

Dielectric Permittivity Calculation of Composites Based on Electrospun Barium Titanate Fibers

H. A. Ávila^{*}, M. M. Reboredo, R. Parra, M. S. Castro^{**}

Instituto de Investigaciones en Ciencia y Tecnología de Materiales (INTEMA)

CONICET – UNMdP

J. B. Justo 4302, B7608FDQ Mar del Plata, Argentina

^{*}*havila@fi.mdp.edu.ar*, ^{**}*mcastro@fi.mdp.edu.ar*

Abstract

On the basis of theoretical predictions and experimental results, an empirical method using upper bound equation of the rule of mixtures (ROM) is reported to predict the dielectric permittivity of barium titanate nanofibers. In addition, composites with low volume fraction of BaTiO₃ fiber layers embedded in epoxy resin were prepared and characterized. The relative permittivities of composites with perpendicular and parallel configurations, with respect to the electrodes, were calculated by means of the ROM model. The predicted permittivities matched precisely the obtained experimental values.

Keywords: *Polymer-matrix composites (PMCs) (A), Ceramic fiber (A), Electrical properties (B), Analytical modeling (C).*

Introduction

Composites have been extensively studied for functional and structural applications. By integrating the advantages of two phases, composite materials can offer enhanced performances with respect to those of the individual components [1]. In the last decades, there has been a great interest in polymer matrix composites that combine polymers with ferroelectric ceramics of high dielectric permittivity. Generally, the performances of this kind of composite materials depend both on the properties of the matrix and on the filler characteristics, such as its nature, geometry, orientation and volume ratio [2] besides the interphase between them. The size and concentration of ceramic particles have a significant effect on the dielectric properties of composites. Indeed, when the amount of BaTiO₃ particles reaches 75 vol%, the dielectric permittivity of nanocomposites may be increased [3]. However, defects such as voids and porosity tend to decrease the dielectric constant. The porosity of composites containing BaTiO₃ microparticles decreases with the ceramic ratio up to 30%, then increases abruptly at 40%. The high porosity of the 40% BaTiO₃/epoxy bulk composites leads to higher losses [4,5].

Several dielectric composites have been proposed by using BaTiO₃ particles as filler and a thermoplastic [5-11] or thermosetting [12,18] polymer as matrix. Composites can be prepared in many different ways, leading to a large variety of morphologies, which strongly influence the macroscopic behavior [19]. Recently, new polymer composites using electrospun BaTiO₃ fibers have been reported with promising results [20-24]. Ceramic fibers are attractive because of the increased anisotropy, improved flexibility and strength over monolithic ceramics. For instance, the potential of nanostructured composites that include BaTiO₃ fibers in their compositions has been demonstrated by applications in miniaturized electronic devices and sensors [21,23,25]. Concerning different device configuration

possibilities, a significant increase in the composite dielectric permittivity was obtained depending on the orientation of fibers inside the matrix [24].

The prediction and measurement of the dielectric properties of composites have been a challenging issue [19,26-28]. Several quantitative rules of mixture models have been proposed to calculate the dielectric permittivity of heterogeneous two-component systems considering the dielectric properties of each component [29]. However, whereas different models have been developed, usually little experimental evidence is provided to support the derived equations. Therefore, it is difficult to choose the correct model to calculate the effective dielectric permittivity of polymer/ceramic composites.

Several models have been proposed and used to predict the effects of second phases on the dielectric properties of composites such as the Maxwell-Garnett's equation [30], the Lichtenecker's equation [31], the Bruggeman's equation [32], the Jayasundere-Smith's equation [33], the Effective Medium Theory (EMT) [31] and the Poon-Shin's equation [34]. All these models take into account a continuous medium filled with spherical particles. However, only a few theoretical models are useful to predict the dielectric permittivity of polymer/ceramic-fiber composites [19].

This work aims to the estimation of the dielectric permittivity of both BaTiO₃ fibers and Epoxy/BaTiO₃-fibers composites. The dielectric permittivity of BaTiO₃ fibers has been calculated considering a fiber air-composite. Then, composite materials made of BaTiO₃ fibers embedded in an epoxy resin matrix were studied. The experimental dielectric values of Epoxy/BaTiO₃-fibers composites were matched with the ones calculated from the rule of mixtures (ROM) model.

Experimental procedure

1. Fiber synthesis

For the synthesis of barium titanate fibers, a sol of titanium isopropoxide (Aldrich, 97%), acetylacetone (Aldrich, 99.9%) and absolute ethanol were mixed, under vigorous stirring inside a glove box, with a solution of barium acetate (Mallinckrodt) in acetic acid (Merk). A solution of poly(vinyl pyrrolidone) (PVP, 1.3MDa, Aldrich) in absolute ethanol was added to the resulting sol precursor, which was loaded into a polypropylene syringe for electrospinning. A DC voltage of 13 kV was applied to the metallic needle by means of a Gamma High Voltage Research (0-30 kV) unit, and a syringe pump (ACTIVA Prestige-equipment) fed the sol at a constant rate of 0.5 ml/h. Random fiber mats were collected 10 cm below the needle tip on aluminum foil. Crystalline BaTiO₃ fibers were obtained after a heat treatment in air at 800 °C for 1 h.

2. Structural and Microstructural Characterization of Fibers

X-ray diffraction (XRD) was performed on fiber samples using a PANalytical, X'pert Pro equipment running with CuK α radiation and step size of 0.02°/min, from 10° to 80° 2 θ . Raman spectra were collected with a Renishaw inVia Raman microscope using a 514 nm Ar-ion laser (50 mW nominal power). The microstructure and morphology of fibers were also characterized by Field Emission – Scanning Electron Microscopy (FE-SEM, Zeiss Supra 35).

3. Preparation of Epoxy/BaTiO₃ Composites

A polymer-ceramic composite was prepared from bisphenol A-type epoxy resin (D.E.R. 331) and D.E.H. 24 (12.5% w/w) curing agent, both from Dow Chemical, with Epodil 747 (10% w/w) as reactive diluent and tetrahydrofuran (THF, Dorwil Chemical) (10vol%) as

solvent. Heat-treated fiber mats, in which fibers are not oriented but distributed at random, were embedded in epoxy resin inside silicone molds. The resulting composite samples were cured in an oven at 100 °C for 2 h for solvent removing and resin polymerization. The obtained composite was cut into specimens with dimensions in thickness, width and length, of 1, 1 and 10 mm, respectively.

4. Dielectric Characterization of Fibers and Composites

4.1. Fibers

In order to determine the effective permittivity of barium titanate fibers, the heat-treated fiber mats were considered as a biphasic composite material made of BaTiO_3 and air. For this purpose, 2 and 4 layers of fiber mats were slightly pressed in order to obtain two ceramic/air composite samples with different volume fraction of ceramic filler (V_f). In this way the composite material can be considered as a capacitor containing two phases that can be approximated to a structure of parallel capacitors with respect to electrodes (Figure 1)

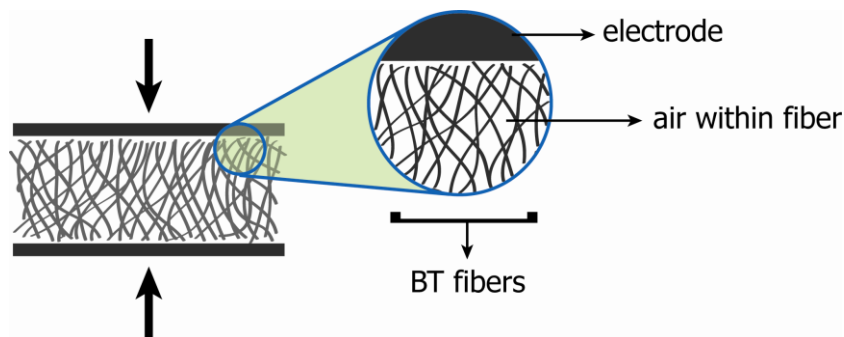


Figure 1. Schematic representation of BaTiO_3 fiber layers/air composite.

4.2. Composites

For the dielectric characterization of composites, silver electrodes were painted on different faces of the composite samples (dimensions 1x1x10 mm), in order to obtain two

different configurations of electrodes (parallel or perpendicular) regarding to the fiber mat layers, as shown in Figure 2, (as already described in Ávila *et al.* [24]).

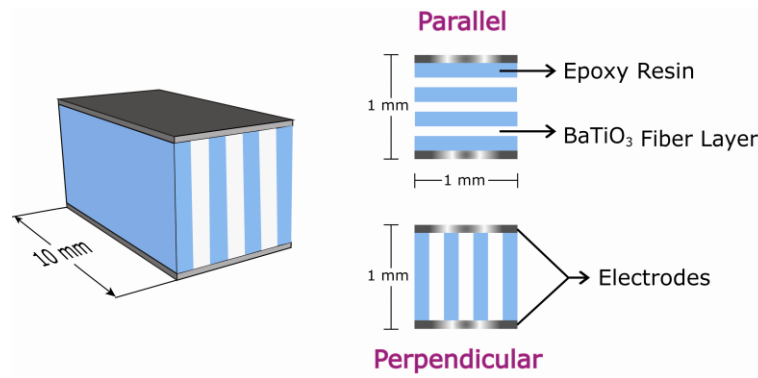


Figure 2. Epoxy/BaTiO₃ fibers composites in parallel and perpendicular configurations.

In order to predict the dielectric properties of the Epoxy/BaTiO₃ composites, equation (1) was used as follows,

$$\epsilon = \epsilon_0 \left(\epsilon_f + \frac{n \epsilon_f \epsilon_m (\epsilon_m - \epsilon_f)}{\epsilon_m + n \epsilon_f} \right) \quad (1)$$

where, n equals -1 for a *parallel configuration*, and +1 for a *perpendicular configuration* [35].

Dielectric measurements were carried out at room temperature in the 0.1 – 10⁷ Hz frequency range using HIOKI 3522-50 LCR HiTester (10 mHz – 100 kHz) and HIOKI 9700-10 Head Amp Unit (100kHz – 120 MHz) instruments.

Results and Discussion

Figure 3(a) shows the X-ray diffraction pattern of BaTiO₃ fibers heat-treated at 800 °C along with the peaks of the BaTiO₃ tetragonal phase from the reference file JCPDS 74-1956. The diffraction pattern corresponds to barium titanate without secondary phases in agreement

with the reference file. The Raman spectrum in Figure 3(b) shows typical bands for the tetragonal phase. These results demonstrate that BaTiO₃ fibers consist of a mixture of cubic and tetragonal phases. The effect of thermal treatment on the crystalline structure of BaTiO₃ fibers was discussed in a previous work [24].

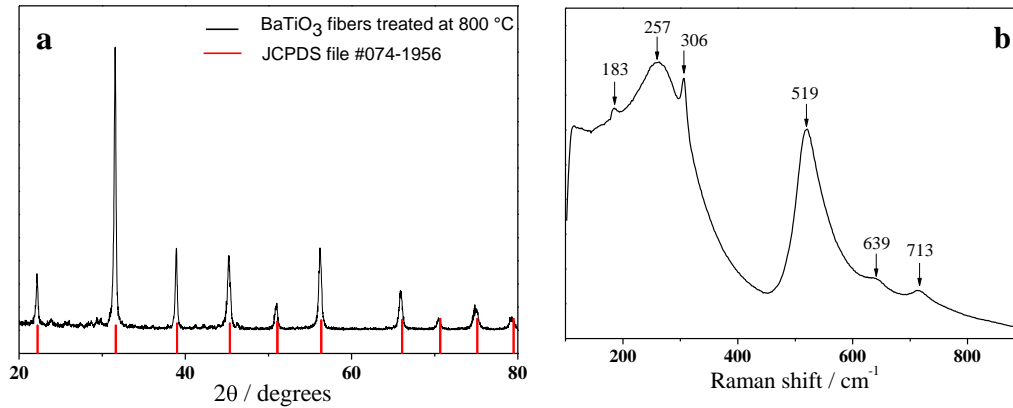


Figure 3. (a) XRD pattern and (b) Raman spectrum of BaTiO₃ heat-treated fibers.

Figure 4 shows the FE-SEM images of heat-treated fibers. Figure 4(a) shows BaTiO₃ fibers with average diameter of 250 nm. The magnification in Figure 4(b) shows uniform sintered grains and mesopores generated during polymer burning. It is important to notice the continuity of fibers, which is the most important feature for determining dielectric properties of composites.

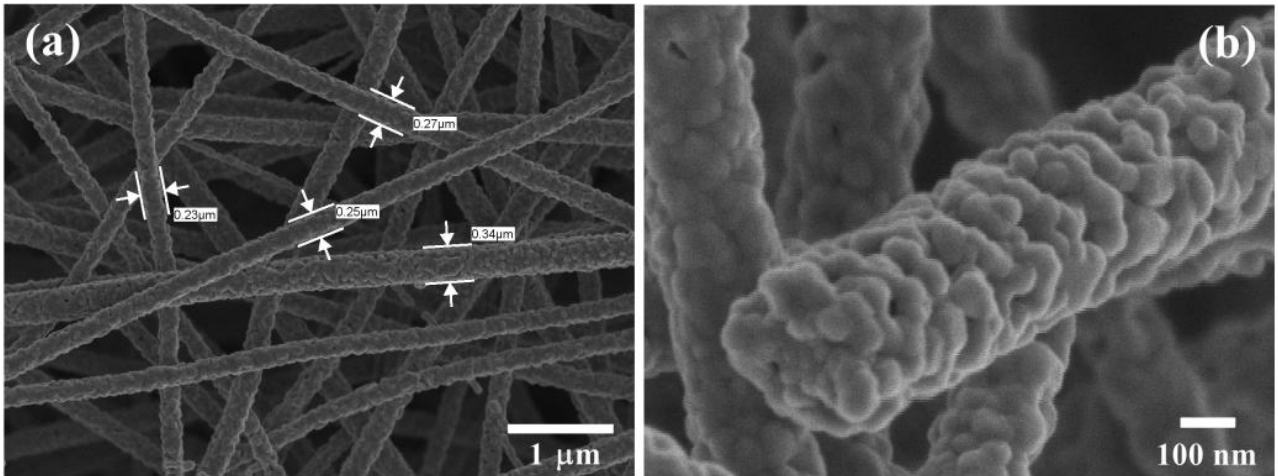


Figure 4. FE-SEM images of heat-treated electrospun BaTiO₃ fibers.

Figure 5 shows a cross-section image of the Epoxy/BaTiO₃-fiber composite fabricated with heat treated fibers. The micrograph shows dark and bright regions that belong to the epoxy resin matrix and BT fiber layers, respectively. It is clear that the epoxy resin fully covers the fibers mats and that the presence of voids or porosity in Epoxy/BT-fibers interphase is negligible.

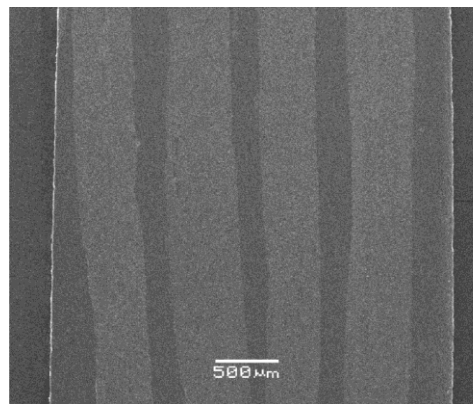


Figure 5. Cross-sectional SEM image of an Epoxy/BT-fiber composite showing fiber layers (vertical bright regions) embedded in epoxy resin (dark regions).

The permittivity as a function of frequency for BaTiO₃ fiber layers/air composites is shown in Figure 6. Taking into account the low volume fraction of fibers and the important changes measured in the permittivity values, it can be assumed that the permittivity can be approximated by a linear regression using three points with V_f : 0, 0.0234 and 0.157, respectively. Then, the measured permittivity of the composite (ϵ_c) can be related with the BaTiO₃ (filler) permittivity (ϵ_f) as follows,

$$\epsilon_c = \epsilon_m + \frac{\epsilon_f - \epsilon_m}{V_f} V_f \quad (2)$$

where, ϵ_m is the permittivity of the matrix (air relative permittivity = 1.00058). Then, the permittivity of the ceramic fibers can be determined from the slope of ϵ_c vs. V_f plots. The permittivity values of BaTiO₃ fibers/air composite at specific frequency are shown in Table 1, as well as the respective linear correlation coefficients.

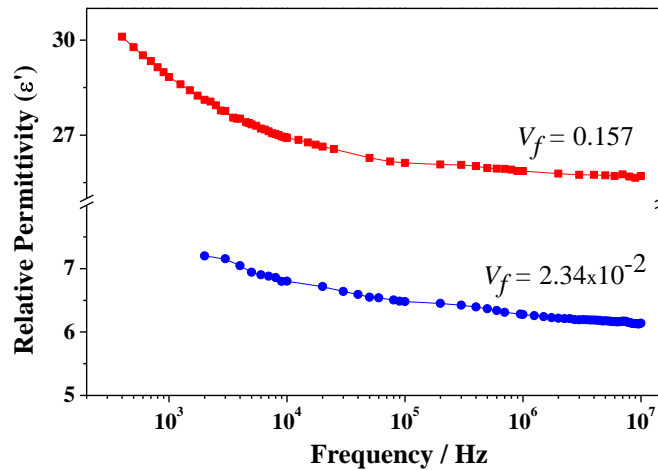


Figure 6. Relative permittivity of BaTiO₃ fibers/air composites of two different volume fractions.

Table 1. Relative permittivity values (ϵ_c) of BaTiO₃/air composites with different fiber volume fractions (V_f), calculated permittivity of fibers (ϵ_f) at different frequencies using $\epsilon_{air} = 1.00058$, and linear correlation coefficient (r^2).

Frequency	ϵ_c ($V_f = 2.34 \times 10^{-2}$)	ϵ_c ($V_f = 0.157$)	ϵ_f	Correlation coefficient r^2
10 kHz	6.8	26.9	161.0	0.9971
100 kHz	6.5	26.1	156.4	0.9974
1 MHz	6.3	25.9	155.5	0.9979
10 MHz	6.1	25.7	154.7	0.9983

The permittivity values of BaTiO₃ fibers are smaller than those of BaTiO₃ powders reported by various authors as Fang *et al.* ($\epsilon_r = 6100$ at 1 kHz) [36], Dutta *et al.* ($\epsilon_r = 500$ -6000 at 1 kHz) [37], Tsurumi *et al.* ($\epsilon_r = 3000$) [38], Ying *et al.* ($\epsilon_r = 800$ at 1 kHz) [39], George *et al.* ($\epsilon_r = 1223$ at 1 MHz) [40]. However, physical-morphological factors as diameter, grain size, and tetragonality could significantly affect the permittivity values of BaTiO₃ fibers. Then, a direct model to predict relative permittivities, which uses few parameters and fits the experimental data, is put forward hereafter. In a recent work Wei *et al.* calculated the dielectric permittivity of BaTiO₃ nanofibers using the Maxwell-Granett's model by considering a porous bulk specimen as a BaTiO₃ fibers/air composite [41]. The high dielectric permittivity ($\epsilon_r \sim 820$) reported by the authors may be associated with the experimental procedure, by means of which fibers were pressed into pellets at 2MPa and heat treated at 1050°C. It is worth noting that the applied pressure and the high temperature treatment can destroy the original fiber morphology, breaking fibers into small pieces and favoring grain growth and sintering with the consequent increase in dielectric permittivity.

The model used in this work is simpler than that proposed by Wei *et al.* [41], and the experimental procedure followed does not modify or destroy the original fiber morphology.

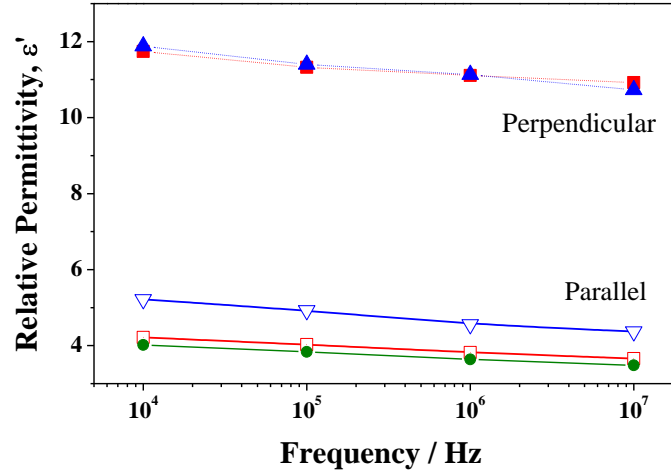


Figure 7. Permittivity of Epoxy/BaTiO₃ fibers composites for 4.92 vol% showing the experimental (blue) and calculated (red) data. The curve of the epoxy resin (green) is also shown.

In a previous work [24], a new kind of polymer/ceramic fibers composite using BaTiO₃ fiber layers in parallel and perpendicular configurations with regard to the electrodes was introduced. One of the problems in predicting the effective dielectric permittivity of polymer/ceramic composites is the fact that the dielectric permittivity of the ceramic filler is not available. For biphasic composites, the highest permittivities may be calculated with equation (1) using $n = 1$.

Figure 7 shows both experimental and theoretical permittivities of Epoxy/BaTiO₃ fibers composites calculated using equation 2 with perpendicular ($n = 1$) and parallel ($n = -1$) electrode configurations. The variation observed with frequency is due to the normal decay of the epoxy resin permittivity, as shown by the green curve obtained from a pure epoxy specimen.

Some restrictions must be considered before applying the ROM model. These are: (a) the assumption that fiber layers are completely embedded in the matrix so there is perfect bonding between fibers and matrix; (b) the absence of relaxation processes in the analyzed frequency range; (c) the proposed model is only applicable for low filler fractions; (d) the absence of voids and the electrical conductivity values of filler and matrix are restricted by percolation theory.

Despite the restrictions of the ROM, the calculated values for both configurations are concomitant with experimental data. However, slight differences between experimental and theoretical permittivities, especially in the parallel arrangement, can be brought about by extrinsic defects such as air bubbles, discontinuous fibers, random orientation of fibers within the layers and residual solvent in composites.

Conclusions

A simple method has been used to determine the relative permittivity of ceramic (BaTiO_3) fibers. Then, the permittivity values obtained by means of the rule of mixtures were employed to calculate the dielectric permittivity of Epoxy/ BaTiO_3 fibers composites with two different configurations. Estimations matched closely the experimental results for this kind of composites. The dielectric constant of a composite with alternating fiber-layers can be described by the rule of mixtures model equations.

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