SUPPLEMENTARY INFORMATION

Amphiphilic electrospun scaffolds of PLLA-PEO-PPO block copolymers: preparation, characterization and drug-release

behaviour

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PELA and PEPELA amphiphilic block copolymers were synthesized by ring-opening polymerization of LLA in a coordination-insertion mechanism from diol PEO and PEO-*b*-PEO polyether macroinitiators using $Sn(Oct)_2$ as catalyst. The equimolar $Sn(Oct)_2/hydroxyl$ groups ratio ($n_{catalyst}:n_{macroinitiator} = 2:1$) was used to produce high molar mass block copolymers and large yields (79-91%). ^{(1), (2), (3)}

Considering symmetrical PLLA block insertions during synthesis, ¹H NMR spectra (Figure S1) were used to determine the degree of polymerization of PLLA and hence the overall PELA and PEPELA copolymer molar masses (Table 1), calculated from the integration of specific signals at δ = 5.2 ppm (-*CH*(CH₃)COO-) for PLLA blocks and at δ = 3.6 ppm (-*CH*₂*CH*₂-O-) for PEO blocks, and applying the previously calculated molar masses from the esterified macroinitiators (Table 1). ⁽⁴⁾ The ratio between the number of LLA units and the total number of EO and PO repeating units in the copolymers

 (n_{LLA}/n_{EO+PO}) was also determined by ¹H NMR, and the values were similar to those obtained for the reaction medium (Table 1). ^(4–6)



Figure S1: ¹H NMR spectra of PELA (a) and PEPELA (b) copolymers. The characteristic peaks at approximately 5.2 ppm (1H, *q*) and 1.6 ppm (3H, *d*) are assigned to the methine (-CH) and methyl (-CH₃) protons of the LLA units, respectively; the peak at the 3.6 ppm (2H, *s*) is assigned to the methylene protons (-CH₂) of the EO units, and the peak at the 1.1 ppm (3H, *m*) is assigned to the methyl protons (-CH₃) of the PO units.

GPC analysis of PELA and PEPELA amphiphilic copolymers (Figure S2) provide numberaverage molar masses (M_n) relative to the PS standards used for calibration. Because of this, the values calculated by ¹H NMR are preferable. The polydispersity index (M_w/M_n) obtained from GPC data (Table 1) represent the molar mass distribution of polymers independent of the nature of the standards used for calibration.



Figure S2: Chromatogram from GPC analysis of the PELA (○) and PEPELA (■) copolymers directly after synthesis (before electrospinning processing).



Figure S3: XRD patterns of PELA and PEPELA scaffolds without drugs (a), loaded with 2.5 wt.% (b) and 5 wt.% (c) of acetaminophen, and loaded with 2.5 wt.% (d) and 5 wt.% (e) of celecoxib.

The addition of AC or CL drugs to PELA and PEPELA results in changes in the fibre diameter, and in number, size and shape of the beads of the scaffolds - Table 2 and Figure S4:



Figure S4: Mean fibre diameter (*D*) as a function of the initial mass fraction of the drug in the spinning solutions (x_{Di}) for (∞) PELA-AC_x, (**O**) PELA-CL_y, (**D**) PEPELA-AC_x and (∞) PEPELA-CL_y series.

Under DSC cooling, PLLA phase of the copolymers in the neat and loaded scaffolds partially crystallized, which is depicted by exothermic peaks at approximately $T_c = 90$ °C. The partially crystallized PLLA copolymer phase undergoes further cold crystallization at approximately 100°C only for PELA-AC₁₀, PEPELA-AC₅ and PEPELA-AC₁₀ scaffolds during the subsequent DSC 2nd heating scan. Also in the 2nd heating curves, PELA and PEPELA neat and loaded scaffolds present glass transition at approximately -15°C, referring to the polyether phase. In some cases, the polyether phase of the scaffolds was capable to crystallize, and a low intensity melting peak at approximately 35°C was observed for the scaffolds, with the exception of loaded PEPELA. Melting temperature at approximately 160°C and melting enthalpy of PLLA phase of all scaffolds did not vary significantly.

Besides the drug effect, the electrospinning process also influences the crystallization of the copolymer, since the polymer chains are elongated by the applied electric field along the solution jet direction and quenched upon rapid solvent evaporation, which restricts the chain segments mobility, folding and consequently reduces the degree of crystallization. ⁽⁷⁾

The molten liquids were held at high temperatures (200°C) ensuring the elimination of all crystals, and they did not crystallize upon the cooling step at 20°C min⁻¹, remaining as amorphous drugs. As the amorphous AC sample is heated again, it presents a glass transition at 30°C and then two exothermic peaks, referring to the AC amorphous-totype III transition ($T_{a-III,onset} = 80°C$) and type III-to-type II transition ($T_{III-II,onset} = 127°C$), followed by the melting at $T_{m, onset} = 159°C$ of the polymorphic mixture of AC. ^{(8), (9)} The amorphous CL presents in the 2nd heating a glass transition at 64°C, followed by an exothermic peak ($T_{cc,onset} = 133°C$) and an endothermic peak ($T_{m,onset} = 162°C$), referring to the cold crystallization and melting of CL. ^{(10), (11)}



Figure S5: DSC cooling and 2^{nd} heating curves at 20°C/min for PELA and PEPELA scaffolds: (a) without drug, (b) acetaminophen and acetaminophen-loaded scaffolds (c) AC_{2.5}, (d) AC₅, (e) AC₁₀; (f) celecoxib and celecoxib-loaded scaffolds (g) CL_{2.5}, (h) CL₅, (i)

CL_{10.}



Figure S6: Relative storage (E'_R) and loss (E''_R) moduli and loss factor $(\tan \delta)$ as a function of temperature for PEPELA scaffolds loaded with acetaminophen (a, b, c) or celecoxib (d, e, f) drugs at concentrations of 0.0 (\Box), 2.5 (\blacksquare), 5.0 (\circ) and 10 wt.% (\bullet) of initial drug load.

Table S1: Thermal properties of AC and CL drugs and PELA-AC_x, PELA-CL_y, PEPELA-AC_x

Material	DSC 1 st Heating								Со	oling	2 nd Heating							DMA	
	Tg ⁱ ∕°C	T _g ⁱⁱ ∕°C	T _m ⁱ ∕°C	∆H _m i ∕Jg ⁻¹	T _c ⁱⁱ ∕°C	∆H _c ⁱⁱ /Jg ⁻¹	T _m ⁱⁱ ∕°C	∆H _m ⁱⁱ /Jg ⁻¹	T _c ⁱⁱ ∕°C	ΔH _c ⁱⁱ /Jg ⁻¹	Tg ⁱ ∕℃	Tc ⁱⁱ ∕°C	ΔH _c ⁱⁱ /Jg ⁻¹	T _m i ∕°C	∆H _m i ∕Jg ⁻¹	T _m ⁱⁱ ∕°C	ΔH _m ⁱⁱ /Jg ⁻¹	<i>T</i> g ⁱ ∕°C	<i>T_g</i> [™] ∕°C
AC ⁱⁱⁱ	-	-	-	-	-	-	171	187	-	-	30	80 127	112 7.5	-	-	159	174	-	-
CL ⁱⁱⁱ	-	-	-	-	-	-	162	95	-	-	64	133	49	-	-	162	51	-	-
PELA	-	52	-	-	98	19	163	49	95	27	-10	-	-	39	35	163	47	-43	37
PELA-AC _{2.5}	-10	46	-	-	96	19	162	43	88	28	-11	-	-	39	5.5	161	46	-45	39
PELA-AC ₅	-13	42	54	8.7	93	15	162	46	94	-	-9	99	1.1	-	-	161	49	-47	50
PELA-AC ₁₀	-	-	56	14	94	15	162	45	-	0	-19	90	14	-	-	161	47	-43	51
PELA-CL _{2.5}	-11	50	68	4.2	97	16	164	43	88	22	-13	-	-	39	3.3	164	43	-13 broad	
PELA-CL ₅	-14	57	-	-	96	19	163	42	86	15	-11	-	-	-	-	163	45	-21 broad	
PELA-CL ₁₀	-14	45	62	5.9	96	16	161	45	90	-	-18	96	1.2	-	-	161	47	-24 broad	
PEPELA	-	60	-	-	98	19	164	47	88	34	-9	-	-	29	8.1	164	47	-28 broad	
PEPELA-AC _{2.5}	-	47	-	-	95	18	162	49	84	28	-13	-	-	-	-	161	49	-40	42
PEPELA-AC ₅	-10	35	53	7.3	95	16	161	47	79	10	-18	98	3.4	-	-	160	49	-41	45
PEPELA-AC ₁₀	-	53	-	-	96	13	161	46	-	-	-18	100	10	-	-	160	49	-44	47
PEPELA-CL _{2.5}	-	59	-	-	97	18	163	47	87	25	-18	-	-	-	-	163	46	-13 broad	
PEPELA-CL ₅	-10	59	-	-	95	16	163	47	86	32	-13	-	-	-	-	162	48	-27 broad	
PEPELA-CL ₁₀	-11	57	-	-	95	14	161	45	87	26	-14	-	-	-	-	161	47	-14 broad	

and PEPELA-CL_v scaffolds, as determined by DSC and DMA.

ⁱTransition referring to the polyether (PEO or PEO-*b*-PPO-*b*-PEO) phase in PELA and PEPELA scaffolds.

ⁱⁱTransition referring to the polyester PLLA phase in PELA and PEPELA scaffolds.

 ΔH^i and ΔH^{ii} were calculated, respectively related to polyether and PLLA phase mass.

ⁱⁱⁱT_{onset} for the chemically defined substances AC e CL.

AC spectrum exhibits the characteristic Raman shifts at 1650 (C=O stretching - $v_{C=O}$), 1620 (N-H bending - δ_{N-H}), 1560 ($v_{H-N-C=O}$), 1321 (δ_{C-H}), and the exclusive of AC type I 1236 (v_{C-N}) and 835 cm⁻¹ (δ_{C-H} out of the plane) ^{(9), (12)}, while the CL type III characteristic peaks are at 3234 and 1551 cm⁻¹ (respectively v_{N-H} and δ_{N-H} in sulfonamides), 1345 and 1161 cm⁻¹ ($v_{S=O}$), 1276 and 1228 cm⁻¹ (v_{C-F}). ^{(10), (11)} Unloaded PELA and PEPELA scaffold spectra present similar Raman shifts at 3002, 2946 and 2880 cm⁻¹ (v_{C-H}); 1765 cm⁻¹ ($v_{C=O}$), 1452 and 1295 cm⁻¹ (δ_{C-H}); 1130, 1090 and 1042 cm⁻¹ (v_{C-C} and v_{C-O}). ⁽¹³⁻¹⁶⁾



Figure S7: Full-range Raman spectra for acetaminophen (a) and celecoxib (b), and the unloaded PELA (c) and PEPELA (d) scaffolds. The green line depicts the copolymers' selected peak in the Raman shift range of 1765-1766 cm⁻¹. The red line corresponds to selected range of 1619-1620 cm⁻¹ for AC and of 1618-1619 cm⁻¹ for CL.

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