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Influence of AlN passivation on thermal performance of AlGaN/GaN High-Electron Mobility Transistors on Sapphire substrate: A simulation study

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Abstract:

This work describes the self-heating effects on the behaviour of AlGaN/GaN-based high-electron Mobility Transistors (HEMTs), **which are grown on Sapphire substrate**, using electro-thermal TCAD simulations. The proposed device, passivated with AlN/SiN, demonstrates more excellent thermal performance than the conventional one with SiN passivation due to the introduction of additional AlN on top of the device, which acts as a heat spreader. The electro-thermal simulations have carried out for different AlN thicknesses (0 μm to 25 μm), and the device with 5 μm AlN shows better performance compared to others. **The proposed AlN/SiN stacked passivation HEMT shows a comparatively small lattice temperature of 418 K, whereas the conventional HEMT with SiN passivation shows 578 K.** All the devices (gate length, $L_G = 1 \mu\text{m}$) switch from OFF- to ON-states using the voltage, V_{GS} from -10 V to 0 V with fixed bias, $V_{DS} = 5 \text{ V}$. **The values of saturation drain current density (I_{DSS}) and transconductance (g_m) are 0.7 A/mm and 173 mS/mm for the proposed HEMT with 5 μm AlN considering the thermal simulation model. In contrast, the conventional device demonstrates those of 0.42 A/mm and 109 mS/mm, respectively.** The $\sim 0.32 \text{ A/mm}$ of drain current recover for the proposed device with 5 μm AlN from a conventional device because of the reduction of self-heating effects. **Our study reveals that the AlN/SiN passivation HEMTs are a promising technology for high-power switching and microwave applications without significant reduction in device performance at high drain bias.**

Keywords:

Self-heating; lattice temperature; AlN; thermal management; sapphire; DC characteristics; HEMT;

1. Introduction:

Nowadays, GaN-based HEMTs have attracted enormous attention for high-power and high-frequency applications [1-4]. Due to their high breakdown field and wider bandgap, GaN-based heterostructure devices have been proved to have high-power handling capability to withstand excessive temperatures and corrosive ambient [5]. These devices have gained colossal research focus because of their exemplary performance in terms of very high saturation velocities of the charge carriers and enormous charge densities achieved even without intentional doping, the breakdown voltage of few 100's of volts, and operational frequencies ranging from few GHz to THz [6-9]. The GaN HEMTs have already achieved a commercial breakthrough in optical, RF device, and power markets. However, the self-heating effects (SHEs), increase in crystal temperature, by the elevated lattice temperature at high-power operation limits the device performance by the degradation of electron mobility, breakdown voltage, and radio frequency (RF) characteristics [10-14]. The SHEs have hindered the manufacturing of high-quality HEMTs that become dominant when drain bias is applied, leading to sub-optimal breakdowns. Extensive researches have been conducted to overcome the SHEs by various research groups using advanced field plate structures, excessive thermal stability material, and air-water cooling device framework [15-19]. However, even after lots of research, an optimized device has not been proposed to meet the market power-requirements. Almost all of the HEMT designs are found to have the limitation of a suboptimal breakdown even when the GaN material has a very high breakdown field of 3 MV/cm [20-22]. Usually, GaN has grown on the substrate, such as Sapphire, Si, and SiC, for high-power applications. Though the SiC has high thermal conductivity (TC) (330 W/m-K @ room temperature, RT), the costs and lack of availability in large sizes and ~13% mismatch in the lattice parameter for GaN on SiC [23] are the bottlenecks of SiC substrate. On the other hand, Si material is available in larger wafer size at low costs, and its more excellent TC (150 W/m-K at RT) makes it one of the choices. However, due to the non-mature process technologies for GaN-on-Si combined with a high-lattice mismatch (~16%) and significant difference in thermal expansion coefficients (where the values of $2.6 \times 10^{-6} \text{ K}^{-1}$ and $5.59 \times 10^{-6} \text{ K}^{-1}$ for Si and GaN, respectively), Si substrate reduces the reliability of the devices. Besides, the AlN nucleation layer or multiple layers of AlN and AlGaIn with varying concentrations of Al forming a transition scheme are necessary to reduce the lattice mismatch for GaN grown on Si [24,25]. Despite low-TC (42 W/m-K at RT), many research-groups have recently demonstrated the outstanding performance of the GaN-material based devices with the sapphire substrate [23-32]. Therefore, sapphire has become an excellent choice as a substrate due to its high crystalline quality, lattice constant matching with GaN, low cost of wafers, and availability in larger wafer sizes.

Theoretical and experimental results have strongly confirmed that reduction of SHEs on AlGaIn/GaN-based devices are a critical issue in various regards. Varghese et al. demonstrated that the AlGaIn/GaN HEMTs exhibited outstanding device performance due to the introduction of triple AlN inter-layers, those employed as a spacer layer, nucleation layer, and interfacial passivation [25]. Huang et al. claimed that the decrease of dynamic ON-resistance and current-collapse suppression seen in AlGaIn/GaN HEMTs with the AlN passivation [33]. Several research groups reported various techniques for thermal optimization for GaN-based devices and circuits. Dasom Jeon et al. demonstrated the reduction of self-heating in nano-interfaces by using h-BN based 2D interface [34]. GaN on Si substrate HEMTs based RF amplifier thermal dissipation improved by plastic package [35]. Etched AlGaIn barrier HEMTs reduces the channel temperature by generating a new hot-spot in the thermal field [36]. Neama et.al. demonstrated a thermal management technique for GaN-HEMTs using mini-channel heat sinks [37]. A copper filled Micro-Trench structure used for thermal management of GaN on Si HEMT [38]. Different passivation materials had been employed to reduce the SHEs of GaN HEMTs [39-41]. Chander et al. studied the SHEs of AlGaIn/GaN HEMT with field plate structure on SiC substrate for different passivation materials [39]. Haghshenas et al. reported that the passivation thickness and material had a significant role in the hot spot temperature of AlGaIn/GaN HEMTs with a single passivation layer [40]. Our previous works reported the effective suppressing of SHEs on AlGaIn/GaN-based HEMT with the filed-plated structure on SiC substrate using stacked passivation [41]. Still, inadequate data on the SHEs on the device performance are available, and studies on AlGaIn/GaN HEMTs grown on sapphire substrate with stacked passivation are lacking. However, such results are vital to design the next-generation high-power and high-frequency devices. Therefore, a more detailed understanding and the proper inclusions of the SHEs on device performance are immensely crucial.

In this work, we proposed a cost effective AlN/SiN stacked passivation for thermal optimization of GaN-HEMTs grown on sapphire substrate. We have explored the electro-thermal analysis of AlGaIn/GaN HEMTs using TCAD simulations on the SILVACO ALTAS platform [42] for with and without top AlN layer. TCAD simulations carry out with and without a self-heating model. Both lateral- and vertical-lattice temperatures of devices are also studied. The proposed device shows promising results compared to a conventional one due to better heat dissipation, which is achieved by introducing AlN on top. The AlN layer in the passivation scheme reduces interface traps at passivation/AlGaIn and plays a vital role in enhancing the drain current by lowering the SHEs. All the simulations have been done at room temperature.

2. Materials, Methods, and Models

The illustrations of GaN/AlGaN HEMTs without (conventional) and (proposed) top AlN are demonstrated in Fig. 1(a) and (b), respectively. Fig. 1 (c) demonstrates the expected device fabrication process steps. The composition, $x = 0.25$ is considered for the barrier layer, $\text{Al}_x\text{Ga}_{1-x}\text{N}$. The geometric and material parameters are provided in Table I and Table II, respectively. The conventional device defines as an GaN/AlGaN HEMT with SiN passivation, whereas the GaN/AlGaN HEMT with dual passivation, AlN/SiN, is the proposed device. The conventional device has been simulated without and with the self-heating model, which define as CM (conventional model simulation) and SHM (self-heating model simulation), respectively. The simulations have been done with the proposed self-heating model, which is mentioned as SHM with AlN.

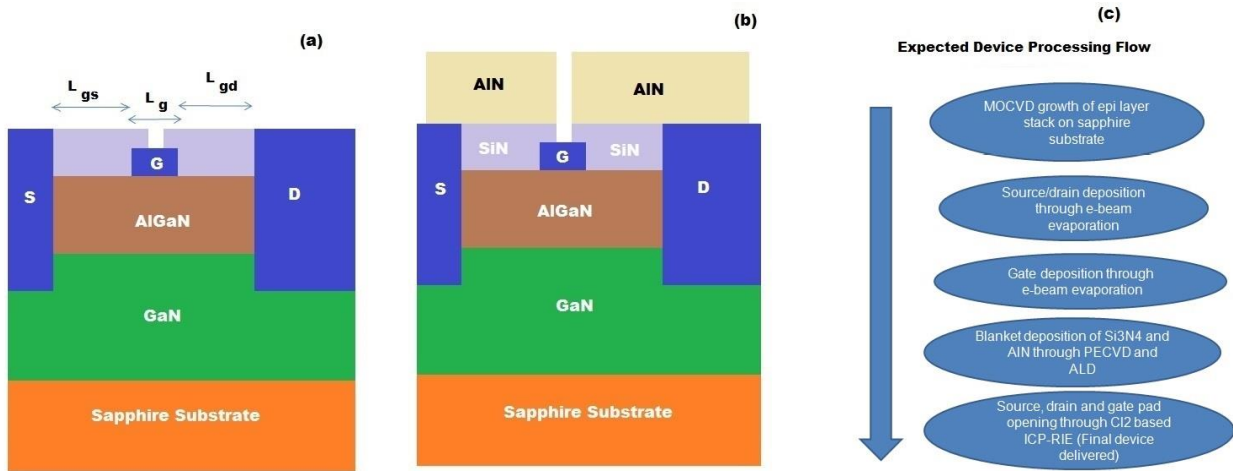


Figure 1(a). Illustrations of a conventional GaN/AlGaN-HEMT. Figure 1(b). Illustrations of the proposed device design with AlN layer on top. Figure 1(c). Process flow to make feasible the proposed device fabrication.

Table I. Geometric parameters

Parameter	Value
AlGaN barrier layer thickness	26.7 nm
GaN channel layer thickness	47.3 nm
GaN buffer layer thickness	1.0 μm
SiN passivation thickness	5.5 μm
Top AlN	0 ~ 25 μm
L_G (length of gate)	1.0 μm
L_{sg} (distance between source & gate)	2.0 μm
L_{gd} (distance between gate & drain)	2.0 μm
Gate Width (W)	100 μm

A two-dimensional (2D) TCAD simulation is carried out using lattice temperature model to investigate the self-heating-effects (SHEs) of GaN/AlGaN HEMT. Additionally, we consider the essential physical models such as interface trap models, surface trap state model, nitride low & high field mobility models, and SRH (Shockley–Read–Hall) recombination models. The thermal boundary resistances (TBRs) are considered for each material in the simulation [42-50].

Table II. Material parameters

Parameter	GaN	AlGaN
E_g (Semiconductor band gap in eV)	3.4	3.96
μ (Low field electron mobility in $\text{cm}^2/\text{V.s}$)	1460	300
ϵ_r (Relative permittivity)	9.5	9.5
Saturation velocity of electron in cm/S	2×10^7	1.12×10^7
Electron affinity (eV)	4	3.82
Saturation velocity of hole in cm/S	1.9×10^7	1.0×10^6

Conduction band DOS ($\times 10^{18}$) in cm^{-3}	1.07	2.07
SRH lifetime	1.0×10^7	1.0×10^7
Valence band DOS ($\times 10^{19}$) in cm^{-3}	1.16	1.16

DOS: Density of states; SRH: Shockley Read Hall Recombination

DD (Drift-Diffusion) model of charge transport, Fermi-Dirac, Boltzman carrier statistics & interface traps models are used in this work, which is adequate for all micrometer range devices that are technologically feasible. The more conventional form of drift-diffusion equations are as follows [42]:

$$J_n = qn\mu_n E_n + qD_n \nabla_n \quad (1)$$

$$J_p = qp\mu_p E_p - qD_p \nabla_p \quad (2)$$

Where μ_n and μ_p are electron and hole mobility, respectively, and E stands for the effective electric field. D_n and D_p are written as [42]:

$$D_n = \frac{kT_L}{q} \mu_n \quad (3)$$

$$D_p = \frac{kT_L}{q} \mu_p \quad (4)$$

Where T_L & k represents lattice temperature & Boltzman constant respectively. The self-heating model includes the primary lattice heat flow model, heat generation and recombination models, and thermal conductivity model [42]:

Lattice heat flow [42]:

$$C_{th} \frac{\partial T_L}{\partial t} = \nabla(k \nabla T_L) + H \quad (5)$$

Where C_{th} & k respectively represents the thermal capacitance & TC (thermal conductivity). In general, the electro-thermal effects are modeled by thermal capacitance $C_{th} = \frac{\Delta Q \text{ (rate of change of heat flow)}}{\Delta T \text{ (rate of change of temperature)}}$ and thermal resistance R_{th} parallel network consists of thermal current source (I_{diss}). High thermal conductivity of the semiconductors transports the generated heat energy faster than lower TC materials and R_{th} is the TC dependent parameter.

Heat generation and recombination [42]:

$$H = \left(\frac{|\vec{j}_n|^2}{q\mu_n n} + \frac{|\vec{j}_p|^2}{q\mu_p p} \right) + q(R - G)[\varphi_p - \varphi_n + T_L(P_p - P_n)] \quad (6)$$

Thermal conductivity [42]:

$$k(T_L) = (TC.CONST)/(T_L/300)^{TC.NPOW} \quad (7)$$

The Material thermal conductivity constant is $TC.NPOW$ and the thermal resistance $\left(R_{th} = \frac{\Delta T(\text{rise in the channel temperature})}{P_{diss}} \right)$ links the channel temperature to power dissipation. The thermal constants are listed in Table III.

Temperature dependent nitride low field mobility model [39]:

$$\mu_0(T, N) = \mu_{min} \left(\frac{T}{300} \right)^{\beta_1} + \frac{(\mu_{max} - \mu_{min}) \left(\frac{T}{300} \right)^{\beta_2}}{1 + \left[\frac{N}{N_{ref} \left(\frac{T}{300} \right)^{\beta_3}} \right]^{\alpha (T/300)^{\beta_4}}} \quad (8)$$

Where N & T indicates the total doping density & temperature respectively and $\alpha, \beta_1, \beta_2, \beta_3, \beta_4, \mu_{min}, \mu_{max}$, and N_{ref} are fitting parameters [42].

Table III. Thermal conductivity constant [42]

Parameter	Unit	AlGaN	GaN	AlN	Sapphire
Thermal conductivity constant ($TC.CONST$)	W/cm-k	0.4	1.3	1.8	0.3

Nitride high field mobility (temperature dependent) model [42]:

$$\mu = \frac{\mu_0(T,N) + v^{sat} \frac{E^{n1-1}}{E_c^{n1}}}{1 + a \left(\frac{E}{E_c}\right)^{n2} + \left(\frac{E}{E_c}\right)^{n1}} \quad (9)$$

Where, μ_0 & E indicates the low field mobility & electric field respectively, v^{sat} , E_c , $n1$, $n2$ and a are fitting parameters of Faramand modifier Thomas Nitride model [42].

SRH (Shockley-Read Hall) Model [42]:

$$R_{SRH} = \frac{pn - n_{ie}^2}{TAUP0 \left[n + n_{ie} \exp\left(\frac{ETRAP}{kT_L}\right) \right] + TAUN0 \left[p + n_{ie} \exp\left(\frac{ETRAP}{kT_L}\right) \right]} \quad (10)$$

Where, $ETRAP$ is the difference in energy levels between trap state and Fermi level, and the electron & hole lifetimes are denoted by $TAUN0$ and $TAUP0$ respectively. The electron & hole carrier concentrations (cm^{-3}) are defined as n and p , respectively. The n_{ie} is an effective intrinsic concentration. T_L and k are defined as lattice temperature and Boltzmann constant, respectively [42].

3. Results and discussions

The DC output characteristics of conventional HEMT has been done without incorporating the thermal models for gate bias (V_{GS}), swept from 0 V to -5 V and drain-bias (V_{DS}), swept from 0 V to 15 V (Fig.2 (a)). The peak I_{DS} is ~ 1.2 A/mm **at the bias voltage of** $V_{GS} = 0$ V and $V_{DS} = 15$ V. The extracted 2-DEG (2-dimensional electron gas) and mobility (μ) of the channel are $1.14 \times 10^{13} \text{ cm}^{-2}$ and $1260 \text{ cm}^2/\text{V-s}$, respectively. Figures 2 (b) and (c) demonstrate the device characteristics of conventional and proposed HEMTs with self-heating model, respectively. The value of drain current dispersion is higher for the conventional devices as compared with the proposed one. This result can be reasoned by the SHE (self-heating effect) that becomes dominant in the conventional device at higher drain biases, leading to noticeable degradation in the current density. The conventional device shows a peak current of 0.42 A/mm at $V_{GS} = 0$ V. As a result of self-heating effects, an increase in lattice temperature by an applied electric field and thus thermal scattering and degradation in the electron mobility occurs. At high drain voltage, the effect of scattering of the carriers increased, and therefore in the region of saturation, there will be a negative slope of drain current. Figure 2 (c) represents the proposed device's output characteristics with SHM for 5 μm AlN thickness. Inclusion of AlN with high TC of 1.8 W/cm-K as a heat spreader in the conventional HEMT reduces the SHEs in the device channel's access

region, thereby improving the normalized drain current density to 0.7 A/mm at $V_{GS} = 0$ V. Besides, this also reduces the electron traps in the device.

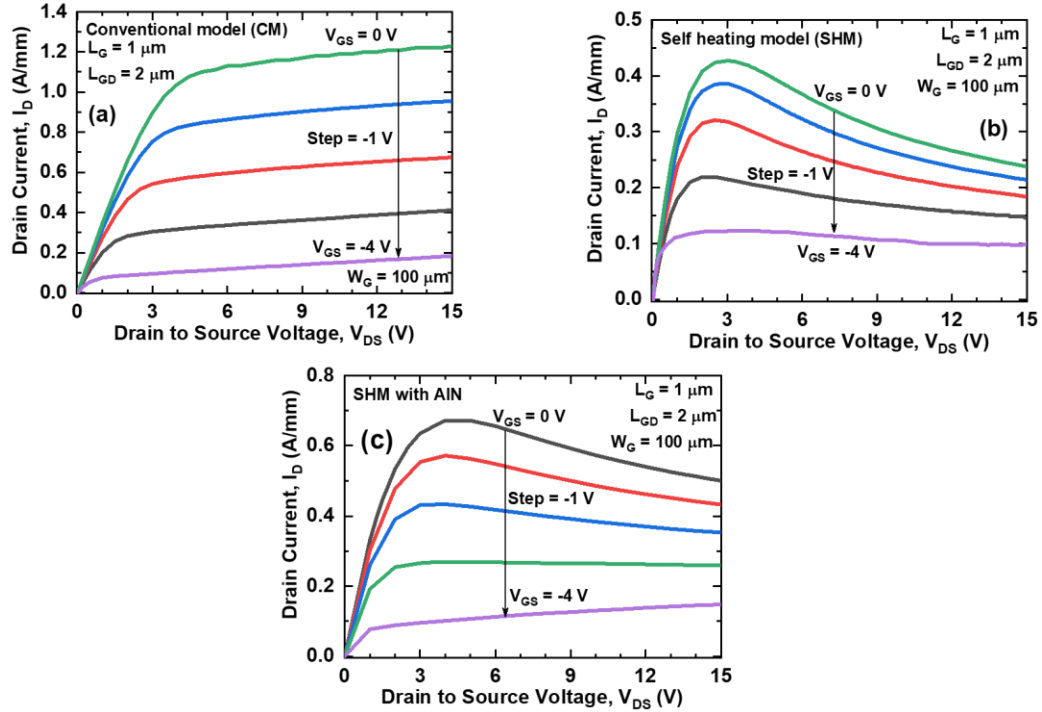


Figure 2. DC characteristics of conventional and proposed HEMTs with $L_G = 1 \mu\text{m}$. The characteristics are shown as (a) conventional device without self-heating model (SHM), (b) conventional device with self-heating model (SHM), and (c) proposed device with self-heating model (SHM) for 5 μm AlN thickness.

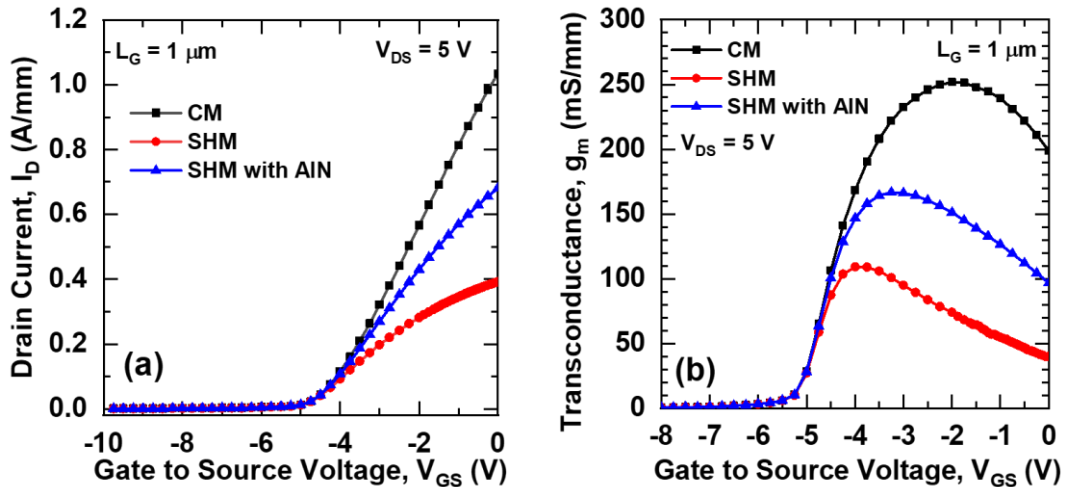


Figure 3. Transfer (a) and transconductance (b) characteristics of a conventional device without self-heating model (SHM), a conventional device with self-heating model (SHM,) and the proposed device with self-heating model (SHM) for 5 μm AlN thickness.

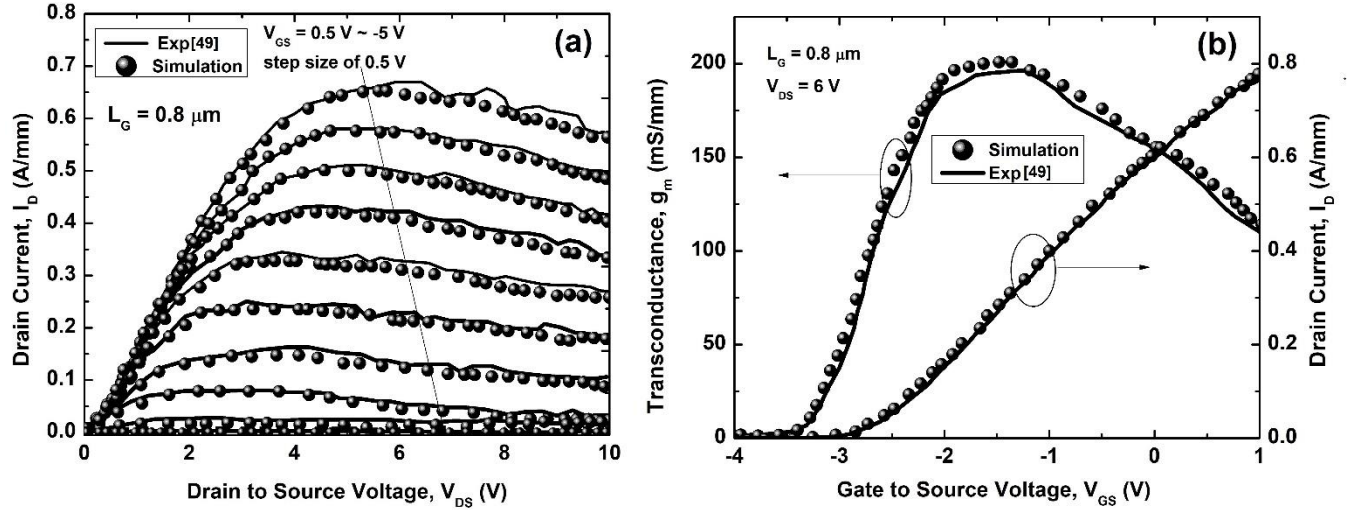


Figure 4(a). DC characteristics of conventional AlGaIn/GaN HEMT on sapphire substrate with a $L_G = 0.8 \mu\text{m}$ [49]. **Figure 4(b).** Transfer characteristics.

Fig.3 (a) demonstrates the comparison of transfer curves of a conventional HEMT, a conventional HEMT with self-heating model simulation (SHM), and the proposed device with self-heating model (SHM) for V_{GS} swept from -10 V to 0 V and $V_{DS} = 5 \text{ V}$. Fig.3 (b) shows the transconductance characteristics of the conventional device, the conventional device with self-heating model simulation, and proposed device with self-heating model simulation for $5 \mu\text{m}$ AlN thickness for V_{GS} swept from -10 V to 0 V at fixed drain bias, $V_{DS} = 5 \text{ V}$. It is found that there is a significant drop in the both I_{DS} and g_m for the devices with self-heating model simulation. The off-state gate leakage currents are $3.3 \times 10^{-16} \text{ A/mm}$, $1.4 \times 10^{-9} \text{ A/mm}$, and $2.1 \times 10^{-12} \text{ A/mm}$ for a conventional device without self-heating model simulation, a conventional device with self-heating model simulation, and the proposed device with self-heating model simulation, respectively. The peak values of transconductance (g_m) are 273 and 109 mS/mm for a conventional device with and without self-heating model simulation. In comparison, it shows 173 mS/mm for the proposed device with the top $5 \mu\text{m}$ AlN for self-heating model simulation. The threshold voltages are almost the same as -5 V for all the devices.

The increase of crystal temperature causes thermal scattering, which diminished the electrons' mobility in the channel. Further, at high drain voltage, the SHE increased, reducing the I_{DS} and g_m . The overall output current density and the maximum transconductance improved in the proposed device with AlN passivation thermal structures. TCAD simulation is carried out for $L_G = 0.8 \mu\text{m}$ GaN/AlGaIn HEMT on the sapphire substrate reported in [49], and its DC, and transfer curves are displayed in Fig.4 (a) and Fig 4(b), respectively. The results show a perfect match with experimental results.

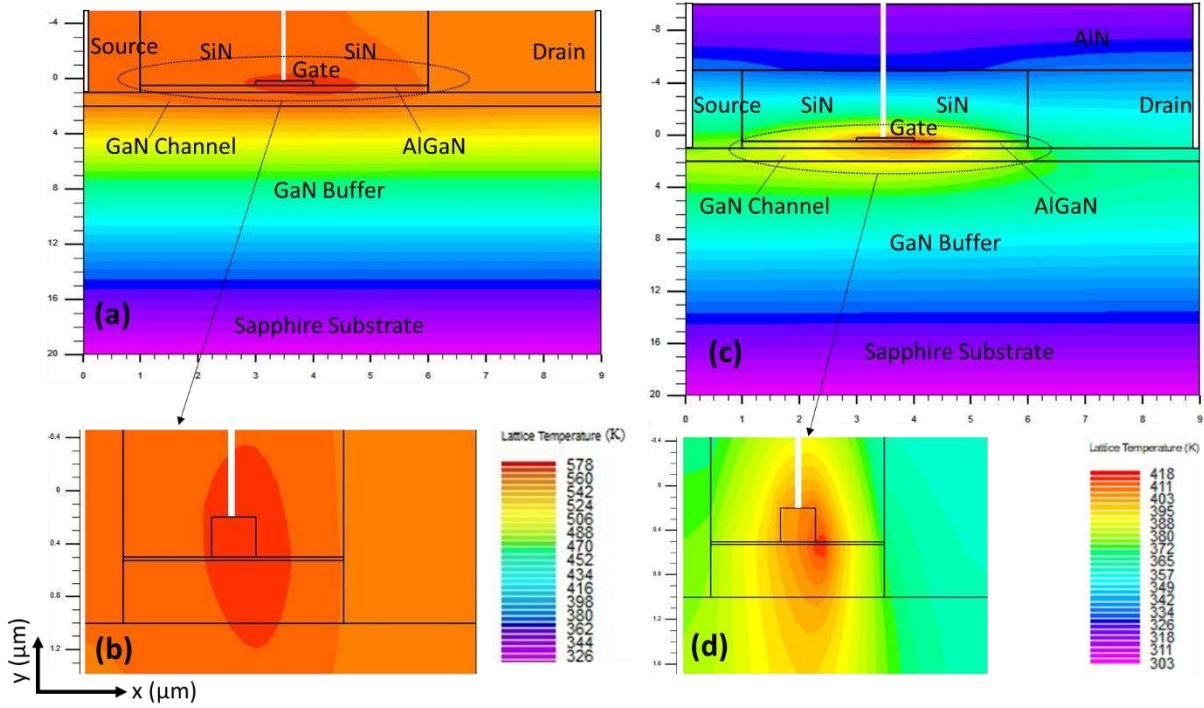


Figure 5(a). Temperature gradient profile of conventional device with self-heating model simulation for drain bias (V_{DS}) 15 V. Figure 5(b). Channel temperature gradient profile of conventional device with self-heating model simulation. Figure 5(c). Temperature gradient profile of proposed device with self-heating model simulation for 5 μm AlN top layer. Figure 5(d). Channel temperature gradient profile of proposed devices with self-heating model simulation for 5 μm AlN top layer.

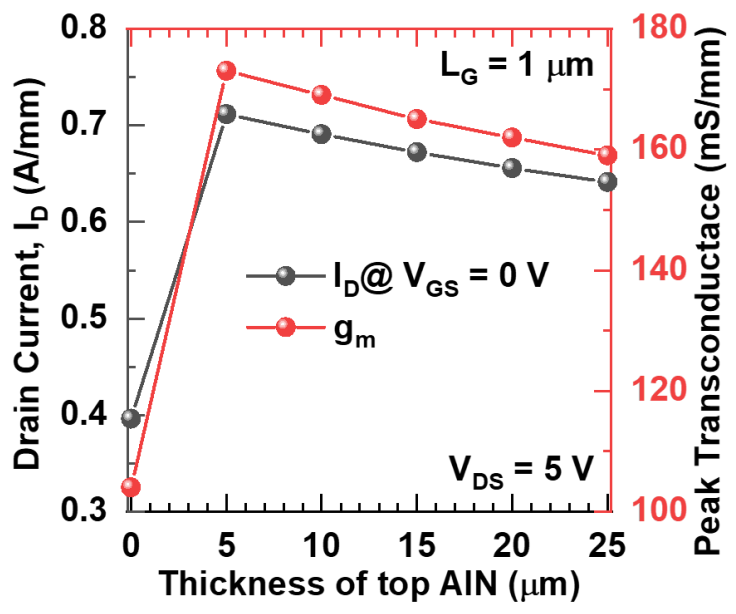


Figure 6. Effect of top AlN thickness on drain current and peak transconductance

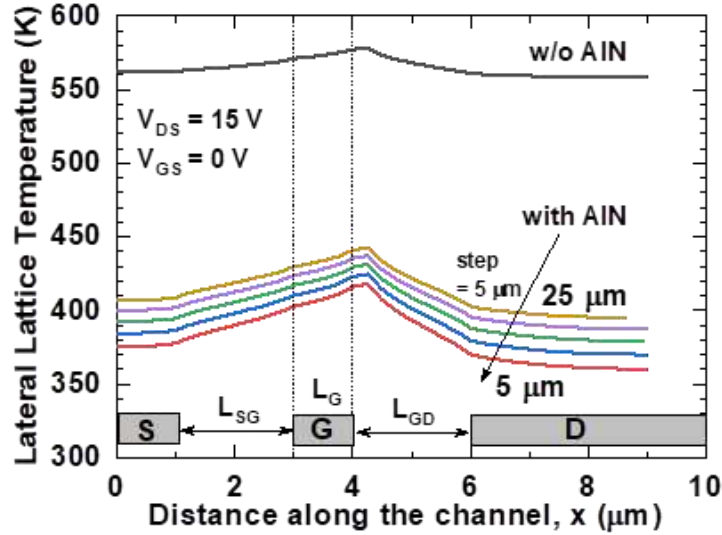


Figure 7. Lattice temperature variation along the channel, x of the conventional device, and the proposed device with self-heating model simulation for different AlN thickness.

The temperature gradient profile of a conventional device with self-heating model simulation at $V_{GS} = 0$ V and $V_{DS} = 15$ V is depicted in Fig. 5 (a). A maximum lattice temperature of 578 K is observed in the device (Fig.5 (b)). The proposed device's temperature gradient profile with SHM for 5 μm AlN is demonstrated in Fig.5(c). The peak-temperature of 418 K observed in the device, as shown in Fig. 5(d). TCAD simulation of the device's temperature gradient profile with 20 μm Sapphire substrate and 5 μm AlN thicknesses is presented in Fig.5. However, TCAD simulation is performed for various substrate thicknesses (20 μm to 300 μm) and confirmed that the device characteristics and channel temperature profile are the same for various substrate thicknesses for a particular AlN top layer thickness. The top AlN thickness-dependent drain current and transconductance are depicted in Fig. 6. The I_{DS} is enhanced due to the introduction of AlN on top. However, it starts to reduce above 5 μm by increasing AlN's thickness, and similar effects are observed in transconductance. The proposed device shown high drain current and transconductance for 5 μm AlN thickness results from reduced lattice temperature shown in Fig. 7.

Comparative lattice temperature variation along the channel of a conventional device with self-heating model simulation and proposed device with self-heating model simulation for different thicknesses of top AlN is shown in Fig 7. The overall vertical lattice temperature distribution of region near the gate end of the drain side of conventional and proposed devices is shown in Fig. 8(a) and Fig. 8(b), respectively. Figure 8 (c) shows the effects of the top AlN thickness on vertical lattice temperature. The

proposed device near the gate electrode showed the low lattice temperature. The additional AlN passivation effectively decreases the temperature in 2DEG. As a result, the proposed thermal structure improved the overall DC characteristics.

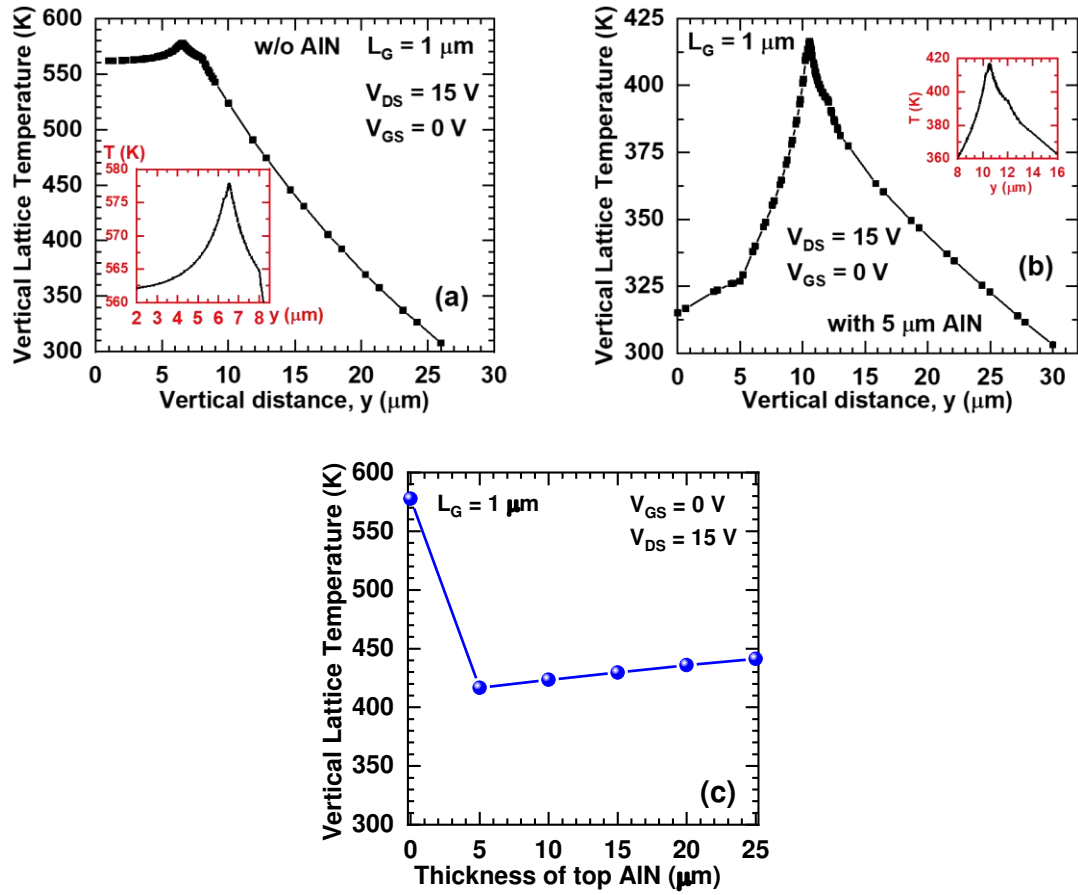


Figure 8. Vertical lattice temperature gradient of the conventional (a) and the proposed (b) devices with self-heating model simulation. Inset figures show the enlarged view of the region near the gate end of drain side. Figure 8 (c) effects of the top AlN thickness on vertical lattice temperature.

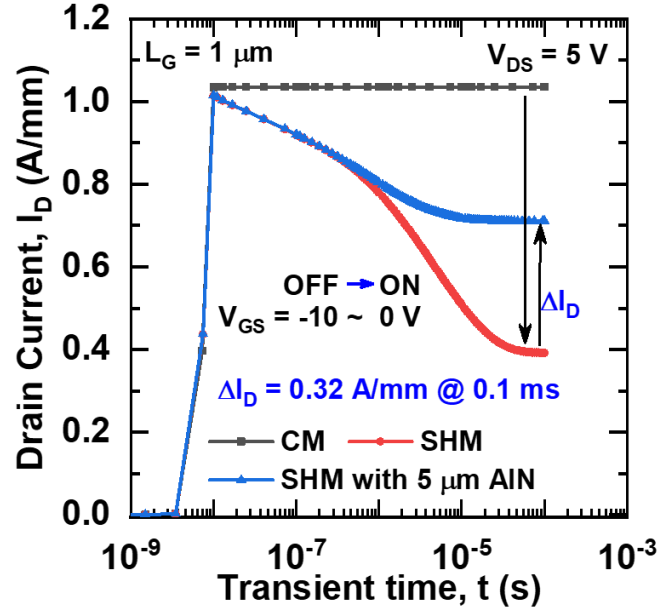


Figure 9. Transient time drain current responses of a conventional device with and without self-heating model simulation, and the proposed device with self-heating model simulation for 5 μm AlN top layer.

Fig. 9 shows the transient characteristics of the current drain, I_D for AlGaIn/GaN HEMTs with different models. The devices switch from OFF-state ($V_{GS} = -10\text{ V}$) to ON-state ($V_{GS} = 0\text{ V}$) with a fixed $V_{DS} = 5\text{ V}$. The devices show different times when the I_{DS} reached a steady state at the same electric bias. The conventional AlGaIn/GaN HEMT transient current drops significantly using self-heating model simulation due to temperature rise in the channel when it reaches steady-state conditions. Wang et al. [49], Zhang et al. [50] and Yachao Zhang et al. [51] reported that the current drop in the GaN/AlGaIn HEMTs occurred for the reduction of electron mobility in the channel. One of the key factors of this reduction is self-heating in the devices. At the very beginning of switching, the devices show almost the same drain current using the thermal models and start to change with the time due to reduction of the channel mobility, caused by self-heating. However, the thermal optimized HEMT using a high thermal conductivity AlN passivation (5 μm) has shown improved I_D transient response of $\Delta I_D = 0.32\text{ A/mm}$ at 0.1 ms. The above results indicate that the additional AlN deposited on SiN passivation improves the performance device due to the reduction of SHEs.

4. Conclusion

The studies of self-heating effects on AlGaIn/GaN HEMT DC characteristics with a dual SiN/AlN passivation have been presented. The proposed thermally optimized HEMT device design incorporating an AlN passivation layer significantly improved the device's electrical characteristics over the conventional ones. The overall device temperature in crystal lattice within the 2DEG channel and temperature in the vertical lattice is found to get reduced due to the temperature handling capability of AlN. An investigation of the impact of AlN thickness on device performance has been presented. The drain current, transconductance, and proposed device are also improved compared to the conventional one. The proposed dual SiN/AlN passivation HEMT ($L_G = 1\mu\text{m}$) TCAD simulation with self-heating model shown 60% improvement in drain current density and 63% improvement in transconductance. The device's overall performance is enhanced with the incorporation of 5 μm AlN making the device reliable and stable for a wider range of operation, which is desirable in the high-power domain.

Data availability

The raw/processed data required to reproduce these findings cannot be shared at this time as the data also forms part of an ongoing study.

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