Synthesis of Phosphorus Included Multiwalled Carbon Nanotubes by **Pyrolysis of Sucrose**

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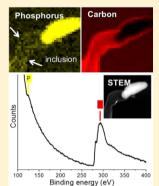
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ABSTRACT: Multiwalled carbon nanotubes (MWCNT) were synthesized by a pyrolysis route which involves a dehydration step using phosphoric acid. The resultants were found to be mostly containing amorphous carbon with scattered MWCNTs using scanning electron and transmission electron microscopy techniques. These MWCNTs were smaller in size and limited in quantity compared to the ones synthesized using other common precursors. Energy dispersive X-ray and electron energy loss spectroscopy analysis indicated the presence of phosphorus both at the MWCNT sidewalls and in the catalyst particles encapsulated inside the MWCNTs. In addition, a comparative investigation for sulfur and phosphorus inclusion to the sidewalls of MWCNTs was carried out using density functional theory calculations. The results of the computational study showed that both phosphorus and sulfur atoms prefer to bind among themselves rather than adsorbing directly on carbon nanotubes (CNT). Furthermore, cluster calculations revealed that phosphorus atoms more likely form carbonaceous clusters which result in a decrease in the number of free carbon atoms that can be used for CNT formation. Therefore, we concluded that



MWCNT growth might be hindered (promoted) in a phosphorus (sulfur) rich environment which results in needle like phosphorus containing MWCNTs.

1. INTRODUCTION

Boron, nitrogen, and phosphorus are commonly used as dopants for tailoring the electronic properties of carbon nanotubes (CNT).^{1–3} All boron doped CNTs display p-type properties, whereas nitrogen and phosphorus doped CNTs have n-type electronic structures caused by incorporated electron donor sites.⁴ Doping plays a key role for the building of electronic devices using CNTs. Hence, there are reports of CNTs with modified electronic properties to find future applications as field emitters, fuel cell electrodes, and protein immobilizators.5-7

Beside their effect on the electronic properties, changes in the morphology of CNTs are also reported due to the inclusion of dopants.^{8–10} For instance, it is reported that "bamboo" and "matchstick" morphologies encountered for CNTs are a direct result of doping by nitrogen and phosphorus, respectively.^{9,11} Moreover, substitutional inclusion of phosphorus atoms enhances the chemical affinity of CNTs toward acceptor molecules due to the favorable change in localized charge densities.¹² This may allow the use of phosphorus doped CNTs as ultrasensitive molecular sensors and catalyst supports.^{13,14} However, in situ doping during the synthesis^{13,15,16} has added challenges of overcoming the adverse effect of phosphorus on the catalytic activity of metallic catalysts and increased disorder in the carbon network.⁴

In the present study, we report a new route for the synthesis of phosphorus containing multiwalled carbon nanotubes (MWCNT) by the pyrolysis of dehydrated sucrose. Phosphoric acid (H_3PO_4) was used both as the dehydrating agent and the phosphorus source throughout the experiments. In our previous study, we investigated the effect of sulfur on the growth and the morphology of CNTs.¹⁷ It was found that the sulfur only affected the catalytic activity of catalyst particles used for the synthesis of CNTs. Although sulfur and phosphorus elements are neighbors in the periodic table, we did not find any evidence for the inclusion of sulfur to the carbon network of CNTs. In this study, we provide experimental evidence for the presence of phosphorus in the CNTs synthesized by the pyrolysis method, and also present a theoretical underpinning for this through density functional theory (DFT) calculations. Furthermore, our computational study explains the markedly different behavior observed between sulfur and phosphorus.

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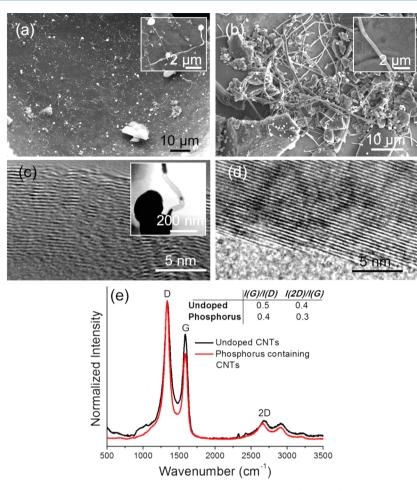


Figure 1. SEM images of MWCNTs synthesized using (a) H_3PO_4 (magnified image as inset) and (b) H_2SO_4 (magnified image as inset). HRTEM images of the MWCNTs synthesized using (c) H_3PO_4 (STEM image as inset) and (d) H_2SO_4 . (e) Raman analysis of MWCNT containing pyrolysis product synthesized using H_2SO_4 and H_3PO_4 (black and red lines represent the undoped and phosphorus containing MWCNTs, respectively.). The intensity ratios of D, G, and 2D bands are provided in Raman spectra.

2. MATERIAL AND METHODS

2.1. Experimental Procedure. MWCNTs were synthesized using the pyrolysis technique as described in our previous study.¹⁷ In the first step of MWCNT synthesis, 5 g of carbon source (sucrose) and 25 mg of catalyst (Fe, Ni, and Cr powder mixture) precursors were mixed and ground to a fine powder using a mortar. A 1:1 mixture of powdered Fe (≥99.9%) and Ni $(\geq 99.9\%)$ was used for the catalyst. Subsequently, 5 mL of concentrated H_3PO_4 (85%) solution was added to the mixture. H₃PO₄ is a strong dehydrating acid, therefore sucrose mixture gets dehydrated and thus carbonized upon the introduction of the acid. In the final step, the carbonized mixture was pyrolyzed in an atmosphere controlled furnace under flowing Ar (99.99%) atmosphere at 1330 °C for 4 h. The resultant pyrolysis product was analyzed using scanning electron microscope (SEM), transmission electron microscope (TEM), energy dispersive Xray spectroscopy (EDX), electron energy loss spectroscopy (EELS), and Raman spectroscopy.

2.2. Theoretical Calculations. We have performed DFT^{18,19} based pseudo-potential plane wave calculations using Vienna ab initio simulation package program (VASP)^{20,21} and have investigated the geometrical changes in the system by the introduction of phosphorus or sulfur atoms. The ions were described by projector-augmented-wave potentials.^{22,23} Generalized gradient approximation (Perdew–

Wang 91 type)²⁴ was employed for exchange-correlation potential. Energy cutoff for plane waves was set to 500 eV. K-points were defined by Monkhorst-Pack mesh²⁵ to carry out the Brilluoin zone integrations with the smearing parameter set as 0.08 eV. The energy convergence was within 10^{-5} eV. The systems were relaxed to their minimum energy optimal geometry. From the theoretical point of view, modulating the structure and electronic properties of CNTs is a well-studied subject. For example, doping^{12,14,26–31} and chemical functionalization^{30,31} are tried for this purpose. Among these, doping of CNTs with phosphorus atom is promising for further applications and also investigated through several computational studies.^{4,12,14,26} Note that graphene, a two-dimensional structure, can be used to describe CNTs having large diameters with small curvature.^{12,26,32} After all, we used graphene models to describe the CNTs in the computational part of the study.

3. RESULTS AND DISCUSSION

SEM image of the pyrolysis product carbonized by H_3PO_4 shows that the yielding material contains both amorphous carbon and randomly scattered tiny MWCNTs (Figure 1a). Although the diameter of these MWCNTs reached 60 nm, their lengths extended to several micrometers. In our previous study, it was shown that MWCNTs were successfully synthesized using sulfuric acid (H_2SO_4) instead of H_3PO_4 as a carbon-

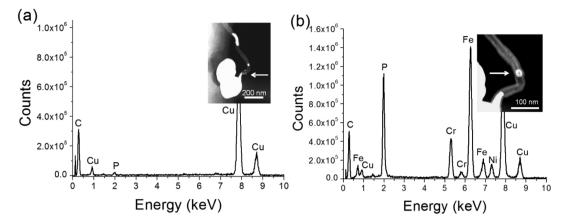


Figure 2. EDX analysis performed using TEM both from (a) the sidewalls and (b) the catalyst particle; iron, nickel, and chromium mixture, indicated the presence of phosphorus both inside the sidewalls and the catalyst particle. The Cu signal originated from TEM grid.

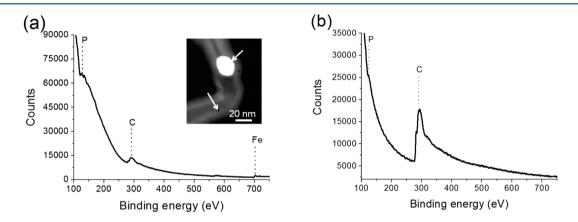


Figure 3. (a) EELS analysis on the catalyst particle trapped inside the CNT showing a high amount of phosphorus edge and (b) EELS data from the sidewalls of CNT showing a lower amount of phosphorus edge. Arrows at the inset image point where spectra in panels a and b were taken.

ization agent in the same manner, and it was found that sulfur was only present inside the catalyst particle but not in the carbon network¹⁷ (Figure 1b). In the rest of this paper, CNTs synthesized using H_2SO_4 are called as undoped MWCNTs.

When the SEM images of the pyrolysis products synthesized by H_2SO_4 and H_3PO_4 are compared, a significant difference in the size and length of MWCNTs was observed (Figure 1a,b). MWCNTs synthesized using H₃PO₄ were much shorter and smaller in diameter than the undoped MWCNTs (inset images in Figure 1a,b). Furthermore, examination of high resolution TEM (HRTEM) images shows that the graphitization of sidewalls of the MWCNTs grown in phosphorus rich environment was poor in quality compared to the undoped ones (Figure 1, panels c and d, respectively). The effect of phosphorus on the structure of CNTs was also apparent in the Raman spectrum as presented in Figure 1e. The resultant pyrolysis product from H₃PO₄ possessed relatively more structural defects as indicated by the lower I(G/D) ratio (Figure 1c). In addition, the lower I(2D/G) ratio pointed out that MWCNTs had a poorly crystallized structure compared to the undoped MWCNTs which was consistent with the HRTEM images.

According to SEM images, the observed differences between undoped and phosphorus containing MWCNTs were not limited to their size and structure, but their morphology was also strikingly differing. MWCNTs synthesized using H_3PO_4 generally emerged from a small catalyst particle which is significantly larger than the diameter of MWCNT, and the diameter was found to narrow through the "tail" resembling a needle shape with the catalyst particle acting as the "head of the needle" (inset of Figure 1a). Similar growth behavior was also observed by other research groups, and it was attributed to the phosphorus inclusion.^{11,33,34} By developing this idea further, we also address the shorter and smaller needle like CNT growth with phosphorus inclusion. Phosphorus content could deactivate the large portion of catalyst particle which can result in carbon coating on the catalyst surface and left only a small active part for the CNT growth. In order to develop a better understanding for the role of phosphorus in the growth, EDX analysis using TEM was performed on the MWCNT synthesized using H₃PO₄. A poor signal for phosphorus was detected from the sidewalls of MWCNT (Figure 2a), whereas there was a significant amount of phosphorus signal coming from the catalyst (Figure 2b). Also, EDX spectra from the catalyst shows that it is a mixture of iron and nickel trapped into the hollow part of a MWCNT (inset of Figure 2b). The phosphorus concentration inside the active metal catalyst trapped into the CNT was found to be around 13% (at.) by quantifying the EDX spectrum. Note that the eutectic point for this percentage of phosphorus is at 1048 °C.35 Therefore, in our growth temperature (1300 °C), the catalyst formed eutectic with phosphorus (Fe/P) at liquid state. The hollow parts of CNTs were partially filled by the catalyst due to its liquid state during the growth.

Phosphorus is known to lower the catalytic activity of metal catalysts used in the oxidative alkanes conversion.³⁶ In addition,

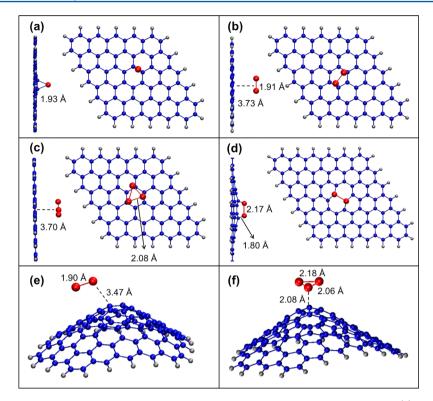


Figure 4. Optimized structures of various phosphorus atoms interacting with the carbon network. Adsorption of (a) single phosphorus atom in the bridge position, (b) two and (c) three phosphorus atoms over the graphene layer, (d) two phosphorus atoms on the double vacancy of the graphene layer, and (e) two and (f) three phosphorus atoms over the kink structure. (Small blue and large red dots represent carbon and phosphorus atoms, respectively.)

Terrones et al. reported that the iron phosphide particles (Fe_2P and Fe_3P) were not active compared to the metallic particles in CNT growth.⁸ On the contrary, phosphorus was also reported as a cocatalyst for the sequential growth of CNTs.¹¹ In this study, we found that phosphorus did not fully hinder the growth yet it yielded tiny MWCNTs compared to the sulfur case.¹⁷ Since the catalyst is in a liquid state during the synthesis, the catalyst surface would be refreshed, and new active growth sites would appear. Hence, this might keep the growth continuous even if the phosphorus is present in the media.

Although EDX analysis showed a significant signal for phosphorus, it is not enough to claim the presence of phosphorus inside the sidewalls. In order to provide further evidence, EELS analysis was performed on the MWCNTs synthesized using H_3PO_4 (Figure 3). According to the spectrum in Figure 3a, there is an apparent phosphorus edge for the catalyst which was trapped inside the hollow part of the MWCNT along with the Fe edge at 720 eV. This indicated that there is phosphorus content inside the Fe catalyst which supports the outcome of the EDX spectrum (Figure 2b). Furthermore, EELS data shown in Figure 3 pointed out that phosphorus was not only found at the catalyst particle but also present inside the sidewalls of CNTs.

In order to provide a further understanding of the phosphorus-carbon network interaction at the atomistic scale, we have performed DFT calculations. We needed to place phosphorus atoms in various positions over sidewalls of MWCNTs to describe their interaction. Since the diameter of MWCNTs we synthesized was around 30 nm, it was very difficult to model such large structures from first principle calculations. Fortunately, a larger diameter means smaller curvature so we could represent them by planar graphene

layers. A 6×6 graphene unit cell was used to simulate the outmost layer of CNT and then phosphorus atoms were brought in contact with it. Moreover, there were defected and bent regions as well as the junctions on the synthesized CNTs. Therefore, defected graphene and kink structures were also included in the investigation of the phosphorus interaction with CNTs. We placed different number of phosphorus atoms at different possible positions over these graphene layers. For several initial geometries, we performed structure optimization by minimizing total energy. Then, we calculated the binding energy (E_{binding}) of these final structures as follows:

$$E_{\rm binding} = E_{\rm total} - E_{\rm grap} - n_{\rm P}\mu_{\rm P}$$

where E_{total} is the total energy of the system, E_{grap} is the energy of the graphene layer, n_{p} is the number, and μ_{p} is the chemical potential of phosphorus atoms. In addition, we have also calculated the formation energy ($E_{\text{formation}}$) of the structures by changing the reference from the separated phosphorus atoms to corresponding phosphorus molecules such as dimer, trimer, etc., i.e., replacing $n_{\text{p}}\mu_{\text{p}}$ by E_{molecule} .

Figure 4 summarizes the selected optimized structures with the highest binding energies which are provided in detail in Table 1. A single phosphorus atom was found to be adsorbed over two carbons in the bridge position as seen in Figure 4a. When a second phosphorus atom was introduced, they tended to attract each other to form a phosphorus dimer (Figure 4b). Hence, the interaction between the phosphorus and carbon network was weakened and the carbon—phosphorus distance was increased. In a similar manner, a trimer structure was formed in the presence of the third phosphorus atom and moved away from the carbon surface (Figure 4c). This indicates that the phosphorus—phosphorus interaction is stronger than

Table 1. Binding Energies for Optimized Configurations

	_		-			
	phosphorus		sulfur			
system	E _{binding} (eV)	$egin{array}{c} E_{ m formation} \ (eV) \end{array}$	E _{binding} (eV)	$E_{ m formation} \ (eV)$		
Graphene						
single atom	-1.66	-1.66	-1.69	-1.69		
dimer (physisorption)	-9.02	-0.06	-6.49	-0.22		
dimer (chemisorption)	-6.58	2.37				
separated atoms	-3.77	5.18	-3.43	2.84		
trimer	-13.52	-0.12	-10.53	0.30		
chain	-12.61	0.80	-10.41	0.42		
Grapl	nene with V	acancy				
single vacancy + single atom	-9.86	-9.86	-8.08	-8.08		
double vacancy + single atom	-7.50	-7.50	-5.25	-5.25		
double vacancy +2 atoms (opposite sides)	-12.42	-3.46	-9.71	-3.44		
double vacancy +2 atoms (same sides)	-12.57	-3.61	-7.62	-1.35		
Kink Structure						
on top of a C–C bond in pentagon	-2.68	-2.68	-2.78	-2.78		
on top of a C–C bond away from pentagon	-2.72	-2.72	-2.75	-2.75		
dimer (physisorption)	-9.05	-0.09				
dimer (single atom chemisorption)	-6.80	2.15	-6.89	-0.62		
dimer (chemisorption)	-8.44	0.51	-6.40	-0.12		
trimer (physisorption)	-13.63	-0.23	-10.53	0.30		
trimer (chemisorption)	-14.44	-1.05				
chain	-12.59	0.81	-11.08	-0.24		
dimer (physisorption) + single atom	-11.90	1.50				
dimer (chemisorption) + single atom	-11.67	1.73	-9.54	1.30		

the carbon-phosphorus interaction; thus, phosphorus atoms could not be incorporated into the sidewalls of CNTs. However, when there is a defected carbon network, such as double carbon vacancy shown in Figure 4d, phosphorus atoms (dimer) saturated the dangling bonds of carbon atoms around the vacancies. Apart from these, introduction of pentagon rings led to the bending of graphene structures as shown in Figure 4e,f. Bent regions or junctions of CNTs can be modeled by these kink structures. Interaction of phosphorus atoms with these structures was similar to those of the planar cases discussed previously. In conclusion, in the absence of defects on the graphene layer, phosphorus atoms have a tendency to aggregate and get away from the carbon network.

In order to further quantify the phosphorus interaction with CNTs, we compared these results to the sulfur containing cases since we observed differences in CNT morphologies even though a similar synthesis technique was conducted using sulfur containing precursor as reported in our previous study.¹⁷ In that study, similar DFT results have been obtained for the sulfur–carbon network.¹⁷ This indicates that phosphorus and sulfur atoms do not join to the surface of a complete CNT, and there is not a significant difference in their behaviors toward a vacancy as well. Therefore, in order to explain the effect of phosphorus (sulfur) on CNTs, we need to investigate the details of the growth by checking different sized phosphorus–carbon and sulfur–carbon containing clusters.

Our cluster calculations were not related directly to CNT but about the environment during the growth. We calculated the stabilities of $C_x P_y$ and $C_x S_y$ clusters having five atoms at most. We formed several initial structures with initial bond lengths around 2 Å. Then, we discussed energetics of the optimized structures (Table 2) which are presented in Figure 5. The

Table 2. Total Energies for the Systems Shown in Figure 5

Figure 5 system $X = P$ $X = S$ a C_1X_1 -8.60 -10.20 b C_1X_2 -15.49 -16.56 c C_2X_1 -16.86 -16.12 d C_1X_3 -20.75 -18.14 e C_2X_2 -23.34 -20.52 f C_3X_1 -25.33 -23.67 g C_1X_4 -27.01 -23.34 h C_2X_3 -28.92 -27.02 i C_3X_2 -32.27 -32.85 j C_4X_1 -33.65 -31.30 k C_1X_4 -21.37 1 1 C_2X_3 -26.75 m C_4X_1 -29.22 n C_4X_1 -28.95 dimers E_{total} (eV) -28.95			$E_{\rm total}~({\rm eV})$		
b C_1X_2 -15.49 -16.56 c C_2X_1 -16.86 -16.12 d C_1X_3 -20.75 -18.14 e C_2X_2 -23.34 -20.52 f C_3X_1 -25.33 -23.67 g C_1X_4 -27.01 -23.34 h C_2X_3 -28.92 -27.02 i C_3X_2 -32.27 -32.85 j C_4X_1 -33.65 -31.30 k C_1X_4 -21.37 l C_2X_3 -26.75 m C_4X_1 -29.22 n C_4X_1 -28.95 dimers E_{total} (eV) C_2 -9.00 P_2 -9.02 -9.02 -9.02	Figure 5	system	X = P	X = S	
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	a	C_1X_1	-8.60	-10.20	
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	b	C_1X_2	-15.49	-16.56	
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$	с	C_2X_1	-16.86	-16.12	
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$	d	C_1X_3	-20.75	-18.14	
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	e	C_2X_2	-23.34	-20.52	
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$	f	C_3X_1	-25.33	-23.67	
$\begin{array}{ccccccc} i & C_{3}X_{2} & -32.27 & -32.85 \\ j & C_{4}X_{1} & -33.65 & -31.30 \\ k & C_{1}X_{4} & -21.37 \\ l & C_{2}X_{3} & -26.75 \\ m & C_{4}X_{1} & -29.22 \\ n & C_{4}X_{1} & -28.95 \\ \hline dimers & E_{total} (eV) \\ \hline C_{2} & -9.00 \\ P_{2} & -9.02 \end{array}$	g	C_1X_4	-27.01	-23.34	
$\begin{array}{ccccccc} j & C_4 X_1 & -33.65 & -31.30 \\ k & C_1 X_4 & -21.37 \\ l & C_2 X_3 & -26.75 \\ m & C_4 X_1 & -29.22 \\ n & C_4 X_1 & -28.95 \\ \hline dimers & E_{total} (eV) \\ \hline C_2 & -9.00 \\ P_2 & -9.02 \\ \end{array}$	h	C_2X_3	-28.92	-27.02	
k C_1X_4 -21.37 l C_2X_3 -26.75 m C_4X_1 -29.22 n C_4X_1 -28.95 dimers E_{total} (eV) C_2 -9.00 P_2 -9.02	i	C_3X_2	-32.27	-32.85	
$\begin{array}{cccc} l & C_{2}X_{3} & -26.75 \\ m & C_{4}X_{1} & -29.22 \\ n & C_{4}X_{1} & -28.95 \\ \hline dimers & E_{total} \ (eV) \\ \hline C_{2} & -9.00 \\ P_{2} & -9.02 \end{array}$	j	C_4X_1	-33.65	-31.30	
$\begin{array}{cccc} m & C_{4}X_{1} & -29.22 \\ n & C_{4}X_{1} & -28.95 \\ \hline dimers & E_{total} \ (eV) \\ \hline C_{2} & -9.00 \\ P_{2} & -9.02 \end{array}$	k	C_1X_4		-21.37	
n C_4X_1 -28.95 dimers E_{total} (eV) C_2 -9.00 P_2 -9.02	1	C_2X_3		-26.75	
dimers E_{total} (eV) C_2 -9.00 P_2 -9.02	m	C_4X_1		-29.22	
$C_2 = -9.00$ $P_2 = -9.02$	n	C_4X_1		-28.95	
P ₂ -9.02	dimers	$E_{\rm total}$ (eV)			
-	C ₂	-9.00			
S648	P_2	-9.02			
0.10	S ₂	-6.48			

carbon-phosphorus and carbon-sulfur bond lengths (~1.55 Å) measured from their dimer structures, shown in Figure 5a, were used as references for the growth of their clusters. Carbon monosulfide are known as the metastable intermediated phase.³⁷ When second phosphorus or sulfur atoms were introduced (Figure 5b), the bond lengths were increased slightly to form a chain with carbon atom in the center instead of a triangular structure. Carbon-carbon bond lengths in Figure 5c show that the phosphorus atom pulls the electron of the carbon atom in the middle more than the sulfur atom, so the carbon-phosphorus interaction is a strong one. As we increased the total number of atoms, various planar geometries appeared as their minimum energy configurations (Figure 5dj). In our $C_x S_v$ calculations, we have observed additional configurations with close energies to the optimum ones (Figure 5g,h,j) as presented in Figure 5k-n. In these, larger clusters split into smaller ones including carbon monosulfide frequently. Such behavior was not observed for $C_x P_y$ cases in which atoms ended up intact clusters. These findings would provide an explanation to our following experimental results. First, phosphorus atoms make cluster with carbon atoms decreasing the number of free carbon that can be used for CNTs to be constructed. Therefore, CNTs cannot grow much and they become tinier and low in number. The situation is just opposite for $C_x S_y$ systems because they tend to break up; hence, the number of free carbon atoms increases which might be the cause of the promotion of the CNT growth.¹⁷ Second, when a phosphorus atom enters into the structure of a wall during the growth, it may tend to form a cluster, and this would result in a local bump inside the CNT walls. Furthermore, this might be preventing the structure to form a well graphitized surface. Third, since clusters with sulfur atoms have a tendency to split, sulfur clusters are rarely observed inside the CNT walls compared to the phosphorus case. In conclusion, CNTs synthesized in the phosphorus rich environment are expected

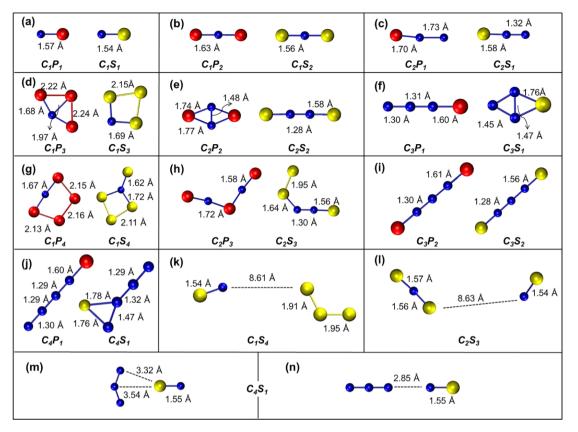


Figure 5. Various $C_x P_y$ and $C_x S_y$ clusters with (a-j) optimized minimum energy geometry and (k-n) additionally observed $C_x S_y$ structures with close energy to the optimum ones. (Small blue, large red, and large yellow dots represent carbon, phosphorus, and sulfur atoms, respectively.)

to be smaller in size, lower in quantity, and poorly graphitized in contrast to the ones synthesized in the sulfur rich environment. Furthermore, our findings support that phosphorus atoms more likely form a carbonaceous cluster over the catalyst particle.

4. CONCLUSIONS

Needle like phosphorus containing MWCNTs were synthesized by the pyrolysis method using H3PO4 as a dehydrating agent. According to the SEM and TEM images, these MWCNTs were small in size, low in quantity, and poorly graphitized with respect to the ones synthesized using sulfur containing precursors. The EDX and EELS analysis provided strong evidence for the phosphorus inclusion to the sidewalls of MWCNTs, besides it was present at the catalyst particle inside the hollow part of the CNT. These experimental findings were further supported by DFT calculations. Accordingly, both phosphorus and sulfur atoms prefer to bind among themselves rather than adsorbing directly on CNTs. Indeed, phosphorus atoms more likely form a carbonaceous cluster over the catalyst particle indicated by DFT calculations of C_xP_y and C_xS_y cluster systems. Phosphorus atoms make cluster with carbon atoms decreasing the number of free carbon atoms that can be used for CNT formation and hinders the growth. Whereas, sulfur atoms tend to disassociate the $C_x S_y$ hence the number of free carbon atoms increases that promotes the CNT growth. In conclusion, phosphorus inclusion is realized on sidewalls and/ or inside the MWCNTs; however, these are smaller in size, lower in quantity, and poorly graphitized as compared to the undoped MWCNTs synthesized in the sulfur rich environment.

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Notes

The authors declare no competing financial interest.

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