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Self-assembled monolayers of polyoxovanadates with phthalocyaninato lanthanide moieties on gold surfaces†

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The two first representatives of phthalocyaninato (Pc) lanthanide-ligated polyoxovanadate cages $\{[V_{12}O_{32}(Cl)](LnPc)_n\}^{n-5}$ ($n = 1$ or 2 , $Ln = Yb^{3+}$) were synthesised and fully characterised. These magnetic complexes form two-dimensional self-assembled monolayers exhibiting electrical conductivity on gold substrate surfaces, as assessed by using an EGaIn tip.

The possibility of forming stimuli-responsive two-dimensional (2D) monolayers¹ paves the way for implementation of nanostructured molecule–surface interfaces into highly sought-after molecule-based computer memory cells. Self-assembled monolayers (SAMs) consisting of tailor-made metal complexes² constitute one of the promising ways of addressing the challenges³ in the area of molecular electronics.^{4,5} To transit to the actual device fabrication, experimentalists need first and foremost to achieve active control of contact-surface impacted single-molecule structure–property stability and of the reproducibility of molecular conductivity measurements. A continuous optimisation of coordination compounds is furthermore required to eventually minimise resistivity and improve the conductivity of 2D SAMs in the applied electrode environment.

Herein we report on our efforts towards the development of such SAMs whose molecular electronic states can be manipulated on air-stable substrate surfaces by electrical means and their working functions can be tracked by (micro)-spectroscopic methods. Specifically, we elaborate on polyoxovanadates⁶ (POVs)

and, specifically, their organic–inorganic hybrid derivatives that have been identified⁷ as being able to implement multilevel data storage and processing functions in so-called “More than Moore” IT devices. We describe the synthesis, structure and magnetism of nBu_4N^+ -charged balanced complexes with the general formulas $\{[V_{12}O_{32}(Cl)](LnPc)\}^{4-}$ (**mono**⁴⁻) and $\{[V_{12}O_{32}(Cl)](LnPc)_2\}^{3-}$ (**bis**³⁻) where Pc is the phthalocyanine dianion (Fig. 1). This Communication is a focused case study with $Ln = Yb^{3+}$, emphasising the significance of covalent grafting of late and early lanthanide–Pc moieties onto the lacunary POV core^{8,9} and its far-reaching implications on molecular charge transport characteristics. To the best of our knowledge, the direct interaction between metal–Pc moieties and the polyoxometalate^{10,11} (POM) unit was shown only once. Drain and co-workers synthesised¹² hybrid $(nBu_4N)_5[PW^{VI}_{11}O_{39}(MPc)]$ compounds where the

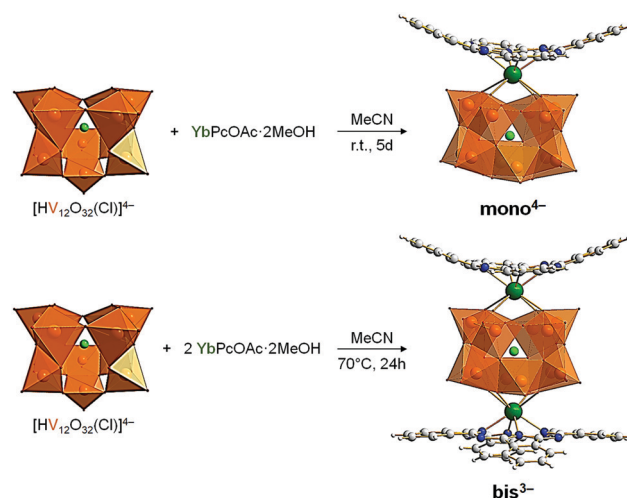


Fig. 1 Synthesis of fully-oxidised $\{[V_{12}O_{32}(Cl)](YbPc)_n\}^{n-5}$ hybrid poly-anions with $n = 1$ (top, **mono**⁴⁻) and $n = 2$ (bottom, **bis**³⁻). The H^+ proton in $[HV_{12}O_{32}(Cl)]^{4-}$ is not shown. r.t. = room temperature. Colour code: Yb = dark green, V = orange polyhedra, Cl = pale green, O = red, N = blue, C = grey, and H = white. See the ESI,† for details of synthesis and crystallographic data (Table S3).

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fully-oxidised, lacunary Keggin POM is capped by MPCe^{2+} with $\text{M} = \text{Zr}^{\text{IV}}$ or Hf^{IV} . Note that there are however plenty of hybrid compounds^{13,14} where the POM unit and the metal-Pc moieties are separated by electrostatic interactions or by an organic linker.

Hybrids **mono**⁴⁻ and **bis**³⁻ were isolated as polycrystalline powders in high yields (95 and 91%) from the reactions of the $(n\text{Bu}_4\text{N})_4[\text{HV}_{12}\text{O}_{32}(\text{Cl})]$ precursor with $\text{YbPcOAc}\cdot 2\text{MeOH}$ (OAc = acetate) in a 1:1 and 1:2.2 ratio in MeCN, respectively (Fig. 1). Single-crystal X-ray diffraction studies indicated reactivity-induced changes of the POV building unit from a “closed”-to a “tube”-like $\text{V}_{12}\text{O}_{32}$ structure^{15,16} that is thus capped by one (in **mono**⁴⁻) or sandwiched by two (in **bis**³⁻) YbPc^+ moieties (Fig. 1). The formal oxidation states of the vanadium and lanthanide atoms were confirmed by bond valence sum calculations [$\sum(\text{V}^{\text{V}}) = 4.99\text{--}5.13$, $\sum(\text{Yb}^{\text{III}}) = 2.81$] for **mono**⁴⁻ and [$\sum(\text{V}^{\text{V}}) = 4.80\text{--}5.26$, $\sum(\text{Yb}^{\text{III}}) = 2.80$] for **bis**³⁻. The formation, composition and electronic structure of the fully-oxidised **mono**⁴⁻ and **bis**³⁻ are furthermore supported by elemental and thermogravimetric analyses, electrospray ionisation mass spectrometry and SQUID magnetometry (for details see the ESI†).

Magnetism of both compounds arises from Yb^{3+} ions and is dominated by spin-orbit interaction. Because Yb^{3+} has a more than half-filled 4f valence shell, its ground multiplet should correspond to a high total magnetic moment $J = 7/2$.¹⁷ Since the energy gap between the ground and the first excited multiplet in Yb^{3+} is supposed to be around 14000 K¹⁸ only the ground multiplet was considered in modelling of the magnetism. In Fig. 2 the molar susceptibility and magnetisation of **mono**⁴⁻ are depicted together with theoretical fits. The details of magnetic modelling of **bis**³⁻ (Fig. S12) can be found in the ESI.†

The electronic structure of **mono**⁴⁻ was also assessed by density functional theory calculations (for details see the ESI†) and it is in line with the obtained magnetic data. As illustrated in Fig. 3, the negative charge is transferred from the Pc to the POV subunit, while a doublet spin state is strongly localised at

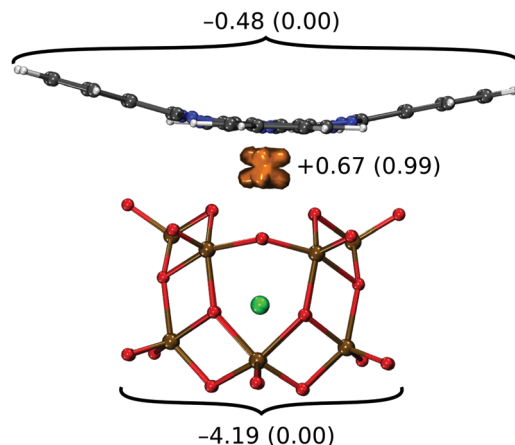


Fig. 3 Hirshfeld partial charges and regions with increased α -spin density (orange isosurface) in **mono**⁴⁻. Sum of partial charges is shown without brackets, while the spin density is indicated in brackets.

the Yb atom. The computed $\text{Yb}\cdots\text{Cl}$ distance of 4.06 Å matches the experimental value of 4.1001(29) Å determined by X-ray diffraction which is *ca.* 0.34 Å longer than the one in the **bis**³⁻ derivative (exptl. 3.760(2)–3.761(2) Å). This is due to the fact that the Cl atom is pushed away from Yb by *ca.* 0.21 Å in **mono**⁴⁻ if the POV centre is taken as a reference point. The asymmetry within the POV subunit is induced by the coordinated Yb atom contracting the host-guest vanadium-oxo cluster. The largest O–O distance between the Yb-coordinated oxygen atoms is 3.99 Å (exptl. 4.0611(279) Å), and it is increased to 4.71 Å (exptl. 4.9368(85) Å) on the opposite POV side for the comparable oxygen atoms.

Next, we succeeded in growing SAMs by immersing freshly cleaved templated-stripped¹⁹ gold substrates (atomically smooth Au^{TS}) in a ~ 0.1 mM methanolic solution of each tested, thermally stable (up to *ca.* 200 °C) coordination compound overnight. The formation of SAMs is shown in Fig. 4A–C. SAMs were contacted with an EGaIn²⁰ tip to form tunnel junctions with the structure $\text{Au}^{\text{TS}}//\text{SAM}/\text{Ga}_2\text{O}_3/\text{EGaIn}$, where “/” denotes the interface defined by chemisorption and “//” by physisorption. The SAM quality was assessed by measuring the morphology of the fabricated sample surfaces (Fig. 4A–C and Fig. S13 in the ESI†) using atomic force microscopy (AFM). The observed particles exhibiting the size of physisorbed **mono**⁴⁻ and **bis**³⁻ are homogeneously distributed over the Au^{TS} substrate surface (Fig. 4A and Fig. S13 in the ESI†) that is characterised by the absence of molecular aggregates (Fig. 4B). The roughness values of SAMs are estimated to be 0.417 nm for **mono**⁴⁻ and 0.382 nm for **bis**³⁻. They are thus higher than those (0.1 nm) obtained for the bare Au^{TS} substrates, suggesting that the target coordination compounds were successfully immobilised on gold. The formation of monolayers was confirmed by ellipsometric measurements of the SAM thickness, resulting in values 0.96 ± 0.07 nm for **mono**⁴⁻ and 1.17 ± 0.07 nm for **bis**³⁻ that are in excellent agreement with the expected height values (*ca.* 0.90 vs. 1.20 nm) for these hybrid polyoxoanions.

The measured current density–voltage (J – V) characteristics of the engineered SAMs are depicted in Fig. 4D. Interestingly,

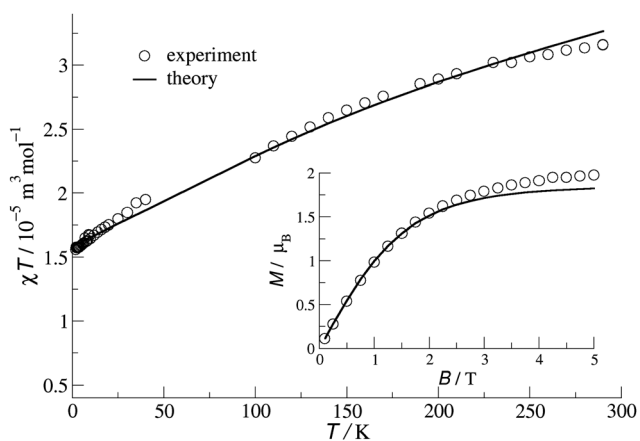


Fig. 2 Molar susceptibility ($B = 0.1$ T) and magnetisation ($T = 2$ K) for the polycrystalline powder sample of **mono**⁴⁻ (circles) with theoretical fits (solid lines).

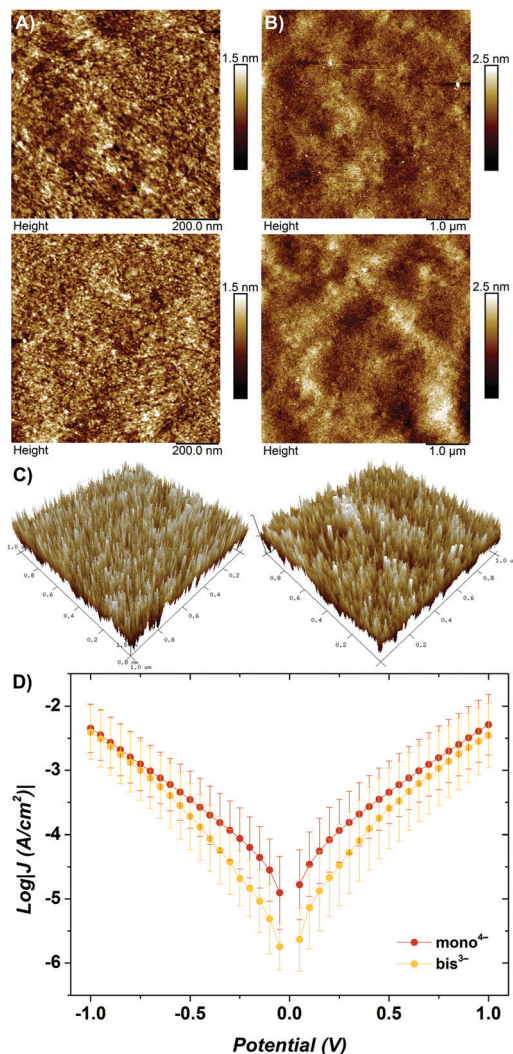


Fig. 4 AFM height images of SAMs on Au^{TS} (scanned at A) 500 μm and (B) 5 μm and the 3D morphology of the sample surfaces (C) for **mono**⁴⁻ (top/left) and **bis**³⁻ (bottom/right). (D) Plots of log current density vs. applied potential for SAMs of **mono**⁴⁻ and **bis**³⁻ on Au^{TS}. Values of $\log|J|$ at $V = 0$ V are omitted for clarity. $J = I/A$ where I and A are the current and area of the junction, respectively. Error bars represent the standard deviation of Gaussian fits.

the SAM of **bis**³⁻ demonstrates somewhat lower molecular conductivity than **mono**⁴⁻ at a low bias voltage. However, at a higher bias $\geq \pm 1$ V the difference in the rate of tunnelling charge-transport vanishes, likely due to the molecular states coming into resonance with the electrode/substrate Fermi level with the increasing bias voltage.²¹ Although the magnetism of **mono**⁴⁻ and **bis**³⁻ is dominated by the spin-orbit effects of Yb³⁺ ions, the molecular charge-transport characteristics are expected to be strongly influenced by electron transport²² through the fully-oxidised V^V centres. This is in accordance with our latest results²³ indicating that the Cu(II) centre is the main transmission channel across tunnel junctions formed of a series of charge-neutral dinuclear 3d–4f coordination complexes [CuLn(L-SME)₂(OOCMe)₂(NO₃)]₂·xMeOH (Ln = Gd, Tb, Dy and Y; L = Schiff base; $x = 0.75$ –1). The latter were similarly

immobilised on Au^{TS} in the form of monolayers; however, these SAMs are characterised by lower molecular conductivity compared to the SAMs of **mono**⁴⁻ and **bis**³⁻. In view of these data, further efforts towards the grafting of covalently bound early lanthanide–Pc moieties onto (reduced) POM cores need to be devoted in order to substantially break the energy level alignment of molecular magnetic states with respect to the Fermi level of the applied electrode surfaces.

The findings described herein offer guidance to customizing a hybrid material that can be assembled of any stable lacunary POM core and phthalocyaninato lanthanide moieties. The structure-directing and -stabilising role of the charge balancing counteranions is also worth mentioning.²⁴ A comparative analysis of the molecular charge transport characteristics of the hybrid materials and their precursors ((*n*Bu₄N)₄[HV₁₂O₃₂(Cl)] and LnPcOAc) will be published elsewhere. The potential possibilities of controlled fine-tuning of their structural, electrochemical and magnetic characteristics in the applied electrode environment are expected to make 2D monolayers of such POM-based coordination compounds suitable candidates for practical integration into memory cells.^{25,26} To achieve this technology transfer-oriented goal, we will further seek to evoke the SAM conductivity switching response to the electric field of a scanning probe microscope as a function of oxidation and magnetic states of metal ions or temperature-controlled molecular ordering and/or adsorption geometry in monolayers.

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Conflicts of interest

There are no conflicts to declare.

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