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X-ray Free Electron Laser-Induced Synthesis of ε Iron Nitride at High Pressures

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ABSTRACT The ultrafast synthesis of ε -Fe₃N_{1+x} in a diamond-anvil cell (DAC) from Fe and N₂ 77 78 under pressure was observed using serial exposures of an X-ray free electron laser (XFEL). When 79 the sample at 5 GPa was irradiated by a pulse train separated by 443 ns, the estimated sample 80 temperature at the delay time was above 1400 K, confirmed by in-situ transformation of α - to γ iron. Ultimately, the Fe and N₂ reacted uniformly throughout the beam path to form Fe₃N_{1.33}, as 81 82 deduced from its established equation of state (EOS). We thus demonstrate that the activation 83 energy provided by intense X-ray exposures in an XFEL can be coupled with the source time 84 structure to enable exploration of time-dependence of reactions under high pressure conditions. 85

86 TOC GRAPHICS



88 **KEYWORDS** X-ray free electron laser • Diamond-anvil cell • Solid-gas reaction • Iron • Iron

89 nitride

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93 Solid-gas reactions at high pressures and in high radiation environments are important in space 94 and on (exo-) planets.¹ In such environments copious amounts of molecular nitrogen are found 95 together with common refractory materials, such as iron alloys. On Earth, better understanding 96 of the processes guiding to the evolution of atmospheric nitrogen on Earth, from its presence in a 97 protoplanetary disc² to its present-day appearance in deep mantle reservoirs, is needed.^{3,4}

In addition, recent theoretical work on solid-gas reactions such as Mg with Xe, Kr and Ar^5 ,

99 iron and nickel with xenon⁶, Kr with oxygen⁷ and the possible formation of Helium compounds⁸

100 warrants extensive exploratory synthesis efforts. The recent synthesis of a NiAr Laves phase⁹

101 and $(Xe,Si)O_2^{10}$ supports changes of electronegativity at high pressures¹¹ and the opportunity to

102 synthesize new materials with unusual properties.

High-pressure chemical synthesis is important since it permits the exploration of novel compounds at extreme conditions encountered in Earth and space science.¹² Diamond-anvil cell (DAC) experiments are suitable for such studies since the contained gas can function simultaneously as a pressure-transmitting medium (PTM) and a reactant. To trigger a chemical reaction, the energy of intense electromagnetic radiation pulses can be used.¹³ Time-resolved diffraction and spectroscopic studies in a DAC at XFEL facilities have the potential to elucidate the formation and dynamics of new materials under extreme conditions.

Here we present an experimental setup using a DAC containing an Fe foil and N_2 as a PTM and reactant to demonstrate that a reaction to form ε -Fe₃N_{1+x} can be induced by appropriate XFEL pulse-probe conditions.

In addition to cosmochemical and planetary aspects,¹⁴ iron nitrides have important technological applications due to their greater magnetization compared to iron oxides and lower manufacturing cost compared to iron alloys.^{15,16} In general, iron nitrides are synthesized by

116	reacting Fe and NH_3 above 400 °C where Fe catalyzes the decomposition of ammonia. Atomic
117	nitrogen then diffuses into bulk Fe with reaction times in the order of 100 seconds to form a
118	nitride surface layer. ¹⁷ Our experiments here show that we can synthesize uniform ϵ -Fe ₃ N _{1+x}
119	within nanoseconds without using conventional heating at an initial pressure of 5 GPa within a
120	DAC. Previously, Hasegawa and Yagi synthesized Fe ₂ N using a laser heated DAC at pressures
121	up to 10 GPa and temperatures near $1800 \pm 50 \text{K}$. ¹⁸ Additional high-pressure and high-
122	temperature syntheses of iron nitrides have led to the discovery of new Fe-N compounds such as
123	$\alpha''-Fe_{16}N_2, \alpha'-Fe_8N, \gamma'-Fe_4N, \epsilon-Fe_3N_{1+x} (-0.40 < x < 0.48), Fe_7N_3, \zeta-Fe_2N, \gamma'''-FeN, and most$
124	recently, Fe_3N_2 , FeN_2 and FeN_4 . ¹⁸⁻²⁴ On the other hand, Laniel et al. revealed that after
125	decompression of a laser-heated DAC to 5 GPa only Fe_2N was found, while above 17.7 GPa
126	NiAs-type FeN was the most stable phase found after pressure release. ²³ No indications for the
127	presence of ϵ -Fe ₃ N _{1+x} was found in these studies up to pressures of 128 GPa. However, work by
128	Clark et al. using ⁵⁷ Fe-Mössbauer studies revealed a mixture of ζ -Fe ₂ N and ϵ -Fe ₃ N _{1+x} after laser
129	heating at 1300 K below 10 GPa. ¹⁹
130	Our experiments were performed at the High Energy Density (HED) instrument at the
131	European X-ray Free Electron Laser (EuXFEL) facility in Schenefeld, Germany. ²⁵⁻²⁸ Pulses of
132	20 fs duration at 17.8 keV were generated in a train of 2-20 pulses at a 2.25 MHz repetition rate,
133	repeating at 10 Hz. The fluence incident on the DAC at 100% transmission was 187(48)
134	μ J/pulse over a spot size of 14 ± 1 μ m FWHM, focused by the compound refractive lens (CRL),
135	as measured using damage imprinting in freestanding Ta foil. At this repetition rate, XFEL
136	pulses were so close in time (443 ns) that they may be used to pump and probe <i>in-situ</i> chemical
137	reactions. ^{13,29}

138 A DAC with 500 μ m culets was used as a pre-compression chamber. A small piece of a 4 μ m 139 thick Fe foil (< 250 μm × 250 μm, 99.99% purity) was loaded into a cylindrical chamber of 140 310 µm diameter and 50 µm height made by electro-spark erosion in a pre-indented rhenium 141 gasket (Figure 1). A small spherical ruby crystal with $\sim 10 \mu m$ diameter was placed into the 142 sample chamber to determine the pressure.^{30,31} We then loaded N_2 gas as a pressure transmitting 143 medium (PTM) and a reactant using the gas loading system at the Extreme Conditions Science 144 Infrastructure (ECSI) of PETRA III. The pressure of the sample inside the DAC was determined 145 by recording the shift of the R1 emission line of ruby (precision: ± 0.1 GPa). The initial pressure 146 of the sample before XFEL irradiation was set at 5.0(1) GPa. Prior to the experiment, the sample 147 was characterized by synchrotron X-ray diffraction and optical techniques at the Extreme 148 Conditions Beamline, P02.2, at PETRA III to confirm the presence of pure N₂ and Fe in the 149 sample. 150 The horizontally polarized EuXFEL X-ray beam was directed along the central axis of the 151 DAC, through both diamond culets (Figure 1). Powder X-ray diffraction patterns were collected 152 at 10 Hz using two VAREX XRD4343 area detectors placed above and below the perpendicular 153 direction of the XFEL beam (Figure 1). Patterns comprise a superposition of all scattering during 154 each pulse train, allowing reactions to be tracked in a pump and probe fashion, while repeatedly 155 exposing the sample. We recorded the X-ray diffraction pattern using different combinations of 156 the XFEL fluence and pulses as summarized in Table 1. Dioptas software was used to convert 2-157 dimensional diffraction images to 1-dimensional diffraction patterns.³² Following the *in- situ* 158 XFEL measurements, *ex-situ* X-ray powder diffraction data of the reaction product were 159 collected at the P02.2 beamline at PETRA III using a monochromatic synchrotron X-ray beam 160 with 0.2898(1) Å wavelength and a Perkin Elmer XRD 1621 flat-panel detector. The incident X-

161	ray beam on the sample was focused to 2 μm FWHM beam using a pair of Kirkpatrick Baez
162	(KB) mirrors. The CeO2 powder standard from NIST SRM 674b was used to calibrate the
163	detector parameters using Dioptas. A 200 μ m × 200 μ m sample area was grid-scanned in 21 × 21
164	steps with 1 second exposure time per step.
165	Figure 2 shows photographic images of the sample before and after the XFEL experiments.
166	Areas irradiated by the three different XFEL pump-and-probe modes are marked with A, B, and
167	C (Table 1). While area A appears to be intact after exposure during run A (increasing pulse
168	fluence), area B and C visibly show partial and complete alteration of the sample, after 20
169	consecutive pulses and continuous exposure with 2 consecutive pulses for 11 seconds,
170	respectively. It should be noted that during exposure runs B and C, in-situ XRD patterns
171	showed the presence of both the reaction product and compressed solid N_2 (Figure 3c). The
172	observed damage on area C in Figure 2b is interpreted to result from the indirect effects of the
173	X-ray absorption and heating in the sample leading to reaction.
174	Real-time changes in the X-ray diffraction patterns as a function of XFEL irradiation are
175	summarized in Figure 3. First, we measured the XRD patterns using two consecutive pulses by
176	changing the XFEL fluence to 10%, 30%, 50%, 70%, and 100%, (run A) to check the stability of
177	the sample and the diamond anvil cell. In each two-pulse exposure, the first pulse probed the
178	sample at ambient temperature and also increased its temperature as a result of the X-ray
179	absorption. The second X-ray pulse probed the sample 443 ns after the first, measuring
180	diffraction peaks that are shifted to lower q values, where the relative shift was used to estimate
181	the temperature at this time (530 to 1424 K). The peak sample temperature was higher due to
182	cooling between exposures, ranging from 700 K to 5700 K in the Fe for 10 to 100% transmitted
183	power. ²⁹

184	As the XFEL fluence increased from 10 to 100 % transmission ($\sim 10^{10}$ to $\sim 10^{11}$ photons per
185	pulse), the diffraction peak intensities of the compressed solid N_2 and bcc-Fe (α -iron) increase as
186	expected, which demonstrates that the anvils remain intact and maintain the original compression
187	conditions (Figure 3a). At 30% transmission, a shoulder peak starts to form at the low-q side of
188	the α -iron (110) peak, indicating an XFEL-induced thermal expansion. At 50% transmission, this
189	shoulder peak grows and shifts further toward the low-q side, and above 70 % transmission,
190	high-temperature fcc-Fe (γ -iron) is observed by the appearance of its (111) peak. The
191	transformation of α - to γ -iron would require at least 900 K at 5 GPa. ³³ Based on the refined peak
192	positions and the established EOS of the iron phases, ³⁴ we estimate residual temperatures of 750
193	and 1270 K at 50 and 70% fluence (2400 and 3900 K calculated peak temperatures), respectively
194	(Figure 3a). The reaction between Fe and N_2 , however, did not occur even at the full 100%
195	transmission in run A, at which point the residual temperature is estimated to be 1424 K (5700 K
196	peak temperature). The \sim 70% cooling between the pump and the probe is consistent with
197	expectations for Fe samples. ²⁹ The pulse to pulse energy variance of the XFEL, of order 30%,
198	could lead to differing heating in shorter duration experiments, whereas in longer duration
199	studies, where reactions products appear, only the averaged heating and duration is important.
200	Thus, temperatures measured following single pulses (pump-probe) are consistent with expected
201	values including initial volumetric X-ray absorption and heat transfer between pulses ²⁹ ;
202	somewhat higher temperatures are expected after serial exposures in a MHz pulse train, due to
203	heat accumulation ²⁹ , thus the temperature conditions reported following single pulses are lower
204	bounds on those achieved after second and subsequent pulses in a train.
205	In contrast, XFEL-induced chemical reaction between Fe and N_2 is observed in the runs B and
206	C (Figure 3a). The new peaks observed both <i>in-situ</i> and <i>ex-situ</i> after exposure during run B are

207	indexed as an ε -Fe ₃ N _{1+x} phase (Figure 3b). When we exposed the sample with serial XFEL trains
208	for 11 seconds (run C), the formation of the ϵ -Fe ₃ N _{1+x} was observed from the third train by the
209	growth of its (002) and (111) peaks (Figure 3c). In the first and second trains, we observed the
210	growth of (111) and (002) Bragg reflections of fcc-Fe indicating temperatures of 1490 K and
211	1680 K, respectively. From the third train where ϵ -Fe ₃ N _{1+x} formed, the temperature was
212	maintained between 1340 K and 1780 K up to the fourteenth train where the Bragg peaks of fcc-
213	Fe disappeared. The ϵ -Fe ₃ N _{1+x} (and residual bcc-Fe) remained until the last train without any
214	systematic changes in the relative peak intensities and positions. We estimate a cumulative
215	absorbed energy of 3 mJ for the chemical reaction of Fe and N_2 at the pre-compression
216	conditions of 5 GPa.
217	The sample pressure after run C increased to 7.8(1) GPa, where the unit cell parameters of ϵ -
218	Fe_3N_{1+x} are refined to a = 4.707(1) Å and c = 4.357(1) Å in the space group P6 ₃ 22 (no. 182)
219	(Figure 3d). It is well established that the unit-cell volume of ϵ -Fe ₃ N _{1+x} is linearly proportional to
220	the increasing nitrogen content x at ambient pressure. ^{35,36} The refined unit cell volume of the
221	recovered ε -Fe ₃ N _{1+x} , e.g., 83.59(3) Å ³ at 7.8(1) GPa, would correspond to ε -Fe ₃ N _{1+x} with x = 0.33
222	at ambient conditions, assuming an EOS of $B_0 = 172$ GPa and B' = 5.7 as derived previously. ³⁶
223	We therefore conclude that the composition of the XFEL-induced synthesis product is $Fe_3N_{1.33}$.
224	To investigate the cross-sectional textures and chemical distribution of the $Fe_3N_{1.33}$ reaction
225	product, we have prepared the sample from run B using a focused Ga ion beam (FIB, 30 kV and
226	10 pA to 30 nA for surfacing) and performed scanning electron microscopy (SEM) (ZEISS
227	Crossbeam 540 at Yonsei University) and scanning transmission electron microscope imaging
228	(STEM, JEOL JEM-F200 combined with energy-dispersive X-ray spectroscopy) (Figure 4). The
229	sample was mounted on a Cu grid and measured in the annular dark-field mode. A two-

230 dimensional elemental distribution of the cross-sectioned sample shown in Figure 4c and 4d 231 reveal that over the 4 µm thickness of the original Fe-foil, the distribution of Fe and N appears to 232 be uniform with regularly spaced holes representing degassed N_2 after the recovery of the 233 sample. STEM data corroborate the results of the *in-situ* and *ex-situ* XRD by showing the 234 distribution of the ABAB type stacking and distance between the Fe atoms, as expected for the ε -235 Fe_3N_{1+x} structure (Figure 4b). It is remarkable that such a homogeneous composition is obtained 236 after such a short cumulative heating time as atomic nitrogen diffusion is in general a much 237 slower process.38

238 In conclusion, our work demonstrates that the required activation energy for chemical 239 reactions controlled by the XFEL radiation is an important experimental parameter. Despite only 240 a limited number of exposures due to constraints and availability of in- and ex-situ 241 characterizations we are confident that the XFEL parameters of our studies are well within 242 parameters that are reproducible and in agreement with the known behavior of the Fe-N₂ system 243 under high pressure and temperature. In contrast the kinetics of the XFEL-induced synthesis of 244 ϵ -Fe₃N_{1,33} are noteworthy and unprecedented. We found a remarkably homogeneous reaction 245 product after the ultrafast reaction between Fe and nitrogen in a DAC at pressures above 5 GPa 246 and temperatures exceeding 1400 K as pumped and probed by consecutive XFEL pulses 247 separated by 443 ns. Following chemical reactions between gas and solid at high pressures and 248 temperatures in a DAC using a tailored pump-probe setup at an XFEL open up a new parameter 249 space for the exploration of new materials forming on fast timescales at high pressure.

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- **Figure 1.** Experimental setup of the XFEL pump-and-probe using the DAC. Schematic diagram
- 257 of the experimental setup at HED (High Energy Density science) Beamline at European XFEL.



- Figure 2. Photos of the sample inside DAC (a) before and (b) after a series of XFEL exposures as
 described in Table 1. The red line in (b) has been cross-sectioned by FIB for SEM and STEM
 imaging (see Figure 4).



Figure 3. The changes of X-ray diffraction patterns of iron and nitrogen during and after in-situ chemical reaction by XFEL pump-and-probe. (a) Changes in diffraction patterns of the Fe foil under N₂ PTM pre-compressed to 5 GPa in a DAC. The percentages in run A indicate the transmission (fluence) of the XFEL. At 100% XFEL transmission, run B contains 20 consecutive pulses while run C has consecutive pulses over 11 seconds. (b) Synchrotron X-ray diffraction patterns collected at beamline P02.2 at PETRA III after run B, compared to the XFEL data measured right after run B (top pattern). After the XFEL experiments, the sample pressure has changed from 5.0(1) GPa to 7.8(1) GPa. (c) Changes in the diffraction patterns of the Fe foil during run C. (d) Profile fitting of the ex-situ XRD pattern measured after run B as shown in (b). ϵ -Fe₃N_{1+x} (P6₃22) and δ -Nitrogen (Pm3n) were fitted to an agreement index of R_{wp} = 1%. Observed data are shown in black crosses and the calculated pattern in a red line. Tick-marks under the pattern indicate the (hkl) reflection positions of the composing phases (green: ε -Fe₃N_{1+x}, blue: δ -Nitrogen). The red asterisk is an unidentified shoulder peak.



Figure 4. Electron microscope images and elemental mapping of the recovered sample. (a) A cross-sectional SEM image after exposure run B (the red line in Figure 2b). (b) The positions of the red, orange, and green area in the SEM image are shown by the STEM images. (c) Energy dispersive spectroscopy (EDS) elemental mapping of iron (left), nitrogen (middle), and their overlay (right) at the middle of the recovered sample (the green area in the SEM image).

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	Experimental run	XFEL pump and probe condition					
	A		With increasing XFEL transmission from 10 to 100 % (2 pulses per train)				
	В	2.25 MHz 0.19 mJ/pulse 17.8 keV XFEL beam diameter:	20 consecutive pulses at 100% transmission (20 pulses in one train)				
	С	14 $\mu m \varnothing$	2 consecutive pulses per train at 100% transmission for 11 s (220 exposures)				
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Table 1. Pump and probe conditions used in our experiment.

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- 338 Notes
- 339 The authors declare no competing financial interests.

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