



## Research papers

Binary composite (TiO<sub>2</sub>-Gr) based nano-enhanced organic phase change material: Effect on thermophysical properties

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## ABSTRACT

The latent heat storage technology using Phase Change Materials (PCMs) has recently been extensively utilized in energy conservation and management to reduce energy consumption. To improve the thermal conductivity of PCMs, they have been incorporated with nanoparticles. In this article, we report, Titanium dioxide-Graphene (TiO<sub>2</sub>:Gr) binary composite (1 wt% TiO<sub>2</sub>: 0.1, 0.5, 1 and 2 wt% of Graphene (Gr)) with Paraffin wax (PW) to improve the thermophysical properties added with sodium dodecylbenzene sulphonate (SDBS) as surfactant. Ultraviolet-visible spectrometer (UV-VIS), Fourier transform infrared spectroscopy (FT-IR), Differential scanning calorimeter (DSC), Thermogravimetric analyzer (TGA), Field Emission Scanning Electron Microscopy (FESEM) and Thermal property analyzer (TEMPOS) were used for material characterizations. The latent heat capacity of the PW/Titanium oxide (TiO<sub>2</sub>) composite and the PW/TiO<sub>2</sub>-Gr binary composites were improved by 8.62% and 10.02%, respectively, in comparison to base PCM. The thermal conductivity of the composite PCMs with PW/TiO<sub>2</sub>-1.0 and PW/TiO<sub>2</sub>Gr-1.0 is 120% and 179% higher than base PW. The FT-IR spectra demonstrated no chemical interaction between the PW and the nanoparticles. TGA results presented improved thermal stability by integration of the TiO<sub>2</sub>-Graphene particles into the matrix of paraffin wax. The light transmission of the prepared composite was reduced by 58.30% related to base PW, resulted increased light absorption and, subsequently, enhanced photothermal conversion. The composite's improved thermal conductivity and enthalpy make it a strong contender for use in TES and solar photovoltaics thermal systems.

## 1. Introduction

Energy security and climate change are the two most critical issues in the world today, and they are intimately connected. On the one hand, today's global energy system is mainly reliant on conventional fossil fuels, which are rapidly depleting; however, the emissions from using this fuel constitute a significant contributor to climate change [1,2]. Using surplus heat from solar and other energy sources, thermal energy storage systems (TES) developed one of the required means for dropping dependence on fossil fuel and leading to more effective and environmentally sustainable energy storage [3]. Phase change materials (PCM's)

recommended as possible energy storage mediums have gotten much attention recently. When PCM absorb and release significant quantities of latent heat storage (LHS) during phase change, the technology allows for storage, transportation and use of thermal energy. The PCM can be classified as Organic, Inorganic and Eutectics based on material properties. Organic PCMs have become standard for energy storage in recent years due to their high thermal stability. Some PCMs have a higher energy density, which means they can store and release a tremendous amount of energy [4]. However, some issues with PCM's, such as low thermal conductivity (TC) and lower thermophysical properties (TPP), will arise in use. It is essential to integrate nanoparticles into PCMs to enhance their TC [5]. Additionally, nanoparticles could increase the

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### Nomenclature

<b>CuO</b>	Copper oxide nanoparticles
<b>DSC</b>	Differential scanning calorimetric
<b>EDX</b>	Energy-dispersive X-ray spectroscopy
<b>FESEM</b>	Field emission scanning electron microscope
<b>FT-IR</b>	Fourier transform infrared spectroscopy
<b>Gr</b>	Graphene nanoparticles
<b>LHS</b>	Latent heat storage
<b>MWCNT</b>	Multi-wall carbon nanotubes
<b>NEOPCM</b>	Nano-enhanced organic Phase change materials
<b>NEPCM</b>	Nano-enhanced phase change materials
<b>PCM</b>	Phase change materials
<b>PV</b>	Photovoltaics
<b>PVT</b>	Photovoltaic thermal systems
<b>PW</b>	Paraffin wax
<b>PW/TiO<sub>2</sub></b>	Paraffin titanium oxide composite
<b>PW/TiO<sub>2</sub>Gr</b>	Paraffin Titanium oxide-Graphene binary composite

<b>SST</b>	solid to solid transition
<b>SLT</b>	solid to liquid transition
<b>SDBS</b>	Sodium dodecylbenzene sulphate
<b>SA</b>	stearic acid
<b>TC</b>	Thermal conductivity
<b>TGA</b>	Thermal gravimetric analyzer
<b>TiO<sub>2</sub></b>	Titanium oxide
<b>TPP</b>	Thermo-physical properties
<b>UV-VIS</b>	Ultra-violet visible spectroscopy

### Greek letters

$\alpha$	Absorptance
$\beta$	Temperature coefficient
$\rho$	Density kg/m <sup>3</sup>
$\eta$	Efficiency
$H$	Photovoltaic efficiency
$\mu$	Micron
$\phi$	Concentration ratios (%)

surface area to volume ratio and decrease the solidification and melting rate.

High conductive nanoparticles dispersed in the PCM offer a promising solution for improving TC and sustaining thermal efficiency [6]. Therefore, Nano-enhanced organic phase change materials (NEOPCM's) have been found to be some of the most promising energy storage materials, surpassing the limitations of PCM's [7–9]. The TC of NEOPCM is influenced by nanoparticles' inherent TC and the PCMs and nanoparticles interactional compatibility [10]. Carbon-based nanotechnology such as carbon nanotubes, oxides and metallic nanoparticles have already been used in heat transfer products, which can be combined to increase their performance [11], for example, using dispersants with PCMs to improve (TC) and heat transfer [12]. Due to their lower energy charge/discharge ratios, the need for the PCM to improve their TC's is obvious [13]. Therefore, nanoparticles that improve their TCs or that help influence composite PCMs are being sought to generate high energy charging/discharging speed.

Paraffin with polyaniline nanocomposite was created by dispersion with simple sonification [14]. The improved dispersion of conducting polymers-based nanocomposite gave a 15.9% increase in LHS. The PCM will store additional thermal energy when the latent heat is increased. The TC improved by 20.4% resulted increases charging performance because decreased interfacial resistance results in increased efficacy of PCM. The thermal energy storage study was directed for thermal efficiency using carboxyl functionalize Graphene doped with paraffin wax (PW) [15]. Volume concentrations were increased by 0.25 to 1%. By using doped PCM materials, it was observed that an enhancement in TC result increased the charging time decrease rate up to 20.0% to 30.0% due to variation in HTF inlet temperature. When the charging time was decreased by 30% to 50% using 1% composite in Paraffin, a considerable quantity of energy was stored concisely. The carbon nanoparticles combined with graphite speed up the solidification process [16]. The melting process for the nano enhanced PCM was increased by 21% as related to PW, while the composition of carbon foam/PW improved by 31%. That being said a 42% increase in TC for the composite PCM. The reported work [17] demonstrated that expanded graphite (EG)-paraffin wax composite PCMs solar absorption capacity increased by introducing small quantity of carbon nanotubes (CNTs) as the light absorbent. The produced composite PCMs containing 5% EG and 2% CNTs exhibit exceptional thermal conductivity a high solar absorption capacity, and a high latent heat capacity. Additionally, these enhanced thermal and optical characteristics are stable after 1000 melting-freezing cycles. Paraffin with reduced graphene oxide and zirconium carbide was created for enhanced light transmission and thermophysical properties

[18]. The composites PCMs with varying amounts of zirconium carbide exhibited outstanding photo absorption, thermal storage capacity, and heat transmission properties. When doped with 0.01 wt% zirconium carbide, the photothermal conversion efficiency increases to 81%. Composite PCMs found to be maximum thermal conductivity of 121% greater than PW. Chen et al. [19] examined the impact of CuO nanoparticles' light absorption capabilities on PW. The composites were prepared with a mass fraction of CuO varying from 0.01 to 0.1%, and found that the composites increased the steady state temperature by 2.3 times. The ability to absorb light was increased when the mass fraction of CuO nanoparticles increased. A novel method was used to make a nanoparticle enhanced PCM stearic acid SA TiO<sub>2</sub> [20]. The finding showed they had higher TC values compared with base PCM. NEPCM composite was made with weight concentration 0.09, 0.26, 0.33 and 0.36 wt percentage. The highest TC was reported as 27% compared with pure SA PCM. The work reported is directly associated with using different combinations of Al<sub>2</sub>O<sub>3</sub>-CuO hybrid nanoparticles (75:25, 50:50 and 25:75) [21]. These hybrid nanoparticle combinations were integrated with varied volume concentrations of 0.5, 1, 2, and 3% and then subjected to TC and melting point tests utilizing the KD-2 Pro and digital melting point equipment, respectively. The inclusion of hybrid nanoparticles improved the TC and melting temperature (MT) of the PCM composite, according to the study's findings. Sun et al. [22] characterized the TPP of PW and nanocarbon materials. The paraffin/nanocarbon of three different samples, 0.02, 0.06 and 0.10%, were characterized. The highest increment in NEPCM TC was found at 0.06, and 0.10% were 25% and 12%, respectively. It was also pointed out that the 20% decrease in the melting temperature by adding nanocarbon to Paraffin PCM. Paraffin were doped at the level of 0, 0.5, 1.0, 1.5 and 2.0% with TiO<sub>2</sub> to improve thermal response [23]. Results demonstrated improved results of latent heat as compared with paraffin LHS. The increment in the TC values 33, 66, 84 and 101% was also seen by adding the TiO<sub>2</sub> concentration without disturbing the chemical edifice. Binary Graphene-Silver (Gr-Ag) nanomaterial used for the first time with PW PCM to improve its TPP [24]. By 0.3% of binary Gr-Ag nanomaterial loaded in PCM, the most significant increase in TC as well as the improvement in LHS was around 40% noticed. The examination of the thermal performance of solid-solid PCM with binary composite and multi-nanoparticles addition composite was evaluated by [25]. The thermal behavior of improved neopentyl glycol (NPG) composites has been reported and discussed in various cases. It was found that the influence of the element concentration ratio in a multi nanoparticles combination on the thermal performance of NPG. SiO<sub>2</sub>, Al, and TiO<sub>2</sub> had higher TC and LHS.

While numerous studies have focused on preparing NEPCM with increased TC of composites, none of them had addressed the TiO<sub>2</sub> and Graphene binary combination to prepare PW-TiO<sub>2</sub>Gr composite. The novelty lies in the fact that the paraffin TiO<sub>2</sub>-Graphene is the first time to the authors knowledge that composite can be made in the current study. This study aims to investigate the impact of TiO<sub>2</sub>-Graphene for energy storage application on the TPP of the paraffin RT47. According to the literature review, TC and LHS of composites are critical when it comes to using energy storage. The TC allows heat from the materials to go through the PCM more quickly while LHS is the quantity of energy it can hold before it overheats significantly. Therefore, we have performed a comprehensive characterization of the materials in this study using techniques such as FESEM, EDX, FTIR, UV-Vis, and TGA. This allows for a detailed analysis of their morphological, chemical compatibility, thermal stability and other properties. Further, a wide temperature range is included in medium temperature-based applications using the produced material, such as cooling of PV (40 °C–70 °C), PVT (40 °C–90 °C) [26,27], electronic devices (40 °C–80 °C) [28], automotive battery cooling (37 °C–67 °C) [29], hot water application (40–65 °C).

## 2. Materials and methods

### 2.1. Materials

Paraffin RT47 Organic PCM with a melting point around 47–48 °C and excellent phase change enthalpy of 160 kJ/kg is obtained from RUBITHERM ®Germany. TiO<sub>2</sub> nanoparticles having particles size of 25 nm purchased from Sigma-Aldrich Company and Graphene nanoparticles 8 nm-sized obtained from Graphene Supermarket. Surfactant SDBS was purchased from Sigma-Aldrich company Germany. The physiochemical properties of the materials used are listed in Table 1.

### 2.2. Preparation of NEOPCM and binary NEOPCM

A two-step technique achieved the preparation of TiO<sub>2</sub>-Graphene binary nanoparticles dispersed in RT47 Paraffin organic PCM, as seen in Fig. 1. The appropriate weight of base PCM, TiO<sub>2</sub> nanoparticles, and SDBS surfactant (0.1%, 0.5%, 1%, and 2%) measured by analytical microbalance. Paraffin RT47 PCM was melted at 70 °C by using a hotplate (RCT, BASIC IKA), then TiO<sub>2</sub> nanoparticles and SDBS was added with melted base PCM and sonicated for 30 min by using probe-sonicator (Model; FS-1200N) for proper mixing of composite.

The composite was permissible to cool at ambient temperature. The same method is used to make binary composites NEOPCM Paraffin/TiO<sub>2</sub>-Graphene (PW/TiO<sub>2</sub>Gr). The composite 1.0-PW/TiO<sub>2</sub> was melted at 70 °C by using a hotplate (RCT BASIC IKA). Graphene nanoparticle was added at a different wt per cent (0.1%, 0.5%, 1% and 2%). Probe sonication has been done for proper mixing of NEOPCM binary composite. Then it was allowed to let the composition reach room temperature. The various concentrations of RT47 PW with TiO<sub>2</sub> are represented as PW, PW/TiO<sub>2</sub>-0.1, PW/TiO<sub>2</sub>-0.5, PW/TiO<sub>2</sub>-1.0, and PW/TiO<sub>2</sub>-2.0 for weight concentrations of TiO<sub>2</sub> at 0%, 0.10%, 0.50%, 1.0%, and 2.0%, correspondingly. Likewise, binary composites comprising RT47-TiO<sub>2</sub>

added Graphene are signified as PW/TiO<sub>2</sub>Gr-0.1, PW/TiO<sub>2</sub>Gr-0.5, PW/TiO<sub>2</sub>Gr-1.0, and PW/TiO<sub>2</sub>Gr-2.0 for Graphene weight concentrations of 0.1, 0.5, 1%, and 2.0%.

### 2.3. Characterization methods

To investigate the morphological, chemical, and thermal characteristics of NEOPCM, a variety of facilities were utilized. The morphology of the PW base, PW/TiO<sub>2</sub> and (PW/TiO<sub>2</sub>-Graphene) composite was studied with the help of Field Emission Scanning Electron Microscopy (FE SEM). The scanning electron microscope was equipped with an Energy Dispersive X-ray Spectroscopy (EDX) system, which was utilized to determine the elemental compositions of the samples directly. An FTIR technique with a wavenumber range of 400–4000 cm<sup>-1</sup>, a spectral resolution of 0.1 cm<sup>-1</sup>, and a wave number precision of 0.01 cm<sup>-1</sup> were used to determine the chemical compositions and functional groups. The ATR-RX1 spectrometer was used to create comprehensive FT-IR spectra curves in transmittance mode. Under an N<sub>2</sub> environment and at a heating rate of 10 °C/min, composites' mass loss and thermal stability anticipated using the Thermal gravimetric analyzer (TGA 400 Perkin Elmer 4000). The LHS and MP of material properties were measured using a DSC-1000 M (Linseis Inc., USA). The DSC equipment was calibrated with the help of the Linseis TA software. We conducted a TC test on NEOPCM (PW/TiO<sub>2</sub> and PW/TiO<sub>2</sub>Gr) utilizing a TEMPOS dual needle SH3 sensor Thermal analyzer, which measured TC. Optical absorbance measurement was performed using UV-Vis' spectroscopy (LAMBDA 1050, Perkin Elmer, USA). The measurements are taken at room temperature and in the wavelength range 200–800 nm.

## 3. Results and discussion

### 3.1. Morphological characterization of NEOPCM and binary NEOPCM

The morphological analysis for TiO<sub>2</sub> nanoparticles, Graphene nanoparticles, base PW, nanocomposite (1.0-PW/TiO<sub>2</sub>) with 1% TiO<sub>2</sub> relative to paraffin wax and binary composite (1.0-PW/TiO<sub>2</sub>Gr) was carried out, and FESEM images are shown in Fig. 2. According to Fig. 2(a), TiO<sub>2</sub> nanoparticles may be readily observed at a magnification of 150 times the standard magnification. The diameter of these nanotubes ranges from 20 to 40 nm, and their surface is porous. The size of Graphene nano powder is below 10 nm, FESEM analysis of Graphene nanoparticles demonstrated a flakes-like structure, as illustrated in Fig. 2(b). The FESEM of pure PW is presented in Fig. 2(c), which only reveals a lamellar layered structure and no other features. When paraffin wax is doped with TiO<sub>2</sub> and binary TiO<sub>2</sub>-Graphene, it can be seen in Fig. 2(d) and (e), the wax's smooth surface become uneven, revealing pores and particles contained in the matrix—TiO<sub>2</sub> functions as an irregular particle uniformly distributed over the Graphene surface [30]. Even though TiO<sub>2</sub> is larger than graphene nano powder, high magnification of the composite material is not possible as the material starts to melt as the magnification increases. The elemental mapping of the sample is shown as evidence of the presence of TiO<sub>2</sub>. However, as Carbon is the element for paraffin wax and graphene, to distinguish the presence of graphene from paraffin wax is not possible through elemental mapping. Another difficulty in identifying graphene nano powder is the layer-by-layer structure as evident in SEM image. Therefore, during the preparation of the composite, the graphene layers may have been embedded on the waxy structure of paraffin wax leading to difficulty in recognizing graphene from paraffin wax. However, TiO<sub>2</sub> is clearly seen from the change in smooth surface to small round uneven surface from the addition of titanium dioxide (Fig. 2e).

Nevertheless, it can be viewed in Fig. 2(e) that TiO<sub>2</sub> has grown on the surface of the Graphene. It is because of the functional groups on the Graphene surface, which act as anchoring sites while inhibiting nanoparticles agglomeration [30,31]. Consequently, the FESEM examination of the nanocomposite reveals that the TiO<sub>2</sub>-Graphene particles have

**Table 1**  
Physiochemical properties of PW RT-47 PCM (47–48 °C).

Properties	
PCM	RT47
Class	Organic ester PCM
Latent heat capacity	160 kJ/kg
Thermal conductivity	0.20 W/m/K
Colour	Bluish-white
Density	0.88 kJ/l (solid) 0.77 kJ/l (liquid)
Solubility	Insoluble in water

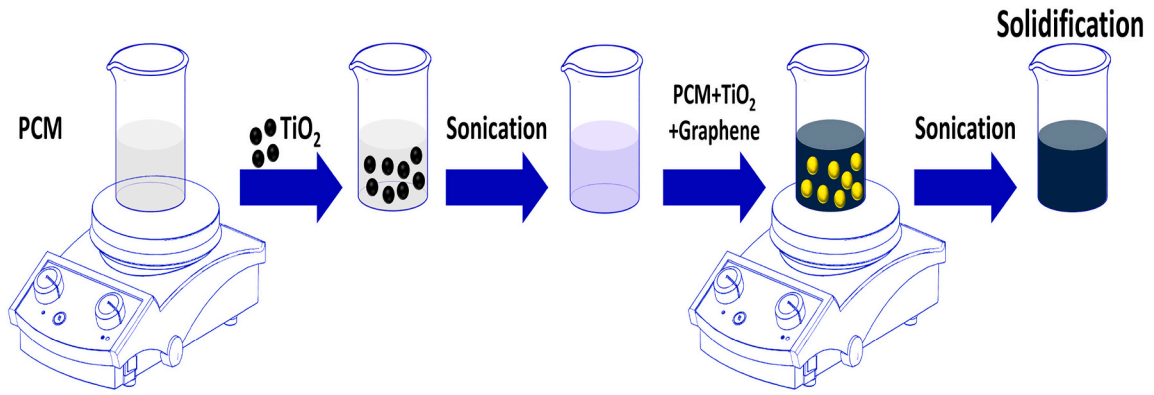


Fig. 1. Preparation of binary NEOPCM composite.

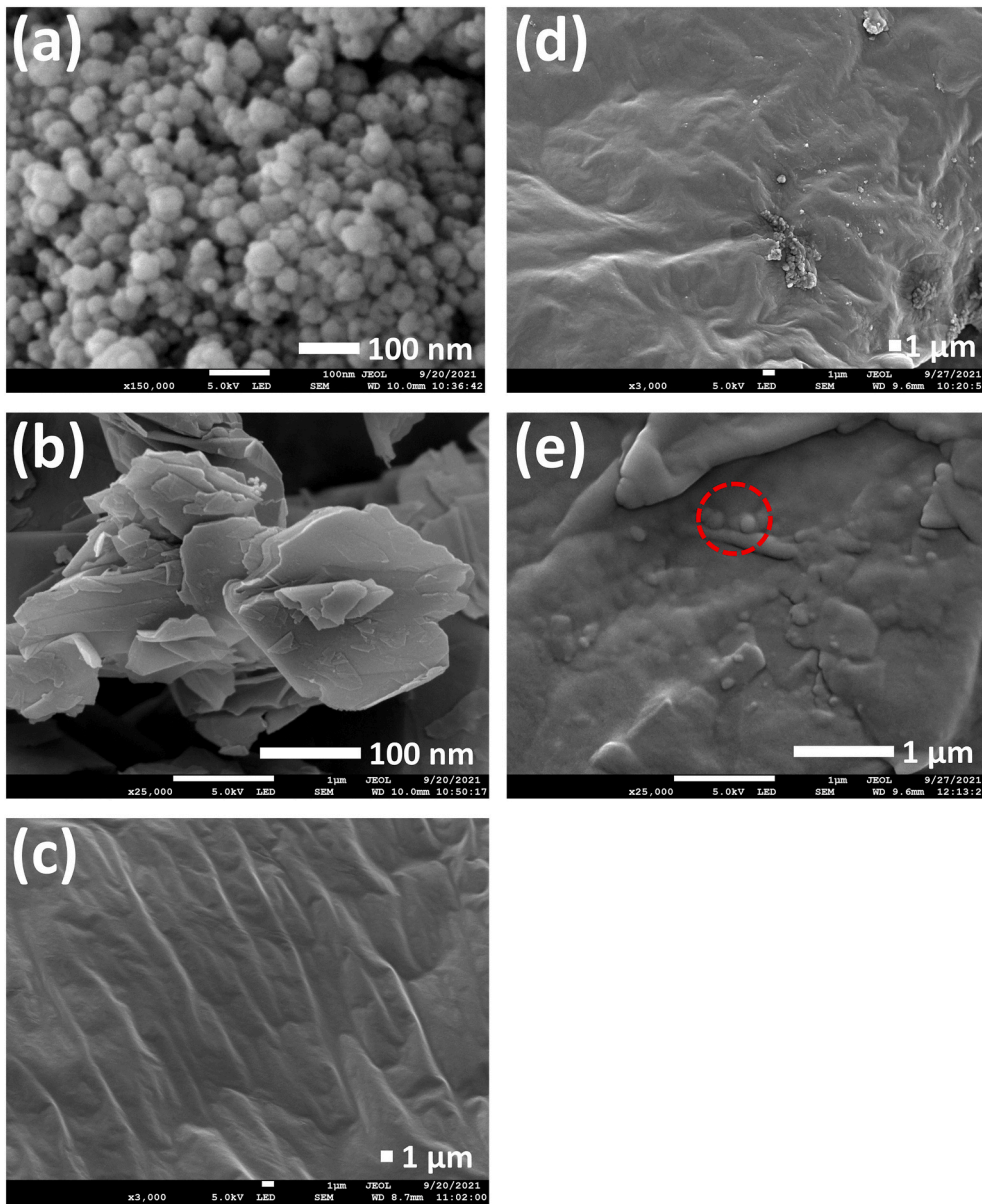


Fig. 2. FESEM images (a)  $\text{TiO}_2$  nano particles. (b) Graphene nano particles. (c) Paraffin wax RT47. (d) PW/ $\text{TiO}_2$  composite. (e) PW/ $\text{TiO}_2$  graphene binary composite.

been integrated into the matrix of paraffin wax.

The element mapping approach is beneficial for visualizing element distributions; it was selected to produce a more accurate representation of the dopant's dispersion in PW. Paraffin/TiO<sub>2</sub> nanocomposite mapping is shown in Fig. 3, and PW/TiO<sub>2</sub>-Graphene composite element mapping is shown in Fig. 4. From the figures, it is seen that the dopant has been evenly disseminated throughout the paraffin matrix. Fig. 3 shows the EDX spectrum's peaks of the elements such as C, O, Ti, Na, Ag, and Zr. Additionally, the weight per cent of C, O, Na, Mg, S, Ti, and Zr in the 1.0-PW/TiO<sub>2</sub> nanocomposite was 77.59, 7.45, 0.69, 9.59, 1.83, 1.09, and 10.75, respectively. Similarly, Fig. 4 shows the elemental structure and elements of the PW/TiO<sub>2</sub> Gr nanocomposite. The weight percentages of C, Na, S, and Ti in the 1.0-PW/TiO<sub>2</sub>Gr nanocomposite were determined to be 90.83, 6.35, 0.37, 0.63, and 1.82%, respectively.

### 3.2. Latent heat storage

Fig. 5 a shows enthalpy outcome with different mass fractions of PW/TiO<sub>2</sub> composite and pure PW RT 47. The base PCM supplier's data shows that the enthalpy, melting and solidifying temperature of base PW PCM and nanocomposite materials are slightly less. All the prepared nanocomposites presented an enhancement in the LHS except 1.0-PW/TiO<sub>2</sub> and 2.0-PW/TiO<sub>2</sub>. When the loading exceeds 1%, the LHS decrease below those of the base PW, due to two opposing factors cause changes in the composite's LHS. 1) The LHS of the composites increases due to the interaction of PW molecules and TiO<sub>2</sub> particles, and 2) the LHS decreases by the addition of TiO<sub>2</sub> particles because nanoparticles may absorb or release more energy during the composite melting or solidification processes. Thus, if the first component is more vital than the second, the composite's LHS enhances, and vice versa [32]. The downward trend of the initial peak shows the change of phase from solid to liquid. PW and PW/TiO<sub>2</sub> composites show solid-solid transition (SST) in the range of 30 °C–40 °C temperature. In comparison to the solid-liquid transition (SLT), the solid-to-solid peak is smaller. The SST occurs when the molecular structure changes from ordered to disordered. The second peak corresponds to the phase change from solid to liquid. As can be observed, the phase change begins at around 30 °C and finishes at approximately 80 °C. The area beneath the phase transition peaks of PW and PW/TiO<sub>2</sub> composites is used to determine the total LHS. The melting point is determined by finding a tangent on the left slope that passes through the DSC curve's apex.

The lower curve depicts the melting processes of the composites; it indicates that adding TiO<sub>2</sub> nanoparticles has little effect on the melting point of paraffin wax. The melting point has been increased by 47 °C to 50 °C by adding TiO<sub>2</sub> nanoparticles. The appropriate enthalpies were determined by integrating the peaks over the DSC software's baseline. As seen in this figure, the melting and crystallization enthalpies of paraffin nanocomposites with a mass fraction of 0.1% are pretty higher than those of pure wax. The Van der Waals interactions between the TiO<sub>2</sub>

nanoparticles and the PW are responsible for this latent heat increase [33]. The heat absorbed during the solid-to-liquid transition is utilized to overcome the PCM's weak intermolecular forces. If the particle concentration is high enough, the presence of nanoparticles can change these forces so that the interaction potential between the paraffin wax and the nanoparticles is greater than the interaction potential between the wax molecules themselves. Due to the nanoparticles' exceptionally high surface to volume ratio, a low concentration of them can improve molecular interaction and raise the latent heat of the nanocomposites. The latent temperatures of fusion of NEOPCM are deteriorated at increasing concentrations due to the accumulative replacement of paraffin wax by nanoparticles.

The highest LHS enhancement was seen for the composite 0.1-PW/TiO<sub>2</sub> at 7.98% 157.0 ± 3.14 J/g and melting temperature 46.20 ± 0.2 °C, 1% enhancement in the LHS recorded for the composite 0.5-PW/TiO<sub>2</sub> at 146.30 ± 2.92 J/g. All the results of latent heat capacity for 0.1-PW/TiO<sub>2</sub>, 0.5-PW/TiO<sub>2</sub>, 1.0-PW/TiO<sub>2</sub> and 2.0-PW/TiO<sub>2</sub> are 1%, 7.98%, –16% and –15% respectively as can be seen in Fig. 5(a). The melting point of all composites is essentially the same as the base PW. However, except 2.0 PW/TiO<sub>2</sub>, all composites exhibit a decrease in melting temperature compared to pure PW. It can be explained by the nature of the interaction between the composite's components. TiO<sub>2</sub> nanoparticles have a high surface area, and the intermolecular interaction between the molecules in the PW/TiO<sub>2</sub> dispersion might account for the increase in latent heat energy [34]. The shift in melting temperature was caused by the weak attractive contact between the fluid and the matrix surface [35].

Fig. 5(b) shows enthalpy outcome with different mass fractions of PW/TiO<sub>2</sub>Gr composite. Except for PW/TiO<sub>2</sub>Gr-1.0, PW/TiO<sub>2</sub>Gr-2.0, every produced nanocomposite shown an increase in latent heat capacity, as illustrated in Fig. 5(b). The downward trend of the first peak indicates the transition from the solid to the liquid phase. PW and PW/TiO<sub>2</sub>Gr composites have an SST temperature range of 30 °C–40 °C. The SST has a smaller peak than the solid-liquid transition. When the molecular structure shifts from organized to disordered, the solid-to-solid transition occurs. The second peak indicates the phase transition from solid to liquid. As evidenced by the preceding, the phase transition begins at about 30 °C and ends around 80 °C. Total latent heat is calculated by measuring the area beneath the phase transition peaks of PW and PW/TiO<sub>2</sub>Gr composites. The melting point is calculated by locating a tangent on the left slope that crosses through the peak of the DSC curve. The highest enhancement was 10.02% for the composite 0.1-PW/TiO<sub>2</sub>Gr, a latent heat capacity of 160.0 ± 3.20 J/g and a melting temperature of 49.12 ± 0.2 °C, 4.74%. The LHS capacity of 0.1-PW/TiO<sub>2</sub>, 0.5-PW/TiO<sub>2</sub>Gr, 1.0-PW/TiO<sub>2</sub>Gr, and 2.0-PW/TiO<sub>2</sub>Gr are 160.0 ± 3.20 J/g, 158 ± 3.16 J/g, 130.04 ± 2.60 J/g and 122 ± 2.44 J/g. All composites have a melting point that is nearly identical to that of the base PW. However, all composites, except for 1.0 PW/TiO<sub>2</sub>Gr and 2.0 PW/TiO<sub>2</sub>Gr, have a lower melting point than base PW. It can be explained by

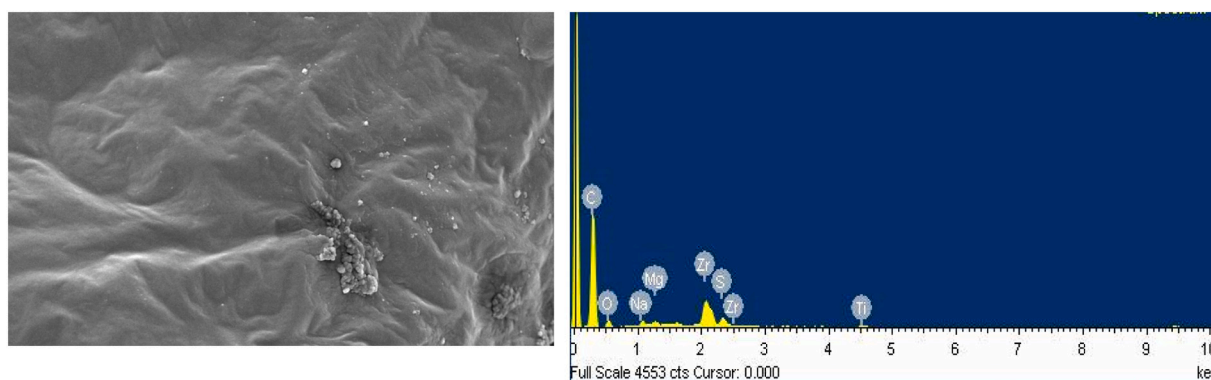


Fig. 3. The element mapping PW/TiO<sub>2</sub> composite. Left image: showing the elemental mapping. Right image: elements and their weight percentage.

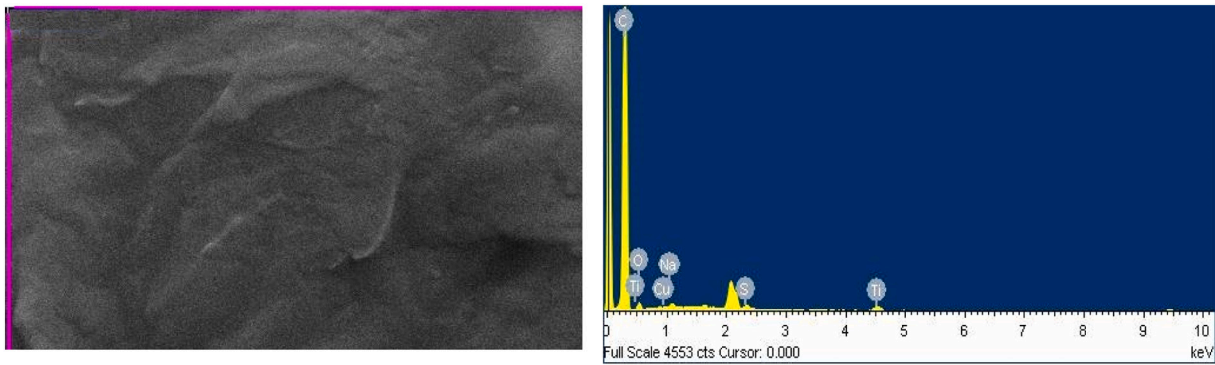


Fig. 4. Element mapping of PW/TiO<sub>2</sub> graphene binary composite. Left image: showing the elemental mapping. Right image: elements and their weight percentage.

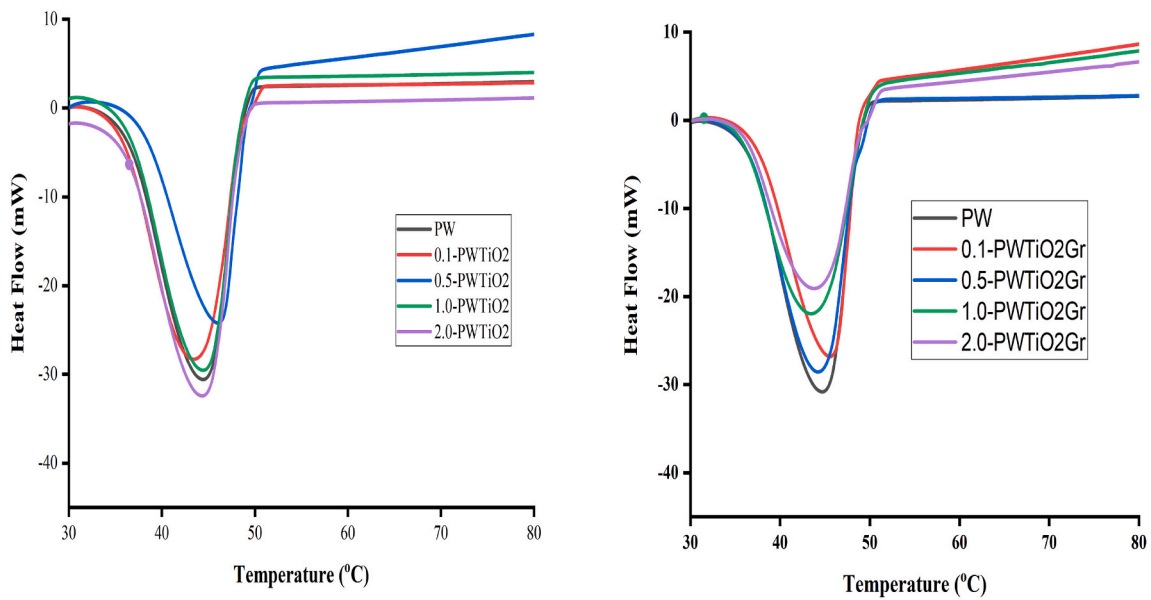


Fig. 5. DSC Curve of heat flow vs temperature (a) PW-TiO<sub>2</sub> (b) PW-TiO<sub>2</sub> Gr.

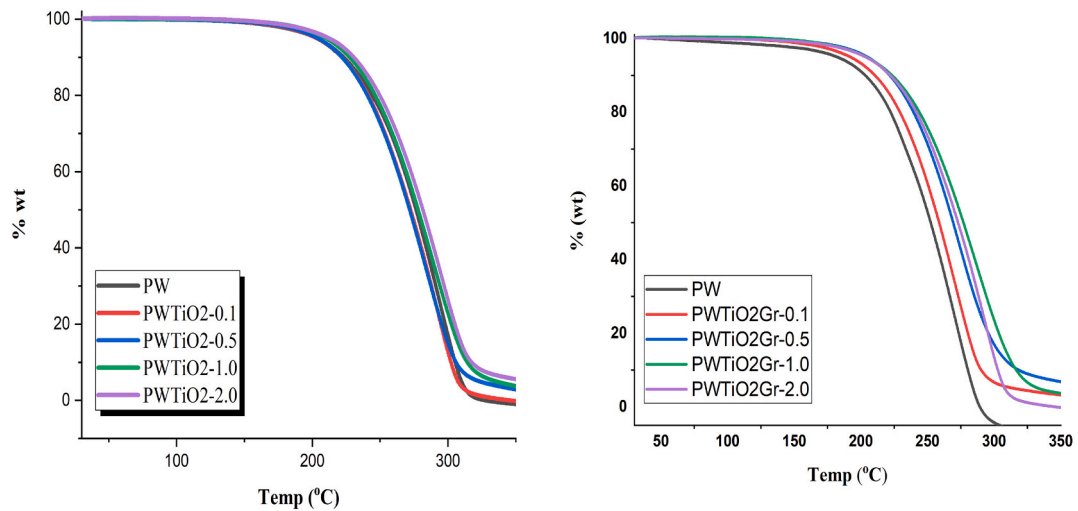


Fig. 6. TGA curve of weight % vs temperature for PW/TiO<sub>2</sub> in (a) and for PW/TiO<sub>2</sub>Gr in (b).

the way the composite's components interact. The shift in melting temperature might have been induced by the fluid's weak attractive interaction with the matrix surface. TiO<sub>2</sub>-Graphene nanoparticles have a high surface area, and the intermolecular interaction between the molecules in the PW/TiO<sub>2</sub>Gr dispersion might account for the increase in latent heat energy.

### 3.3. Thermal stability

In various concentrations, the thermogravimetric analyzer (TGA) evaluates the durability of the prepared composites PW/TiO<sub>2</sub> and PW/TiO<sub>2</sub>Gr added Graphene are presented in Fig. 6a and b respectively. TGA measures are performed to ensure the precision of results obtained using a mass value of 10 mg and a heating rate of 10 °C/min in the 450-temperature range. Weight deterioration over temperature is illustrated in the Fig. 6; the downward trend indicates that the weight loss arises for PW and PW/TiO<sub>2</sub> composites. The increase in nanoparticle concentration enhances the initial and final degradation temperatures of pure PW and PW/TiO<sub>2</sub> composites. This research shows the addition of TiO<sub>2</sub> nanoparticles improves the thermal stability of PW. The temperature at which deterioration begins for PW, PW/TiO<sub>2</sub>-0.1, PW/TiO<sub>2</sub>-0.5, PW/TiO<sub>2</sub>-1.0 and PW/TiO<sub>2</sub>-2.0 are 309.24 °C, 309.98 °C, 309.21 °C, 313.31 °C and 316.29 °C respectively.

The focal reduction temperature of PW/TiO<sub>2</sub>-2.0 is found to be 325.25 °C, which shows a slight increment in the final degradation temperature of nanocomposite using TiO<sub>2</sub> with pure PW. At temperatures higher than pure paraffin PW, more excellent thermal stability is seen, whereas a lower value indicates a decrease in thermal stability [36]. The onset degradation temperature for binary composite by addition of Graphene nanoparticles with PW/TiO<sub>2</sub> composites PW/TiO<sub>2</sub>-Gr-0.1, PW/TiO<sub>2</sub>-Gr-0.5, PW/TiO<sub>2</sub>-Gr-1.0 and PW/TiO<sub>2</sub>-Gr-2.0 are 315.90 °C, 316.44 °C, 320.22 °C and 325.25 °C respectively. With the addition of Graphene nanoparticles to NEOPCM composite, the thermal degradation occurs higher than the base PW. Admittedly, all deterioration temperatures are within 10 °C of the temperature of pure PW, indicating only a minor variance in the thermal degradation of binary composites that have been produced. It demonstrates that Graphene can improve the thermal stability of the RT47 PW base. Because the TiO<sub>2</sub>-Graphene nanoparticles form a thermal barrier when heated, this may be what causes the improved thermal stability.

### 3.4. Thermal conductivity

Fig. 7(a) and (b) shows the thermal conductivity of PW, PW/TiO<sub>2</sub> and PW/TiO<sub>2</sub>Gr composites, respectively. The measurement demonstrated the composite PCM's values PW, PW/TiO<sub>2</sub>-0.1, PW/TiO<sub>2</sub>-0.5,

PW/TiO<sub>2</sub>-1.0 and PW/TiO<sub>2</sub>-2.0 are  $0.1958 \pm 0.019$  W/mK,  $0.3247 \pm 0.032$  W/mK,  $0.3632 \pm 0.036$  W/mK,  $0.4304 \pm 0.043$  W/mK,  $0.3655 \pm 0.036$  W/mK, respectively. All samples show greater TC as related to base PW. Due to TiO<sub>2</sub>'s exceptionally high TC compared to PW, the composites' TC rose as the mass ratio of TiO<sub>2</sub> nanoparticles increased. SDBS was used as a surfactant to improve the TC of composites. TC has a significant effect on the pace at which energy is stored. It is seen in Fig. 7 (a) that the TC of PW/TiO<sub>2</sub> nanocomposite upsurge with the concentration of TiO<sub>2</sub> up to 1% and then decrease at PW/TiO<sub>2</sub>-2.0. Results obtained at high concentrations imply that the samples are not stable. The agglomeration of nanoparticles changes the measuring region's composition and properties, resulting in thermal conductivities that are significantly nearby the base PCM. Since a result of this, efficacy is diminished, resulting in the collapse of the thermal network and decreased TC. [37].

In Fig. 7(b), since Graphene has a much higher TC than RT47 PCM, the TC of the binary composites PW/TiO<sub>2</sub>Gr-0.1, PW/TiO<sub>2</sub>Gr-0.5, PW/TiO<sub>2</sub>Gr-1.0 and PW/TiO<sub>2</sub>-2.0% increased as 0.2361, 0.2796, and 0.3491, which are 120.58%, 142.80% and 179.03% increment concerning base PW PCM. The measurement showed the composite PCM's values PW/TiO<sub>2</sub>Gr-0.1, PW/TiO<sub>2</sub>Gr-0.5, PW/TiO<sub>2</sub>Gr-1.0 and PW/TiO<sub>2</sub>Gr-2.0 are  $0.4319 \pm 0.043$  W/mK,  $0.475 \pm 0.475$  W/mK,  $0.544 \pm 0.054$  W/mK and  $0.396 \pm 0.039$  W/mK, respectively. The addition of TiO<sub>2</sub>, Graphene and SDBS led to an increase in the TC of the composites. All the nanocomposite shows a greater increment in the TC by adding TiO<sub>2</sub> and Graphene nanoparticles. PW/TiO<sub>2</sub>Gr-1.0 showed the greater TC at  $0.5441 \pm 0.054$  W/mK, which is an increment of ~179.03%. The amount of TiO<sub>2</sub> and Graphene in the composite PCM increased, and the TC increased dramatically. The TC of the composite PCMs with PW/TiO<sub>2</sub>-1.0 and PW/TiO<sub>2</sub>Gr-1.0 is 2.20 times and 2.78 times, respectively than PW. The increase in TC by inserting nanoparticles into PCM is evident compared to introducing nanoparticles into base PW. The TC of composite enhances with concentration up to 1%, and then decrement trend is seen at PW/TiO<sub>2</sub>-2.0 and PW/TiO<sub>2</sub>Gr-2.0. The reduction of TC at higher concentrations is because of the agglomeration of nanoparticles [38]. That is the key reason for efficacy as it led to the breakdown of the thermal network and reduction in TC [39,40]. The enhancement in the TC for PW/TiO<sub>2</sub> and PW/TiO<sub>2</sub>Gr binary composite related to Paraffin may be ascribed to the improved thermal network within the PCM matrix due to highly conductive nanoparticles [41]. But as concentration increases above 1%, the agglomeration occurs, creating thermal breakdown and decreasing electron thermal transport. There are two types of thermal heat transfer mechanisms: electron transport and phonon transport. PW is based on phonon-phonon transport, which is similar to electron transport in metals and conducting polymers. Phonons are critical for heat transmission in materials. Generally, the

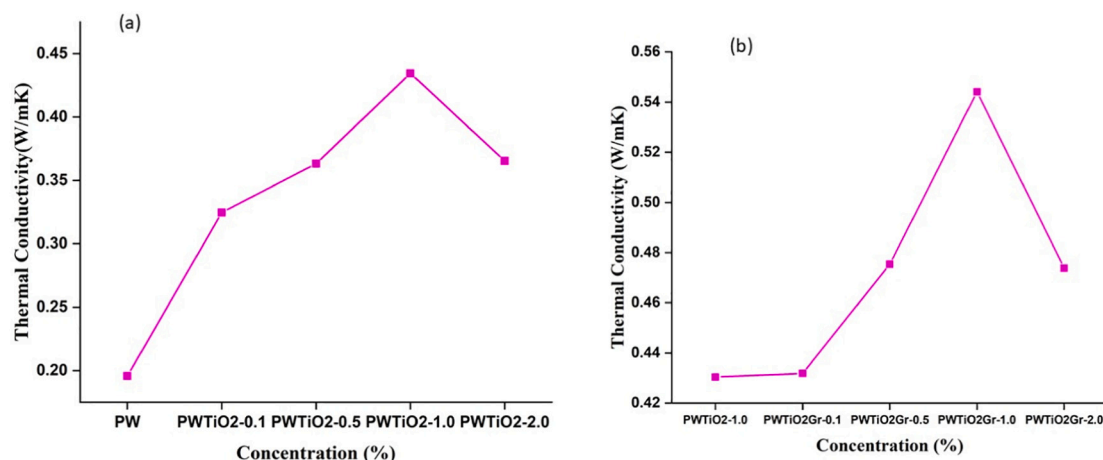


Fig. 7. The curves of thermal conductivity vs concentration (%) of PW, PW/TiO<sub>2</sub> in (a) and of PW/TiO<sub>2</sub>Gr in (b).

greater the strength of phonon diffusion in a material, the greater its capacity to transfer heat and hence the higher its TC, and vice versa. The diverse phonon scatterings inside materials are the primary cause of phonon diffusion degradation. Phonon scattering may be classified into three types: phonon-phonon scattering, phonon-defect scattering, and phonon-interface scattering. [37]. In phonon transfer, the impurities develop more with concentration, increasing the phonon scattering impurities, resulting in a shorter mean free path and thus a decrease in TC. The reduction in TC of PW/TiO<sub>2</sub> and PW/TiO<sub>2</sub>Gr-1.0 could result from a tradeoff between electron and phonon thermal transport. As a result, the optimal concentration for increased TC is PW/TiO<sub>2</sub>-1.0 and PW/TiO<sub>2</sub>Gr-1.0.

### 3.5. Chemical stability

The FT-IR study was done to see whether any chemical or physical reactions between the various components of the composites. The FT IR transmission spectrum of PW/TiO<sub>2</sub> NEOPCM and PW/TiO<sub>2</sub>Gr binary composite are given in Fig. 8a and b correspondingly. Based on the current evidence, composite PW/TiO<sub>2</sub> exhibits four distinct transmission or absorption bands, as depicted in Fig. 8a: CH<sub>2</sub> rocking vibration at 720 cm<sup>-1</sup>, CH<sub>2</sub> and CH<sub>3</sub> deformation vibration at 1467 cm<sup>-1</sup>, and intense peaks at 2850 cm<sup>-1</sup> and 2917 cm<sup>-1</sup>, which correspond to CH<sub>2</sub>—CH<sub>3</sub> stretching vibrations [42] and intense peaks at 2850 cm<sup>-1</sup> and 2917 cm<sup>-1</sup>, respectively. When it comes to C—H stretching vibrations, the first two peaks correspond to the aliphatic and methyl groups. In contrast, the third peak corresponds to the deformation vibration of the —CH<sub>2</sub> and —CH<sub>3</sub> bond and the fourth peak corresponds to the rocking vibration of the —CH<sub>2</sub> bond. PW and PW/TiO<sub>2</sub> composites designated as PW/TiO<sub>2</sub>-0.1, PW/TiO<sub>2</sub>-0.5, PW/TiO<sub>2</sub>-1.0, and PW/TiO<sub>2</sub>-2.0 were used to study the FTIR spectra at 4000–400 cm<sup>-1</sup> frequencies and were shown to be effective. The PW and PW/TiO<sub>2</sub> composites have almost similar peak heights, and there is no evidence of a new peak. The existence of a similar peak indicates that the particle's additives and the

surfactant are not reacting chemically and that only physical contact occurs. As shown in Fig. 6b, the main absorption peaks of composite PW/TiO<sub>2</sub>Gr are 2917 cm<sup>-1</sup>, 2850 cm<sup>-1</sup>, 1467 cm<sup>-1</sup>, and 720 cm<sup>-1</sup>, respectively. The C—H<sub>2</sub> bonding of the Paraffin has a peak at 2917 cm<sup>-1</sup> and 2850 cm<sup>-1</sup>, indicating that it is highly stable.

The peak defines the Paraffin contains the stable balance of attractive and repulsive forces between carbon and hydrogen atoms [43]. Due to C—H<sub>3</sub> stretching vibration, oxygen functionalities were observed in the composite peaks 1467 cm<sup>-1</sup> and 720 cm<sup>-1</sup>. It is noticeable no new peaks are formed in both PW/TiO<sub>2</sub> composites preparation. Fig. 8b shows FT IR spectrum of PW, TiO<sub>2</sub> nanoparticles, Graphene nanoparticles and the composites PW/TiO<sub>2</sub>-1.0, PW/TiO<sub>2</sub>Gr-0.1, PW/TiO<sub>2</sub>Gr-0.5, PW/TiO<sub>2</sub>Gr-1.0 and PW/TiO<sub>2</sub>Gr-2.0. Based on the current evidence, binary composites PW/TiO<sub>2</sub>Gr have four distinct transmission or absorption bands the same as in the case of PW/TiO<sub>2</sub> composites. According to the current evidence, composite PW/TiO<sub>2</sub>Gr displays four different transmission or absorption bands, as illustrated in Fig. 8b. CH<sub>2</sub> rocking and deformation vibrations vary at 720 cm<sup>-1</sup>, 1467 cm<sup>-1</sup>, and 2850 cm<sup>-1</sup>, respectively, which correspond to CH<sub>2</sub>—CH<sub>3</sub> stretching vibrations. Intense peaks occur at 2850 cm<sup>-1</sup> and 2917 cm<sup>-1</sup>, which correspond to CH<sub>2</sub>—CH<sub>3</sub> stretching vibrations [42].

Regarding stretching vibrations between atoms of the C—H bond, the first two peaks belong to the aliphatic and methyl groups, respectively. The third peak corresponds to deformation vibrations of the CH<sub>2</sub> and —CH<sub>3</sub> bonds, while the fourth peak refers to rocking vibrations of the —CH<sub>2</sub> bond. In order to investigate the FTIR spectra at 4000–400 cm<sup>-1</sup> frequencies, the PW/TiO<sub>2</sub> composites PW/TiO<sub>2</sub>-0.1, PW/TiO<sub>2</sub>-0.5, PW/TiO<sub>2</sub>-1.0, and PW/TiO<sub>2</sub>-2.0 were employed. The results showed that the composites were effective. The peak heights of the PW and PW/TiO<sub>2</sub>Gr composites are virtually identical, and there is no indication of the formation of a new peak. The presence of a comparable peak shows that the particle's additives and the surfactant are not interacting chemically and that only physical contact is taking place between the two substances. Adding the SDBS to PW/TiO<sub>2</sub> composite and added Graphene

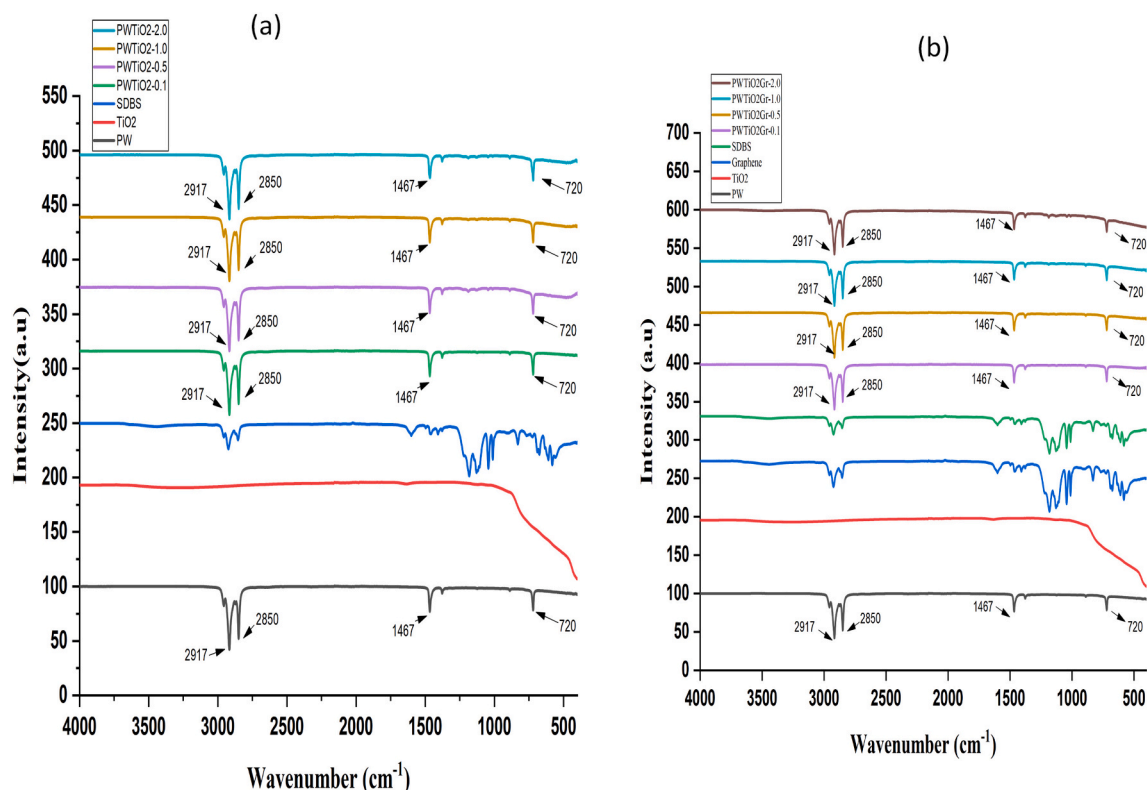


Fig. 8. FTIR Spectrum curves for intensity v/s wavenumber (cm<sup>-1</sup>) of (a) PW/TiO<sub>2</sub> (b) PW/TiO<sub>2</sub>Gr.



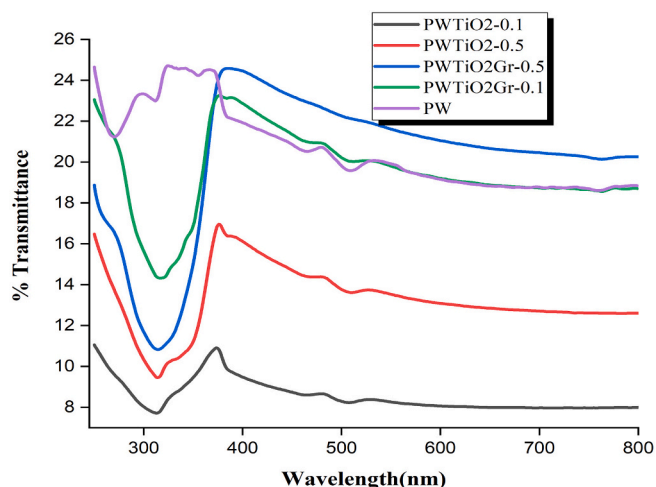


Fig. 9. Light transmission curve of PW/TiO<sub>2</sub> and PW/TiO<sub>2</sub>Gr.

nanoparticles did not result in new peaks in the FTIR spectrum of PW/TiO<sub>2</sub>Gr composite, as can be seen from Fig. 8b.

### 3.6. UV-Vis light transmission capability

The transmissibility of the UV-VIS spectra of PW/TiO<sub>2</sub> and PW/TiO<sub>2</sub>Gr binary composites is depicted in Fig. 9. PW Base PCM, PW/TiO<sub>2</sub>-0.1, PW/TiO<sub>2</sub>-0.5, PW/TiO<sub>2</sub>Gr-0.1, PW/TiO<sub>2</sub>Gr-0.5 have transmittance of 19.81%, 16.53%, 12.50%, 10.35% and 8.30% respectively as compared with solar spectrum. The prepared composites reduced light transmission by 58.30% compared to pure PW; maximum reduction is due to the introduction of TiO<sub>2</sub> and Graphene nanoparticles into PW. The decrease in solar spectrum transmission indicates increased light absorption and, subsequently, enhanced photothermal conversion.

Results of the UV-VIS spectrum of the PW/TiO<sub>2</sub>Gr binary composite show a reduction in the sample's transmissibility is accompanied by dispersion of nanoparticles in PW [44]. A maximum reduction in transmission of 36.90% is achieved with the PW/TiO<sub>2</sub> composite. The

transmittance of PW/TiO<sub>2</sub> has been reduced by 33.2% compared with PW/TiO<sub>2</sub> composite. When the findings were compared, it was found that the PW/TiO<sub>2</sub>Gr binary composite performs significantly better than the PW/TiO<sub>2</sub> in terms of direct light absorption. The enhanced absorptivity of the binary composite can be related to the Graphene nanoparticles' outstanding photothermal properties.

### 3.7. Thermal reliability

The Thermal cycling tests were conducted to confirm the prepared nano enhanced phase change material's (NePCMs) thermal reliability. The samples were placed in a custom-built thermal cycler equipped with three chambers to allow for simultaneous testing of three samples. The samples were placed in a crucible and an electronically controlled thermocouple was introduced into each one. The maximum temperature is kept at 65 °C, at least 10 °C above the melting point of the sample, to ensure full melting. To obtain the maximum temperature, the samples were heated from below the crucible holder. When the maximum temperature has been reached, the heater is auto turned off. This initiates the cooling cycle, during which the sample is cooled from above using a Peltier cooling system. The minimum temperature for the cooling cycle is kept as 25 °C. Due to the less quantity of sample size used, approximately 500 mg–800 mg, the maximum temperature is reached in 2 min, while the cooling cycle takes slightly longer, approximately 3 min. Thermal reliability was assessed for the PW, TiO<sub>2</sub> added composite, and binary composite using accelerated thermal cycler. The thermal cycling properties of paraffin and the selected composite are determined using TGA, DSC, and FTIR data taken at 0 and 300 cycles. Due to their high thermal conductivity values, two composites, PWTiO<sub>2</sub>-1.0 and

Table 2  
DSC data after 0 and 300 thermal cycles.

Composite	Latent heat J/g		Melting point °C	
	0 cycle	300 cycles	0 cycle	300 cycles
PW	145	142.29	48.77	47.0
PWTiO <sub>2</sub> -1.0	121	116	48.61	47.1
PWTiO <sub>2</sub> Gr-1.0	130	128	48.24	47.2

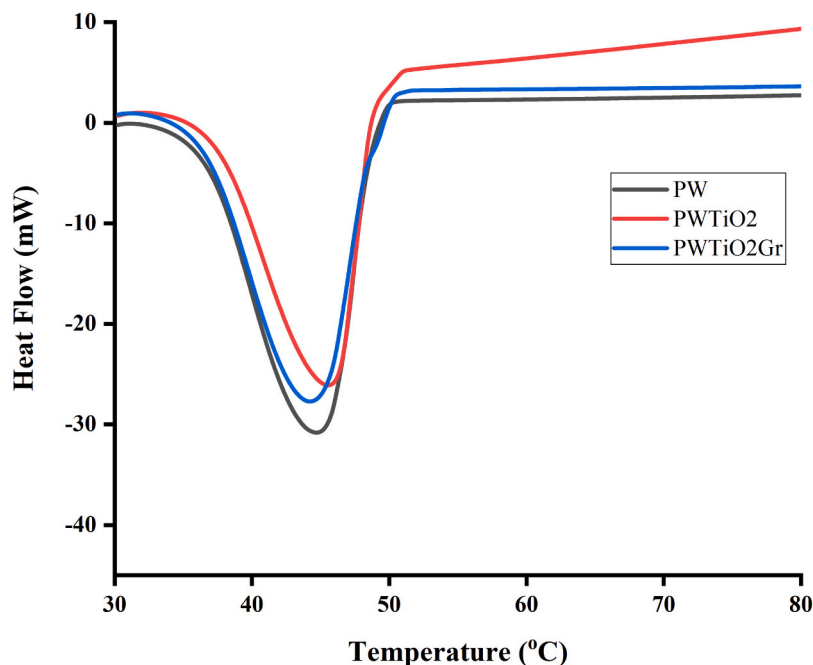


Fig. 10. DSC Curve of heat flow vs temperature 300 cycles PW, PW-TiO<sub>2</sub> and PW-TiO<sub>2</sub> Gr.

PWTiO<sub>2</sub>Gr-1.0, were chosen for accelerated thermal cycling. Fig. 10 illustrates the DSC curve of thermally cycled paraffin wax PW, PW-TiO<sub>2</sub> and PW-TiO<sub>2</sub> Gr at 300 thermal cycles and the data is summarized in Table 2. At 300 thermal cycles, paraffin wax PW, PW-TiO<sub>2</sub> and PW-TiO<sub>2</sub> Gr has a latent heat storage capacity of  $142.61 \pm 3.07$  J/g and  $116.71 \pm 2.91$  J/g and  $128.71 \pm 2.91$  J/g, respectively. Additionally, paraffin wax PWTiO<sub>2</sub>-1.0 and PWTiO<sub>2</sub>Gr-1.0 melted at  $47.0 \pm 0.2$  °C,  $48.1 \pm 0.2$  °C, and  $48.2 \pm 0.2$  °C, respectively. The thermal cycled samples lack the solid-solid transition area, even if the solid-to-solid transition phase is not visible in the DSC curve, this may be owing to particle agglomeration caused by repeated phase changes may account for the decrease in latent heat at 300 cycles from  $145.31 \pm 2.90$  J/g to  $142.71 \pm 2.84$  J/g,  $121 \pm 2.42$  J/g to  $116.71 \pm 2.32$  J/g and  $130 \pm 2.60$  J/g to  $128 \pm 2.56$  J/g respectively. The percent decrease in performance is attributable to the faster thermal cycling.

The FT-IR analysis was performed to determine whether any chemical or physical reactions occurred between the composites' constituent components after 300 cycles. Fig. 11 illustrates the FTIR transmission spectra of 300 cycles PW, PW/TiO<sub>2</sub>-1.0, and PW/TiO<sub>2</sub>Gr-1.0 binary composites. Composite and PW display four same transmission or absorption bands, as illustrated in Fig. 11: CH<sub>2</sub> rocking vibration at  $720$  cm<sup>-1</sup>, CH<sub>2</sub> and CH<sub>3</sub> deformation vibration at  $1467$  cm<sup>-1</sup>, and intense peaks at  $2850$  cm<sup>-1</sup> and  $2917$  cm<sup>-1</sup>, respectively, which correspond to CH<sub>2</sub>–CH<sub>3</sub> stretching vibrations and intense peaks at  $2850$  cm<sup>-1</sup> and  $2917$  cm<sup>-1</sup>. The first two peaks correspond to the aliphatic and methyl groups, respectively, in terms of stretching vibrations between C–H bond atoms. The third peak relates to the deformation vibrations of the CH<sub>2</sub> and –CH<sub>3</sub> bonds, whereas the fourth peak belongs to the –CH<sub>2</sub> bond rocking. The PW and composites were used to analyze the FTIR spectra at  $4000$ – $400$  cm<sup>-1</sup> frequencies. The composites were found to be effective. The peak heights of the 300 cycles PW, PWTiO<sub>2</sub> and PW/TiO<sub>2</sub>Gr composites are almost comparable, and there is no evidence of a new peak forming. The decrease in intensity could have been due to particle agglomeration, whereas the new peak could be due to the development of a C=C.

The initial degradation temperature for PW, PW/TiO<sub>2</sub>-1.0, and PW/TiO<sub>2</sub>Gr-1.0 after 300 thermal cycles can be seen in Fig. 12 and are similar to the initial decomposition temperature of uncycled zero cycle. The maximal decomposition temperature of PW, PW/TiO<sub>2</sub>-1.0, and PW/TiO<sub>2</sub>Gr-1.0 after 300 thermal cycles are  $317.0$ ,  $329.25$  °C and  $334.3$  °C, which is higher than the of zero cycle of PW ( $309.32$  °C), PW/TiO<sub>2</sub>-1.0 ( $313.31$  °C), and PW/TiO<sub>2</sub>Gr-1.0 ( $320.22$  °C) samples. This clearly demonstrates the chemical and thermal stability of the nanocomposite dispersed Paraffin, indicating that it can be applied for TES while overcoming paraffin's low thermal conductivity.

#### 4. Conclusions

In this study, we determined the thermal behavior of the PW as PCM, PW/TiO<sub>2</sub> composite and PW/TiO<sub>2</sub>Gr binary composite. The enhanced thermal characteristics of PW and composites were tailor-made via two-step preparation methods, commonly employed for TES. The characterization and thermophysical properties of the novel binary composite were explored; the FTIR study demonstrated no chemical interaction between the PW and the nanoparticles. Morphological characterization was done using FESEM and EDX to confirm imaging and observational spatial of all the elements. The EDX spectrum's peaks correspond to the elements carbon, oxygen, titanium, sodium and silver. Outcomes demonstrated that TiO<sub>2</sub> and Graphene are integrated into the paraffin matrix. The temperature and enthalpy of the melting and solidifying processes enhance the chemical and structural stability of the composites. The 0.1-PW/TiO<sub>2</sub> and the 0.1-PW/TiO<sub>2</sub>Gr binary composite's latent heat capacity was improved by 8.62% and 10.02%, respectively. The TC shows a maximum increment of 120% (PW/TiO<sub>2</sub>-1.0) and 179% (PW/TiO<sub>2</sub>Gr-1.0) concerning base PCM. The obtained TGA results confirm that increment in the loading weight percentage of nanoparticles

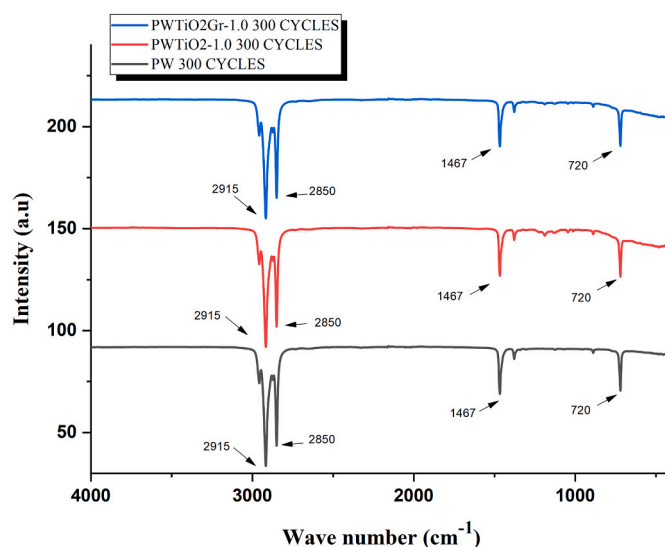


Fig. 11. FTIR Spectrum curves for intensity v/s wavenumber (cm<sup>-1</sup>) of 300 cycles PW, PW/TiO<sub>2</sub> 1.0 and PW/TiO<sub>2</sub>Gr-1.0.

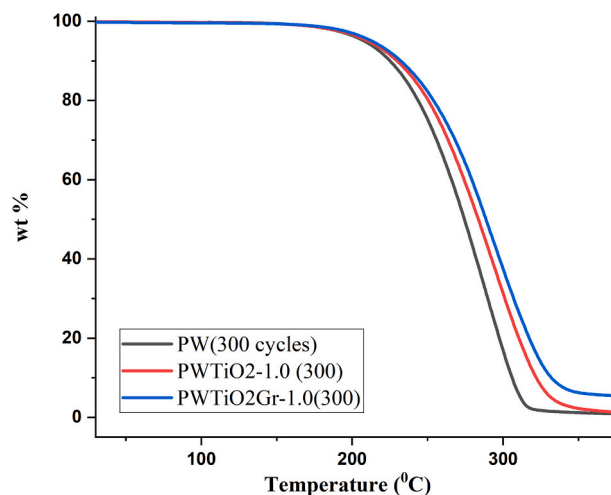


Fig. 12. TGA Curve of weight % vs temperature for 300 cycles PW, PW/TiO<sub>2</sub> 1.0 and PW/TiO<sub>2</sub>Gr-1.0.

increases the thermal durability of the nanocomposite PW/TiO<sub>2</sub> and PW/TiO<sub>2</sub>Gr. A thermophysical change in pure PW and NEOPCM composite was noted regarding initial and final degradation. The light transmission of the prepared composite was reduced by 58.30% related to base PW. All composite showed thermal degradation within 10 °C with compared to PW PCM. Thus, with the higher TC and enhanced thermal reliability, the PW/TiO<sub>2</sub> and PW/TiO<sub>2</sub>Gr composites presented promising prospects for energy-efficient application of solar photovoltaic thermal, domestic hot water and cooling application of electric drives.

#### CRediT authorship contribution statement

**Imtiaz Ali Laghari:** Data Curation, Investigation, Formal analysis, Writing - Original Draft

**M. Samykano:** Validation, Review & Editing, Methodology, Supervision, Project administration

**A K Pandey:** Conceptualization, Methodology, Formal analysis, Validation, Supervision, Project administration

**K. Kadirgama:** Writing - Review & Editing, Supervision

**Yogeshwar Nath Mishra:** Validation, Writing - Review & Editing

## Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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## Data availability

Data will be made available on request.

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