

Enhanced Light Scattering with Energy Downshifting Using 16 nm Indium Nitride Nanoparticles for Improved Thin-film a-Si N-i-P Solar Cells

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In this work the effect of Indium nitride (InN) nanoparticles (NPs) on the performance of a-Si:H solar cells has been investigated. The average J_{sc} of InN NPs coated cells was found 6.76 mA/cm² which is 16.69% higher than the average J_{sc} of the reference cell which was 5.79 mA/cm². Average efficiency of InN NPs coated cells showed 14.16% increase from 3.32% to 3.79%. Peak EQE has increased from 44.8% at 500 nm to 51.67% at 510 nm and peak IQE has increased from 51.70% at 510 nm to 68.38% at 500 nm for InN NPs coated cell. Further study shows that EQE change is larger between 510 nm-700 nm compared to IQE change indicating a surface scattering mechanism that reduces the reflectivity. However, between 400 nm-510 nm IQE change is larger than EQE change which indicates that energy downshifting mechanism is dominating. So overall performance enhancement can be attributed to the scattering and photoluminescence properties of InN NPs that enhances absorption inside a-Si:H solar cells.

Introduction

Thin film solar cells have gained a lot of attention due to the reduced material cost. However, decreasing the thickness also decreases the absorption of light and reduces the overall current the solar cell can generate. As a result, the use of such as nanoparticles to improve the performance of thin film solar cells is a topic of increased research and interest [1-7].

There are two approaches to this work: metal nanoparticles (Au or Ag) that supports localized surface plasmons (LSPs) to increase the optical absorption length or semiconductor nanoparticles that have photoluminescence property. Recently, semiconductor nanoparticles have also received great interest for enhancing the performance of optoelectronic device such as photodetectors and thin film solar cells [8-11]. Unlike metal nanoparticles, semiconductor nanoparticles can absorb higher energy photons and emit photons of lower energy. InN NPs can absorb ultra violet (UV) light that excites electron inside the nanoparticle from ground state to excited state. This excited state is unstable so electron lose energy via non-radiative relaxation and relax

back to lower energy state. During this relaxation electron loses energy by the emission of vibrational energy. After that it further relaxes to ground state from lower energy state which involves emission of light. The energy of the emitted light is lower than the energy of the light being absorbed by InN NPs initially [12, 13].

A recent study of silicon nanoparticles in polycrystalline silicon solar cell has shown a power enhancement of 60% in the ultraviolet (UV)/blue range [8]. In the visible range, particles of 2.85 nm size showed ~10% enhancement and particles of 1 nm size showed ~3% enhancement [8]. This efficiency enhancement is mainly attributed to photoluminescence from UV/blue to visible/near infrared as a result of radiative recombination of photo-excited excitons. In this work, the effect of luminescent 16 nm InN NPs spin coated on top of thin film a-Si:H solar cells is studied.

Experiment

Indium Nitride Nanoparticles Fabrication

The generation of InN NPs was carried out using a nanosecond pulsed ND:YLF laser operated at 527 nm with pulse duration of 100 ns and a pulse repetition rate of 1 kHz. The laser output power was 16 W with pulse energy of 16 mJ. The laser beam was focused on High-Pressure Chemical Vapor Deposition (HPCVD) grown InN sample [14, 15] target containing 20 ml pure ethanol using a plano-convex lens with a focal length of 50 mm. The deposition process and optical properties of this HPCVD grown InN layer can be found elsewhere [16]. The height of liquid layer over the InN target was 5 mm. The laser ablation was carried out for 5 minutes [14]. Figure 1 shows a transmission electron microscopy (TEM) image of the laser synthesized InN NPs in ethanol solution. InN NPs are spherical and are within 3.24 nm-36 nm size range with an average size of 16 nm. The optical and spectroscopic properties of InN NPs along with their crystalline qualities are reported in detail elsewhere [14]. The Raman spectrum of InN NPs shows an E_2 peak at 496.76 cm^{-1} with a full width half maximum (FWHM) value of 5.36 cm^{-1} that is consistent with the experimental spectrum [14]. The optical absorption of InN NPs show absorption from 200 nm-1200 nm with prominent shoulders at 260 nm, absorption edge starting around 500 nm [14]. Figure 2 shows the photoluminescence spectra of InN NPs under 250 nm and 300 nm excitation. For both excitations two peaks are visible, one in UV spectrum and the other one in visible spectrum.

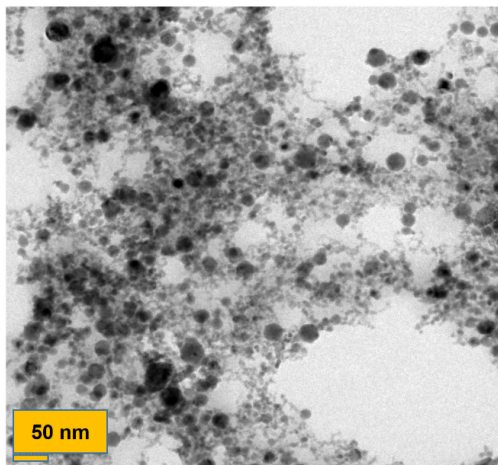


Figure 1. Transmission electron microscopy (TEM) image of laser synthesized InN NPs.

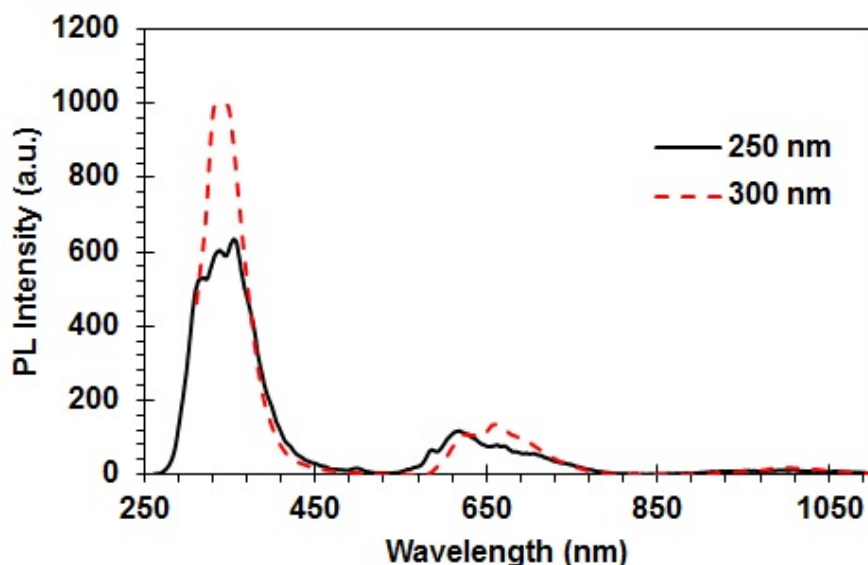


Figure 2. Photoluminescence spectra of 16 nm InN NPs. Excitation at 250 nm and 300 nm show picks in UV and visible spectrum.

Solar Cell Structure and Fabrication

In order to study the effect of InN NPs on the performance of solar cells, a-Si:H solar cells were fabricated. Figure 3 shows cross section of the fabricated a-Si:H solar cells. 20 nm heavily doped p-type a-Si, 100 nm undoped a-Si, 20 nm heavily doped n-type a-Si and 80 nm Tin doped Indium Oxide (ITO) is grown on a p⁺ type Si substrate. The p⁺ type Si substrate serves as back contact and does not contribute significantly to the electron hole pair generation. The n⁺, p⁺ and intrinsic layers were grown using the plasma enhanced chemical vapor deposition (PECVD) tool. ITO with a thickness of 80 nm was sputtered immediately afterwards to avoid interface defects using the RF sputtering tool [17-19]. More details of solar cell fabrication steps can be found in [6, 7, 20, 21]. InN NPs in ethanol solution were then spin coated on top of ITO. Different spin coating recipes were studied and the optimized recipe turned out to be 2000 rpm speed, 1000 rpm acceleration and spinning time of 60s. After the NPs coating, metal evaporation was done to define the silver finger contacts. Silver was deposited using electron-beam evaporator tool and lift off technique was used to form the contacts. In this work, results presented are from solar cells of size 0.25 cm × 0.25 cm unless otherwise stated.

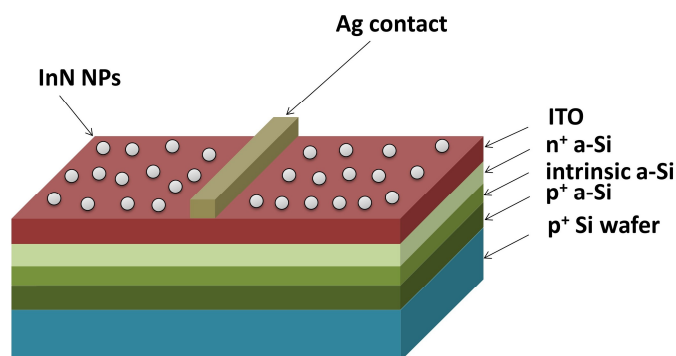


Figure 3. Layer stack of the fabricated n-i-p a-Si:H solar cell with InN NPs deposited on the top surface (Figure not to scale).

Experimental Results

The enhancement was evaluated by comparing InN NPs coated cells with uncoated reference cells. For electrical (J-V under one sun i.e. AM1.5G) measurements 5 solar cells (0.25 cm x 0.25 cm) were measured. Figure 4 shows the J-V curves of the best performing solar cells (reference cell and cell coated with InN NPs). The J_{sc} increases from 5.90 mA/cm² to 7.30 mA/cm². The efficiency also increases from 3.45% to 3.95% due to the increase in J_{sc} for this cell. The V_{oc} remains 0.89V and does not change significantly with the InN NPs.

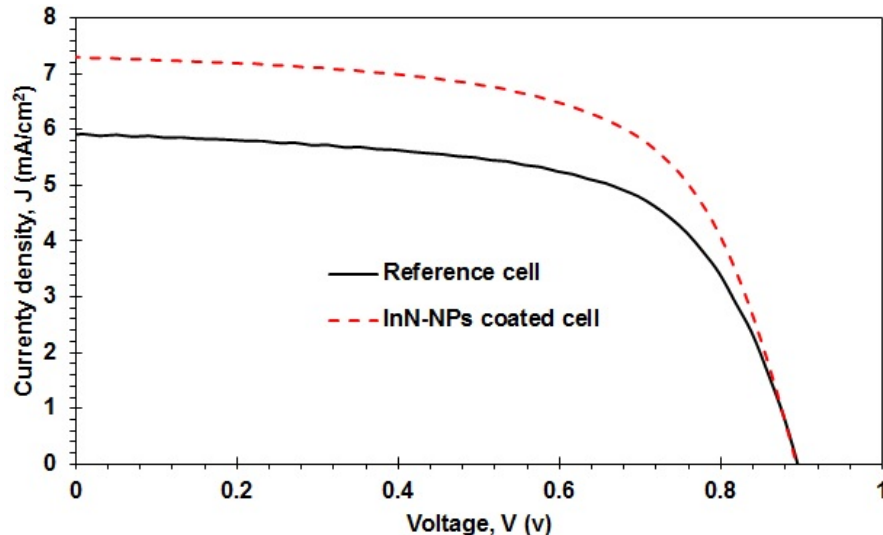


Figure 4. J-V characteristics of n-i-p a-Si:H reference cell and cell with InN NPs under one sun i.e. AM1.5G.

For further study the effect of InN NPs on a-Si:H cell, external quantum efficiency (EQE), internal quantum efficiency (IQE) and reflectance of both reference cell and InN NPs coated cells were measured. From the EQE and reflectance of the cells the following formula is used to calculate IQE:

$$\text{IQE} = \text{EQE} / (1 - \text{Reflectance}) \quad (1)$$

EQE, IQE vs. wavelength is plotted in figure 5 for a-Si:H cells. Figure 5 shows that InN NPs coated cells have improved spectral response compared to reference cell. Peak EQE increases from 44.8% at 500 nm to 51.67% at 510 nm for InN NPs coated cells and peak IQE increases from 51.70% at 510 nm to 68.38% at 500 nm. Figure 6 shows the reflectance of the InN NPs coated a-Si:H cells compared to reference cell. It shows a decrease in reflection from 500 nm onwards due to presence of InN NPs on top of the cells which means InN NPs are promoting light scattering. For a-Si:H cell, decrease in reflection is 26.14%.

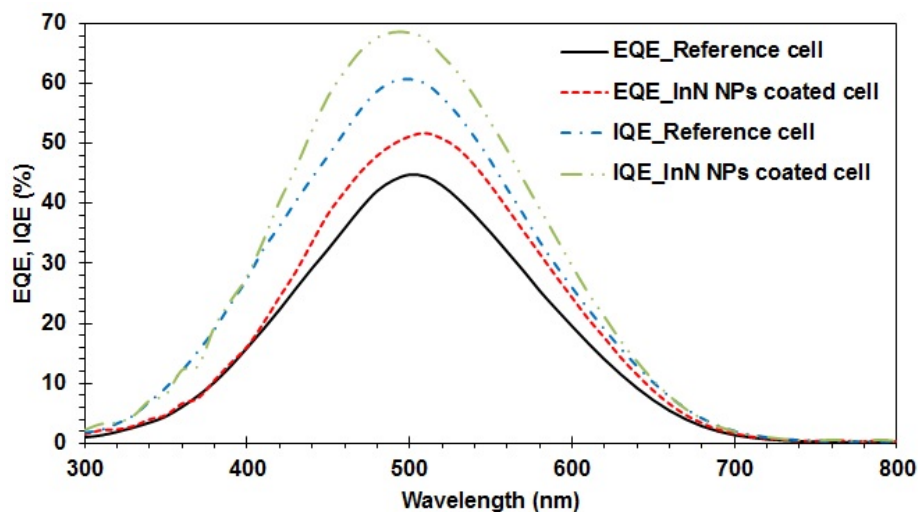


Figure 5. EQE and IQE of n-i-p a-Si:H reference cell and cells with InN NPs.

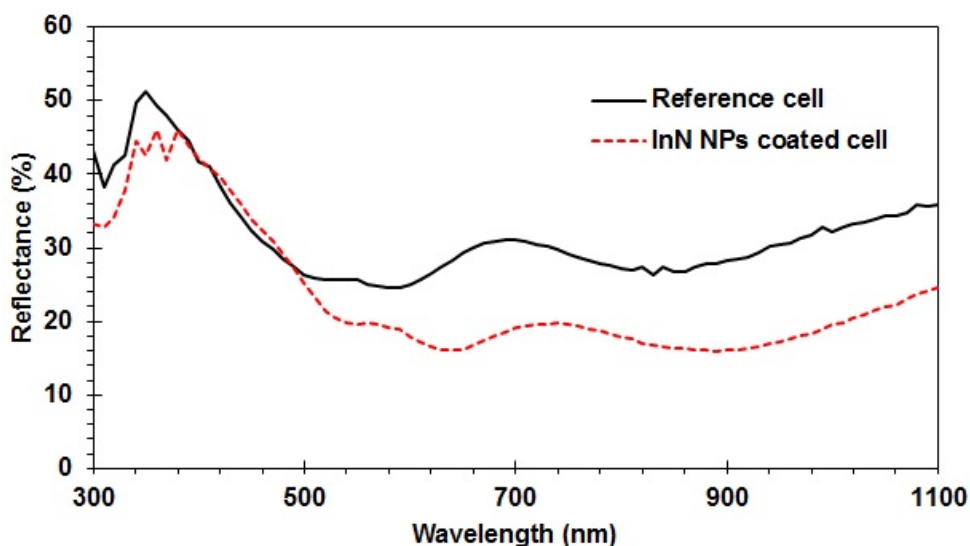


Figure 6. Reflectance of n-i-p a-Si:H reference cell and cells with InN NPs.

Table 1 summarizes main findings of this work, listing average J_{sc} , efficiency and fill factor of the reference and InN NPs coated solar cells. Average J_{sc} of InN NPs coated

cells is found to be 6.76 mA/cm^2 which is 16.69% higher than the reference cells which has an average J_{sc} of 5.79 mA/cm^2 . And the average efficiency of InN NPs coated cells is 3.79%, which is 14.16% higher than the average efficiency of the reference cells which is 3.32%. The average fill factor of InN NPs coated cell improves from 62.97% to 63.10% as well.

Table 1. Summary of results

	Short Circuit current density (J_{sc})	Efficiency (%)	Fill Factor (%)
Reference cell	5.79	3.32	62.97
InN NPs coated cell	6.76	3.79	63.10

Discussion

In order to discuss the physics of the effect of the InN NPs on the solar cells the EQE and IQE are further analyzed. Figure 7 shows change in EQE and IQE vs. wavelength for a-Si:H cell.

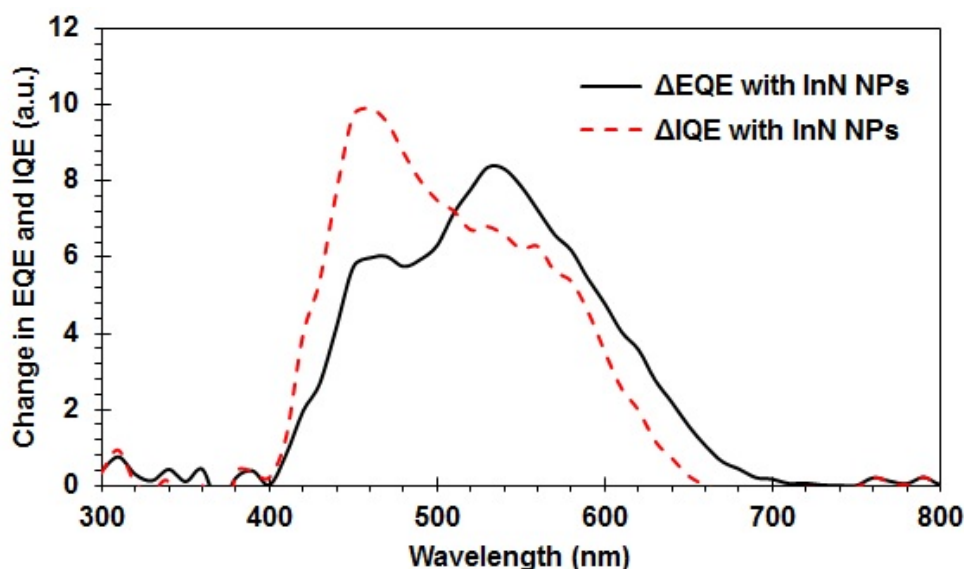


Figure 7. Change in external and internal quantum efficiency of n-i-p a-Si:H cells with InN NPs.

For InN NPs coated cell, figure 7 shows that EQE change is larger between 510 nm-700 nm compared to IQE change indicating a surface scattering mechanism that reduces reflectivity. Lower reflection of InN NPs coated cell from 500 nm onwards also supports this argument. However, from 400 nm-510 nm the IQE change is larger than EQE change which indicates an energy downshifting mechanism is dominating here. The InN NPs, with 0.7 eV band gap, can absorb a high energy photon and downshift it to a lower energy photon that can be absorbed by a-Si material. The PL intensity measurement of the InN NPs also shows a high peak in the shorter wavelength region that supports our

findings. Also, downshifting is effective in a-Si cell since the absorption depth indicates that 100 nm of a-Si material absorbs up to 540 nm of light (~ 2.30 eV). In figure 8 we have shown an energy level diagram of InN NPs coated a-Si:H cell.

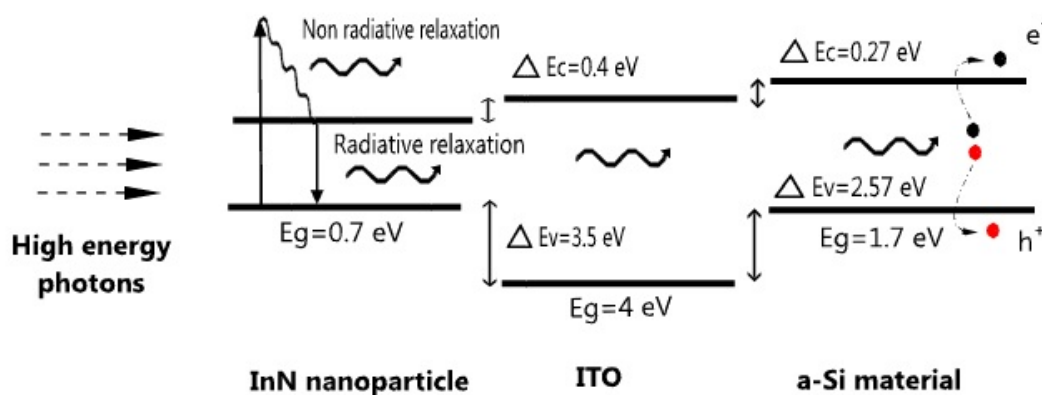


Figure 8. Energy band diagram of InN NPs coated a-Si:H cell. High energy photons excite the electrons which is then undergo a non-radiative relaxation by emitting vibrational energy and radiative relaxation by emitting photons.

Conclusion

In summary, efficiency enhancement of thin-film a-Si:H solar cell using InN NPs is reported. Exploiting the scattering and photoluminescence properties of InN NPs short circuit current density and efficiency have been improved. Compared to the reference cells, average improvement of short circuit current density and efficiency of InN NPs coated cells are 16.69% and 14.16% respectively. The cells show higher change in IQE between 400 nm-510 nm which indicates that down conversion by InN NPs is the reason for performance enhancement in this region and higher change in EQE between 510 nm-700 nm indicates scattering of light by InN NPs is the reason for performance enhancement in this region. Finally, these results highlight a simple and promising use of nanotechnology for enhancement of future thin film solar cells.

Acknowledgments

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