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# Vibrational dynamics of fullerene molecules adsorbed on metal surfaces studied with synchrotron infrared radiation

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**Abstract.** Infrared (IR) spectroscopy of chemisorbed  $C_{60}$  on Ag (111), Au (110) and Cu (100) reveals that a non-IR-active mode becomes active upon adsorption, and that its frequency shifts proportionally with the charge transferred from the metal to the molecule by about 5 cm<sup>-1</sup> per electron. The temperature dependence of the frequency and the width of this IR feature have also been followed for  $C_{60}/Cu$  (100) and were found to agree well with a weak anharmonic coupling (dephasing) to a low-frequency mode, which we suggest to be the frustrated translational mode of the adsorbed molecules.

Additionally, the adsorption is accompanied by a broadband reflectance change, which is interpreted as due to the scattering of conduction electrons of the metal surface by the adsorbate. The reflectance change allows determination of the friction coefficient of the  $C_{60}$  molecules, which results in rather small values ( $\sim 2 \times 10^9 \, \text{s}^{-1}$  for Ag and Au, and  $\sim 1.6 \times 10^9 \, \text{s}^{-1}$  for Cu), consistent with a marked metallic character of the adsorbed molecules.

Pre-dosing of alkali atoms onto the metal substrates drastically changes the IR spectra recorded during subsequent  $C_{60}$  deposition: anti-absorption bands, as well as an increase of the broadband reflectance, occur and are interpreted as due to strong electron–phonon coupling with induced surface states.

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Electron–phonon coupling plays an important role in determining the physical properties of fullerene-based materials, its most important manifestation being the superconductivity exhibited by fullerides ( $A_xC_{60}$ , A=alkali or alkaline-earth metal) at transition temperatures only exceeded by those of the cuprates. Not all of the  $A_xC_{60}$  salts are superconductors; they can also be either semiconductors or metals and present a variety of crystallographic phases, depending on the

concentration and the type of metal atoms used to form the compound [1]. Electron-phonon coupling does not only determine the transport properties of these materials, it also has a strong impact on the vibrational spectra, as was first found by Fu et al. who discovered the giant vibrational resonances in  $K_6C_{60}$  [2]. Later on, it became clear that the vibrational spectra of all metal fullerides [3] exhibit mode-specific shifts and intensity variations associated with electron donation to the fullerene molecules from the metal atoms intercalated into the interstitial sites of the lattice. The physical origin of the effects of the charge transfer on the phonon spectrum was clarified by ab initio calculations [4] for K<sub>6</sub>C<sub>60</sub>: the excess electrons are accommodated in the lowest unoccupied  $(t_{1u})$ orbitals of C<sub>60</sub>, which have anti-bonding character [5], leading to a structural relaxation of the molecule. This structural relaxation is primarily responsible for the frequency shifts, more specifically for the softening of the tangential modes. The intensity variations of the infrared (IR) spectrum are instead mainly brought about by electron-phonon coupling, which plays an important role in the vibrational excitation process. A first theoretical approach to the problem was the charge-phonon model [6], which considered the coupling of the dipole-active  $t_{1u}$  modes to the  $t_{1u} - t_{1g}$  electronic transitions. It was subsequently suggested that this model had been oversimplified, since couplings to many different electronic excitations have to be considered to get a consistent picture [4]. However, the detailed understanding of the electron phonon coupling in the fullerides is hampered by the complicated interplay between structural, vibrational and electronic properties in these compounds.

In this paper we contribute to this subject by investigating the electron–phonon coupling signatures in the vibrational spectra of  $C_{60}$  deposited onto metal surfaces. There the fullerenes are chemisorbed but the degree of hybridization between  $C_{60}$  orbitals and the substrate electronic states and the amount of charge transfer vary greatly from substrate to substrate [7]. However, adsorbing the fullerene molecules on well-defined substrates enables one to control the physical properties of the overlayer with a precision unmatched in bulk materials. Mode-specific shifts and intensity variations similar to those

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in fullerides have been observed in high-resolution electronenergy loss [8-10] and infrared [11] spectra of C<sub>60</sub> monolayers on bare and alkali-covered metal surfaces. Our aim here was to take advantage of intrinsic high brightness, high stability and proportionality to the stored electron beam of synchrotron IR spectroscopy to elucidate the physical origin of these changes by also recording the IR broadband absorption on which the specific vibrational bands are superimposed. Both the vibrational bands and the broadband absorption revealed a surprisingly rich scenario for the vibrational dynamics of C<sub>60</sub>, comprising both electron–phonon and purely vibrational coupling. During the adsorption on Ag (111), Au (110) and Cu (100) we found a broadband reflectance decrease typical of what Persson and Volokitin [12] were able to explain as a friction phenomenon between the adsorbate vibrations, with a component of velocity oscillating parallel to the surface (such as the frustrated translations and rotations) and motion of the conduction electrons within the skin depth and driven by the external electric field of the infrared light (see also [13]). At the same time a vibrational mode which is not IR-active for the isolated molecule appears and shows a temperature-dependent frequency and bandwidth change characteristic of a change in adsorption state and an anharmonic coupling [14] to a low-frequency mode. This low-frequency mode lies within the energy domain of the frustrated translation of C<sub>60</sub>. When the metal surfaces were pre-dosed with potassium we found an anti-absorption line shape for the new IR-active mode, suggesting that now the coupling of the vibrational mode to an induced surface state has become the process which dominates the IR spectrum.

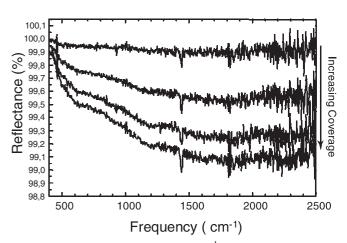
## 1 Experimental

The IR spectra were collected with the synchrotron infrared reflection-absorption spectroscopy set-up of beamline U4IR at the National Synchrotron Light Source, which is described in detail elsewhere [15]. The use of a synchrotron source is strictly necessary for these experiments since a thermal infrared source is subject to temperature fluctuations, which cause uncertainty in the incident-beam intensity. The synchrotron-radiation brightness, on the other hand, is determined only by the stored current, which can be accurately measured for the normalization of each of the spectra. This normalization procedure can be adopted only if another crucial requirement is fulfilled, i.e. the electron-beam position in the ring remains stable during the whole experiment. At the National Synchrotron Light Source this stability is achieved by means of a global feedback system [15] developed some years ago, which continues to be improved. Using a Cu: Ge detector, the broadband IR-reflectance change and the absorption bands were measured during C<sub>60</sub> deposition by collecting a spectrum (256 co-added scans) at a resolution of 4 cm<sup>-1</sup> over each 90 s of dosing. In order to avoid instabilities, which can result in intensity oscillations within the spectral distribution of the broadband reflectance change, we adopted a stringent procedure of data selection: five spectra of 256 scans each were recorded before starting the fullerene deposition; the last spectrum was ratioed to the first, and exposure to C<sub>60</sub> was initiated only if a 'flat' baseline was obtained between 300 and 3000 cm<sup>-1</sup>. C<sub>60</sub> was deposited onto the metal surfaces by resistive heating of a tantalum crucible to about 700 K. The pressure in the chamber rose from  $1 \times 10^{-10}$  Torr to  $9 \times 10^{-10}$  Torr during dosing. In order to avoid an additional source of baseline drift in the spectra due to the thermal loading of the metallic substrate, a deposition rate of approximately one monolayer (ML) per 15 min was used. The deposition time corresponding to the completion of one fullerene monolayer was calibrated by saturating the substrate kept at or slightly above (570 K) the desorption temperature of the multilayer (455 K) [16] and measuring the ratio of the overlayer/substrate Auger peaks. Auger spectra were also collected after each IR experiment to check the fullerene coverage. At the beginning of each experiment, the Ag (111), Au (110) or Cu (100) single crystal was cleaned by cycles of Ar<sup>+</sup> sputtering followed by annealing to restore the surface ordering, until no impurities could be detected by Auger spectroscopy and sharp low-energy electron-diffraction (LEED) patterns were observed. For alkali deposition during which the pressure in the chamber never exceeded  $2 \times 10^{-10}$  Torr, a thoroughly outgassed K getter was used. For each exposure, the same data-selection procedure was followed to make sure that no drift was present. The substrate temperature was monitored by a thermocouple inserted in a hole in the substrate and could be varied by filling the sample manipulator with liquid N<sub>2</sub> and by resistive heating.

#### 2 Results and discussion

Figure 1 shows the IR spectra collected during  $C_{60}$  deposition on a  $Cu\,(100)$  substrate held at room temperature. One clearly distinguishes only one IR band at  $1440\,\mathrm{cm^{-1}}$  superimposed on a broadband IR reflectance change. The amplitudes of both features increase with fullerene coverage and saturate when a complete monolayer is reached. We note that with our slow deposition rate the growth at room temperature proceeds layer by layer, as verified by measuring the ratio of the overlayer/substrate Auger peaks vs. deposition time (not shown). The overall reflectance shape remains unchanged.

Similar experiments were performed on Ag (111) and Au (110) and the broadband reflectance change associated



**Fig. 1.** Infrared spectra in the  $400-2500\,\mathrm{cm}^{-1}$  region of a Cu(100) surface collected during C<sub>60</sub> deposition at room temperature. A broadband reflectance change as well as one absorption band at  $1440\,\mathrm{cm}^{-1}$  are clearly seen

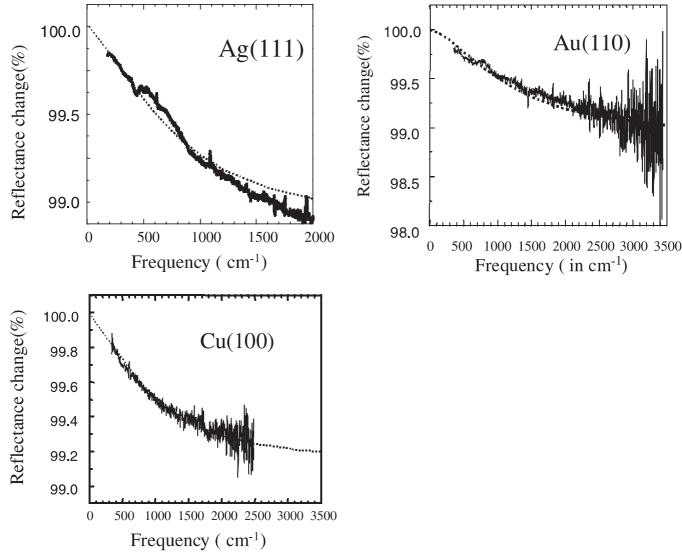


Fig. 2. Broadband reflectance change for 1 ML  $C_{60}$  on Ag (111) (top left), on Au (110) (top right) and on Cu (100) (bottom). The broken line is a fit according to the Persson–Volokitin model [12]. The dotted lines have been obtained with the plasma frequencies of 74 560, 65 760 and 87 100 cm<sup>-1</sup> and the skin depths of 31.5, 35.8 and 27 nm for Ag, Au and Cu, respectively

with the deposition of one monolayer of C<sub>60</sub> on each of the three single-crystal substrates is shown in Fig. 2. The specific vibrational bands also visible in the spectra will be discussed later. Here we note that the broadband IR reflectivity change has a frequency dependence in good agreement with the theory, which considers the electronic energy dissipation arising from the scattering of the substrate conduction electrons by the adsorbate [12, 17]. The relative motion between the sliding adsorbate and the electron 'fluid' will give rise to a dragforce action on the adsorbate. So, one can define an electronic friction coefficient which has been set to be equal to the inverse lifetime of the frustrated translation of the adsorbate due to creation of electron-hole pairs. The microscopic theory of the friction [18], based on the Newns-Anderson model of chemisorption, expresses this electronic friction coefficient in terms of the width of the adsorbate-induced resonance state,  $\Gamma$ , the adsorbate-induced density of states at the Fermi level,  $\varrho(E_{\rm F})$ , and a term which accounts for the electron-transfer interaction between the adsorbate orbital and metal states. According to the diffuse-scattering model, the friction coef-

ficient can be obtained from the asymptotic 'high-frequency' value of the IR reflectance change

$$\Delta R_{\rm p} = -\frac{4}{c} \frac{n_{\rm a}}{n} \frac{M}{m} \frac{\eta}{\cos \theta} F, \qquad (1)$$

where F is a correction factor which is important at grazing incidence geometry [19],  $F = \cos^2\theta/(\cos^2\theta + \omega^2/\omega_p^2)$ , and  $\theta$ , m,  $\omega_p$ ,  $\omega$  and  $n_a$  are the angle of incidence, the electron mass, the plasma frequency, the frequency of the light and the density of the adsorbate, respectively. The broken lines in Fig. 2 are the fits of the experimental data to this theory, which were obtained using the plasma frequencies of 74560, 65760 and 87 100 cm<sup>-1</sup> and the skin depths of 31.5, 35.8 and 27 nm for Ag, Au and Cu, respectively. The asymptotic limits of the reflectance change are  $1 \pm 0.1\%$  for Ag (111) and Au (110) and  $0.8 \pm 0.1\%$  for Cu (100). Accounting for the  $C_{60}$  mass (720 × 1.66 × 10<sup>-27</sup> kg) and the average density for one monolayer of  $C_{60}$  on noble-metal surfaces ( $\approx 1.15 \times 10^{-2} \text{ Å}^{-2}$ ), the friction coefficient is found

to be  $2.08\times10^9\,s^{-1}$  for  $C_{60}$  on Ag (111) and Au (110) and  $1.66\times10^9\,s^{-1}$  for  $C_{60}$  on Cu (100).

The same scattering mechanism of the substrate conduction electrons by the adsorbate will also cause a resistance change in a metallic thin film when adsorbates are chemisorbed on the surface [12, 15, 18]. Our results can therefore be compared with resistance-change measurements during C<sub>60</sub> deposition on copper films evaporated on sapphire substrates, where a value of 5 Å<sup>2</sup> for the scattering cross section of the adsorbed molecules has been found [20]. Using the bulk parameters of copper, a friction coefficient of  $9.4 \times 10^8 \,\mathrm{s}^{-1}$  can be deduced, which is in fairly good agreement with the value obtained from IR reflectance measurements, considering the differences in surface structure. Thus we can deduce that the diffuse-scattering model [12, 15, 18] also accounts for the shape and amplitude of the broadband IR reflectance change induced by the adsorption of large molecules like  $C_{60}$ .

The friction coefficients obtained for all three metal surfaces are proportional to the inverse lifetimes of the frustrated translation of the adsorbate due to creation of electron-hole pairs of the order of hundreds of picoseconds, and this can be considered an independent confirmation of the metallic character of the C<sub>60</sub> monolayer, previously deduced from direct [21,22] and inverse [23,24] photoemission data. The values of the friction coefficient are about six times smaller than that found for CO adsorbed on Cu (111) films [15]. Thus, the product  $\Gamma \varrho(E_{\rm F})$  is about four times higher for C<sub>60</sub> adsorbed on noble-metal surfaces, compared to CO. For the CO/Cu (111) system, the  $2\pi^*$  state is known to lie at  $\sim 3.35 \, \text{eV}$  above the Fermi level, with a width of  $0.1 \, \text{eV} < \Gamma < 0.6 \, \text{eV}$  [25, 26]. For  $C_{60}$  on the three metallic substrates, the width of the LUMO (lowest unoccupied molecular orbital) ( $\pi^*$  character) is difficult to evaluate: direct [23] and inverse [24, 25] photoemission data show rather broad spectral features both for the occupied and the unoccupied parts of the LUMO. However, by comparison with the HOMO (highest occupied molecular orbital) band, it seems plausible that most of this width is due to vibronic coupling and that the actual bandwidth amounts to less than 0.1 eV [27]. One therefore deduces that the adsorbate-induced  $\varrho(E_{\rm F})$  is much larger for C<sub>60</sub> than for CO.

Let us turn our attention now to the specific vibrational bands of the molecule: Fig. 3 shows the spectral region between 1250 and 1600 cm<sup>-1</sup> of the IR spectra acquired while dosing C<sub>60</sub> onto Ag (111) at room temperature. As the coverage increases, a band at 1442 cm<sup>-1</sup> appears and shifts to higher wavenumbers. Only very close to monolayer coverage (last spectrum) can a second band centered at 1388 cm<sup>-1</sup> be discerned. No additional bands are seen at lower wavenumbers. The spectra for room-temperature adsorption on the other two metals are similar: a detailed inspection of the spectrum for 1 ML C<sub>60</sub> on Cu (100) (see Fig. 1) shows only one strong band at 1440 cm<sup>-1</sup>, while for 1 ML C<sub>60</sub> on Au (110) (see Fig. 2) only one strong band at 1445 cm<sup>-1</sup> can be seen. In order to better understand these features, we performed another experiment in which the fullerene molecules were prevented from chemisorption onto the Cu(001) substrate by a pre-coverage of 1 ML of CO. The IR spectrum shown in Fig. 4 corresponds to a nominal coverage of 1 ML of C<sub>60</sub>. The very strong band at 2083 cm<sup>-1</sup> is the CO stretching vibration. One can also very clearly see the first three  $t_{1u}$  modes at prac-

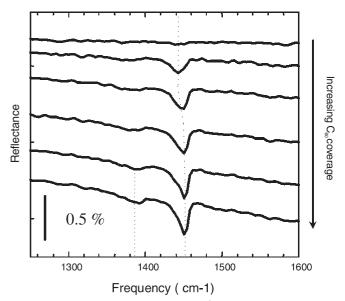
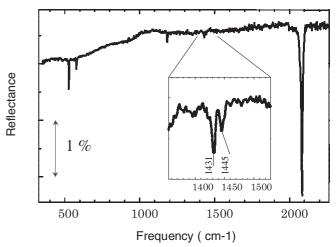


Fig. 3. Infrared spectra in the  $1250-1600\,\mathrm{cm}^{-1}$  region of a Ag (111) surface collected during  $C_{60}$  deposition at room temperature

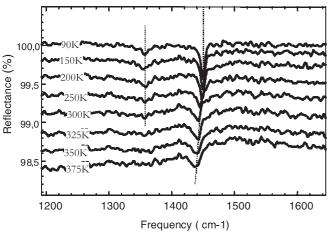


**Fig. 4.** Infrared spectra in the  $400-2250\,\mathrm{cm^{-1}}$  region of a Cu (100) surface pre-covered with 1 ML CO collected during  $C_{60}$  deposition at 90 K

tically the same frequency as for bulk  $C_{60}$ . However, in the spectral region of the  $t_{1u}(4)$  of bulk  $C_{60}$ , two bands are seen at 1431 cm<sup>-1</sup> and 1445 cm<sup>-1</sup>, respectively, but it is the former which is more intense. The breaking of symmetry due to the presence of the surface, therefore, induces the appearance of a new infrared-active mode, as already seen previously for C<sub>60</sub> adsorption on H-terminated Si (111) [28] and at the surface of  $C_{60}$  and fulleride crystals [29, 30]. This new mode has been attributed [30, 31] to the Ag(2) Raman mode, and this assignment has also been adopted in a combined IR and SFG (sum frequency generation) study [11] of C<sub>60</sub>/Ag. From the comparison of Figs. 1 and 4 one deduces that the visibility of all C<sub>60</sub>-induced IR bands does not depend on instrumental sensitivity, and that in the case of chemisorbed fullerenes on metals the intensity of the t<sub>1u</sub> modes is depressed while that of the Ag(2) is enhanced due to electron-phonon coupling effects similar to those observed in fullerides [2, 31]. We note that the intensity enhancement of the Ag(2) mode for  $C_{60}/Ag$ has been attributed to the dynamical charge transfer between

the metallic substrate and the  $C_{60}$  molecule [31] but we believe, as will be clear in the following, that this mechanism does not explain the phenomenon.

We have also studied the temperature dependence of the IR spectra as shown in Fig. 5. The top-left panel shows the evolution of the spectral region between 1200 and 1650 cm<sup>-1</sup> when a C<sub>60</sub> monolayer deposited on Cu (100) at 90 K is heated stepwise to 375 K. While the  $t_{1u}(4)$  mode stays at constant frequency but decreases in intensity, the Ag(2) mode is seen to shift from 1451 cm<sup>-1</sup> to 1440 cm<sup>-1</sup> and to broaden considerably. The detailed frequency behavior of the Ag(2) mode is plotted in the top-left panel of Fig. 6: if one performs repeated heating-cooling cycles between 90 K and 600 K the Ag(2) mode does not shift back to the initial frequency after the first heating. This indicates that an irreversible phase transition to a different adsorption state has occurred. Similar phase transitions have been observed for C<sub>60</sub> on other metal surfaces and are a signature of the existence of activation barriers between metastable and stable adsorption geometries, sometimes accompanied by reconstruction of the substrate [32]. However, the reversible part of the frequency shift and the broadening of the Ag(2) band are typical of an anharmonic coupling to a low-frequency vibrational mode, also called dephasing, which becomes more and more important



**Fig. 5.** Temperature dependence of the IR spectra in the region between 1200 and  $1650\,\mathrm{cm^{-1}}$  of 1 ML  $C_{60}/Cu\,(100)$  deposited at 90 K and heated stepwise to 375 K

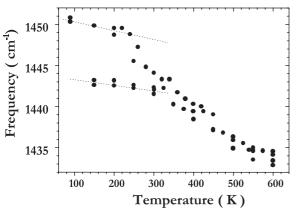


Fig. 6. Frequency behavior of the Ag(2) mode of 1 ML  $C_{60}/Cu$  (100) during repeated heating–cooling cycles between 90 K and 600 K

as the coupled mode becomes populated with increasing temperature. The theory for this mechanism is a phase relaxation exchange model [14] and allows one to extract the frequency,  $\omega_0 \approx 45\text{--}75~\text{cm}^{-1}$ , of the coupled mode from the temperature dependence of the FWHM of the Ag(2) band which is plotted in Fig. 7. Given that this value of  $\omega_0$  is too low to be attributed to an internal mode of the fullerene, we suggest that it might be the frustrated translation of the  $C_{60}$  on the metal surface which couples to the Ag(2) mode [33].

In order to gain further insight into the dependence of the IR spectra on charge transfer we have also studied the case when more electronic charge is made available to the chemisorbed  $C_{60}$  molecules through co-adsorbed potassium, as illustrated in Fig. 8, 9 and 10. When K dosing has been performed on top of a room-temperature-deposited 1 ML  $C_{60}/Cu$  (100) as shown in Fig. 8, the Ag(2) band is seen to shift to lower frequency while its line width decreases. At K coverages very close to saturation the Ag(2) band decreases drastically in intensity while still red-shifting, while the  $T_{1u}(4)$  mode suddenly increases in intensity. This behavior is very similar to what has been observed in analogous experiments [11] where K was dosed on 1 ML  $C_{60}/Ag$  (111). Given that highest charge transfer obtainable by this doping procedure on Au and Ag surfaces [8, 10, 11] has been found

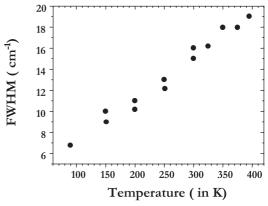
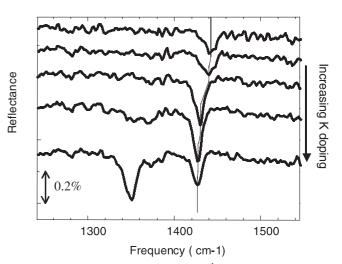


Fig. 7. FWHM of the Ag(2) mode of 1 ML  $C_{60}/Cu\,(100)$  during the reversible part of the heating–cooling cycles

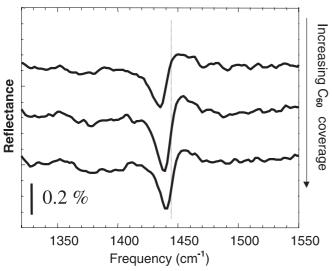


**Fig. 8.** Infrared spectra in the 1225–1550 cm<sup>-1</sup> region collected during K dosing on top of 1 ML C<sub>60</sub>/Cu (100) deposited at room temperature

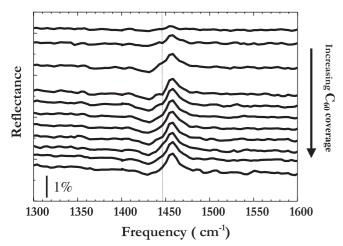
to be six electrons per fullerene, we assume that the same is true on Cu (100). This leads us to conclude that the Ag(2) mode shifts linearly by about 5 cm<sup>-1</sup> per transferred electron, while the  $T_{1u}(4)$  mode shifts in a strongly non-linear fashion as already observed by HREELS (high resolution electron energy loss spectroscopy) [10] for  $C_{60}+Cs/Au$  (110).

Based on the linear shift of the Ag(2) mode the charge transfer per fullerene molecule in 1 ML  $C_{60}$  on Au(110) and Ag(111) (see Figs. 2 and 3) can be estimated to be about one electron, while it amounts to about two electrons per fullerene in 1 ML  $C_{60}$  on Cu(100). These values are in very good agreement with previous estimates based on HREELS [9] and photoemission [23] data; however they have to be considered with caution since the anharmonic coupling of the Ag(2) mode to the frustrated translational mode might influence the shift differently on the different surfaces.

Let us now consider the potassium doping of  $C_{60}$  on Au (110) and Ag (111), which is illustrated in Figs. 9 and 10 where we plot the evolution of the IR spectra during fullerene deposition onto the metal surfaces pre-covered by 1



**Fig. 9.** Infrared spectra in the  $1300-1600\,\mathrm{cm}^{-1}$  region collected during room-temperature  $C_{60}$  dosing on top of 1 ML K/Au (110)



**Fig. 10.** Infrared spectra in the  $1325-1550\,\mathrm{cm}^{-1}$  region collected during room-temperature  $C_{60}$  dosing on top of 1 ML K/Ag (111)

ML of K. For the adsorption on gold (Fig. 9) we notice that, similar to K dosing on top of 1 ML C<sub>60</sub>/Cu (100) discussed above, the Ag(2) band is red-shifted compared to the undoped monolayer. This agrees well with previous studies which find that the charge transfer for 1 ML C<sub>60</sub>/1 ML K/Au (110) is about three electrons per fullerene [8]. However, a new characteristic of this system is that the shape of the Ag(2) band is now derivative-like, such as for an antiabsorption line. For C<sub>60</sub> adsorption on K-pre-dosed Ag (111) (Fig. 10), the anti-absorption line character even becomes the dominant feature in the spectrum. These features point again to strong electron-phonon coupling and resemble those observed previously for hydrogen adsorption on transition metals [34], issuing from the coupling of the adsorbate vibration to an adsorbate-induced electronic surface state. This coupling mechanism is characterized by an increase in broadband reflectance [35]. We therefore tried to take advantage of the possibilities of the synchrotron radiation IR set-up to measure the broadband reflectance change during K dosing onto the C<sub>60</sub> monolayers deposited on the three metal substrates, but we were unfortunately unable to record reproducible results. In fact, the problem of reproducibility arises from thermal loading of the metallic substrate by the K getter source which is kept at  $\sim 900$  K. In order to definitely confirm the coupling of the Ag(2) mode to an electronic surface state it will be necessary to record the broadband reflectance change in a set-up with adequate shielding around the getter source.

### 3 Summary

Synchrotron infrared studies of chemisorbed C<sub>60</sub> on bare and potassium-pre-covered Ag (111), Au (110) and Cu (100) have revealed several new insights into electron-phonon and purely vibrational couplings in these systems. In the case of adsorption on the clean metal substrates, the overall behavior of the IR reflectance change is dominated the diffuse-scattering mechanism of the conduction electrons of the metallic substrate by the adsorbate. This broadband reflectance change allows determination of the friction coefficient between the sliding adsorbate and the free conduction electrons of the metal driven by the external electric field associated with the incident IR photons. The values obtained for the friction coefficient give an independent confirmation of the metallic character of the fullerene-substrate bond and of the high density of adsorbate-induced states close to the Fermi level. Superimposed on the broadband reflectance are the individual vibrational bands of the adsorbate: a non-IR-active mode becomes active upon adsorption and its frequency shifts proportionally to the charge transferred from the metal to the molecule by about 5 cm<sup>-1</sup> per electron. The temperature dependence of the frequency and the width of this IR feature, attributed to the normally Raman-active Ag(2) mode, have also been followed for  $C_{60}/Cu(100)$  and were found to agree well with a weak anharmonic coupling (dephasing) between this newly IR-active mode and the frustrated translational mode of the adsorbed molecules. When potassium is co-adsorbed with C<sub>60</sub>, allowing for a larger charge transfer to the fullerene, a new electron-phonon coupling mechanism becomes dominant in determining the shape of the IR spectra, namely the coupling

of the Ag(2) mode to induced surface states. This coupling manifests itself through a derivative-like shape of the Ag(2) band. In addition, the comparison with co-adsorption experiments of CO and  $C_{60}$  allowed us to identify intensity variations of the  $t_{1u}$  infrared bands that occur both with and without co-adsorption of K, similar to those observed in alkali fullerides. Given the hybridization between  $C_{60}$  and metal substrate electronic states, these intensity variations are probably caused by the coupling to induced surface states.

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