



# Synergistic conversion of CO<sub>2</sub> into C1 and C2 gases using hybrid in-doped TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> photocatalysts

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## ABSTRACT

Achieving high-efficiency photocatalytic conversion of CO<sub>2</sub> into value-added chemicals remains a challenge. This study synthesizes In-doped TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> composites (In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>) via a facile and reliable method. The as-synthesized In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> produces CO, CH<sub>4</sub>, and C<sub>2</sub>H<sub>4</sub> under UV, and CO and CH<sub>4</sub> under visible light from gaseous CO<sub>2</sub> and H<sub>2</sub>O vapor. A prolonged photocatalysis results in the continuous production of the same set of carbonaceous compounds over 30 h, with a photonic yield of ~40%. The yield of C<sub>2</sub>H<sub>4</sub> with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> is ~11-times greater than the sum of In-TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>. The CO<sub>2</sub> adsorption isotherms show that In-TiO<sub>2</sub> acts as a CO<sub>2</sub> adsorbent and photocatalyst whereas g-C<sub>3</sub>N<sub>4</sub> mainly works as a photocatalyst. In-situ FTIR study reveals the formation of CH<sub>4</sub> and C<sub>2</sub>H<sub>4</sub> on In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>. Time-resolved photoluminescence indicate that In-doping facilitates charge transfer and a strongly coupled g-C<sub>3</sub>N<sub>4</sub> induces cascaded charge transfer. This leads to inhibited charge recombination and long-lived charge carriers.

## 1. Introduction

Over the last few decades, photocatalytic CO<sub>2</sub> conversion has gained popularity as one of the artificial photosynthesis technologies [1–5]. It can upgrade captured CO<sub>2</sub> to value-added chemicals, such as CO, methanol, aliphatic acids, and hydrocarbons, using n-type and p-type semiconductor photocatalysts [6–10]. Despite extensive studies, most photocatalysts still suffer from low selectivity (i.e., mixed carbon products), non-stoichiometric redox reactions (i.e., limited or absent water oxidation reactions), short durability, and poor solar conversion efficiency. Some p-type semiconductors demonstrated an exceptionally high solar conversion efficiency of ~5% over a month under simulated sunlight, with 100%-selective formic acid production and stoichiometric O<sub>2</sub> evolution in the absence of any sacrificial chemicals and electrical biases [6,11–13].

Among n-type semiconductors, TiO<sub>2</sub> has been widely studied as a model material because of its low cost, photochemical stability, and environmentally friendly properties [14–16]. Nevertheless, TiO<sub>2</sub> has several photochemical limitations for driving CO<sub>2</sub> reduction reactions (CO<sub>2</sub> RR), including a wide bandgap, poor charge transfer, and

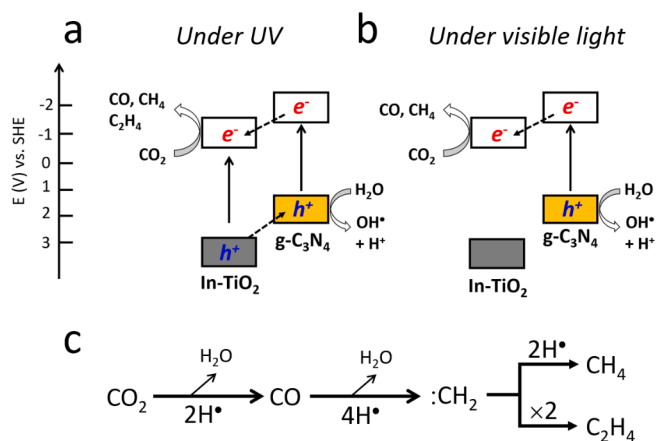
inefficient charge injection. The first two issues are common for most reduction reactions (e.g., O<sub>2</sub> reduction and H<sub>2</sub> evolution). However, the last is critical, particularly for CO<sub>2</sub> RR because the conduction band (CB) has a significantly lower energy level than the one-electron reduction potential of CO<sub>2</sub> ( $E^\circ(\text{CO}_2/\text{CO}_2^{\bullet-}) = -1.97 \text{ V}$ ) [5]. Nevertheless, the formation of aliphatic acids and C<sub>1</sub> hydrocarbons at irradiated TiO<sub>2</sub>/liquid and TiO<sub>2</sub>/gas interfaces indicates that proton-coupled electron transfer (PCET) to adsorbed CO<sub>2</sub> is the dominant process. Furthermore, the presence of C<sub>≥2</sub> hydrocarbons indicates that the adsorbed C<sub>1</sub> intermediates are sufficiently long-lived for C–C coupling.

Many surface modifications (e.g., doping, heterojunction, and coupling with co-catalysts) have been attempted to enhance the photocatalytic activity for CO<sub>2</sub> RR (Table S1). Among them, doping with In<sup>3+</sup> is effective in creating oxygen vacancies in TiO<sub>2</sub> and reducing the charge recombination process [17]. Furthermore, indium-doping can increase the surface area and enlarge the bandgap ( $E_g$ ) of TiO<sub>2</sub> [18]. Previous studies showed that the inhibited charge recombination significantly increased CH<sub>4</sub> production with In-doped TiO<sub>2</sub>, whereas CO was the sole product with bare TiO<sub>2</sub> [17–19]. Coupling with carbon-based materials (e.g., graphitic carbon derivatives, graphene, graphene

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**Scheme 1.** Schematic illustration of the charge transfer mechanism with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> under (a) UV and (b) visible light. (c) Proposed elementary reaction pathway of photocatalytic CO<sub>2</sub> conversion into CO and hydrocarbons. H<sup>•</sup> represents a proton/electron (H<sup>+</sup>/e<sup>-</sup>) pair.

oxide, and carbon nanotubes) can also improve the TiO<sub>2</sub> activity. The electrically conductive carbon materials with a large surface area can serve as an adsorbent of CO<sub>2</sub> and catalyst of CO<sub>2</sub> RR because of their unique electronic properties [20,21]. Among carbon materials, graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) has a semiconductor property with a narrow E<sub>g</sub> (~2.7 eV). When coupled to TiO<sub>2</sub>, g-C<sub>3</sub>N<sub>4</sub> captures visible light (λ ~ 460 nm) and facilitates photogenerated charge transfer (Scheme 1) [22–24].

Therefore, this study synthesized In-doped TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> using sol-gel and thermal polymerization processes, respectively, and coupled them using an impregnation method. Considering different E<sub>g</sub>s of TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>, the photocatalytic activity of the as-synthesized samples (TiO<sub>2</sub>, In-TiO<sub>2</sub>, g-C<sub>3</sub>N<sub>4</sub>, TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>, and In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>) was examined for CO<sub>2</sub> RR under UV (λ = 365 nm) and visible light (λ > 420 nm). Particularly, the yield, total charge consumption, and photoelectrochemical features were analyzed, and charge carrier transfer dynamics were examined. Based on the experimental results, photocatalytic reaction mechanisms under UV and visible light were proposed. Finally, the as-synthesized composites were characterized using various surface analysis tools (XRD, XPS, Raman, FTIR, UV-vis absorption, BET, and HR-TEM).

## 2. Experimental section

### 2.1. Synthesis of materials

A series of In-TiO<sub>2</sub> samples were synthesized using a sol-gel method. Titanium tetrabutoxide (Ti(C<sub>4</sub>H<sub>9</sub>O)<sub>4</sub>, 4.8 mL, Sigma-Aldrich reagent grade, 97%) was added to a vigorously stirred aqueous solution containing *tert*-butanol (12.6 mL, Sigma-Aldrich, ACS reagent, 99.0%), acetic acid (1.4 mL, Sigma-Aldrich, ACS reagent, ≥99.7%), deionized water (18 MΩ cm, 1.2 mL), and various amounts of indium nitrate (In(NO<sub>3</sub>)<sub>3</sub>·xH<sub>2</sub>O, Sigma-Aldrich, 99.9%). The mole fractions of In (In / (Ti + In)) were 0.01–0.15; unless otherwise specified, the mole fraction was fixed to be 0.05. The as-obtained gel was aged for 24 h and dried at 105 °C, followed by calcination at 450 °C in the air for 4 h. g-C<sub>3</sub>N<sub>4</sub> was synthesized by thermal polymerization of melamine (Sigma-Aldrich, 99%) at 550 °C for 4 h in air. The as-synthesized In-TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> were mixed at a weight ratio of 3:7 in 50 mL acetone in an ultrasound bath for 60 min. Finally, the mixture was stirred in a fume hood for 24 h to remove the acetone [25]. The as-obtained yellow solid powder was ground and calcined in a muffle furnace at 300 °C for 2 h.

### 2.2. Surface characterizations

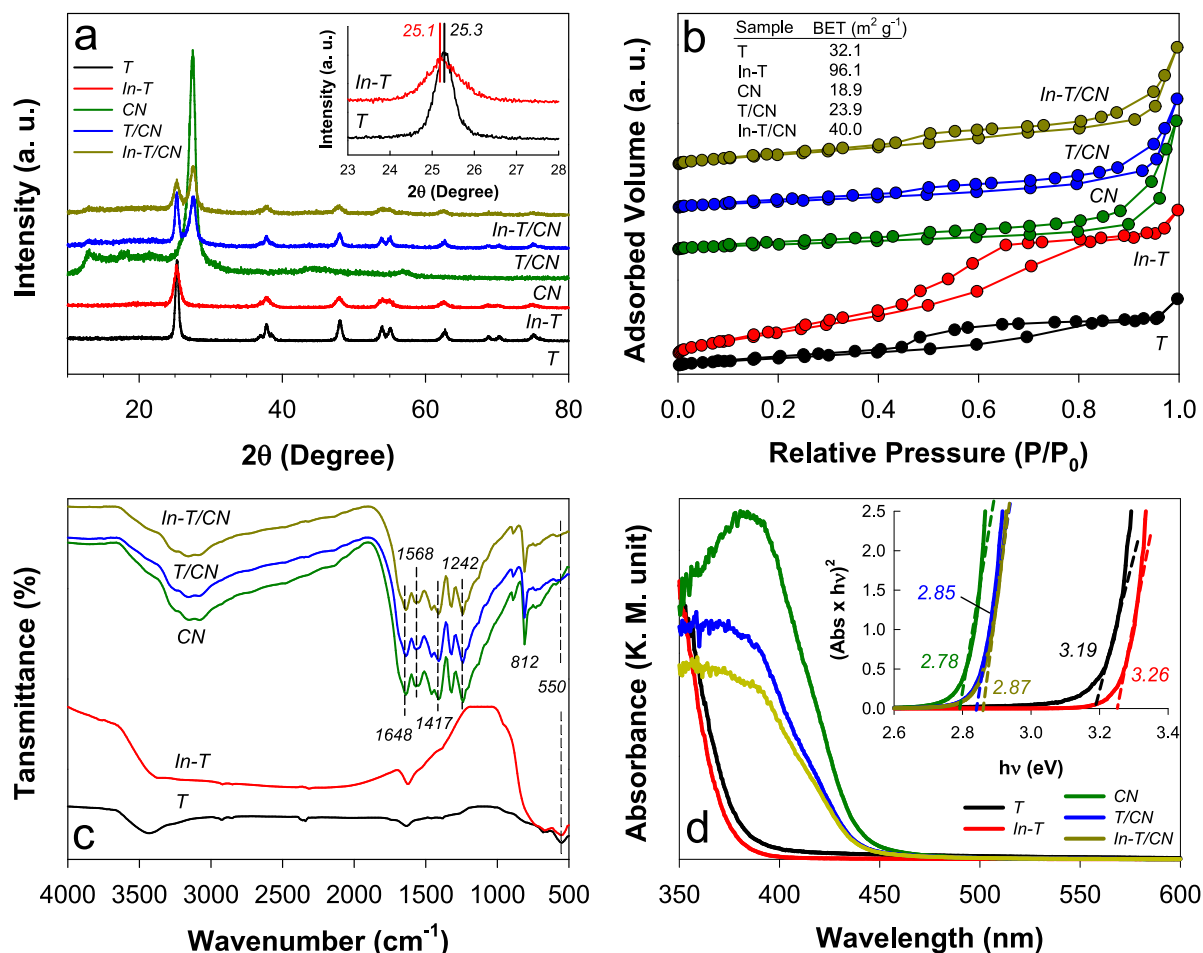
A low-temperature N<sub>2</sub> adsorption-desorption method (Quantachrome, Autosorb-iQ & Quadrasorb SI) was used to measure the BET surface area, pore-volume, and average pore diameter of the samples. High-resolution transmission electron microscopy (HR-TEM, FEI Company Titan G2) equipped with high-angle annular dark-field scanning TEM (HAADF-STEM) and energy-dispersive X-ray spectroscopy (EDS) was used to examine the morphology, lattice fringe, and elemental composition of the samples. The crystalline patterns and elemental states of the samples were examined using X-ray powder diffraction (XRD, Rigaku D/Max-2500) equipped with Cu Kα X-ray source (λ = 1.5408 Å) and X-ray photoelectron spectroscopy (XPS, ULVAC-PHI Quantera SXM) with Mg Kα monochromator X-ray source at 14 kV, respectively. The binding energies of the elements were calibrated with regards to C 1 s (284.8 eV). The diffuse reflectance UV-vis absorption spectra of the samples were obtained (Shimadzu, UV-2540). Sample powders were mixed with BaSO<sub>4</sub> and their reflectance spectra were measured with respect to BaSO<sub>4</sub>. Then, the reflectance (R) was converted into absorbance using the Kubelka-Munk equation; absorbance = (1 - R)<sup>2</sup>/2R. The Fourier transform infrared spectra (FTIR, PerkinElmer, Frontier) were collected in the wavenumber range of 400 cm<sup>-1</sup> and 4000 cm<sup>-1</sup> at a spectral resolution of 4 cm<sup>-1</sup>. An inverted-type scanning confocal microscope (MicroTime-200, Picoquant, Germany) with a 40× (air) objective was used for the time-resolved photoluminescence lifetime (TRPL) study. Lifetime measurements were performed at the Korea Basic Science Institute, Daegu Center, Korea. Single-mode pulsed diode lasers (λ = 375 nm with a ~ 30 ps pulse width) were used as excitation sources. More information on the analytical conditions can be found elsewhere [12,26].

The adsorption-desorption isotherms of CO<sub>2</sub> gas on the as-synthesized photocatalysts were obtained using a BET analyzer (BELSORP-mini II, Japan Microtrac-BEL, Corp). Highly pure CO<sub>2</sub> gas (99.99%) at a pressure of 0–1 bar was adsorbed and desorbed at 298 K. The saturation vapor pressure was 1 bar when the relative pressure (p/p<sub>0</sub>) was 1. The *in-situ* FTIR spectroscopic analysis was performed during photocatalytic reactions. The photocatalyst samples (50 mg) were placed in a 1 × 1 cm sample mount (RefractorReactor, Harrick). Before CO<sub>2</sub> adsorption, the samples were irradiated with simulated sunlight (100 mW cm<sup>-2</sup>) for 1 h while purging high purity air to remove carbonaceous impurities adsorbed onto the samples. After the pre-phototreatment, no FTIR signals associated with carbonaceous chemicals were found. The reactor was purged with high-purity of CO<sub>2</sub> (99.99%) for 30 min and sealed. The FTIR spectra were obtained before and during the irradiation in the wavenumber range of 600–4000 cm<sup>-1</sup> at a resolution of 4 cm<sup>-1</sup> (Thermo Scientific). For each spectrum, 40 scans were automatically averaged. All spectra were referenced to the spectrum with the pre-phototreated surface before CO<sub>2</sub> adsorption.

### 2.3. Photoelectrochemical measurements and photocatalytic CO<sub>2</sub> reduction

The transient photocurrents of the as-synthesized sample electrodes were obtained in a customized glass reactor containing 0.5 M Na<sub>2</sub>SO<sub>4</sub> aqueous solution using a three-electrode system with a saturated calomel electrode (SCE, reference electrode) and Pt wire (counter electrode). The sample electrodes were synthesized using a doctor blade method described elsewhere [27]. In brief, 5 mg of the sample powder was ultrasonicated in ethanol (10 mL) with a Nafion solution (5%, 20 μL, Sigma-Aldrich) and the sticky paste was coated on a fluorine-doped SnO<sub>2</sub>-coated glass electrode. The sample electrodes were dried in air for 1 h and in an oven at 120 °C for 2 h. While a potential of +0.5 V vs. SCE was applied to the sample electrodes, AM 1.5 light (100 mW cm<sup>-2</sup>) was irradiated from a Xenon arc lamp (300 W).

Photocatalytic reactions were performed in a closed gas reactor using a UV LED generating λ = 365 nm (LUNA fiber optics) and visible light (λ



**Fig. 1.** Surface characterization of the as-synthesized  $\text{TiO}_2$  (T),  $\text{In-TiO}_2$  (In-T),  $\text{TiO}_2/\text{g-C}_3\text{N}_4$  (T/CN), and  $\text{In-TiO}_2/\text{g-C}_3\text{N}_4$  (In-T/CN) samples. (a) XRD spectra (inset: magnified spectra of  $\text{TiO}_2$  and  $\text{In-TiO}_2$ ). (b)  $\text{N}_2$  adsorption–desorption isotherms (inset: BET surface areas). (c) FTIR spectra. (d) UV–vis absorption spectra (inset: optical analysis of bandgaps).

> 420 nm) of the simulated sunlight with an intensity of  $100 \text{ mW cm}^{-2}$  (300 W Xe lamp, ABET Technology). The wavelength intensity of the UV LED was estimated to be  $0.9025 \text{ mW cm}^{-2}$ , which was converted to an incident 365 nm-photon flux of  $1.658 \times 10^{18} \text{ s}^{-1}$ . To use visible light, the simulated sunlight passed through a cutoff filter ( $\lambda > 420 \text{ nm}$ , Newport). The catalyst powder (50 mg) was placed in the bottom of the stainless-steel reactor equipped with a quartz window for light penetration. Before photocatalysis, the samples were irradiated with light sources for 1 h while purging with high purity air to remove carbonaceous impurity adsorbed onto the samples. High-purity  $\text{CO}_2$  (99.99%) or  $\text{N}_2$  (99.999%) gases were then passed through a deionized water column and the reactor for 1 h. During irradiation (irradiated area:  $1 \text{ cm}^2$ ), the head-space gases (i.e., CO,  $\text{CH}_4$ , and  $\text{C}_2\text{H}_4$ ) were intermittently sampled and quantified using gas chromatography with a thermal conductivity detector (GC-TCD, Agilent 7820) and flame ionization detector (Young Lin ACME 6100 and Agilent 7820). To quantify CO and hydrocarbons, each standard gas (Fluka) was allowed to flow through the GC, and standard curve fits between each gas concentration and corresponding spectral area were obtained. Detailed analytical methods for CO and hydrocarbons have been described elsewhere [7,11,28]. After photocatalysis for 6 h, the photonic yield was estimated using the following equation:  $[2 \times \text{CO} (\text{mol}) + 8 \times \text{CH}_4 (\text{mol}) + 12 \times \text{C}_2\text{H}_4 (\text{mol})] \times 100\% / [\text{photon flux} \times 6 \text{ h}]$ , where 2, 8, and 12 represent the equivalent numbers of electron for production of CO,  $\text{CH}_4$ , and  $\text{C}_2\text{H}_4$ , respectively, from  $\text{CO}_2$ .

### 3. Results and discussion

#### 3.1. Surface characterization of the as-synthesized materials

The XRD patterns of the as-synthesized samples ( $\text{TiO}_2$ ,  $\text{In-TiO}_2$ ,  $\text{g-C}_3\text{N}_4$ ,  $\text{TiO}_2/\text{g-C}_3\text{N}_4$ , and  $\text{In-TiO}_2/\text{g-C}_3\text{N}_4$ ) were compared (Fig. 1a).  $\text{TiO}_2$  and  $\text{In-TiO}_2$  showed an anatase crystalline structure (e.g.,  $2\theta = 25.3^\circ$ ,  $37.6^\circ$ , and  $47.8^\circ$  for (101), (004), and (200) planes, respectively; JCPDS no. 21–1272) [29]. The  $\text{In-TiO}_2$  sample did not contain any indium-oriented peaks (e.g.,  $\text{In}_2\text{O}_3$ ) [30]. However, the (101) plane broadened and shifted to a low angle because of the In-oriented disorder of the  $\text{TiO}_2$  lattice (Fig. 1a inset). The XRD pattern of  $\text{g-C}_3\text{N}_4$  showed two characteristic peaks at  $2\theta = 13.1^\circ$  and  $27.4^\circ$ , corresponding to (100) and (002) planes of  $\text{g-C}_3\text{N}_4$ , respectively (JCPDS no. 87–1526) [31]. The  $\text{TiO}_2$  and  $\text{g-C}_3\text{N}_4$  composite samples (i.e.,  $\text{TiO}_2/\text{g-C}_3\text{N}_4$  and  $\text{In-TiO}_2/\text{g-C}_3\text{N}_4$ ) also showed the copresence of the crystal planes observed in the  $\text{TiO}_2$  and  $\text{g-C}_3\text{N}_4$ . This indicates that  $\text{g-C}_3\text{N}_4$  did not influence the  $\text{TiO}_2$  structure during the synthetic processes of the composites. In line with this, the Raman spectroscopy of the composites did not show any new vibration modes (Fig. S1).

The  $\text{N}_2$  adsorption–desorption isotherm measurement was used to evaluate pore distributions and specific surface areas of the as-synthesized samples (Fig. 1b). All samples showed mesoporous structures with hysteresis loops [32] and their textural properties were summarized in Table S2. Notably, In-doping increased the surface area of  $\text{TiO}_2$  by three times ( $32 \text{ m}^2 \text{ g}^{-1}$  to  $96 \text{ m}^2 \text{ g}^{-1}$ ) because of inhibited crystal growth. The as-synthesized, bulky  $\text{g-C}_3\text{N}_4$  showed the lowest

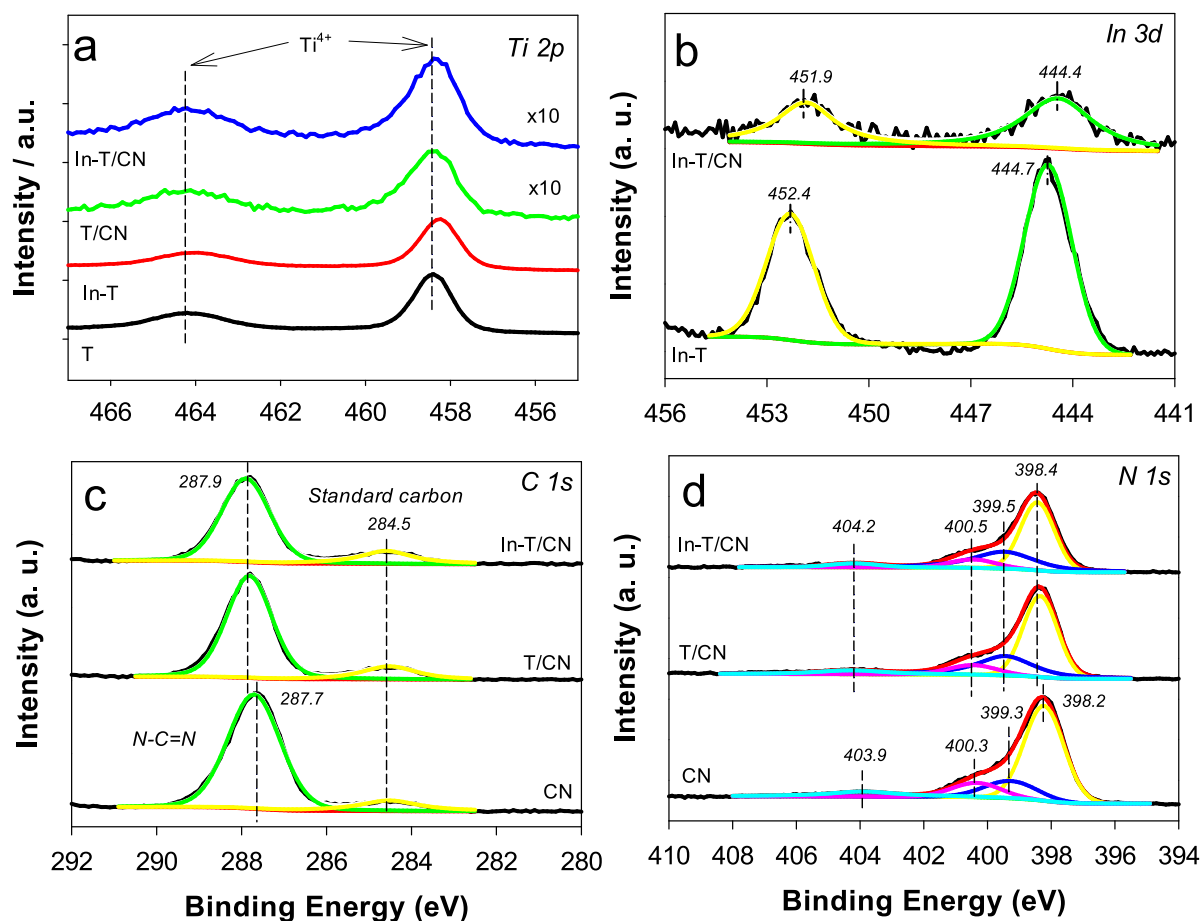


Fig. 2. XPS spectra of (a) Ti 2p, (b) In 3d, (c) C 1 s, and (d) N 1 s bands for In-TiO<sub>2</sub>, g-C<sub>3</sub>N<sub>4</sub>, TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>, and In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>.

surface area of  $\sim 18 \text{ m}^2 \text{ g}^{-1}$ . When g-C<sub>3</sub>N<sub>4</sub> was coupled with In-TiO<sub>2</sub>, its surface area increased twofold.

The surface functional groups of the composites were also examined using FTIR (Fig. 1c). TiO<sub>2</sub> and In-TiO<sub>2</sub> samples showed broad absorption bands at  $\sim 3500 \text{ cm}^{-1}$ , which were attributed to stretching vibrations of surface OH groups and adsorbed water molecules (H–O–H). The absorption at  $\sim 550 \text{ cm}^{-1}$  was attributed to the Ti–O–Ti stretching vibration [33]. For g-C<sub>3</sub>N<sub>4</sub>, a broad absorption band at  $3000\text{--}3600 \text{ cm}^{-1}$  was associated with N–H stretching vibration of amino groups and/or O–H stretching of adsorbed water molecules [29]. Furthermore, many absorption peaks in the wavenumber region of  $1200\text{--}1650 \text{ cm}^{-1}$  were attributed to the typical stretching modes of g-C<sub>3</sub>N<sub>4</sub>. For example, the aromatic C–N stretching resulted in absorption peaks at 1242, 1417, and  $1568 \text{ cm}^{-1}$ . The peaks at 1648 and  $812 \text{ cm}^{-1}$  could be attributed to the C–N stretching and characteristic breathing mode of triazine units, respectively [34]. All these characteristic bands of TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> were found in the composite samples, indicating that the chemical structures of both components remained intact during the synthetic processes of the composites.

The UV–vis reflectance absorption spectra of the as-synthesized samples were obtained to estimate the optical  $E_g$  values (Fig. 1d and S2). Compared to bare TiO<sub>2</sub>, In-TiO<sub>2</sub> exhibited a slightly blue-shifted absorption spectrum without any new absorption band. However, g-C<sub>3</sub>N<sub>4</sub> showed an absorption peak at  $\sim 400 \text{ nm}$  and an edge at  $\sim 450 \text{ nm}$ . The spectrum of TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> was similar to that of g-C<sub>3</sub>N<sub>4</sub>, but with a slight blue shift in the absorption edge. In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> followed the same tendency as TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>, with the reduced absorbance attributed to the In-doping effect [18]. Based on these spectra, the optical  $E_g$  values were estimated to be 3.19 eV (TiO<sub>2</sub>), 3.26 eV (In-TiO<sub>2</sub>), 2.78 eV (g-C<sub>3</sub>N<sub>4</sub>), 2.85 eV (TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>), and 2.87 eV (In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>) (Fig. 1d

inset). The increase in  $E_g$  caused by In-doping was partly attributed to In<sub>2</sub>O<sub>3</sub> with a large  $E_g$  (3.7 eV) [18]. In some cases, absorption in the visible region of 400–800 nm was observed with In-TiO<sub>2</sub> synthesized with an indium chloride precursor [35] because of the surface states of O–In–Cl<sub>x</sub> species. However, in this study, a different precursor (indium nitrate) was used and no visible light absorption with In-TiO<sub>2</sub> alone was observed.

The elemental states of the as-synthesized samples were examined using XPS analysis (Fig. 2 and S3). The bare TiO<sub>2</sub> spectrum showed Ti 2p bands at  $\sim 458.4 \text{ eV}$  (2p<sub>3/2</sub>) and  $\sim 464.2 \text{ eV}$  (2p<sub>1/2</sub>) associated with Ti<sup>4+</sup> states (Fig. 2a). In-TiO<sub>2</sub> showed the same Ti 2p bands, at 0.2 eV-lower binding energies [36]. This sample also contained two In 3d-bands at  $\sim 452.4 \text{ eV}$  (3d<sub>3/2</sub>) and  $\sim 444.7 \text{ eV}$  (3d<sub>5/2</sub>) (Fig. 2b), which were attributed to In<sup>3+</sup> species [35,37]. It appears that In<sup>3+</sup>-doping of the TiO<sub>2</sub> lattice partially reduced Ti<sup>4+</sup> to Ti<sup>3+</sup>, creating oxygen vacancies favorable for CO<sub>2</sub> adsorption [38–40]. Notably, the coupling of In-TiO<sub>2</sub> (or TiO<sub>2</sub>) and g-C<sub>3</sub>N<sub>4</sub> significantly influenced the binding energies of the elements in both materials. For example, the binding energy of the In 3d-bands decreased by  $\sim 0.4 \text{ eV}$  (Fig. 2b) and the C 1 s band in the bare g-C<sub>3</sub>N<sub>4</sub> was shifted to high binding energy by 0.2 eV. However, the binding energy of standard carbon was unchanged (Fig. 2c). Considering that the binding energy of C 1 s band in g-C<sub>3</sub>N<sub>4</sub> is susceptible to electronegativity of neighboring elements [29,41], such shift indicates a strong interaction between In-TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>. The bare g-C<sub>3</sub>N<sub>4</sub> also demonstrated N 1 s bands originating from C–N=C, N–(C)<sub>3</sub>, C–N–H, and  $\pi$ -excitation at binding energies of 398.6, 399.4, 400.9, and 404.2 eV, respectively (Fig. 2d) [42–44]. The sub-N 1 s bands were also shifted to high binding energy by 0.2 eV. Accordingly, these shifts in the binding energies of the component elements indicate strong electronic interaction between In-TiO<sub>2</sub> (or TiO<sub>2</sub>) and g-C<sub>3</sub>N<sub>4</sub>.

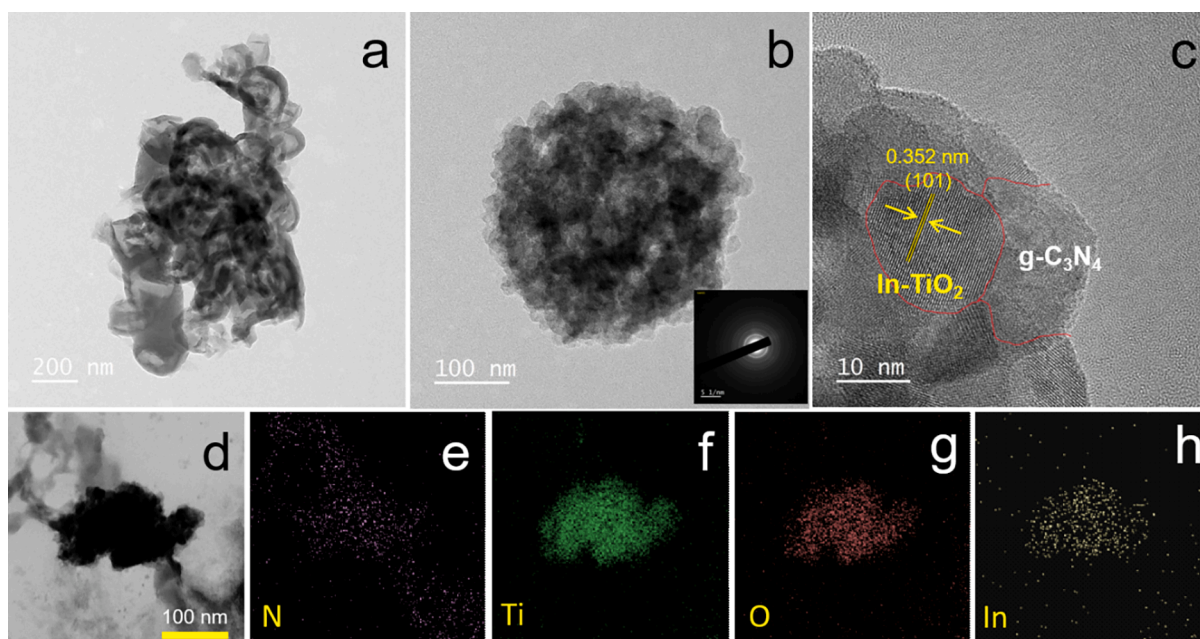


Fig. 3. (a and b) TEM, (c) HR-TEM, (d) HAADF-STEM image, and (e-h) EDS elemental mappings of In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>.

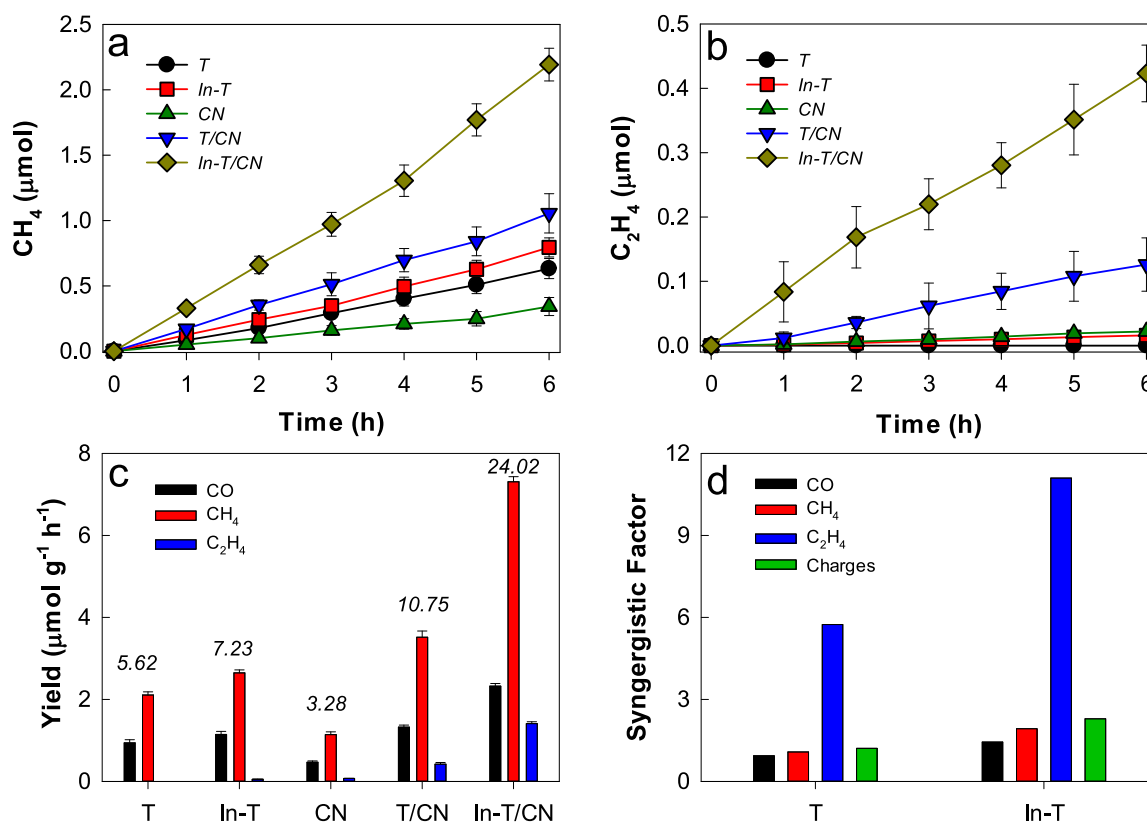


Fig. 4. Photocatalytic CO<sub>2</sub> conversions with the as-synthesized TiO<sub>2</sub>, In-TiO<sub>2</sub>, TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>, and In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> samples under UV irradiation. (a) CH<sub>4</sub> productions. (b) C<sub>2</sub>H<sub>4</sub> productions. (c) Comparison of product yields after 6 h. The numbers on the bars stand for the total elementary charges (μmol) used for the products (2e<sup>-</sup>, 8e<sup>-</sup>, and 12e<sup>-</sup> for CO, CH<sub>4</sub>, and C<sub>2</sub>H<sub>4</sub>, respectively). (d) Synergistic factors of TiO<sub>2</sub> and In-TiO<sub>2</sub> activities by g-C<sub>3</sub>N<sub>4</sub> for CO<sub>2</sub> conversion products and elementary charges. Synergistic factor = activity of TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> (or In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>) divided by the sum of TiO<sub>2</sub> (or In-TiO<sub>2</sub>) activity and g-C<sub>3</sub>N<sub>4</sub> activity.

The morphology of In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> composites was examined using TEM and HR-TEM (Fig. 3). The various morphologies of the same sample were attributed to the impregnation method which often results in irregular configuration. The lattice plane of anatase (101) with a lattice

spacing of 0.325 nm was observed, indicating In-TiO<sub>2</sub> inside the composites. While forming interfacial contact with In-TiO<sub>2</sub>, ~ 10 nm-thick g-C<sub>3</sub>N<sub>4</sub> was located on the edge of the composite. EDS mapping of the In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> composites showed that In and Ti were uniformly

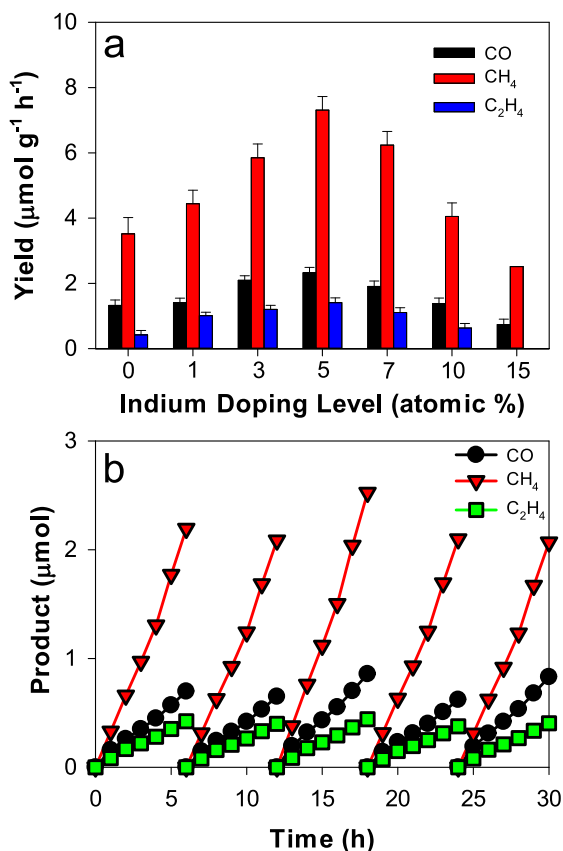


Fig. 5. Photocatalytic CO<sub>2</sub> conversions with the as-synthesized In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> samples under UV irradiation. (a) Effect of In-doping level. (b) Prolonged photocatalysis.

distributed in the same region. N atoms also were found in the In and Ti regions (Fig. S4) and further spread from the regions. This indicates that g-C<sub>3</sub>N<sub>4</sub> acts as a linker of In-TiO<sub>2</sub> particles while forming a composite.

### 3.2. Photocatalytic CO<sub>2</sub> conversion under UV and visible light

The photocatalytic CO<sub>2</sub> conversion was conducted using the as-synthesized samples under UV ( $\lambda = 365$  nm) (Fig. 4). The primary CO<sub>2</sub> conversion products were CO, methane, and ethylene, which were linearly produced with time (see Fig. S5 for CO production). The yield was 2–3 times greater than the CO yield for all samples. Trace levels of ethylene were produced on irradiated TiO<sub>2</sub>, In-TiO<sub>2</sub>, and g-C<sub>3</sub>N<sub>4</sub>, whereas the amount of ethylene was comparable to CO with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>. When the reactor was filled with N<sub>2</sub> instead of CO<sub>2</sub>, no carbonaceous compound was detected (Fig. S6). This implies that the purged CO<sub>2</sub> was the sole carbon source of the compounds.

The photocatalytic activity of bare and modified samples was also compared. TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> showed poor activity, whereas TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> showed significant photocatalytic activity (Fig. 4a and 4b). For example, TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> produced 1.7 times and 3 times more methane than those with TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>, respectively. A similar enhancement was similarly found for CO (Fig. S5). In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> demonstrated a more pronounced coupling effect. The amount of methane with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> was  $\sim 2.8$  times and 6.4 times larger than those with In-TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>, respectively (Table S3). Notably, ethylene production with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> was 26.5 times and  $\sim 19$  times larger than that with In-TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>, respectively. The synergistic factors [i.e., the activity of In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> (or TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>) divided by the sum of In-TiO<sub>2</sub> (or TiO<sub>2</sub>) activity and g-C<sub>3</sub>N<sub>4</sub> activity] were estimated for CO, methane, and ethylene (Fig. 4c). The synergistic factors for CO, methane, and ethylene by g-C<sub>3</sub>N<sub>4</sub> with TiO<sub>2</sub>, were 0.9, 1.1, and 5.7, respectively; with In-TiO<sub>2</sub>,

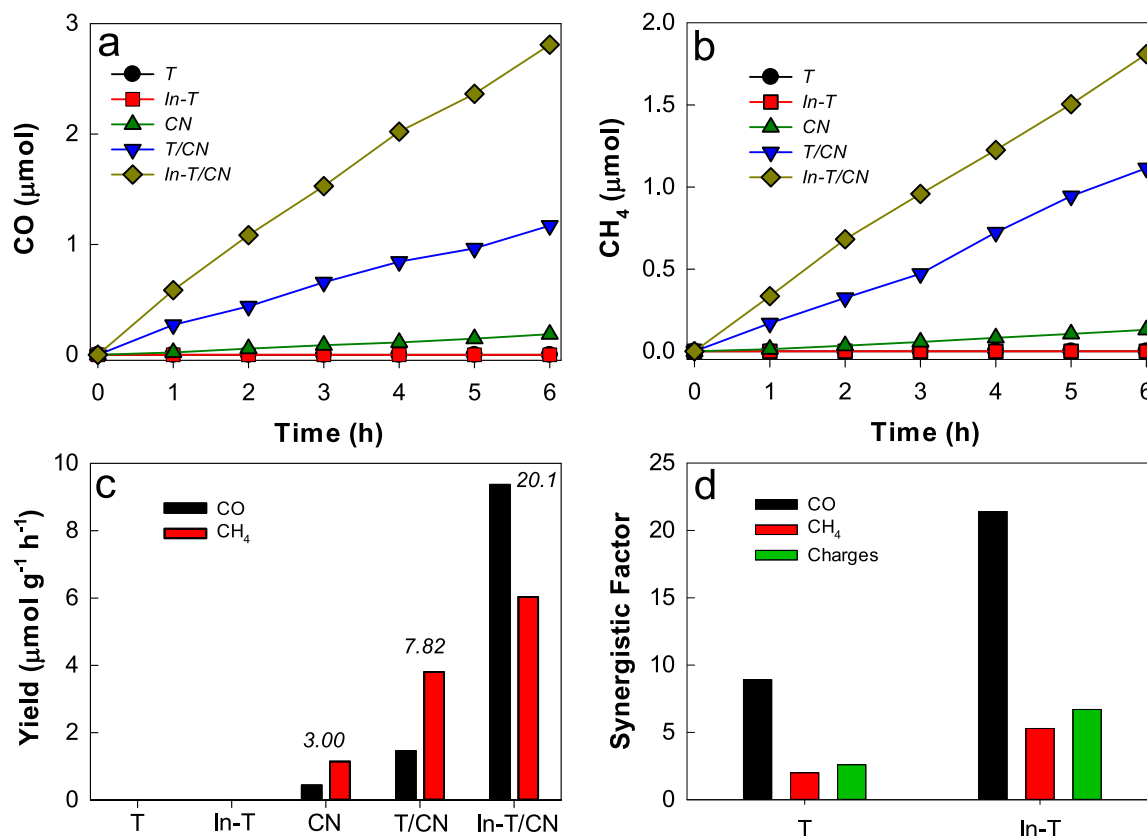
they were 1.4, 1.9, and 11, respectively. This shows that the coupling effect of g-C<sub>3</sub>N<sub>4</sub> became more significant as more reduced chemicals were produced ( $12e^-$ ,  $8e^-$ , and  $2e^-$ -transfers for ethylene, methane, and CO, respectively). Notably, the fractional ratios of CO, methane, and ethylene were  $\sim 29\%$ ,  $68\%$ , and  $< 4\%$ , respectively, with each In-TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>, whereas with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>, they changed to 21%, 66%, and 13%, respectively. This indicates that In-doping facilitated the charge transfer kinetics in increasing the ethylene selectivity. The total number of photogenerated charges used for CO<sub>2</sub> conversion (for 6 h) was  $\sim 10.8$   $\mu\text{mole}$  with TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> and 24.0  $\mu\text{mole}$  with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> (Fig. 4c), leading to the synergistic factors of  $\sim 1.2$  and 2.3, respectively (Fig. 4d). The overall photonic yield with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> was 40.4% (2.4% for CO, 29.5% for methane, and 8.5% for ethylene) (Table S3).

The as-observed activity of In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> significantly depended on the In-doping level (Fig. 5a). The yield products of CO, methane, and ethylene were greatest with the In-doping of 5 atomic%. The existence of the optimal doping level indicates that In played a critical role in the photogenerated charge transfer kinetics (discussed below). CO was always observed regardless of the In-doping level, whereas neither H<sub>2</sub> nor other carbonaceous compounds (e.g., formate) were detected in all cases. The proton-coupled electron ( $H^+/e^-$  or  $H^*$ ) transfer (PCET) appeared to be the predominant pathway of methane and ethylene production and the direct CO<sub>2</sub> reduction by photogenerated electrons did not occur (Scheme 1). A prolonged photocatalysis over 30 h with the as-synthesized In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> showed continuous production of carbonaceous compounds with no sign of material deactivation (Fig. 5b). This stability with high activity is impressive because materials with high activities usually suffer from rapid deactivation whereas those with low activities exhibit relatively prolonged stability [6,11–13]. It appears that an efficient charge transfer between In-TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> effectively inhibits the photo-induced deactivation (e.g., photocorrosion) of each component.

The photocatalytic activity of the as-synthesized samples was also examined under visible light ( $\lambda > 420$  nm) (Fig. 6). Any CO<sub>2</sub> conversion product was not observed with TiO<sub>2</sub> and In-TiO<sub>2</sub> under visible light (Fig. 6a and 6b), because of their  $E_g$  of  $\sim 3.2$  eV. However, g-C<sub>3</sub>N<sub>4</sub> with  $E_g$  of 2.78 eV generated photo charges and produced CO and CH<sub>4</sub> at the ratio of  $\sim 3:7$ . Therefore, g-C<sub>3</sub>N<sub>4</sub> was essential for driving the CO<sub>2</sub> reduction reaction under visible light. The overall tendency of the activity was similar to that under UV. In-doping and g-C<sub>3</sub>N<sub>4</sub>-coupling enhanced TiO<sub>2</sub> activity, leading to the highest activity with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> (Fig. 6c). The optimal In-doping level was the same as that observed under UV (Fig. S7). However, CO was the most abundant with  $\sim 60\%$  selectivity, whereas ethylene was not produced. The synergistic factors with TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> for the CO and methane production were 9 and 2, respectively; the values with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> were 21 and 5, respectively (Fig. 6d). The total number of photogenerated charges used for CO<sub>2</sub> conversion (for 6 h) was 7.8  $\mu\text{mole}$  with TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> and 20  $\mu\text{mole}$  with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> (Fig. 6c), resulting in synergistic factors of 2.6 and 6.7, respectively (Fig. 6d).

### 3.3. Photoinduced charge transfer and photocatalytic CO<sub>2</sub> conversion mechanism

To examine the In-doping and g-C<sub>3</sub>N<sub>4</sub>-coupling effects on the photoinduced behavior, the PL emission spectra of the as-synthesized samples were compared (Fig. 7a). Upon excitation at  $\lambda = 375$  nm, all the samples exhibited the similar spectral responses with broad emission bands in the wavelength range from 400 to 700 nm. The main emission band with bare TiO<sub>2</sub> was found at  $\lambda = \sim 630$  nm because of the recombination of trapped electrons and holes [45,46]. The emission band was shifted to  $\lambda \sim 560$  nm and its intensity was reduced because of the In 5 s state [17]. In contrast to these TiO<sub>2</sub>-based PL spectra, the g-C<sub>3</sub>N<sub>4</sub> generated a significant emission at  $\lambda \sim 470$  nm. Notably, this band was significantly reduced when coupled to In-TiO<sub>2</sub>, indicating that the



**Fig. 6.** Photocatalytic CO<sub>2</sub> conversions with the as-synthesized TiO<sub>2</sub>, In-TiO<sub>2</sub>, TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>, and In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> samples under visible light. (a) CO productions. (b) CH<sub>4</sub> productions. (c) Comparison of product yields after 6 h. The numbers on the bars stand for the total elementary charges (μmol) used for the products (2e<sup>-</sup>, 8e<sup>-</sup>, and 12e<sup>-</sup> for CO, CH<sub>4</sub>, and C<sub>2</sub>H<sub>4</sub>, respectively). (d) Synergistic factors of TiO<sub>2</sub> and In-TiO<sub>2</sub> activities by g-C<sub>3</sub>N<sub>4</sub> for CO<sub>2</sub> conversion products and elementary charges.

charge recombination was inhibited in In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> [29,47].

TRPL decay spectra of the samples excited at  $\lambda = 375$  nm were compared (Fig. 7b) to gain insights into the charge transfer kinetics. The PL intensity of TiO<sub>2</sub> decayed rapidly with an average decay lifetime ( $\tau$ ) of 3.19 ns, whereas In-TiO<sub>2</sub> showed a twofold slower  $\tau$  (7.4 ns). This confirms that charge carriers survived for longer periods and their recombination is retarded with In-TiO<sub>2</sub>. Notably,  $\tau$  with g-C<sub>3</sub>N<sub>4</sub> increased from 5.8 ns to 6.73 ns and 18 ns when g-C<sub>3</sub>N<sub>4</sub> was coupled with TiO<sub>2</sub> and In-TiO<sub>2</sub>, respectively. The slowest PL decay with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> was attributed to the synergetic effect of In-associated trap sites and delocalized  $\pi$ -electrons over the molecular framework in g-C<sub>3</sub>N<sub>4</sub> [48]. This should inhibit the recombination of photogenerated charge carriers by facilitating the interfacial charge transfer, allowing the charge carriers to survive long enough for reaction with adsorbed CO<sub>2</sub>. The 2D PL images of the as-synthesized samples further revealed that the charge carrier transfer kinetics were relatively uniform over the composite samples (Fig. 7b inset). The strong junction between In-TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> appears to effectively separate electrons and holes, resulting in a low PL emission.

A comparison of photocurrent density ( $J_{ph}$ ) verified the inhibited charge recombination with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> (Fig. 7c). Bare TiO<sub>2</sub> generated small  $J_{ph}$  of  $< 2 \mu\text{A cm}^{-2}$ , and In-doping slightly enhanced  $J_{ph}$ . g-C<sub>3</sub>N<sub>4</sub> showed a lower activity than TiO<sub>2</sub> and In-TiO<sub>2</sub>; however,  $J_{ph}$  with g-C<sub>3</sub>N<sub>4</sub> was stable over time.  $J_{ph}$  with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> was  $5 \mu\text{A cm}^{-2}$ , which was greater than each In-TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>, as well as the sum of both. Furthermore, it was maintained over time without a sign of decrease. TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> showed the same tendency.

The CO<sub>2</sub> adsorption isotherms were also obtained to examine the CO<sub>2</sub> adsorption capacities of the as-synthesized photocatalysts (Fig. 8). All adsorption-desorption isotherms were found to be reversible without hysteresis (Fig. S8), indicating that CO<sub>2</sub> is physically adsorbed

without forming chemical bonds. The CO<sub>2</sub> adsorbed on bare TiO<sub>2</sub> was estimated to be  $0.13 \text{ mmol g}^{-1}$ , whereas In-doping enhanced CO<sub>2</sub> adsorption capacity by nearly twofold ( $0.25 \text{ mmol g}^{-1}$ ). This In-doping effect was similarly found in the BET surface area, which was  $\sim 3$ -fold enhanced upon In-doping (Fig. 1b). Note that the CO<sub>2</sub> adsorption capacity of g-C<sub>3</sub>N<sub>4</sub> ( $\sim 0.08 \text{ mmol g}^{-1}$ ) was only 25% that of In-TiO<sub>2</sub>. In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>, with a weight ratio of 3/7 had a lower capacity than that for In-TiO<sub>2</sub> without g-C<sub>3</sub>N<sub>4</sub> (the same for TiO<sub>2</sub>). Therefore, In-TiO<sub>2</sub> should act as a CO<sub>2</sub> adsorbent and photocatalyst, whereas g-C<sub>3</sub>N<sub>4</sub> mainly acts as a photocatalyst in the CO<sub>2</sub> RR with the In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> composites. The tendency for the CO<sub>2</sub> adsorption capacity was similar to that for the BET area (Fig. 8 inset).

To gain insight into the reaction mechanism, in-situ FTIR spectra of In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> during photocatalysis under UV and visible light were obtained (Fig. 9). The surface without CO<sub>2</sub> adsorption was clean with no specific absorption signals. However, the surface exposed to CO<sub>2</sub> gas exhibited many absorption signals at wavenumbers of 944–1076 cm<sup>-1</sup> (CO<sub>2</sub>·H<sub>2</sub>O, HCO<sub>3</sub><sup>-</sup>, and CO<sub>3</sub><sup>2-</sup>), 2345 cm<sup>-1</sup> (CO<sub>2</sub>), and 3625 cm<sup>-1</sup> (HCO<sub>3</sub><sup>-</sup> or CO<sub>2</sub>) [49,50]. With UV irradiation, two absorption bands appeared at 2985 cm<sup>-1</sup> and 3155 cm<sup>-1</sup> corresponding to C–H stretching vibrations of CH<sub>4</sub> and C<sub>2</sub>H<sub>4</sub>, respectively (Fig. 9a). The wavenumber of 1295 cm<sup>-1</sup> was also attributed to the C–H stretching vibration of CH<sub>4</sub>. The surface irradiated with visible light showed similar spectral patterns, except for the absence of absorbance at 3155 cm<sup>-1</sup> (C<sub>2</sub>H<sub>4</sub>) (Fig. 9b). This is consistent with the gaseous products obtained under UV (CO, CH<sub>4</sub>, and C<sub>2</sub>H<sub>4</sub>) and visible light (CO and CH<sub>4</sub>). The absence of CO stretching mode in the wavenumber range of 2100–2200 cm<sup>-1</sup> was attributed to the limited adsorption of CO on In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> (Fig. S9).

All observed results indicate that In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> is active for photoinduced charge transfer and photocatalytic CO<sub>2</sub> conversion under UV and visible light. Under UV, In-TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> are both activated;

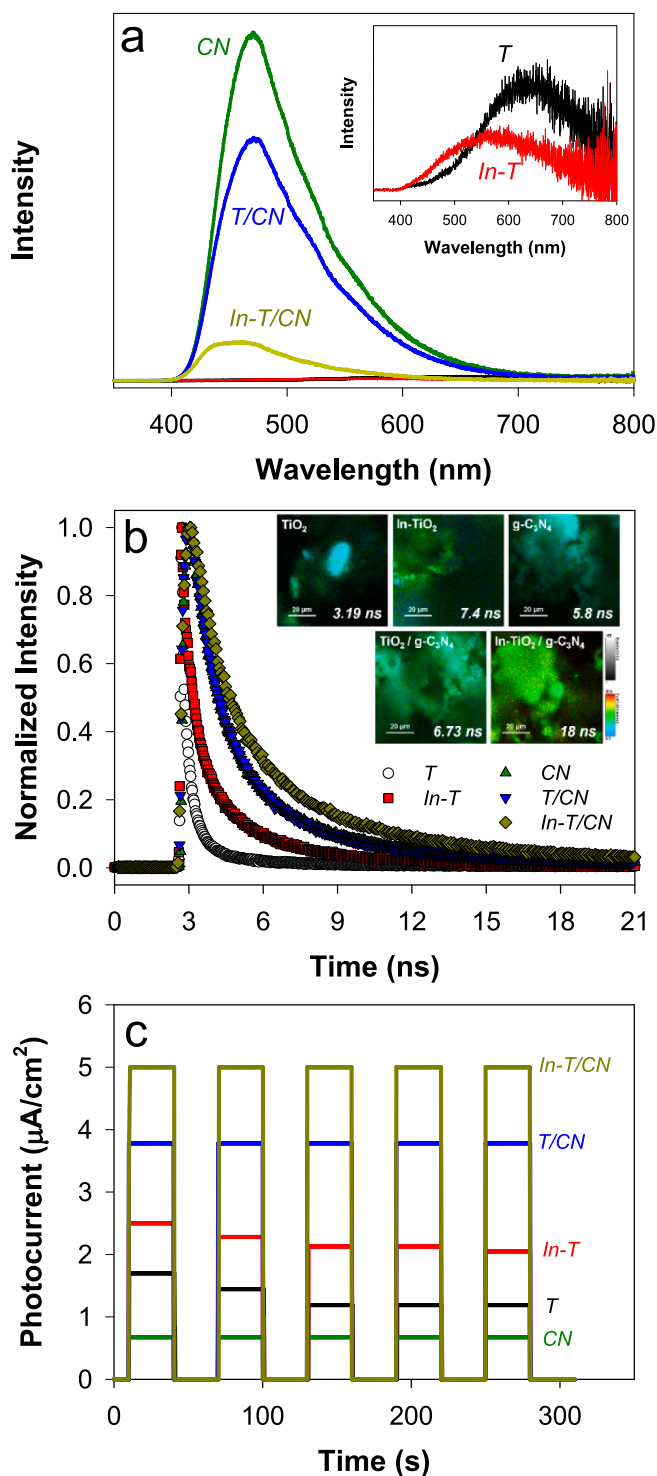


Fig. 7. Spectroscopic and photoelectrochemical analysis of the as-synthesized samples. (a) Photoluminescence spectra excited at  $\lambda = 375$  nm. The inset shows the magnified spectra of  $\text{TiO}_2$  and  $\text{In-TiO}_2$ . (b) Time-resolved photoluminescence emission decay spectra. Excited at  $\lambda = 375$  nm. The inset shows the two-dimensional emission intensity and lifetime images. The numbers in the images represent the average lifetimes ( $\tau$ ). (c) Chopped photocurrent profiles at  $E = 0.5$  V vs. SCE in 0.5 M sodium sulfate solutions.

under visible light, only  $\text{g-C}_3\text{N}_4$  is activated (Scheme 1). The different band energy levels of the two semiconductors can induce cascaded charge transfer under UV [51]. Therefore, regardless of the irradiation condition, the photogenerated electrons in  $\text{g-C}_3\text{N}_4$  are transferred to  $\text{In-TiO}_2$  with high  $\text{CO}_2$  adsorption capacity and the photogenerated hole-

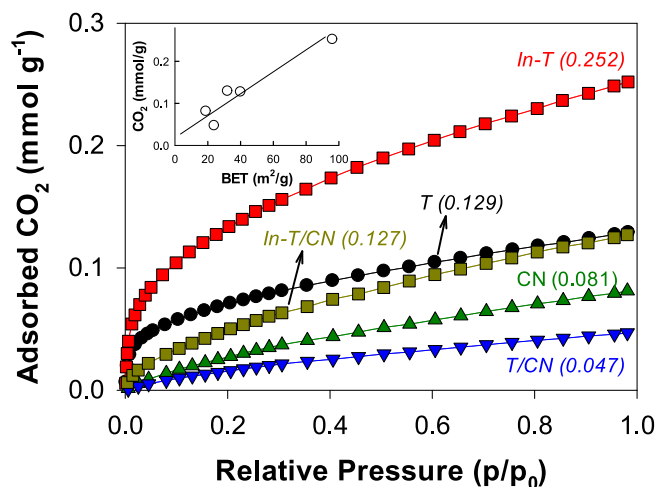


Fig. 8.  $\text{CO}_2$  adsorption isotherms with the as-synthesized samples at 298 K. The numbers in parentheses represent the amounts of adsorbed  $\text{CO}_2$  at 1 atm  $\text{CO}_2$  gas ( $\text{mmol g}^{-1}$ ). Inset shows the plot between the BET surface areas and the amounts of adsorbed  $\text{CO}_2$  for the same samples.

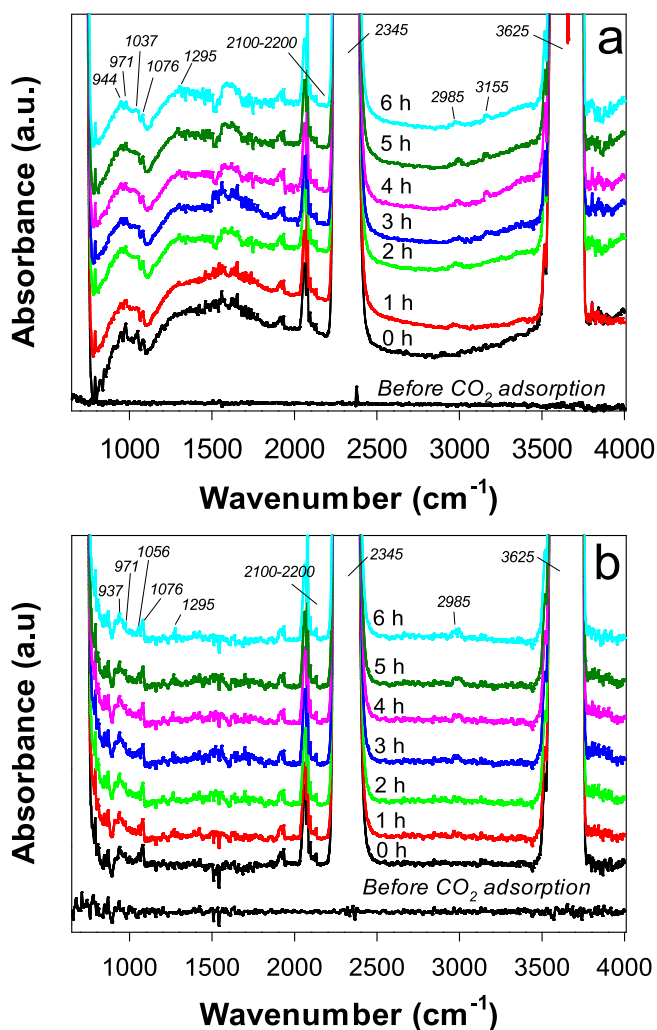


Fig. 9. In-situ FTIR spectra of  $\text{In-TiO}_2/\text{g-C}_3\text{N}_4$  surface in the  $\text{CO}_2$  atmosphere under (a) UV and (b) visible light.



reaction occurs on g-C<sub>3</sub>N<sub>4</sub> with low CO<sub>2</sub> adsorption capacity. The high  $J_{ph}$  and long  $\tau$  with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> (see Fig. 7) indicate that the charge separation occurs efficiently, and the separated charge carriers sufficiently survive for transfer to adsorbed CO<sub>2</sub> (\*CO<sub>2</sub>). According to density functional theory computations, the CO<sub>2</sub> adsorption energy on TiO<sub>2</sub> surface is strongly influenced by crystalline structures and oxygen vacancies [38,52,53]. For the anatase (101) surface, the monodentate binding modes (e.g., O-C-O – Ti<sub>surf</sub> and O-C – O<sub>surf</sub>) have the exothermic adsorption energies between – 7.8 and – 11 kcal mol<sup>-1</sup>, whereas the adsorption energy for the bidentate modes significantly decreases. Notably, the presence of oxygen vacancies largely enhances the CO<sub>2</sub> adsorption energies for both binding modes. For example, the adsorption energy of CO<sub>2</sub> via the monodentate mode for the anatase TiO<sub>2</sub> (101) with oxygen vacancies (–15.5 kcal mol<sup>-1</sup>) is greater than the adsorption energy of CO<sub>2</sub> with g-C<sub>3</sub>N<sub>4</sub> (–9.64 kcal mol<sup>-1</sup>) [38,40,54]. Considering that In-doping induced Ti<sup>3+</sup> species with oxygen vacancies (see Fig. 2a), In-TiO<sub>2</sub> should be better than g-C<sub>3</sub>N<sub>4</sub> for CO<sub>2</sub> adsorption.

The CO production under UV and visible light also indicates that the photogenerated electrons are transferred to a carbon atom of \*CO<sub>2</sub> in the initial stage [19,32,55]. If an oxygen atom of \*CO<sub>2</sub> is an electron acceptor, the aliphatic acids should have been produced instead [6,11,13,56]. In-TiO<sub>2</sub> has a lower CB than the one-electron reduction potential of free CO<sub>2</sub> [ $E^{\circ}(\text{CO}_2/\text{CO}_2^{\bullet-}) = -1.97 \text{ V}$ ] [57]. Accordingly, PCET (not electron transfer) should be the primary pathway for \*CO<sub>2</sub> reduction. Despite similar thermochemical potentials for H<sub>2</sub> and CO production [ $E^{\circ}(\text{H}_2\text{O}/\text{H}_2) = 0 \text{ V}$ ;  $E^{\circ}(\text{CO}_2/\text{CO}) = -0.106 \text{ V}$ ], the absence of H<sub>2</sub> production indicates that the dimerization of two H<sup>•</sup>s did not occur and that the reaction between H<sup>•</sup> and CO<sub>2</sub> was dominant (Scheme 1c). The produced CO should undergo stepwise PCET, forming methane and ethylene. Such photocatalyzed Fischer-Tropsch reaction was also observed in previous studies [7,28]. Finally, the photogenerated holes were speculated to oxidize adsorbed water to surface-bound OH radicals (>OH<sup>•</sup>) and H<sup>+</sup>. The dimerization of the two radical species can form hydrogen peroxide, which can be actively involved in photochemical and photocatalytic reactions. The absence of O<sub>2</sub> in the reactor headspace supports this speculation.

#### 4. Conclusions

This study demonstrated that In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> composites synthesized using a facile impregnation method converted CO<sub>2</sub> into value-added chemicals with high efficiency under UV and visible light. The primary CO<sub>2</sub> conversion products were CO, methane, and ethylene under UV, and CO and methane under visible light. A prolonged photocatalysis with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> produced the same carbonaceous compounds for 30 h, with a photonic yield of ~ 40% (2.4% for CO, 29.5% for methane, and 8.5% for ethylene). The enhanced UV and visible light activities were attributed to synergistic effects of doping with In and coupling with g-C<sub>3</sub>N<sub>4</sub>. While In-doping facilitated charge transfer via In<sup>3+</sup>-associated trap sites, a strongly coupled g-C<sub>3</sub>N<sub>4</sub> induced cascaded charge transfer. The photoelectrochemical response and time-resolved photoluminescence spectra confirmed the significantly inhibited charge recombination and prolonged charge carriers, particularly with In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>. The CO<sub>2</sub> adsorption isotherms indicated that In-TiO<sub>2</sub> acts as a primary CO<sub>2</sub> adsorbent and photocatalyst, whereas g-C<sub>3</sub>N<sub>4</sub> mainly acts as a photocatalyst. In-situ FTIR study further revealed the formation of CH<sub>4</sub> and C<sub>2</sub>H<sub>4</sub> under UV and CH<sub>4</sub> under visible light on In-TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>. Considering that In-TiO<sub>2</sub> surface is the same CO<sub>2</sub> reduction site under UV and visible light, the absence of ethylene production under visible light was attributed to insufficient PECT in the last stage of the stepwise CO<sub>2</sub> reduction (CO<sub>2</sub> → CO → CH<sub>4</sub> → C<sub>2</sub>H<sub>4</sub>).

#### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence

the work reported in this paper.

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#### Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.cej.2022.135388>.

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