PAPER • OPEN ACCESS

Magnetocaloric properties of multicomponent Laves phase compounds and their composites

To cite this article: J wik et al 2021 J. Phys.: Conf. Ser. 1758 012009

View the article online for updates and enhancements.

You may also like

- <u>Ionic Conduction and Fuel Cell</u> <u>Performance of Ba_{0.97}Ce_{0.8}Ho_{0.2}O₃ <u>Ceramic</u> Li-gan Qiu</u>

- Temperature probing and emission color tuning by morphology and size control of <u>upconverting</u> -<u>NaYb_{0.07}Gd_{0.30}F₄:Tm_{0.015}:Ho_{0.015} nanoparticles Emille M Rodrigues, Diogo A Gálico, Italo O Mazali et al.</u>

- <u>Competing effects in the magnetic</u> <u>polarization of non-magnetic atoms</u> R Boada, M A Laguna-Marco, C Piquer et al.



This content was downloaded from IP address 89.247.147.32 on 23/03/2022 at 08:41

Magnetocaloric properties of multicomponent Laves phase compounds and their composites

J Ćwik¹, Yu Koshkid'ko¹, K Nenkov² and N Kolchugina³

¹ Institute of Low Temperature and Structure Research, Polish Academy of Sciences, 50-950, Wroclaw, Poland; j.cwik@intibs.pl

1758 (2021) 012009

² IFW Dresden, Institute of Metallic Materials, D-01171, Dresden, Germany

³ Baikov Institute of Metallurgy and Materials Science, Russian Academy of Sciences, 119334, Moscow, Russia

Abstract. Heat capacity measurements have been performed for multicomponent $(Ho_{0.9}Er_{0.1})_{1,x}Gd_xCo_2$ compounds with x = 0.05, 0.1, and 0.15. The isothermal magnetic entropy change, ΔS_{max} , allowing the estimation of the magnetocaloric effect, was determined based on the heat capacity measurements in magnetic fields up to 2 T. A numerical method, with the magnetic entropy change of individual $(Ho_{0.9}Er_{0.1})_{1-x}Gd_xCo_2$ compounds, was used to calculate the optimal molar composition of the constituents and the resulting change of the isothermal magnetic entropy of composite, ΔS_{comp} . The results show that proposed composite can be considered as a refrigerant material in magnetic refrigerators performing an Ericsson cycle in a temperature range of 90-130 K.

1. Introduction

The magnetocaloric effect (MCE) consists in the magnetic materials' thermal response when they are subjected to magnetic field variations. It can be quantified as the reversible change in temperature, ΔT_{ad} , during adiabatic process, or the reversible change of the magnetic entropy, ΔS_{mag} , if the change in field takes place in near isothermal process. Both ΔT_{ad} and ΔS_{mag} are characteristic values of the MCE, according to the initial temperature and the value of change in the magnetic field. This effect is particularly pronounced at temperatures and fields corresponding to magnetic phase transitions.

It has been found that in the case of RCo_2 with heavy rare earth elements (R = Ho, Dy and Er) the magnetic phase transitions are of the first order and these compounds exhibit large magnetic entropy changes [1,2]. Numerous investigations of the MCE of RCo₂ compounds with substitutions either in rare - earth sublattice (RR')Co₂ or in Co - sublattice $RCo_{2-x}M_x$ (where M = Si, Ga, Ge, Al, Fe, Ni, Mn) have been performed [3,4], while the multicomponent (RR'R")Co₂ alloys lack systematic investigations of their thermal properties [5,6]. In one of our previous works, the magnetic and magnetocaloric properties of such compositions were studied [7]. In the present work, $Ho_{0.9}Er_{0.1}Co_2$ compound was selected as a potential candidate for refrigerant in magnetic refrigerators. This compound is characterized by the first-order magnetic phase transitions with $T_{\rm C} = 72$ K, and $\Delta S_{\text{mag}} = 14.2 \text{ J/kgK}$ at a magnetic field change of 0 to 3 T [7]. In order to expand the range of applications, the GdCo₂ compound was used for the partial substitution for Ho_{0.9}Er_{0.1}Co₂ to form (Ho_{0.9}Er_{0.1})_{1-x}Gd_xCo₂ since the T_C temperature of GdCo₂ is about 400 K [8].

The purpose of this paper is, using heat capacity measurements, to present the magnetocaloric properties of the composite material based on (Ho_{0.9}Er_{0.1})_{1-x}Gd_xCo₂ compounds,.

Content from this work may be used under the terms of the Creative Commons Attribution 3.0 licence. Any further distribution of this work must maintain attribution to the author(s) and the title of the work, journal citation and DOI. Published under licence by IOP Publishing Ltd 1

IOP Publishing

2. Materials and experiment

Polycrystalline samples of $(Ho_{0.9}Er_{0.1})_{1-x}Gd_xCo_2$ compounds, with x = 0.05, 0.1 and 0.15 were prepared by arc-melting using a water-cooled copper crucible and a high purity argon atmosphere. The starting materials (Co 99.99% purity and rare-earth metals 99.9% purity) were taken in stoichiometric proportions. The alloys were melted repeatedly (four times) to ensure the adequate homogeneity of buttons. The mass losses after the melting were less than 1 wt %. The buttons obtained were wrapped in a tantalum foil, sealed in evacuated quartz ampoules, and annealed at 850°C for four weeks.

The heat capacity was measured in a temperature range of 2-300 K in zero, 1 and 2 T magnetic fields using a PPMS 14 Heat Capacity System (Quantum Design). X-ray powder diffraction analysis showed that the $(H_{0.9}Er_{0.1})_{1-x}Gd_xCo_2$ samples are single-phase and have the cubic C15 structure (space group Fd-3m) and the substitution of gadolinium for rare earth leads to the increase in the lattice parameter. At room temperature, we have obtained a = 0.7172 nm, for $(H_{0.9}Er_{0.1})_{0.95}Gd_{0.05}Co_2$, a = 0.7175 nm, for $(H_{0.9}Er_{0.1})_{0.95}Gd_{0.1}Co_2$ and a = 0.7178 nm, for $(H_{0.9}Er_{0.1})_{0.85}Gd_{0.15}Co_2$.

3. Results and discussion

The heat capacity measurements performed as a function of temperature in a given magnetic field provide the most complete characterization of any magnetic materials with respect to their magnetocaloric properties. It is well known that the heat capacity of metals can be considered as the sum of independent electronic, lattice (phonon), and magnetic contributions:

$$C_{p}(T) = C_{el}(T) + C_{ph}(T) + C_{mag}(T).$$
(1)

The electronic and phonon contributions to the heat capacity can be calculated by expression [9]:

$$C_{el+ph}(T) = \gamma T + 9NR \left(\frac{T}{\Theta_{D}}\right)^{3} \int_{0}^{\Theta_{D}/T} \frac{x^{4}e^{x}}{\left(e^{x}-1\right)^{2}} dx , \qquad (2)$$

where the first term represents the electronic heat capacity and the second term corresponds to the phonon Debye contribution; γ is the Sommerfeld coefficient; Θ_D is the Debye temperature; N = 3 is the number of atoms per formula unit; R is the universal gas constant; and $x = \hbar \omega / k_B T$. The $C_{el+ph}(T)$ dependence for each of the investigated samples was calculated by Eq. 2. The best fitting for the wide temperature range could be obtained by fixing the parameter $\gamma \sim 30 \text{ mJ/molK}^2$, while the Debye temperature fluctuated around 230 - 235 K, and is very close to that reported by Oliveira et al. [10] for pure DyCo₂ ($\Theta_D = 230 \text{ K}$).

For all $(Ho_{0.9}Er_{0.1})_{1-x}Gd_xCo_2$ compounds, the magnetic contribution $C_{mag}(T)$ was estimated by subtracting the sum of electronic and phonon contributions from the total heat capacity $C_P(T)$. As the examples, the temperature dependences of the heat capacity, $C_P(T)$, the sum of electronic and phonon, $C_{el+ph}(T)$, and magnetic contributions, $C_{mag}(T)$, measured for $(Ho_{0.9}Er_{0.1})_{0.95}Gd_{0.05}Co_2$ in zero magnetic field are shown in Figure 1. In the case of all multicomponent compounds under study, two specific heat peaks are observed in the curves. The first of them is attributed to a spin-reorientation transition and the other well-defined maximum corresponds to the magnetic ordering temperature, T_C . A possible mechanism for the relatively small smooth step-like anomaly in $C_P(T)$ plots at low temperatures might be a spin reorientation in the R 4*f*-electron subsystem, which is similar to that observed for HoCo₂ ($T_{SR} = 15$ K) and NdCo₂ ($T_{SR} = 43$ K) [11]. The inset in Fig. 1 shows temperature dependences of the heat capacity of (Ho_{0.9}Er_{0.1})_{0.95}Gd_{0.05}Co₂ measured, additionally, in magnetic fields. It is seen that the maximum of peaks in the $C_P(T)$ dependences measured in zero, 1 and 2 T magnetic fields shifts to the high-temperature range and gradually decreases with increasing magnetic field.

IOP Publishing



Figure 1. Temperature dependences of heat capacity $C_P(T)$ of $(Ho_{0.9}Er_{0.1})_{0.95}Gd_{0.05}Co_2$ measured in zero magnetic field; the calculated sum of electronic and phonon contribution, $C_{el+ph}(T)$, as well as estimated magnetic contribution, $C_{mag}(T)$ are shown. Inset shows $C_P(T)$ dependences measured in zero, 1 and 2 T magnetic fields.

Figures 2 (a-c) show the temperature dependences of C_{mag} for $(Ho_{0.9}Er_{0.1})_{1-x}Gd_xCo_2$ compounds in zero, 1 and 2 T magnetic fields. The observed maximum in the $C_{mag}(T)$ appears in the vicinity of T_C and then seems to be the result of the magnetic phase transitions. The introduction of Gd in $Ho_{0.9}Er_{0.1}Co_2$ leads to the increase in the magnetic ordering temperature. The T_C temperature increases from 89.2 K for $(Ho_{0.9}Er_{0.1})_{0.95}Gd_{0.05}Co_2$ to 127 K for $(Ho_{0.9}Er_{0.1})_{0.85}Gd_{0.15}Co_2$. For all compositions, the observed maximum decreases in height, broadens and moves toward the high-temperature range in applying the magnetic field (see Figs. 2b and 2c).



Figure 2. Temperature dependences of C_{mag} for $(Ho_{0.9}Er_{0.1})_{1-x}Gd_xCo_2$ compounds in zero (a), 1 T (b) and 2 T (c) magnetic fields.

The value of magnetic entropy, $S_{mag}(T)$, was calculated from the experimental heat capacity data using $C_{mag}(T)$ curves by standard formula :

$$S_{mag}(T) = \int_{0}^{T} \frac{C_{mag}(T)}{T} dT$$
(3)

IOP Publishing

and, from the calculated curves $S_{mag}(T, \mu_0 H = 0)$ and $S_{mag}(T, \mu_0 H \neq 0)$, we obtain the magnetic entropy change $\Delta S_{mag} = S_{mag}(T, \mu_0 H) - S_{mag}(T, 0)$. Temperature dependences of the magnetic entropy change, $\Delta S_{mag}(T)$, for investigated compounds, caused by the magnetic field changes $\mu_0 \Delta H = 1$ T and $\mu_0 \Delta H = 2$ T are shown in Fig. 3 and Fig. 4, respectively. At 1 T magnetic field change, the maximum magnetic entropy change reaches values : $\Delta S_{mag} = 5.1$ J/kgK, for $(Ho_{0.9}Er_{0.1})_{0.95}Gd_{0.05}Co_2$, 3.3 J/kgK for $(Ho_{0.9}Er_{0.1})_{0.9}Gd_{0.1}Co_2$, and 2.6 J/kgK for $(Ho_{0.9}Er_{0.1})_{0.85}Gd_{0.15}Co_2$. At 2 T magnetic field change, $\Delta S_{mag} = 9.1$ J/kgK, for $(Ho_{0.9}Er_{0.1})_{0.95}Gd_{0.05}Co_2$, 6.0 J/kgK for $(Ho_{0.9}Er_{0.1})_{0.9}Gd_{0.1}Co_2$, and 4.6 J/kgK for $(Ho_{0.9}Er_{0.1})_{0.85}Gd_{0.15}Co_2$. It should be noted that the maximum values of ΔS_{mag} obtained from the heat capacity data are similar to those reported based on magnetization measurements [7].

According to literature data, the Ericsson cycle is the appropriate one to be used in magnetic refrigerator [12]. This cycle employs a constant value for the ΔS_{mag} in the temperature range of refrigeration, which is necessary for the improvement of the regenerative processes [13].

Unfortunately, as can be seen from Figures 3 and 4, $\Delta S_{mag}(T)$ for single multicomponent compounds is not constant in the temperature range. To overcome this difficulty, the numerical simulations, according to the procedure proposed in Refs. [14-16] to construct a composite material formed by [(Ho_{0.9}Er_{0.1})_{0.95}Gd_{0.05}Co₂]y₁; [(Ho_{0.9}Er_{0.1})_{0.9}Gd_{0.1}Co₂]y₂ and [(Ho_{0.9}Er_{0.1})_{0.85}Gd_{0.15}Co₂]y₃ was done. The factors y_i represent the mass ratio of each multicomponent compounds under the condition y₁+y₂+y₃=1. The effective isothermal magnetic entropy change upon magnetic field variation of the composite material is given by [15];

$$\left|\Delta \mathbf{S}_{\text{comp}}\right| = \sum_{i=1}^{3} \mathbf{y}_{i} \left| (\Delta \mathbf{S}_{\text{mag}})_{i} \right|,\tag{4}$$

where $(\Delta S_{mag})_i$ represent the isothermal magnetic entropy change of the single materials $(Ho_{0.9}Er_{0.1})_{0.95}Gd_{0.05}Co_2$; $(Ho_{0.9}Er_{0.1})_{0.9}Gd_{0.1}Co_2$ and $(Ho_{0.9}Er_{0.1})_{0.85}Gd_{0.15}Co_2$, respectively.



Figure 3. Temperature dependences of the magnetic entropy change, $\Delta S_{mag}(T)$, for investigated compounds and composite (solid line) based on them, for magnetic field change 0 to 1 T. The dotted lines represent ΔS_{mag} vs. temperature for $(Ho_{0.9}Er_{0.1})_{1-x}Gd_xCo_2$ compounds after the numerical simulations using $y_1 = 0.219$, $y_2 = 0.328$ and $y_3 = 0.453$, for x = 0.05, 0.1 and 0.15, respectively.

The solids lines in Figures 3 and 4 show the calculated isothermal entropy change for these composite materials using $y_1 = 0.219$, $y_2 = 0.328$ and $y_3 = 0.453$ for a magnetic field variation from 0 to 1 T and $y_1 = 0.222$, $y_2 = 0.290$ and $y_3 = 0.487$ for a magnetic field variation from 0 to 2 T, respectively. From those figures, one can notice that the peaks corresponding to the magnetic transition temperature of the three multicomponent compounds and the magnitude of the peak values decreases and the shape of the peaks gets broadened. The maximum magnetic entropy change fluctuates around 1.4 J/kgK, at $\mu_0 \Delta H = 1$ T, and 2.7 J/kgK for $\mu_0 \Delta H = 2$ T in a temperature range from 90 to 127 K.



Figure 4. Temperature dependences of the magnetic entropy change, $\Delta S_{mag}(T)$, for investigated compounds and composite (solid line) based on them, for magnetic field change from 0 to 2 T. The dotted lines represent ΔS_{mag} vs. temperature for $(Ho_{0.9}Er_{0.1})_{1-x}Gd_xCo_2$ compounds after the numerical simulations using $y_1 = 0.222$, $y_2 = 0.290$ and $y_3 = 0.487$, for x = 0.05, 0.1 and 0.15, respectively.

4. Conclusions

On the basis of the experimental heat capacity measurements, the isothermal magnetic entropy changes for multicomponent $(Ho_{0.9}Er_{0.1})_{1-x}Gd_xCo_2$ compounds (x = 0.05, 0.1 and 0.15) were calculated. Also, it was determined theoretically the optimum molar fraction of these three compounds in order to form a composite to be considered as an optimal refrigerant material working in an Ericsson cycle in the temperature range 90-130 K. Moreover, the obtained molar concentrations of the composite showed the slight dependence on $\mu_0\Delta H$ in the magnetic field range under study.

Acknowledgments

The work was supported by the National Science Center, Poland through the OPUS Program under Grant No. 2019/33/B/ST5/01853.

References

- [1] Du N H, Kim Anh D T 2002 J. Magn. Magn. Mater. 873-875 242
- [2] Wang D H, Tang S L, Liu H D, Gao W L, Du Y W 2002 Intermetallics 10 819
- [3] Tishin A M, Spichkin Y I 2003 *The Magnetocaloric Effect and Its Applications* (IOP Publishing, Bristol and Philadelphia)
- [4] Liu X B, Altounian Z 2005 J. Magn. Magn. Mater. 292 83
- [5] Tereshina I S, Cwik J, Tereshina E A, Politova G A, Burkhanov G S, Chzhan V B, Ilyushin A S, Miller M, Zaleski A, Nenkov K, Schultz L 2014 *IEEE Trans. Magn.* **50** 2504604
- [6] Tereshina I, Chzhan V, Tereshina E, Khmelevskyi S, Burkhanov G, Ilyushin A, Paukov M, Havela L, Karpenkov A, Cwik J, Koshkid'ko Yu, Miller M, Nenkov K, Schultz L 2016 J. Appl. Phys. 120 01390
- [7] Cwik J, 2014 J. Solid State Chem. 209 13
- [8] Kirchmayr H R, Burzo E, 1990 in: Landolt Börnstein, New Series III/19d2 (H.P.J. Wijn, Berlin)
- [9] Gopal E S R, 1966 in: *The International Cryogenics Monograph Series*, Ed K Mendelssohn, K D Timmerhaus (Heywood Books, London)
- [10] de Oliveira N A, von Ranke P J 2003 J. Magn. Magn. Mater 264 55
- [11] Gratz E, Nowotny H 1982 J. Magn. Magn. Mater 29 127
- [12] Gómez J R, Garcia R F, Miguel Catoira A, Gómez M R 2013 Renewable and Sustainable Energy Reviews 17 74
- [13] Z. Yan and J. Chen, J. Appl. Phys. 72 (1992) 1.
- [14] Diguet G, Lin G, Chen J 2013 Int. J. Refrigeration 36 958

1758 (2021) 012009 doi:10.1088/1742-6596/1758/1/012009

- [15] Smaïli A, Chahine R, 1996 Adv. Cryog. Eng. 42 445
- [16] Hashinomoto T, Kuzuhara T, Sahashi M, Inomata K, Tomokiyo A, Yayama H 1987 J. Appl. Phys. 62 3873