\left[\frac{-128000}{RT}\right]} \quad \text{Equation 1}$$

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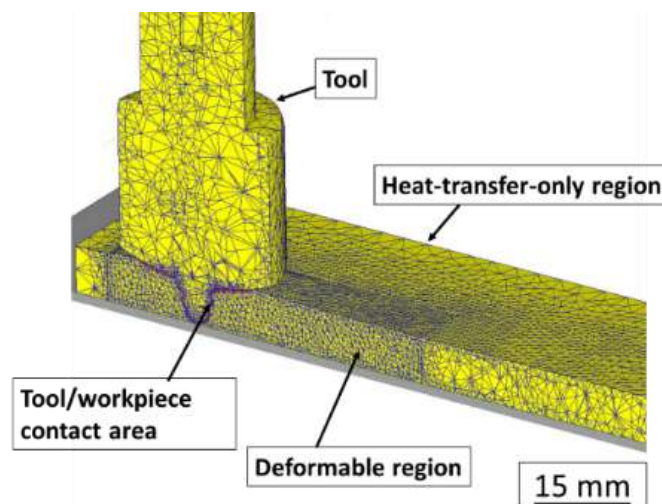
138 2.3. Numerical modeling considerations

139 The simulations were performed using the finite element analysis software DEFORM-
140 3D. This software can simultaneously perform material deformation, and thermal history using
141 an Arbitrary Lagrangian-Eulerian (ALE) formulation. The arbitrary Lagrangian-Eulerian (ALE)
142 is a finite element formulation in which the computational system is not a prior fixed in space
143 (e.g., Eulerian-based finite element formulations – CFD models) or attached to the material. In
144 the ALE method, the mesh nodes can move arbitrarily, bringing the freedom of the mesh
145 movement and allowing significant distortions of the continuum. This feature allows precise
146 material distortion and deformation emulation (and also material flow defects - voids), which are
147 the core events in the solid-state hot deformation process of friction stir welding.

148 The welding tool was assumed rigid and the workpiece as a rigid viscoplastic material.
149 Additional details on governing equations can be found in the works of Buffa et al. [22,27] and
150 Jain et al. [28]. The mesh workpiece was divided into two regions with different assignments, a
151 deformable region located under the tool pin and shoulder, and an only-heat-transfer rigid region
152 domain field. The thermal contact interaction between the two regions was considered perfect.
153 The deformable region was designed with a 20 mm width. No significant alteration is expected
154 beyond this distance in the base material. Thus, the mesh was refined locally (tool and workpiece
155 contact area), and the simulation processing time optimized. The mesh schematics can be seen in
156 Figure 2. The model considered one single plate instead of two stacked plates to reduce the contact
157 interactions and save computational time.

158 The deformable region was meshed with about 60000 tetrahedral elements with a mean
159 element size of 0.35 mm. The heat-transfer-only region was meshed with about 10000 elements.
160 The tool was meshed with about 15000 tetrahedral elements. Figure 2 illustrates the contact area
161 between the workpiece and tool, where the friction model was applied.

162 The heat transfer between the workpiece and ambient, and the tool and ambient was set
163 equal to $20 \text{ W/m}^2\text{K}$, and the Heat transfer between the workpiece and tool was set equal to 22000
164 $\text{W/m}^2\text{K}$ according to Asadi et al. [11]. The material flow was treated with an adaptive re-meshing
165 finite element formulation.
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169 **Figure 2: Mesh schematics indicating the tool domain, deformable and non-deformable**
170 **regions, and the tool/workpiece area**

2.4. Friction

In the FSW process, the friction coefficient is affected by temperature, pressure, among other variables, which makes it complex to determine its value. A trial-and-error method was adopted in this study due to the lack of experimental evidence of the high-temperature friction coefficient. To define the best friction coefficient, axial forces, and modeling temperatures, four different friction factors were calculated: 0.4, 0.6, 1, and 1.3, under 200 mm/min traverse speed and 800 rpm rotational speed. By comparing the simulated temperature results with the experimental ones, a good match was observed at the friction coefficient of 1.3.

Several authors have shown proper matching with experimental data using friction coefficient values below 1, e.g., 0.45 for Pashazadehet al. [19] and 0.3 for Gao et al. [16]. However, it is observed that all CSM models in literature explored only tools with small dimensions (below 6mm length) and simplified geometries [16–22]. In these cases, it is hypothesized that the selected low friction values are inaccurate; however, wrongly providing the right temperature according to calculations. These tools provide a simplified material flow that is easily achievable using low friction values. For large/complex tool geometries, the material movement tends to be more complicated as sticking/adhesion phenomena take a higher spot, which is achievable with low friction coefficients values.

Friction coefficient values above 1 are not very common in conventional metal forming processes. However, some studies sought to measure friction in typical conditions of friction stir welding such as pressure, tool geometry, and temperature. Duffin and Bahrani [29] determined that the friction coefficient can exceed the value of 1 by studying linear welding of steels. Similar results were found by Kumar et al. [30], who conducted experiments to study the interaction between the FSW tool and AA 7020-T6 sheets at various axial pressures and spindle speeds. The elevated contact pressures and temperatures experienced during the friction stir and similar processes promote a steep increase in the coefficient of friction identified to be due to the seizure phenomenon [31]. This phenomenon is observed when the contact condition between the rotating tool and the workpiece material changes from the slipping mode to a sticking/adhesion mode.

Figure 3 provides the tool surface covered with the AZ31B magnesium alloy. This result suggests that some of the processed material adhered to the tool surface, which supports the high friction values hypothesis during the friction stir processing.



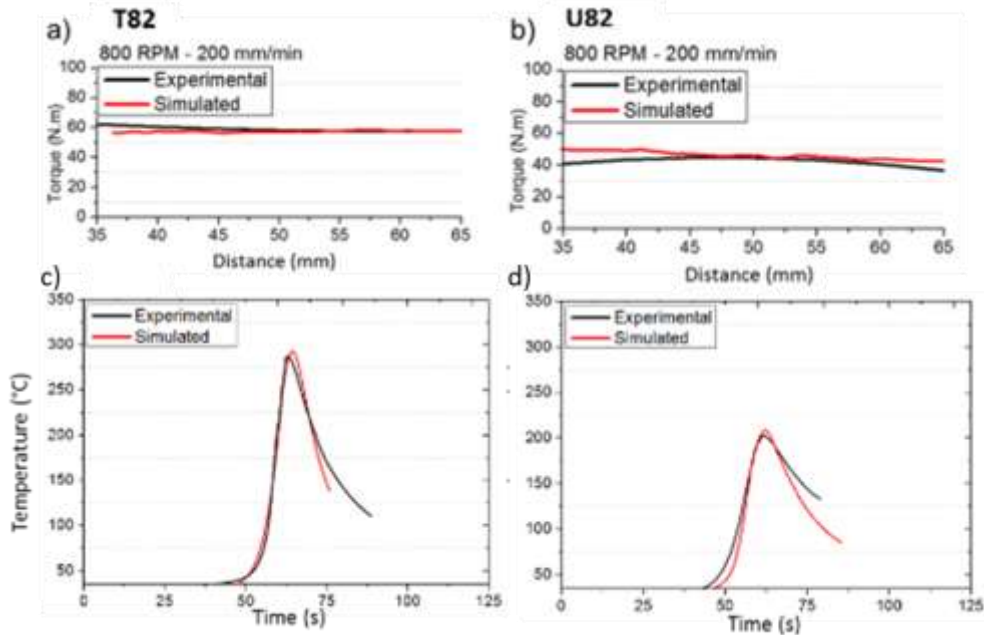
Figure 3: Tools surface after the friction stir welding processing covered with the magnesium alloy

3. RESULTS AND DISCUSSION

3.1. Model validation

To verify the results obtained in the simulations, the temperature of the sheet and torque of tools from welding conditions T82 and U82, were exposed together, corresponding to the threaded and unthreaded tools with the spindle speed of 800 RPM and 200 mm/min, respectively.

215 The results in Figure 4 (a) and Figure 4 (b) showed proper matching with torque values for both
 216 conditions. Figure 4 (c) and Figure 4 (d) show the comparison between the thermal cycle of the
 217 experimental welding and simulated welding. The simulation agrees well with the thermal cycle
 218 of the experiment, for heating and cooling rates, and peak temperature. It was noticed an increase
 219 in torque for the threaded tool, maximum values of 60 N.m for T82, and 50 N.m for U82, which
 220 is related to a higher material resistance for intricate geometries than the smooth probe as
 221 mentioned by Jonckheere et al. [32].
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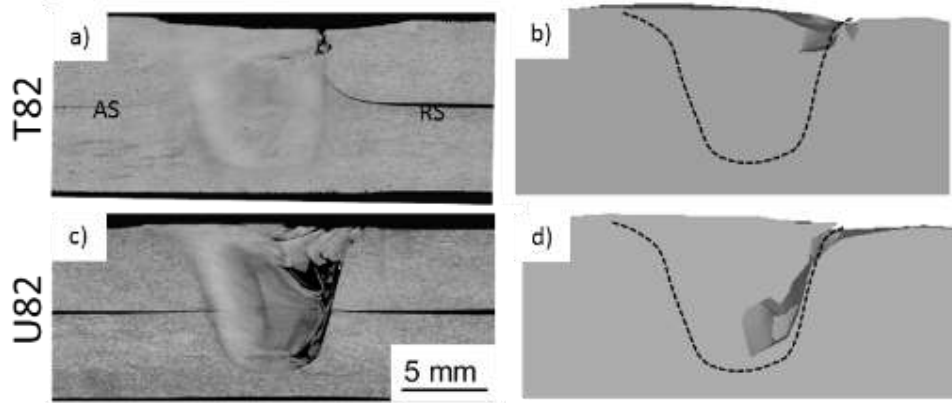


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 225 **Figure 4: Comparison of tool torque values between simulation and experiment for (a)**
 226 **T82 condition and (b) U82 condition. Comparison of peak temperature between**
 227 **simulation and experiment at 15 mm far from the weld center and 2.5 mm depth from the**
 228 **surface for (c) T82 and (d) U82**
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230 231 3.2. Joint morphology

232 In the DEFORM software, the 3D slicing feature was used to observe internal features
 233 evolution along the cross-section of the weld. Experimental and simulated cross-sections joints
 234 were compared in Figure 5. As observed, the model successfully represented the experimental
 235 results, where even the defects position and approximately volume were calculated. It is possible
 236 to observe a significant void defect in the unthreaded condition. Even though there is a difference
 237 in the morphology of the defects, this numerical model could predict the tendency of defect
 238 production.

239 According to many related studies, void defects always occurred in the stir zone (SZ) on
 240 the advancing side (AS) and have a substantial relationship with the heat input generated by pin
 241 geometry and parameters, providing a lack of material flow and insufficient plastic deformation
 242 selected [33–37]. Failla [35] related void defect formation with the heat input applied to the weld.
 243 Increasing the linear travel speed reduced the heat input, which resulted in faster cooling of the
 244 material before this region becomes filled with stirred material. Toumpis et al. [36] stated that
 245 void defects in FSW of DH36 steel appear when the heat input from the tool is insufficient, and
 246 thus, a lack of material flow may occur. Rasti [37] performed an analytical study linking the heat
 247 input and void area for several welds in 1060 aluminum alloy. The author was able to identify a
 248 minimum heat input generation to create the tunnel void-free welds. The potential causes for the
 249 defects observed and simulated, such as temperatures and material deformation behavior, will be
 250 discussed in the following sections.



251 **Figure 5: Experimental and simulated welded cross-sections for different welding**
 252 **parameters and pin geometries. Tools positions indicated in simulation for visualization**
 253 **proposes**
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 257 **3.3. Material thermomechanical history evaluation using a point tracking technique**

258 For the material history evaluation, a point tracking technique was applied. The material
 259 behavior plays a critical role in determining the weld quality of FSW, which can be observed by
 260 the marker tracking method in the simulation results [38]. An arbitrary point was chosen at the
 261 center of the weld bead ($X=0$ mm) and a specified height ($Z=7.25$ mm) from the baseline for each
 262 condition, and its position, deformation, velocity, and temperature were evaluated, as seen in
 263 Figure 5.

264 In both conditions, observing the X curve, the movement around the tool is noted. Here,
 265 the simulations reveal that the material performs one semi-cycle around the tool, returning its
 266 initial transverse position. In the literature [17,39], similar flow behavior is also predicted by
 267 simulations and experimental observations.

268 However, the points presented different responses on vertical movement, as the threads
 269 provided an intense material downflow. It was calculated almost 4-mm vertical amplitude from
 270 maximum to the minimum height position in threaded condition. The downflow took almost 0.5
 271 seconds. Then, the material returns to the initial vertical position. On unthreaded condition, a
 272 slight vertical movement was noted without the downward flow observed on the threaded tool.
 273 The material is retained in the weld bead by the tool shoulder action. Here we notice the difference
 274 of the trajectory imposed by the different pin geometries.
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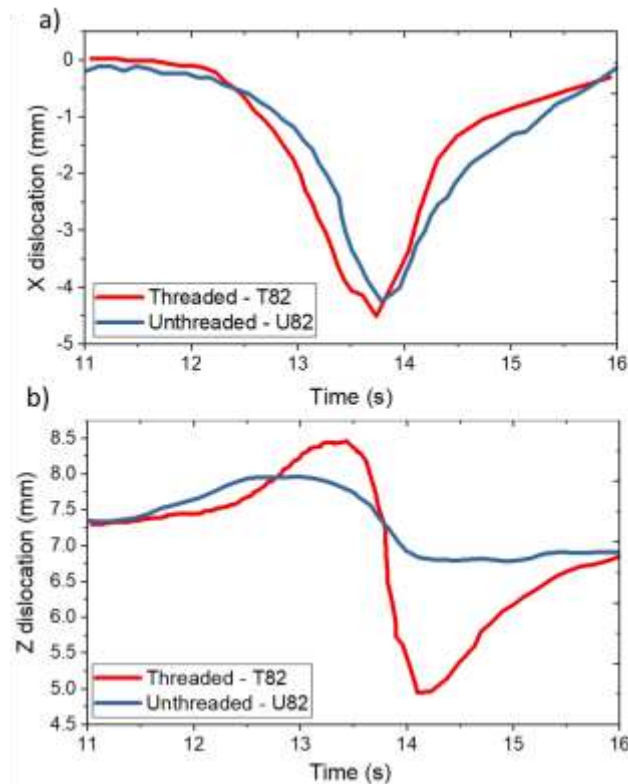


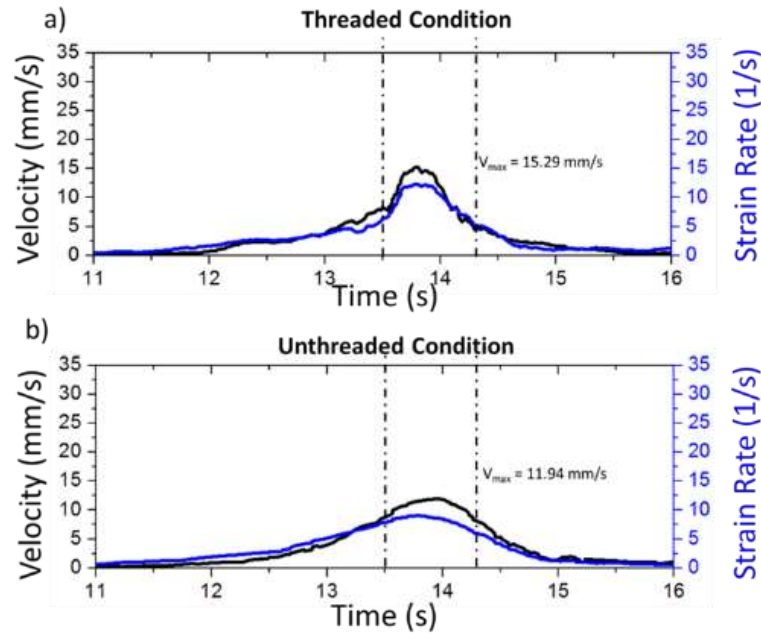
Figure 6: Comparison of the point position between the (a) threaded and (b) unthreaded condition

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After evaluating point positions, the deformation and velocity on these points were analyzed. Figure 6 displays the points' velocity and strain rate during their processing. It was noticed that both velocity and strain rate have higher values in the threaded condition, reaching maximum values at the tool domain region (demarcated by dotted lines). Also, the peak velocity and strain rate values occur during the material downflow movement. In terms of velocity and strain-rate values, the threaded provides an almost 30% increase for the same rotation and welding speed.

Moreover, the overall values of strain rate and velocity are smaller and show less difference between threaded and unthreaded conditions than those found in the literature of CFD simulations. According to the velocity values, the maximum velocity observed was 15 mm.s^{-1} . Results found in the literature indicate higher values when performed in CFD, such as 3.43 m.s^{-1} found by Kadian et al. [15], between 2 and 4 m.s^{-1} by Ji et al. [40], indicating a difference in orders of magnitude from the present results. The higher values, as mentioned before, are due to the liquid assumptions for CFD models. CFD also showed higher values for strain-rate compared to the present work, for example, 500 s^{-1} , according to Chen et al. [41].

Through visualization of markers with X-rays, Morisada et al. [42] in aluminum welds at 1000 RPM estimate the material velocity and strain-rate during friction stir welding. The material's velocity varied in a range between $50\text{-}300 \text{ mm.s}^{-1}$, and the strain-rate reached a maximum of 15 s^{-1} , values similar to those found in the present work, and in similar works using the CSM approach [21,24]. It is valuable to mention that the presented friction coefficient assumption did not provide unreliable strain-rate and velocity values for the material according to both experimental and similar models found in the literature [21,24,42].

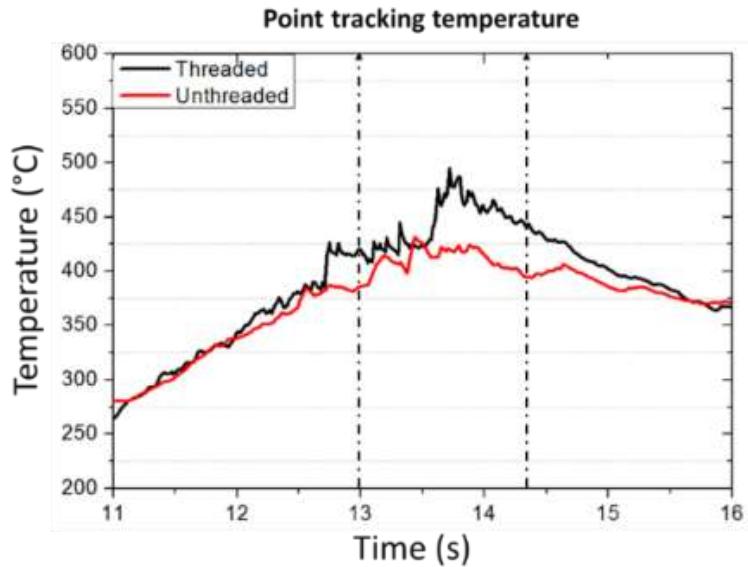


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 307 **Figure 7: Comparison of the simulated point velocity and strain rate between (a) threaded**
 308 **and (b) unthreaded pins**
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 311 The temperature profile for both points was also evaluated and is shown in Figure 7. The
 312 threaded condition presented higher values of temperature during welding, mostly during the tool
 313 domain region. The threaded condition point reached almost 500 °C, while the unthreaded point
 314 reached 425 °C. The same results are also presented in Figure 8, which shows a longitudinal cross-
 315 section and the calculated temperatures during processing. Its observed the higher temperatures
 316 are around the pin, and, higher values for the threaded condition.

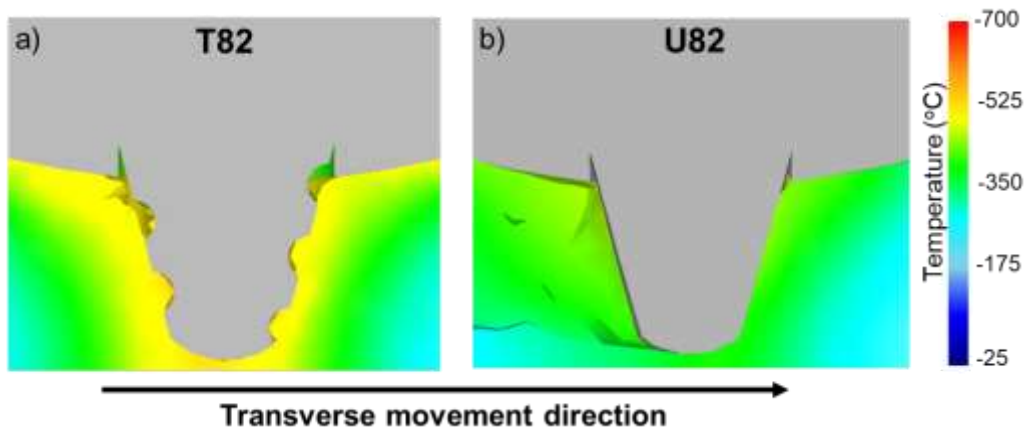
317 The result was already expected due to the higher deformation velocities applied by tool
 318 threads. The peak temperature in threaded condition coincides with the downflow movement,
 319 maximum velocity, and maximum strain-rate values. Hence, the simulated values show the
 320 geometrical changes in the tool profile corresponds to almost 15% of the maximum temperature
 321 gain during material processing, using the same parameters.

322 In that way, tool threads provided an additional downflow movement, which comes with
 323 an additional increase in velocity and strain-rate. The increase in the deformation rate leads to a
 324 temperature increase during the process. A higher temperature and material velocity promote a
 325 higher capacity to the material to flow and fill the void generated by the tool during the process.
 326 Finally, as shown in the experimental welding trials, these effects combined lead to fewer defect
 327 formation probabilities.



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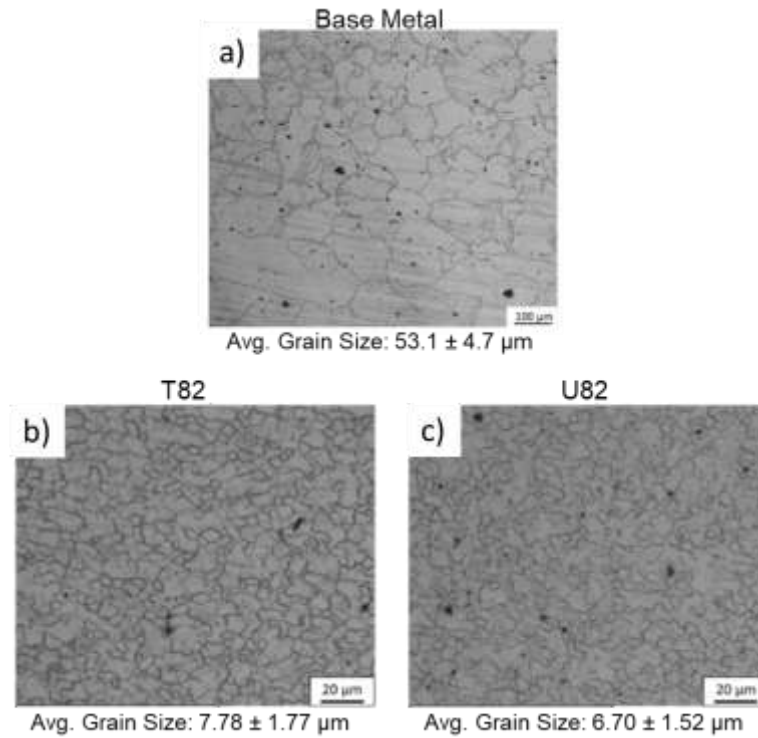
Figure 8: Simulated point tracking temperature profiles for threaded and unthreaded conditions



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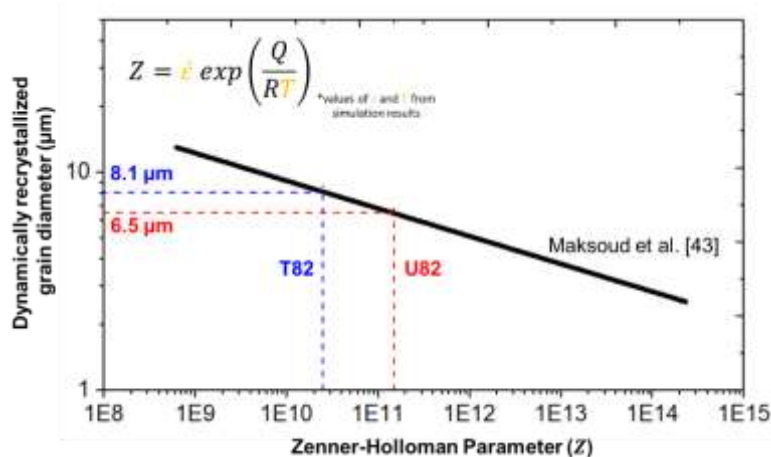
Figure 9: Longitudinal cross-section revealing the local temperatures during processing for the (a) threaded condition and (b) unthreaded condition

338 The microstructure of both conditions was compared to check if the temperature calculated
339 by the model was consistent. Figure 10 (a) shows the microstructure for base metal condition,
340 Figure 10 (b) for the T82 condition, and Figure 10 (c) for the U82 condition. Both processed
341 microstructures show an equiaxed grain structure for the average grain size of 7.78 μm for T82
342 and 6.70 μm for U82. As observed, no critical microstructural difference occurred between the
343 two conditions.



344 **Figure 10: (a) Stir zone microstructure for T82; (b) Stir zone microstructure for U82**

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348 Finally, to verify the reliability of temperature and strain rate calculated by the model, the
349 Zener-Holloman parameter (Z) was calculated using averages temperatures and strain-rate values
350 provided by the present simulation and compare to hot deformation analysis in literature. The
351 calculation results were compared to Maksoud et al., [43] work, where the author delivered a
352 Zener-Holloman (Z) and dynamic recrystallized grain size relationship for the AZ31 magnesium
353 alloy. According to results in Figure 11, the temperature and deformation behavior calculated by
354 the model provided similar values of grain sizes from those measured in Figure 10, which suggests
355 accuracy on the present model assumptions and results.



358 **Figure 11: Calculated Zener-Holloman parameters and expected recrystallized grain**
359 **diameters**

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362 **4. Conclusions**

364 In this research work, friction stir welding of AZ31B Magnesium alloy was successfully
365 simulated using the DEFORM-3D software. The simulation could calculate temperature, tool
366 response, cross-section geometry, strain, strain rate, and material flow. Regarding the
367 methodology and presented results, some conclusions are presented:
368

- 369 • The simulation indicates that pin with threads provided a material downward flow. The
370 additional movement and deformation offered by threads increased the temperature of the
371 material for the same welding parameters.
- 372 • The peak temperature in the threaded condition coincides with the point of downward
373 flow, maximum velocity, and maximum strain rate values. On the other hand, the
374 unthreaded tool generated a slight vertical movement without a significant downward
375 flow.
- 376 • Calculated values of velocity and strain-rate show much lower values compared to widely
377 employed CFD simulations. However, the present results are equivalent to the
378 experimental results found in the literature. These results also support the friction
379 coefficient assumption taken on the present work.
- 380 • Zener-Holloman parameters calculated using calculated values of deformation rate and
381 temperature shows the consistency of the values presented by the model.
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