

Ion-Beam induced grain rotation in nanocrystalline alumina

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Single crystals of α -alumina (orientation $[11\bar{2}0]$, $15 \times 10 \times 1 \text{ mm}^3$) were bombarded at room temperature with 4.8 MeV/u Au ions. The transformation of the single crystals into nanocrystals was followed by monitoring the $(11\bar{2}0)$ -diffraction peak by $\text{CuK}\alpha$ -radiation at the in-situ 3-circle-diffractometer at the UNILAC M2-beamline. The ω -circle (incident x-ray beam) and the θ -circle (diffracted beam) define the diffraction plane. The χ -circle enables a sample tilt relative to the diffraction plane; for $\chi = 0$ the sample normal lies in the diffraction plane and coincides with the ion beam direction. During irradiation χ_i was $\pm 45^\circ$. X-ray measurements were done by varying χ and $\omega = \theta$ fixed at the Bragg angle θ_B . The single peak originally located at $\chi = 0$ splits with increasing fluence into two peaks, one located at $\chi_1 = 0$, originating from unmodified material at greater specimen depths, and a second one, located at χ_2 , resulting from tilted $(11\bar{2}0)$ -lattice planes. The tilt angle $\Omega = -\chi_2$ versus fluence is shown in fig. 1. At a fluence of $1 \times 10^{13} \text{ Au/cm}^2$ χ_i has moved from $+45^\circ$ to -45° . The rotation direction also changed its sign. A symmetric Bragg scan at $\chi = \chi_2$ yields the precise position of θ_B and the peak width of the modified alumina. θ_B decreased by about 0.1° , which is attributed to a volume increase by dislocation production. The width of the modified peak is shown in fig. 2. The change in χ_i is not visible in the behavior of the width.

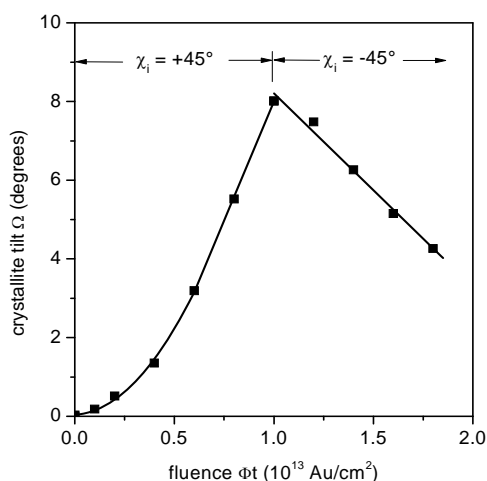


Figure 1: Crystallite tilt Ω versus ion fluence.

Both grain rotation and the square-root dependence of the width (see fig. 2) indicated the action of dislocations. Because the number density of dislocations is of the order of Φt , most of the dislocations are concentrated in newly formed grain boundaries. Thus, at $\Phi t \sim 10^{12} \text{ Au/cm}^2$ the

single crystal is transformed into an aggregate of nanocrystals. The evolution of the grain size is depicted in fig. 3 for two data evaluation strategies.

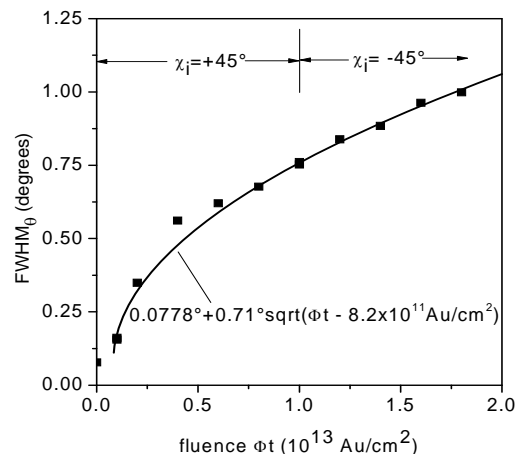


Figure 2: The full width at half maximum FWHM_θ of the $(11\bar{2}0)$ -Bragg peak of α -alumina as function of ion fluence.

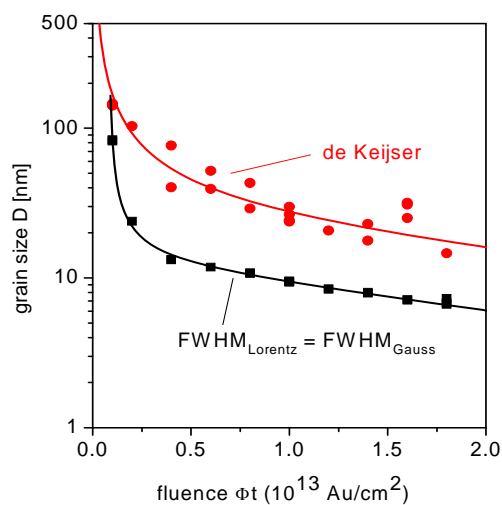


Figure 3: Grain size D versus ion fluence derived from the separation of grain size and strain contributions to FWHM_θ . The black curve has been obtained with the usual assumption that the Gaussian and the Lorentzian widths to the Pseudo-Voigt peak widths are equal. The red curve has been obtained by a procedure proposed by de Keijser [1]

[1] de Keijser et al., J. Appl. Cryst. 15(1982) 308-314.

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