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## Heisenberg spins on an anisotropic triangular lattice: PdCrO<sub>2</sub> under uniaxial stress

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Pascal Manuel<sup>4</sup> , Fabio Orlandi<sup>4</sup> , Dmitry D Khalyavin<sup>4</sup> , Andrew P Mackenzie<sup>1,5</sup>   
and Clifford W Hicks<sup>1,6,\*</sup> <sup>1</sup> Max Planck Institute for Chemical Physics of Solids, Nöthnitzer Str. 40, 01187 Dresden, Germany<sup>2</sup> Stewart Blusson Quantum Matter Institute, The University of British Columbia, Vancouver, V6T 1Z4 Canada<sup>3</sup> H. H. Wills Physics Laboratory, University of Bristol, Bristol, BS8 1TL, United Kingdom<sup>4</sup> ISIS Facility, Rutherford Appleton Laboratory, Chilton, Didcot, OX11 0QX, United Kingdom<sup>5</sup> Scottish Universities Physics Alliance (SUPA), School of Physics and Astronomy, University of St. Andrews, St. Andrews KY16 9SS, United Kingdom<sup>6</sup> School of Physics and Astronomy, University of Birmingham, Birmingham B15 2TT, United Kingdom

\* Author to whom any correspondence should be addressed.

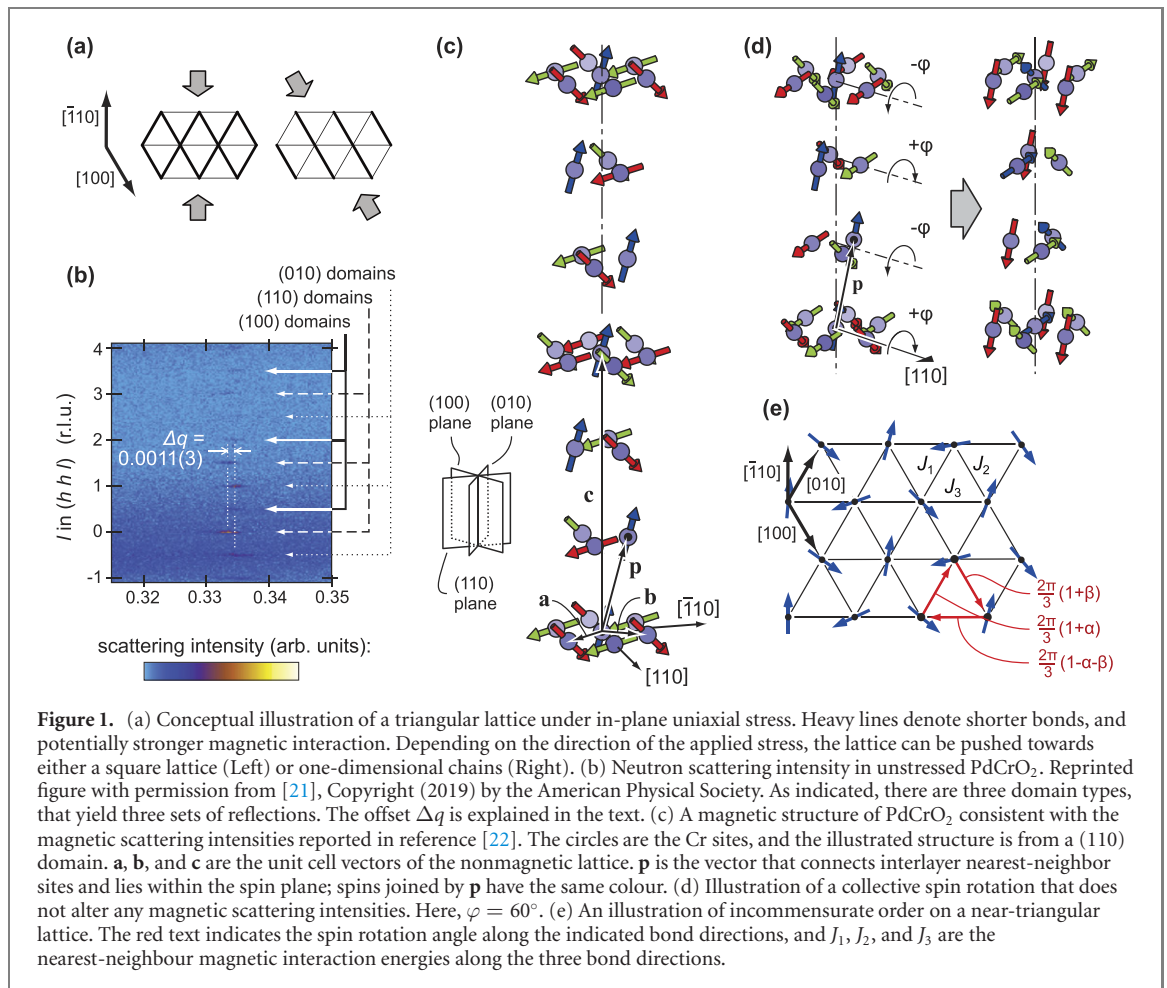
E-mail: [C.Hicks.1@bham.ac.uk](mailto:C.Hicks.1@bham.ac.uk)**Keywords:** triangular lattice, delafossite, uniaxial stress, antiferromagnetism, neutron scattering

## Abstract

When Heisenberg spins interact antiferromagnetically on a triangular lattice and nearest-neighbor interactions dominate, the ground state is 120° antiferromagnetism. In this work, we probe the response of this state to lifting the triangular symmetry, through investigation of the triangular antiferromagnet PdCrO<sub>2</sub> under uniaxial stress by neutron diffraction and resistivity measurements. The periodicity of the magnetic order is found to change rapidly with applied stress; the rate of change indicates that the magnetic anisotropy is roughly forty times the stress-induced bond length anisotropy. At low stress, the incommensuration period becomes extremely long, on the order of 1000 lattice spacings; no locking of the magnetism to commensurate periodicity is detected. Separately, the magnetic structure is found to undergo a first-order transition at a compressive stress of ~0.4 GPa, at which the interlayer ordering switches from a double- to a single-*q* structure.

Antiferromagnetic interactions are frustrated on a triangular lattice. When the spins are Heisenberg spins and nearest-neighbour interactions dominate, balancing interactions across the three bond directions leads to 120° order, in which the spin orientation rotates by  $2\pi/3$  from site to site [1, 2]. Lifting the triangular symmetry may cause the magnetism to become incommensurate [3, 4]. However, propagation vectors tend to lock to commensurate values, because for commensurate orders the spins can find a single optimum configuration, while for incommensurate orders the state energy must be averaged over a  $2\pi$  rotation of the spin orientation. A symmetry-breaking field may therefore need to exceed a threshold strength for the magnetism to become incommensurate. Such ‘lock-in’ to commensurate periodicities has been observed in several non-triangular systems, such as, for example, CaFe<sub>4</sub>As<sub>3</sub> [5, 6], Ba(Fe<sub>1-x</sub>Co<sub>x</sub>)<sub>2</sub>As<sub>2</sub> [7], and lightly-doped Cr [8]; in all of these cases, the transition between commensurate and incommensurate states is first-order. Independently of this potential commensurate-to-incommensurate transition, the possibility of tuning a triangular lattice through uniaxial stress has been considered before, and there are two possible directions, illustrated in figure 1(a) [3, 4, 9, 10]. By applying stress along a  $\langle\bar{1}10\rangle$  direction, the atomic positions are pushed towards a square lattice, and the magnetic interactions may similarly evolve towards a square lattice with weak diagonal bonds. Alternatively, by applying stress along a  $\langle 100\rangle$  direction, the atomic positions are pushed in the direction of 1D chains.

Here, we study the triangular antiferromagnet PdCrO<sub>2</sub> under tunable uniaxial stress applied along the  $[\bar{1}10]$  lattice direction. We will show that the magnetic interactions, like the atomic positions, evolve towards those of a square lattice. In other words, the interaction energy increases as bond length is reduced,



which is intuitive. In this compound, insulating CrO<sub>2</sub> layers are separated by highly conducting Pd sheets. The Cr configuration is  $3d^3$ , and due to strong Hund's coupling the three electron spins align to form robust  $S = 3/2$  moments; the CrO<sub>2</sub> layers are Mott insulating [11–13]. Coupling between the Cr and Pd layers means that the magnetic structure in the Cr layers affects electronic transport properties. A downturn in resistivity shows that short-range correlation between Cr spins appears starting at around room temperature [14–16]. By  $\sim 100$  K the correlation length becomes long enough to measurably influence the Hall and Nernst effects [17], and to yield substantial magnon drag in the thermopower [18]. By 40 K the in-plane correlation area reaches  $\sim 1000$  sites, without interlayer correlation [19, 20]. Then, at  $T_N = 37.5$  K, the layers lock together in a transition that appears to be first-order, and the order becomes long-range correlated in three dimensions [19].

The magnetic order spontaneously lifts the three-fold rotational symmetry of the non-magnetic lattice [16, 21]. The delafossite structure is rhombohedral: Cr sites in each layer are equidistant from three Cr sites in adjacent layers, so interlayer interactions balance and the layers cannot couple [23]. The rotational symmetry breaking provides a preferred orientation that allows coupling; the related delafossite compound CuCrO<sub>2</sub> also shows this rotational symmetry breaking [24, 25]. The spin plane for PdCrO<sub>2</sub> and CuCrO<sub>2</sub> contains the *c* axis, and the azimuthal angle takes one of three symmetrically equivalent possible orientations [21, 22].

We report results of neutron scattering and electrical resistivity measurements. The neutron diffraction data show that the rate of change of the magnetic structure with applied stress is large: the magnetic anisotropy is roughly forty times the stress-induced bond length anisotropy. The neutron data also reveal a stress-driven first-order transition in the interlayer ordering, which shows that the interlayer ordering probably reflects a delicate balance of interactions. This transition is apparent in the resistivity data. The resistivity data also reveal structure in the transition at  $T_N$ , which encourage further experimentation.

## 1. Background

We begin with a description of the magnetic structure of PdCrO<sub>2</sub> as it is currently understood. Shown in figure 1(b) is neutron scattering data from nominally unstressed PdCrO<sub>2</sub>, reported in reference [21]. As

indicated in the figure, there are three sets of reflections, originating from three domain types, which we label by their spin plane. In figure 1(c), we illustrate a coplanar magnetic structure that gives a good fit to the scattering intensities reported in reference [22]. An anomalous Hall effect has been interpreted as evidence for a non-coplanar magnetic structure [26], and an observation of nonreciprocal transport could indicate that inversion symmetry is broken [27]. However, the quality of the fit of the co-planar structure in figure 1(c) is very good [21]. The Hall effect might be influenced by magnetic breakdown [28].

In the structure illustrated in figure 1(c), the spin plane is the (110) plane. The (100) and (010) planes are symmetrically equivalent, resulting in the three domain types. To understand the magnetic structure, it is useful to define a vector  $\mathbf{p}$ , that connects interlayer nearest-neighbour sites and lies within the spin plane. Between spins that are connected by  $\mathbf{p}$ , the components of the spins that are parallel to  $\mathbf{p}$  are aligned ferromagnetically, and those perpendicular to  $\mathbf{p}$  are aligned antiferromagnetically. This combination of ferromagnetic and antiferromagnetic alignment means that the helicity alternates from layer to layer. It means that neutron reflections from a single domain are spaced along the  $l$  axis by  $\Delta l = \frac{3}{2}$  r.l.u. (defined using a three-layer unit cell), rather than the  $\Delta l = 3$  that would be obtained for a fully ferromagnetic or antiferromagnetic interlayer ordering. For  $h = k = \frac{1}{3}$ , (110) domains give reflections at  $l = \frac{3}{2}n$ , (100) domains at  $\frac{1}{2} + \frac{3}{2}n$ , and (010) domains at  $1 + \frac{3}{2}n$ , where  $n$  is an integer [21].

In figure 1(d) we illustrate a degree of freedom that is unconstrained by data so far: rotation of spins in alternating planes by angles  $+\phi$  and  $-\phi$  does not alter any neutron scattering intensities, for any value of  $\phi$ . Incommensuration can therefore emerge as a spatial gradient in  $\phi$ . In figure 1(e) we illustrate a generic construction of incommensurate order on a triangular lattice. If the spin rotation angle upon translation from site to site along the [010] direction is  $\frac{2\pi}{3}(1 + \alpha)$ , and that along [100]  $\frac{2\pi}{3}(1 + \beta)$ , then to linear order in  $\alpha$  and  $\beta$  the reflection at  $[h, k, l] = [\frac{1}{3}, \frac{1}{3}, l]$  moves to  $[\frac{1}{3}(1 + \alpha), \frac{1}{3}(1 + \beta), l]$ . In the illustration, the nearest-neighbor interaction strengths are  $J_1 > J_2 > J_3$ , which yields  $\alpha > 0$  and  $\alpha + \beta > 0$ .

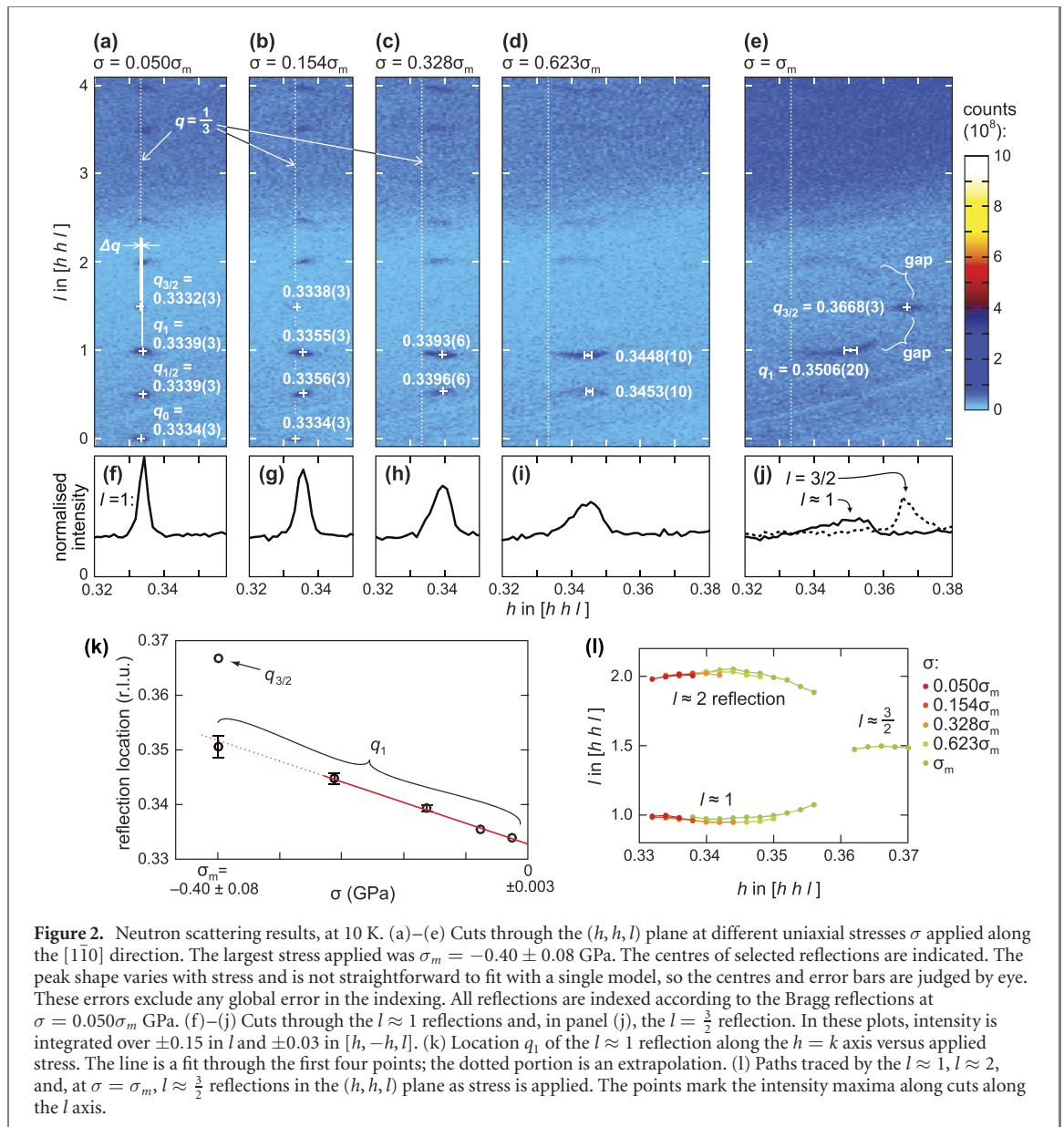
## 2. Results: neutron scattering

A sample of cross section  $1.37 \times 0.21 \text{ mm}^2$  was mounted into piezoelectric-based uniaxial stress apparatus adapted for use in neutron scattering and muon spin rotation measurements, described in reference [29]. The central, exposed portion of the sample was 1.3 mm in length. The end portions of the sample were embedded in epoxy for transmission of force to the sample. These portions were screened from the neutron beam with cadmium foil. Growth of single crystals of PdCrO<sub>2</sub> more than 1–2 mm in extent remains a challenge, and there were voids in the sample that will have introduced stress inhomogeneity.

Measurements were performed using the WISH diffractometer at the ISIS Pulsed Neutron and Muon Source [30]. Compressive stress was applied along the  $[\bar{1}10]$  direction. Strain gauges were affixed to the sample holder to measure the applied stress, and the cell incorporates a mechanism for finding the zero-stress reading. We estimate an error of  $\pm 3$  MPa on this zero-stress reading. The largest compressive stress we applied was  $\sigma_m = -0.40 \pm 0.08$  GPa. (Negative values denote compression.) Measurements were performed at 10 K, and temperature was kept constant, without thermal cycling, during changes in the applied force. In analysis of the neutron data, we index the reflections at all strains to the Bragg peaks at the smallest stress applied,  $0.050\sigma_m$ . In other words, there is no correction for lattice strain, which we show later to be a reasonable approximation.

Cuts through the  $(h, h, l)$  plane (which is perpendicular to the stress axis) under different applied stresses  $\sigma$  are shown in figures 2(a)–(e). At  $\sigma = 0.050\sigma_m$ , reflections from all three domain types are present. The reflections from the (100) and (010) domains are shifted rightward relative to those from the (110) domains by  $\Delta q = 0.0006 \pm 0.0003$  r.l.u. Such an offset can also be seen in the data of nominally unstressed PdCrO<sub>2</sub> in figure 1(b), from reference [21], where  $\Delta q = 0.0011 \pm 0.0003$  r.l.u.

As stress is applied the (110) reflections fade and disappear, which is in agreement with previous results [21]: in-plane uniaxial compression favours domains whose spin planes are more perpendicular to the stress axis. The (100) and (010) reflections shift further rightward. As  $|\sigma|$  becomes large the reflections also become streaked, which is most likely a consequence of strain inhomogeneity. Strain homogeneity should also make the nuclear Bragg peaks streaked, but we find that  $q$  shifts rapidly with applied strain, so that streaking is far more pronounced in the magnetic peaks.  $q_1$ , the location of the  $l = 1$  reflection, versus applied stress  $\sigma$  is plotted in figure 2(k). Within resolution, the dependence is linear, and no locking to the commensurate value  $1/3$  is resolved. Between  $\sigma = 0$  and  $-0.4$  GPa,  $q_1$  shifts by 5%. For a Young's modulus of  $\sim 100$  GPa, the longitudinal strain under  $\sigma = -0.4$  GPa would be around  $-0.4\%$ , and the Poisson's ratio expansion in the scattering plane perhaps half of this, or  $+0.2\%$ . In other words, the stress-driven shift in  $q_1$  exceeds plausible strain in the sample by an order of magnitude, and we are justified in not correcting the indexing for lattice strain.



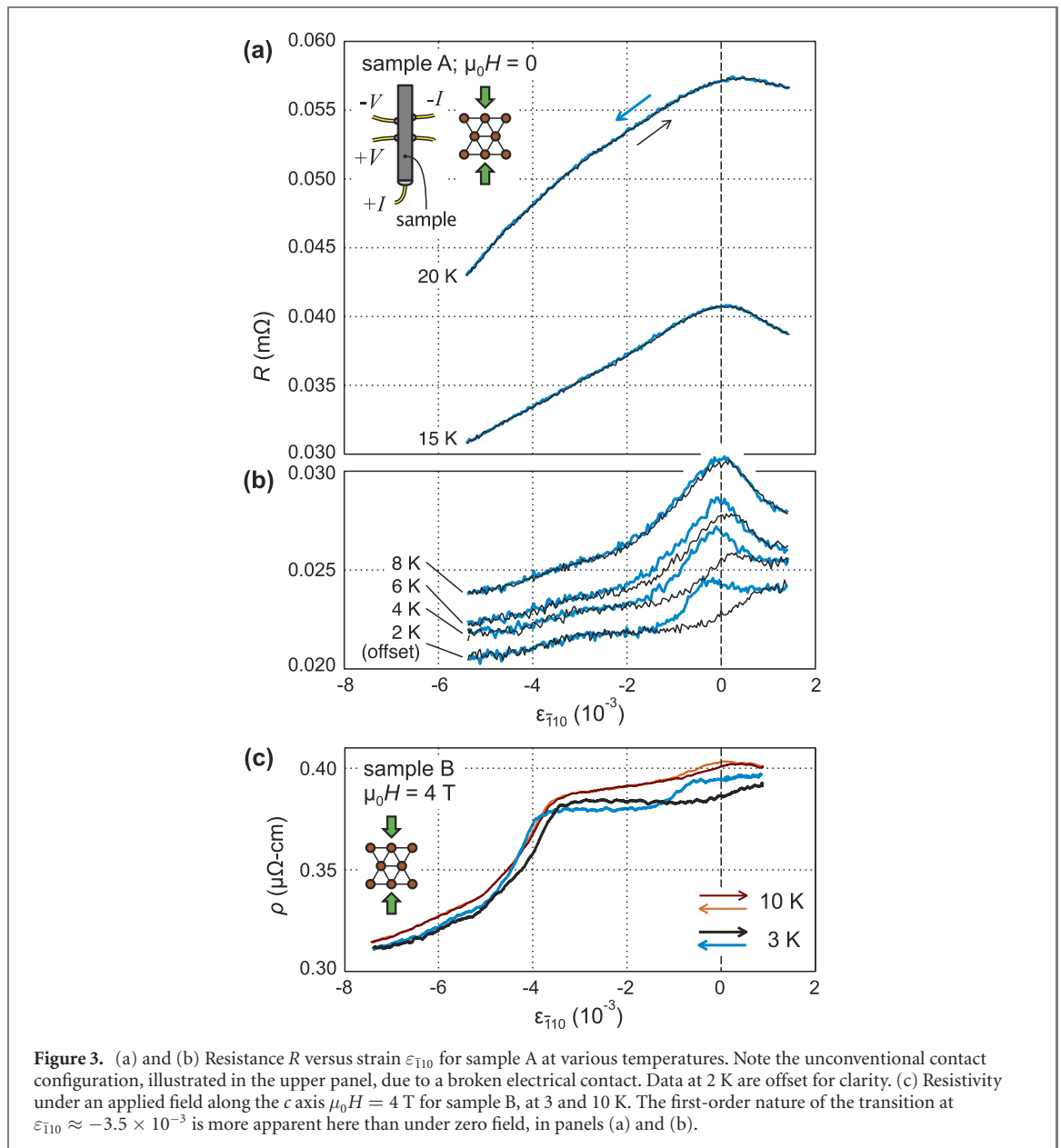
**Figure 2.** Neutron scattering results, at 10 K. (a)–(e) Cuts through the  $(h, h, l)$  plane at different uniaxial stresses  $\sigma$  applied along the  $[110]$  direction. The largest stress applied was  $\sigma_m = -0.40 \pm 0.08$  GPa. The centres of selected reflections are indicated. The peak shape varies with stress and is not straightforward to fit with a single model, so the centres and error bars are judged by eye. These errors exclude any global error in the indexing. All reflections are indexed according to the Bragg reflections at  $\sigma = 0.050\sigma_m$  GPa. (f)–(j) Cuts through the  $l \approx 1$  reflections and, in panel (j), the  $l \approx \frac{3}{2}$  reflection. In these plots, intensity is integrated over  $\pm 0.15$  in  $l$  and  $\pm 0.03$  in  $[h, -h, l]$ . (k) Location  $q_1$  of the  $l \approx 1$  reflection along the  $h = k$  axis versus applied stress. The line is a fit through the first four points; the dotted portion is an extrapolation. (l) Paths traced by the  $l \approx 1$ ,  $l \approx 2$ , and, at  $\sigma = \sigma_m$ ,  $l \approx \frac{3}{2}$  reflections in the  $(h, h, l)$  plane as stress is applied. The points mark the intensity maxima along cuts along the  $l$  axis.

Because the spin planes of the (100) and (010) domains are neither aligned with nor perpendicular to the stress axis, there is no symmetry requirement for the reflections from these domains to have  $h = k$ ; referring to figure 1(e), we do not require  $\alpha = \beta$ . In the appendix cuts through the  $(h, k, l)$  plane are shown, which show that  $h$  and  $k$  of the reflections nevertheless remain approximately equal as stress is applied.

The fact that  $\Delta q$  is non-zero at the smallest applied stress, and also in the nominally unstressed sample shown in figure 1(b), indicates that the magnetic order is incommensurate even at very low applied stress, with an incommensuration period of around 1000 lattice spacings. To resolve such long-period incommensuration, it is useful to be able to compare reflections from different domains in this way, because it means that we do not rely solely on the global indexing of the reflections. A nonzero  $\Delta q$  was visible in the data of reference [21], but was not commented upon: the observation that  $\Delta q$  grows under stress gives confidence that it is a real feature of the data.

The paths traced by the  $l = 1$  and  $l = 2$  reflections in the  $(h, h, l)$  plane as stress is applied are shown in figure 2(l). It is apparent that as stress is applied the magnetism also becomes incommensurate along the  $c$  axis. The reflection that was at  $l = 1$  shifts initially to an incommensurate position at  $l < 1$ , and then reverses direction to  $l > 1$  as  $|\sigma|$  increases further. The  $l = 2$  reflection follows an opposite stress dependence, shifting initially to  $l > 2$ , then  $l < 2$ .

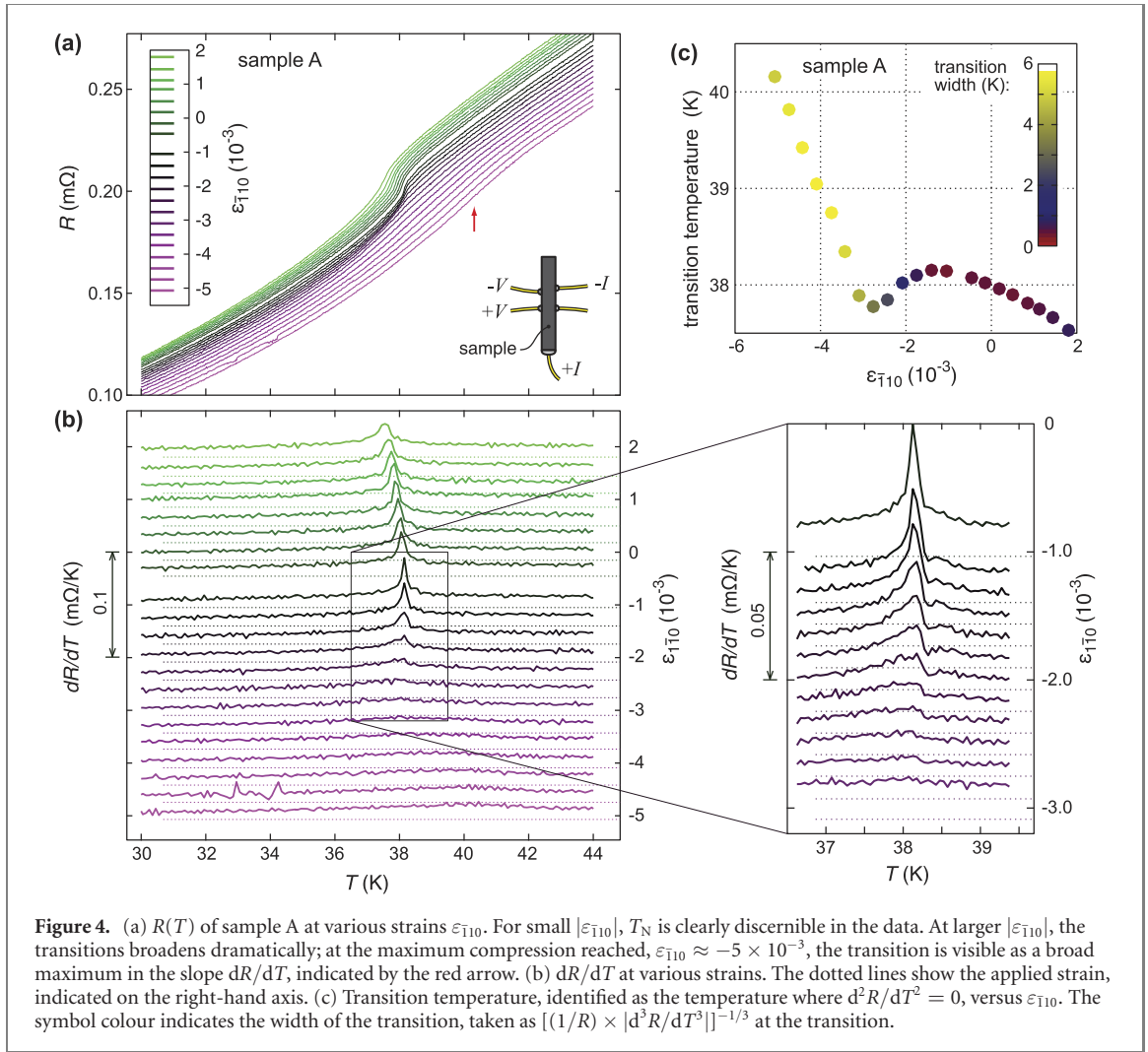
We do not detect any scattering intensity for  $h$  between 0.357 and 0.367, where there is then a strong reflection at  $l = \frac{3}{2}$ . This gap is highlighted in figure 2(e). We interpret these data as showing a first-order transition in the magnetic structure. In this interpretation, the reflections at  $l \approx 1$  and  $l \approx 2$  in the small- $|\sigma|$



structure shift continuously to larger  $q$  as  $|\sigma|$  is increased, eventually reaching  $q = 0.357$ . But magnetic order is unstable for  $0.357 < q < 0.367$ . The low- and high-stress phases can coexist in equilibrium due to strain inhomogeneity. The cuts shown in figure 2(j) support this interpretation: the  $l \approx 1$  reflection has a tail extending to  $h < 0.357$ , and the  $l = \frac{3}{2}$  has a tail extending to  $h > 0.367$ , but scattering intensity is cut off sharply between these two values.

The reflection  $(0.367, 0.367, \frac{3}{2})$  is not accompanied by reflections at  $l = 0$  and  $l = 3$ , showing that in the high- $|\sigma|$  phase the interlayer correlation is strictly antiferromagnetic. Also, from the fact that the high- $|\sigma|$  reflection is at  $l = \frac{3}{2}$ , it appears that strong compression along  $[\bar{1}10]$  favours (110) domains, whereas weak compression favoured (100) and (010) domains. The way that the reflections at  $l \approx 1$  and  $l \approx 2$  curve inwards towards  $l = \frac{3}{2}$  as  $|\sigma|$  increases suggests a rotation of the spin planes of the (100) and (010) domains towards the (110) plane as stress is applied. However, the data here are not sufficient for accurate refinement of the magnetic structure under stress.

There is certain to be some structural distortion associated with this first-order magnetic transition, and taking this distortion into account will shift the positions of the reflections. It is unlikely to have a substantial effect on the appearance of the data, however: the jump from  $h = 0.357$  to  $0.367$  is an increase of 3%, and any structural distortion is certain to be much smaller than that.



### 3. Results: resistivity

We now show that this transition in interlayer ordering has an observable effect on in-plane resistivity. For resistivity measurements, a stress cell was used that had a sensor of the displacement but not the force applied to the sample, so strain rather than stress is the measured control variable. The applied strain is taken as the applied displacement divided by the exposed sample length, and then multiplied by a factor of 0.8 as an estimate for the loss due to elastic deformation of the epoxy.

Strain is applied along the length of the samples, and resistivity is measured along the same axis. Two samples were measured, labelled A and B. Some data on sample A were also shown in reference [21], where it was labelled sample 1. In figures 3(a) and (b), we show the resistance versus  $\varepsilon_{110}$  for sample A at temperatures up to 20 K. (Resistance rather than resistivity is shown because a contact to the sample broke—this sample was small—and so an unconventional contact configuration was used.) For temperatures below  $\sim 8$  K, the dominant feature in the data is a first-order transition that was identified in reference [21] as reorientation of domains as the applied stress switches between compressive and tensile, and we therefore fix the neutral strain point  $\varepsilon_{110} = 0$  at the centre of this transition.

For temperatures below  $\sim 8$  K, a feature is also discernible at  $\varepsilon_{110} \approx -3.5 \times 10^{-3}$ . In the zero-field data from sample A it is subtle, but under a field it becomes much more prominent: see the data from sample B in figure 3(c). It also becomes clear that it is a first-order transition: there is definite hysteresis at 3 K, that mostly closes by 10 K. This is almost certainly the transition in interlayer ordering that is seen in the neutron data. In the neutron data, the transition occurs at a stress of  $\sim -0.4$  GPa, and if this is equated with the transition strain  $\varepsilon_{110} = -3.5 \times 10^{-3}$  seen in resistivity a Young's modulus of  $\sim 100$  GPa is implied, which is a typical value for inorganic solids.

Resistivity data also show that strain has a nontrivial effect on the Néel transition.  $R(T)$  of sample A at  $T \sim T_N$  and at various fixed strains is shown in figure 4(a). At small applied strains, the Néel transition is apparent as a sharp drop in resistivity. Although no hysteresis was resolved, the step-like nature of the

transition strongly suggests that it is first-order. As the sample is compressed, there is initially little change in this feature, but then it fades away. Under strong compression, a much broader feature is apparent in the data: a broad maximum in the slope  $dR/dT$ , indicated by the red arrow in figure 4(a) for the largest applied compression. Identifying this feature as the Néel transition, in figure 4(c) we show  $T_N$  versus strain, with the data points coloured by a measure of transition width. The crossover from a first-order to a broad transition is associated with a change in the stress dependence of  $T_N$ : from a concave-downward, parabolic dependence to a steep linear dependence. Under tensile stress, the Néel transition began to broaden, suggesting that it would evolve similarly under strong tensile strain as under strong compressive strain. However, the sample then fractured as we attempted to tension it further.

#### 4. Discussion

We begin our discussion with consideration of the rate of change of the magnetic structure with strain. The fact that  $q$  increases with compression along  $[\bar{1}10]$  means that the magnetic lattice, like the real-space lattice, is tuned towards a square net: referring to figure 1(e),  $J_3$  shrinks relative to  $J_1$  and  $J_2$ . The largest stress we achieve,  $\sigma_m = -0.4$  GPa, is modest. If the Young's modulus is  $\sim 100$  GPa, bonds along the stress axis would be compressed by  $\sim 0.4\%$ , and bonds perpendicular to the stress axis stretched by  $\sim 0.2\%$ , yielding a bond length anisotropy of  $\sim 0.6\%$ . The induced magnetic anisotropy appears to be much larger than this. At  $\sigma = -0.4$  GPa,  $q = 0.367$  for portions of the sample in the high- $|\sigma|$  state and up to 0.357 for portions in the low- $|\sigma|$  state. If interactions beyond nearest-neighbor are neglected, and  $J_1$  and  $J_2$  are taken to be equal, then  $h = 0.367$  is obtained for  $J_3/J_1 = 0.746$ , and 0.357 for  $J_3/J_1 = 0.803$  [3, 4]—the stress-induced anisotropy in nearest-neighbor interactions is about forty times the bond length anisotropy. Higher-order interactions will alter these values, but the point is that magnetic anisotropy grows very rapidly and so very substantial tuning of the magnetic structure is technically possible. For classical spins, spiral order is expected to undergo transition to linear antiferromagnetism at  $J_3/J_1 = 0.5$  [4], and quantum corrections are predicted to shift this transition to higher  $J_3/J_1$  [9, 10]. Extrapolation from the data here suggests that this transition should occur at  $\sigma \sim -1$  GPa, which is an achievable stress [31].

We did not reach such strong compression here, however, and the observed first-order transition, the change in interlayer ordering, was unexpected. It suggests that interlayer ordering reflects a balance of interactions and so is somewhat delicate, an idea supported by the fact that other delafossite compounds with  $\text{CrO}_2$  layers have different interlayer orderings: in  $\text{LiCrO}_2$ , the helicity alternates from layer to layer [32], as in unstressed  $\text{PdCrO}_2$ , but in  $\text{CuCrO}_2$  and  $\text{AgCrO}_2$  the interlayer correlation is ferromagnetic [24, 33, 34], while, as reported above, the interlayer ordering in  $\text{PdCrO}_2$  under in-plane uniaxial stress is antiferromagnetic.

We discuss whether the magnetism of unstressed  $\text{PdCrO}_2$  could be incommensurate. Unstressed  $\text{AgCrO}_2$  and  $\text{CuCrO}_2$  are both incommensurate, with  $q = 0.327$  for the former and 0.329 for the latter [24, 34, 35].  $q$  in unstressed  $\text{PdCrO}_2$  appears to be much closer to  $\frac{1}{3}$  than in these materials. The spontaneous rotational symmetry breaking in unstressed  $\text{PdCrO}_2$  will have a lattice distortion associated with it, and it is possible that this distortion itself is sufficient to make the order incommensurate. Due to the possibility of residual thermal stresses on the MPa level, both here and in [21], we cannot definitively resolve whether the magnetic order would be incommensurate under truly zero stress. Doing so could be a subtle measurement challenge: with, for example, tetragonal to orthorhombic lattice distortions, a configuration of twin boundaries, parallel lines, is possible that allows each domain to fully relax elastically, but with three domain types such an arrangement of domain walls is not possible. Therefore, for measurement under truly zero stress, the sample would need to be polarised into a single domain.

Regardless of whether or not the magnetic order of unstressed  $\text{PdCrO}_2$  is commensurate, the fact that the incommensuration period can reach 1000 lattice spacings shows that the net spin-lattice coupling in  $\text{PdCrO}_2$  is extraordinarily weak. It may be weak enough that classical dipole–dipole interactions determine the orientation of the spin plane [36]. The spins in  $\text{PdCrO}_2$  appear to be almost perfect Heisenberg spins, which places  $\text{PdCrO}_2$  in an interesting regime for study of the thermodynamics of magnetic ordering.

We conclude our discussion with some notes on the effect of strain on the Néel transition, which provides hints of interesting dynamics in the formation of magnetic order. As shown in figure 4(c), the first-order transition at  $T_N$  extends out to  $|\varepsilon_{\bar{1}10}| \sim 3 \times 10^{-3}$ . The transition might be driven first-order by fluctuations: fluctuations are expected to drive rotational-symmetry-breaking transitions first-order when the orientation is selected, as in  $\text{PdCrO}_2$ , from among three or more equivalent options [37, 38]. The strain range over which the transition is first-order does not correspond to fluctuations at  $T < T_N$ : the domain reorientation transition at  $\varepsilon = 0$  has been found to be sharp for temperatures below, and right up to,  $T_N$  [21]. It may therefore correspond to the strain required to polarise the fluctuations at  $T \gtrsim T_N$ . In the mean field, the linear dependence of  $T_N$  on uniaxial stress observed for  $|\varepsilon_{\bar{1}10}| \gtrsim 3 \times 10^{-3}$  is the expected form for



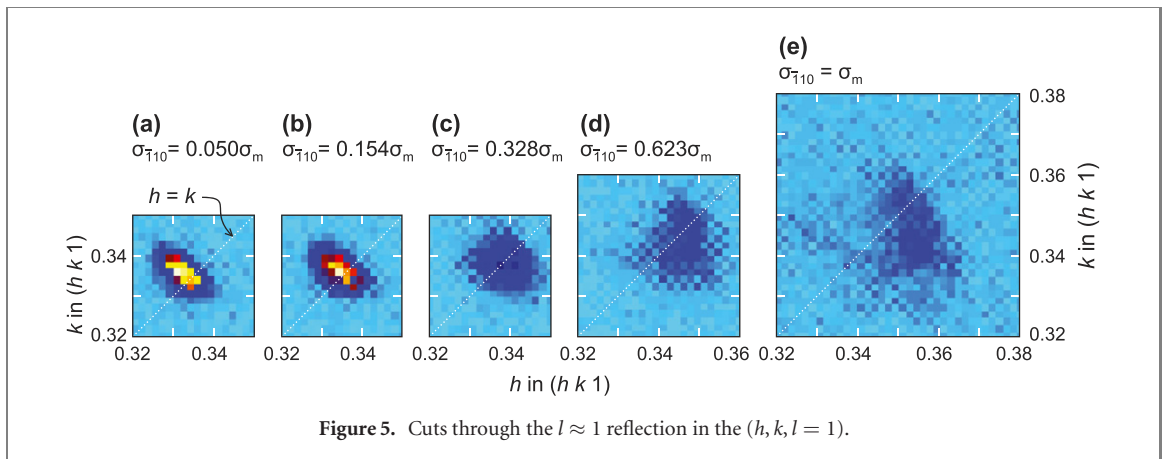


Figure 5. Cuts through the  $l \approx 1$  reflection in the  $(h, k, l = 1)$ .

transition into a uniaxial order.  $\text{PdCrO}_2$  may therefore provide an interesting, and very well-controlled, test case for study of the thermodynamic effect of fluctuations on ordering processes, which motivates further study of the transition at  $T_N$ .

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## Data availability statement

The datasets for the neutron experiment are available from the Science and Technology Facilities Council ISIS Neutron and Muon Source Journal [39], and the transport data presented in figures 3 and 4 are available at [40].

## Appendix A

### A.1. Cuts in the $h, k, l = 1$ plane









In figure 5, we show cuts through the  $l \approx 1$  reflection of sample 1 in the  $h, k$  plane. At the smallest applied stress, in panel (a), the reflection lies close to the commensurate position  $[\frac{1}{3}, \frac{1}{3}, l]$ . As stress is applied, it moves outward to larger  $h$  and  $k$ . It appears not to follow the  $h = k$  line precisely, though remains close enough that, within the resolution of these measurements, the cuts through the  $(h, h, l)$  plane in figures 2(a)–(e) give an accurate view of the evolution of the magnetic structure with stress.

### A.2. Error on the stress calibration for neutron measurements

In the stress cell employed for neutron scattering, a mechanical contact can be opened between the piezoelectric actuators and the sample in order to bring the force applied to the sample to nearly zero. However, the device incorporates a series of flexures, described in reference [29], and the spring constant of these flexures, in combination with the differential thermal contraction between the sample and titanium, from which the sample holder is fabricated, can still result in a residual stress on the sample. The total flexure spring constant is  $\approx 0.3 \text{ N}/\mu\text{m}$ . A thermal contraction of 0.75% between room and cryogenic temperatures has been measured for  $\text{PdCrO}_2$  [22], but this is extremely large;  $\text{CuCrO}_2$  contracts by 0.25% between 300 and 10 K [35]. Ti contracts by 0.15%, so we estimate a differential thermal contraction between  $\text{PdCrO}_2$  and Ti of  $\sim 0.1\%$ , and an effective sample length over which this differential operates of  $\sim 2 \text{ mm}$ . The force on the sample would then be of order  $2 \mu\text{m} \times 0.3 \text{ N}/\mu\text{m} \sim 0.6 \text{ N}$ , for a stress of  $\sim 3 \text{ MPa}$ .

The strain gauges on the sample holder were calibrated after the neutron scattering measurement as described in reference [29]. However, the response from the gauges was roughly half the expected value, based on finite element analysis of the deformation of the holder under applied stress. We therefore assign a relatively large error,  $\pm 20\%$ , to  $\sigma_m$ .

## ORCID iDs

Dan Sun  <https://orcid.org/0000-0001-6771-6300>  
Dmitry A Sokolov  <https://orcid.org/0000-0003-3282-7144>  
Seunghyun Khim  <https://orcid.org/0000-0002-9495-7406>  
Pascal Manuel  <https://orcid.org/0000-0002-8845-6576>  
Fabio Orlandi  <https://orcid.org/0000-0001-6333-521X>  
Dmitry D Khalyavin  <https://orcid.org/0000-0002-6724-7695>  
Andrew P Mackenzie  <https://orcid.org/0000-0001-8000-4949>  
Clifford W Hicks  <https://orcid.org/0000-0002-1736-1406>

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