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Pd decorated Co–Ni nanowires as a highly efficient catalyst for direct ethanol fuel cells

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HIGHLIGHTS

- Co-Ni with Pd NPs exhibit exceptional catalytic activity towards the EOR.
- Co-Ni in the ratio of 3:7 showed best catalytic properties.
- The balance between oxidized, disordered and crystalline phases of Co-Ni is the key.
- Synergistic effect of 1D morphology and structure of the Co–Ni with Pd NPs.
- The presence of Pd, PdO₂, Co-Ni and their oxides enhances catalytic activity.

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GRAPHICAL ABSTRACT



ABSTRACT

The storage and conversion of energy necessitates the use of appropriate electrochemical systems and chemical reaction catalysts. This work presents newly developed catalysts for electrooxidation of ethanol in an alkaline medium. Nanocatalysts composed of Co–Ni nanowires (Co–Ni NWs) decorated with Pd nanoparticles (Pd NPs) were made at varying metal ratios and their chemical composition and structure was investigated in detail. The synthesis involved a wet chemical reduction assisted by a magnetic field, which led to the generation of NWs, followed by the deposition of spherical Pd NPs on their surface. The

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0360-3199/© 2021 The Author(s). Published by Elsevier Ltd on behalf of Hydrogen Energy Publications LLC. This is an open access article under the CC BY-NC-ND license (http://creativecommons.org/licenses/by-nc-nd/4.0/). Keywords: Co—Ni nanowires Pd nanoparticles Ethanol oxidation reaction Electrocatalyst Direct ethanol fuel cells Metallic oxides best catalytic activity was obtained for the catalyst made of Co_3 —Ni₇ decorated with Pd NPs, which exhibited EOR of 8003 mA/mg_{Pd} for only 0.86 wt% of Pd loading. The results can be explained by the synergistic effect between the morphology of the bimetallic support and the favorable interaction of oxophilic Co, Ni with catalytic Pd.

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Introduction

The electricity demand constantly increases along with the technological development of the countries. At the same time, progressive climate change in our environment urges the society to replace energy obtained from conventional sources (fossil fuels) with those of renewable origins. Such sustainable energy may come from the sun, wind, hydropower, tidal forces, etc. [1–4]. The generated green electricity can then be directly transmitted to customers to fulfill the current needs. However, due to the intermittent nature of these energy sources and the fact that the electricity generation time does not fully coincide with the demand, there is a need to preserve it for later use. Alternatively, the energy can be extracted from chemical transformations of replenishable chemical compounds to generate electricity inside of fuel cells. Therefore, the process of energy storage and conversion requires appropriate electrochemical systems and catalysts to carry it out effectively. Due to the tremendous social and economic importance of the energy deficit problem, scientists focus on the development of new catalysts which can promote this process [5].

Direct alcohol fuel cells (DAFCs) are certainly among the most efficient fuel cells operating at ambient temperature [6-8]. The principle of their work is based on the direct conversion of chemical energy accumulated in alcohols into electric energy, which may constitute an alternative power source for vehicles, portable electronics, or other devices [9-12]. Furthermore, it is easy to store, handle, and transport liquid fuels (methanol, ethanol, 1-propanol, ethylene glycol), which can feed for fuel cells. These chemicals exhibit high availability and low costs, as compared to gaseous fuels [13-17]. Some of them can even be generated by sustainable routes. Ethanol stands out among these substrates, as it is characterized by high energy density (6.8 kWh/L), low toxicity and high bioavailability. For instance, it can be obtained by fermentation of biomass that contains sugar [11,13,18–20]. Therefore, it is not surprising that the scientific community takes such a great interest in the topic of direct ethanol fuel cells (DEFCs) [12,14,21-23]. Commonly used ethanol oxidation catalysts operating in DEFCs include Pt-based systems. However, despite its advantages such as high catalytic activity at low temperature, platinum suffers severe limitations [14,24,25]. The most important of them include high cost, low availability, and, above all, time-decreasing kinetics of ethanol oxidation reaction caused by poisoning of the catalyst with

intermediate products (CHx and CO) [12,14,26-30]. Simultaneously, the use of ethanol as a fuel is associated with an additional difficulty. It is energetically challenging to break down the C–C bond to enable its conversion into CO₂. Consequently, the reaction efficiency does not match expectations even when Pt is engaged [31,32]. Therefore, it is essential to devise catalytic systems for ethanol oxidation to CO₂, which could address this problem [33].

The complete oxidation of an ethanol molecule to CO₂ becomes possible upon optimizing several factors such as substrate concentration, reaction temperature, type of environment (acid or alkaline), and the morphology/chemical composition of the catalyst [26,34-36]. Earlier studies conducted on the type of using medium have shown that the kinetics of both ethanol oxidation (EOR) and oxygen reduction (ORR) reactions occur faster and more efficiently in alkaline media [37-40]. Using alkaline conditions makes it possible to reduce the amount of used catalysts or enables one to employ non-Pt catalysts (nickel, silver) in the reactions [24,41], which reduces the overall cost of the process. To further alleviate the cost burden, numerous groups of researchers [29,42,43] have focused on finding a worthy replacement for Pt in alkaline direct ethanol fuel cells (ADEFCs). Higher availability which leads to lower costs, higher stability and poisoning tolerance, and above all, higher electrocatalytic activity towards EOR in alkaline medium, is provided by palladium [12,29,44–46].

Furthermore, alloying of palladium with other metals (Cu, Ag, Au, Sn, Co or Ni) [14,45,47] or metal oxides (CeO₂, NiO, Co₃O₄) [13] can improve its stability and catalytic activity in alkaline environments due to a bifunctional mechanism. In addition, it can also facilitate the formation of OH_{ads} species on its surface, which minimizes the effect of catalyst poisoning by CO [13,14,45,47,48]. With this in mind, the ongoing work focuses on the preparation of efficient and stable catalysts for the oxidation of alcohols that should be less dependent on the noble metals. For instance, a proper selection of the catalyst substrate may be beneficial. Its composition and morphology were found to significantly impact on the catalytic performance [49-52]. Nowadays, the commonly used supports for metallic catalysts (Pt, Pd, Ru) include nanostructured carbon materials such as graphene, graphene oxide or carbon nanotubes. Their advantages include high specific surface area and electrical conductivity [26,53-57]. However, they cannot catalyze ethanol oxidation themselves. Their second major limitation is related to

corrosion and, consequently, to the detachment of catalyst particles from the support. Therefore, they have been looking for alternatives in the form of other supports. This work also seeks to make a contribution in this field.

It was recently found that metallic supports are capable of changing this situation. For instance, a system consisting of a Ni sponge decorated with Pd exhibited high stability and superior catalytic activity [58]. Furthermore, Pd-based catalyst systems with different morphology (PdNi nanosphere, PdNi film and PdCu aerogel) also demonstrated appreciable performance in this area [59,60]. Recently, nanowires (NWs) have attracted particular interest among researchers for this application. In the literature, there are many examples of nanocomposites containing metallic NWs, which were then evaluated for the electro-oxidation performance of methanol or ethanol such as Pt NPs/NbC NWs [61], Pd NPs/rGO-C@TiC NWs [62], Pd NWs/C [63], and Pt-Mo-Ni NWs [64]. Unfortunately, most of the presented supports are expensive and complicated in production. Therefore, this challenge asks for more suitable NW materials to be used substrates for the catalysts. Ideally, NWs should also actively participate in the catalytic reaction to increase its efficiency.

It seems that non-precious metals can play such a role. For example, nickel nanowires (Ni NWs) have already been successfully tested in urea oxidation reactions [65] and others [66]. Moreover, it was observed that the deposition of selected co-catalysts in the form of noble metal nanoparticles on the surface of Ni NWs significantly increases their catalytic activity. For example, it was shown that Ni NWs decorated with 10 wt % Pd nanoparticles have notable catalytic performance (1479.79 mA/mg_{Pd} EOR) [67]. In another study [68], an efficient electrocatalyst constructed from Ni NWs with flower-like Pd NPs operating in alkaline medium showed excellent activity (765 mA/mg_{Pd} EOR) also based on deposited on them. Cobalt is another example of a non-precious metal used with Pd in the nanostructure form (Co-Pd NWs), for which the obtained catalyst exhibited several times better stability and catalytic activity compared to commercial Pd/C [69]. In light of the foregoing, inexpensive and stable non-precious metals Co and Ni are highly promising materials to support precious metal catalysts. The use of metallic support in the form of nanowires (1D) can promote faster diffusion through liquid alcohol, reduce surface energy, and improve electron transport [67,68]. Importantly, Ni and Co have oxophilic nature, which increases the tolerance to the poisoning of the noble metal by CO, and thus increases its catalytic efficiency and longevity [67,70]. Therefore, an interesting issue seems to be the synthesis of bimetallic NW supports, containing both Ni and Co, to further increase the catalytic properties of the catalyst, such as Pd, deposited onto them.

The study presents two series of nanocomposites made of Co–Ni bimetallic nanowires (CoNi NWs) at varying Co:Ni ratios (5:5, 3:7, 1:9, 7:3, 9:1) with Pd NPs deposited on their surface at two precursor concentrations of 0.5 mM and 0.75 mM. The material synthesized by our group is a new generation catalyst. When developing the catalyst, it has taken into account its crystallographic structure, chemical composition and shape. These three parameters appear to be key in the context of designing next-generation catalysts. In addition, the new approach of replacing the carbon support with bimetal nanowires has improved the catalytic performance by actively participating the support in the catalysis reaction and avoiding its corrosion. The one of the goal was to develop an efficient EOR catalyst while minimizing the loading of the Pd nanoparticles. Furthermore, the metal ratios of the Co:Ni support were varied to thoroughly analyze how this affects the catalytic performance of the system. The synthesis of CoNi NWs was based on wet chemical reduction assisted with the magnetic field, on which spherical Pd NPs were then deposited. The relationship between the physicochemical properties of the obtained materials and their electrocatalytic activity in alkaline medium was studied in detail. The choice of Co and Ni as support materials was related to their oxophilic nature, which should enhance the catalytic activity of the Pd catalyst [70,71]. To the best of the authors' knowledge, it is the first demonstration of the use of bimetallic CoNi NWs as support in EOR reactions. The obtained results highlight that the 1D morphology of the support and bifunctional mechanism between CoNi and Pd improve the catalytic activity.

In this manuscript, nanocatalysts made of Co-Ni NWs with Pd NPs deposited on their surface were synthesized. Subsequently, these materials were characterized by XRD, STEM, HRTEM and XPS techniques. Finally, the catalytic properties of the obtained materials were evaluated by three-electrode electrochemical system.

Experimental

Materials

Nickel (II) chloride hexahydrate (Acros Organics, Geel, Belgium, 97%), cobalt (II) chloride hexahydrate (Chempur, Piekary Slaskie, Poland, pure p.a.), chloroplatinic acid hexahydrate (Sigma Aldrich, ACS reagent, >37.5% Pt basis), hydrazine monohydrate (Alfa Aesar, 98%), ethylene glycol (EG; Chempur, Piekary Slaskie, Poland, pure p.a.), sodium hydroxide (Chempur, Piekary Slaskie, Poland, pure p.a.), sodium hydroxide (Chempur, Piekary Slaskie, Poland, pure p.a.), polyvinylpyrrolidone (PVP; Alfa Aesar M.W. 1 300 000, pure p.a.) were all procured from the above-mentioned vendors and used without any further purification.

Synthesis of cobalt-nickel nanowires (CoNi NWs)

The modified synthesis procedure was based on the work of Zhang et al. [72]. 70 mL of 0.1 M NaOH solution in ethylene glycol (EG) was mixed with 5 mL of 64% hydrazine hydrate aqueous solution in a beaker. Then, the mixture was heated up to 90 °C, and 22 mL of precursor solution (various volume ratios of $CoCl_2$ and $NiCl_2$ salts - 1:9, 3:7, 5:5, 7:3, 9:1 dissolved in EG, and 2 mL of 0.01 M H_2PtCl_6 in water) was slowly dripped over 5 min. The formation of NWs was conducted for 10 min in the presence of a magnetic field established by a neodymium magnet. The magnet was enclosed by a second beaker positioned in the centre of the first beaker, where the reaction occurred (Fig. 1). After the reaction, the black magnetically active powder was separated and washed several times with distilled water and acetone to remove the by-products. Once the material was dry, it was stored in a desiccator. The samples were labelled according to the ratio of the metal precursor



Fig. 1 - Laboratory workstation used for the synthesis of Co-Ni NWs.

used, i.e., a ratio of 1:9 corresponds to a Co_1 -Ni₉ sample, and so on.

Synthesis of CoNi NWs_Pd NPs nanocomposites

100 mL of K_2PdCl_4 water solution in specified concentration was poured into a three-necked round-bottom flask. 100 mg of PVP and 120 mg CoNi NWs were added to the solution, and the flask was purged with inert gas. Then the flask was inserted into the ultrasonic water bath (Bandelin Sonorex RK 102H, 320 W, 35 kHz). The mixture was heated up to 60 °C and sonicated for 1 h under an inert atmosphere. After the reaction, Pd/CoNi nanocomposites were separated and washed with copious amounts of distilled water and acetone. Once the material was dry, it was stored in a desiccator.

Materials characterization

The structure and morphology of synthesized nanocomposites were determined based on transmission electron microscope (TEM) and scanning - transmission electron microscope (STEM) images, electron diffraction patterns (ED) and Energy-dispersive X-ray spectra (EDS), which were collected using Transmission Electron Microscope S/TEM TITAN 80-300. The phase composition and purity of the materials were determined by using X-ray diffractometer Rigaku Mini-Flex 600 with a copper tube Cu Ka (k = 0.15406 nm). The data were analyzed by the Rietveld refinement using the Rigaku PDXL software suite. The chemical composition and electronic structure of the samples were examined by X-ray photoelectron spectroscopy (XPS) using a monochromatic Xray source (Al K_{α} , hv = 1486.6 eV; PHI5700 Physical Electronics spectrometer). The measurements were performed directly from the oxidized surface of obtained samples and after

etching by Ar⁺ ion beam with energy 2 kV for 20 min. The survey spectra and core lines of Ni2p_{3/2}, Co2p_{1/2}, Ni3s, Co3s, Pd3d, O1s, C1s, and valence band were recorded. The Co2p_{1/2} line was chosen for analysis because the Co2p_{3/2} line was overlapped with the NiLMM Auger transition. Atomic concentration calculations and fitting process were performed using MULTIPAK software from Physical Electronics (ver. 9.9.0.8). Metal concentrations in the samples were determined by ICP/OES (PerkinElmer, Optima 2100 DV). The samples were diluted in order to reach the appropriate concentration range. The calibrations were made with standards purchased from Sigma-Aldrich company.

Electrochemical measurements

All electrochemical measurements were conducted on a BIO-LOGIC SP-200 potentiostat. A conventional three-electrode electrochemical system was used with a glassy carbon electrode as the working electrode, a platinum wire as the counter electrode, and a silver chloride electrode (Ag/AgCl) as the reference electrode, respectively. Uniform catalyst inks were prepared by ultrasonically mixing 2 mg of catalysts powder (in the form of CoNi NWs with different Co:Ni mass ratios decorated with Pd NPs) with the solution containing isopropyl alcohol and 5 wt % Nafion solution for 30 s. Next, 10 μl of asprepared ink suspension was deposited onto a polished glassy carbon electrode (GCE) to obtain the working electrode with a homogenous thin catalyst layer after the solvent dried naturally in the air. All the electrochemical experiments were conducted at ambient temperature. Prior to electrochemical measurements, the electrolytes containing fresh-prepared 1 M NaOH with 0.5 M or without ethanol, were deoxygenated by bubbling with Ar for 30 min. Then, CV measurements in ethanol-containing solution were carried out to determine

Table 1 — Deposition of Pd NPs on CoNi NWs by varying concentrations of Pd ions during reactions.					
Sample	NWs composition	C(K ₂ PdCl ₄) [mM]	Pd content [% wt]		
1	Co ₁ -Ni ₉	0.5	0.88		
2	Co ₃ —Ni ₇	0.5	0.81		
3	Co ₅ —Ni ₅	0.5	0.83		
4	Co ₇ —Ni ₃	0.5	1.54		
5	Co ₉ –Ni ₁	0.5	0.85		
6	Co ₁ –Ni ₉	0.75	1.92		
7	Co ₃ —Ni ₇	0.75	0.86		
8	Co ₅ —Ni ₅	0.75	3.35		
9	Co ₇ -Ni ₃	0.75	2.96		
10	Co ₉ –Ni ₁	0.75	1.41		

these catalysts' ethanol oxidation reaction (EOR) activity. The electrochemical data were normalized by the Pd mass of each catalyst (in mg) obtained by inductively coupled plasma (ICP) measurements.

Results and discussions

Structure and morphology analysis

The synthesis of Pd NPs/CoNi NWs revealed that even if the same concentration of Pd precursor is used for the deposition, the resulting concentration of Pd is different depending on the employed conditions (Table 1). This shows that the Pd loading on the surface of CoNi NWs is structure- or morphologydependent. To determine the role of these factors on the possibility of deposition of Pd NPs, the structural analysis was performed using transmission electron microscopy, X-ray diffraction method, and X-ray photoelectron spectroscopy. First of all, XRD patterns were measured and compared for two series of CoNi NWs with different Co:Ni ratios and Pd precursor concentrations (Fig. 2a and b). The obtained XRD patterns confirm that the structure of CoNi NWs is strongly Co:Ni ratio-dependent. Additionally, the Rietveld refinement method was used to determine the absence of the other phases and the presence and cell parameters of the identified ones. The analysis results were presented in Table 2 together with the fitting parameters: weighted profile residual (R_{wp}) , profile residual (R_p), and goodness of fit (χ^2). As one can see, the fitting parameters confirm the presence of two phases in all samples, generally, Ni(Co) and Co_{0.75}Ni₂₅ ones. When the concentration of Ni precursor is much higher than Co²⁺ precursor 9:1 and 7:3 (Co1-Ni9 - orange curve and Co3-Ni7 - green curve), the dominant phase is the cubic Ni(Co) (space group: Fm-3m). With increasing content of Co²⁺ precursor, the crystallization of this phase becomes difficult, and thus the highly disordered hexagonal Co_{0.75}Ni_{0.25} phase (space group: P63/ mmc) crystallizes (Co₉-Ni₁ - black curve, Co₇-Ni₃ - red curve). Interestingly, when the concentration of the palladium precursor is equal to 0.75 mM, it is possible to identify the characteristic for crystalline CoPd (space group: Fm-3m) (111) peak. However, only for the sample with the highest Ni concentration (Fig. 2b - Co1-Ni9 - orange curve). Probably, it is related to the growth of Pd NPs on the surface of Ni(Co) in well-ordered





crystallites (both phases crystallize in similar unit cells). In contrast, the same situation cannot occur when Pd growth occurs on the NWs composed predominantly from hexagonal $Co_{0.75}Ni_{0.25}$ phase.

To determine the co-existence of both identified phases and their role in the formation of Pd NPs, TEM and EDX analyses were also performed. Firstly, in Fig. 3a-f, the morphology of three samples (NWs with deposited Pd NPs) with different Co:Ni ratios (9:1, 5:5, and 3:7) was compared. It is evident that all analyzed samples form NWs, however, their surface morphology is different. The needle-like surface shape can be observed for samples containing high nickel concentration (composed of almost pure Ni(Co) cubic phase). On the other hand, for samples with high Co concentration, the wires are composed of spherical-shaped particles, which are probably related to the existence of highly disordered Co_{0.75}Ni_{0.25} phase. Moreover, the surface of these wires is surrounded by a plate-like structure (Fig. 3b). The presence of both ball-like and needle-like particles is related to the metallic Ni-Co, whereas the observed plates result from the presence of cobalt and nickel oxides, which was confirmed through EDX spectra analysis (Fig. 3g and h). Interestingly, as shown in Fig. 3g and h, the palladium presence was observed at the edges of the cobalt and nickel oxide layers (area 1 on Fig. 3g). According to that, while the concentration of Ni and Co plays a crucial role in the formation of wires with two different surface morphologies, the existence or absence of an oxidized surface should reflect in the formation of Pd NPs.

The concentration of this oxidized surface is also Co:Ni ratio-dependent. When NWs are composed of ultra-fine crystallites of hexagonal $Co_{0.75}Ni_{0.25}$ phase, the oxidized layers form a plate-like structure. However, when the

Table 2 $-$ Identified, on XRD patterns, phases with the cell parameters and fitting parameters (R _{wp} , R _p , and χ^2).						
Sample	Identified phases	Space group	Cell parameters	R _{wp} (%)	R _p (%)	χ^2
0.5 mM of I	K ₂ PdCl ₄					
Co_9-Ni_1	Ni(Co)	Fm-3m	$a=b=c=$ 3.528 Å; $lpha=eta=\gamma=$ 90°	1.46	1.15	1.04
	Co _{0.75} Ni ₂₅	P63/mmc	a = b = 2.721 Å; c = 4.078 Å; $\alpha = \beta = 90^\circ$; $\gamma = 120^\circ$			
Co ₇ –Ni ₃	Ni(Co)	Fm-3m	$a=b=c=$ 3.5404 Å; $lpha=eta=\gamma=$ 90°	1.64	1.29	1.34
	Co _{0.75} Ni ₂₅	P63/mmc	a = b = 2.452 Å; c = 4.182 Å; $\alpha = \beta = 90^\circ$; $\gamma = 120^\circ$			
Co ₅ –Ni ₅	Ni(Co)	Fm-3m	$a=b=c=$ 3.5404 Å; $lpha=eta=\gamma=$ 90°	1.72	1.32	1.43
	Co _{0.75} Ni ₂₅	P63/mmc	a = b = 2.516 Å; c = 4.407 Å; $\alpha = \beta = 90^{\circ}$; $\gamma = 120^{\circ}$			
Co ₃ –Ni ₇	Ni(Co)	Fm-3m	$a=b=c=$ 3.5389 Å; $lpha=eta=\gamma=$ 90°	1.81	1.33	1.55
	Co _{0.75} Ni ₂₅	P63/mmc	a = b = 2.602 Å; c = 3.814 Å; $\alpha = \beta = 90^{\circ}$; $\gamma = 120^{\circ}$			
Co ₁ –Ni ₉	Ni(Co)	Fm-3m	$a=b=c=$ 3.5316 Å; $lpha=eta=\gamma=$ 90°	2.19	1.53	1.96
	Co _{0.75} Ni ₂₅	P63/mmc	a = b = 2.355 Å; c = 3.830 Å; $\alpha = \beta = 90^\circ$; $\gamma = 120^\circ$			
0.75 mM of	K ₂ PdCl ₄					
Co_9-Ni_1	Ni(Co)	Fm-3m	$a=b=c=$ 3.537 Å; $lpha=eta=\gamma=$ 90°	1.65	1.30	1.26
	Co _{0.75} Ni ₂₅	P63/mmc	a = b = 2.548 Å; c = 4.305 Å; $\alpha = \beta = 90^{\circ}$; $\gamma = 120^{\circ}$			
Co ₇ –Ni ₃	Ni(Co)	Fm-3m	a = b = c = 3.5460 Å; $\alpha = \beta = \gamma = 90^{\circ}$	1.47	1.16	1.01
	Co _{0.75} Ni ₂₅	P63/mmc	a = b = 2.539 Å; c = 4.095 Å; $\alpha = \beta = 90^{\circ}$; $\gamma = 120^{\circ}$			
Co ₅ –Ni ₅	Ni(Co)	Fm-3m	$a=b=c=$ 3.53 Å; $lpha=eta=\gamma=$ 90°	1.78	1.37	1.49
	Co _{0.75} Ni ₂₅	P63/mmc	$a=b=2.564$ Å; $c=4.114$ Å; $lpha=eta=90^\circ$; $\gamma=120^\circ$			
Co ₃ –Ni ₇	Ni(Co)	Fm-3m	$a=b=c=$ 3.54 Å; $lpha=eta=\gamma=$ 90°	2.11	1.63	2.00
	Co _{0.75} Ni ₂₅	P63/mmc	a = b = 2.526 Å; c = 4.329 Å; $\alpha = \beta = 90^{\circ}$; $\gamma = 120^{\circ}$			
Co ₁ -Ni ₉	Ni(Co)	Fm-3m	$a=b=c=$ 3.5276 Å; $lpha=eta=\gamma=$ 90°	2.50	1.74	2.72
	CoPd	Fm-3m	$a=b=c=3.907$ Å; $lpha=eta=\gamma=90^\circ$			

concentration of this phase decrease, only the thin oxidized layer can be observed. To determine the role of oxidized surface on the palladium nanoparticles nucleation and growth, HRTEM images were investigated for different samples. The presence of two areas, in which Pd nanoparticles occur was confirmed for all analyzed samples and presented in Fig. 4. The performed analysis of HRTEM image of CoNi NWs with Co:Ni ratio equal to 9:1 confirmed the existence of two observed on STEM images areas related to the highly disordered core Co_{0.75}Ni_{0.25} phase (green-marked area in Fig. 4a and corresponding Fig. 4b) and to the presence of oxidized platelike surface of PdO2 nanoparticles (blue marked area in Fig. 4a and corresponding Fig. 4c). Whereas the core is associated with the metallic phase, the palladium was observed especially in the form of PdO₂ on an oxidized surface. While PdO₂ crystallizes especially on the oxidized surface, some palladium NPs and atomic clusters can also be observed on the highly crystalline Ni(Co) phase (Fig. 4d-f). Furthermore, the same analysis confirms the possibility of the crystallization of metallic Pd ultra-fine particles in the oxidized surface of CoNi NWs with the Co:Ni ratio equal to 5:5 (Fig. 4g and h). To summarize, the coexistence of the Pd and PdO₂ NPs is related to the Pd²⁺ ions nucleation and growth on both oxidized and metallic phases. However, the oxidized form crystallizes only on the oxidized NWs surface.

To confirm the coexistence of pure Pd and PdO_2 NPs, XPS spectra were collected for Pd/CoNi NWs with Co:Ni ratio equal to 3:7. The analysis was performed for two different Pd loadings to determine the role of Pd precursor concentration on the formation of Pd⁰ and Pd⁴⁺ states. For this purpose, the XPS spectra were collected and analyzed for NWs in an assynthesized state and also after removing the oxidized surface using Ar⁺ ion etching. The survey spectra (Fig. A.1) confirm the existence of Ni, Co, and Pd in both oxidized surface and metallic core. While the concentration of Co and Ni is similar in the surface and core (Table A1) the palladium atomic concentration is much higher on the surface than in the metallic core. The performed above analysis of TEM images assumed that all nanowires have an oxidized surface. The collection of the Ni2p photoemission spectra obtained for all investigated samples are depicted in Fig. 5a and b. All presented shapes of the $2p_{3/2}$ levels are complex because of three components at binding energies of 856.6 eV, 854.4 eV and 852.8 eV, which corresponded to the presence of Ni₂O₃ and NiO oxides on the oxidized surface and Ni atoms in the core of CoNi NWs, respectively [24,67,73,74]. The intensities or lack of particular components are different for all samples indicates their various contents in outer layers. The additional component at binding energies above 857 eV corresponded to the satellite structure of the main $2p_{3/2}$ line. The shapes of the $Co2p_{1/2}$ spectra presented in Fig. 5d and e consists of two components. The first of them at 797.6 eV of binding energy corresponded to Co₃O₄ oxides layer, while the second represented satellite structure. In order to check of the chemical composition of the core of investigated samples the ion treatment was applied by using Ar⁺ ion gun. The obtained results were presented in Fig. 5c, f and i. In the case of nickel, the shape of $2p_{3/2}$ photoemission line indicates a metallic form of nickel in the core of CoNi alloy (Fig. 5c), while in the case of Co2p_{1/2}, except for pure cobalt, a small contribution of oxides form still existed (Fig. 5f). This may be due to the significant surface roughness of the core, which does not allow complete removal of the oxidized outer layer of the core. Interestingly, also the analysis of Pd 3d_{3/2} and Pd 3d_{5/2} highresolution XPS spectra confirm the possibility of crystallization of both Pd and PdO₂ nanoparticles. Moreover, as can be seen in Fig. 5g and h, the signal from PdO₂ is strongly related to the concentration of the used precursor. While the concentration of Pd precursor was equal to 0.5 mM, the existence of a strong peak from the PdO₂ can be visible. For 0.75 mM the



Fig. 3 – STEM HAADF (a, c, e) and BF-DF (b, d, f) images of CoNi NWs synthesized using different Co:Ni precursor ratios: 9:1 (a, b), 5:5 (c, d) and 3:7 (e, f); (g) HAADF STEM image of CoNi NW synthesized using Co:Ni precursor ratio equal to 9:1 with marked areas and corresponding to them EDX spectra (h) confirming existence of Co and Ni oxides and Pd nanoparticles deposited on this oxidized surface.

intensity of peak related to the PdO₂ is much smaller. With the increasing concentration of Pd²⁺ ions, it is more favorable to crystallize the metallic Pd NPs. For lower concentrations, firstly, PdO₂ crystallizes, especially on the oxidized surface of NWs (Fig. 5g–i). While the formation of the PdO₂ is strongly related to the interaction with Ni₂O₃, NiO, and Co₃O₄ oxides, the formation of metallic Pd is related to their crystallization of the surface of the CoNi core, which manifests itself in the formation of ultra-fine NPs and atomic clusters of Pd on the surface of highly crystalline Ni. Probably the formation of metallic Pd nanoparticles on the oxides is limited. A comparison of the binding energy of particular components of the photoemission lines of Ni 2p3/2, Co $2p_{1/2}$ and Pd 3d5/2 for samples Co₃–Ni₇ with deposited Pd NPs in concentration of 0.5 and 0.75 mM of K₂PdCl₄ was presented in Table 3.

Electrochemical analysis

The cyclic voltammetry measurements were used to determine the electrochemical behavior and verify the catalytic properties of the catalysts synthesized by using two different concentrations of Pd precursor, 0.5 and 0.75 mM. Two series of electrocatalyst samples were prepared by depositing Pd NPs on Co–Ni NWs with different Co:Ni ratios. Fig. 6 shows the cyclic voltammograms (CVs) of Co:Ni NWs decorated by Pd NPs, obtained in a standard three-electrode electrochemical cell at a scan rate of 20 mVs⁻¹ in 1 M NaOH without ethanol. CV results for catalysts synthesized by using 0.5 mM palladium precursor were marked by a dashed line and for 0.75 mM by a solid line, Co₇–Ni₃ – red, Co₃–Ni₇ – blue, and Co₁–Ni₉ – magenta curve, respectively. The results for Co₅–Ni₅ and



Fig. 4 – HRTEM micrographs of CoNi NWs synthesized using Co:Ni precursors ratio: 9:1 (a-c), 3:7 (d-f) and 5:5 (g and h); (b) enlarged green marked area on (a) confirming existence of highly disordered $Co_{0.75}Ni_{0.25}$ hexagonal structure; (c) enlarged blue-marked area on (a) with identified interplanar distance characteristic for PdO₂ (101) planes; (e) enlarged green-marked area on (d) with identified interplanar distance characteristic for Ni (111) planes; (f) enlarged marked blue area on (d) confirming the existence of Pd atomic clusters at the edges of highly crystalline Ni phase; (h) enlarged green marked area on (g) with identified interplanar distance characteristic for Pd (111) planes. (For interpretation of the references to colour in this figure legend, the reader is referred to the Web version of this article.)

Co₉-Ni₁ nanowires for both Pd precursor concentrations are presented in the supplementary information (Fig. A.2). Fig. A.2 shows an overview of CV curves for all five catalysts obtained by using 0.5 (a) and 0.75 (b) mM Pd precursor for easier comparison. The profiles presented in Fig. 6 correspond to the fifth cycle of the respective catalysts. According to the literature reports, the peaks above 0.3 V were assigned to the oxidation of Pd to palladium oxide, and the reduction of the oxides thus formed [78]. It should be noted that the palladium percentage is only about 1%. In comparison to other works, where the mass fraction of Pd is 10% [67] or even 22% [73], may be noticeable differences in the intensity of these peaks. Because nano-structural architecture plays a crucial role in the catalytic activity, decoration method in synthesis of electrocatalysts and the synthesis of nanocatalysts with sophisticated shapes, became very popular and allow to obtain very high electrocatalytic activity. Therefore, inspired by these findings, we tested our catalysts in 1.0 M NaOH + 0.5 M C₂H₅. OH solution in order to investigate their catalytic ability towards ethanol oxidation. After adding ethanol into the electrolyte, the solution was bubbled with inert gas (Ar) for 30 min, and the experiment was continued. Fig. 7a shows the cyclic voltammograms of ethanol oxidation under alkaline conditions (0.5 M C₂H₅OH and 1 M NaOH) for the two series of

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Fig. 5 – Ni2p_{3/2} (a, b and c), Co2p_{1/2} (d, e and f) and Pd3d_{5/2} (g, h and i) photoemission spectra recorded for CoNi NWs (Co:Ni ratio equal to 3:7) after deposition of Pd NPs using 0.5 mM of K_2 PdCl₄ (a, d, g) and 0.75 mM of K_2 PdCl₄ (b, e, h); the spectra on (c, f and i) were obtained after Ar⁺ ion treatment procedure.

Table 3 – The binding energy of particular components of the photoemission lines of Ni 2p3/2, Co $2p_{1/2}$ and Pd 3d5/2 for samples Co_3 -Ni ₇ with deposited Pd NPs in concentration of 0.5 and 0.75 mM of K ₂ PdCl ₄ .					
		Sample 0.5 mM K ₂ PdCl ₄	: CoNi (3:7) & 0.75 mM K ₂ PdCl ₄		
		Binding Energy (eV)		References	
Photoemission line	Ni 2p _{3/2}	Co 2p _{1/2}	Pd 3d _{5/2}		
Ni atoms in CoNi core	852.8 eV			This work	
NiO	854.4 eV			[24,67,73,74]	
Ni ₂ O ₃	856.6 eV				
Co atoms in CoNi core		793.3 eV		This work	
Co ₃ O ₄		797.6 eV		[72,75,76]	
Pd atoms in CoNi core			335.8 eV	This work	
PdO ₂			337.5 eV	[14,77]	

CoNi NWs with different Co:Ni ratios and palladium precursor concentrations. CVs were recorded at 20 mV per second in the potential range from -0.4 to 1 V vs Ag/AgCl at room temperature for all synthesized catalysts. The current was normalized to the amount of Pd obtained by ICP, thus reporting mass activities. The results for Co5-Ni5 and Co9-Ni1 nanowires for both Pd precursor concentrations, similar to the CV curves, were presented in Supplementary Information (Fig. A.3 and Fig. A.4). In the case of both sample series (for different Pd precursor concentrations), Co₅-Ni₅ and Co₉-Ni₁ NWs decorated by Pd NPs showed very similar, and at the same time, low catalytic activity (Fig. A.3 and Fig. A.4). In the case of ultrahigh Co:Ni ratio equal to 9:1, the palladium form ultra-fine PdO₂ only at the edges of Co₃O₄, NiO, and Ni₂O₃ plate-like oxides. According to that, this oxidized palladium cannot react in any way with a metallic nanowire core. With

increasing concentration of Ni precursor, this plate-like structure disappears progressively. When Co:Ni ratio is equal to 1:9, palladium can crystallize in metallic form, crystallites are large (detectable even by XRD method). These well crystalized Pd NPs are responsible for very good catalytic activity (2500 mA/mg_{Pd} for 1.92% of Pd). However, which was confirmed herein, to achieve the highest catalytic activity, it is necessary to obtain a balance between highly crystalline metallic Ni(Co) cubic phase and highly disordered Co_{0.75}Ni_{0.25} phase. As it can be noticed, depending on the Co:Ni ratio, the electrocatalytic activity towards ethanol oxidation reaction is changing. The different activities of studied samples could be attributed to the changes in the structure and morphology of NWs and NPs which, as shown by the XRD and TEM results, are strongly dependent on the Co:Ni ratio. According to the Fig. A.3, the catalytic activities were increasing in the



Fig. 6 – Cyclic voltammetry curves of Co:Ni NWs decorated by Pd NPs in argon-saturated 1.0 M NaOH solution at a scan rate of 20 mV/s at room temperature. The solid lines present the results for electrocatalysts synthesized by using 0.5 mM palladium precursor, while the dashed lines by using 0.75 mM palladium precursor.

following order: $Co_9 - Ni_1 < Co_5 - Ni_5 < Co_7 - Ni_3 < Co_1 - Ni_9 < Co_3$. $-\mathrm{Ni}_7$ for 0.5 mM concentration of Pd precursor and Co_5- $-Ni_5 < Co_9 - Ni_1 < Co_7 - Ni_3 < Co_1 - Ni_9 < Co_3 - Ni_7$ for 0.75 mM concentration of Pd precursor (Fig. A.4), respectively. In both K₂PdCl₄ concentrations, the highest catalytic activity showed Co₃-Ni₇ NWs decorated by Pd NPs nanocatalysts. Mass activity for Co₃-Ni₇Pd (0.75 mM) catalyst is more than 3 times higher compared to the other catalysts (Fig. 7). For Co₃-Ni₇Pd (0.5 mM) and Co₃-Ni₇Pd (0.75 mM) catalysts, the mass activities were up to 5252 mA/mg_{Pd} and 8003 mA/mg_{Pd}, respectively. Additionally, Fig. 7c and d outlines mass activities at 0.8 and 0.9 V. As it was demonstrated in Fig. 7c and d, the Co₇-Ni₃ (red curve) and Co₃-Ni₇ NWs (blue curve) decorated by Pd NPs (0.75 mM) showed higher mass activities in comparison of sample series obtained by using 0.5 mM Pd precursors. The opposite is true only for a sample with a Co:Ni ratio of 1:9 (magenta curve).

Another important parameter describing the activity of the catalysts is the so-called onset potential, which is defined as the value of the potential at which the current rapidly rises. The lower the onset potential is, the faster the reaction starts. The EOR curves for all the catalysts in an ethanol-containing



Fig. 7 – (a) Comparison of the cyclic voltammograms (EOR curves) of electrocatalysts in the form of cobalt-nickel nanowires with different Co:Ni ratios decorated by Pd in argon-saturated 1.0 M NaOH + 0.5 M CH_3CH_2OH solution at a scan rate of 20 mV/s; (b) plot showing an enlargement of the data between 0 and 3500 mA/mg_{Pd}; (c) EOR activities of all tested catalysts at a potential of 0.8 V and (d) 0.9 V vs. Ag/AgCl. The solid lines present the results for electrocatalysts synthesized by using 0.5 mM palladium precursor, while the dashed lines by using 0.75 mM palladium precursor. The current values were normalized by the Pd content.

electrolyte, indicate that the onset potential of ethanol electrooxidation for the Co_1-Ni_9-Pd (0.75 mM) is the lowest. Despite that, in comparison to Co_3-Ni_7-Pd (0.75 mM) sample, the value of current densities is almost three times lower. As was deduced from the results in Fig. 7b, also for Co_3-Ni_7-Pd (0.75 mM) sample, the value of onset potential is shifted toward less positive values compared to the other four catalysts, and above 0.2 V vs Ag/AgCl has the highest value of current densities.

This superior catalytic activity of Co3-Ni7-Pd (0.75 mM) catalyst might be attributed to the presence of two different crystalline phases: metallic Ni(Co) cubic phase and highly disordered Co_{0.75}Ni_{0.25} phase, which can be easily oxidized to the cobalt and nickel oxides. Moreover, the nanometric size of Pd nanoparticles provides a large surface area, which is desirable for electrocatalytic activity. According to the literature, all metals: palladium, nickel and cobalt are oxophilic [14,67,70,79]. Therefore, due to the oxophilic nature of Ni and Co it could spontaneously convert Ni, Co to their oxides [14,67,70], which was confirmed by XPS results (Fig. 5). These properties cause that Ni and Co have the capacity to generate OH_{ads} at a lower potential and facilitate the oxidative desorption of the intermediate products, thus enhancing the catalytic activity of catalysts containing palladium, nickel, and cobalt [80,81]. Furthermore, besides the chemical composition of the tested catalysts and properties of palladium, cobalt, and nickel, definitely, the morphology of the obtained catalysts also play a crucial role in the ethanol oxidation reaction: a good dispersion of palladium nanoparticles causes larger exposed active surface area and possibly better interaction with the three metals [14].

The catalysts in the form of nanowires have already been previously reported in the literature [67,82,83]. Du et al. [82] described nanoporous Pd57Ni43 alloy NW catalysts with the mass activity of about 800 mA/mg_{Pd}. The authors underlined nanowires are very interesting structures as they can act as catalysts, so they do not require any support. Consequently, the detachment problems and corrosion of support that catalytic systems based on nanoparticles and support often suffered can be avoided. Similar structures based on nanowires were also synthesized by Bin et al. [83]. Nevertheless, catalyst synthesized by this group was prepared as Pd-Ni bimetallic NW networks supported on reduced graphene oxide, and the mass activity of this system was up to 604.3 mA/mg_{Pd}. However, in both cases, the electrocatalytic oxidation of formic acid was investigated. Kottayintavida and Gopalan synthesized [67] the Pd modified Ni NW catalyst towards methanol and ethanol electro oxidation. The authors showed that the PdNi catalyst obtained by galvanic replacement reaction achieved mass activity of about 1500 mA/mg_{Pd}, while for the Co₃-Ni_{7-Pd} (0.75 mM) catalyst synthesized in this work, the catalytic activity in EOR was more than five times higher (8003 mA/mg_{Pd}) (Fig. 7a). A catalyst with the same chemical composition as synthesized in this work, but a different structure, was synthesized by Zhang group [73]. However, the NiCoPd nanocatalyst was dispersed on carbon nanotubes and used as catalyst towards methanol oxidation reaction with current densities of 862 mA/mg_{Pd} (9 times less than the catalyst obtained in this work).

Big advantage of catalyst synthesized in this work is lack of platinum. In case of designing catalyst often the key challenge is removal of platinum from the catalyst composition. It is related with that the platinum is susceptible to poisoning by CO-like intermediates, leading to the blockage of their active sites and the reduction of catalytic activity [84,85]. Moreover platinum is very expensive metal. Therefore a lot of studies are focused on replacement of platinum by another metals like palladium. Zhai et al. [75] also synthesized catalyst with the unique 1D nanowire structure. They system was based on Co and Pt. As they showed PtCo nanowires synthesized by them has the electrocatalytic activity in ethanol oxidation reaction of 2203 mA/mg_{Pt}, what is about four times lower in comparison of Co₃–Ni₇_Pd (0.75 mM) catalyst synthesized in this work.

Based on the literature, we have grounds to speculate that the synergistic interactions between Pd and CoNi nanowires promote significant enhancement of catalytic activity when the Co:Ni ratio is equal to 3:7. Thus, the enhanced performance can be attributed to the direct physical contact between CoNi NWs with Pd. The crucial role of physical contact between the respective nanoparticles was underlined in the literature frequently [26,86-88]. As was shown by Drzymala et al. [26], by choosing the appropriate metal composition and ensuring the physical contact between the nanoparticles forming the nanocatalysts, the overall catalytic activity towards the ethanol oxidation reaction and durability could be improved to a large extent. Crabb et al. [88] also noticed that catalysts are more successful when two metals are in contact with each other. Therefore, we suggest that the presence of interfaces between the NiCo NWs and Pd NPs could induce charge transfer, which enhances the catalytic performance. A comparison of our catalysts with other catalyst systems based on Pd and Ni (and one case in Co) reported in the literature is shown in Table 4, demonstrating the excellent performance of our catalysts.

Conclusions

This work presented a new bimetallic CoNi support for Pd nanocatalyst, fabricated by wet chemical reduction assisted with magnetic field and using H₂PtCl₆ as the nucleating agent. Pd NPs at two precursor concentrations (K_2PdCl_4) of 0.5 and 0.75 mM were produced and deposited on the as-prepared metallic supports at varying Co:Ni ratios (9:1, 7:3, 5:5, 3:7, 1:9). The key challenge during the synthesis of Co_x-Ni_y_Pd catalyst was, first of all, preparing these structures in such a way that the CoNi nanowires are covered with palladium nanoparticles. The second limitation was necessity to obtain a balance between highly crystalline metallic Ni(Co) and highly disorder Co_{0.75}Ni_{0.25} phases in the structure of NiCo NWs. XRD, STEM, HRTEM, and XPS analyses confirmed the incorporation of palladium on CoNi bimetallic NWs and its appearance in metallic Pd and oxide PdO₂ forms. The catalytic activity tests

Table 4 – Comparison of the mass activities of the as-prepared Co:Ni_Pd catalyst with other catalyst systems based on Pd, Ni or Co.					
Catalyst	Electrolyte	Mass activity $[mA/mg_{Pd}]$	Reference		
Pd modified Ni NW	0.5 M KOH + 0.5 M Ethanol	1480	[67]		
Pd ₂ Ni ₁ /C	1.0 M NaOH + 1 M Ethanol	2957	[89]		
NiCoPd/CNT	1.0 M NaOH + 1 M Methanol	862	[73]		
Pd ₈₃ Ni ₁₇ HNS aerogel electrocatalyst	1.0 M NaOH + 1 M Ethanol	3630	[90]		
Co ₁ -Ni ₉ Pd (0.5mmol)	1 M NaOH + 0.5 M Ethanol	3148	This work		
Co ₃ -Ni ₇ Pd _(0.5mmol)	1 M NaOH + 0.5 M Ethanol	5252	This work		
Co ₃ -Ni ₇₋ Pd (0.75mmol)	1 M NaOH + 0.5 M Ethanol	8003	This work		

showed that the obtained system can be successfully used as a catalyst in fuel cells powered by ethanol. The decoration of CoNi NWs by Pd NPs leads to a much higher availability of the palladium surface deposited on CoNi nanowires. As was shown, it has the effect of improving catalytic activity, because the connection between Pd and CoNi NWs plays a crucial role in the catalyst response. When Co:Ni ratio was equal to 1:9, palladium crystallized in metallic form, and the crystallites were large. The existence of these well crystalized Pd nanoparticles was responsible for excellent catalytic activity (2500 mA/mg_{Pd} for 1.92% of Pd).

However, the best response was obtained for NWs, in which Co:Ni ratio was equal to 3:7 for both Pd loadings 0.5 and 0.75 mM. For this material, it was confirmed that the palladium crystallized mainly in the PdO₂ form at the oxidized surface and as metallic Pd atomic clusters on the metallic support. The coexistence of these two forms (Pd and PdO₂) and balance between metallic Ni(Co) and oxidized layer composed of Co₃O₄, NiO, and Ni₂O₃ allowed to achieve unexpectable high catalytic activity equal to 8003 mA/mgPd for only 0.86 wt% of Pd loading on the CoNi nanowire surface. Summarizing, the results obtained in the present work showed that by simultaneous consideration of the optimal chemical composition, crystallographic structure and welldefined shape of the novelty designed catalysts, the overall catalytic activity could be definitely improved. These three features represent a very important potential of newly designed catalysts described in this work. Thanks to combining of both composition and shape, the synthesis of a novel generation of efficient and more active catalyst toward ethanol oxidation reaction turned out to be possible. In addition, a novelty is also the replacement of the carbon support by bimetallic nanowires, what allow to avoid the detachment problems and corrosion of support. Both of selected metals (Ni and Co) have oxophilic nature and both of them take part in the catalytic reaction. Due to their oxophilic nature, it could spontaneously convert Ni and Co to their oxides. It is associated with the generation of active OH species, which oxidize the molecules adsorbed on the surfaces, facilitate CO removal, increase the tolerance to the poisoning of the noble metal by CO and finally enhance the catalytic activity. The innovativeness of the obtained system is also related to the lack of platinum in the proposed catalytic system, what is one of the biggest challenges when designing catalysts especially due to the economic terms. Moreover platinum is susceptible to poisoning by EOR byproducts. As was shown in this work, it is possible to

obtain efficient novelty platinum-free nanocatalysts with high catalytic activity in EOR. As a conclusion, this study clearly indicates that the developed Pd/CoNi catalytic system can be successfully used as an efficient electrocatalyst for ethanol oxidation in DEFCs (Direct Ethanol Fuel Cells) in an alkaline medium. In the future, the others metal nanoparticles will be tested with connection of CoNi nanowires.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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Appendix A. Supplementary data

Supplementary data related to this article can be found at https://doi.org/10.1016/j.ijhydene.2021.11.177.

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